


 Cite this: *RSC Adv.*, 2020, 10, 38033

## Photo-induced self-catalysis of nano-Bi<sub>2</sub>MoO<sub>6</sub> for solar energy harvesting and charge storage†

 Jiangju Si,<sup>ab</sup> Changmeng Guo,<sup>a</sup> Haojie Liu,<sup>a</sup> Weiwei Li,<sup>ab</sup> Xiaowei Guo,<sup>ab</sup> Peidong Bai,<sup>a</sup> Yanghong Liu,<sup>a</sup> Gairong Chen<sup>\*ab</sup> and Ningbo Sun<sup>\*ab</sup>

Efficient, sustainable, and integrated energy systems require the development of novel multifunctional materials to simultaneously achieve solar energy harvesting and charge storage. Bi-based oxysalt aurivillius phase materials are potential candidates due to their typical photovoltaic effect and their pseudo-capacitance charge storage behavior. Herein, we synthesized nano-Bi<sub>2</sub>MoO<sub>6</sub> as a material for both solar energy harvesting and charge storage due to its suitable band gap for absorption of visible light and its well-defined faradaic redox reaction from Bi metal to Bi<sup>3+</sup>. The irradiation of visible light significantly affected the electrochemical processes and the dynamics of the Bi<sub>2</sub>MoO<sub>6</sub> electrode. The photo-induced self-catalytic redox mechanism was carefully explored by adding sacrificial agents in photocatalysis reaction. In accordance with the rule of energy matching, the photo-generated holes oxidized the Bi metal to Bi<sup>3+</sup>, and the corresponding peak current increased by 79.5% at a scanning rate of 50 mV s<sup>-1</sup>. More importantly, the peak current retention rate remained higher than 92.5% during the entire 200 cycles. The photo-generated electrons facilitated a decrease of 184 mV in the overpotential of the reduction process. Furthermore, the irradiation of visible light also accelerated the ionic diffusion of the electrolyte. These investigations provide a unique perspective for the design and development of new multifunctional materials to synergistically realize solar energy harvesting and charge storage.

 Received 15th August 2020  
 Accepted 8th October 2020

DOI: 10.1039/d0ra07020c

[rsc.li/rsc-advances](http://rsc.li/rsc-advances)

Highly efficient and sustainable green energy conversion and storage are available solutions to the intensifying energy and environmental conservation issues.<sup>1–17</sup> New integrated energy systems with simplified processes, that realize the complementarity between multiple clean energy technologies, are attracting increasing amounts of attention.<sup>1,4–6,9,12–14,17</sup> Among the numerous energy conversion and storage technologies available, solar energy is one of the most promising as an abundant, low-cost, eco-friendly, and renewable energy source. Electrochemical rechargeable devices can achieve highly efficient energy conversion and storage. Therefore, it is necessary to develop an integrated system that synchronously achieves solar energy harvesting and charge storage within one device.<sup>1,11,18</sup> Some photo-rechargeable electrochemical energy storage devices, such as photo-rechargeable batteries<sup>7,8,12,13,16,19–22</sup> and photo-rechargeable capacitors,<sup>2–6,9–11,14,15,18,23–27</sup> have been emerged. These new devices usually contain an independent solar energy harvesting unit (such as photovoltaic solar cells that include a dye-sensitized

solar cell,<sup>4</sup> organometal halide perovskite cells,<sup>23</sup> polymer solar cell,<sup>9</sup> and Si solar cells<sup>13</sup>), as well as separate rechargeable energy storage units (such as lithium-ion batteries,<sup>5,7,12,13,21</sup> lithium oxygen batteries,<sup>22</sup> zinc-ion batteries<sup>8</sup> and flow batteries,<sup>19</sup> supercapacitors,<sup>6,14,23</sup> and hybrid capacitors<sup>25</sup>), which are expensive and lead to the energy-level mismatch and ohmic transport losses at the interface between the two units.<sup>3,17</sup> Thus, it is highly desirable to explore new materials that are capable of harvesting and storing energy simultaneously.<sup>8,25</sup> Bi-based oxysalt aurivillius phase materials are the potential candidates due to their typical photovoltaic effect<sup>28</sup> and electrochemical energy storage behavior.<sup>29–40</sup> However, there are few reports on the integrated photo-rechargeable devices that use Bi-based oxysalt,<sup>41</sup> and the synergy mechanism between the photovoltaic effect and the electrochemical process has not yet been revealed. Here, we selected nano-Bi<sub>2</sub>MoO<sub>6</sub> as the material for both solar energy harvesting and charge storage because it has a suitable band gap for the absorption in visible light,<sup>28</sup> a layered structure that favors a high electron transfer rate, and a well-defined faradaic redox from Bi metal to Bi(III).<sup>36</sup> The irradiation of visible light significantly influenced the electrochemical processes and the dynamics of the Bi<sub>2</sub>MoO<sub>6</sub> electrode, and the photo-induced self-catalytic redox mechanism was carefully explored.

Nano Bi<sub>2</sub>MoO<sub>6</sub> was prepared by a simple hydrothermal reaction, and the specific method was shown in the ESI.† The X-

<sup>a</sup>School of Chemistry and Materials Engineering, Xinxiang University, Xinxiang, Henan 453003, China. E-mail: sunningbo682@163.com

<sup>b</sup>Henan Photoelectrocatalytic Material and Micro-Nano Application Technology Academician Workstation, Xinxiang, Henan 453003, China

† Electronic supplementary information (ESI) available. See DOI: 10.1039/d0ra07020c



ray diffraction (XRD) test was conducted to confirm the phase of the as-prepared  $\text{Bi}_2\text{MoO}_6$  sample. As shown in Fig. 1a, the sharp and intense diffraction peaks demonstrated that the as-prepared  $\text{Bi}_2\text{MoO}_6$  powder had high crystallinity, and the characteristic diffraction peaks at  $10.9^\circ$ ,  $23.6^\circ$ ,  $28.4^\circ$ ,  $32.6^\circ$ ,  $33.3^\circ$ ,  $36.0^\circ$ ,  $39.8^\circ$ ,  $47.2^\circ$ ,  $55.7^\circ$ ,  $56.3^\circ$ ,  $58.6^\circ$ , and  $76.2^\circ$  were well indexed to (020), (111), (131), (200), (002), (151), (062), (331), (133), (262), and (400) planes of the pure orthorhombic phase of the aurivillius  $\text{Bi}_2\text{MoO}_6$  (JCPDS card no. 21-0102), respectively.<sup>35</sup> The morphology of the  $\text{Bi}_2\text{MoO}_6$  was investigated by scanning electronic microscopy (SEM), transmission electron microscopy (TEM) and scanning transmission electron microscopy (STEM) (Fig. 1b–h). The SEM and TEM images of the  $\text{Bi}_2\text{MoO}_6$  showed a mixed morphology consisting of nanorods with diameters in 50–100 nm and nanosheets of 20–30 nm thickness; the elements of Bi, Mo, and O had a uniform distribution on  $\text{Bi}_2\text{MoO}_6$ 's surface by STEM coupled with energy dispersive X-ray spectroscopy (EDX) (Fig. 1d–g). Further, the lattice interplanar spacing of 0.24 nm in the high-resolution (HR)-TEM image corresponded to the (022) crystal plane of the orthorhombic  $\text{Bi}_2\text{MoO}_6$  (Fig. 1h).<sup>36</sup> The nanostructures of the aurivillius  $\text{Bi}_2\text{MoO}_6$  facilitate fast transportation of electrons and ions during the electrochemical and photo reactions. In addition, Fig. 1i showed the photo absorption behavior of the nano- $\text{Bi}_2\text{MoO}_6$ , its visible-light response could extend to 600 nm, and the band gap was 2.6 eV, thus confirming that the as-prepared nano- $\text{Bi}_2\text{MoO}_6$  was able to harvest the solar energy.

The electrochemical processes of the  $\text{Bi}_2\text{MoO}_6$  electrodes were measured using a standard three-electrode system, which contained the working electrode of  $\text{Bi}_2\text{MoO}_6$ , the platinum counter electrode, and the Hg/HgO reference electrode in a 1 M KOH electrolyte at room temperature and atmospheric pressure. The experimental measurements of the irradiation of

visible light were carried out in a 100 ml quartz electrochemical cell. A 300 W xenon arc lamp was utilized as a visible-light source, and the light intensity is  $8 \text{ mW cm}^{-2}$ . In the dark, the cyclic voltammetry (CV) curves of the  $\text{Bi}_2\text{MoO}_6$  electrodes exhibited well-defined faradaic redox peaks at  $-0.35 \text{ V}$ ,  $-0.50 \text{ V}$ , and  $-0.75 \text{ V}$ , corresponding to the conversion of the Bi metal to Bi(III) being mediated by  $\text{OH}^-$  from the electrolyte, which clearly indicated the typical pseudo-capacitance charge storage behavior of the  $\text{Bi}_2\text{MoO}_6$  electrode (Fig. S1†).<sup>31,42</sup> Moreover, the cathodic and anodic peak currents depended linearly on the square roots of the scan rates (Fig. S2†), revealing that the diffusion of the electrolyte was a rate-controlling step during the pseudo-capacitance charge storage process of the  $\text{Bi}_2\text{MoO}_6$  electrodes.

Under illumination of visible light, the  $\text{Bi}_2\text{MoO}_6$  electrode showed similar pseudo-capacitance charge storage behavior to that exhibited in the dark from the CV results (Fig. S3†), but the intensity and position of anodic and cathodic peaks showed significant changes (Fig. 2a–f). Specifically, the anodic peak at  $-0.35 \text{ V}$  ( $P_{a2}$ ) became much sharper and its intensity increased significantly. Correspondingly, the charging platform at  $-0.46 \text{ V}$  appeared much more obvious from galvanostatic charge–discharge (GCD) result (Fig. 2h). These results clearly manifested that the irradiation of visible light improved the reversibility of the oxidation reaction from  $\text{Bi}^0$  to  $\text{Bi}^{3+}$  and enhanced the peak current of  $P_{a2}$  dramatically.<sup>41</sup> Furthermore, we quantitatively compared the peak current of  $P_{a2}$  at different scan rates, and found that the maximum gain reached 79.5% at the scanning rate of  $50 \text{ mV s}^{-1}$  under the irradiation of visible light (Fig. 2g). In addition, the anodic peak at  $-0.5 \text{ V}$  ( $P_{a1}$ ) shifted negatively, demonstrating that the oxidation from metal bismuth located near the electrode–electrolyte interface to  $\text{Bi}^{3+}$

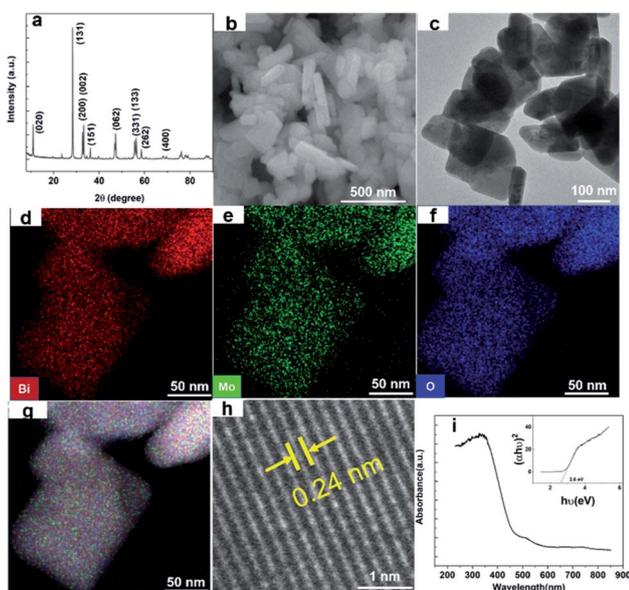


Fig. 1 (a) XRD pattern, (b) FESEM images, (c) TEM graphics, (d–g) STEM images, (h) high-resolution TEM graphic, and (i) optical absorption spectrum of the as-synthesized  $\text{Bi}_2\text{MoO}_6$ .

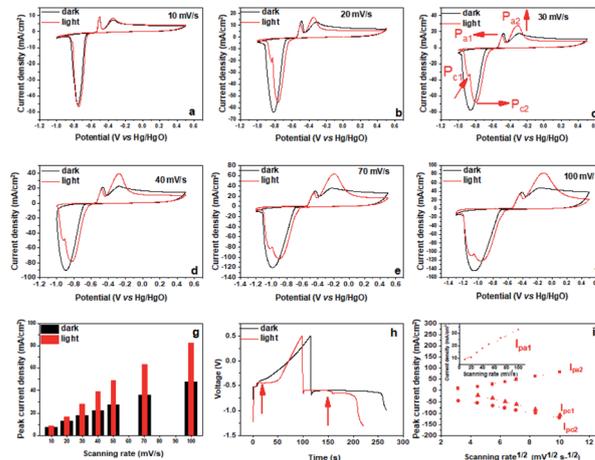


Fig. 2 (a–f) The CV curves of the  $\text{Bi}_2\text{MoO}_6$  electrode and (g) the  $P_{a2}$  current density at different scan rates in the dark and under the irradiation of visible light; (h) the GCD profiles at  $1 \text{ A g}^{-1}$  of the  $\text{Bi}_2\text{MoO}_6$  electrode in the dark and under the irradiation of visible light with a light intensity of  $8 \text{ mW cm}^{-2}$ ; (i) the relationship between the peak currents of the anodic and cathodic peaks and the sweep rate of the  $\text{Bi}_2\text{MoO}_6$  electrode (the currents of the anode and cathode are abbreviated as  $I_{P_{a1}}$ ,  $I_{P_{a2}}$ ,  $I_{P_{c1}}$  and  $I_{P_{c2}}$ , respectively).



occurred more easily under the irradiation of visible light.<sup>43</sup> For the reduction process, the cathodic peak at  $-0.74$  V ( $P_c$ ) had a positive shift, demonstrating that the reduction of  $\text{Bi}^{3+}$  had a smaller overpotential. Meanwhile,  $P_c$  split into a shoulder peak ( $P_{c1}$ ) and a main peak ( $P_{c2}$ ), also,  $P_{c1}$  became increasingly strong as the sweep rate increased.  $P_{c2}$  is attributed to the reduction of  $\text{Bi}^{3+}$  the dissolved species  $\text{Bi}^{3+}$  to  $\text{Bi}^0$ , and the shoulder peak  $P_{c1}$  may correspond to water splitting to produce hydrogen. Simultaneously, two charging platforms arose during the discharging (Fig. 2h). These results highlighted that the irradiation of visible light had a great influence on the electrochemical behavior of the  $\text{Bi}_2\text{MoO}_6$  electrode, and that it can improve the reversibility, decrease the overpotential and increase the peak current during the anodic and cathodic processes. Furthermore, the irradiation of visible light also affected the dynamics of the processes at the  $\text{Bi}_2\text{MoO}_6$  electrode (Fig. 2i). As that in the dark, the transformation between  $\text{Bi}^0$  and  $\text{Bi}^{3+}$  is still a diffusion of the electrolyte dominant process under illumination, because  $I_{P_{a2}}$  and  $I_{P_{c2}}$  depend linearly on the square roots of the scan rates. However, the oxidation of metal bismuth switched to a capacitive process, because  $I_{P_{a1}}$  was linearly dependent on the scan rates. This result illustrated the irradiation of visible light also accelerated the mobility of ions, which enabled the diffusion of the electrolyte opposed to the rate-controlling step. Further, the electrochemical impedance spectroscopy (EIS) of the  $\text{Bi}_2\text{MoO}_6$  electrode is performed in the dark and under the irradiation of visible light, and the results are shown in Fig. S5.† Compared with the quasi-semicircle plot in the dark condition, the  $\text{Bi}_2\text{MoO}_6$  electrode shows a sharper plot in the low frequency region under the irradiation of visible light, verifying that the irradiation of visible light accelerated the ionic diffusion of the electrolyte again.

The above results demonstrated that the irradiation of visible light influenced the electrochemical processes and dynamics of the  $\text{Bi}_2\text{MoO}_6$  electrode remarkably, but this influence mechanism has rarely been reported because of the complexity of the photo processes. As a typical n-type semiconductor,  $\text{Bi}_2\text{MoO}_6$  can generate excitons, and electron-hole pairs after absorbing incident photons. Then the holes and electrons, separating in the electric field, play important roles as oxidants and reducing agents in the following process, respectively. The same photo process of the  $\text{Bi}_2\text{MoO}_6$  also occurs in the photocatalysis reactions, and the catalytic mechanism of photo-generated carriers has been studied in depth by adding appropriate sacrificial agents.<sup>28,44</sup>

Inspired by the sacrificial agents in photocatalysis, we added triethanolamine (TEA) to the electrolyte as the hole sacrificial agent to explore the influence mechanism of the photo-generated holes and electrons on the electrochemical behavior of  $\text{Bi}_2\text{MoO}_6$  electrode. Under the irradiation of visible light, the photo-generated holes on the  $\text{Bi}_2\text{MoO}_6$  electrode were exhausted by the TEA, avoiding the recombination of photo-generated electrons at the same time. The valence band (VB) edge of  $\text{Bi}_2\text{MoO}_6$  located at  $-0.32 + 2.6$  V (vs. NHE, pH 7), which is much lower than the oxidation potential of  $\text{Bi}^0$ , therefore, the photo-generated holes on the  $\text{Bi}_2\text{MoO}_6$  electrode can oxidize  $\text{Bi}^0$ . However, there is no obvious change for the anodic peak  $P_{a1}$

when compared with the CV curves without the TEA (Fig. 3), and only the  $P_{a2}$  declined dramatically. This result clearly confirmed that the photo-generated holes, preferring to accumulate in the bulk electrode, selectively enhanced the oxidation from  $\text{Bi}^0$  in the electrode bulk to  $\text{Bi}^{3+}$ , but the Bi-metal in the interface cannot be oxidized. Summarizing the above oxidation process, we can find that it is the holes induced by photo on the  $\text{Bi}_2\text{MoO}_6$  electrode, catalyze the oxidation of  $\text{Bi}_2\text{MoO}_6$ , which means that the above oxidation process is a typical photo-induced self-catalytic oxidation reaction of  $\text{Bi}_2\text{MoO}_6$  (Fig. 4).<sup>45</sup> The conduction band (CB) edge of the  $\text{Bi}_2\text{MoO}_6$  is located at  $-0.32$  V (vs. NHE, pH 7), which is much higher than the reduction potential of  $\text{Bi}^{3+}$ , thus, the photo-generated electrons on the  $\text{Bi}_2\text{MoO}_6$  electrode are unable to reduce the  $\text{Bi}^{3+}$ . Meanwhile, the cathodic peak of the  $\text{Bi}_2\text{MoO}_6$  electrode had a further positive shift, signifying that the overpotential of the reduction reaction decreased continuously, and this phenomenon became much more obvious as the scan rate increased. When the scan rate increased to  $100$   $\text{mV s}^{-1}$ , the decrement in the reduction overpotential reached  $184$  mV (Fig. 3h), verifying the photo-induced self-catalytic reduction reaction of  $\text{Bi}_2\text{MoO}_6$  (Fig. 4). In other words, the photo-generated electrons on the electrode can transfer rapidly to an external circuit and accelerate the reduction of the  $\text{Bi}_2\text{MoO}_6$  electrode.

The stability of the  $\text{Bi}_2\text{MoO}_6$  electrode under the irradiation of visible light was further investigated by cycling at a scan rate of  $50$   $\text{mV s}^{-1}$ , and the results were shown in Fig. 5 and S4.† For the entire 200 cycles, the peak current retention rate of  $P_{a2}$  maintained higher than 92.5%, suggesting that photo-generated holes on the  $\text{Bi}_2\text{MoO}_6$  electrode had a good stability. For the reduction process, the potential of  $P_{c2}$  continued to shift slightly to positive direction, but the  $P_{c2}$  had an obvious decay (Fig. S4†), the possible reason was the reduction of water.

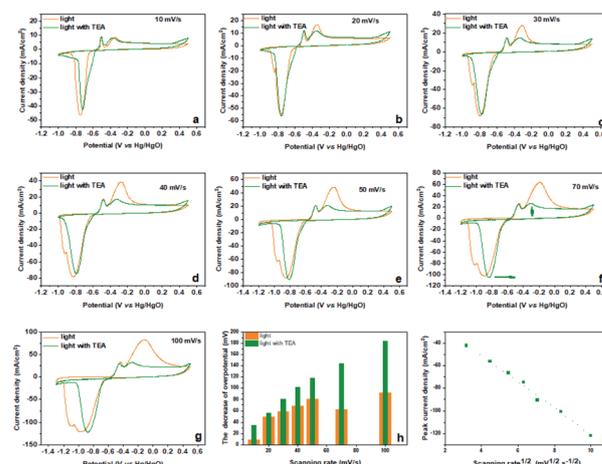
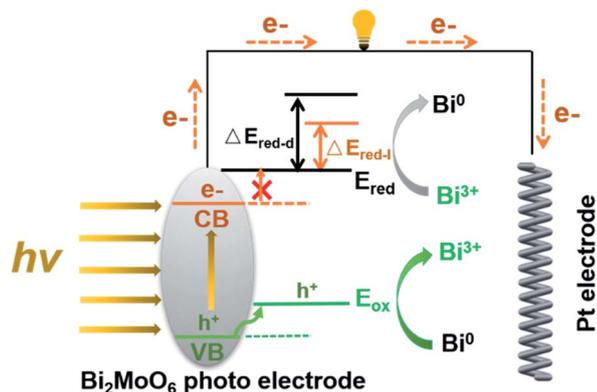


Fig. 3 (a–g) The CV curves of the  $\text{Bi}_2\text{MoO}_6$  electrode at different scan rates and (h) the decrement in the reduction overpotential of the reduction reaction under the irradiation of visible light before and after the addition of the TEA, and the visible light intensity is  $8$   $\text{mW cm}^{-2}$ ; (i) the relationship between the peak current of the cathodic peak and the sweep rate of the  $\text{Bi}_2\text{MoO}_6$  electrode.





$E_{ox}$ : Potential of oxidation;  $E_{red}$ : Potential of reduction

$\Delta E_{red-d}$ : Overpotential of reduction in dark condition

$\Delta E_{red-l}$ : Overpotential of reduction in light condition

Fig. 4 The proposed photo-induced self-catalytic redox mechanism of the  $\text{Bi}_2\text{MoO}_6$  electrode under the irradiation of visible light.

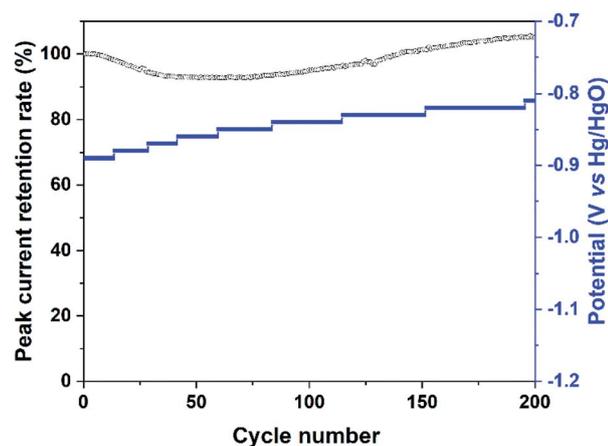


Fig. 5 The peak current retention rate of  $P_{a2}$  and the potential of  $P_{c2}$  in the CV curves of the  $\text{Bi}_2\text{MoO}_6$  electrode cycling at a scan rate of  $50 \text{ mV s}^{-1}$  under the irradiation of visible light.

In summary, we aimed to develop novel multifunctional materials to achieve solar energy harvesting and charge storage synchronously. Nano- $\text{Bi}_2\text{MoO}_6$  was synthesized as a material for both solar energy harvesting and charge storage due to its suitable band gap for the absorption in visible light and its well-defined faradaic redox from Bi metal to  $\text{Bi}^{3+}$ . The irradiation of visible light had a considerable influence on the electrochemical processes and the dynamics of the  $\text{Bi}_2\text{MoO}_6$  electrode. In accordance with the rule of energy matching, the photo-generated holes could oxidize the Bi metal to  $\text{Bi}^{3+}$ , and the photo-generated electrons could decrease the overpotential of the reduction process. Furthermore, the irradiation of visible light also accelerated the ionic diffusion of the electrolyte. This work provides a unique perspective for the development of new multifunctional materials to simultaneously achieve solar energy harvesting and electrical energy storage, and thereby

open pathways towards the efficient integrated energy systems with simplified processes.

## Conflicts of interest

There are no conflicts to declare.

## Acknowledgements

This work was financially supported by grants from the National Natural Science Foundation of China (No. 51901200, U1904189, 51801171), Key Scientific Research Project Plans of Higher Education Institutions in Henan Province (No. 20B430015), and the Regional Joint Fund of Basic and Applied Basic Research of Guangdong Province (No. 2019A1515110725).

## References

- 1 T. Chen, L. Qiu, Z. Yang, Z. Cai, J. Ren, H. Li, H. Lin, X. Sun and H. Peng, *Angew. Chem., Int. Ed.*, 2012, **51**, 11977–11980.
- 2 Y. Jin, Z. Li, L. Qin, X. Liu, L. Mao, Y. Wang, F. Qin, Y. Liu, Y. Zhou and F. Zhang, *Adv. Mater. Interfaces*, 2017, **4**, 1700704.
- 3 E. Navarrete-Astorga, D. Solis-Cortes, J. Rodriguez-Moreno, E. A. Dalchiele, R. Schrebler, F. Martin and J. R. Ramos-Barrado, *Chem. Commun.*, 2018, **54**, 10762–10765.
- 4 T. Song and B. Sun, *ChemSusChem*, 2013, **6**, 408–410.
- 5 Y. Fu, H. Wu, S. Ye, X. Cai, X. Yu, S. Hou, H. Kafafy and D. Zou, *Energy Environ. Sci.*, 2013, **6**, 805–812.
- 6 X. Chen, H. Sun, Z. Yang, G. Guan, Z. Zhang, L. Qiu and H. Peng, *J. Mater. Chem. A*, 2014, **2**, 1897–1902.
- 7 A. Lee, M. Vörös, W. M. Dose, J. Niklas, O. Poluektov, R. D. Schaller, H. Iddir, V. A. Maroni, E. Lee, B. Ingram, L. A. Curtiss and C. S. Johnson, *Nat. Commun.*, 2019, **10**, 4946.
- 8 B. D. Boruah, A. Mathieson, B. Wen, S. Feldmann, W. M. Dose and M. De Volder, *Energy Environ. Sci.*, 2020, **13**, 2414–2421.
- 9 Z. Zhang, X. Chen, P. Chen, G. Guan, L. Qiu, H. Lin, Z. Yang, W. Bai, Y. Luo and H. Peng, *Adv. Mater.*, 2014, **26**, 466–470.
- 10 F. Zhou, Z. Ren, Y. Zhao, X. Shen, A. Wang, Y. Y. Li, C. Surya and Y. Chai, *ACS Nano*, 2016, **10**, 5900–5908.
- 11 S. Safshekan, I. Herraiz-Cardona, D. Cardenas-Morcoso, R. Ojani, M. Haro and S. Gimenez, *ACS Energy Lett.*, 2017, **2**, 469–475.
- 12 W. Guo, X. Xue, S. Wang, C. Lin and Z. L. Wang, *Nano Lett.*, 2012, **12**, 2520–2523.
- 13 H.-D. Um, K.-H. Choi, I. Hwang, S.-H. Kim, K. Seo and S.-Y. Lee, *Energy Environ. Sci.*, 2017, **10**, 931–940.
- 14 R. Liu, Y. Liu, H. Zou, T. Song and B. Sun, *Nano Res.*, 2017, **10**, 1545–1559.
- 15 H. Meng, S. Pang and G. Cui, *ChemSusChem*, 2019, **12**, 3431–3447.
- 16 A. Gurung and Q. Qiao, *Joule*, 2018, **2**, 1217–1230.
- 17 B. Luo, D. Ye and L. Wang, *Adv. Sci.*, 2017, **4**, 1700104.
- 18 A. Das, S. Deshagani, R. Kumar and M. Deepa, *ACS Appl. Mater. Interfaces*, 2018, **10**, 35932–35945.



- 19 N. F. Yan, G. R. Li and X. P. Gao, *J. Mater. Chem. A*, 2013, **1**, 7012.
- 20 P. Liu, H. X. Yang, X. P. Ai, G. R. Li and X. P. Gao, *Electrochem. Commun.*, 2012, **16**, 69–72.
- 21 A. Paoletta, C. Faure, G. Bertoni, S. Marras, A. Guerfi, A. Darwiche, P. Hovington, B. Commarieu, Z. Wang, M. Prato, M. Colombo, S. Monaco, W. Zhu, Z. Feng, A. Vijh, C. George, G. P. Demopoulos, M. Armand and K. Zaghib, *Nat. Commun.*, 2017, **8**, 14643.
- 22 H. Gong, T. Wang, H. Xue, X. Fan, B. Gao, H. Zhang, L. Shi, J. He and J. Ye, *Energy Storage Materials*, 2018, **13**, 49–56.
- 23 R. Liu, C. Liu and S. Fan, *J. Mater. Chem. A*, 2017, **5**, 23078–23084.
- 24 B. J. Trzeźniewski and W. A. Smith, *J. Mater. Chem. A*, 2016, **4**, 2919–2926.
- 25 D. B. D. Boruah, A. Mathieson, B. Wen, C. Jo, F. Deschler and M. De Volder, *Nano Lett.*, 2020, **20**(8), 5967–5974.
- 26 B. D. Boruah and A. Misra, *ACS Appl. Energy Mater.*, 2019, **2**, 278–286.
- 27 T. N. Murakami, N. Kawashima and T. Miyasaka, *Chem. Commun.*, 2005, **26**, 3346–3348.
- 28 H. Yu, L. Jiang, H. Wang, B. Huang, X. Yuan, J. Huang, J. Zhang and G. Zeng, *Small*, 2019, **15**, e1901008.
- 29 X. Hu, W. Zhang, X. Liu, Y. Mei and Y. Huang, *Chem. Soc. Rev.*, 2015, **44**, 2376–2404.
- 30 J. Wen, S. Sun, B. Zhang, N. Shi, X. Liao, G. Yin, Z. Huang, X. Chen and X. Pu, *RSC Adv.*, 2019, **9**, 4693–4699.
- 31 B. Senthilkumar, R. K. Selvan, L. Vasylechko and M. Minakshi, *Solid State Sci.*, 2014, **35**, 18–27.
- 32 Z.-Q. Liu, L.-Y. Tang, N. Li, K. Xiao, J. Wang, J.-H. Zhang, Y.-Z. Su and Y.-X. Tong, *J. Electrochem. Soc.*, 2012, **159**, D582–D586.
- 33 D. Zhu, W. Wang, J. Zhu, S. Chen and X. Liu, *J. Solid State Electrochem.*, 2016, **21**, 403–408.
- 34 F. Wu, X. Wang, W. Zheng, H. Gao, C. Hao and C. Ge, *Electrochim. Acta*, 2017, **245**, 685–695.
- 35 T. Yu, Z. Li, S. Chen, Y. Ding, W. Chen, X. Liu, Y. Huang and F. Kong, *ACS Sustainable Chem. Eng.*, 2018, **6**, 7355–7361.
- 36 K. J. Samdani, J. H. Park, D. W. Joh and K. T. Lee, *ACS Sustainable Chem. Eng.*, 2018, **6**, 16702–16712.
- 37 P. V. Shinde, N. M. Shinde, J. M. Yun, R. S. Mane and K. H. Kim, *ACS Omega*, 2019, **4**, 11093–11102.
- 38 S. S. Patil, D. P. Dubal, V. G. Deonikar, M. S. Tamboli, J. D. Ambekar, P. Gomez-Romero, S. S. Kolekar, B. B. Kale and D. R. Patil, *ACS Appl. Mater. Interfaces*, 2016, **8**, 31602–31610.
- 39 A. Martínez-de la Cruz, S. Obregón Alfaro, E. López Cuéllar and U. Ortiz Méndez, *Catal. Today*, 2007, **129**, 194–199.
- 40 J. Hu, Y. Xie, J. Zheng, Y. Lai and Z. Zhang, *Nano Res.*, 2020, **13**, 2650–2657.
- 41 M. Zargazi and M. H. Entezari, *Ultrason. Sonochem.*, 2020, **67**, 105145.
- 42 V. D. Nithya, R. Kalai Selvan, D. Kalpana, L. Vasylechko and C. Sanjeeviraja, *Electrochim. Acta*, 2013, **109**, 720–731.
- 43 V. Vivier, A. Régis, G. Sagon, J. Y. Nedelec, L. T. Yu and C. Cachet-Vivier, *Electrochim. Acta*, 2001, **46**, 907–914.
- 44 Z. Wang, H.-C. Chiu, A. Paoletta, R. Gauvin, K. Zaghib and G. P. Demopoulos, *Sustainable Energy Fuels*, 2020, **4**, 4789–4799.
- 45 S. H. DuVall and R. L. McCreery, *J. Am. Chem. Soc.*, 2000, **122**, 6759–6764.

