

Cite this: *Nanoscale Adv.*, 2024, 6, 1781

Construction of a BiVO₄/V₅-MoS₂ S-scheme heterojunction for efficient photocatalytic nitrogen fixation†

Han-Ying Luo,^{‡a} Zhao-Lei Liu,^{‡a} Meng-Ran Zhang,^a Yan-Fei Mu^{ID} ^{*b} and Min Zhang^{ID} ^{*a}

Photocatalytic nitrogen (N₂) reduction to ammonia (NH₃), adopting H₂O as the electron source, suffers from low efficiency owing to the sluggish kinetics of N₂ reduction and the requirement of a substantial thermodynamic driving force. Herein, we present a straightforward approach for the construction of an S-scheme heterojunction of BiVO₄/V₅-MoS₂ to successfully achieve photocatalytic N₂ fixation, which is manufactured by coupling an N₂-activation component (V₅-MoS₂ nanosheet) and water-oxidation module (BiVO₄ nanocrystal) through electrostatic self-assembly. The V₅-MoS₂ nanosheet, enriched with sulfur vacancies, plays a pivotal role in facilitating N₂ adsorption and activation. Additionally, the construction of the S-scheme heterojunction enhances the driving force for water oxidation and improves charge separation. Under simulated sunlight irradiation (100 mW cm⁻²), BiVO₄/V₅-MoS₂ exhibits efficient photocatalytic N₂ reduction activity with H₂O as the proton source, yielding NH₃ at a rate of 132.8 μmol g⁻¹ h⁻¹, nearly 7 times higher than that of pure V₅-MoS₂. This study serves as a noteworthy example of efficient N₂ reduction to NH₃ under mild conditions.

Received 8th December 2023
Accepted 19th February 2024

DOI: 10.1039/d3na01091k

rsc.li/nanoscale-advances

1. Introduction

Ammonia (NH₃) serves as both a promising energy storage intermediary and an essential raw material affecting agricultural and industrial production.^{1,2} However, given the molecular inertness of N₂ (dissociation energy of 945 kJ mol⁻¹), the current industrial synthesis of NH₃ predominantly relies on the Haber-Bosch process.^{3,4} This traditional method requires high temperature and pressure (300–550 °C and 15–25 MPa), resulting in significant investment costs and high energy consumption.^{5,6} Therefore, the pursuit of synthetic processes under mild conditions has become a paramount objective. Photocatalytic N₂ fixation, employing semiconductor photocatalysts that harness sunlight as the energy source while utilizing N₂ and H₂O as reactants, stands out as a sustainable approach for NH₃ production.^{7–10} Nonetheless, because of the poor N₂ adsorption–activation and carrier separation in

semiconductor photocatalysts, achieving efficient photocatalytic conversion of N₂ to NH₃ remains a scientific challenge.^{11–13}

Efficient N₂ adsorption–activation and good light absorption of photocatalysts are the fundamental prerequisites for realizing the efficient photoreduction of N₂ to produce NH₃.¹⁴ Consequently, the initial step involves the selection of potential semiconductor materials to reasonably design photocatalysts with suitable energy levels for N₂ reduction and favorable properties for N₂ adsorption–activation.^{15–17} Schrauzer and co-workers conducted the pioneering systematic investigation into the photocatalytic NH₃ synthesis using TiO₂-based materials.¹⁸ In recent years, numerous studies have explored traditional metal oxides, carbonaceous materials, and layered double hydroxide semiconductors as photocatalysts for N₂ photofixation.^{19–22} However, these semiconductor materials often exhibit wide band gaps and narrow spectral absorption characteristics, which are not conducive for efficient photon utilization in photocatalytic reactions. Inspired by the MoFe-cofactor in natural nitrogenase, molybdenum disulfide (MoS₂) has garnered significant attention in the field of N₂ fixation due to its structural resemblance to nitrogenase with a Mo–S configuration.^{23,24} As a two-dimensional direct bandgap semiconductor material, MoS₂ not only possesses good photon capture ability but also can be used as a carrier to construct composite catalysts to achieve more efficient catalytic conversion. Nonetheless, MoS₂ exhibits poor activity in photocatalytic N₂ reduction systems using water as the electron source due to

^aMOE International Joint Laboratory of Materials Microstructure, Institute for New Energy Materials and Low Carbon Technologies, School of Materials Science and Engineering, School of Chemistry and Chemical Engineering, Tianjin University of Technology, Tianjin 300384, China. E-mail: zm2016@email.tjut.edu.cn

^bSchool of Chemistry and Chemical Engineering, Yangzhou University, Yangzhou, Jiangsu 225009, China. E-mail: 007916@yzu.edu.cn

† Electronic supplementary information (ESI) available: Materials, characterization approaches, and some supporting data. See DOI: <https://doi.org/10.1039/d3na01091k>

‡ Han-Ying Luo and Zhao-Lei Liu contributed equally to this work.

its lack of water oxidation capacity.²⁵ In addition, akin to other semiconductor photocatalysts, its weak photogenerated carrier separation ability also hinders the progress of photocatalytic N₂ reduction.^{26,27} To improve the photocatalytic activity of semiconductor photocatalysts, extensive research efforts have focused on bolstering charge separation through the construction of p–n, II-type, and S-scheme heterojunctions.^{28–31} Mimicking natural photosynthesis, the S-scheme charge transfer system has received special attention, which can simultaneously facilitate spatial carrier separation and improve redox ability.^{32,33} Additionally, monoclinic BiVO₄ as a “star” material has a broad range of visible light utilization and excellent photoelectric stability, and has been widely used in the study of photoanodic water oxidation.³⁴ Based on this, BiVO₄ has potential to serve as a water oxidation unit for MoS₂ to construct heterojunctions, further enhancing the efficiency of photocatalytic N₂ fixation.

Herein, we present an innovative S-scheme heterojunction prepared by coupling sulfur vacancy enriched MoS₂ nanosheets (V_S-MoS₂) with BiVO₄ nanocrystals through electrostatic self-assembly for photocatalytic N₂ reduction. Under simulated sunlight irradiation (100 mW cm^{−2}), BiVO₄/V_S-MoS₂ exhibits efficient photocatalytic N₂ reduction activity, with a NH₃ yield of 132.8 μmol g^{−1} h^{−1}, nearly 7 times that of pure V_S-MoS₂ (20 μmol g^{−1} h^{−1}). In addition, the experimental results from *in situ* X-ray photoelectron spectroscopy and electron spin resonance spectroscopy verify the presence of sulfur-rich vacancies on the surface of V_S-MoS₂ nanosheets and the successful formation of an S-scheme heterojunction, which are conducive to the improvement of N₂ adsorption–activation and driving force of water oxidation as well as charge separation. The mechanism of the N₂ reduction reaction is speculated using *in situ* diffuse reflectance infrared Fourier transform spectroscopy.

2. Experimental

2.1. Sample preparation

2.1.1. Preparation of V_S-MoS₂ nanosheets. Na₂MoO₄·2H₂O (243 mg) and CH₄N₂S (304 mg) were dissolved in ultrapure water (35 mL). After 30 min of stirring, the mixture was transferred to a 50 mL Teflon autoclave and heated at 180 °C for 48 h. Subsequently, the products were washed 3 times with a 0.1 M HCl solution to eliminate any residual nitrogen in the sample. The V_S-MoS₂ nanosheet powder can be obtained after vacuum drying at 60 °C for 12 h.

2.1.2. Preparation of BiVO₄ nanocrystals. In a three-neck flask, a mixture containing 484 mg of Bi(NO₃)₃·5H₂O, 20 mL of octadecene, 3 mL of oleylamine, and 3 mL of oleic acid was heated to 170 °C until it became transparent, and then stored at 100 °C. NaVO₃ (242 mg) was dissolved in ultrapure water (20 mL) at 100 °C, and added to the above three-neck flask. The resulting mixture was poured into ethanol to induce precipitation. Afterward, it was subjected to three rounds of washing with a hexane and ethanol mixture, followed by vacuum drying at 60 °C for 12 h to obtain a yellow-green BiVO₄ nanocrystal. For further purification, the obtained BiVO₄ quantum dots (20 mg) were dispersed in a solution comprising 10 mL of isopropanol

and 1 mL of KI in DMSO (0.1 M). Ultrasonic treatment was performed for 10 min to remove the ligands introduced during the synthesis process and prevent their interference in the photocatalytic N₂ fixation experiment.

2.1.3. Preparation of BiVO₄/V_S-MoS₂. V_S-MoS₂ nanosheets (30 mg) and BiVO₄ nanocrystals (10 mg) were dispersed in toluene (10 mL). The mixture was ultrasonicated for 10 min and stirred for 3 h in the dark. The suspension was centrifuged followed by washing 3 times with hexane. The BiVO₄/V_S-MoS₂ heterojunction can be obtained after vacuum drying at 60 °C for 12 h. In addition, a series of samples can be obtained by adjusting the mass ratio of BiVO₄ and V_S-MoS₂ as 10 mg : 10 mg, 10 mg : 20 mg, 10 mg : 40 mg, denoted as BiVO₄/V_S-MoS₂ (1/1), BiVO₄/V_S-MoS₂ (1/2) and BiVO₄/V_S-MoS₂ (1/4), respectively.

2.2. Photocatalytic N₂ reduction experiment

The photocatalytic N₂ reduction reactions were carried out in a gas–solid reaction system (25 °C). 20 mg of photocatalyst was placed on the sample table of the reactor. 200 μL of water was injected into the bottom of the reactor as the proton source. Before initiating the photocatalytic reactions, the system was heated to produce water vapor. The system was meticulously degassed to remove air with N₂ (99.999%), where N₂ was bubbled through 1 M HCl as well as a mixed solution of potassium permanganate and KOH to ensure the removal of potential contaminants. A 300 W xenon lamp equipped with a UVVIS-CUT400 filter was employed as the light source (light intensity of 100 mW cm^{−2}). For detection, 2 mL of H₂O was injected into the system to dissolve the product to form the product solution.

2.3. Product detection

2.3.1. Detection of NH₄⁺. The NH₄⁺ concentration was determined with the indophenol blue method as follows: three chromogenic solutions were first prepared, divided into chromogenic agent A, chromogenic agent B, and chromogenic agent C. Chromogenic agent A was prepared by mixing NaOH (4 g), salicylic acid (5 g), and sodium citrate (5 g); chromogenic agent B was obtained by preparing a 0.05 M NaClO aqueous solution; chromogenic agent C was prepared by using 0.1 g sodium nitroferrocyanide solution. The product solution (2 mL) was mixed with reagent A (2 mL), reagent B (1 mL), and reagent C (200 μL), and then left for two hours to be detected. Finally, the absorbance at a wavelength of 655 nm was measured to quantify the NH₄⁺ content according to the established standard curve of the NH₄Cl solution.

2.3.2. Detection of NO₃[−]. The specific process is as follows: first, prepare 1.0 ppm, 0.5 ppm, 0.2 ppm, and 0.1 ppm solutions of NaNO₃ as the standard solutions to establish the standard curve. Then, the reaction solution (2 mL) was detected by using a model NEXION300 ion chromatograph.

2.3.3. Detection of NO₂[−]. The colorimetric method for the detection of NO₂[−] concentration is as follows: color reagent A was prepared by dissolving sulfonamide (0.5 g) in 50 mL of HCl solution (2 M); color reagent B was prepared by dissolving *N*-(1-naphthyl) ethylenediamine hydrochloride (20 mg) in 20 mL of



ultrapure water. The reaction solution (2 mL) was mixed with color reagent A (40 μ L) and color reagent B (40 μ L), and then left in the dark for 10 min. The absorbance was characterized by UV-vis absorption spectroscopy, and the content of NO_2^- was quantified according to the established standard curve at a wavelength of 540 nm.

2.3.4. Detection of N_2H_4 . The Watt-Chrisp method was employed to determine the concentration of hydrazine hydrate as follows: the color reagent was prepared by mixing *N,N*-dimethyl-4-aminobenzaldehyde (2 g), concentrated HCl (10 mL), and ethanol (100 mL). The reaction solution (2 mL) was mixed with the color reagent (2 mL), and then left in the dark for 15 min. The absorbance at ~ 458 nm was obtained, and the content of N_2H_4 was quantified according to the established standard curve.

2.4. *In situ* diffuse reflectance infrared Fourier transform spectra (DRIFTS)

In situ diffuse reflectance infrared Fourier transform spectra were recorded using a Bruker IFS 66v Fourier transform spectrometer. The samples were mixed with KBr in a quartz mortar and then placed in an infrared reaction chamber. Pure N_2 (99.999%) containing water vapor was continuously introduced into the experimental chamber during *in situ* characterization. The IR spectrum of pure KBr was first collected as a background spectrum. The final spectra were obtained by subtracting the background spectrum from the spectrum of the sample.

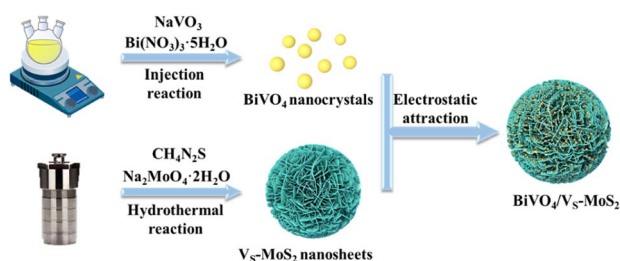
3. Results and discussion

3.1. Preparation and structure of $\text{BiVO}_4/\text{V}_5\text{-MoS}_2$

The synthetic process of $\text{BiVO}_4/\text{V}_5\text{-MoS}_2$ is illustrated in Scheme 1. Briefly, $\text{V}_5\text{-MoS}_2$ and BiVO_4 were initially prepared through the hydrothermal method and thermal injection method, respectively. The measured zeta potential (ζ) of $\text{V}_5\text{-MoS}_2$ is negative (-45 mV), opposited with that of BiVO_4 (33 mV) in toluene (Fig. S1†). This opposite surface charging characteristic of $\text{V}_5\text{-MoS}_2$ and BiVO_4 should facilitate the spontaneous assembly to form $\text{BiVO}_4/\text{V}_5\text{-MoS}_2$ in solution. X-ray diffraction (XRD) patterns (Fig. 1a) show that the as-prepared $\text{V}_5\text{-MoS}_2$ and BiVO_4 can be indexed to the $P6_3/mmc(194)$ hexagonal space group (PDF card no. 01-075-1539) and $I2/a(15)$ monoclinic space group (PDF card no. 00-014-0688),

respectively. The characteristic diffraction peaks at 14.39° , 32.8° , 39.65° and 58.56° of $\text{V}_5\text{-MoS}_2$ correspond to the (002), (100), (103), and (110) crystal planes. The diffraction peaks of BiVO_4 are located at 18.67° , 28.95° , 35.22° , 39.78° , 46.71° , 49.96° , and 59.26° , corresponding to the (110), (121), (002), (211), (240), (-202) , and (123) crystal planes. Notably, the diffraction pattern of $\text{BiVO}_4/\text{V}_5\text{-MoS}_2$ contains the characteristic peaks of both $\text{V}_5\text{-MoS}_2$ and BiVO_4 , indicating the successful loading of BiVO_4 nanocrystals onto $\text{V}_5\text{-MoS}_2$ nanosheets. Scanning electron microscopy (SEM) and transmission electron microscopy (TEM) measurements revealed that $\text{V}_5\text{-MoS}_2$ exhibits a nanoflower morphology (Fig. 1b), which is formed by the gathering of nanosheets (Fig. 1c). Atomic force microscopy (AFM) analysis in Fig. S2† further confirmed that $\text{V}_5\text{-MoS}_2$ has a nanosheet morphology with a mean thickness of ~ 3 nm. TEM images of pure BiVO_4 and $\text{BiVO}_4/\text{V}_5\text{-MoS}_2$ (Fig. 1d and e) show that BiVO_4 nanocrystals are interspersed on $\text{V}_5\text{-MoS}_2$ nanosheets, with an average size of ~ 5 nm, which is almost identical to that of the pure BiVO_4 nanocrystal (Fig. S3†). Further high-resolution TEM (HRTEM) analysis (Fig. 1f) showed clear lattice fringes of 2.72 Å and 3.08 Å, corresponding to the (100) plane of $\text{V}_5\text{-MoS}_2$ (ref. 35) and the (121) plane of monoclinic BiVO_4 (ref. 36) (Fig. S4†), respectively, confirming the presence of BiVO_4 nanocrystals on the surface of $\text{V}_5\text{-MoS}_2$ nanosheets. The corresponding energy-dispersive X-ray spectroscopy (EDS) mapping analysis (Fig. 1g) demonstrated the homogeneous distribution of Mo, S, Bi, V, and O elements on $\text{BiVO}_4/\text{V}_5\text{-MoS}_2$, further verifying the construction of $\text{BiVO}_4/\text{V}_5\text{-MoS}_2$.

To gain insights into the structural characteristics of the as-prepared photocatalysts, electron paramagnetic resonance (EPR) measurements were performed initially. As depicted in Fig. 1h, a distinct EPR signal at $g = 2.003$ can be recognized for $\text{V}_5\text{-MoS}_2$, signifying the formation of the sulfur vacancies on the surface of $\text{V}_5\text{-MoS}_2$.³⁷ In the case of $\text{BiVO}_4/\text{V}_5\text{-MoS}_2$, a stronger EPR signal is detected, providing evidence for the formation of heterojunction interfaces, which can turn spin-coupled electron pairs into unpaired electron states.³⁸ Sulfur vacancies can serve as active sites to promote carrier capture and N_2 adsorption and activation.³⁹ The X-ray photoelectron spectra (XPS) were further analyzed to investigate the potential interfacial interaction and charge transfer processes within the $\text{BiVO}_4/\text{V}_5\text{-MoS}_2$ heterojunction. Doublet XPS characteristic peaks can be identified at 229.3 eV and 232.4 eV in pure $\text{V}_5\text{-MoS}_2$, attributed to Mo 3d_{5/2} and Mo 3d_{3/2} (Fig. S5a†), which can confirm the reduction of Mo^{6+} to Mo^{4+} during the growth of $\text{V}_5\text{-MoS}_2$ nanosheets.⁴⁰ In addition, a comparison between pure $\text{V}_5\text{-MoS}_2$, BiVO_4 , and $\text{BiVO}_4/\text{V}_5\text{-MoS}_2$ reveals noticeable shifts in binding energy for Mo 3d, S 2p, Bi 4f, and V 2p. Specifically, Mo 3d and S 2p exhibit positive shifts of 0.50 eV and 0.20 eV, while Bi 4f and V 2p display negative shifts of 0.20 eV and 0.20 eV, respectively (Fig. S5b–d†). These charge redistributions within the $\text{BiVO}_4/\text{V}_5\text{-MoS}_2$ heterojunction reflect a strong chemical interaction at the interface of the $\text{BiVO}_4/\text{V}_5\text{-MoS}_2$ heterojunction. This robust interfacial electronic coupling in $\text{BiVO}_4/\text{V}_5\text{-MoS}_2$ is expected to facilitate the charge transfer between the $\text{V}_5\text{-MoS}_2$ nanosheets and BiVO_4 nanocrystals.



Scheme 1 Schematic illustration of the preparation process of $\text{BiVO}_4/\text{V}_5\text{-MoS}_2$.



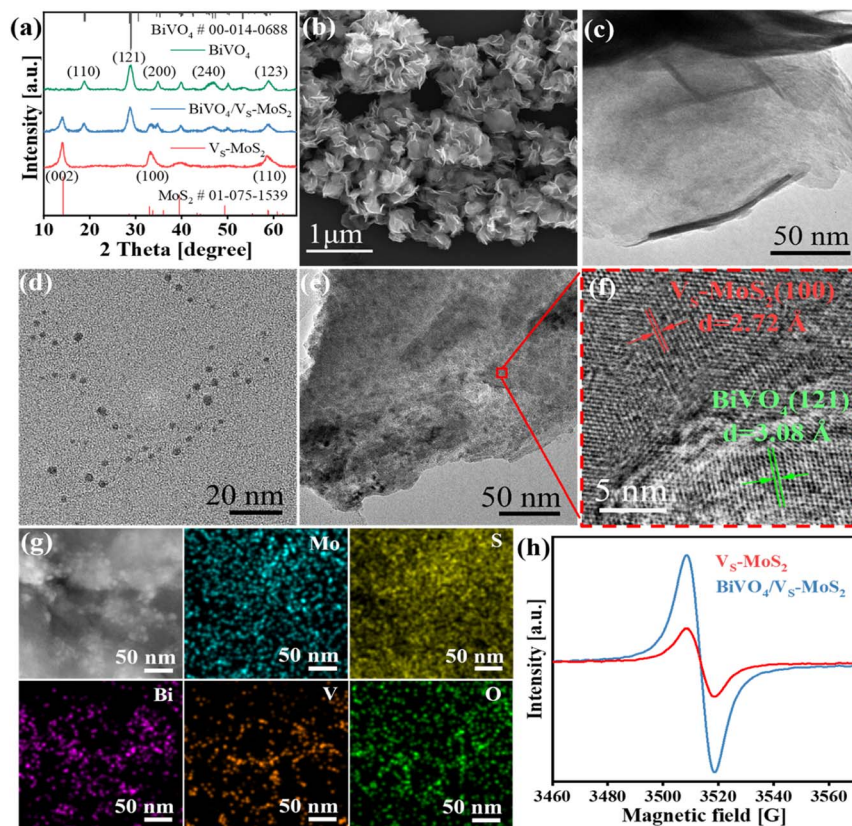


Fig. 1 (a) XRD patterns of $\text{V}_5\text{-MoS}_2$, BiVO_4 , and $\text{BiVO}_4/\text{V}_5\text{-MoS}_2$. (b and c) SEM and TEM images of a $\text{V}_5\text{-MoS}_2$ nanosheet. (d) TEM image of a BiVO_4 nanocrystal. (e–g) TEM, HRTEM, and corresponding EDS images of $\text{BiVO}_4/\text{V}_5\text{-MoS}_2$. (h) EPR spectra of $\text{V}_5\text{-MoS}_2$ and $\text{BiVO}_4/\text{V}_5\text{-MoS}_2$.

3.2. S-scheme mechanism of the $\text{BiVO}_4/\text{V}_5\text{-MoS}_2$ heterojunction

The energy levels and interfacial charge transfer processes of $\text{BiVO}_4/\text{V}_5\text{-MoS}_2$ were explored to evaluate the photocatalytic N_2 reduction potential of the as-prepared photocatalysts. Fig. 2a and b show the UV-vis diffuse reflectance spectra (DRS), revealing a good light response for both the $\text{V}_5\text{-MoS}_2$ nanosheets and BiVO_4 nanocrystals. From the corresponding Tauc plots (insets of Fig. 2a and b), the band gaps (E_g) of $\text{V}_5\text{-MoS}_2$ and BiVO_4 can be deduced to be 1.71 eV and 2.22 eV, respectively. To ascertain the energy band structures of $\text{V}_5\text{-MoS}_2$ and BiVO_4 , the onset edge (E_i) and the secondary electron cutoff (E_{cutoff}) were determined by carrying out ultraviolet photoelectron spectroscopy (UPS) (Fig. 2c). According to the equation $E_{\text{VB}} = 21.22 - (E_{\text{cutoff}} - E_i)$, the valence band edge potentials (E_{VB}) can be calculated to be 1.35 V and 2.23 V *versus* the standard hydrogen electrode (vs. NHE) for $\text{V}_5\text{-MoS}_2$ and BiVO_4 , respectively. By combining the values from E_g and E_{VB} analysis, the corresponding conduction band edge potentials (E_{CB}) can be calculated to be -0.36 V and 0.01 V (vs. NHE) for $\text{V}_5\text{-MoS}_2$ and BiVO_4 , respectively. The resultant energy band structures of $\text{V}_5\text{-MoS}_2$ and BiVO_4 are presented in Fig. 2d. The photogenerated electrons in $\text{V}_5\text{-MoS}_2$ possess sufficient energy to drive the photo-reduction of N_2 to NH_3 (-0.13 V vs. NHE),⁴¹ while their driving force for H_2O oxidation is relatively weak. Meanwhile, the photogenerated holes in BiVO_4 can trigger H_2O photooxidation

to O_2 (0.82 V vs. NHE, pH = 7), while they are insufficient for the photoreduction of N_2 . In addition, the Fermi levels were also estimated according to the UPS spectra, to be -4.62 eV and -5.27 eV (vs. vacuum) for $\text{V}_5\text{-MoS}_2$ and BiVO_4 , respectively. These results reveal that both $\text{V}_5\text{-MoS}_2$ and BiVO_4 are n-type semiconductors, consistent with the results from the Mott-Schottky tests, where the slopes of Mott-Schottky curves for both BiVO_4 and $\text{V}_5\text{-MoS}_2$ are positive (Fig. S6†). From the above results and analysis, we can preliminarily deduce that, given the higher Fermi level of $\text{V}_5\text{-MoS}_2$ compared to that of BiVO_4 , the free electrons in $\text{V}_5\text{-MoS}_2$ would spontaneously flow to BiVO_4 to establish a Fermi level equilibrium upon contact (Fig. S7(I–II)†). A built-in electric field and band bending of the interface in $\text{BiVO}_4/\text{V}_5\text{-MoS}_2$ would be formed. Driven by the interfacial built-in electric field, photogenerated electrons in BiVO_4 would recombine with the photogenerated holes in $\text{V}_5\text{-MoS}_2$ (Fig. S7(III)†), facilitating the formation of the S-scheme charge transfer pathway.

To scrutinize the S-scheme charge transfer pathway in the $\text{BiVO}_4/\text{V}_5\text{-MoS}_2$ heterojunction, *in situ* irradiated XPS (ISI-XPS) spectra were first measured to detect photoinduced changes in the electron cloud density around the nuclei of elements. As depicted in Fig. S8a and b,† the binding energies of Mo 3d and V 2p in $\text{BiVO}_4/\text{V}_5\text{-MoS}_2$ demonstrate a negative displacement (~ 0.40 eV) and positive movement (~ 0.30 eV) under illumination, respectively, compared with those in the dark. These



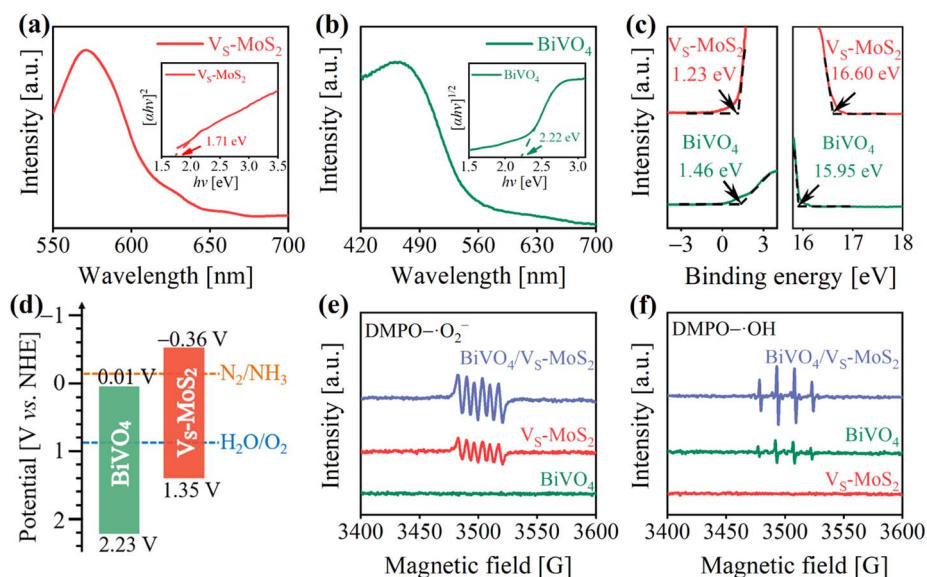


Fig. 2 (a and b) UV-vis DRS spectra of V_5 - MoS_2 and $BiVO_4$. The insets show the corresponding Tauc plots. (c) UPS spectra of V_5 - MoS_2 and $BiVO_4$. (d) Energy band structures of V_5 - MoS_2 and $BiVO_4$ (pH = 7). EPR signals of (e) $DMPO-\cdot O_2^-$ and (f) $DMPO-\cdot OH$ over V_5 - MoS_2 , $BiVO_4$, and $BiVO_4/V_5$ - MoS_2 under illumination.

binding energy changes indicate the accumulation of photo-generated electrons on V_5 - MoS_2 and photogenerated holes on $BiVO_4$, revealing the occurrence of S-scheme interfacial charge transfer within the $BiVO_4/V_5$ - MoS_2 heterojunction. To further confirm this S-scheme interfacial charge transfer mechanism in the $BiVO_4/V_5$ - MoS_2 heterojunction, the EPR spectra were recorded using 5,5-dimethyl-1-pyrroline *N*-oxide (DMPO) as the trapping agent.⁴² In the case of pure V_5 - MoS_2 , only the $DMPO-\cdot O_2^-$ characteristic signal can be identified (Fig. 2e), as its E_{VB} potential is insufficient to drive the oxidation of H_2O to $\cdot OH$. Conversely, in pure $BiVO_4$, only the $DMPO-\cdot OH$ characteristic signal can be recognized (Fig. 2f), due to the insufficient driving force of photogenerated electrons in E_{CB} to trigger the reduction of O_2 to $\cdot O_2^-$ (-0.33 V vs. NHE). For the $BiVO_4/V_5$ - MoS_2 heterojunction, more obvious $DMPO-\cdot O_2^-$ and $DMPO-\cdot OH$ signals can be simultaneously identified, providing compelling evidence for the establishment of an interfacial S-scheme charge transport channel. That is to say, the photogenerated electrons in $BiVO_4$ and the photogenerated holes in V_5 - MoS_2 initially recombine through the interfacial S-scheme channel, leading to the accumulation of photogenerated holes on $BiVO_4$ and photogenerated electrons on V_5 - MoS_2 . This spatial separation of charge carriers in the $BiVO_4/V_5$ - MoS_2 heterojunction would more effectively promote their participation in the redox reaction.

3.3. Photogenerated carrier evolution

In general, the evolution of photogenerated charges within photocatalysts plays a pivotal role in determining their photocatalytic activity.^{43,44} To evaluate the behavior of photogenerated carriers in $BiVO_4/V_5$ - MoS_2 , steady-state photoluminescence (PL) spectrum measurements were first performed. As presented in Fig. 3a, the PL spectrum of $BiVO_4$ displays an intrinsic emission

band at ~ 490 nm under 365 nm excitation, which is related to photogenerated electron-hole recombination. Upon the construction of the $BiVO_4/V_5$ - MoS_2 heterojunction, the PL spectrum exhibits significant quenching, indicating the effective transfer of photogenerated carriers between the $BiVO_4$ nanocrystals and V_5 - MoS_2 nanosheets. In addition, the photo-electrochemical properties of the samples were characterized to assess the potential positive influence of the construction of a heterostructure on charge evolution behavior. Notably, under light irradiation, $BiVO_4/V_5$ - MoS_2 exhibits a smaller charge-transport resistance compared with both pure $BiVO_4$ and V_5 - MoS_2 (Fig. 3b and Table S1†), indicating that the construction of the electric field in the heterojunction improves the charge transport characteristics. From the transient photocurrent ($I-t$) measurements (Fig. 3c), it is evident that $BiVO_4/V_5$ - MoS_2 has a higher current density than both $BiVO_4$ and V_5 - MoS_2 , providing further evidence of expedited charge separation kinetics within the $BiVO_4/V_5$ - MoS_2 heterojunction. Moreover, to more intuitively reveal the space charge separation, Kelvin probe force microscopy (KPFM) was utilized to measure the surface photovoltage (SPV) response of photocatalysts under light. Atomic force microscope (AFM) images (Fig. 3d and f) present the morphology outline of V_5 - MoS_2 and $BiVO_4/V_5$ - MoS_2 nanosheet clusters, which are basically consistent with the SEM results (Fig. 1b). Under illumination, the SPV response zone (Fig. 3e and g) emerges, aligning with the morphology outlines of the samples. These observations also indicate the spatial charge separation and redistribution after light irradiation.⁴⁵ As shown in Fig. 3h, pure V_5 - MoS_2 presents a feeble SPV signal under light irradiation, primarily due to rapid photogenerated carrier recombination. Benefiting from the construction of heterojunctions, $BiVO_4/V_5$ - MoS_2 displays a noticeable enhancement of the SPV response, of ~ 100 mV, far surpassing

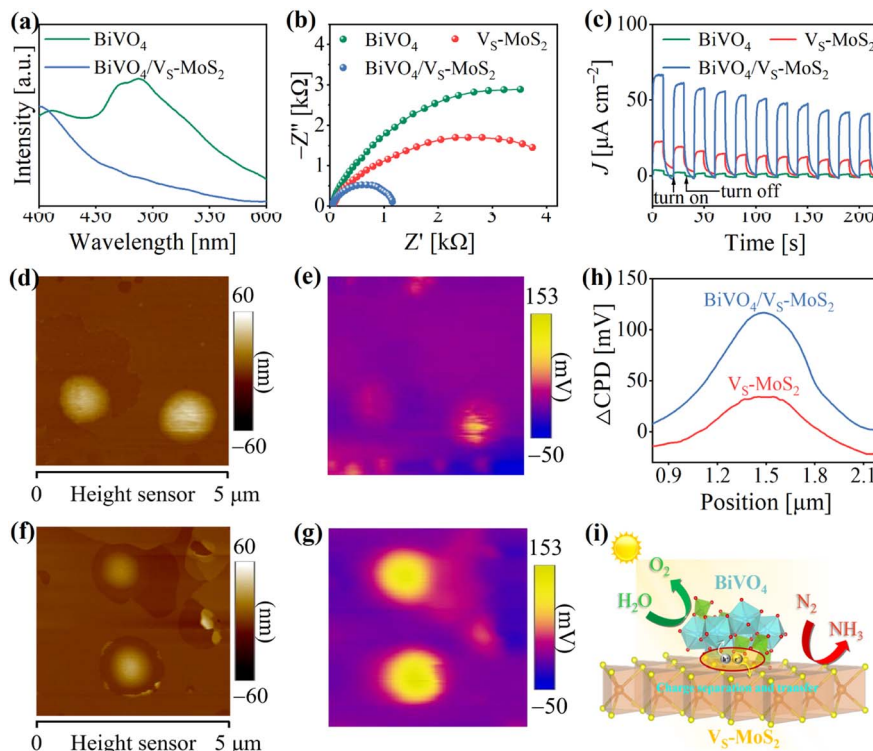


Fig. 3 (a) Steady-state photoluminescence spectra of BiVO₄ and BiVO₄/V₅-MoS₂. (b and c) EIS plots and $I-t$ curves of BiVO₄, V₅-MoS₂, and BiVO₄/V₅-MoS₂ plotted at a bias potential of -0.4 V vs. Ag/AgCl. AFM height images of (d) V₅-MoS₂ and (f) BiVO₄/V₅-MoS₂. SPV images of (e) V₅-MoS₂ and (g) BiVO₄/V₅-MoS₂ under illumination. (h) Potential difference changes (Δ CPD) of V₅-MoS₂ and BiVO₄/V₅-MoS₂ by subtracting the potential under dark conditions from that under illumination. (i) Schematic illustration of the photogenerated carrier separation.

that of pure V₅-MoS₂. Ultimately, these characterization studies underscore that the construction of heterojunctions can promote charge separation, and enhance the accumulation of electrons/holes on V₅-MoS₂ and BiVO₄, respectively, which is beneficial for these carriers to participate in the subsequent multi-electron transfer processes during the photoreduction N₂ reaction (Fig. 3i).

3.4. Photocatalytic N₂ reduction activity

The photocatalytic N₂ reduction reactions were conducted in a gas-solid reaction apparatus (Fig. S9†) containing N₂ and H₂O vapor under simulated sunlight irradiation (100 mW cm^{-2}). To ensure the accuracy of our experiments, the samples were pre-treated with a DMSO solution of KI. This step is crucial to remove any residual ligands introduced during the synthesis process, preventing any interference with the experimental results (Fig. S10,† details are described in the Experimental section). The photocatalytic activities were evaluated by evaluating the NH₄⁺ production using the indophenol blue method (Fig. S11,† details are described in the Experimental section).⁴⁶ As illustrated in Fig. S12,† a prominent absorption signal at ~ 655 nm can be observed, indicating that N₂ is effectively photoreduced to NH₃ in the presence of light for BiVO₄/V₅-MoS₂. The NH₃ generation performances for the as-prepared photocatalysts are presented in Fig. 4a and S13.† For pure BiVO₄, there is virtually no NH₃ synthesis activity due to the

insufficient driving force of electron energy in its conduction band (Fig. 2d). Meanwhile, V₅-MoS₂ exhibits a weak N₂ photoreduction to NH₃ activity, with a rate of $20.0 \pm 1.1 \mu\text{mol g}^{-1} \text{ h}^{-1}$, resulting from its insufficient thermodynamic driving force for H₂O oxidation and serious charge recombination. Upon combining the V₅-MoS₂ nanosheets with the BiVO₄ nanocrystals, the corresponding photocatalytic N₂ reduction activity improves significantly, reaching $132.8 \pm 4.2 \mu\text{mol g}^{-1} \text{ h}^{-1}$ within 4 hours, which is nearly 7-fold that of pure V₅-MoS₂. The apparent quantum efficiency at 575 nm can reach 0.3% (Fig. S14†). Certainly, we also analyzed the NH₄⁺ production by ion chromatography (Fig. S15†), and the results are in agreement with those obtained by the indophenol blue method. Obviously, this improvement of NH₃ evolution activity can be attributed to the increased driving force for water oxidation and the enhanced separation of photogenerated carriers in the BiVO₄/V₅-MoS₂ heterojunction, compared with pure V₅-MoS₂ nanosheets. By adjusting the mass ratio of BiVO₄ and V₅-MoS₂ (Fig. S16†), the NH₃ evolution activities of the samples were screened, in which BiVO₄/V₅-MoS₂ with a mass ratio (BiVO₄ : V₅-MoS₂) of 1 : 3 exhibits the best performance (Fig. 4a).

Moreover, the O₂ evolution activity of BiVO₄/V₅-MoS₂ was also measured through a gas chromatograph, to be $110.0 \pm 4.5 \mu\text{mol g}^{-1} \text{ h}^{-1}$ (Fig. S17†). This result suggests that the rate of consumed electrons (R_{electron}) for N₂ reduction to NH₃ is nearly equal to the rate of consumed holes (R_{hole}) for water oxidation to O₂, in accordance with the formulae of $R_{\text{electron}} = 3 \times R_{\text{NH}_3}$ and



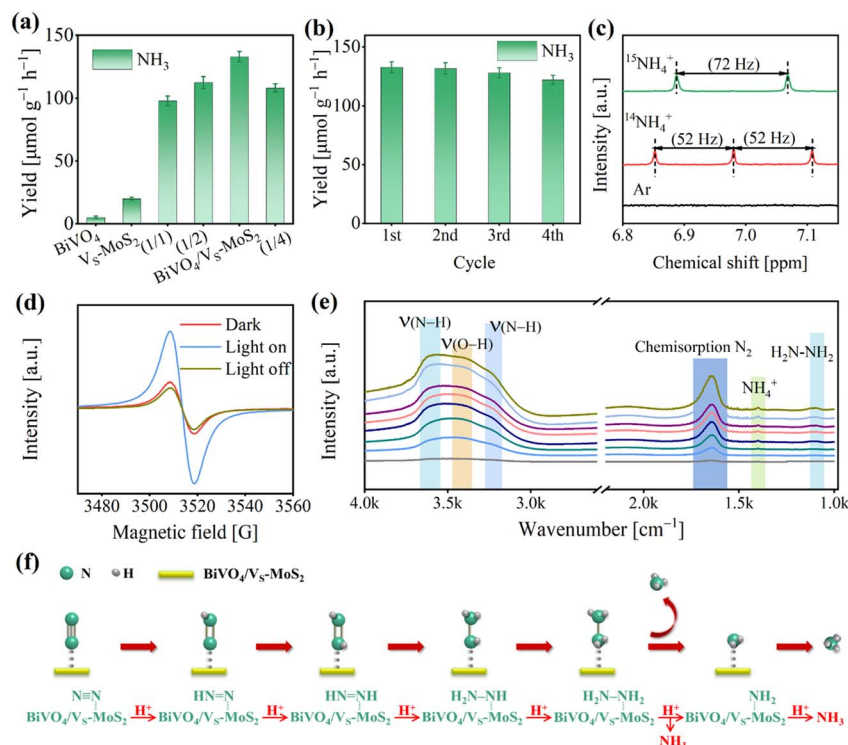


Fig. 4 (a) Photocatalytic NH_3 production rates for BiVO_4 , $\text{V}_5\text{-MoS}_2$, $\text{BiVO}_4/\text{V}_5\text{-MoS}_2$, $\text{BiVO}_4/\text{V}_5\text{-MoS}_2$ (1/1), $\text{BiVO}_4/\text{V}_5\text{-MoS}_2$ (1/2), and $\text{BiVO}_4/\text{V}_5\text{-MoS}_2$ (1/4) under 4 h light irradiation. (b) NH_3 synthesis for four cyclic tests of $\text{BiVO}_4/\text{V}_5\text{-MoS}_2$. (c) The ^1H NMR spectra of products with $^{14}\text{N}_2$ or $^{15}\text{N}_2$ as the reaction atmosphere. (d) *In situ* EPR spectra of $\text{BiVO}_4/\text{V}_5\text{-MoS}_2$ in the dark or under light irradiation in a N_2 atmosphere or light off in a N_2 atmosphere. (e) *In situ* DRIFTS spectra of the photoreduction N_2 reaction. (f) The proposed reaction pathway of the photoreduction N_2 reaction.

$R_{\text{hole}} = 4 \times R_{\text{O}_2}$. This balance signifies the effective utilization of the photogenerated electrons and holes. Based on the above analysis, the whole photocatalytic reaction including N_2 reduction and H_2O oxidation by $\text{BiVO}_4/\text{V}_5\text{-MoS}_2$ can be described as $2\text{N}_2 + 6\text{H}_2\text{O} \rightarrow 4\text{NH}_3 + 3\text{O}_2$. Additionally, the robustness of $\text{BiVO}_4/\text{V}_5\text{-MoS}_2$ was evaluated by reemploying the photocatalyst in four cycles of tests (Fig. 4b), which reveals little decrement in activity. In addition, the XRD patterns (Fig. S18†) and TEM images (Fig. S19†) of $\text{BiVO}_4/\text{V}_5\text{-MoS}_2$ after the photocatalytic reaction exhibit no obvious changes, confirming the good stability of the photocatalyst in this gas-solid photocatalytic system. The Mo 3d and Bi 4f XPS spectra of $\text{BiVO}_4/\text{V}_5\text{-MoS}_2$ after catalysis show almost no significant changes, further suggesting the maintenance of the local coordination environment of the Bi and Mo sites.

To elucidate the origin of NH_3 and O_2 during the photocatalytic reaction, several controlled experiments were carried out using $\text{BiVO}_4/\text{V}_5\text{-MoS}_2$ as the photocatalyst (Fig. S21†). In the absence of a photocatalyst ($\text{BiVO}_4/\text{V}_5\text{-MoS}_2$), N_2 , light, or H_2O , there is nearly no NH_3 evolution, unequivocally demonstrating that the N_2 reduction reaction is indeed a photocatalytic reaction using $\text{BiVO}_4/\text{V}_5\text{-MoS}_2$ as the photocatalyst with H_2O as the proton source. According to the characterization by the colorimetric method and ion chromatography (the details are described in the Experimental section), nearly no NO_2^- and NO_3^- signals in the product system can be detected before and

after the photocatalytic reaction (Fig. S22 and S23†). In addition, we have further measured the N content in the obtained photocatalyst by EDS analysis (Fig. S24 and Table S2†), and no N atoms are present. These results confirm that NH_3 remained unoxidized, and substantiate that NH_3 originated solely from N_2 reduction, excluding impurity conversion as a source. To further verify the origin of NH_3 in the product system, isotope labeling experiments were performed using $\text{BiVO}_4/\text{V}_5\text{-MoS}_2$ as the photocatalyst in atmospheres containing $^{15}\text{N}_2$, $^{14}\text{N}_2$, or Ar, respectively (Fig. S25†). More specifically, no NH_4^+ signal is detected under a Ar atmosphere in the nuclear magnetic resonance (NMR) spectra, while distinct signals belonging to $^{15}\text{NH}_4^+$ or $^{14}\text{NH}_4^+$ can be observed in a $^{15}\text{N}_2$ or $^{14}\text{N}_2$ atmosphere, respectively (Fig. 4c). These results directly confirm that NH_4^+ is indeed generated from the N_2 in the system. In addition, the mass spectrometry analysis of the product (Fig. S26†) was also performed, revealing an $^{18}\text{O}_2$ signal at $m/z = 36$, which confirms that O_2 is produced from H_2O oxidation.

To gain insight into the photocatalytic N_2 reduction process of the $\text{BiVO}_4/\text{V}_5\text{-MoS}_2$ heterojunction, we conducted further characterization to explore the active sites and monitor the evolution of intermediates during the photocatalytic reaction, by using *in situ* EPR spectra and *in situ* diffuse reflectance infrared Fourier transform spectra (DRIFTS), respectively. As presented in Fig. 4d, the intensity of the sulfur vacancy signal for $\text{BiVO}_4/\text{V}_5\text{-MoS}_2$ exhibits an increment when exposed to light

irradiation for 15 min, and decreases once the light was turned off for 10 min, suggesting that the sulfur vacancies facilitate the trapping of photogenerated electrons from the conduction band.^{47,48} The exposed Mo site at the sulfur vacancy is in a coordination unsaturated state, and can act as the photocatalytic active site bonding with N₂. The trapped photogenerated electrons at sulfur vacancies can transfer to N₂ molecules to achieve N₂ reduction and transformation. As the photocatalytic N₂ reduction reaction proceeded, a series of absorption signals were detected belonging to intermediates adsorbed on BiVO₄/V_S-MoS₂ (Fig. 4e). Specifically, an obvious peak at 1648 cm⁻¹ belonging to the chemisorbed N₂ (-N≡N) appeared,⁴⁹ which can be attributed to the efficient adsorption and activation of N₂ by the Mo sites at the vacancies of the nanosheets.⁵⁰ The peak at 3432 cm⁻¹ is attributed to ν(O-H) of H₂O, which serves as the proton source for N₂ activation. The overlapped signal bands at 3230 cm⁻¹ and 3555 cm⁻¹ can be attributed to ν(N-H) stretching modes of NH₃, and the peak at 1403 cm⁻¹ can be attributed to NH₄⁺ adsorption. The enhancement of these characteristic signals implies the progression of N₂ activation and conversion to NH₃ via a multistep proton-coupled electron transfer (PCET) process over BiVO₄/V_S-MoS₂. In addition, a peak at 1102 cm⁻¹ can be identified, originating from the H₂N-NH₂ characteristic signal.⁵¹ Given that there is no hydrazine formation in the product (Fig. S27†), we thus speculate that the photocatalytic N₂ fixation process on BiVO₄/V_S-MoS₂ follows an associative alternating pathway (Fig. 4f). In this pathway, hydrogenation occurs alternatively on two N atoms, with the final step involving the cleavage of the N-N bond to generate the first NH₃ molecule, followed by the last hydrogenation and another NH₃ desorption.

4. Conclusions

In summary, we have successfully prepared an S-scheme heterojunction of BiVO₄/V_S-MoS₂ by coupling the N₂-activated component (V_S-MoS₂) and the water-oxidized module (BiVO₄) to facilitate the process of photocatalytic N₂ reduction. BiVO₄/V_S-MoS₂ exhibits efficient NH₃ synthesis activity using H₂O as a proton source under illumination (100 mW cm⁻²), with an NH₃ yield of 132.8 μmol g⁻¹ h⁻¹, nearly 7 times higher than that of pure V_S-MoS₂. Our experimental findings underscore the pivotal roles played by the V_S-MoS₂ nanosheet with sulfur-rich vacancies in enhancing N₂ adsorption and activation. Furthermore, the construction of the S-scheme heterojunction substantially elevates the driving force for H₂O oxidation and significantly improves charge separation within the system. Moreover, an alternate association pathway for N₂ photoreduction reaction is proposed according to the *in situ* DRIFTS characterization. This study presents an efficient example for achieving the N₂ conversion process under mild conditions.

Author contributions

Han-Ying Luo: data curation, formal analysis, validation. Zhao-Lei Liu: data curation, formal analysis, validation. Meng-Ran

Zhang: formal analysis, supervision. Yan-Fei Mu: validation, writing – original draft preparation, funding acquisition. Min Zhang: conceptualization, funding acquisition, supervision, validation, writing – reviewing and editing.

Conflicts of interest

There are no conflicts of interest to declare.

Acknowledgements

This work was financially supported by the National Natural Science Foundation of China (U21A20286 and 22305214), the National Key R&D Program of China (2022YFA1502902), the National Science Foundation of Tianjin City (17JCJQC43800), the Jiangsu Funding Program for Excellent Postdoctoral Talent, the 111 Project of China (D17003), and the Tianjin Research Innovation Project for Postgraduate Students (2022SKY178).

References

- 1 J. W. Erisman, M. A. Sutton, J. Galloway, Z. Klimont and W. Winiwarter, *Nat. Geosci.*, 2008, **1**, 636–639.
- 2 X.-B. Li, Z.-K. Xin, S.-G. Xia, X.-Y. Gao, C.-H. Tung and L.-Z. Wu, *Chem. Soc. Rev.*, 2020, **49**, 9028–9056.
- 3 Y. Yang, S.-Q. Wang, H. Wen, T. Ye, J. Chen, C.-P. Li and M. Du, *Angew. Chem., Int. Ed.*, 2019, **58**, 15362–15366.
- 4 S. L. Foster, S. I. P. Bakovic, R. D. Duda, S. Maheshwari, R. D. Milton, S. D. Minter, M. J. Janik, J. N. Renner and L. F. Greenlee, *Nat. Catal.*, 2018, **1**, 490–500.
- 5 X. Zheng, Y. Yan, X. Li, Y. Liu and Y. Yao, *J. Hazard. Mater.*, 2023, **446**, 130679.
- 6 S. Wang, Y. Wang, T. C. Zhang, X. Jia and S. Yuan, *Nanoscale*, 2023, **15**, 16219–16226.
- 7 Y. Yu, Y. Li, Y. Fang, L. Wen, B. Tu and Y. Huang, *Appl. Catal., B*, 2024, **340**, 123161.
- 8 Z. Zhao, H. Ren, D. Yang, Y. Han, J. Shi, K. An, Y. Chen, Y. Shi, W. Wang, J. Tan, X. Xin, Y. Zhang and Z. Jiang, *ACS Catal.*, 2021, **11**, 9986–9995.
- 9 S. H. W. Kok, J. Lee, W.-K. Chong, B.-J. Ng, X. Y. Kong, W.-J. Ong, S.-P. Chai and L.-L. Tan, *J. Alloys Compd.*, 2023, **952**, 170015.
- 10 J. Di, C. Chen, Y. Wu, Y. Zhao, C. Zhu, Y. Zhang, C. Wang, H. Chen, J. Xiong, M. Xu, J. Xia, J. Zhou, Y. Weng, L. Song, S. Li, W. Jiang and Z. Liu, *Adv. Mater.*, 2022, **34**, 2204959.
- 11 S. Zhang, Y. Zhao, R. Shi, C. Zhou, G. I. N. Waterhouse, Z. Wang, Y. Weng and T. Zhang, *Angew. Chem., Int. Ed.*, 2021, **60**, 2554–2560.
- 12 S. Wang, X. Hai, X. Ding, K. Chang, Y. Xiang, X. Meng, Z. Yang, H. Chen and J. Ye, *Adv. Mater.*, 2017, **29**, 1701774.
- 13 Z. Lu, H. Wang, Y. Tao, S. Zhu, W. Hao, X. Liu, Y. Min and J. Fan, *Nanoscale*, 2023, **15**, 14847–14857.
- 14 C. J. M. van der Ham, M. T. M. Koper and D. G. H. Hetterscheid, *Chem. Soc. Rev.*, 2014, **43**, 5183–5191.
- 15 Y. Xiong, B. Li, Y. Gu, T. Yan, Z. Ni, S. Li, J.-L. Zuo, J. Ma and Z. Jin, *Nat. Chem.*, 2023, **15**, 286–293.



- 16 S. Liu, Z. Teng, H. Liu, T. Wang, G. Wang, Q. Xu, X. Zhang, M. Jiang, C. Wang, W. Huang and H. Pang, *Angew. Chem., Int. Ed.*, 2022, **61**, e202207026.
- 17 Y. Lv, S.-W. Ke, Y. Gu, B. Tian, L. Tang, P. Ran, Y. Zhao, J. Ma, J.-L. Zuo and M. Ding, *Angew. Chem., Int. Ed.*, 2023, **62**, e202305246.
- 18 G. N. Schrauzer and T. D. Guth, *J. Am. Chem. Soc.*, 1977, **99**, 7189–7193.
- 19 S. Cao, N. Zhou, F. Gao, H. Chen and F. Jiang, *Appl. Catal., B*, 2017, **218**, 600–610.
- 20 L. Zhang, S. Hou, T. Wang, S. Liu, X. Gao, C. Wang and G. Wang, *Small*, 2022, **18**, 2202252.
- 21 S. Hu, Y. Li, F. Li, Z. Fan, H. Ma, W. Li and X. Kang, *ACS Sustainable Chem. Eng.*, 2016, **4**, 2269–2278.
- 22 Y. Zhao, Y. Zhao, G. I. N. Waterhouse, L. Zheng, X. Cao, F. Teng, L.-Z. Wu, C.-H. Tung, D. O'Hare and T. Zhang, *Adv. Mater.*, 2017, **29**, 1703828.
- 23 H. Su, L. Chen, Y. Chen, R. Si, Y. Wu, X. Wu, Z. Geng, W. Zhang and J. Zeng, *Angew. Chem., Int. Ed.*, 2020, **59**, 20411–20416.
- 24 G. Zhan, F. Quan, Y. Yao, S. Zhao, X. Liu, H. Gu, Y. Huang, X. Liu, F. Jia and L. Zhang, *Appl. Catal., B*, 2023, **323**, 122186.
- 25 Q. Li, Y. Shi, Z. Wang, C. Liu, J. Bi, J. C. Yu and L. Wu, *J. Colloid Interface Sci.*, 2023, **652**, 1568–1577.
- 26 T. He, Z. Zhao, R. Liu, X. Liu, B. Ni, Y. Wei, Y. Wu, W. Yuan, H. Peng, Z. Jiang and Y. Zhao, *J. Am. Chem. Soc.*, 2023, **145**, 6057–6066.
- 27 X. Dong, K. Wang, Z. Cui, X. Shi, Z. Wang and F. Dong, *Nano Res.*, 2023, **16**, 6679–6686.
- 28 Z. Ai, M. Huang, D. Shi, M. Yang, H. Hu, B. Zhang, Y. Shao, J. Shen, Y. Wu and X. Hao, *Appl. Catal., B*, 2022, **315**, 121577.
- 29 Y. Chen, M. Yu, G. Huang, Q. Chen and J. Bi, *Small*, 2022, **18**, 2205388.
- 30 C. Cheng, J. Zhang, B. Zhu, G. Liang, L. Zhang and J. Yu, *Angew. Chem., Int. Ed.*, 2023, **62**, e202218688.
- 31 E. Aslan, M. K. Gonce, M. Z. Yigit, A. Sarilmaz, E. Stathatos, F. Ozel, M. Can and I. H. Patir, *Appl. Catal., B*, 2017, **210**, 320–327.
- 32 Y. Zhang, J. Di, X. Zhu, M. Ji, C. Chen, Y. Liu, L. Li, T. Wei, H. Li and J. Xia, *Appl. Catal., B*, 2023, **323**, 122148.
- 33 Y. Wang, S. Wang, J. Gan, J. Shen, Z. Zhang, H. Zheng and X. Wang, *ACS Sustainable Chem. Eng.*, 2023, **11**, 1962–1973.
- 34 Y. Lu, Y. Yang, X. Fan, Y. Li, D. Zhou, B. Cai, L. Wang, K. Fan and K. Zhang, *Adv. Mater.*, 2022, **34**, 2108178.
- 35 A. Shan, X. Teng, Y. Zhang, P. Zhang, Y. Xu, C. Liu, H. Li, H. Ye and R. Wang, *Nano Energy*, 2022, **94**, 106913.
- 36 C. Liu, S. Mao, M. Shi, X. Hong, D. Wang, F. Wang, M. Xia and Q. Chen, *Chem. Eng. J.*, 2022, **449**, 137757.
- 37 X. Guo, E. Song, W. Zhao, S. Xu, W. Zhao, Y. Lei, Y. Fang, J. Liu and F. Huang, *Nat. Commun.*, 2022, **13**, 5954.
- 38 P. Xia, X. Pan, S. Jiang, J. Yu, B. He, P. M. Ismail, W. Bai, J. Yang, L. Yang, H. Zhang, M. Cheng, H. Li, Q. Zhang, C. Xiao and Y. Xie, *Adv. Mater.*, 2022, **34**, 2200563.
- 39 Z. Li, G. Gu, S. Hu, X. Zou and G. Wu, *Chin. J. Catal.*, 2019, **40**, 1178–1186.
- 40 X. Xu, X. Liu, J. Zhao, D. Wu, Y. Du, T. Yan, N. Zhang, X. Ren and Q. Wei, *J. Colloid Interface Sci.*, 2022, **606**, 1374–1379.
- 41 C. Zhao, X. Li, L. Yue, X. Ren, S. Yuan, Z. Zeng, X. Hu, Y. Wu and Y. He, *ACS Appl. Nano Mater.*, 2023, **6**, 15709–15720.
- 42 Y.-F. Mu, C. Zhang, M.-R. Zhang, W. Zhang, M. Zhang and T.-B. Lu, *ACS Appl. Mater. Interfaces*, 2021, **13**, 22314–22322.
- 43 R. Chen, D. Zhang, Z. Wang, D. Li, L. Zhang, X. Wang, F. Fan and C. Li, *J. Am. Chem. Soc.*, 2023, **145**, 4667–4674.
- 44 A. Kumar, M. Sharma, S. Sheoran, S. Jaiswal, A. Patra, S. Bhattacharya and V. Krishnan, *Nanoscale*, 2023, **15**, 11667–11680.
- 45 R. Chen, Z. Ren, Y. Liang, G. Zhang, T. Dittrich, R. Liu, Y. Liu, Y. Zhao, S. Pang, H. An, C. Ni, P. Zhou, K. Han, F. Fan and C. Li, *Nature*, 2022, **610**, 296–301.
- 46 M.-M. Shi, D. Bao, B.-R. Wulan, Y.-H. Li, Y.-F. Zhang, J.-M. Yan and Q. Jiang, *Adv. Mater.*, 2017, **29**, 1606550.
- 47 Z. Wang, J. Zhu, X. Zu, Y. Wu, S. Shang, P. Ling, P. Qiao, C. Liu, J. Hu, Y. Pan, J. Zhu, Y. Sun and Y. Xie, *Angew. Chem., Int. Ed.*, 2022, **61**, e202203249.
- 48 L. Luo, X. Han, K. Wang, Y. Xu, L. Xiong, J. Ma, Z. Guo and J. Tang, *Nat. Commun.*, 2023, **14**, 2690.
- 49 Y. Zhang, J. Di, X. Qian, M. Ji, Z. Tian, L. Ye, J. Zhao, S. Yin, H. Li and J. Xia, *Appl. Catal., B*, 2021, **299**, 120680.
- 50 Y. Bo, H. Wang, Y. Lin, T. Yang, R. Ye, Y. Li, C. Hu, P. Du, Y. Hu, Z. Liu, R. Long, C. Gao, B. Ye, L. Song, X. Wu and Y. Xiong, *Angew. Chem., Int. Ed.*, 2021, **60**, 16085–16092.
- 51 Y. Yao, S. Zhu, H. Wang, H. Li and M. Shao, *J. Am. Chem. Soc.*, 2018, **140**, 1496–1501.

