

Cite this: *Chem. Sci.*, 2022, 13, 10725

All publication charges for this article have been paid for by the Royal Society of Chemistry

# Rational design *via* dual-site aliovalent substitution leads to an outstanding IR nonlinear optical material with well-balanced comprehensive properties†

He-Di Yang,<sup>‡ab</sup> Mao-Yin Ran,<sup>‡ac</sup> Sheng-Hua Zhou,<sup>ac</sup> Xin-Tao Wu,<sup>id ad</sup> Hua Lin<sup>id \*ad</sup> and Qi-Long Zhu<sup>id \*ad</sup>

The acquisition of a non-centrosymmetric (NCS) structure and achieving a nice trade-off between a large energy gap ( $E_g > 3.5$  eV) and a strong second-harmonic generation (SHG) response ( $d_{\text{eff}} > 1.0 \times$  benchmark  $\text{AgGaS}_2$ ) are two formidable challenges in the design and development of infrared nonlinear optical (IR-NLO) candidates. In this work, a new quaternary NCS sulfide,  $\text{SrCdSiS}_4$ , has been rationally designed using the centrosymmetric  $\text{SrGa}_2\text{S}_4$  as the template *via* a dual-site aliovalent substitution strategy.  $\text{SrCdSiS}_4$  crystallizes in the orthorhombic space group  $Ama2$  (no. 40) and features a unique two-dimensional  $[\text{CdSiS}_4]^{2-}$  layer constructed from corner- and edge-sharing  $[\text{CdS}_4]$  and  $[\text{SiS}_4]$  basic building units (BBUs). Remarkably,  $\text{SrCdSiS}_4$  displays superior IR-NLO comprehensive performances, and this is the first report on an alkaline-earth metal-based IR-NLO material that breaks through the incompatibility between a large  $E_g$  ( $> 3.5$  eV) and a strong phase-matching  $d_{\text{eff}}$  ( $> 1.0 \times \text{AgGaS}_2$ ). In-depth mechanism explorations strongly demonstrate that the synergistic effect of distorted tetrahedral  $[\text{CdS}_4]$  and  $[\text{SiS}_4]$  BBUs is the main origin of the strong SHG effect and large birefringence. This work not only provides a high-performance IR-NLO candidate, but also offers a feasible chemical design strategy for constructing NCS structures.

Received 6th July 2022  
Accepted 23rd August 2022

DOI: 10.1039/d2sc03760b

rsc.li/chemical-science

## Introduction

The infrared (IR) laser occupies a more and more important position in military weapons, information storage, precision micromanufacturing and other scientific research.<sup>1–5</sup> Generally, the IR nonlinear optical (NLO) crystal is an integral part for IR laser generation, which requires a phase-matching (PM) feature, a broad energy gap ( $E_g$ ), a strong second-harmonic-generation (SHG) intensity ( $d_{\text{eff}}$ ), a large laser-induced damage threshold (LIDT), an appropriate birefringence ( $\Delta n$ ), and a favorable physical and chemical stability.<sup>6</sup> Chalcopyrite-type materials,  $\text{AgGaS}_2$ ,<sup>7</sup>  $\text{AgGaSe}_2$ ,<sup>8</sup> and  $\text{ZnGeP}_2$ ,<sup>9</sup> are commercially available and exhibit sufficient  $d_{\text{eff}}$  and wide transmittance in

the IR region. However, they still suffer from several fatal drawbacks, *e.g.* low LIDT of  $\text{AgGaS}_2$ , non-phase-matching (NPM) behavior of  $\text{AgGaSe}_2$  and unexpected multi-phonon absorption of  $\text{ZnGeP}_2$ , which hinder their further applications in far-IR regions and high-power lasers. Therefore, it is necessary and urgent to explore new IR-NLO candidates with excellent comprehensive properties.

A non-centrosymmetric (NCS) structure is the prerequisite for a NLO crystal, which is also the first challenge in the design and synthesis of IR-NLO candidates. In order to overcome this problem, various strategies have been developed in the past decades. Among them, chemical substitution is considered to be the simplest and most effective method.<sup>10,11</sup> On one hand, it can greatly improve the IR-NLO properties of chalcogenides, such as,  $\text{Li}_{0.6}\text{Ag}_{0.4}\text{GaS}_2$  ( $d_{\text{eff}} = 1.1 \times \text{AgGaS}_2$ ) *versus*  $\text{LiGaS}_2$  ( $d_{\text{eff}} = 0.4 \times \text{AgGaS}_2$ ),<sup>12</sup>  $\text{Cu}_5\text{Zn}_{0.5}\text{P}_2\text{S}_8$  ( $d_{\text{eff}} = 0.3 \times \text{AgGaS}_2$ ) *versus*  $\text{Cu}_3\text{PS}_4$  ( $d_{\text{eff}} = 0.03 \times \text{AgGaS}_2$ ),<sup>13</sup> and  $\text{Sr}_{1.3}\text{Pb}_{0.7}\text{GeSe}_4$  ( $d_{\text{eff}} = 16 \times \text{SiO}_2$ ) *versus*  $\text{Pb}_2\text{GeSe}_4$  ( $d_{\text{eff}} = 2 \times \text{SiO}_2$ ).<sup>14</sup> On the other hand, it can realize centrosymmetric (CS)-to-NCS structure evolution in the chalcogenide system. Some classic examples include CS  $\text{La}_2\text{CuInS}_5$  *versus* NCS  $\text{La}_2\text{CuSbS}_5$ ,<sup>15</sup> CS  $\text{Ba}_2\text{GaAsSe}_5$  *versus* NCS  $\text{Ba}_2\text{As}_2\text{Se}_5$ ,<sup>16</sup> CS  $\text{BaGa}_2\text{Se}_4$  *versus* NCS  $\text{K}_{0.38}\text{Ba}_{0.81}\text{Ga}_2\text{Se}_4$ ,<sup>17</sup> CS  $\text{K}_2\text{Sb}_4\text{S}_7$  *versus* NCS  $\text{K}_2\text{Ag}_3\text{Sb}_3\text{S}_7$ ,<sup>18</sup> CS  $\text{Rb}_4\text{Hg}_2\text{Ge}_2\text{S}_8$  *versus* NCS  $(\text{Na}_3\text{Rb})\text{Hg}_2\text{Ge}_2\text{S}_8$ ,<sup>19</sup> CS  $\text{SrGeO}_3$  *versus* NCS  $\text{SrGeOSe}_2$ ,<sup>20</sup> CS  $\text{SnBr}_2$

<sup>a</sup>State Key Laboratory of Structural Chemistry, Fujian Institute of Research on the Structure of Matter, Chinese Academy of Sciences, Fuzhou, Fujian 350002, China. E-mail: linhua@fjirsm.ac.cn; qlzhu@fjirsm.ac.cn

<sup>b</sup>College of Chemistry, Fuzhou University, Fujian 350002, China

<sup>c</sup>University of the Chinese Academy of Sciences, Beijing 100049, China

<sup>d</sup>Fujian Science & Technology Innovation Laboratory for Optoelectronic Information of China, Fuzhou 350002, China

† Electronic supplementary information (ESI) available. CCDC 2167628 and 2168083. For ESI and crystallographic data in CIF or other electronic format see <https://doi.org/10.1039/d2sc03760b>

‡ H. D. Yang and M. Y. Ran contributed equally to this work.



versus NCS  $\text{Sn}_7\text{Br}_{10}\text{S}_2$ ,<sup>21</sup> and CS  $\text{Rb}_4\text{P}_2\text{S}_6$  versus NCS  $\text{RbBiP}_2\text{S}_6$ .<sup>22</sup> Notably, compared to a large number of structural transformations achieved by single-site substitution mentioned above, examples of dual-site and multi-site substitution are rarely reported.<sup>23,24</sup>

Among the essential conditions for a promising IR-NLO candidate, a large  $E_g$  and a strong  $d_{\text{eff}}$  are not only the most vital factors but also the most challenging to achieve concurrently due to their incompatibility. Metal chalcogenides have been considered as promising candidates for IR-NLO materials, and nearly a thousand novel NLO-active chalcogenides have been discovered in the past few decades.<sup>25–31</sup> Unfortunately, there are only 6 PM chalcogenides that can meet the preferred requirement for a useful IR-NLO crystal, that is, a nice trade-off between a large  $E_g$  ( $> 3.5$  eV) and strong  $d_{\text{eff}}$  ( $> 1.0 \times$  benchmark  $\text{AgGaS}_2$ ), see Table S1 in the ESI for details.† As summarized in Table S1,† some useful information can be obtained as follows: (1) all of them are sulfides; (2) most significant structural features are two-dimensional (2D) or three-dimensional (3D) structures that are constructed from tetrahedral  $[\text{MS}_4]$  basic building units (BBUs) ( $M =$  metal elements); (3) the filled cations are mainly alkali metals (A) or polycations. Nevertheless, a similar example based on an alkaline-earth metal (AE) as a filled cation is still not reported to date.

Recently, our research focuses on the ternary  $\text{AE-M}^{\text{III}}\text{-Q}$  system ( $M^{\text{III}} =$  group IIIA metal Ga, In), hoping to obtain NCS chalcogenides. The tetrahedral  $[\text{M}^{\text{III}}\text{Q}_4]$  BBUs are the beneficial NLO-active units for achieving a large  $d_{\text{eff}}$ , while the introduction of AE elements into this system may have the additional advantage of enlarging the  $E_g$ , which may help to increase the LIDT once an IR-NLO crystal is obtained.<sup>32–34</sup> Our systematic exploratory efforts have led to the discovery of a known ternary sulfide in this family, namely,  $\text{SrGa}_2\text{S}_4$ .<sup>35</sup> It exhibits a unique 2D  $[\text{Ga}_2\text{S}_4]^{2-}$  layer that is constructed from common NLO-active  $[\text{GaS}_4]$  units and possesses a wide optical  $E_g$  (3.93 eV) and a large theoretical birefringence ( $\Delta n = 0.147@2050$  nm). Unfortunately, the CS space group of  $Fddd$  (no. 70) makes this sulfide NLO inert, that is, it does not display any SHG signal under laser irradiation. Inspired by the aforementioned chemical substitution strategy and detailed structural analysis, we are eager to realize the CS-to-NCS structural evolution *via* the replacement of two  $\text{Ga}^{\text{III}}$  sites by “ $M^{\text{I}} + M^{\text{V}}$ ” or “ $M^{\text{II}} + M^{\text{IV}}$ ” in such a 2D layer. We term this the “dual-site aliovalent substitution” strategy.

Guided by a dual-site aliovalent substitution strategy, a novel quaternary NCS sulfide  $\text{SrCdSiS}_4$  was successfully discovered herein. Remarkably,  $\text{SrCdSiS}_4$  exhibits the PM feature and excellent IR-NLO performances, including a strong  $d_{\text{eff}}$  ( $1.1 \times \text{AgGaS}_2$ ), wide  $E_g$  (3.61 eV), ultra-high LIDT ( $20.4 \times \text{AgGaS}_2$ ), broad transmission range (0.33–18.19  $\mu\text{m}$ ) and suitable  $\Delta n$  (0.158@2050 nm), which indicates that it is a promising candidate for IR-NLO materials and eliminates the disadvantageous factors of commercial chalcopyrite-type chalcogenides. Moreover,  $\text{SrCdSiS}_4$  is also the first example of an alkaline-earth metal-based IR-NLO material that breaks through the incompatibility between a large  $E_g$  ( $> 3.5$  eV) and a strong PM  $d_{\text{eff}}$  ( $> 1.0 \times \text{AgGaS}_2$ ). In this work, a systematic study of the syntheses,

structural evolution, NLO and linear optical properties, and the in-depth mechanism is reported as well.

## Results and discussion

In this study, light-yellow crystals of  $\text{SrCdSiS}_4$  were prepared by a high-temperature solid-state reaction between stoichiometric  $\text{SrS}$ ,  $\text{CdS}$ ,  $\text{Si}$ , and  $\text{S}$  at 1123 K using  $\text{CsI}$  as the flux. The purity of the polycrystalline sample was checked by powder X-ray diffraction (XRD) analysis (Fig. 1a), and energy-dispersive X-ray spectroscopy (EDX) provides average atomic ratios of 1.09/1.10/1/4.13 for Sr, Cd, Si, and S elements (Fig. S1†), which are close to theoretical values determined from single-crystal XRD results. As shown in Fig. 1b,  $\text{SrCdSiS}_4$  exhibits desirable thermal stability below 1207 K under  $\text{N}_2$  condition and decomposes to  $\text{Sr}_2\text{SiS}_4$  and  $\text{CdS}$  at higher temperatures (Fig. S2†). The UV-vis and near-IR absorption spectra of  $\text{SrCdSiS}_4$  reveal an optical  $E_g$  of 3.61 eV (see Fig. 1c) based on the Kubelka Munk function,<sup>36</sup> which is not only keeping the advantage of the wide  $E_g$  of the parent compound  $\text{SrGa}_2\text{S}_4$  (3.93 eV, as plotted in Fig. S3†) but is also considerably wider than those of commercial IR-NLO materials  $\text{AgGaS}_2$  (2.56 eV),<sup>37</sup>  $\text{AgGaSe}_2$  (1.83 eV)<sup>38</sup> and  $\text{ZnGeP}_2$  (2.0 eV).<sup>39</sup> Notably, such an ultra-wide  $E_g$  of  $\text{SrCdSiS}_4$  can effectively avoid two- or multi-photon absorptions under the incident normal laser, which is helpful to obtain a high LIDT. In addition, the transmittance spectrum (Fig. 1d) recorded from a well-polished single crystal piece indicates that  $\text{SrCdSiS}_4$  exhibits a wide transparent window from 0.33  $\mu\text{m}$  (UV-vis region) to 18.19  $\mu\text{m}$  (far-IR region), which can cover two notable atmospheric windows (3–5  $\mu\text{m}$  and 8–12  $\mu\text{m}$ ). Remarkably, such a transparent range is wider than those of distinguished IR-NLO materials  $\text{AgGaS}_2$  (0.48–11.4  $\mu\text{m}$ ),<sup>39</sup>  $\text{ZnGeP}_2$  (0.74–12  $\mu\text{m}$ ),<sup>39</sup>  $\text{AgGaSe}_2$  (0.76–17  $\mu\text{m}$ )<sup>39</sup> and other recently reported IR-NLO candidates.<sup>40–45</sup>

The structural evolution from CS  $\text{SrGa}_2\text{S}_4$  to NCS  $\text{SrCdSiS}_4$  based on the dual-site aliovalent substitution strategy is illustrated in Fig. 2. Comparison of their structures shows that they belong to the same orthorhombic system and possess tetrahedral  $[\text{MS}_4]$  BBUs in their 2D layered structures. However, they still have several significantly different characteristics in their structures: (i)  $\text{SrCdSiS}_4$  crystallizes in the space group of  $Ama2$  (no. 40), while  $\text{SrGa}_2\text{S}_4$  adopts the space group of  $Fddd$  (no. 70), see Table 1 for details; (ii) the asymmetric unit of  $\text{SrCdSiS}_4$  has 6 crystallographically independent sites (*i.e.*, 1 Sr, 1 Cd, 1 Si, and 3 S atoms) and the  $Z$  value (number of molecules in a unit cell) is 4, which are different from those of  $\text{SrGa}_2\text{S}_4$  [9 unique sites, namely, 3 Sr, 2 Ga, and 4 S atoms) and  $Z = 32$ ], see Tables 1 and 2 for details; (iii) note that the repeated functional primitive, namely the 12-member-ring  $[\text{Cd}_3\text{Si}_3\text{S}_{16}]$  (including 3  $[\text{CdS}_4]$  and 3  $[\text{SiS}_4]$  BBUs, see the dashed part in Fig. 2f) exists in each 2D  $[\text{CdSiS}_4]^{2-}$  layer of  $\text{SrCdSiS}_4$  (Fig. 2d), but in  $\text{SrGa}_2\text{S}_4$ , the repeated 12-member-ring  $[\text{Ga}_6\text{S}_{16}]$  in each 2D  $[\text{Ga}_2\text{S}_4]^{2-}$  layer consists of 2  $[\text{Ga}(1)\text{S}_4]$  and 4  $[\text{Ga}(2)\text{S}_4]$  BBUs (see the dashed part in Fig. 2c and e); (iv) the  $[\text{SrS}_8]$  polyhedra are more highly distorted in  $\text{SrCdSiS}_4$  than those in  $\text{SrGa}_2\text{S}_4$ , *e.g.*, the larger difference ( $\Delta d$ ) between the Sr–S bonds ( $\Delta d$  (Sr–S) = 0.15 Å) in  $\text{SrCdSiS}_4$  than that (0.03 Å) in  $\text{SrGa}_2\text{S}_4$ , and a similar trend also



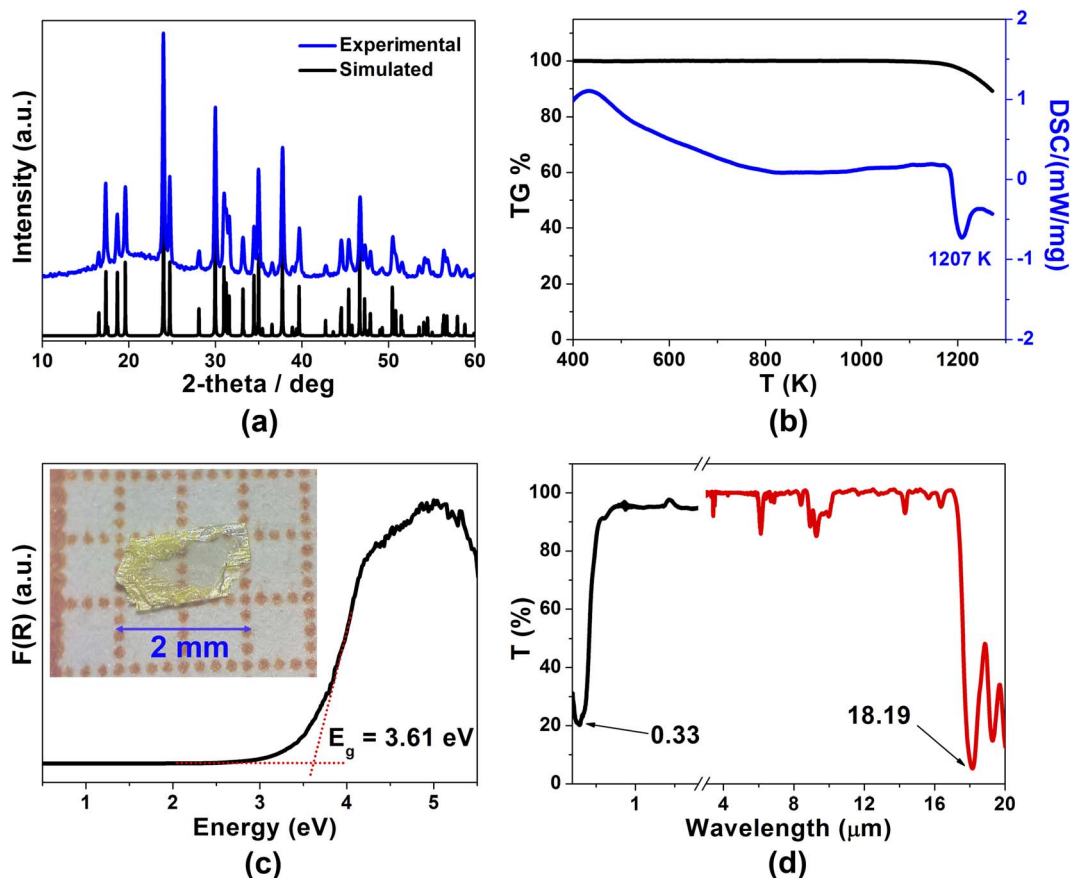


Fig. 1 Experimental characterization results of SrCdSi<sub>4</sub>: (a) experimental (blue) and simulated (black) powder XRD curves; (b) TG and DSC diagrams; (c) solid-state UV-vis-NIR diffuse reflectance spectrum (inset: photograph of a polished crystal); (d) optical transmittance spectra from UV-vis to IR region.

occurred in the tetrahedral [MS<sub>4</sub>] BBUs, see Fig. S4 and Tables S2 and S3 for details.<sup>†</sup> In a word, the dual-site aliovalent substitution led to the above-mentioned obvious changes in their crystal structures, thus realizing the CS-to-NCS structural transformation from ternary SrGa<sub>2</sub>Si<sub>4</sub> to quaternary SrCdSi<sub>4</sub>. Moreover, the detailed symmetric operation change shown in Fig. 2g and h clearly displays the evolution of symmetry breaking, that is, the loss of the different glide planes and the inversion centre from CS SrGa<sub>2</sub>Si<sub>4</sub> [high symmetry *Fddd* (no. 70)] to NCS SrCdSi<sub>4</sub> [low symmetry *Ama2* (no. 40)].

Owing to SrCdSi<sub>4</sub> possessing the NCS polar structure, we exhaustively investigated and analyzed the NLO performance. Size-dependent SHG effect measurements were performed by using the Kurtz-Perry method<sup>46</sup> at five different particle size ranges. As illustrated in Fig. 3a, the SHG intensity strength increases with the increase of particle size, indicating that SrCdSi<sub>4</sub> can achieve type-I PM in the IR region. Under the same particle size of 150–210 μm, the  $d_{\text{eff}}$  is around 1.1 times that of AgGaS<sub>2</sub> under a 2050 nm Q-switched laser. We also measured the SHG signals under a 1064 nm laser due to the shorter UV absorption edge of SrCdSi<sub>4</sub> (*ca.* 330 nm), giving it the potential to be applied in the UV-vis range. As indicated in Fig. 3b, SrCdSi<sub>4</sub> shows a large SHG effect of  $4.5 \times \text{KH}_2\text{PO}_4$  (KDP) with type-I PM nature. Therefore, SrCdSi<sub>4</sub> is an excellent dual-band

NLO candidate that can be used in both the IR and UV-vis regions. Apart from an adequate SHG response, a large LIDT is also vitally important for an IR-NLO material. So, its LIDT was measured by a single-pulse power technology.<sup>47</sup> As shown in Fig. S5,<sup>†</sup> the experimental LIDT of SrCdSi<sub>4</sub> of 57.14 MW cm<sup>-2</sup> in the particle size range of 150–210 μm is around 20.4 times higher than that of benchmark AgGaS<sub>2</sub> (2.8 MW cm<sup>-2</sup>) under the same condition (1064 nm, 1 Hz, 10 ns). Such a value shows the outstanding laser tolerance of SrCdSi<sub>4</sub>, indicating its potential in high-power laser applications. As a new member of the XM<sup>II</sup>M<sup>IV</sup>Q<sub>4</sub> (X = Eu, Sr, Ba; M<sup>II</sup> = Mn, Zn, Cd, and Hg; M<sup>IV</sup> = group-14 elements; and Q = chalcogen) system,<sup>24,48–63</sup> it is necessary to make a detailed comparison with other compounds. A summary of the two key performance parameters (*i.e.*,  $d_{\text{eff}}$  and  $E_{\text{g}}$ ) of the XM<sup>II</sup>M<sup>IV</sup>Q<sub>4</sub> family is provided in Fig. 4 and details are listed in Table S4.<sup>†</sup> Remarkably, SrCdSi<sub>4</sub> displays superior IR-NLO comprehensive performances, and this is the first report on an alkaline-earth metal-based IR-NLO material that breaks through the incompatibility between a large  $E_{\text{g}}$  (>3.5 eV) and a strong phase-matching  $d_{\text{eff}}$  (>1.0 × AgGaS<sub>2</sub>) in this system. Furthermore, a more comparative study with other state-of-the-art IR-NLO candidates is worthwhile.<sup>32,64–67</sup> As shown in Fig. S6 and Table S1,<sup>†</sup> there are 7 PM IR-NLO chalcogenides with  $E_{\text{g}} > 3.5$  eV and  $d_{\text{eff}} > 1.0 \times \text{AgGaS}_2$



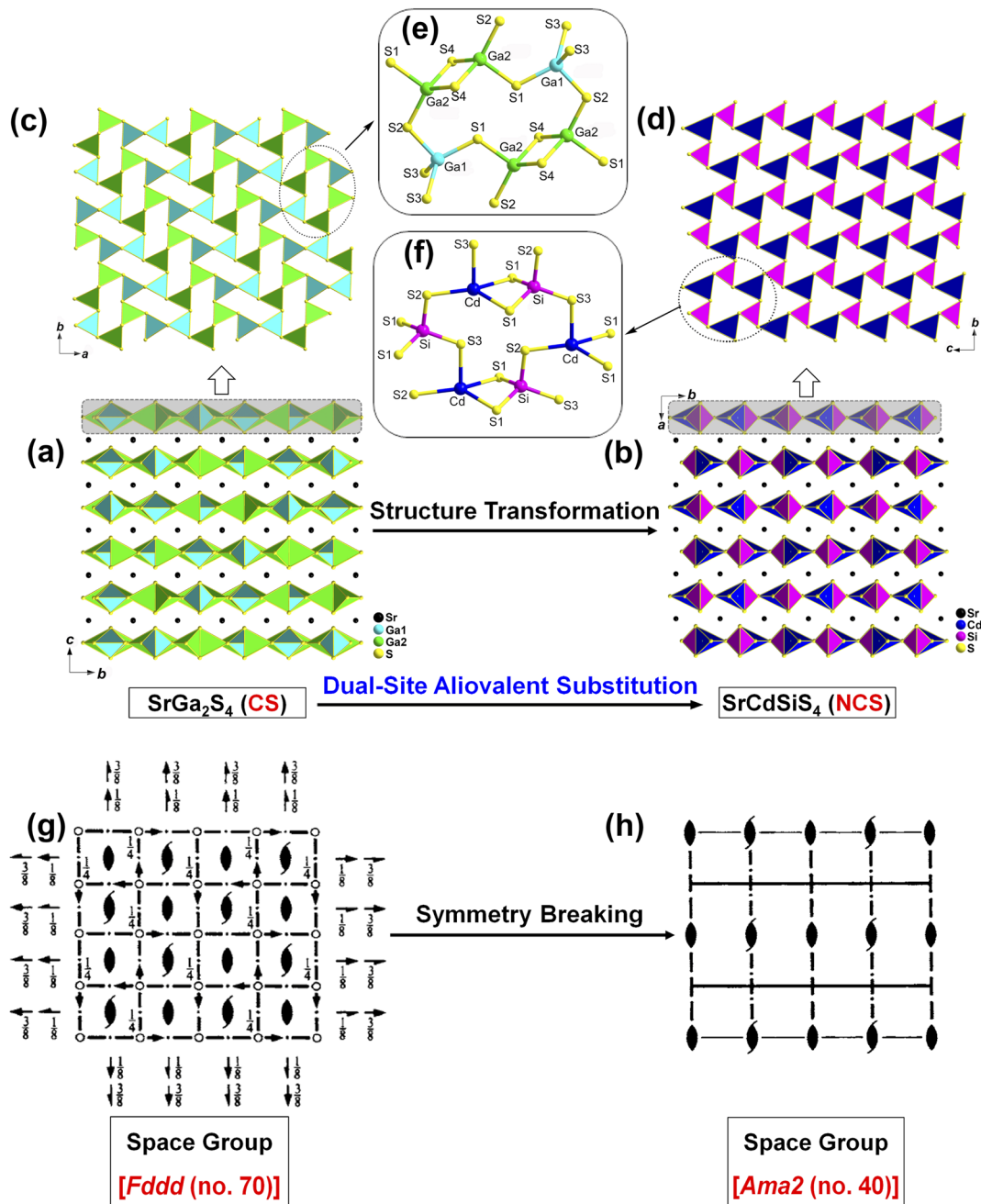


Fig. 2 Structural evolution from CS  $\text{SrGa}_2\text{S}_4$  to NCS  $\text{SrCdSi}_4$ : (a and b) view of the crystal structure of  $\text{SrGa}_2\text{S}_4$  and  $\text{SrCdSi}_4$  along the  $bc$  and  $ab$  planes, respectively; (c and d) 2D tetrahedral-stacking  $[\text{Ga}(1)\text{Ga}(2)\text{S}_4]^{2-}$  and  $[\text{CdSi}_4]^{2-}$  layers via edge- and corner-sharing with the repeat 12-member rings (dashed part) marked viewed along the  $ab$  and  $bc$  planes, respectively; (e and f) the ball-and-stick models of the 12-member-ring  $[\text{Ga}(1)_2\text{Ga}(2)_4\text{S}_{16}]$  and  $[\text{Cd}_3\text{Si}_3\text{S}_{16}]$  functional primitives with the atom number marked; (g and h) spatial symmetry operation change from CS [high symmetry  $Fddd$  (no. 70)] to NCS [low symmetry  $Ama2$  (no. 40)].

(Fig. S6a†), which have been selected on the basis of literature research. From the perspective of structural dimension, they are mainly constructed in 3D framework (43%) and 2D layered (43%) structures, and only  $\text{K}_2\text{BaP}_2\text{S}_6$ <sup>67</sup> possess a zero-dimensional (0D) cluster structure (14%) (Fig. S6b†). In addition, they can be divided into four categories according to the kind of filled cation: polycation-based (43%), alkali-metal-based (29%), mixed-cation-based (14%) and alkaline-earth-metal-

based (14%) (Fig. S6c†). Note that the central atoms in most of the BBUs are main group elements [e.g., Ga (20%), P (20%), Si (13%), Li (13%) and Ge (7%)] and transition metal elements [e.g., Zn (20%), and Cd (7%)] (Fig. S6d†). The production of  $\text{SrCdSi}_4$  not only enlarges the proportion of Cd and Si acting as favorable framework cations but also represents the first report of an alkaline-earth metal-based IR-NLO material that breaks through the wall of  $E_g > 3.5$  eV and  $d_{\text{eff}} > 1 \times \text{AgGaS}_2$ .



**Table 1** Crystallographic data and refinement details for SrCdSi<sub>4</sub> and SrGa<sub>2</sub>S<sub>4</sub>

Empirical formula	SrCdSi <sub>4</sub>	SrGa <sub>2</sub> S <sub>4</sub>
Formula weight	356.35	355.30
Temperature (K)	293(2)	293(2)
Crystal system	Orthorhombic	Orthorhombic
Space group	<i>Ama2</i> (no. 40)	<i>Fddd</i> (no. 70)
<i>a</i> (Å)	10.2821(7)	12.2216(5)
<i>b</i> (Å)	10.1551(9)	20.5008(9)
<i>c</i> (Å)	6.3699(5)	20.8426(10)
<i>V</i> (Å <sup>3</sup> )	665.12(9)	5222.2(4)
<i>Z</i>	4	32
<i>D<sub>c</sub></i> (g cm <sup>-3</sup> )	3.559	3.615
<i>μ</i> (mm <sup>-1</sup> )	12.520	17.481
GOOF on <i>F</i> <sup>2</sup>	1.204	1.174
<i>R</i> <sub>1</sub> , <i>wR</i> <sub>2</sub> ( <i>I</i> > 2σ( <i>I</i> )) <sup>a</sup>	0.0212, 0.0533	0.0228, 0.0591
<i>R</i> <sub>1</sub> , <i>wR</i> <sub>2</sub> (all data)	0.0217, 0.0594	0.0338, 0.0627
Largest diff. peak and hole (e Å <sup>-3</sup> )	0.81, -0.49	0.69, -1.03
Flack parameter	0.007(12)	

$$^a R_1 = \frac{\sum |F_o| - |F_c|}{\sum |F_o|}, wR_2 = \left[ \frac{\sum w(F_o^2 - F_c^2)^2}{\sum w(F_o^2)^2} \right]^{1/2}$$

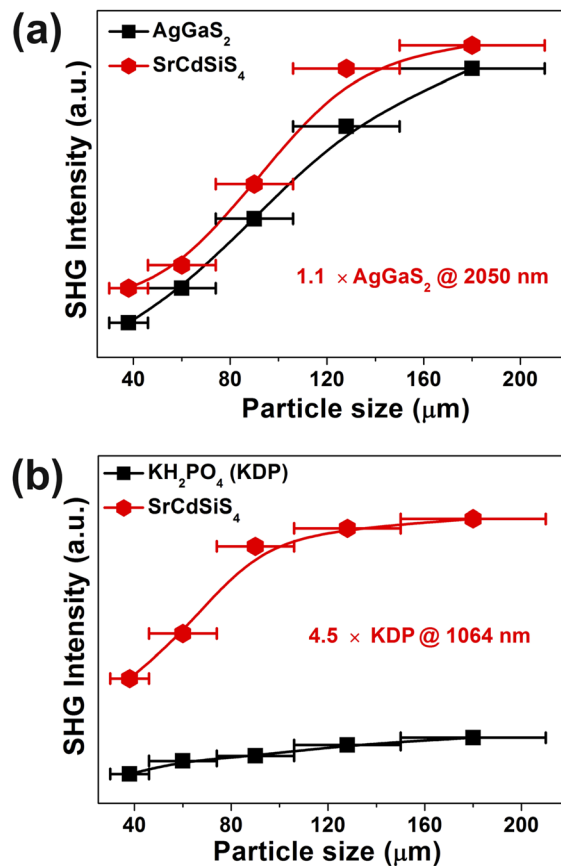
**Table 2** Atomic coordinates and equivalent isotropic displacement parameters (Å<sup>2</sup>) of SrCdSi<sub>4</sub> and SrGa<sub>2</sub>S<sub>4</sub>

Atom	Wyckff	<i>X</i>	<i>y</i>	<i>Z</i>	<i>U<sub>eq</sub></i> (Å) <sup>a</sup>
<b>SrCdSi<sub>4</sub></b>					
Sr	4 <i>a</i>	0	0	0	0.0165(4)
Cd	4 <i>b</i>	0.75	0.83086(9)	0.4430(2)	0.0251(4)
Si	4 <i>b</i>	0.75	0.7178(3)	0.9879(5)	0.0124(7)
S1	8 <i>c</i>	0.4166(2)	0.7754(2)	0.6887(3)	0.0165(5)
S2	4 <i>b</i>	0.75	0.8972(3)	0.8119(4)	0.0150(7)
S3	4 <i>b</i>	0.75	0.5662(2)	0.7620(5)	0.0171(7)

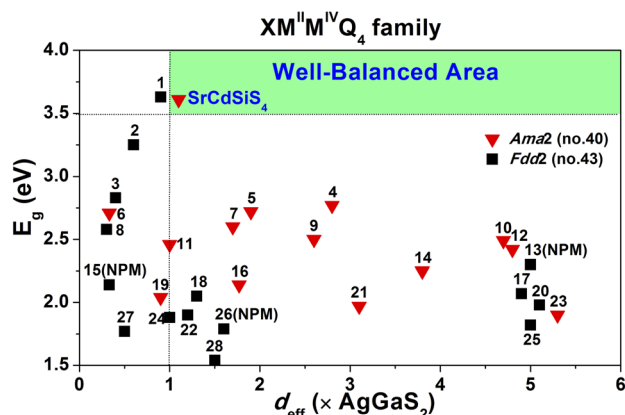
<b>SrGa<sub>2</sub>S<sub>4</sub></b>					
Sr1	16 <i>g</i>	0.875	0.375	0.62685(2)	0.01154(9)
Sr2	8 <i>b</i>	0.625	0.625	0.625	0.0111(2)
Sr3	8 <i>a</i>	0.375	0.375	0.875	0.01104(2)
Ga1	32 <i>h</i>	0.37362(2)	0.51258(2)	0.74942(2)	0.00896(6)
Ga2	32 <i>h</i>	0.66403(2)	0.44634(2)	0.74984(2)	0.00917(6)
S1	32 <i>h</i>	0.49464(4)	0.59489(3)	0.74888(4)	0.01004(9)
S2	32 <i>h</i>	0.48491(4)	0.42197(2)	0.74897(3)	0.00918(9)
S3	32 <i>h</i>	0.25159(6)	0.50054(3)	0.83226(3)	0.0099(9)
S4	32 <i>h</i>	0.75071(7)	0.49983(3)	0.66738(3)	0.00993(9)

<sup>a</sup> *U<sub>eq</sub>* is defined as one third of the trace of the orthogonalized *U<sub>ij</sub>* tensor.

Theoretical computations were adopted to better understand the structure–activity relationships of the title compound. According to the electronic structures, the valence band minimum (VBM) and the conduction band maximum (CBM) are at different *k*-points for SrGa<sub>2</sub>S<sub>4</sub> (Fig. 5a) and SrCdSi<sub>4</sub> (Fig. 5b), which indicates that they are indirect *E<sub>g</sub>* semiconductors. Theoretical results exhibit that the calculated *E<sub>g</sub>* values are 2.85 eV for SrGa<sub>2</sub>S<sub>4</sub> and 2.77 eV for SrCdSi<sub>4</sub>. Such values are smaller than the experimental ones (3.93 eV for SrGa<sub>2</sub>S<sub>4</sub> and 3.61 eV for SrCdSi<sub>4</sub>, respectively), which is mainly due to the discontinuity of the exchange correlation energy of the GGA functional.<sup>68–70</sup> In addition, their partial density of states (PDOSs) in the energy field from –10 to 10 eV are shown



**Fig. 3** Phase-matching curves for SrCdSi<sub>4</sub> and inserted values are the SHG intensities in the particle size range of 150–210 μm: (a) AgGaS<sub>2</sub> as the benchmark under 2050 nm radiation; (b) KDP as the reference under 1064 nm radiation. The solid curves are drawn as a guide to the eye.



**Fig. 4** Comparison of *d<sub>eff</sub>* and *E<sub>g</sub>* of the XM<sup>II</sup>M<sup>IV</sup>Q<sub>4</sub> system (*X* = Eu, Sr, Ba; *M*<sup>II</sup> = Mn, Zn, Cd, and Hg; *M*<sup>IV</sup> = group-14 elements; and *Q* = chalcogen) and the green shaded region represents the well-balanced (*E<sub>g</sub>* > 3.5 eV and *d<sub>eff</sub>* > 1.0 × AgGaS<sub>2</sub>) area for IR-NLO materials. Details are listed in Table S4.†

in Fig. 5c and d. From the PDOSs, it is found that the contribution in the VBM is mainly from Ga-4s, S-3p orbitals for SrGa<sub>2</sub>S<sub>4</sub> and S-3p, Si-3p orbitals for SrCdSi<sub>4</sub>, while the CBM consists of Ga-4p, S-3p orbitals for SrGa<sub>2</sub>S<sub>4</sub> and Cd-5s, S-3p



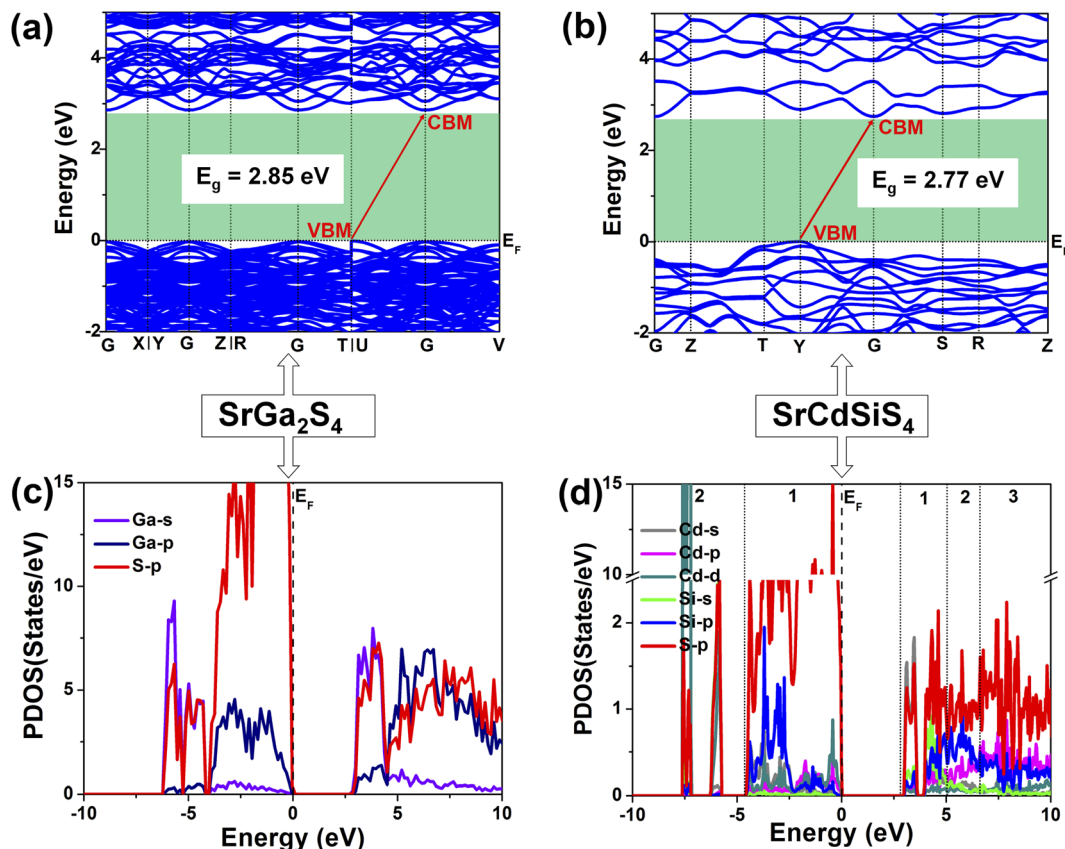


Fig. 5 Changes in (a and b) electronic band structures and (c and d) PDOSs (states with less contributions are omitted for better view) from SrGa<sub>2</sub>S<sub>4</sub> to SrCdSiS<sub>4</sub> caused by dual-site aliovalent substitution.

orbitals for SrCdSiS<sub>4</sub>. The main source of contribution transformed from Ga-4s to Si-3p for the VB and Ga-4p to Cd-5s for the CB after dual-site aliovalent substitution. Accordingly, diverse orbital states finally account for the little difference in optical  $E_g$  and the electron transfer mainly depends on [GaS<sub>4</sub>] (for SrGa<sub>2</sub>S<sub>4</sub>) converting to [CdS<sub>4</sub>] and [SiS<sub>4</sub>] (for SrCdSiS<sub>4</sub>). Moreover, the origin of the SHG response and birefringence ( $\Delta n$ ) as the two important NLO indexes were also analyzed in detail. As seen from Fig. 6a, SrCdSiS<sub>4</sub> has three nonzero independent second-order susceptibility tensors based on the rule of Kleinman's symmetry,<sup>71</sup> namely,  $d_{33}$ ,  $d_{24}$  and  $d_{15}$ . The calculated values at 2050 nm are  $d_{33} = 18.52$ ,  $d_{24} = 9.99$ , and  $d_{15} = 4.23$  pm V<sup>-1</sup>, respectively. In general, a larger  $E_g$  is usually accompanied by a smaller  $d_{\text{eff}}$ , but the largest one is 1.4 times that of reference AgGaS<sub>2</sub> ( $d_{14} = 13.6$  pm V<sup>-1</sup> at 2050 nm), which is basically consistent with experimental results (about 1.1 times that of AgGaS<sub>2</sub>). In addition, the theoretical static  $\Delta n$  values for SrCdSiS<sub>4</sub> are 0.158@2050 nm and 0.165@1064 nm, which are higher than those of SrGa<sub>2</sub>S<sub>4</sub> ( $\Delta n = 0.147$ @2050 nm and 0.153@1064 nm) and sufficiently large to ensure PM features in both UV-vis and IR regions (Fig. 6b). Typically, a significant anisotropic structure is beneficial to produce a large  $\Delta n$ , that is, dual-site aliovalent substitution induces greater structural distortion from SrGa<sub>2</sub>S<sub>4</sub> to SrCdSiS<sub>4</sub>. Meanwhile, these calculated values are larger than those of commercialized NLO

materials, such as AgGaS<sub>2</sub> ( $\Delta n = 0.039$ @2050 nm),<sup>72</sup> ZnGeP<sub>2</sub> ( $\Delta n = 0.04$ @2050 nm)<sup>72</sup> and KDP ( $\Delta n = 0.034$ @1064 nm).<sup>20</sup> Besides, the frequency-dependent refractive index diagrams mean that

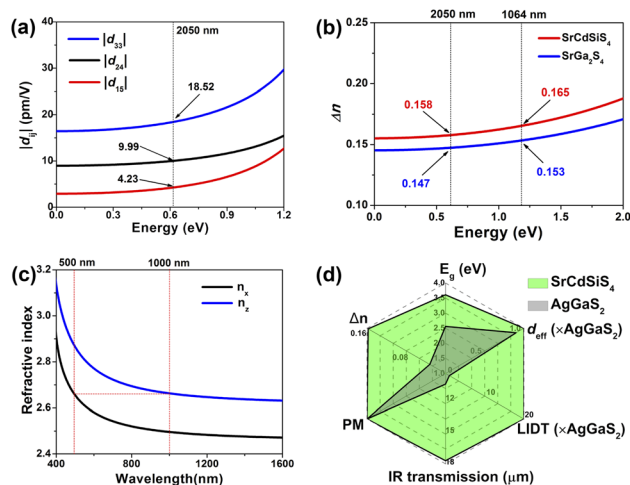


Fig. 6 (a) Frequency-dependent SHG coefficients of SrCdSiS<sub>4</sub>; (b) curves of the calculated  $\Delta n$  as a function of energy (eV) for SrGa<sub>2</sub>S<sub>4</sub> and SrCdSiS<sub>4</sub>; (c) calculated refractive index dispersion curves with the shortest PM cut-off edge at 500 nm; (d) radar chart with six directions (representing  $E_g$ ,  $d_{\text{eff}}$ , LIDT, IR transmission, PM,  $\Delta n$ ) to characterize comprehensive IR-NLO performance of SrCdSiS<sub>4</sub>.



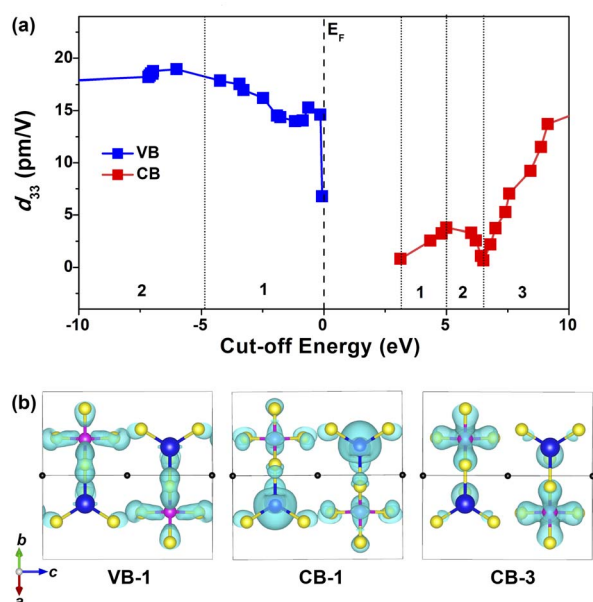


Fig. 7 Theoretical analysis of the SHG source for SrCdSi<sub>4</sub>: (a) variation of cut-off energy (eV) versus the largest static  $d_{33}$ ; (b) charge-density maps in the selections (VB-1, CB-1, CB-3) of major contribution. Black atoms: Sr; blue atoms: Cd; pink atoms: Si; yellow atoms: S.

under the premise of PM determined at 2050 nm, the lower limit of the SHG output wavelength is 500 nm (Fig. 6c). Based on theoretical studies and experimental observations, we compared SrCdSi<sub>4</sub> with the illustrious IR-NLO crystal AgGaS<sub>2</sub>. As illustrated in the radar chart (Fig. 6d), the green colored shadow is larger than the gray indicating the superior performance of SrCdSi<sub>4</sub>, including the PM feature, large  $E_g$  (ca. 3.61 eV), strong  $d_{\text{eff}}$  (ca.  $1.1 \times \text{AgGaS}_2$  at 2050 nm), giant LIDT ( $20.4 \times \text{AgGaS}_2$ ), beneficial  $\Delta n$  (0.158@2050 nm) and broad transparent region (0.33–18.19  $\mu\text{m}$ ).

Furthermore, the cut-off energy dependences of the largest static  $d_{33}$  were analyzed based on a length-gauge formalism method<sup>73,74</sup> with the purpose of revealing the intrinsic source of the SHG response. As shown in Fig. 7a,  $d_{33}$  values are trending upward in the range of VB-1 (dominated by the S-3p and Si-3p states, CB-1 (dominated by the S-3p and Cd-5s states) and CB-3 (dominated by the S-3p and Si-3p states). Distinctly, these three regions have a predominant impact on the overall NLO response. Considering the PDOS (Fig. 5d) and the relevant partial charge density profiles (Fig. 7b), the splendid SHG response comes from the collaborative effect of NLO-active [CdS<sub>4</sub>] and [SiS<sub>4</sub>] BBUs, i.e., the 2D [CdSi<sub>4</sub>]<sup>2-</sup> alternating arrangement layer.

## Conclusions

In conclusion, employing the ternary CS SrGa<sub>2</sub>S<sub>4</sub> as the parent structure, a new NCS quaternary SrCdSi<sub>4</sub> was successfully designed and synthesized *via* a dual-site aliovalent substitution strategy, whose 2D layered structure consisted of alternately connected [CdS<sub>4</sub>] and [SiS<sub>4</sub>] BBUs through corner- and edge-sharing S atoms. Detailed performance analyses indicated

that SrCdSi<sub>4</sub> could be a promising candidate for the UV-vis and IR-NLO crystal due to its advantages including a strong SHG intensity ( $d_{\text{eff}} = 4.5 \times \text{KDP}$  at 1064 nm, or  $1.1 \times \text{AgGaS}_2$  at 2050 nm) with PM feature, a suitable birefringence ( $\Delta n_{\text{(cal.)}} = 0.165$  at 1064 nm, or 0.158 at 2050 nm), a wide transmission window (0.33–18.19  $\mu\text{m}$ ), a large  $E_g$  (3.61 eV), and an ultra-high LIDT ( $20.4 \times \text{AgGaS}_2$ ). In addition, theoretical calculations reveal that the large  $\Delta n$  and strong  $d_{\text{eff}}$  are mainly contributed by the tetrahedral [CdS<sub>4</sub>] and [SiS<sub>4</sub>] NLO-active motifs that are nicely arranged in a most favorable stacking. Hopefully, such a simple and effective chemical design strategy can accelerate the discovery of novel NCS materials with advanced NLO properties.

## Data availability

Supporting data for this article is presented in the ESI.†

## Author contributions

Synthesis, characterization and original manuscript: H. D. Yang and M. Y. Ran; theoretical calculations: S. H. Zhou; experimental conception, supervision and manuscript editing: X. T. Wu, H. Lin and Q. L. Zhu. H. D. Yang and M. Y. Ran contributed equally to this work. All authors provided comments and approved the final version of the manuscript.

## Conflicts of interest

There are no conflicts to declare.

## Acknowledgements

This work was supported by the National Natural Science Foundation of China (22175175, 21771179 and 21901246), Fujian Science & Technology Innovation Laboratory for Optoelectronic Information of China (2021ZR118), the Natural Science Foundation of Fujian Province (2019J01133) and the Youth Innovation Promotion Association CAS (2022303). The authors thank Prof. Bing-Xuan Li at FJIRSM for helping with the NLO measurements and Prof. Yong-Fan Zhang at Fuzhou University for helping with the DFT calculations.

## Notes and references

- 1 F. J. Duarte, in *Tunable Laser Applications*, CRC Press, Boca Raton, FL, 2nd edn, 2008, ch. 2, pp. 9 and 12.
- 2 V. Petrov, *Prog. Quantum Electron.*, 2015, **44**, 1–106.
- 3 V. A. Serebryakov, E. V. Boiko, N. N. Petrishchev and A. V. Yan, *J. Opt. Technol.*, 2010, **77**, 6–17.
- 4 X.-T. Wu and L. Chen, Structure-Property Relationships in Nonlinear Optical Crystals II The IR Region, *Struct. Bonding*, 2012, **145**, 1–42.
- 5 N. L. B. Sayson, T. Bi, V. Ng, H. Pham, L. S. Trainor, H. G. L. Schwefel, S. Coen, M. Erkintalo and S. G. Murdoch, *Nat. Photonics*, 2019, **13**, 701–706.
- 6 L. Kang, M. Zhou, J. Yao, Z. Lin, Y. Wu and C. Chen, *J. Am. Chem. Soc.*, 2015, **137**, 13049–13059.



- 7 A. Harasaki and K. Kato, *Appl. Phys.*, 1997, **36**, 700–703.
- 8 G. C. Catella, L. R. Shiozawa, J. R. Hietanen, R. C. Eckardt, R. K. Route, R. S. Feigelson, D. G. Cooper and C. L. Marquardt, *Appl. Opt.*, 1993, **32**, 3948–3951.
- 9 G. D. Boyd, E. Buehler and F. G. Storz, *Appl. Phys. Lett.*, 1971, **18**, 301–304.
- 10 H. Lin, W.-B. Wei, H. Chen, X.-T. Wu and Q.-L. Zhu, *Coord. Chem. Rev.*, 2020, **406**, 213150.
- 11 G. Zou and K. M. Ok, *Chem. Sci.*, 2020, **11**, 5404–5409.
- 12 H. M. Zhou, L. Xiong, L. Chen and L. M. Wu, *Angew. Chem., Int. Ed.*, 2019, **58**, 9979–9983.
- 13 B. J. Song, Z. Ma, B. Li, X. T. Wu, H. Lin and Q. L. Zhu, *Inorg. Chem.*, 2021, **60**, 4357–4361.
- 14 L. T. Menezes, A. Assoud, W. Zhang, P. S. Halasyamani and H. Kleinke, *Inorg. Chem.*, 2020, **59**, 15028–15035.
- 15 H. Lin, Y.-Y. Li, M.-Y. Li, Z. Ma, L.-M. Wu, X.-T. Wu and Q.-L. Zhu, *J. Mater. Chem. C*, 2019, **7**, 4638–4643.
- 16 M.-M. Chen, Z. Ma, B.-X. Li, W.-B. Wei, X.-T. Wu, H. Lin and Q.-L. Zhu, *J. Mater. Chem. C*, 2021, **9**, 1156–1163.
- 17 Y. N. Li, Y. Chi, Z. D. Sun, H. Xue, N. T. Suen and S. P. Guo, *Chem. Commun.*, 2019, **55**, 13701–13704.
- 18 C. Liu, S. H. Zhou, Y. Xiao, C. Zhang, H. Lin and Y. Liu, *J. Mater. Chem. C*, 2021, **9**, 15407–15414.
- 19 C. Tang, W. Xing, F. Liang, M. Sun, J. Tang, Z. Lin, J. Yao, K. Chen, J. Wu, W. Yin and B. Kang, *J. Mater. Chem. C*, 2022, **10**, 3300–3306.
- 20 M.-Y. Ran, Z. Ma, H. Chen, B. Li, X.-T. Wu, H. Lin and Q.-L. Zhu, *Chem. Mater.*, 2020, **32**, 5890–5896.
- 21 X. H. Li, Z. H. Shi, M. Yang, W. Liu and S. P. Guo, *Angew. Chem., Int. Ed.*, 2022, **61**, e202115871.
- 22 M.-M. Chen, S.-H. Zhou, W. Wei, M.-Y. Ran, B. Li, X.-T. Wu, H. Lin and Q.-L. Zhu, *ACS Mater. Lett.*, 2022, **4**, 1264–1269.
- 23 Z.-X. Chen, Y.-N. Li, W.-D. Yao, W. Liu and S.-P. Guo, *J. Alloys Compd.*, 2022, **899**, 163255.
- 24 Y.-N. Li, Z.-X. Chen, W.-D. Yao, R.-L. Tang and S.-P. Guo, *J. Mater. Chem. C*, 2021, **9**, 8659–8665.
- 25 I. Chung and M. G. Kanatzidis, *Chem. Mater.*, 2013, **26**, 849–869.
- 26 S.-P. Guo, Y. Chi and G.-C. Guo, *Coord. Chem. Rev.*, 2017, **335**, 44–57.
- 27 K. Wu and S. Pan, *Coord. Chem. Rev.*, 2018, **377**, 191–208.
- 28 P. Gong, F. Liang, L. Kang, X. Chen, J. Qin, Y. Wu and Z. Lin, *Coord. Chem. Rev.*, 2019, **380**, 83–102.
- 29 W. Wang, D. Mei, F. Liang, J. Zhao, Y. Wu and Z. Lin, *Coord. Chem. Rev.*, 2020, **421**, 213444.
- 30 L. Gao, J. Huang, S. Guo, Z. Yang and S. Pan, *Coord. Chem. Rev.*, 2020, **421**, 213379.
- 31 H. Chen, W.-B. Wei, H. Lin and X.-T. Wu, *Coord. Chem. Rev.*, 2021, **448**, 214154.
- 32 H. Chen, Y.-Y. Li, B. Li, P.-F. Liu, H. Lin, Q.-L. Zhu and X.-T. Wu, *Chem. Mater.*, 2020, **32**, 8012–8019.
- 33 Y. Zhang, Q. Bian, H. Wu, H. Yu, Z. Hu, J. Wang and Y. Wu, *Angew. Chem., Int. Ed.*, 2021, **61**, e202115374.
- 34 Q. G. Yue, S. H. Zhou, B. Li, X. T. Wu, H. Lin and Q. L. Zhu, *Inorg. Chem.*, 2022, **61**, 1797–1804.
- 35 T. E. Peters and J. A. Baglio, *J. Electrochem. Soc.*, 1972, **119**, 230–236.
- 36 E. L. Simmons, *Appl. Opt.*, 1975, **14**, 1380–1386.
- 37 H. Lin, L. Chen, L. J. Zhou and L. M. Wu, *J. Am. Chem. Soc.*, 2013, **135**, 12914–12921.
- 38 M. C. Ohmer and R. Pandey, *MRS Bull.*, 1998, **23**, 16–22.
- 39 L. Bai, Z. Lin, Z. Wang, C. Chen and M. H. Lee, *J. Chem. Phys.*, 2004, **120**, 8772–8778.
- 40 M.-Y. Li, B. Li, H. Lin, Z. Ma, L.-M. Wu, X.-T. Wu and Q.-L. Zhu, *Chem. Mater.*, 2019, **31**, 6268–6275.
- 41 Y. Chu, P. Wang, H. Zeng, S. Cheng, X. Su, Z. Yang, J. Li and S. Pan, *Chem. Mater.*, 2021, **33**, 6514–6521.
- 42 J. Yao, D. Mei, L. Bai, Z. Lin, W. Yin, P. Fu and Y. Wu, *Inorg. Chem.*, 2010, **49**, 9212–9216.
- 43 M.-Y. Li, Z. Ma, B. Li, X.-T. Wu, H. Lin and Q.-L. Zhu, *Chem. Mater.*, 2020, **32**, 4331–4339.
- 44 Z. Li, S. Zhang, Z. Huang, L.-D. Zhao, E. Uykur, W. Xing, Z. Lin, J. Yao and Y. Wu, *Chem. Mater.*, 2020, **32**, 3288–3296.
- 45 M.-Y. Ran, S.-H. Zhou, B. Li, W. Wei, X.-T. Wu, H. Lin and Q.-L. Zhu, *Chem. Mater.*, 2022, **34**, 3853–3861.
- 46 S. K. Kurtz and T. T. Perry, *J. Appl. Phys.*, 1968, **39**, 3798–3813.
- 47 M.-J. Zhang, X.-M. Jiang, L.-J. Zhou and G.-C. Guo, *J. Mater. Chem. C*, 2013, **1**, 4754–4760.
- 48 Q. Q. Liu, X. Liu, L. M. Wu and L. Chen, *Angew. Chem., Int. Ed.*, 2022, **62**, e202205587.
- 49 Y. Zhang, D. Mei, Y. Yang, W. Cao, Y. Wu, J. Lu and Z. Lin, *J. Mater. Chem. C*, 2019, **7**, 8556–8561.
- 50 Y. Guo, F. Liang, Z. Li, W. Xing, Z. S. Lin, J. Yao, A. Mar and Y. Wu, *Inorg. Chem.*, 2019, **58**, 10390–10398.
- 51 W. Yin, A. K. Iyer, C. Li, J. Yao and A. Mar, *J. Alloys Compd.*, 2017, **708**, 414–421.
- 52 Y. Dou, Y. Chen, Z. Li, A. K. Iyer, B. Kang, W. Yin, J. Yao and A. Mar, *Cryst. Growth Des.*, 2019, **19**, 1206–1214.
- 53 Y.-J. Lin, R. Ye, L.-Q. Yang, X.-M. Jiang, B.-W. Liu, H.-Y. Zeng and G.-C. Guo, *Inorg. Chem. Front.*, 2019, **6**, 2365–2368.
- 54 W. Xing, N. Wang, Y. Guo, Z. Li, J. Tang, K. Kang, W. Yin, Z. Lin, J. Yao and B. Kang, *Dalton Trans.*, 2019, **48**, 17620–17625.
- 55 Y. Guo, F. Liang, W. Yin, Z. Li, X. Luo, Z.-S. Lin, J. Yao, A. Mar and Y. Wu, *Chem. Mater.*, 2019, **31**, 3034–3040.
- 56 N. Zhen, K. Wu, Y. Wang, Q. Li, W. Gao, D. Hou, Z. Yang, H. Jiang, Y. Dong and S. Pan, *Dalton Trans.*, 2016, **45**, 10681–10688.
- 57 X. Pang, R. Wang, X. Che and F. Huang, *J. Solid State Chem.*, 2021, **297**, 122092.
- 58 W. Xing, C. Tang, N. Wang, C. Li, Z. Li, J. Wu, Z. Lin, J. Yao, W. Yin and B. Kang, *Inorg. Chem.*, 2020, **59**, 18452–18460.
- 59 Y.-J. Lin, B.-W. Liu, R. Ye, X.-M. Jiang, L.-Q. Yang, H.-Y. Zeng and G.-C. Guo, *J. Mater. Chem. C*, 2019, **7**, 4459–4465.
- 60 M. Yan, Z.-D. Sun, W.-D. Yao, W. Zhou, W. Liu and S.-P. Guo, *Inorg. Chem. Front.*, 2020, **7**, 2451–2458.
- 61 F. Hou, D. Mei, Y. Zhang, F. Liang, J. Wang, J. Lu, Z. Lin and Y. Wu, *J. Alloys Compd.*, 2022, **904**, 163944.
- 62 K. Wu, X. Su, Z. Yang and S. Pan, *Dalton Trans.*, 2015, **44**, 19856–19864.
- 63 F.-Y. Yuan, C.-S. Lin, Y.-Z. Huang, H. Zhang, A.-Y. Zhou, G.-L. Chai and W.-D. Cheng, *J. Solid State Chem.*, 2021, **302**, 122352.
- 64 G. Li, Y. Chu and Z. Zhou, *Chem. Mater.*, 2018, **30**, 602–606.





- 65 J.-H. Zhang, D. J. Clark, J. A. Brant, K. A. Rosmus, P. Grima, J. W. Lekse, J. I. Jang and J. A. Aitken, *Chem. Mater.*, 2020, **32**, 8947–8955.
- 66 B. W. Liu, H. Y. Zeng, X. M. Jiang, G. E. Wang, S. F. Li, L. Xu and G. C. Guo, *Chem. Sci.*, 2016, **7**, 6273–6277.
- 67 V. Nguyen, B. Ji, K. Wu, B. Zhang and J. Wang, *Chem. Sci.*, 2022, **13**, 2640–2648.
- 68 K. Burke, *J. Chem. Phys.*, 2012, **136**, 150901.
- 69 K. Govaerts, R. Saniz, B. Partoens and D. Lamoen, *Phys. Rev. B: Condens. Matter Mater. Phys.*, 2013, **87**, 235210.
- 70 N. E. Christensen, A. Svane and E. L. Peltzer y Blancá, *Phys. Rev. B: Condens. Matter Mater. Phys.*, 2005, **72**, 014109.
- 71 D. A. Kleinman, *Phys. Rev.*, 1962, **126**, 1977–1979.
- 72 Z. Qian, Q. Bian, H. Wu, H. Yu, Z. Lin, Z. Hu, J. Wang and Y. Wu, *J. Mater. Chem. C*, 2022, **10**, 96–101.
- 73 C. Aversa and J. E. Sipe, *Phys. Rev. B: Condens. Matter Mater. Phys.*, 1995, **52**, 14636–14645.
- 74 S. N. Rashkeev, W. Lambrecht and B. Segall, *Phys. Rev. B: Condens. Matter Mater. Phys.*, 1998, **57**, 3905–3919.

