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Structural, electronic, and electrocatalytic evaluation of spinel transition metal sulfide supported reduced graphene oxide†

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Development of highly active and durable non-precious spinel transition metal sulfide (STMS)-based electrocatalysts plays a vital role in increasing the efficiency of hydrogen production *via* water electrolysis. Herein, we have synthesized a hierarchical nanostructured ZnCo₂S₄ on reduced graphene oxide (ZCS@rGO) sheet using a cost-effective hydrothermal synthesis method. The prepared ZCS@rGO shows improved hydrogen desorption and adsorption energy of the electrocatalyst surface towards efficient hydrogen evolution reaction (HER). As a result, ZCS@rGO showed lower HER overpotential ($\eta_{10} = 135$ eV) and Tafel slope (47 mV dec⁻¹) and superior durability at 10 mA cm⁻² for 36 h, as compared to the benchmark catalyst of Pt-C. Further, the electronic structure and HER mechanism of the ZCS@rGO catalyst were investigated by density functional theory calculations. This work provides a new pathway for the rational design of highly active and durable non-precious STMS-based electrocatalysts for hydrogen production.

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1. Introduction

With rising concerns in population, energy demands, climate change, and shortages in fossil fuel resources, researchers are in need to explore efficient green and renewable energy systems.^{1,2} Hydrogen is one of the promising green fuels in next generation green vehicles due to its high gravimetric energy density and zero carbon emission.^{3,4} The hydrogen evolution reaction (HER) is the cathode-cell reaction in the water electrolysis device. Developing an active electrocatalyst to minimize the HER overpotential becomes a critical task for improving the hydrogen production from water electrolysis.⁵

Precious metal catalysts of Pt/C and Pt-based alloys provide high HER efficiency, but their commercialization is highly restricted due to their scarcity, high price, sluggish reaction

kinetics, and insufficient durability.^{6–8} Therefore, it is vital to design and fabricate non-precious and low cost catalysts that are highly electroactive and durable for the hydrogen economy.⁹ In this regard, incredible effort has been expended to explore various non-precious metal catalysts such as transition metals,^{10–12} metal sulfides,^{13–16} phosphides,^{6,17,18} nitrides,^{19,20} and hydroxides.^{21,22} Therefore, non-precious metals have been required to increase the fundamental catalytic assets such as surface alteration and phase conversion, in addition to the construction of heterostructures^{23,24} as a strategy focused on regulating electron distribution to improve electrocatalytic activity.^{25,26}

Recently, transition metal sulfides have been one of the highest potential electrocatalysts used in energy storage and conversion devices owing to earth abundance, low cost, eco-friendliness, and superior electrocatalytic properties. Furthermore, the HER activity of transition metal sulfides (TMS) has been extensively studied due to their highly active sites, distinctive structural features, adjustable electronic properties, electrocatalytic activity, and low cost.^{27,28} Chemical state and electronic structure of the TMS's can determine the strength of reactant adsorption, reaction kinetics, and active intermediate formation towards HER. In particular, metallic sulfides exhibit excellent electrocatalytic activity towards HER due to their better redox-active reaction properties.²⁹ Nevertheless, many investigations have shown that metallic sulfides have lower electron conductivity and durability in energy conversion applications. In this context, the electron

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conductivity of metal sulfides could be improved by incorporating carbon nanofillers such as graphene oxide,^{30,31} carbon frameworks,³² and carbon nanofibers.³³ Among the various carbon conducting fillers, graphene sheets provide superb electron conductivity, large surface area, and prevent agglomeration of metal sulfides.^{34,35}

Among transition metal sulfides, cobalt-based binary or ternary sulfides such as CoS₂, MnCo₂S₄/NF, CuCo₂S₄, NiCo₂S₄, and FeCoMoS@N-rGO, have demonstrated promising electrocatalytic performance in OER/HER. It is crucial to identify the actual electrocatalytic active sites of transition metal sulfides for HER due to the presence of various stoichiometric ratios of metal ions with different oxidation states. Furthermore, the catalytic performance of an electrocatalyst could be drastically improved by tuning the morphology of the nanostructure, which can alter the electronic configuration and improve the active sites and area of the electrocatalyst. Based on the above inspiration, the fabrication of cobalt metal sulfides tailored with zinc components could be a promising new electrocatalyst for HER.³⁶

In this work, we have grown hierarchical nanostructures of ZnCo₂S₄ (ZCS) on reduced graphene oxide (rGO) sheet using a cost-effective hydrothermal reaction. The hierarchical nanostructure growth of ZnCo₂S₄ on rGO (ZCS@rGO) provided excellent electron conductivity, extended surface area with more metal sulfide active sites, and improved the synergistic effect between metal sulfides and rGO for HER. The optimized electrocatalyst of ZCS@rGO shows lower HER overpotential ($\eta_{10} = 135$ eV), Tafel slope (47 mV dec⁻¹), and superior durability at 10 mA cm⁻² for 36 h as compared to the prepared Co₃S₄@rGO (CS@rGO) catalyst. Further, we investigated the density of states (DOS) and HER mechanism of ZCS@rGO catalyst by density functional theory (DFT) calculations. Notably, the ZCS@rGO catalyst coated carbon cloth electrode exhibits excellent H₂ molecule evolution with a Faraday efficiency of about 93.74%. This present work provides a new pathway to the rational design of spinal TMS and rGO based electrocatalysts for efficient HER.

2. Methods

2.1 Chemicals

Graphite powder (Sigma Aldrich, 99.99% trace metals), zinc nitrate(II) hexahydrate (Yakuri Pure Chemicals, 95%), cobalt nitrate(II) hexahydrate (Alfa Aesar, 97.7%), urea (Alfa Aesar, 99.0–100.5%), sodium sulfide nonahydrate (Alfa Aesar, 98.0%), hydrazine monohydrate (Alfa Aesar, 98+%), ammonium fluoride (Sigma Aldrich, ≥98.0%), benchmark 20 wt% Pt/C (Alfa Aesar), 5% Nafion solution (Sigma-Aldrich), methanol, and potassium hydroxide were obtained from Samchun Pure Chemicals Co., South Korea.

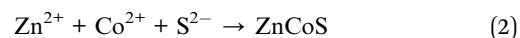
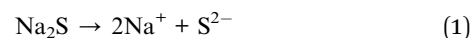
2.2 Synthetic procedure

2.2.1 Synthesis of ZnCo layered double hydroxides (LDH) @rGO. The graphene oxide was synthesized from graphite powder using a modified Hummers' method with a previously reported procedure.³⁷ 30 mg of graphene oxide (GO), 1 mmol of zinc nitrate, 2 mmol cobalt nitrate, and 4 mmol urea were added

to 20 mL of deionized (DI) water, and the mixture was sonicated for 15 min. Then, 2 mmol of ammonium fluoride was added slowly with constant stirring. Then, a sufficient amount of DI water was added to the 60 mL reaction mixture in an autoclave with continuous stirring, and the mixture was transferred into a hydrothermal reaction vessel and kept at 150 °C for 5 h. The system was permitted to cool to ambient temperature, and the product was collected, washed with DI water and ethanol several times, and dried at 60 °C overnight.³⁸

2.2.2 Synthesis of ZnCo₂S₄@rGO catalyst. 5 mL of N₂H₄ was added dropwise to a 6 mmol of Na₂S·9H₂O with constant stirring to obtain a homogeneous solution. Then, 20 mg of ZnCo LDH@rGO was dispersed in 20 mL of DI water in the hydrothermal vessel, and a homogenous mixture of N₂H₄ and Na₂S·9H₂O was added drop by drop. Furthermore, a sufficient amount of DI water was added to 60 mL of autoclave and hydrothermally treated for 4 h at 180 °C. The obtained product was collected and washed several times with DI water and ethanol. The final product was desiccated at 60 °C overnight.^{39,40}

Furthermore, the comparison material of the CS@rGO catalyst was prepared under identical synthesis conditions without adding zinc precursors in the hydrothermal process. The detailed synthesis procedure of CS@rGO is given in ESI†



2.3 Characterization techniques

Morphological examination of the obtained materials was performed by field emission scanning electron microscopy (FE-SEM; SUPRA 40 VP; Carl Zeiss, Germany) and high-resolution transmission electron microscopy (HR-TEM; JEM-2010, JEOL) at the Center for University Wide Research Facilities (CURF) at Jeonbuk National University (JBNU), South Korea. The crystalline nature of materials was investigated using X-ray diffraction (XRD) copper K α radiation ($\lambda = 0.154$ nm), Rigaku Corporation, Japan, and energy dispersive X-ray spectroscopy (EDS; Carl Zeiss, Germany, SUPRA 40 VP). The loading contents of Zn, Co, and S were evaluated by ICP-OES (inductively coupled plasma-optical emission spectrometry) with the iCAP 7000 series (Thermo Fisher Scientific) and the defective nature of materials was revealed by Raman spectroscopy (RAM-HR equipped with a 532 nm HORIBA-Lab helium–neon laser) at CURF, JBNU, Republic of Korea. Then, the binding nature and elemental composition of materials were examined with an X-ray photoelectron spectrometer (XPS; Axis-Nova, Kratos Inc.), and the surface area of the prepared catalyst was investigated using a Brunauer–Emmett–Teller (BET) Autosorb-iQ 2ST/MP physisorption analyzer at the Jeonju Center of the Korea Basic Science Institute (KBSI), Republic of Korea. The R-XAS based extended X-ray absorption fine structure (EXAFS) and X-ray absorption near edge structure (XANES) measurements at the Zn and Co K-edges were performed at the Sc-detector of

Chonnam National University (model R-XAS, Rigaku, Japan) with total electron yield detection.

2.4 Electrochemical characterization

The prepared electrocatalyst (~3 mg) was disseminated in 1 mL of DI water and ethanol (1 : 1) mixture. Further, 50 μL of 5% Nafion was added and sonicated for 60 min to generate a homogenous ink. The prepared ink was coated on carbon cloth (CC) (coated area $1 \times 1 \text{ cm}^2$) and dried at 60 $^\circ\text{C}$ overnight in a vacuum oven. For comparative study, commercial 20% Pt–C catalyst ink was prepared using a similar procedure.

The HER electrochemical performance of prepared electrocatalysts was examined by a distinctive three-electrode system using CC supported catalyst as the working electrode, graphite rod (counter electrode), and saturated Ag/AgCl (reference electrode). The linear sweep voltammetry (LSV) curve was measured with 1 mV s^{-1} (scan rate) and 1.0 M KOH electrolyte. The double-layer capacitance value of the prepared catalyst was evaluated using CV curves with a different scan rate of 20 to 100 mV s^{-1} in the non-faradaic region. The slope of current densities *vs.* scan rate was double the value of double-layer capacitance (C_{dl}). Electrochemical impedance spectra (EIS) of prepared catalysts were measured using potentiostatic impedance with onset potential and a frequency range of 100 kHz to 0.1 Hz in 1.0 M KOH solution. The long-term stability test was measured by chronopotentiometry at a current density of 10 mA cm^{-2} for 36 h. All recorded potentials were converted into reversible hydrogen electrodes (RHE) using the following equation

$$E_{\text{RHE}} = E_{\text{AgCl}} + (0.197 + 0.059 \times \text{pH}) \quad (3)$$

2.5 Computational analysis

The comprehensive density functional theory (DFT) calculation techniques are provided in the ESI.†

3. Results and discussion

3.1 Structural and chemical evaluation of STMS

The ZnCo layered double hydroxide on the rGO (ZnCo LDH@rGO) was synthesized using metal precursors of cobalt nitrate(II) hexahydrate and zinc nitrate(II) hexahydrate in the presence of GO solution through a hydrothermal method at 150 $^\circ\text{C}$ for 5 h as presented in Fig. 1a. The GO has functional groups that generate strong interactions between the metal precursors and the GO sheet. During the hydrothermal reaction, the absorbed metal precursor was uniformly grown as a hierarchical structured ZnCo LDH on rGO, and simultaneously, GO was converted to rGO due to the presence of urea. Here, ammonium fluoride was used as the structure tuning agent to control the formation of hierarchical ZnCo LDH@rGO.³⁶ Therefore, FE-SEM images of ZnCo LDH@rGO display the sphere-like hierarchical structure of ZnCo LDH grown on the rGO sheet, as shown in Fig. 1b–d.

The ZnCo LDH@rGO was successfully converted to zinc cobalt sulfide on rGO (ZCS@rGO) *via* hydrothermal reaction at 180 $^\circ\text{C}$ for 4 h with sodium sulfide as the sulfur source and hydrazine monohydrate as the reducing agent. Fig. 1e shows the agglomerated nanoparticles covered on hierarchical sphere-like structure of ZCS@rGO, which confirms the conversion of ZnCo LDH@rGO to ZCS@rGO during the hydrothermal method.

This hierarchical sphere-like structure of ZCS along with the larger surface area of rGO favors the electron/ion transport with facile absorption of electrolyte during HER activity.¹⁵ Elemental mapping images of SEM-EDS show the structure of ZCS on the rGO sheet and confirm the presence of C, Zn, Co, and S (Fig. 1f–i).

Moreover, the intrinsic morphology of the ZCS@rGO catalyst was analyzed by HR-TEM and STEM-HAADF as presented in Fig. 2a–f. The image shows that the lattice fringe distances of 0.33 and 0.27 nm belong to (111) and (220) planes of ZCS@rGO catalyst, respectively, which is in good agreement with the X-ray diffraction results (see Fig. 2g and S1†). The SAED pattern (Fig. 2b (inset image)) shows a ring with a bright spot revealing the polycrystalline nature of ZCS@rGO, which further confirms the (111), (220), and (311) planes. Additionally, Fast Fourier Transform (FFT) images also confirm the good crystalline nature of the ZCS@rGO catalyst (inset image of Fig. 2c). Subsequently, the EDS spectrum (Fig. 2d–f) of ZCS@rGO catalyst reflects the uniform distribution of Zn, Co, and S which confirms the formation of ZCS on the rGO sheet; Fig. S2† shows the EDX spectrum of ZCS@rGO catalyst and elemental composition of the ZCS@rGO catalyst are given in inset image of Fig. S2.† Fig. S3a–c† illustrates the TEM images of cobalt sulfides, which confirm the nanostructures were homogeneously grown on the rGO sheet. The HR-TEM image of Fig. S3b† confirms the lattice distances of CS@rGO to be 0.56 and 0.26 nm, corresponding to the (111) and (200) planes of CS@rGO catalyst, respectively. Furthermore, FFT images confirm the good crystalline nature of the CS@rGO catalyst. Additionally, STEM-HAADF and EDX spectra confirm the presence of Co and S elements on the rGO sheet (Fig. S3d–g†).

The crystalline states of ZCS@rGO and CS@rGO catalysts were examined by XRD, as presented in Fig. 2g. The XRD pattern displays characteristic CS@rGO catalyst peaks of 18.7 $^\circ$, 31.2 $^\circ$, 36.7 $^\circ$, and 55.5 $^\circ$ belonging to the (111), (311), (400), and (440) planes, respectively. These peaks are reliably a typical diffraction pattern for CS@rGO (cubic structure) (JCPDS – 073-1073). The XRD patterns of ZCS@rGO catalyst exhibited strong peaks at 29.4 $^\circ$, 47.5 $^\circ$, and 59.1 $^\circ$ belonging to the (111), (220), and (222) planes of the cubic ZCS (JCPDS – 047-1656), in agreement with previous literature reports.⁴¹ Notably, a broad peak appeared at $\sim 23.10^\circ$, which is related to the (002) plane of rGO in ZCS@rGO catalyst, which confirmed that GO was effectively transformed to rGO through the hydrothermal process.

Raman spectroscopy is a vital tool to investigate the defective and graphitic environment of ZCS@rGO and CS@rGO catalysts. Raman spectra of ZCS@rGO and CS@rGO catalysts, as revealed in Fig. 2h, showed distinctive peaks of the D band appearing in 1349 cm^{-1} ascribed to a defect nature of graphitic carbon, and G band appearing at 1580 cm^{-1} is associated with the sp^2 nature of carbon atoms in the catalyst. The intensity ratio of D and G

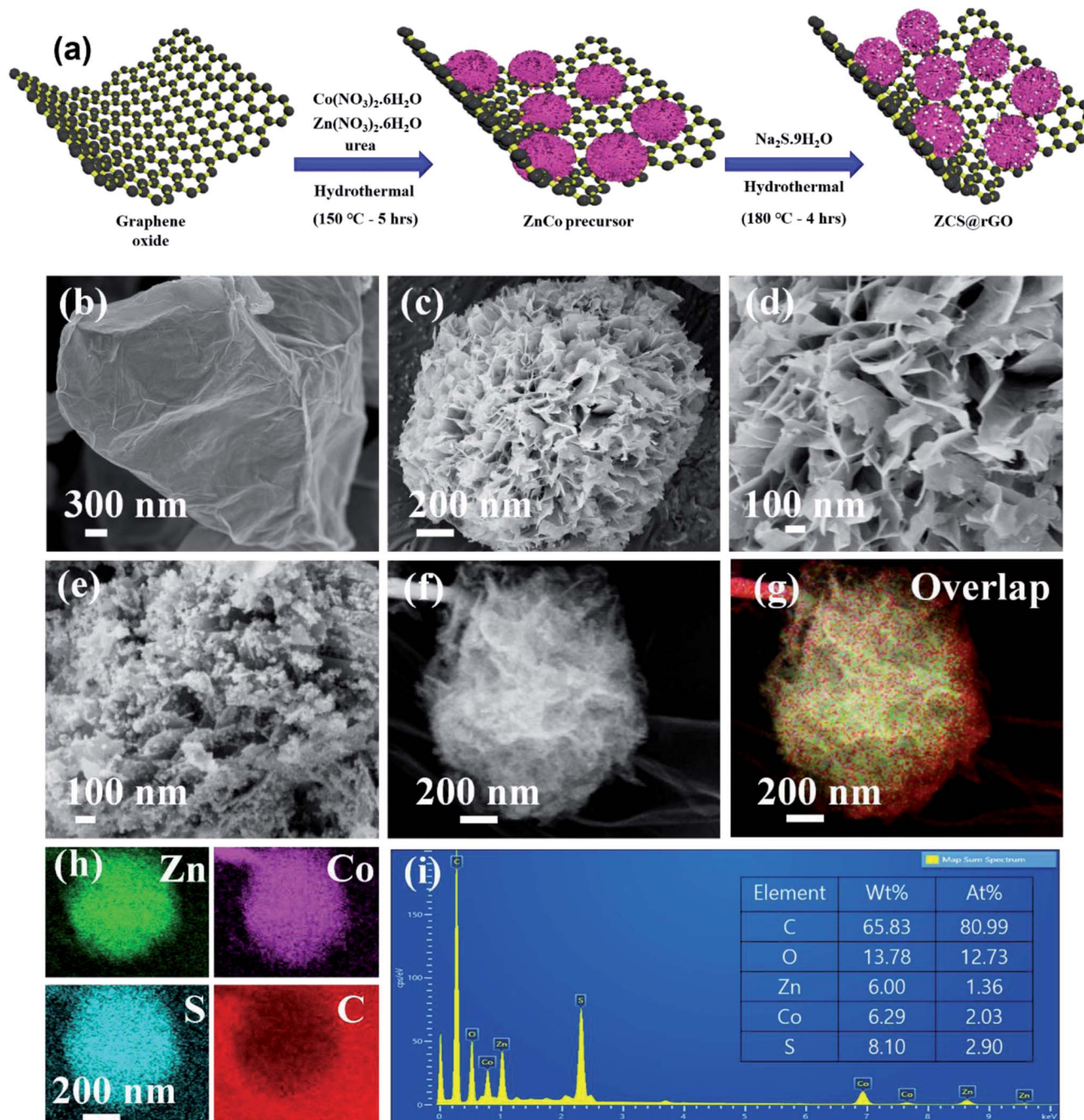


Fig. 1 (a) Schematic diagram for the synthesis of ZCS@rGO. The SEM images of (b) GO, (c and d) ZnCo LDH@rGO and (e) ZCS@rGO. (f–h) FE-SEM-HAADF-EDX elemental analysis of ZCS@rGO catalyst with respect to Zn–K, Co–K, S–K, and C–K and (i) EDS spectrum of ZCS@rGO catalyst.

bands (I_D/I_G) values of CS@rGO catalyst were approximately 1.16, which was slightly higher than that of the ZCS@rGO catalyst ($I_D/I_G = 1.14$), which represents a good interaction between the ZCS and rGO and the successful formation of ZCS@rGO catalyst.^{42,43}

The surface area and pore size of the ZCS@rGO catalyst were examined using a Brunauer–Emmett–Teller (BET) surface area analyzer, and the results are shown in Fig. 2i. ZCS@rGO catalyst displays a large specific surface area value of approximately $156 \text{ m}^2 \text{ g}^{-1}$, which is comparatively greater than that of a recently

reported ZnCoS based nanocomposite.⁴⁴ The hysteresis curve (type IV isotherm) of ZCS@rGO catalyst displayed mesoporous features on a relative pressure of 0.4–1.0. The pore size distribution of the ZCS@rGO catalyst was further examined by a Barret–Joyner–Halenda model (BJH) study, and the curve is shown in the inset image of Fig. 2i. ZCS@rGO catalyst possesses a pore size around $\sim 18.1 \text{ nm}$, which confirms the mesoporous environment. Such a high specific surface area with mesoporous ZCS@rGO catalyst benefits an improved electrocatalytic activity towards HER. The weight percentages of Zn and Co in

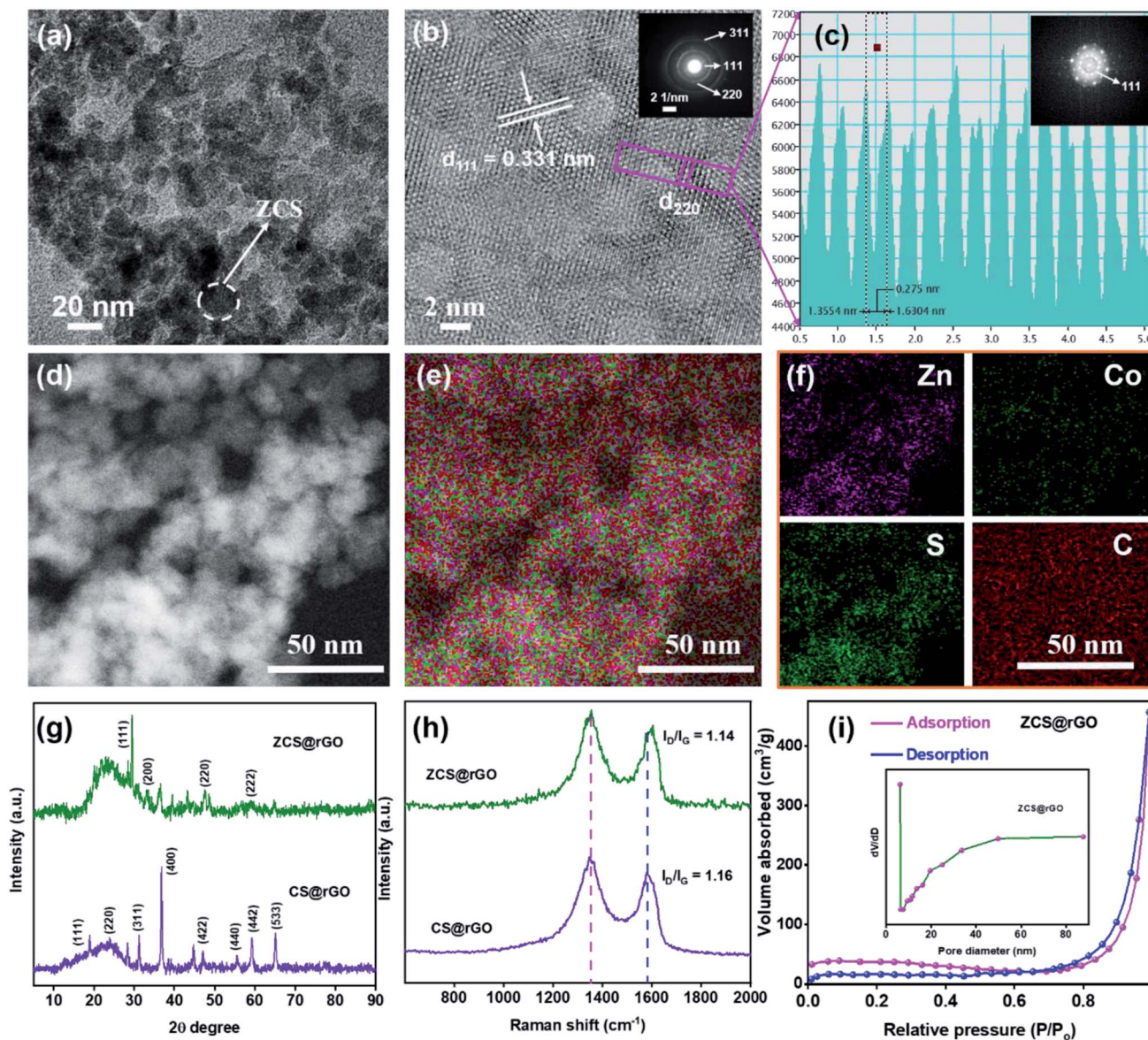


Fig. 2 Morphology analysis (a and b) HR-TEM image (inset image of (b): selected area diffraction (SAED pattern). (c) The lattice spacing of the ZCS@rGO catalyst layer (inset: FFT pattern). (d–f) HAADF-STEM-EDAX elemental mapping of ZCS@rGO. (g) XRD pattern, and (h) Raman analysis of CS@rGO and ZCS@rGO catalyst. (i) BET surface area from nitrogen adsorption–desorption curves (inset: pore size distribution) of ZCS@rGO catalyst.

ZCS@rGO and CS@rGO catalysts were calculated by ICP-OES, indicating loadings of Zn and Co of 2.41% and 5.48%, respectively (Fig. S4†).

The chemical state of the ZCS@rGO catalyst was studied by XPS analysis. XPS survey spectra of ZCS@rGO showed the presence of Zn 2p, Co 2p, and S 2p peaks in Fig. 3a–c, which ensure the presence of Zn, Co, and S. The XPS spectra of Zn 2p reveal peaks at 1020.8 and 1045.9 eV belonging to Zn 2p_{3/2} and Zn 2p_{1/2}, respectively,⁴⁴ demonstrating the existence of Zn²⁺ in ZCS@rGO catalyst (Fig. 3a). The XPS spectra of Co 2p presented in Fig. 3b display peaks at 797.0 and 781.3 eV corresponding to Co 2p_{1/2} and Co 2p_{3/2}, respectively.⁴⁵ Additionally, the satellite (sat.) peaks appear at 786.4 and 802.9 eV, which indicates the existence of Co(III) and Co(II).^{46,47} The XPS spectra of S 2p

(Fig. 3c) at 161.4 and 162.8 eV corresponds to S 2p_{3/2} and S 2p_{1/2}, respectively.^{48,49} The peak positions at 163.8 and 168.4 eV are representative peaks of metal sulfide bonds (S²⁻) and SO₄²⁻/HSO₄⁻, respectively. Further, Fig. S5† depicts the survey spectrum, O 1s, C 1s, and atomic percentage, which clearly shows the peak binding energy and composition of the ZCS@rGO catalyst.

To identify the in-depth elemental positions and electronic configuration of the optimal electrocatalyst, ZCS@rGO was examined by X-ray absorption spectroscopy (XAS) analysis with reference foils such as Zn and Co. The XANES spectra characterize the coordination environment of ZCS@rGO and CS@rGO catalysts,⁵⁰ as shown in Fig. 3. The XANES spectra disclose a characteristic peak at 7710 eV due to Co 4sp – Co 3d

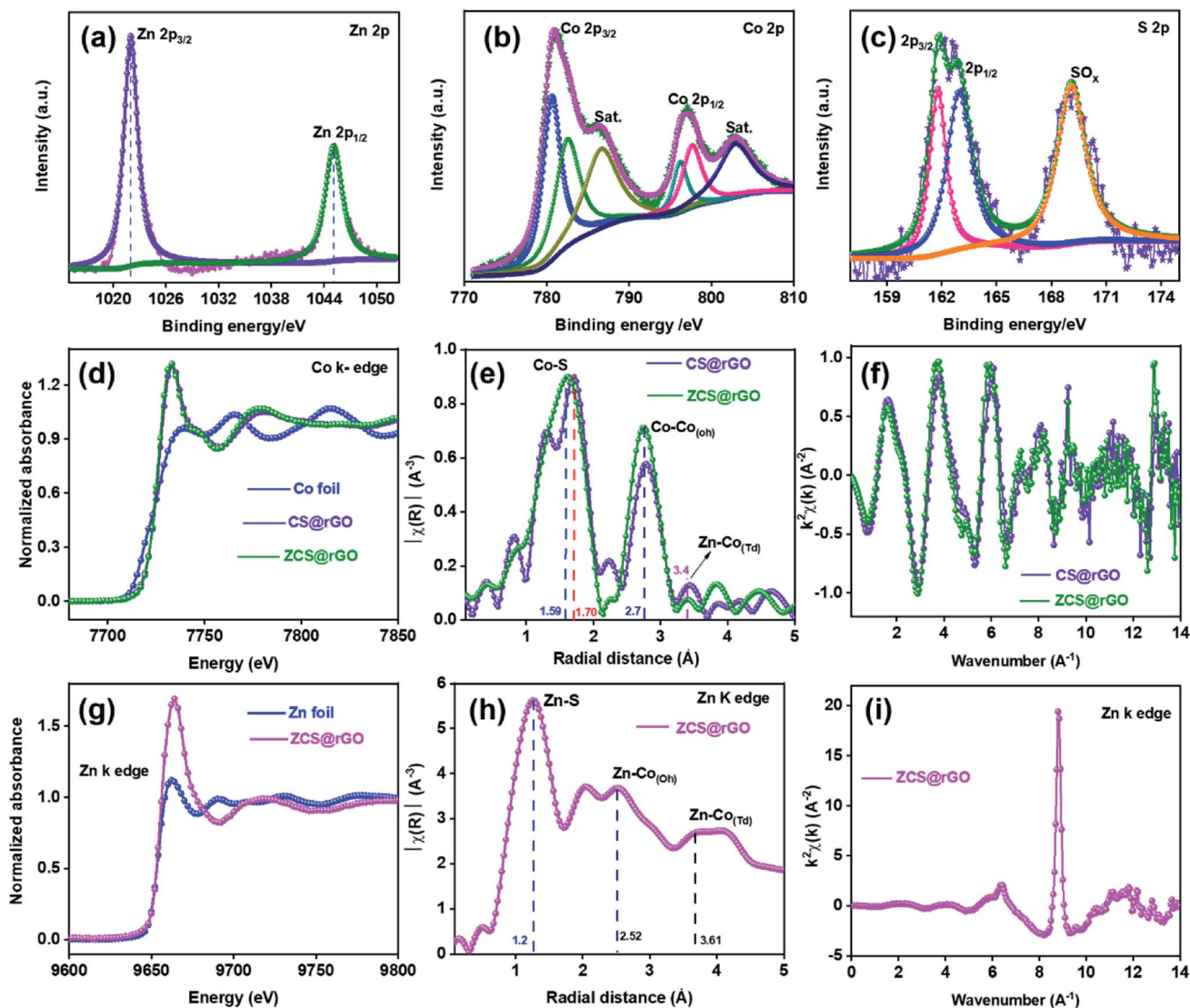


Fig. 3 The XPS analysis of (a) Zn 2p, (b) Co 2p, (c) S 2p spectra of ZCS@rGO catalyst. The XAS analysis, (d) XANES spectra of Co K-edge for CS@rGO and ZCS@rGO, (e) EXAFS spectra of Co K-edge for CS@rGO, and ZCS@rGO catalysts, (f) EXAFS oscillations of Co K-edge for CS@rGO, and ZCS@rGO catalysts. (g) XANES spectra for Zn K-edge of Zn-foil and ZCS@rGO catalyst. (h) EXAFS spectrum, and (i) EXAFS oscillations for Zn K-edge of ZCS@rGO catalyst, respectively.

hybridization,⁵¹ which further confirms the state of Co in CS@rGO and ZCS@rGO catalysts. Further, Fig. 3d shows the main absorption peaks of ZCS@rGO and CS@rGO catalysts appearing at 7730 eV due to electron transition from the occupied orbital of Co 1s to unoccupied orbitals of Co 4p. Interestingly, the peak intensities of ZCS@rGO catalyst are slightly increased as compared to Co-foil and CS@rGO catalyst due to the presence of Zn atoms. This confirms the oxidation states of cobalt are +2 and +3 in ZCS@rGO catalyst.⁵²

EXAFS spectra evaluate the coordination environment of cationic and anionic metal ions in CS@rGO and ZCS@rGO catalysts, as presented in Fig. 3e. The spectra show a strong peak around 1.59 Å, ascribed to the Co–S bonds in ZCS@rGO catalyst, whereas the Co–S peak position of CS@rGO catalyst was shifted to 1.70 Å because of Zn atoms occupying the T_d site in the spinel structure. Further, peaks around 2.7 and 3.4 Å

correspond to Co–Co(O_h) and Zn–Co(T_d) in octahedron and tetrahedron sites, respectively.⁵³ In Fig. 3f, oscillations of the Co K-edge show a neat EXAFS spectra from a good signal to noise percentage of CS@rGO and ZCS@rGO catalysts.⁵⁴ Fig. 3g shows Zn K-edge spectra of the XANES spectrum for Zn foil and ZCS@rGO catalyst, which confirms the coordination environment of zinc atoms (Zn²⁺), in which pre-edge and post-edge peaks appear at 9638 eV and 9664 eV, respectively. Further, the EXAFS spectrum confirms the presence of the Zn–S bond, Zn–Co(O_h) octahedron species, and Zn–Co(T_d) tetrahedron species, with corresponding peaks appearing at 1.2, 2.52, and 3.61 Å for ZCS@rGO catalyst, as observed in Fig. 3h. Fig. 3i displays the oscillation frequency of ZCS@rGO catalyst, with the EXAFS spectra collected with a good signal to noise ratio in the Zn K-edge.⁵⁵

The bond distance of the transition metal sulfide was decreased by increasing the central metal ion redox state because of the decreased effective ionic radius.⁵⁶ The promising electrocatalytic activity performance was based on the mixed oxidation state of Co²⁺ and Co³⁺ cationic valences existing in the octahedron site of ZCS@rGO. From the XAS results, the Zn electronegativity was low (Pauling electronegativity is 1.65) compared with Co (Pauling electronegativity is 1.88), indicating that the sulfide atom has a tendency to gain an electron from Co, confirming S–Zn–S–Co–S structures in ZCS. Fig. S6† shows the spin and structure state change of Co ion from CS@rGO to ZCS@rGO catalysts. For the electron transition, Co atoms exhibit different electron configurations of t_{2g}⁵ e_g² (high spin, Co²⁺) and t_{2g}⁶ e_g⁰ (low spin, Co³⁺) with a stable configuration of Zn²⁺ (t_{2g}⁶ e_g⁴).⁵⁷

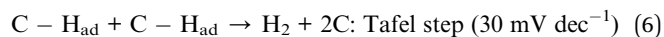
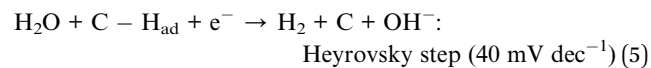
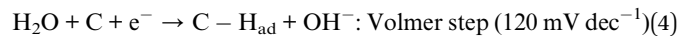
The high spin Co²⁺ (t_{2g}⁵ e_g²) and low spin Co³⁺ (t_{2g}⁶ e_g⁰) are changed in the occupied state when adding Zn atoms. Therefore, the CFSE energy level variance of Co²⁺ is less than that of Co³⁺ in CS@rGO and ZCS@rGO catalysts, and the CFSE energy value of Zn²⁺ ion is about zero as a stable state of ZCS@rGO catalyst. Based on CFSE calculations, it is important to describe Jahn–Teller (JT) effects,⁵⁸ which are important in describing the catalytic properties. Therefore, tetragonal distortion is present in the asymmetry electron of Co²⁺ state (low spin) as shown in Fig. S7, S8, and Table S1.† Improving the electrochemical performance of STMS requires suppressing the effects of Jahn–Teller distortion. Lan *et al.* observed that suppression of the Jahn–Teller distortion in In-doping MnCo₂O₄ enhanced the HER activities.^{59,60} We observed strong electronic interaction between Zn/Co and S in the ZCS@rGO catalyst, which can significantly minimize the Jahn–Teller distortion of Co²⁺ cation in ZCS@rGO, thus leading to improved intrinsic conductivity and electron transfer between metal cations and sulfur. Additionally, the ZCS@rGO catalyst offers enhanced HER activity with fast kinetics and better deprotonation/protonation reaction processes.

3.2 Electrocatalytic evaluation of STMS

The HER activity of the as-prepared electrocatalyst was analyzed using electrochemical techniques of cyclic voltammetry (CV), linear sweep voltammetry (LSV), and chronopotentiometry with 1.0 M KOH solution. Approximately 3 mg of prepared electrocatalyst was coated on cleaned CC, which acts as the working electrode. Fig. 4a shows *iR* corrected LSV comparative curves of ZCS@rGO/CC, CS@rGO/CC, GO/CC, and 20% Pt/C catalysts, and Fig. S9† displays LSV curves of CC and GO/CC. Notably, the optimal ZCS@rGO/CC exhibits a lower overpotential of 135 mV at 10 mA cm⁻² as related to CS@rGO/CC (341 mV) and GO/CC (420 mV) at the same current density, which is higher than the benchmark catalyst of 20% Pt–C (37 mV) as presented in Fig. 4b.

The reaction kinetics of ZCS@rGO/CC, CS@rGO/CC, GO/CC, and Pt–C/CC catalysts were evaluated using the Tafel slope, which is derived from the slope of overpotential vs. log(*j*) plot. As presented in Fig. 4c, the ZCS@rGO/CC catalyst has a smaller Tafel slope value of 47 mV dec⁻¹ when compared to CS@rGO/

CC (70 mV dec⁻¹) and GO/CC (81 mV dec⁻¹) catalysts, which is higher than the benchmark catalyst of 20% Pt–C (33 mV dec⁻¹) catalyst. This result demonstrates that the optimal ZCS@rGO/CC has improved inherent catalytic activity with enhanced HER kinetics. The Tafel results reveal the HER reactions follow the following rate-determining steps.



In the first step, hydrogen is adsorbed onto the ZCS@rGO (C – representing ZCS@rGO catalyst in steps 4–6) surface and generates hydroxide ions. H₂ gas molecules escape from the ZCS@rGO catalyst surface through two distinct reaction paths: (i) the Heyrovsky reaction and (ii) the Tafel reaction, as given in eqn (5) and (6), respectively. According to the above reaction pathway, H₂ molecules are out from the ZCS@rGO catalyst surface *via* the Volmer–Heyrovsky step. The electrochemical impedance spectra (EIS) of ZCS@rGO/CC, CS@rGO/CC, and GO/CC catalysts were measured using the onset potential of HER and frequency ranging between 100 KHz to 0.1 Hz in 1.0 M KOH, as shown in Fig. 4d. The results show that the ZCS@rGO catalyst has a lower *R*_{ct} value of 4.8 Ω when compared to CS@rGO/CC (5.71 Ω) and GO/CC (7.24 Ω) catalysts, indicating that ZCS@rGO/CC catalyst has good conductivity and excellent electron-transport kinetics for HER. Furthermore, ZCS@rGO/CC catalyst indicates a lower overpotential of 135 mV at a current density of 10 mA cm⁻² as compared to recently disclosed non-precious metal electrocatalysts as shown in Table S3† and Fig. 4e. We estimated the electrochemically active surface area (ECSA) of ZCS@rGO/CC, CS@rGO/CC, and GO/CC catalysts by calculating the electrochemical double-layer capacitance (*C*_{dl}). The *C*_{dl} values were calculated by measuring CV curves with different scan rates in the non-faradaic region, as presented in Fig. 5a–d. The double-layer capacitance (*C*_{dl}) value of ZCS@rGO/CC was ~0.032 mF cm⁻², which is higher than those of CS@rGO (0.0027 mF cm⁻²) and GO (0.0016 mF cm⁻²) catalysts. The calculated ECSA value of ZCS@rGO/CC (8.01 cm⁻²) catalyst demonstrates the existence of a greater number of electroactive sites in ZCS@rGO. Furthermore, the ZCS@rGO/CC had a higher turnover frequency (TOF) value of 0.0813 s⁻¹ at 302 mV, which is greater than that of the CS@rGO/CC (0.0129 s⁻¹) catalyst, as shown in Fig. 5e. These results further confirm that the hierarchical structure of zinc cobalt sulfides on reduced graphene sheet provides more active sites for HER.⁶¹

The long-term durability of ZCS@rGO/CC catalyst was examined using chronopotentiometry at 10 mA cm⁻² in alkaline 1.0 M KOH for 36 h. The results indicated that there were no significant changes in potential after 36 h, which confirms the high catalytic stability of the ZCS@rGO/CC catalyst (Fig. 4f). In contrast, the CS@rGO/CC catalyst exhibited a significant change in potential after 36 h, as shown in Fig. S10.†

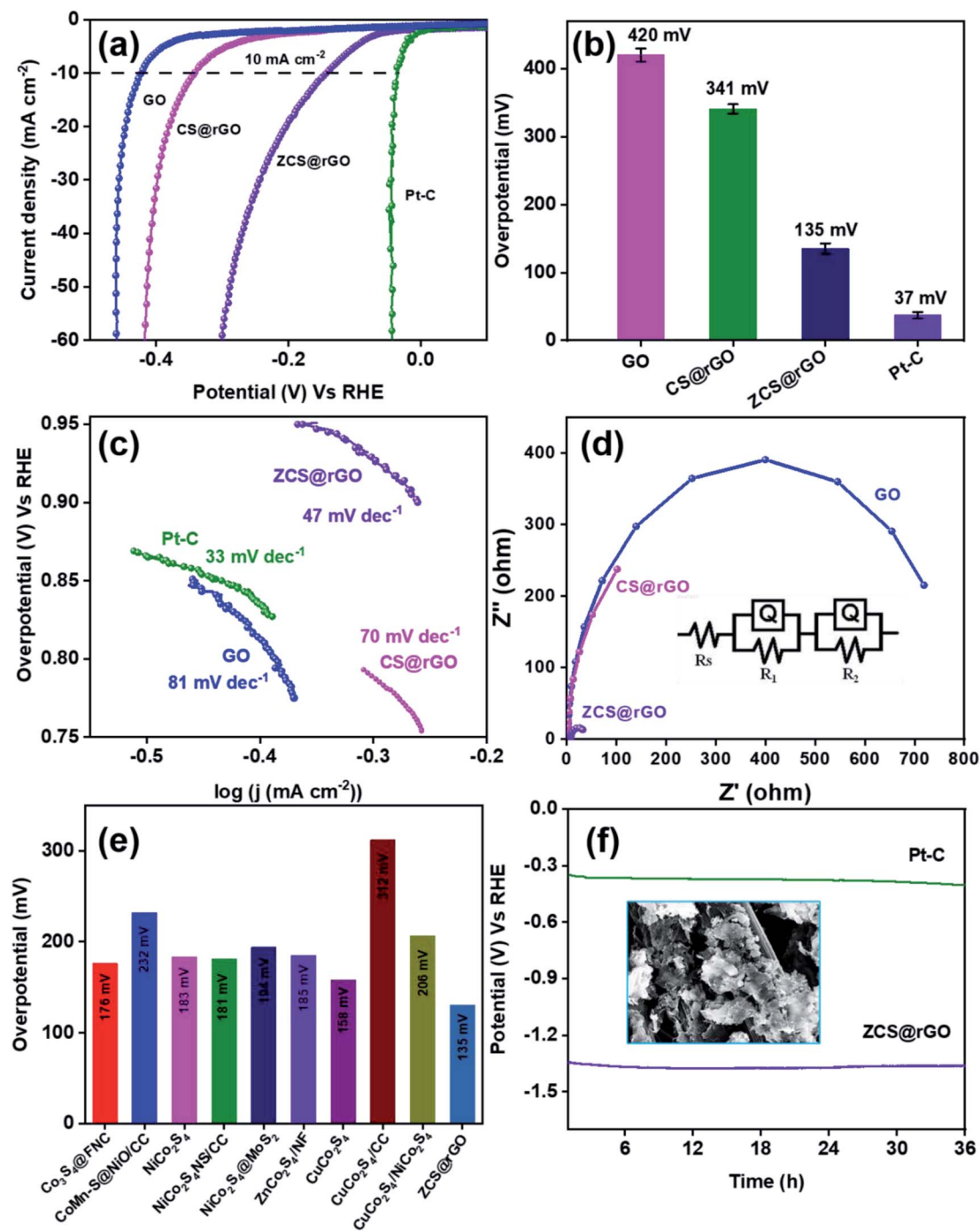


Fig. 4 HER performance. (a) Comparative LSV curves, (b) bar chart exhibiting HER overpotential at 10 mA cm^{-2} and (c) Tafel slope of ZCS@rGO, CS@rGO, GO, and Pt-C in 1.0 M KOH . (d) EIS spectrum of ZCS@rGO, CS@rGO, and GO. (e) The comparison of overpotential of ZCS@rGO with recently reported electrocatalysts and (f) long-term chronopotentiometry curves of ZCS@rGO and Pt-C at 10 mA cm^{-2} .

Fig. 6a shows LSV curves of ZCS@rGO/CC catalyst before and after 1000 cycles of CV in 1.0 M KOH solution (Fig. S11† (inset image of SEM)). The results indicated that there is a small difference in potential at a current density of 10 mA cm^{-2} , showing the improved stability of the ZCS@rGO/CC catalyst for HER. The HER stability test for the optimum ZCS@rGO/CC catalyst was achieved through multi-step chronopotentiometry. Fig. 6b displays the multi-step

chronopotentiometry curve of ZCS@rGO/CC catalyst recorded at different current densities from 10 to 80 mA cm^{-2} with a one-hour time interval for each step. The results display that initially the potential has a slight change at the current density of 10 – 20 mA cm^{-2} , but there were no significant changes in potential at current densities of 30 – 80 mA cm^{-2} over 8 h . This result confirms the ZCS@rGO/CC catalyst has excellent mass transport properties and stability of electrode.⁵²

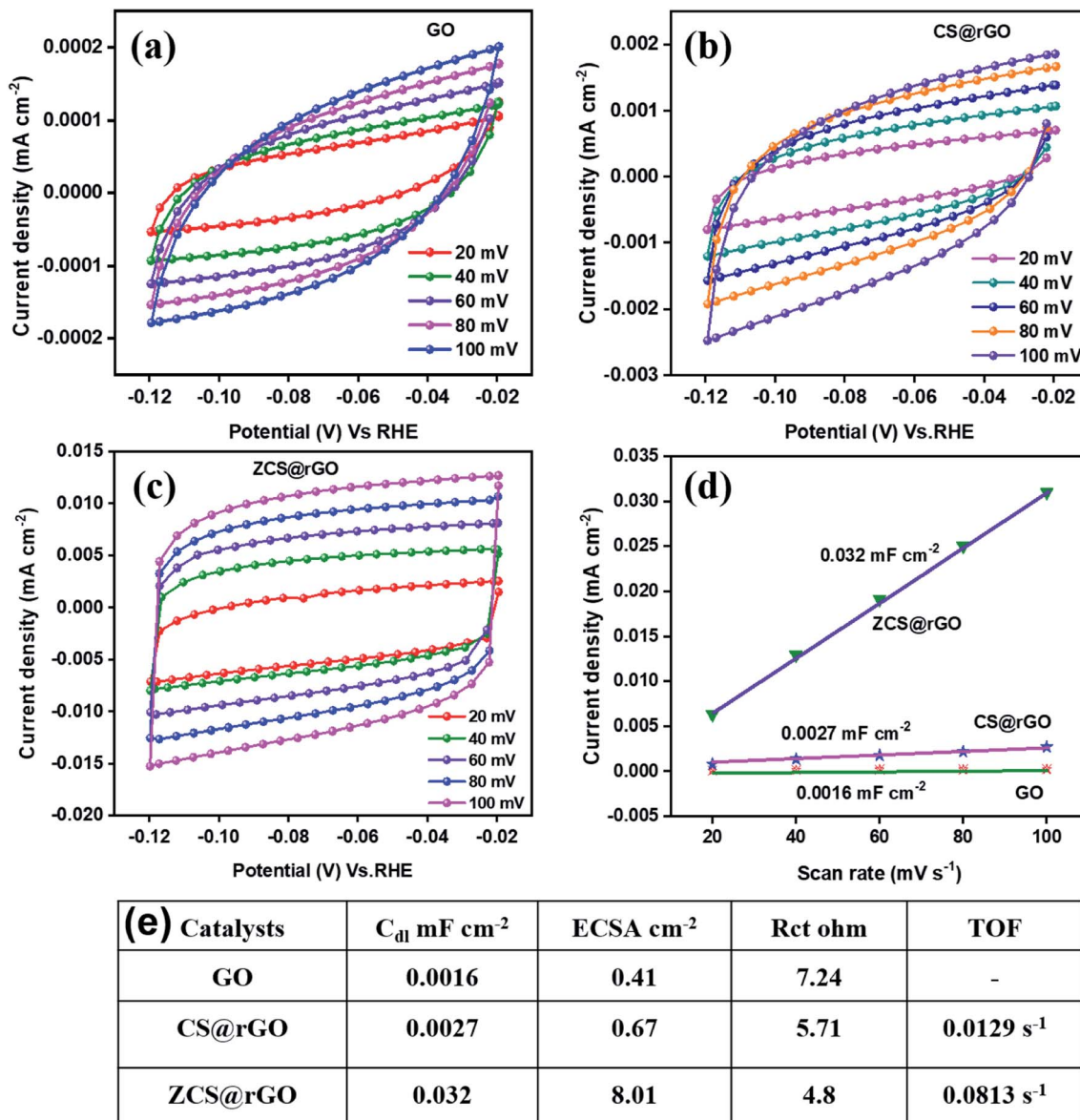


Fig. 5 CV curves with different scan rates from 20–100 mV s⁻¹ (a) GO/CC, (b) CS@rGO/CC, and (c) ZCS@rGO/CC. (d) Double-layer capacitance measurements (scan rate from 20 to 100 mV s⁻¹) of ZCS@rGO, CS@rGO, and GO. (e) The comparative table for electrochemical properties of ZCS@rGO, CS@rGO, and GO.

After the durability test, we investigated the morphology changes and chemical states of ZCS@rGO/CC catalyst through FE-SEM, TEM, and XPS analyses. After the durability test, morphology changes of ZCS@rGO/CC catalyst were investigated through SEM and TEM analyses, as shown in Fig. S12 and S13.† Fig. S12† shows the SEM images of the ZCS@rGO/CC, and there is a change of sphere-like hierarchical structure after the durability test. SEM-EDAX elemental mapping (Fig. S12c and d†) of ZCS@rGO/CC confirms the existence of Zn, Co, S, and C in ZCS@rGO catalyst after durability test. Additionally, TEM images of the zinc cobalt sulfide supported carbon cloth show ZCS is agglomerated on rGO sheet and good crystallinity with a *d*-space value of 0.197 nm belonging to (220) planes of ZCS@rGO catalyst, as

shown Fig. S13a.† Furthermore, HR-TEM-EDAX elemental mapping (Fig. S13d and e†) revealed the existence of Zn, Co, S, and C in ZCS@rGO catalyst after durability testing. After durability, the EDS spectra SEM (inset table in Fig. S12d†) and TEM of ZCS@rGO (inset table in Fig. S13e†) catalyst shows that zinc, cobalt, and sulfur contents were gradually reduced as compared to before durability test, which demonstrates the conversion of sulfide to sulfur oxides with a slight dissolution of the catalyst during the durability test.⁶² Furthermore, we investigated the chemical composition of ZCS@rGO catalyst by using XPS analysis after durability tests (Fig. S14†), which shows the intensity of Co 2p and Zn 2p peaks were reduced and the S 2p peaks confirmed the oxidation of sulfur during hydrogen evolution (see Fig. S14†). Additionally, the element

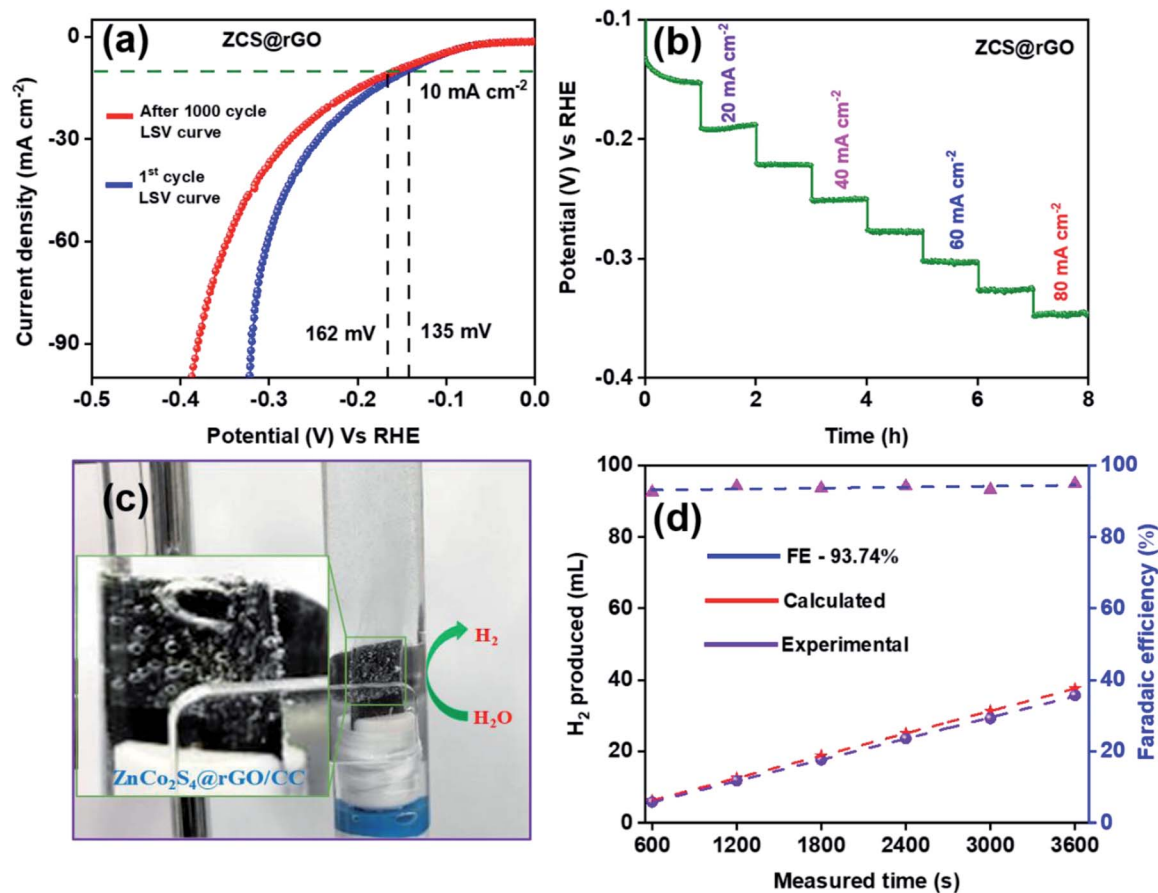


Fig. 6 HER durability test. (a) LSV curve of ZCS@rGO catalyst for before and after 1000 cyclic voltammograms at 100 mV s⁻¹ in 1.0 M KOH. (b) Multi-step chronopotentiometry curve at 10–80 mA cm⁻², (c) digital photograph display of H₂ gas evolution, and (d) faradaic efficiency of HER by using ZCS@rGO/CC.

composition (atomic %) of ZCS@rGO was gradually reduced after the durability test, as shown in the inset table of Fig. S14f.† This result confirms the adsorption of OH active species on the surface of the ZCS@rGO catalyst.⁶¹ The efficiency of hydrogen evolution was measured by calculating the faradaic efficiency of the ZCS@rGO/CC catalyst through the water displacement method, as shown in Fig. 6c. Fig. 6d shows that the faradaic efficiency of H₂ evolution by ZCS@rGO catalyst was 93.74%.

3.3 Computational methodology and results

To get a better understanding of the catalytic performance of ZCS@rGO, DFT calculations were used to investigate the HER mechanism of Co₃S₄ and ZnCo₂S₄. The DFT calculation was performed by using the Vienna *ab initio* simulation package software (VASP). Initially, structural optimization was carried out in order to find the best catalyst structures, which are schematically represented in Fig. S18.† Fig. S15† shows the density of states (DOS) of CS and ZCS, and the results display the highest charge density near the Fermi level, demonstrating a large number of charge carriers. The DOS distribution of ZCS near the Fermi surface increases the electron transfer and good H* adsorption ability of sphere-like hierarchical ZCS in the prepared catalyst.^{63–65}

We investigated the Gibbs free energy value (ΔG_{H^*}) of ZnCo₂S₄ adsorbed with H atoms to describe the HER mechanism.⁶⁶ Based on a previous report, we evaluated the detailed ΔG_{H^*} values of ZnCo₂S₄ and Co₃S₄ to understand the HER catalytic activity. Stable configurations for the adsorption of H atoms on the spinel Co₃S₄ and ZnCo₂S₄ systems are shown in Fig. 7. As shown in Fig. 7a, b and S16,† the hydrogen atoms are adsorbed on the surface of the S atom of Co₃S₄ to form H*-Co₃S₄. The H₂ adsorption occurs similarly in ZnCo₂S₄ to form H*-ZnCo₂S₄, as Fig. 7c, d, and S17† portray. Based on the DFT calculations, formation energy values are listed in Fig. S18 and Table S2,† and ΔG_{H^*} values for H*-Co₃S₄ and H*-ZnCo₂S₄ were determined to be 0.894 eV and -0.329 eV for the adsorption of H atoms on S sites, respectively. Further, the optimized electronic structures are shown in Fig. 7e and f for Co₃S₄ and ZnCo₂S₄ electrocatalysts. Fig. 7g displays the calculated free energy values (ΔG_{H^*}) for hydrogen adsorption on ZnCo₂S₄ and Co₃S₄. In theoretical HER estimation, change in Gibbs free energy during hydrogen adsorption (ΔG_{H^*}) is considered as one of the crucial parameters. The electrocatalyst materials are expected to show excellent HER activity when the ΔG_{H^*} is close to 0 eV. Based on the DFT results, hydrogen atoms prefer to

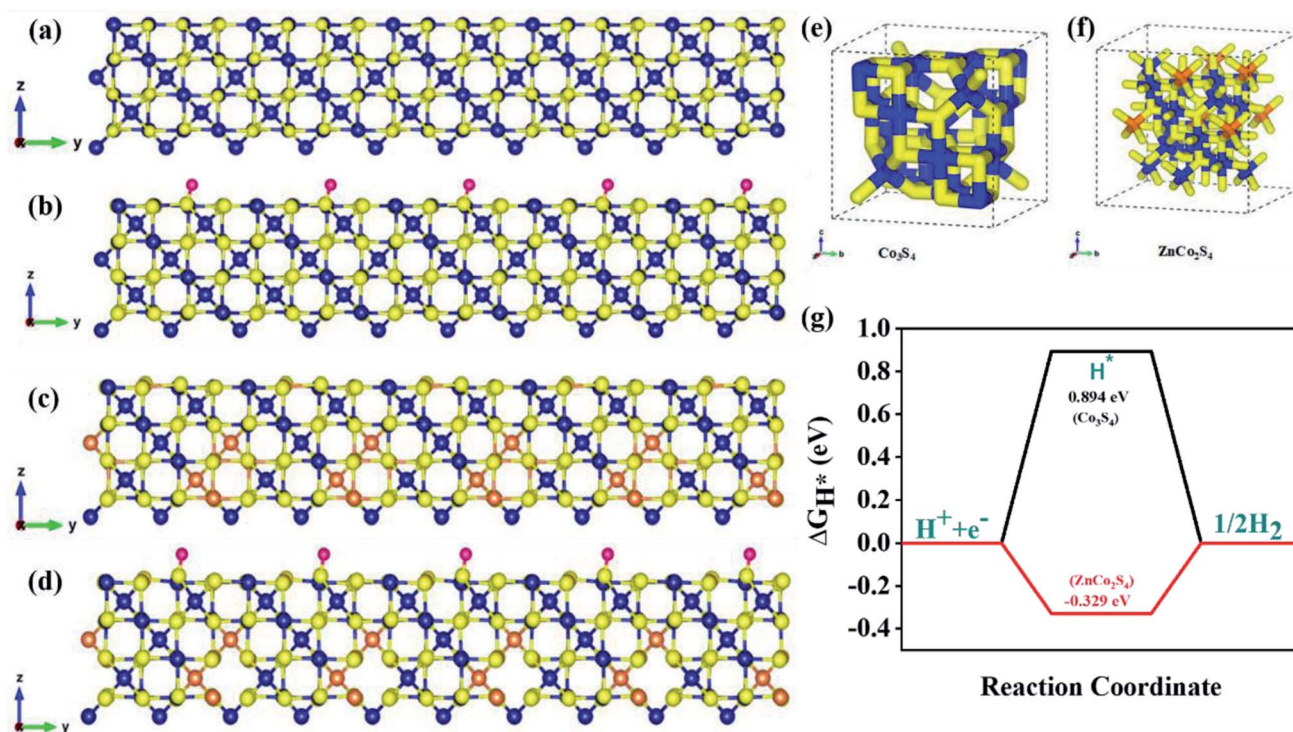


Fig. 7 DFT calculations. The side view of (a) Co_3S_4 and (b) $\text{H}^*\text{-Co}_3\text{S}_4$ nanostructure, side view of (c) ZnCo_2S_4 and (d) $\text{H}^*\text{-ZnCo}_2\text{S}_4$, spinel structure of (e) Co_3S_4 and (f) ZnCo_2S_4 (navy, yellow, orange, and red balls represent Co, S, Zn, and H atoms, respectively). (g) The HER Gibbs free energy profiles for the different active sites at Co_3S_4 and ZnCo_2S_4 .

adsorb on the S atoms of Co_3S_4 and ZnCo_2S_4 layers. We observed that the excellent activity of the spinel ZnCo_2S_4 as compared to Co_3S_4 was mainly due to the fact that Zn atoms occupy T_d sites and Co atoms occupy O_h sites. Therefore, it can be inferred that the transition metal sulfide system enhances H_2 evolution behavior, as a result improving the catalytic efficiency for HER.

4. Conclusion

In summary, we have successfully grown hierarchical ZCS on rGO through a two-step hydrothermal method. The morphology, chemical and physical properties of ZCS@rGO catalyst were investigated using various analytical techniques, including SEM, TEM, XRD, XPS, XAS, and BET analyses, which support the formation of a hierarchical morphology of ZCS on rGO sheets. The key parameters of the hierarchical ZCS@rGO catalyst provided a large surface area with more exposed electrocatalytic active sites and improved the interfacial interactions between hierarchical ZCS and the rGO sheet. As a result, the hierarchical ZCS@rGO catalyst showed low overpotential (135 mV at 10 mA cm^{-2}), Tafel slope value (47 mV dec^{-1}), and outstanding durability as compared to CS@rGO catalyst in alkaline medium. Further, we evaluated the electronic structure and HER mechanism of ZCS using DFT calculations. Thus, the present work paves the way to the development of high surface area, hierarchical structured, bimetal sulfide-based electrocatalysts for green production of hydrogen gas *via* water electrolysis.

Conflicts of interest

The authors declare no competing financial interest.

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References

- 1 Z. W. Seh, J. Kibsgaard, C. F. Dickens, I. Chorkendorff, J. K. Nørskov and T. F. Jaramillo, *Science*, 2017, **355**, eaad4998.
- 2 C. Wei, Y. Sun, G. G. Scherer, A. C. Fisher, M. Sherburne, J. W. Ager and Z. J. Xu, *J. Am. Chem. Soc.*, 2020, **142**, 7765–7775.
- 3 J. Zhu, L. Hu, P. Zhao, L. Y. S. Lee and K.-Y. Wong, *Chem. Rev.*, 2020, **120**, 851–918.
- 4 I. Ledezma-Yanez, W. D. Z. Wallace, P. Sebastian-Pascual, V. Climent, J. M. Feliu and M. T. M. Koper, *Nat. Energy*, 2017, **2**, 17031.

- 5 C. Wei, R. R. Rao, J. Peng, B. Huang, I. E. L. Stephens, M. Risch, Z. J. Xu and Y. Shao-Horn, *Adv. Mater.*, 2019, **31**, 1806296.
- 6 J. Kibsgaard, C. Tsai, K. Chan, J. D. Benck, J. K. Nørskov, F. Abild-Pedersen and T. F. Jaramillo, *Energy Environ. Sci.*, 2015, **8**, 3022–3029.
- 7 C. Hu, L. Zhang and J. Gong, *Energy Environ. Sci.*, 2019, **12**, 2620–2645.
- 8 J. Zhang, Q. Zhang and X. Feng, *Adv. Mater.*, 2019, **31**, 1808167.
- 9 N. Xue, Z. Lin, P. Li, P. Diao and Q. Zhang, *ACS Appl. Mater. Interfaces*, 2020, **12**, 28288–28297.
- 10 J. Kim, H. Jung, S.-M. Jung, J. Hwang, D. Y. Kim, N. Lee, K.-S. Kim, H. Kwon, Y.-T. Kim, J. W. Han and J. K. Kim, *J. Am. Chem. Soc.*, 2021, **143**, 1399–1408.
- 11 Y. Liu, J. Zhang, B. Han, X. Wang, Z. Wang, C. Xue, G. Bian, D. Hu, R. Zhou, D.-S. Li, Z. Wang, Z. Ouyang, M. Li and T. Wu, *J. Am. Chem. Soc.*, 2020, **142**, 6649–6660.
- 12 M. Ferri, J. D. Elliott, M. F. Camellone, S. Fabris and S. Piccinin, *ACS Catal.*, 2021, **11**, 1897–1910.
- 13 X. Zhu, J. Dai, L. Li, D. Zhao, Z. Wu, Z. Tang, L.-J. Ma and S. Chen, *Carbon*, 2020, **160**, 133–144.
- 14 J. Hu, Y. Ou, Y. Li, D. Gao, Y. Zhang and P. Xiao, *ACS Sustainable Chem. Eng.*, 2018, **6**, 11724–11733.
- 15 G. Song, Z. Wang, J. Sun, J. Sun, D. Yuan and L. Zhang, *Electrochem. Commun.*, 2019, **105**, 106487.
- 16 A. R. Puente Santiago, T. He, O. Eraso, M. A. Ahsan, A. N. Nair, V. S. N. Chava, T. Zheng, S. Pilla, O. Fernandez-Delgado, A. Du, S. T. Sreenivasan and L. Echegoyen, *J. Am. Chem. Soc.*, 2020, **142**, 17923–17927.
- 17 B. G. Fiss, N.-N. Vu, G. Douglas, T.-O. Do, T. Friscic and A. Moores, *ACS Sustainable Chem. Eng.*, 2020, **8**, 12014–12024.
- 18 U. P. Suryawanshi, U. V. Ghorpade, D. M. Lee, M. He, S. W. Shin, P. V. Kumar, J. S. Jang, H. R. Jung, M. P. Suryawanshi and J. H. Kim, *Chem. Mater.*, 2021, **33**, 234–245.
- 19 B. Cao, G. M. Veith, J. C. Neuefeind, R. R. Adzic and P. G. Khalifah, *J. Am. Chem. Soc.*, 2013, **135**, 19186–19192.
- 20 J. Li, C. Yao, X. Kong, Z. Li, M. Jiang, F. Zhang and X. Lei, *ACS Sustainable Chem. Eng.*, 2019, **7**, 13278–13285.
- 21 D. Rathore, M. D. Sharma, A. Sharma, M. Basu and S. Pande, *Langmuir*, 2020, **36**, 14019–14030.
- 22 A. Karmakar, K. Karthick, S. Kumaravel, S. S. Sankar and S. Kundu, *Inorg. Chem.*, 2021, **60**, 2023–2036.
- 23 X. Du, J. Huang, J. Zhang, Y. Yan, C. Wu, Y. Hu, C. Yan, T. Lei, W. Chen, C. Fan and J. Xiong, *Angew. Chem., Int. Ed.*, 2019, **58**, 4484–4502.
- 24 M. E. G. Lyons, R. L. Doyle, D. Fernandez, I. J. Godwin, M. P. Browne and A. Rovetta, *Electrochim. Acta*, 2014, **45**, 60–62.
- 25 Y. Jiao, Y. Zheng, M. Jaroniec and S. Z. Qiao, *Chem. Soc. Rev.*, 2015, **44**, 2060–2086.
- 26 Y. Wang, W. Qiu, E. Song, F. Gu, Z. Zheng, X. Zhao, Y. Zhao, J. Liu and W. Zhang, *Natl. Sci. Rev.*, 2018, **5**, 327–341.
- 27 D.-Y. Wang, M. Gong, H.-L. Chou, C.-J. Pan, H.-A. Chen, Y. Wu, M.-C. Lin, M. Guan, J. Yang, C.-W. Chen, Y.-L. Wang, B.-J. Hwang, C.-C. Chen and H. Dai, *J. Am. Chem. Soc.*, 2015, **137**, 1587–1592.
- 28 M. Wang, L. Zhang, Y. He and H. Zhu, *J. Mater. Chem. A*, 2021, **9**, 5320–5363.
- 29 X. Wu, X. Han, X. Ma, W. Zhang, Y. Deng, C. Zhong and W. Hu, *ACS Appl. Mater. Interfaces*, 2017, **9**, 12574–12583.
- 30 S. Sahoo and J.-J. Shim, *ACS Sustainable Chem. Eng.*, 2017, **5**, 241–251.
- 31 X. Yang, H. Sun, P. Zan, L. Zhao and J. Lian, *J. Mater. Chem. A*, 2016, **4**, 18857–18867.
- 32 Z. Zhang, Y. Huang, X. Liu, C. Chen, Z. Xu and P. Liu, *Carbon*, 2020, **157**, 244–254.
- 33 F. Luo, D. Ma, Y. Li, H. Mi, P. Zhang and S. Luo, *Electrochim. Acta*, 2019, **299**, 173–181.
- 34 D. Strmcnik, P. P. Lopes, B. Genorio, V. R. Stamenkovic and N. M. Markovic, *Nano Energy*, 2016, **29**, 29–36.
- 35 B. Hammer and J. K. Nørskov, *Adv. Catal.*, 2000, **45**, 71–129.
- 36 J. Gautam, Y. Liu, J. Gu, Z. Ma, J. Zha, B. Dahal, L.-N. Zhang, A. N. Chishti, L. Ni, G. Diao and Y. Wei, *Adv. Funct. Mater.*, 2021, 2106147.
- 37 R. Santhosh Kumar, K. Govindan, S. Ramakrishnan, A. R. Kim, J.-S. Kim and D. J. Yoo, *Appl. Surf. Sci.*, 2021, **556**, 149765.
- 38 J. Yu, S. Chen, W. Hao and S. Zhang, *ACS Nano*, 2016, **10**, 2500–2508.
- 39 V. Mani, S. Shanthi, T.-K. Peng, H.-Y. Lin, H. Ikeda, Y. Hayakawa, S. Ponnusamy, C. Muthamizhchelvan and S.-T. Huang, *Sens. Actuators, B*, 2019, **287**, 124–130.
- 40 Z. Fang, L. Peng, Y. Qian, X. Zhang, Y. Xie, J. J. Cha and G. Yu, *J. Am. Chem. Soc.*, 2018, **140**, 5241–5247.
- 41 A. Pramanik, S. Maiti, T. Dhawa, M. Sreemany and S. Mahanty, *Mater. Today Energy*, 2018, **9**, 416–427.
- 42 N. Logeshwaran, S. Ramakrishnan, S. S. Chandrasekaran, M. Vinothkannan, A. R. Kim, S. Sengodan, D. B. Velusamy, P. Varadhan, J.-H. He and D. J. Yoo, *Appl. Catal., B*, 2021, **297**, 120405.
- 43 S. Ramakrishnan, J. Balamurugan, M. Vinothkannan, A. R. Kim, S. Sengodan and D. J. Yoo, *Appl. Catal., B*, 2020, **279**, 119381.
- 44 S. Dutta, C. Ray, Y. Negishi and T. Pal, *ACS Appl. Mater. Interfaces*, 2017, **9**, 8134–8141.
- 45 M. Oku and K. Hirokawa, *J. Electron Spectrosc. Relat. Phenom.*, 1976, **8**, 475–481.
- 46 R. Z. Wang and Y. Zhang, *Mol. Syst. Des. Eng.*, 2020, **5**, 565–572.
- 47 P. Shi, Y. Zhang, G. Zhang, X. Zhu, S. Wang and A.-L. Wang, *J. Mater. Chem. A*, 2021, **9**(35), 19719–19724.
- 48 A. Sivanantham, P. Ganesan and S. Shanmugam, *Adv. Funct. Mater.*, 2016, **26**, 4661–4672.
- 49 S. Ramakrishnan, M. Karuppanan, M. Vinothkannan, K. Ramachandran, O. J. Kwon and D. J. Yoo, *ACS Appl. Mater. Interfaces*, 2019, **11**, 12504–12515.
- 50 D. Chen, C.-L. Dong, Y. Zou, D. Su, Y.-C. Huang, L. Tao, S. Dou, S. Shen and S. Wang, *Nanoscale*, 2017, **9**, 11969–11975.
- 51 M. Wang, C.-L. Dong, Y.-C. Huang and S. Shen, *ACS Catal.*, 2020, **10**, 1855–1864.

- 52 Z. Dai, H. Geng, J. Wang, Y. Luo, B. Li, Y. Zong, J. Yang, Y. Guo, Y. Zheng, X. Wang and Q. Yan, *ACS Nano*, 2017, **11**, 11031–11040.
- 53 W. Deeloed, Y. Hanlumyuang, W. Limphirat, S. Suramitr, K. Chansaenpak, P. Kanjanaboos, S. Wannapaiboon and W. Wattanathana, *Crystals*, 2021, **11**(10), 1256.
- 54 L. Liu, D. K. Wang, P. Kappen, D. L. Martens, S. Smart and J. C. Diniz da Costa, *Phys. Chem. Chem. Phys.*, 2015, **17**, 19500–19506.
- 55 Z. Zhao, G. Tian, V. Trouillet, L. Zhu, J. Zhu, A. Missiul, E. Welter and S. Dsoke, *Inorg. Chem. Front.*, 2019, **6**, 1861–1872.
- 56 S. Chen, H. Huang, P. Jiang, K. Yang, J. Diao, S. Gong, S. Liu, M. Huang, H. Wang and Q. Chen, *ACS Catal.*, 2020, **10**, 1152–1160.
- 57 J. K. Burdett, G. D. Price and S. L. Price, *J. Am. Chem. Soc.*, 1982, **104**, 92–95.
- 58 M. D. Kaplan, *J. Am. Chem. Soc.*, 2006, **128**, 10631–10632.
- 59 B. Lan, Y. Xiang, X. Luo, D. Wu, L. Zhang, J. Duan, M. Guo, Y. Ito and Y. Liu, *Chem. Eng. J.*, 2022, **430**, 132886.
- 60 S. Hirai, S. Yagi, A. Seno, M. Fujioka, T. Ohno and T. Matsuda, *RSC Adv.*, 2016, **6**, 2019–2023.
- 61 Y.-H. Fang and Z.-P. Liu, *ACS Catal.*, 2014, **4**, 4364–4376.
- 62 P. W. Menezes, A. Indra, A. Bergmann, P. Chernev, C. Walter, H. Dau, P. Strasser and M. Driess, *J. Mater. Chem. A*, 2016, **4**, 10014–10022.
- 63 G. Kresse and D. Joubert, *Phys. Rev. B*, 1999, **59**, 1758–1775.
- 64 G. Kresse and J. Furthmuller, *Comput. Mater. Sci.*, 1996, **6**, 15–50.
- 65 D. Voiry, H. Yamaguchi, J. Li, R. Silva, D. C. Alves, T. Fujita, M. Chen, T. Asefa, V. B. Shenoy, G. Eda and M. Chhowalla, *Nat. Mater.*, 2013, **12**, 850–855.
- 66 D. Zhao, K. Sun, W.-C. Cheong, L. Zheng, C. Zhang, S. Liu, X. Cao, K. Wu, Y. Pan, Z. Zhuang, B. Hu, D. Wang, Q. Peng, C. Chen and Y. Li, *Angew. Chem., Int. Ed.*, 2020, **59**, 8982–8990.