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# Electrospun porous La–Sr–Co–Ni–O nanofibers for highly sensitive non-enzymatic glucose detection†

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Glucose biosensors are widely used for clinical, industrial, and environmental applications. Nonenzymatic electrochemical glucose biosensors based on metal oxides with a perovskite structure have exhibited high sensitivity, excellent stability, and cost efficiency. In this work, porous La–Sr–Co–Ni–O (LSCNO) nanofibers, with an ABO<sub>3</sub>-type perovskite structure, were prepared through optimizing the A-site and B-site elements by electrospinning, followed with calcination at 700 °C for 5 h. Characterized by scanning and transmission electron microscopy, X-ray diffraction, and X-ray photoelectron spectroscopy, fabricated nanofibers were confirmed to be porous and nanosized polycrystalline grains with high crystallinity. A novel La<sub>0.75</sub>Sr<sub>0.25</sub>Co<sub>0.5</sub>Ni<sub>0.5</sub>O<sub>3</sub>-based nonenzymatic electrochemical biosensor was developed, which is sensitive to glucose because of an electrochemically catalytic mechanism, a mediated electron transfer involving Ni(II)/Ni(III) or Co(II)/Co(III), accompanying with gluconic acid complexation. The glucose biosensor presented a linear response in the range of 0.1–1.0 mM with a calibration sensitivity of 924 ± 28 μA mM<sup>-1</sup>, a proportion of the variance of 0.9926, and a lower limit of detection of 0.083 mM, respectively, demonstrating an outstanding analytical performance. The biosensor showed no response to the most widely used anionic surfactant, sodium dodecyl sulfate, and low sensitivity to other biomolecules, such as fructose, lactose, galactose, mannose, dopamine, and ascorbic acid. A urine sample was tested by this novel nonenzymatic electrochemical biosensor by standard addition method, suggesting a potential application for clinical test.

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## 1. Introduction

Non-enzymatic fast and accurate detection of biological molecules is fascinating because of their potential applications in disease diagnosis and management, biological analysis, food industry, and environmental protection. Numerous electrochemical techniques have been widely used as probes for a routine analysis of body fluids for the detection of pathogens and biomarkers to identify several diseases. However, the search for non-enzymatic sensors for biomarkers is constantly

increasing due to their high selectivity, low detection limit, stability, low cost, and reproducibility as compared to enzymatic counterparts.<sup>1</sup> The non-enzymatic sensors rely on the electrocatalytic oxidation of analytes at the electrode surface in the absence of any expensive enzymes, reducing their cost considerably. Noble metals are identified as promising catalysts for electrochemical catalysis and sensing; however, their high cost and potential catalyst poisoning limit their extensive applications.<sup>2</sup>

Nanostructured metal oxides are considered as the most promising low-cost alternatives to the enzymatic and noble metal sensors for the selective detection of biomarkers. Several transition metal oxides are identified as suitable candidates for glucose<sup>3–7</sup> and dopamine.<sup>8–10</sup> The characteristics of the electrode surface are the primary factor that controls the electrochemical sensing of the above analytes. Thus, morphology, porosity, and size of the electrode materials have a significant impact on the sensitivity, selectivity, response time, and limit of detection of the nanostructured electrochemically active oxides for sensors.<sup>11–13</sup> Transition metal oxides from nickel, copper, and cobalt are widely studied as non-enzymatic sensors for the

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detection of glucose and other biomarkers. There are several attempts to increase the sensitivity of these oxides by adopting different morphologies, increasing the porosity, and forming composites with other conductive materials, such as graphene and carbon nanotubes.<sup>10–12</sup>

Perovskite oxides with the general formula  $ABO_3$  are recognized as electrocatalysts for fuel cells,<sup>14,15</sup> supercapacitors,<sup>16–18</sup> and batteries.<sup>19,20</sup> They are also sensitive to several biomarkers such as glucose,<sup>21</sup>  $H_2O_2$ ,<sup>22</sup> dopamine,<sup>23</sup> etc. The unique characteristics of perovskite oxides are their three-dimensional network structure, which can accommodate abundant oxygen vacancies retaining their structural integrity.<sup>24</sup> It is also important to note that perovskite structures can accommodate multiple elements in the periodic table to their lattice without compromising the crystal structure.<sup>17,25</sup> The incorporation of other elements to the lattice can improve the selectivity and sensitivity towards various analytes through the manipulation of defects present in them. Perovskites containing transition metals such as Ni, Co, and Fe in their B-site and their composites are the most sought electrochemically active materials for sensing.

Nonenzymatic electrochemical glucose biosensors using perovskite oxide have been continuously attracting great attentions due to the growing requirements for diabetes management, food quality control, and bioprocess inspection, monitoring of glucose in various matrixes.<sup>26–29</sup> Although enzyme-based glucose biosensors still prevail in market, intrinsic problems remain in an enzymatic glucose sensor, such as low enzyme stability, high environmental dependency, high price, and complicated immobilization procedures, owing to the inadequate stability, denaturation of enzymes, thermal and chemical instability to environmental factors like moisture, pH and temperature.<sup>30,31</sup> Perovskite oxide non-enzymatic glucose biosensors have shown great advantages, such as high sensitivity, high surface area, porosity, low cost, and flexibility. In particular, non-enzymatic glucose biosensors are stable to environmental change as no enzymes are used.<sup>22,26,32–40</sup>

Herein, lanthanum, strontium, nickel and cobalt perovskite nanofibers, a composition that has not been reported previously, were synthesized by using electrospinning method and characterized by various techniques, such as X-ray Diffraction (XRD), X-ray photoelectron spectroscopy (XPS), scanning electron microscopy (SEM), and transmission electron microscopy (TEM). A nonenzymatic electrochemical glucose biosensor was presented with linear response between 0.1 mM and 1 mM by using cyclic voltammetry. The calibration sensitivity of  $924 \mu A mM^{-1}$  and lower limit of detection (LOD) of 0.083 mM were obtained. This level is much lower than the normal blood glucose level of 3.9–5.6 mM, indicating a potential biomedical application for blood test.

## 2. Experimental

### 2.1 Materials

Polyvinylpyrrolidone (PVP) with a mean molecular weight  $M_w = 1.3 \times 10^6 \text{ g mol}^{-1}$ , lanthanum(III) nitrate hydrate (99.9%), strontium nitrate, nickel(II) nitrate hexahydrate, cobalt(II) nitrate

hexahydrate, manganese(II) nitrate hydrate, iron(III) nitrate non-hydrate, *N,N*-dimethylformamide (DMF), and ethanol were purchased from Sigma Aldrich and were used without further purification to fabricate the precursor composite fibers. NaOH procured from Sigma Aldrich and carbon cloth obtained from MTI were used for the electrochemical measurements. D-(+)-Glucose (Anhydrous, ACS), D-(–)-fructose (Powder, high purity), lactose monohydrate (ACS), D-(+)-galactose ( $\geq 99.0\%$ ), D-(+)-mannose ( $\geq 99\%$ ), dopamine hydrochloride (99%), L-(+)-ascorbic acid (ACS), and other chemicals were purchased from VWR International.

### 2.2 Procedures

The fabrication of the perovskite nanofibers was divided into three steps: (1) preparation of electrospinnable solution containing the stoichiometric quantities of metal salts and gel-forming medium; (2) fabrication of precursor composite fibers by electrospinning process; and finally (3) calcining the precursor composite fibers above the degradation temperature of the volatile components. The electrospinnable precursor solutions were prepared by dissolving 0.07 M of stoichiometric quantities of the metal nitrates in a 20 mL 50/50 ethanol/*N,N*-dimethylformamide solvent mixture. Then 2.0 g of polyvinylpyrrolidone (PVP) was added to the above solution and stirred vigorously for 12 h to ensure uniform mixing. The electrospinning was conducted at atmospheric conditions with an applied voltage of 15 kV, a flow rate of  $300 \text{ mL h}^{-1}$ , and a spinneret to collector distance of 17 cm. The calcination temperature was selected based on the degradation temperature of the precursor composite fibers observed from their thermogravimetric analysis (TGA) and Differential thermal analysis (DTA) in a nitrogen atmosphere at a heating rate of  $10 \text{ }^\circ\text{C min}^{-1}$  (Shimadzu DTG-60). The xerogel fibers obtained through the electrospinning process were subsequently calcined at  $700 \text{ }^\circ\text{C}$  in the air for 5 h at a ramp of  $2 \text{ }^\circ\text{C min}^{-1}$  to get the resulting perovskite nanofibers.

SEM morphological data were collected using a JEOL JSM-6510LV scanning electron microscope. Carbon coated samples were prepared to observe the morphology of the nanofibers in a JEOL field-emission JXA-8530F Electron Probe Microanalyzer (EPMA), which was equipped with an X-ray energy-dispersive spectrometer. For TEM, the nanofibers were dispersed in pure ethanol by sonication for 5 min, and then a drop of the dispersion was placed on a carbon film supported grid and allowed to dry. The grids were observed in the FEI Talos F200X TEM/STEM instrument at 200 kV. The XRD patterns of the calcined nanofibers were recorded by a Rigaku MiniFlex 600 X-ray diffractometer with Cu  $K\alpha$  radiation ( $\lambda = 0.154178 \text{ nm}$ ), with a scan rate of  $0.075 \text{ deg min}^{-1}$  in the  $2\theta$  range of  $10\text{--}90^\circ$ . Surface composition and oxidation states of elements were characterized with XPS by Thermo Scientific ESCALAB 250Xi and the deconvolution of high-resolution XPS spectra were done using OriginPro 2020.

The nanofibers were immobilized on a carbon cloth electrode surface with Nafion as a binder. The nanofiber modified carbon electrode was used as a working electrode to detect glucose with cyclic voltammetry (CV) in the range of 0.2–0.7 Volt



(vs. Ag/AgCl, 1M KCl) (CH Instruments Model 440 Potentiostat, 3-electrodes systems: NMCE as working, a platinum wire as the counter, and Ag/AgCl, KCl (1M) as reference electrode).

### 3. Results and discussion

The TGA and DTA plots of electrospun PVP/metal salts composite nanofibers are shown in Fig. 1. As nanofibers exhibit three stages of thermal degradation during their transformation to LSCNO oxide nanofibers, even though the pure PVP exhibit a single step thermal degradation.<sup>41</sup> During the first step, removal of adsorbed water from the nanofibers as well as the hydrated metal salts takes place up to  $\sim 200$  °C. The onset of degradation of the as-spun precursor nanofibers is  $\sim 300$  °C and the degradation is completed at  $\sim 485$  °C. During the second step of the degradation, the elimination of organic parts of the mixtures, and the formation of subsequent perovskite oxides takes place. The multiple exothermic and endothermic phase transformations are revealed in the DTA plot. Even though, the weight loss above 485 °C is negligible, the exothermic reaction continues at high temperatures as in the DTA curve above 500 °C, therefore the calcination temperature is chosen as 700 °C.

#### 3.1 Crystal structure and morphology

The XRD patterns of representative LSCNO nanofibers are shown in Fig. 2, which confirms the formation of crystalline LSCNO nanofibers after the calcination of electrospun precursor nanofibers. The peaks in the XRD patterns are matching to that of the JCPDS data file no. 034-1181 of LSCNO nanofibers. All the XRD patterns comprise well-defined peaks without any apparent impurity peaks originating from the dopants, which discloses the occupancy of the A and B site dopants in the intended lattice site of the crystal structure. The electrospun ceramic nanofibers are composed of nanosized grains, therefore the XRD peaks are broad in the present study.

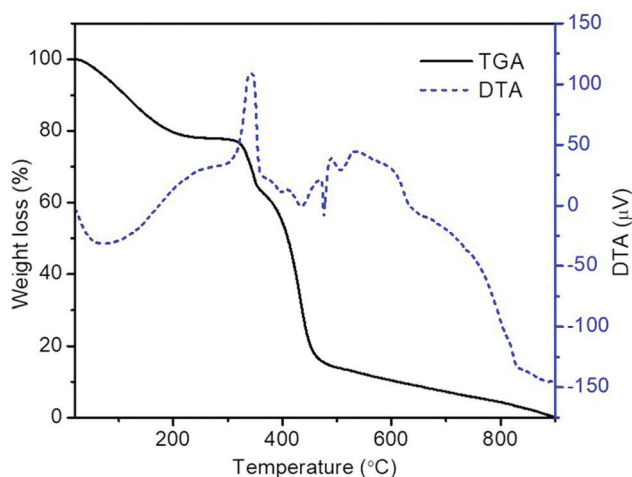


Fig. 1 TGA and DTA plots of PVP/metal nitrate composite nanofibers.

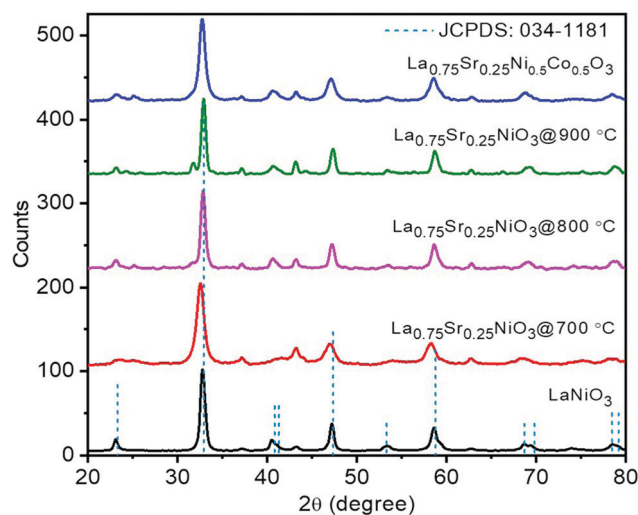


Fig. 2 XRD patterns of LSCNO representative nanofibers.

Fig. 3 shows the SEM images of  $\text{La}_{0.75}\text{Sr}_{0.25}\text{Co}_{0.5}\text{Ni}_{0.5}\text{O}_3$  nanofibers before and after calcination. The SEM images of nanofibers reveal that the surface of the nanofibers is smooth textured in nature. Conversely, the nanofibers after calcination are porous and rough textured with serrated edges. In general, the electrospun nanofibers are composed of myriads of nano-sized grains, which are formed by the nucleation and grain growth during the calcination process.<sup>42</sup> Additionally, the diameter of the nanofibers after calcination is reduced significantly as compared to the PVP/metal salt precursor composite nanofibers.

Fig. 4a and b shows the TEM and high-resolution TEM (HRTEM) images of the  $\text{La}_{0.75}\text{Sr}_{0.25}\text{Co}_{0.5}\text{Ni}_{0.5}\text{O}_3$  nanofibers. The TEM images reveal that the nanofibers are polycrystalline in nature, composed of nanosized grains with a highly porous nature of the nanofibers, as revealed in the magnified TEM image in the inset of Fig. 4a. The measured d-spacing from the HRTEM image (Fig. 4b) of LSCNO is consistent with the XRD results. The intermittent rings in the selected-area electron diffraction (SAED) pattern, as shown in Fig. 4c, reveals the polycrystalline nature of the nanofibers with high crystallinity. The nanosized grains composing the nanofibers were formed due to homogeneous nucleation and the spatial confinement of the nanofibers.

The high-resolution XPS analysis was performed to understand the oxidation states of Co and Ni as  $\text{Sr}^{2+}$  is replacing the

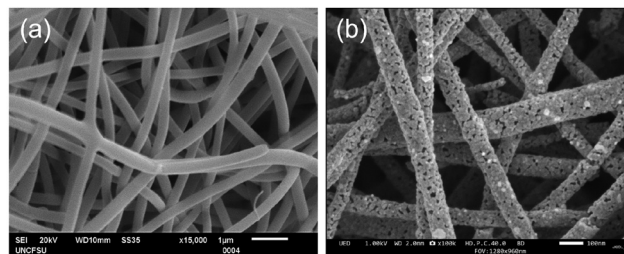


Fig. 3 SEM images of (a) PVP/metal salt precursor nanofibers before calcination, and (b) representative LSCNO nanofibers after calcination.



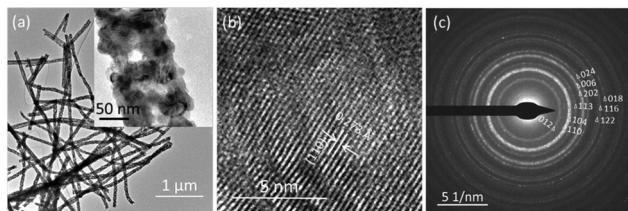


Fig. 4 (a) TEM (magnified in the inset), (b) lattice fringes, and (c) HRTEM images of representative LSCNO nanofibers.

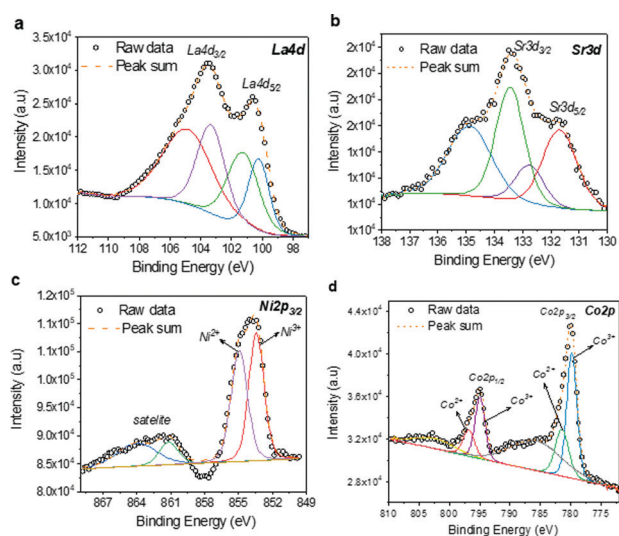


Fig. 5 High-resolution XPS spectra of LSCNO nanofibers: (a) La 4d; (b) Sr. 3d; (c) Ni 2d; and (d) Co 2p.

La<sup>3+</sup> lattice site. The high-resolution spectrum La 4d (Fig. 5a) reveals the peaks corresponding to La 4d<sub>3/2</sub> and La 4d<sub>5/2</sub> at ~103.45 and ~100.5 eV, respectively, in addition to the satellite peaks. Fig. 5b show the high-resolution spectrum of Sr 3d with significant peaks corresponding to Sr 3d<sub>3/2</sub> and Sr 3d<sub>5/2</sub>, respectively. The multiple peaks observed in the Ni 2p spectrum (Fig. 5c) reveals the two oxidation states of Ni<sup>2+</sup> and Ni<sup>3+</sup>. Similarly, in Fig. 5d, Co 2p spectra also exhibit peaks corresponding to Co<sup>2+</sup> and Co<sup>3+</sup> oxidation states. Due to the multiple oxidation states of the B-site cations, the charge exchange interactions among the cations are very likely. Apparently, such charge interactions are identified as the dominant mechanism of glucose sensing in non-enzymatic glucose sensors. Therefore, one can expect an efficient non-enzymatic detection of glucose using this material.

### 3.2 Compositional optimization for glucose response

Various compositions of LSCNO nanofibers were synthesized and then modified on a carbon electrode surface. The electrodes immobilized with different compositions of perovskite LSCNO nanofibers were tested on the correlations of anodic peak current (*I*<sub>p</sub>) vs. glucose concentrations (Fig. 6). As can be seen from Fig. 6 and Table 1, all the compositions displayed

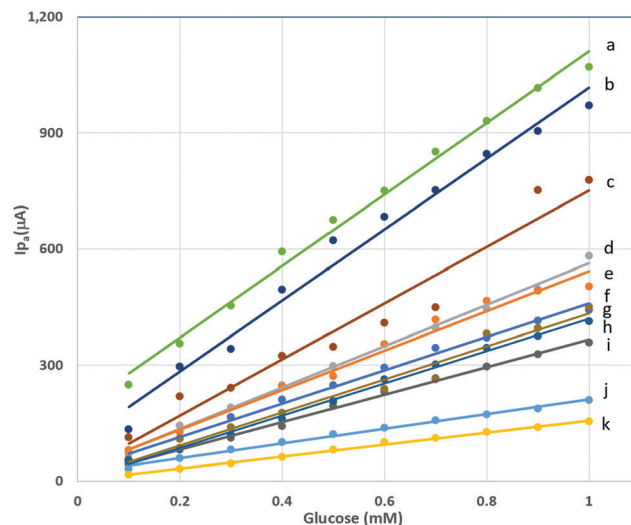


Fig. 6 Influence of LSCNO compositions on calibration curves of non-enzymatic electrochemical glucose biosensors. From top to bottom of the right end: (a) La<sub>0.75</sub>Sr<sub>0.25</sub>Co<sub>0.5</sub>Ni<sub>0.5</sub>O<sub>3</sub>; (b) La<sub>0.75</sub>Sr<sub>0.25</sub>Co<sub>0.25</sub>Ni<sub>0.75</sub>O<sub>3</sub>; (c) La<sub>0.75</sub>Sr<sub>0.25</sub>Co<sub>0.75</sub>Ni<sub>0.25</sub>O<sub>3</sub>; (d) La<sub>0.75</sub>Sr<sub>0.25</sub>Co<sub>0.33</sub>Ni<sub>0.33</sub>Mn<sub>0.33</sub>O<sub>3</sub>; (e) La<sub>0.75</sub>Sr<sub>0.25</sub>Co<sub>0.25</sub>Ni<sub>0.25</sub>Mn<sub>0.25</sub>Fe<sub>0.25</sub>O<sub>3</sub>; (f) La<sub>0.75</sub>Sr<sub>0.25</sub>NiO<sub>3</sub>; (g) La<sub>0.5</sub>Sr<sub>0.5</sub>NiO<sub>3</sub>; (h) La<sub>0.9</sub>Sr<sub>0.1</sub>NiO<sub>3</sub>; (i) LaNiO<sub>3</sub>; (j) La<sub>0.75</sub>Sr<sub>0.25</sub>CoO<sub>3</sub>; and (k) La<sub>0.75</sub>Sr<sub>0.25</sub>Co<sub>0.33</sub>Ni<sub>0.33</sub>Fe<sub>0.33</sub>O<sub>3</sub>.

linear relationship with good proportions of the variance, indicating potential applications as nonenzymatic glucose sensors. However, compositions of nanofibers significantly affect linear functions (Table 1). Among the compositions of LaNiO<sub>3</sub>, La<sub>0.5</sub>Sr<sub>0.5</sub>NiO<sub>3</sub>, La<sub>0.75</sub>Sr<sub>0.25</sub>NiO<sub>3</sub>, and La<sub>0.9</sub>Sr<sub>0.1</sub>NiO<sub>3</sub>, a La/Sr ratio at 0.75/0.25, e.g., La<sub>0.75</sub>Sr<sub>0.25</sub>NiO<sub>3</sub>, exhibited the highest calibration sensitivity of 432 ± 13 μA mM<sup>-1</sup>, thus this ratio was chosen as the La/Sr ratio. Replacing Ni with Co, e.g., La<sub>0.75</sub>Sr<sub>0.25</sub>CoO<sub>3</sub>, gave a calibration sensitivity of 191 ± 5 μA mM<sup>-1</sup>, a much lower value than that of La<sub>0.75</sub>Sr<sub>0.25</sub>NiO<sub>3</sub> based biosensor. Interestingly, coexistence of Co and Ni presented surprisingly high sensitivity. As listed in Table 1, La<sub>x</sub>Sr<sub>1-x</sub>Ni<sub>y</sub>Co<sub>1-y</sub>O<sub>3</sub> nanofibers displayed the best sensitivity, with a value of 924 ± 28 μA mM<sup>-1</sup> for La<sub>0.75</sub>Sr<sub>0.25</sub>Co<sub>0.5</sub>Ni<sub>0.5</sub>O<sub>3</sub>, 917 ± 45 μA mM<sup>-1</sup> for La<sub>0.75</sub>Sr<sub>0.25</sub>Co<sub>0.25</sub>Ni<sub>0.75</sub>O<sub>3</sub>, and 726 ± 62 μA mM<sup>-1</sup> for La<sub>0.75</sub>Sr<sub>0.25</sub>Co<sub>0.75</sub>Ni<sub>0.25</sub>O<sub>3</sub>. Nevertheless, a further addition of other trace metals showed no apparent improvement in sensitivity. For instance, La<sub>0.75</sub>Sr<sub>0.25</sub>Co<sub>0.33</sub>Ni<sub>0.33</sub>Fe<sub>0.33</sub>O<sub>3</sub>, La<sub>0.75</sub>Sr<sub>0.25</sub>Co<sub>0.33</sub>Ni<sub>0.33</sub>Mn<sub>0.33</sub>O<sub>3</sub>, and La<sub>0.75</sub>Sr<sub>0.25</sub>Co<sub>0.25</sub>Ni<sub>0.25</sub>Mn<sub>0.25</sub>Fe<sub>0.25</sub>O<sub>3</sub> presented a calibration sensitivity of 155 ± 3 μA mM<sup>-1</sup>, 536 ± 12 μA mM<sup>-1</sup>, and 513 ± 24 μA mM<sup>-1</sup>, respectively. Factors accounted for the sensitivity discrepancy may include, but not limited to, the ratio of oxidation states, the porous structures, mechanical stability, and surface area of the nanofibers. All the factors largely depend on the compositions and chemical properties of the perovskite nanofibers.

To understand the mechanisms behind the sensitivity connection with compositions, we have analysed the particle shapes and distributions in representatives SEM images (Fig. S2, ESI<sup>†</sup>). The results are shown in Table 2.

The compositions of LSCNO nanocomposites significantly affected shape, diameter, particle distribution, and porous



**Table 1** Calibration sensitivity and  $R^2$  values of nonenzymatic electrochemical glucose biosensors built with different compositions of perovskite LSCNO nanofibers

| Composition (label in Fig. 6)  | Calibration sensitivity ( $\mu\text{A mM}^{-1}$ ) | $R^2$  |
|--|---|--------|
| LaNiO <sub>3</sub> (i)   | 355 ± 7   | 0.9964 |
| La <sub>0.5</sub> Sr <sub>0.5</sub> NiO <sub>3</sub> (g)   | 429 ± 26  | 0.9719 |
| La <sub>0.75</sub> Sr <sub>0.25</sub> NiO <sub>3</sub> (f)   | 432 ± 13  | 0.9926 |
| La <sub>0.9</sub> Sr <sub>0.1</sub> NiO <sub>3</sub> (h)   | 418 ± 9   | 0.9964 |
| La <sub>0.75</sub> Sr <sub>0.25</sub> CoO <sub>3</sub> (j)   | 191 ± 5   | 0.9947 |
| La <sub>0.75</sub> Sr <sub>0.25</sub> Co <sub>0.5</sub> Ni <sub>0.5</sub> O <sub>3</sub> (a)   | 924 ± 28  | 0.9926 |
| La <sub>0.75</sub> Sr <sub>0.25</sub> Co <sub>0.25</sub> Ni <sub>0.75</sub> O <sub>3</sub> (b)                                       | 917 ± 45  | 0.9809 |
| La <sub>0.75</sub> Sr <sub>0.25</sub> Co <sub>0.75</sub> Ni <sub>0.25</sub> O <sub>3</sub> (c)                                       | 726 ± 62  | 0.9521 |
| La <sub>0.75</sub> Sr <sub>0.25</sub> Co <sub>0.33</sub> Ni <sub>0.33</sub> Fe <sub>0.33</sub> O <sub>3</sub> (k)                    | 155 ± 3   | 0.9964 |
| La <sub>0.75</sub> Sr <sub>0.25</sub> Co <sub>0.33</sub> Ni <sub>0.33</sub> Mn <sub>0.33</sub> O <sub>3</sub> (d)                    | 536 ± 12  | 0.9961 |
| La <sub>0.75</sub> Sr <sub>0.25</sub> Co <sub>0.25</sub> Ni <sub>0.25</sub> Mn <sub>0.25</sub> Fe <sub>0.25</sub> O <sub>3</sub> (e) | 513 ± 24  | 0.9821 |

structures. The porous structures were formed in the Co and Ni mixed perovskites provide higher surface areas. La<sub>0.75</sub>Sr<sub>0.25</sub>Co<sub>0.5</sub>Ni<sub>0.5</sub>O<sub>3</sub> nanocomposites show densely porous nanofiber feature with a well-distributed diameter  $0.15 \pm 0.03 \mu\text{m}$ . Consequently, a sensor prepared with this material has higher surface area and well-covered membrane, exhibiting enhanced sensitivity to glucose.

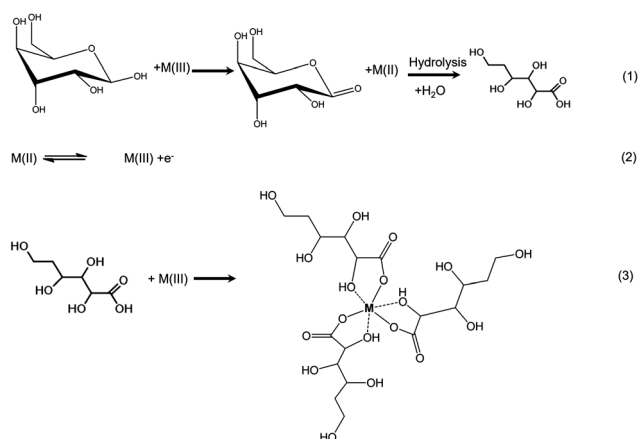
La<sub>0.75</sub>Sr<sub>0.25</sub>Co<sub>0.5</sub>Ni<sub>0.5</sub>O<sub>3</sub>, showed the highest calibration sensitivity of  $924 \pm 28 \mu\text{A mM}^{-1}$ , and good proportion of the variance,  $R^2 = 0.9926$ , in the linear range of 0.1–1.0 mM, therefore, this nanofiber material was selected to build the nonenzymatic glucose biosensor. As the standard error of the signal was obtained to be  $25.6 \mu\text{A}$ , a lower limit of detection (LLOD) was then calculated to be 0.083 mM. The La<sub>0.75</sub>Sr<sub>0.25</sub>Co<sub>0.5</sub>Ni<sub>0.5</sub>O<sub>3</sub> nanofibers described above were synthesized at the preselected calcination temperature 700 °C. In contrast, La<sub>0.75</sub>Sr<sub>0.25</sub>Co<sub>0.5</sub>Ni<sub>0.5</sub>O<sub>3</sub> nanofibers synthesized at 800 °C and 900 °C are also linearly responsive to glucose when immobilized at a carbon electrode surface, though the sensitivity decreases as the calcination temperature increases, and the peak features were changed (ESI, † Fig. S1).

A normal fasting blood glucose concentration is between  $70 \text{ mg dL}^{-1}$  ( $3.9 \text{ mmol L}^{-1}$ ) and  $100 \text{ mg dL}^{-1}$  ( $5.6 \text{ mmol L}^{-1}$ ). A level from 100 to  $125 \text{ mg dL}^{-1}$  ( $5.6$  to  $6.9 \text{ mmol L}^{-1}$ ) is considered prediabetes. A level higher than  $126 \text{ mg dL}^{-1}$  ( $7 \text{ mmol L}^{-1}$ ) on two separate tests allows the diagnosis of diabetes. Accordingly, the nonenzymatic electrochemical glucose biosensor developed in this research is highly sensitive probe which might be applicable in clinic glucose detection for all groups of the patients.

### 3.3 LSCNO nanofibers are electrochemical catalyst for glucose oxidation

Fig. 7 represents cyclic voltammograms of glucose measured on La<sub>0.75</sub>Sr<sub>0.25</sub>Co<sub>0.5</sub>Ni<sub>0.5</sub>O<sub>3</sub> nanofiber modified carbon electrode as the concentration increased from 0.1 to 1 mM. The anodic peak current ( $I_{p_a}$ ) was observed to increase proportionally with the increment of glucose concentration; as a result, it is feasible to employ  $I_{p_a}$  as a sensing parameter of this nonenzymatic glucose sensor.

As glucose is known to be electrochemically inactive within the potential window on a carbon electrode surface, the anodic peak observed above supports the assumption that glucose is catalytically oxidized on the LSCNO nanofiber modified carbon electrode. Similar mechanisms have been reported on carbon electrode modified with Ni-based metal and metal oxide electrodes.<sup>43,44</sup> The electrochemical oxidations were seen to occur in the same potential ranges of oxidation of Ni(II) or Co(II). The oxidation peaks of Ni(II) and Co(II) are overlapped and hardly separated on the selected LSCNO modified electrode. However, electrodes modified with different LSCNO compositions may present different features, sometime with separated peaks. As a result, we suggest a mediated electron transfer mechanism involving nickel or cobalt oxidation states,<sup>12</sup>



where M represents Ni or Co. (1) is a fast, while Step (2) is a slow and rate-determining step.

Although an increase of the cathodic peak current ( $I_{p_c}$ ) with glucose concentration was also observed, the change was much less significant than that of the anodic peak. We predict an increase on surface density of Ni(III) or Co(III) is responsible to the  $I_{p_c}$  increment. Glucose continuously diffuses into the nanostructures and then is oxidized to gluconolactone through

**Table 2** Particle shape and size distribution of perovskite LSCNO nanofibers with different compositions

| Nanofiber composition  | Diameter ( $\mu\text{m}$ ) | Predominant particle feature in SEM images |
|--|----------------------------|--|
| La <sub>0.75</sub> Sr <sub>0.25</sub> CoO <sub>3</sub>                                     | $0.11 \pm 0.02$            | Less porous nanofibers                     |
| La <sub>0.75</sub> Sr <sub>0.25</sub> Co <sub>0.75</sub> Ni <sub>0.25</sub> O <sub>3</sub> | $0.15 \pm 0.09$            | Porous nanofibers                          |
| La <sub>0.75</sub> Sr <sub>0.25</sub> Co <sub>0.5</sub> Ni <sub>0.5</sub> O <sub>3</sub>   | $0.15 \pm 0.03$            | Porous nanofibers                          |
| La <sub>0.75</sub> Sr <sub>0.25</sub> Co <sub>0.25</sub> Ni <sub>0.75</sub> O <sub>3</sub> | $0.20 \pm 0.05$            | Less porous nanofibers and nanosheets      |
| La <sub>0.75</sub> Sr <sub>0.25</sub> NiO <sub>3</sub>                                     | $0.10 \pm 0.02$            | Less porous nanofibers                     |



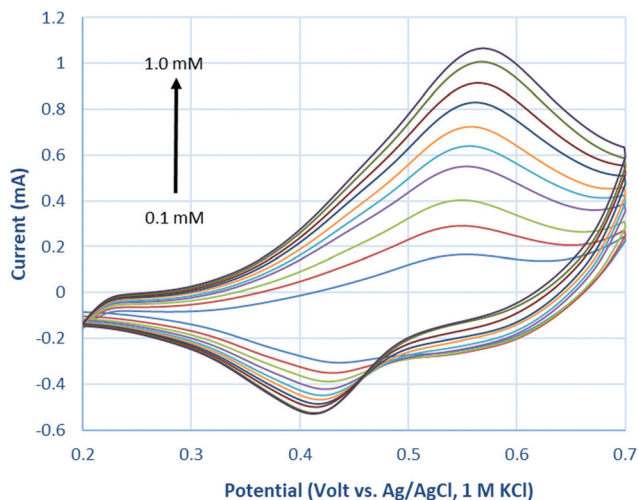


Fig. 7 Cyclic voltammograms of glucose on  $\text{La}_{0.75}\text{Sr}_{0.25}\text{Co}_{0.5}\text{Ni}_{0.5}\text{O}_3$  nanofibers-modified carbon electrode: from bottom (blue) to top (dark purple) positive scans, the CVs represent 0.1, 0.2, 0.3, 0.4, 0.5, 0.6, 0.7, 0.8, 0.9, and 1.0 mM glucose, respectively. Initial potential: 0.2 V, Final Potential: 0.7 V, scan rate:  $25 \text{ mV s}^{-1}$ .

Step (1). Gluconic acid has been formed through hydrolysis.  $\text{Ni(II)}$  or  $\text{Co(II)}$  are quickly formed through Step (1) to give a hike of  $\text{Ip}_a$ , *i.e.*, Step (2). In the meantime, gluconic acid accumulates in the porous LSCNO space of the sensor, chelating metal ions in the lattice sites of the non-stoichiometric perovskite nanomaterials. For example, more  $\text{Ni}^{3+}$  or  $\text{Co}^{3+}$  ions were released by Step (3) from the lattice sites to the electrode surface, giving to a slight peak increase of  $\text{Ip}_c$ .

The anodic peak potentials marginally shifted to the right and the cathodic peak potential to the left with increasing glucose concentration, owing to the increasing overpotential caused by the limited diffusion coefficient of glucose, leading to delayed mass transport at a higher concentration. For same reason, a similar phenomenon was also observed when increasing the potential scan rate in 1 mM glucose.

### 3.4 LSCNO nanofibers are responsive to other biomolecules

Other common biomolecules such as fructose, lactose, galactose, mannose, dopamine, and ascorbic acid were tested with the LSCNO nanofiber modified electrode. As shown in Fig. 8, although the sensitivity for those molecules is lower than that for glucose, these molecules also show linear response by the nonenzymatic biosensor. Similar behavior has also been reported for other metal oxide-based nonenzymatic glucose biosensors.<sup>27,28,38,45–49</sup> Like most available glucose biosensors, glucose monitor interference by the coexisting biomolecules should be considered in real sample measurements.<sup>50</sup> However, a biosensor that is sensitive to multiple carbohydrates may find application for multiple carbohydrates monitoring in human body fluid or fermentation processes.<sup>51–53</sup>

Sodium dodecyl sulphate (SDS), the most common anionic surfactant, is an amphiphilic compound that reduce water surface tension. SDS is commonly used as an active ingredient in household and personal care products as well as in

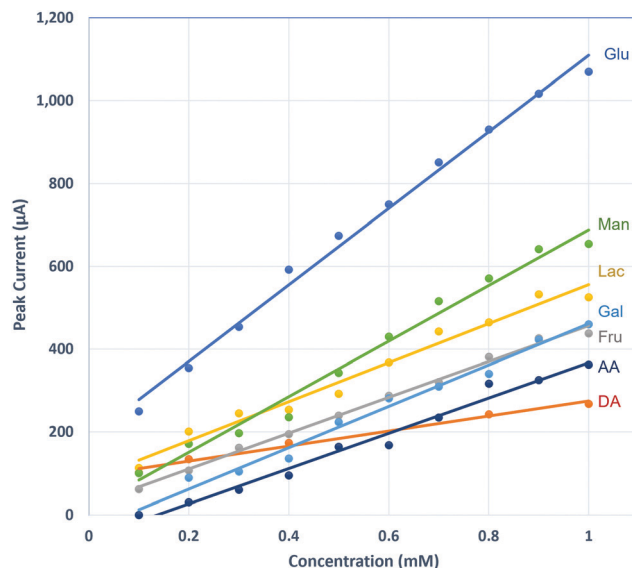


Fig. 8 Linear responses of LSCNO nanofiber modified electrode to common biomolecules: glucose (Glu), fructose (Fru), lactose (Lac), galactose (Gal), mannose (Man), dopamine (DA), and ascorbic acid (AA).

specialized applications to fabrics, carpets, and paper. To check if the SDS exposure could cause the interference to the LSCNO-based glucose biosensor, 0.1% SDS solution was tested but did not show any peak. Presence of 0.1% SDS in a 1 mM glucose solution did not significantly change the CV feature and  $\text{Ip}_a$  value, suggesting such sensor was not interfered by SDS.

To examine the feasibility of the prepared sensor for real human fluid samples, we examined a urine sample that was collected at 3 hours after meal (550 kcal) (Fig. S3, ESI†). The subject has no clinical diabetes treatment including insulin or any other medicines. To avoid the matrix effect from the urine fluid, standard addition method was selected in this test. Glucose was found to be  $0.59 \pm 0.03_6$  mM in the sample. This value is in accord with the normal urine glucose range 0–0.8 mM for a healthy person, suggesting the potential to use this sensor for clinical monitoring.

Other sugars, such as fructose, are at lower concentration level ( $\mu\text{mol}$  daily or  $\mu\text{M}$ ) in urine,<sup>54</sup> which will not cause significant interference to the determination of glucose, considering its lower sensitivity for these sugars than for glucose (Fig. 8). As the sensitivity for ascorbic acid is much lower than that of glucose (Fig. 8), a low level of ascorbic acid in urine would not significantly affect the glucose results. However, in a sample with high ascorbic acid level, an independent analysis of ascorbic acid would be necessary to avoid a false positive result.

## 4. Conclusions

In this work, we fabricated porous LSCNO nanofibers with nanosized polycrystalline grains with high crystallinity, which resulted from homogeneous nucleation and the spatial confinement of the nanofibers. For the first time, LSCNO-based



nonenzymatic electrochemical glucose biosensors have been developed, following an electrochemically catalytic mechanism, a mediated electron transfer involving Ni(II)/Ni(III) and Co(II)/Co(III). LSCNO glucose biosensors presented high sensitivity, which was significantly affected by the metal oxide compositions. An optimized composition,  $\text{La}_{0.75}\text{Sr}_{0.25}\text{Co}_{0.5}\text{Ni}_{0.5}\text{O}_3$  was selected to compose the sensor. This nanocomposite has densely porous nanofiber texture with a well-distributed diameter ( $0.15 \pm 0.03 \mu\text{m}$ ), which causes a high surface area and well-covered sensor surface, leading to the highest sensitivity. The  $\text{La}_{0.75}\text{Sr}_{0.25}\text{Co}_{0.5}\text{Ni}_{0.5}\text{O}_3$  immobilized electrode showed linear response to glucose in the range of 0.1–1.0 mM, along with a calibration sensitivity of  $924 \pm 28 \mu\text{A mM}^{-1}$ , a  $R^2$  of 0.9926, and a LLOD of 0.083 mM, an outstanding analytical performance that is appropriate for a clinical test. The biosensor showed no response to SDS, the most widely used anionic surfactant, but was linearly responsive to other biomolecules, such as fructose, lactose, galactose, mannose, dopamine, and ascorbic acid. Examination of a urine sample by standard addition method showed a  $0.599 \pm 0.036 \text{ mM}$  glucose, which is in accord with the normal urine glucose range for a health person. Consequently, this novel nonenzymatic electrochemical biosensor may also be appealing to the detection of multiple biomolecules.

## Author contributions

KDP and TAH conducted the nanomaterials preparation, characterization, and biosensor evaluation. GG guided KDP and TAH on electrospinning, and conducted XRD, and XPS experiments, materials synthesis and characterization, as well as manuscript preparation and revision. YL and JW conducted the TEM work. SRE participated in research discussion and XPS data analysis. TAH and MA conducted SEM analysis. ZL participated in project design, experiment supervision, and paper revision. SH supervised the research activity, conducted the data analysis and evaluation, and participated in paper preparation and revision.

## Conflicts of interest

There are no conflicts to declare.

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## References

- H. Zhu, L. Li, W. Zhou, Z. Shao and X. Chen, *J. Mater. Chem. B*, 2016, **4**, 7333–7349.
- Z. Li, F. Gao and Z. Gu, *Sens. Actuators, B*, 2017, **243**, 1092–1101.
- Y. Zhang, Y. Q. Liu, Y. Bai, X. L. Li and W. Chu, *Appl. Surf. Sci.*, 2021, 539.
- K. Tian, K. Baskaran and A. Tiwari, *Vacuum*, 2018, **155**, 696–701.
- L. Sinha, S. Pakhira, P. Bhojane, S. Mali, C. K. Hong and P. M. Shirage, *ACS Sustainable Chem. Eng.*, 2018, **6**, 13248–13261.
- M. Palmer, M. Masikini, L. W. Jiang, J. J. Wang, F. Cummings, J. Chamier, O. Inyang and M. Chowdhury, *J. Alloys Compd.*, 2021, 853.
- G. George and S. Anandhan, *Thin Solid Films*, 2016, **610**, 48–57.
- E. Bahrami, R. Amini and S. Vardak, *J. Alloys Compd.*, 2021, 855.
- Y. Y. Li, P. Kang, S. Q. Wang, Z. G. Liu, Y. X. Li and Z. Guo, *Sens. Actuators, B*, 2021, 327.
- J. Liu, L. L. Sun, G. L. Li, J. Hu and Q. G. He, *Mater. Res. Bull.*, 2021, 133.
- T. Tite, E. A. Chiticaru, J. S. Burns and M. Ionita, *J. Nanobiotechnol.*, 2019, 17.
- Q. Cheng, L. D. Ji, K. B. Wu and W. K. Zhang, *Sci. Rep.*, 2016, 6.
- R. Abdel-Karim, Y. Reda and A. Abdel-Fattah, *J. Electrochem. Soc.*, 2020, 167.
- D. M. Bastidas, S. W. Tao and J. T. S. Irvine, *J. Mater. Chem.*, 2006, **16**, 1603–1605.
- S. Afroze, A. Karim, Q. Cheok, S. Eriksson and A. K. Azad, *Front. Energy*, 2019, **13**, 770–797.
- X. Q. Lang, H. Y. Mo, X. Y. Hu and H. W. Tian, *Dalton Trans.*, 2017, **46**, 13720–13730.
- G. George, S. L. Jackson, C. Q. Luo, D. Fang, D. Luo, D. L. Hu, J. G. Wen and Z. P. Luo, *Ceram. Int.*, 2018, **44**, 21982–21992.
- Y. Liu, Z. B. Wang, Y. J. Zhong, X. M. Xu, J. P. M. Veder, M. R. Rowles, M. Saunders, R. Ran and Z. P. Shao, *Chem. Eng. J.*, 2020, 390.
- P. Tan, M. L. Liu, Z. P. Shao and M. Ni, *Adv. Energy Mater.*, 2017, 7.
- L. Kong, X. Chen, B. Q. Li, H. J. Peng, J. Q. Huang, J. Xie and Q. Zhang, *Adv. Mater.*, 2018, 30.
- F. F. Jia, H. Zhong, W. G. Zhang, X. R. Li, G. Y. Wang, J. Song, Z. P. Cheng, J. Z. Yin and L. P. Guo, *Sens. Actuators, B*, 2015, **212**, 174–182.
- J. He, J. Sunarso, Y. Zhu, Y. Zhong, J. Miao, W. Zhou and Z. Shao, *Sens. Actuators, B*, 2017, **244**, 482–491.
- K. Y. Wang, J. Z. Song, X. J. Duan, J. S. Mu and Y. Wang, *New J. Chem.*, 2017, **41**, 8554–8560.
- G. George, S. R. Ede and Z. Luo, *Fundamentals of Perovskite Oxides: Synthesis, Structure, Properties and Applications*, CRC Press, 2020.



- 25 S. R. Ede, C. N. Collins, C. D. Posada, G. George, H. Wu, W. D. Ratcliff, Y. Lin, J. Wen, S. Han and Z. Luo, *ACS Catal.*, 2021, **11**, 4327–4337.
- 26 Y. Wang, Y. Xu, L. Luo, Y. Ding and X. Liu, *J. Electroanal. Chem.*, 2010, **642**, 35–40.
- 27 Q. Dong, H. Ryu and Y. Lei, *Electrochim. Acta*, 2021, **370**, 137744.
- 28 M. H. Hassan, C. Vyas, P. Bartolo and B. Grieve, *Sensors*, 2021, **21**.
- 29 M. Wei, Y. Qiao, H. Zhao, J. Liang, T. Li, Y. Luo, S. Lu, X. Shi, W. Lu and X. Sun, *Chem. Commun.*, 2020, **56**, 14553–14569.
- 30 R. Wilson and A. P. F. Turner, *Biosens. Bioelectron.*, 1992, **7**, 165–185.
- 31 N. J. Ronkainen, H. B. Halsall and W. R. Heineman, *Chem. Soc. Rev.*, 2010, **39**, 1747–1763.
- 32 Y. Wang, Y. Xu, L. Luo, Y. Ding, X. Liu and A. Huang, *Sens. Actuators, B*, 2010, **151**, 65–70.
- 33 E. H. El-Ads, A. Galal and N. F. Atta, *J. Electroanal. Chem.*, 2015, **749**, 42–52.
- 34 E. H. El-Ads, A. Galal and N. F. Atta, *RSC Adv.*, 2016, **6**, 16183–16196.
- 35 M. Sivakumar, K. Pandi, S.-M. Chen, Y.-H. Cheng and M. Sakthivel, *New J. Chem.*, 2017, **41**, 11201–11207.
- 36 N. F. Atta, A. Galal and E. H. El-Ads, *J. Electroanal. Chem.*, 2019, **852**, 113523.
- 37 I. Boubezari, A. Zazoua, F. Bessueille, A. Errachid and N. Jaffrezic-Renault, *Electroanalysis*, 2020, **32**, 1642–1650.
- 38 T.-W. Chen, R. Ramachandran, S.-M. Chen, G. Anushya and K. Ramachandran, *Sensors*, 2020, **20**, 6755.
- 39 A. Senthamizhan, B. Balusamy and T. Uyar, *Anal. Bioanal. Chem.*, 2016, **408**, 1285–1306.
- 40 K. Puttananjegowda, A. Taksi and S. Thomas, *J. Electrochem. Soc.*, 2020, **167**, 37553.
- 41 C. Peniche, D. Zaldivar, M. Pazos, S. Paz, A. Bulay and J. S. Roman, *J. Appl. Polym. Sci.*, 1993, **50**, 485–493.
- 42 G. George and S. Anandhan, *J. Sol-Gel Sci. Technol.*, 2013, **67**, 256–266.
- 43 S. Berchmans, H. Gomathi and G. P. Rao, *J. Electroanal. Chem.*, 1995, **394**, 267–270.
- 44 D.-W. Hwang, S. Lee, M. Seo and T. D. Chung, *Anal. Chim. Acta*, 2018, **1033**, 1–34.
- 45 D. Chen, S. Wang, M. Liu, J. Gao, H. Song and S. Zhang, *J. Electrochem. Soc.*, 2019, **166**, B1653–B1659.
- 46 M. A. Yassin, B. K. Shrestha, R. Ahmad, S. Shrestha, C. H. Park and C. S. Kim, *J. Ind. Eng. Chem.*, 2019, **73**, 106–117.
- 47 S. Shendelman, A. Jonason, C. Martinat, T. Leete and A. Abeliovich, *PLoS Biol.*, 2004, **2**, 1764–1773.
- 48 Y. Shu, B. Li, J. Chen, Q. Xu, H. Pang and X. Hu, *ACS Appl. Mater. Interfaces*, 2018, **10**, 2360–2367.
- 49 V. E. Coyle, A. E. Kandjani, M. R. Field, P. Hartley, M. Chen, Y. M. Sabri and S. K. Bhargava, *Biosens. Bioelectron.*, 2019, **141**, 111479.
- 50 V. B. Juska and M. E. Pemble, *Sensors*, 2020, **20**, 6013.
- 51 D. Lin, S. W. Yang, C. L. Hsieh, K. J. Hsu, T. X. Gong, Q. Wu, S. F. Qiu, S. Y. Feng and K. V. Kong, *ACS Sens.*, 2021, **6**, 1240–1247.
- 52 J. Tkac, P. Gemeiner, J. Svitel, T. Benikovsky, E. Sturdik, V. Vala, L. Petrus and E. Hrabarova, *Anal. Chim. Acta*, 2000, **420**, 1–7.
- 53 Y. Sun, S. Han, X. Song and H. Sun, *Fenxi Yiqi*, 1997, 12–15.
- 54 X. Song, S. L. Navarro, P. Diep, W. K. Thomas, E. C. Razmpoosh, Y. Schwarz, C. Y. Wang, M. Kratz, M. L. Neuhauser and J. W. Lampe, *Nutr. Res.*, 2013, **33**, 696–703.

