



Polyelectrolyte character of rigid rod peptide bundlemer chains constructed via hierarchical self-assembly

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33 subtle balance between short-range attraction and long-range repulsion interactions in proteins and other
34 biomaterials. With computational control of constituent peptide sequences, it is further possible to
35 deconvolute the underlying sequence driven structure-property relationships in the modular bundlemer
36 chains.

37 **Introduction**

38 Supramolecular self-assembly of biological building blocks into multifunctional materials that
39 display hierarchical structural organization is ubiquitous in nature.^{1,2} Synthetic biomolecule self-assembly
40 has emerged as a bioinspired, powerful tool for the bottom-up construction of materials.³ While empirical
41 sequence design of non-natural *de novo* biomolecules based on DNA,⁴ lipids⁵ and proteins⁶ to construct
42 non-natural biomaterials has been widely successful, introduction of the computational toolbox for
43 defining biomolecule design has added an exciting new dimension to material discovery while also
44 enabling tunability of structure and function at the nanoscale in a systematic manner. Consequently, *in*
45 *silico* computational design tools are contributing significantly to the flourishing science of DNA
46 origami^{7,8} and to supramolecular protein assembly,⁹ especially via protein-protein interface modification¹⁰
47 and docking algorithms.¹¹⁻¹³ However, much work remains in terms of sequence-based design and
48 structure predictions for peptide and protein folding and assembly, which is complicated by multiple local
49 interactions, large numbers of potential conformations, and sequence-dependent structural features, the
50 sum of which is commonly referred to as the ‘protein folding problem’.¹⁴ As a result, the potential of *in*
51 *silico* computational design of protein-based biomaterials remains vastly unexploited.

52 Recently, coiled coils, tertiary or quaternary structural domains frequently found in natural
53 proteins, have garnered attention as candidates for novel biomaterial construction.^{15, 16} In nature, coiled
54 coils are commonly left-handed bundles of two to six α -helical peptides, each of which are based on a
55 repeat motif of seven amino acid residues called the heptad (*-abcdefg-*). The coiling of this sequence into
56 a helix produces a ‘hydrophobic stripe’ along one face of the peptide (amino acids at positions *a* and *d*),
57 which promotes assembly of these peptides into supercoiled bundles, burying their hydrophobic faces
58 while exposing the more hydrophilic residues (amino acids at positions *b,c,e,f* and *g*) to the aqueous

59 solvent. In case of synthetic coiled coil forming peptides, the solvent-exposed groups can be carefully
60 picked or modified *in silico* such that the resulting inter-bundle interactions can direct bundlemer self-
61 assembly into predetermined nanostructures.¹⁷⁻¹⁹ Important advantages of using coiled coil-based
62 building blocks for nanomaterial construction include assembly that is triggered under mild aqueous
63 conditions, robustness of the bundle assembly, and the use of nonbiological synthesis techniques.
64 Importantly, non-natural amino-acid and arbitrary chemical functional group incorporation is enabled for
65 strategic placement on solvent-exposed bundle sites, facilitating new pathways for hierarchical
66 biomaterial construction. Consequently, advances in the computational design of coiled coils has
67 leveraged their well-understood structural parameter space and cylindrical structure.²⁰⁻²³

68 Zhang et al. demonstrated that computationally designed peptide sequences that form robust,
69 antiparallel homotetrameric coiled coils, i.e. four identical helices packed in an antiparallel fashion
70 providing the bundle with D_2 symmetry, successfully assembled into non-natural 2D lattices with a
71 targeted crystal symmetry.²⁴ The self-assembled nanostructure of the artificial bundles was responsive to
72 aqueous solution conditions such as pH, salt and temperature.^{25, 26} Haider et al. confirmed that these
73 computationally designed bundles have the structure of nano-cylinders that are ≈ 40 Å in length and ≈ 20
74 Å in diameter in good agreement with their expected computationally designed dimensions.²⁷ Exploiting
75 the robustness of these bundles, Wu et al. modified the N-termini of the peptides to include either a free
76 thiol (via addition of a cysteine) or a free maleimide (Mal-) group resulting in two tetra-functional
77 bundlemers (see **Figure 1**). When the thiol and maleimide-functional bundlemers were reacted via Thiol-
78 Michael *click* reaction, bundlemer chains that have a 20 Å cross-section were formed having extremely
79 long persistence lengths (i.e., on the order of tens of micrometers), herein referred to as rigid rods.²⁸
80 Furthermore, the *click* chemistry-based construction pathway allowed use of a variety of linker moieties
81 and facilitated the control of bundlemer chain length, flexibility, architecture (such as introduction of
82 kinks) and branching; this synthetic approach yields a library of tunable 1D nanomaterials.

83

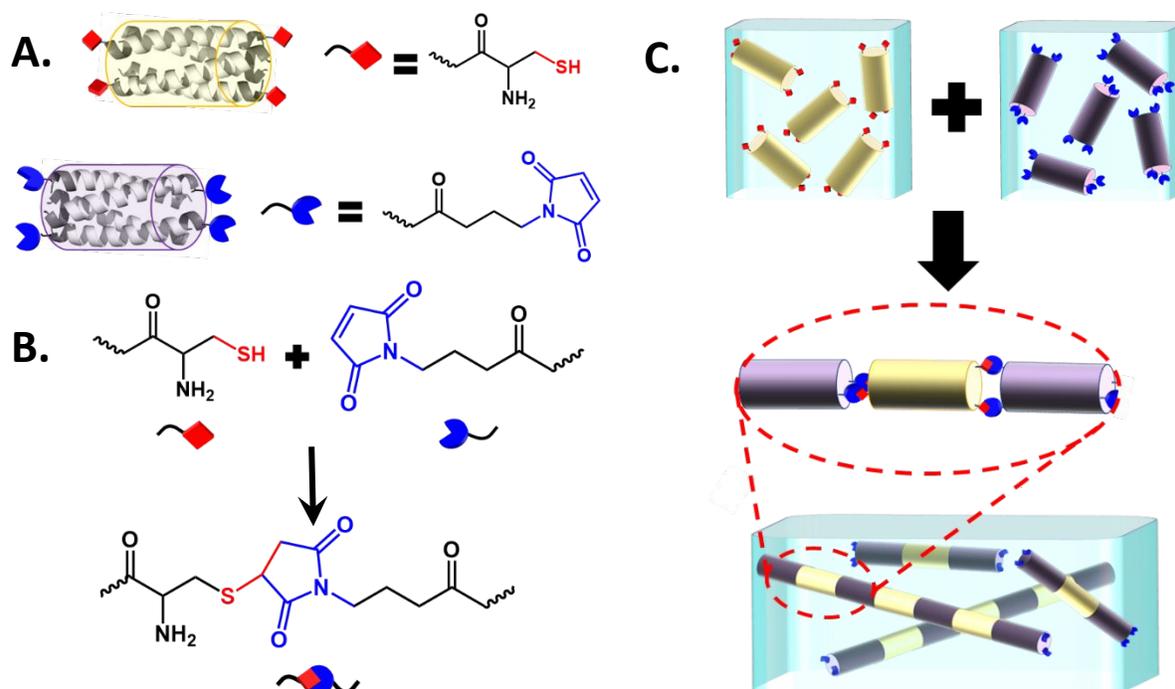


Figure 1: (A) Schematic of homotetramer coiled coils showing the bundlers **Peptide 1** (purple) decorated with maleimide groups (blue cut circles) and **Peptide 2** (yellow) decorated with thiol groups (red diamonds). (B) Reaction scheme of Thiol-Michael *click* reaction between a thiol group (on cysteine) and maleimide group that yields a thiol-maleimide adduct. (C) Reaction protocol for hybrid physical-covalent assembly via Thiol-Michael *click* reaction between assembled bundlers. The thiol-decorated **Peptide 2** bundlers (yellow with red diamonds) are reacted with Mal-decorated **Peptide 1** bundlers (purple with blue cut circles) resulting in formation of rigid rod-like chains with alternating **Peptide 1** and **Peptide 2** bundlers.

84

85 The bundlemer chains include physical interactions to stabilize individual bundlers and
 86 covalent connections for hierarchical rigid rod formation. These chains and their hierarchical assembly
 87 pathways present unique opportunities to utilize them as model systems to study structure-property
 88 relationships of specific amino acid sequences, a knowledge critical in furthering our understanding of 1D
 89 assemblies such as collagen, microtubules, Tobacco Mosaic Virus (TMV), fd bacteriophage, actin and
 90 amyloid fibers²⁹⁻³¹. These biological assemblies are usually rod-like polyelectrolytes that are inherently
 91 different from their synthetic counterparts, such as colloidal gold nanorods, wormlike micelles and simple
 92 polymeric polyelectrolytes such as polystyrene sulfonate, in that they display patchiness in local
 93 interactions due to multiple polar, hydrophobic, and oppositely charged side groups.³²⁻³⁴ The complexity
 94 in interactions in biological assemblies is readily mimicked in the computationally-designed bundlers

95 due to their sequence tunability, folding, and assembly. We present herein a study of the solution-state
96 structure and interactions of bundlemer rigid rods to shed light on specific sequence driven structure-
97 property relationships relevant to rod-like biological assemblies. Small-angle neutron scattering (SANS)
98 is uniquely suited to studying the solution structure of soft matter systems in a non-destructive manner
99 and can be used alongside transmission electron microscopy (TEM) as together, they unambiguously
100 yield information about the geometric shape of the nanomaterial while also giving insight into interactions
101 between components in solution (obtained via SANS). Three different rigid rod systems with increasing
102 average rod lengths have been investigated for their solution-state structure as a function of peptide
103 concentration. For the long rigid rod system, the impact of monovalent sodium chloride (NaCl) salt and
104 acidic pH on the solution structure has also been investigated. For the interpretation of the SANS data, we
105 model the rigid rods as straight cylinders that are repulsive at low ionic strength and neutral pH conditions
106 using a mean-field expression derived by Schneider et al.³⁵ Clusters of rods that are evidenced by a
107 characteristic upturn corresponding to large length scales in SANS curves are modeled using Teixeira's
108 fractal scattering expression.³⁶ Implications of the composite SANS modeling results and important
109 similarities and differences between the rigid rods and other 1D assemblies are discussed.

110

111 **Experimental section**

112 **1. Synthesis and purification of bundlemers:**

113 Two computationally designed peptide sequences that have been previously reported to form
114 bundlemers were employed to construct rigid rods via a hybrid physical-covalent self-assembly
115 pathway.^{24, 27} Here, these peptide sequences were modified to include functional groups that enable
116 assembly of polymers of covalently linked bundlemers. The peptide sequences, **Peptide 1** with a
117 maleimide (Mal)-modified N-terminus and amidated C-terminus (Mal DEIKNM ADQIKHM
118 AWMIDRM AEKIDRE A -NH₂) and **Peptide 2** with a cysteine at the N-terminus and amidated C-
119 terminus (C DEEIRRM AEEIRQM AERIQQM AEQIQQE A -NH₂) were synthesized as reported
120 previously.^{24,28} Specifically, peptides were synthesized at a 0.25×10^{-3} mol scale via microwave-assisted

121 fmoc-based solid-phase peptide synthesis (fmoc-SPPS) on a CEM Liberty Blue™ Automated Microwave
122 Peptide Synthesizer. Rink-amide resin (ChemPep) that yields an amidated C-terminus after the final
123 cleaving step (see below) was utilized for all syntheses. Amino-acids were purchased from ChemPep and
124 used as is. The solvent for all synthesis steps was analytical-grade di-methyl formamide (DMF) which
125 was purchased from Fischer scientific. Coupling reagents 0.5 mol/L N, N-diisopropylcarbodiimide (DIC,
126 Sigma) and 0.5 mol/L ethyl 2-cyano-2-(hydroxyimino) acetate (Oxyma, CEM) were prepared in DMF
127 while piperidine (Sigma) in DMF having a volume fraction of 20 % was utilized for deprotection cycles.
128 Default microwave and heating cycles were employed for the coupling and deprotection steps which are
129 also recommended by CEM. For **Peptide 1**, the functionalizing agent N-carboxy propyl maleimide
130 (Sigma) was coupled to the N-terminus of the peptidoresin by performing an additional amino-acid
131 coupling cycle on the instrument itself at the end of the synthesis.

132 The final peptidoresin was washed multiples times with analytical-grade dichloromethane (DCM,
133 Fischer) and fresh DMF. The peptide was then cleaved from the resin by suspending the peptidoresin in
134 10 ml cleaving solution containing a volume fraction of 95 % trifluoroacetic acid (TFA, Sigma), 2.5 %
135 Milli-Q water and 2.5 % triisopropylsilane (TIPS, Sigma) and shaking the cocktail for 2 hours. The
136 cleaved peptide solution was collected and precipitated using fresh anhydrous ethyl-ether (Fischer) at
137 least three times. The crude was dried, resuspended in solvent containing a volume fraction of 70 % water
138 and 30 % acetonitrile (HPLC-grade, Fischer) and lyophilized for subsequent purification.

139 The lyophilized crude peptide was dissolved in optimal water-acetonitrile solvent and purified on
140 a high-performance liquid chromatography (HPLC) instrument (Quaternary Gradient Module (Waters
141 2545), Waters Corporation) using a reverse-phase BEH130 Prep C18 10 µm column (XBridge, Waters
142 Corporation, Milford, MA). The elution gradient employed was 85 % solvent A to 15 % solvent B over
143 60 minutes, where solvent A was Milli-Q Water containing 0.1 % TFA and solvent B was acetonitrile
144 with 0.1 % TFA (all percentages are volume fractions). The elutant was continuously monitored for
145 absorbance at 214 nm and 280 nm using a UV-Vis photo-detector (Waters 2489, Waters Corporation).
146 The peptide elution peak was collected and checked for the correct molecular weight using electron-spray

147 ionization mass spectroscopy (ESI-MS) on Waters Xevo G2-XS QToF Quadrupole Time-of-Flight Mass
148 Spectrometry instrument (see SI for ESI-MS results). The purified peptides were then combined and
149 lyophilized.

150

151 **2. Hybrid physical-covalent assembly of bundlemers:**

152 Due to the antiparallel packing of four identical α -helical peptides within a coiled coil bundle, N-
153 termini modification yielded two functional groups at each end of the bundlemer. These bundlemers were
154 reacted together via Thiol-Michael *click* reaction resulting in polymers comprised of alternating **Peptide 1**
155 and **Peptide 2** bundlemers linked by two thiol-maleimide adducts between them (**Figure 1**). For this
156 study, the stoichiometric ratio r of thiol- groups afforded by cysteines on **Peptide 2** to Mal-groups on
157 **Peptide 1** for constructing polymeric rigid rods having a length distribution centered at a desired length L ,
158 *i.e.* short ($r=0.50$, $L\sim 105$ Å), medium ($r\sim 0.88$, $L\sim 525$ Å) and long ($r\sim 0.94$, $L\sim 1085$ Å), were calculated
159 using Flory's equation assuming full conversion (See SI).^{28,37}

160 A step-wise hybrid physical-covalent assembly pathway was utilized to construct rigid rods of
161 bundlemers that has been optimized and reported in our previous publication. Stock solutions of
162 lyophilized pure peptides were prepared by suspending them in Milli-Q water at 1×10^{-3} mol/L peptide
163 concentration. Tricarboxyethylphosphine hydrochloride (TCEP.HCl, Sigma) was added to **Peptide-2**
164 stock solution at a final concentration of 0.5×10^{-3} mol/L to prevent disulfide formation via oxidation of
165 the free thiol groups on the cysteine.

166 Calculated volumes of **Peptide 1** and **Peptide 2** stock solutions required to assemble rigid rods of
167 target length distributions were mixed together and reacted overnight under constant shaking at 30°C. The
168 success of the reaction was confirmed via ESI-MS wherein peaks corresponding to the **Peptide 1-2** dimer
169 of the correct molecular weight were observed due to formation of the Thiol-Michael adduct between
170 bundlemers (see SI for discussion on ESI-MS). The resulting solution containing the assembled rigid rods
171 was lyophilized and re-suspended in Deuterium Oxide (D₂O, atom fraction of 99.5 %, Aldrich) at desired
172 concentration for subsequent TEM and SANS characterization.

173

174 3. Characterization of rigid rod-like bundlemers**175 a. Transmission Electron Microscopy (TEM):**

176 Before TEM analyses, samples were diluted to a mass by volume fraction of 0.1 %. Sample grids
177 for analyses were prepared on 200-mesh carbon-coated copper grids purchased from Electron Microscopy
178 Solutions Inc. All grids were plasma treated under vacuum for 30 seconds and used within 2 minutes of
179 treatment. A 3-5 μL drop of the respective rod sample was added to the carbon-coated side of a plasma-
180 treated grid. After 60 seconds, excess sample was blotted and 3-5 μL of freshly prepared 1 % (mass by
181 volume fraction) sodium phosphotungstate staining solution (NaPWO_4 , pH 7.0) immediately was added
182 to the same side of the grid. The excess stain solution was blotted after 60 seconds and the sample grid
183 was dried under air for at least 15 minutes before TEM analyses. TEM was performed on FEI Talos
184 F200X Transmission Electron Microscope using a 200×10^3 V accelerating voltage. A Ceta 16 M camera
185 from Gatan was used to record 4 k x 4 k images. ImageJ analysis software was utilized to measure rod
186 lengths on the micrograph and calculate length statistics for each rigid rod sample.³⁸

187

188 b. Small-Angle Neutron Scattering (SANS):

189 Samples for SANS measurements were prepared by suspending lyophilized rigid rod samples of a
190 target length (short, medium or long) in D_2O and subsequent dialysis against 50 ml D_2O using Thermo
191 Scientific Dialysis Devices (3500 Da Molecular weight cutoff) to remove any small molecule,
192 unassembled peptide, and unreacted bundlemer impurities. The sample with the highest concentration of
193 rigid rods was measured first and subsequently diluted to a lower concentration using fresh D_2O for the
194 next measurement. At each dilution step, the samples were degassed (5 minutes) and sonicated (10
195 minutes) in a sonication bath at 30°C and rested for at least 30 minutes prior to SANS measurement. All
196 samples and pure D_2O had a pH in the 6 to 7 range as measured using Hydrion pH paper. For studies of
197 the effect of ionic strength on the long rods, a concentrated aqueous solution of sodium chloride (NaCl ,
198 salt) was added resulted in a salt concentration of 200×10^{-3} mol/L. Similarly, for the acidic pH study,

199 deuterium chloride (DCl, atom fraction of 99 % D, Sigma-Aldrich) in D₂O was added to the rigid rod
200 solution to prepare a 200 x 10⁻³ mol/L DCl sample with an acidic pH of ca. 1.0 as measured by Hydriion
201 pH paper.

202 SANS experiments were performed at the National Institute of Standards and Technology (NIST)
203 Center for Neutron Research, Gaithersburg, Maryland, USA. Beamlines NG7 SANS and CHRNS
204 NGB30m were utilized for conducting the measurements. A wavelength of 6 Å of cold neutrons with a
205 wavelength spread of ca. 12 % was employed. Sample-to-detector distance of 1 m and 4 m were utilized
206 to collect scattering data in the high to mid- Q region ($0.01 < Q < 0.3 \text{ \AA}^{-1}$). Here, Q is the momentum
207 transfer of the scattered neutron, defined as $Q = \frac{2\pi \sin\theta}{\lambda}$, where λ is the wavelength of the incoming beam
208 of neutrons and 2θ is the scattering angle. For low Q data collection, samples having peptide
209 concentration C greater than a mass by volume fraction of 0.5 %, were measured with either 13 m
210 sample-to-detector distance configuration and neutron lenses on NGB30 beamline or with 15 m sample-
211 to-detector-distance and neutron lenses on the NG7 beamline, both using a neutron wavelength of 8 Å,
212 resulting in a low- Q cutoff of 0.002 Å⁻¹. Due to low counting statistics for rod samples with concentration
213 C less than mass by volume fraction of 0.5 %, low- Q scattering data was collected at 13 m sample-to-
214 detector distance without the use of neutron lenses, resulting in a larger low- Q cutoff of 0.005 Å⁻¹.

215 All scattering data were reduced using NIST NCNR's IgorPro data reduction software³⁹. For
216 initial analyses, SASView scattering data analysis software was utilized.⁴⁰ For in-depth data modeling, an
217 in-house python code was written that uses a non-linear least-squares package to fit the normalized SANS
218 data to model $I(Q)$ expressions described in the next section. Best fit results with the lowest possible
219 reduced chi-squared (χ^2_R) values were obtained after multiple fitting routines. The mean and standard
220 deviation (σ) for each fit parameter was calculated and is summarized along with the reduced chi-squared
221 values in the **SI**. Error bars in the SANS data and fit parameters represent $\pm 1\sigma$ throughout this article.

222

223 **Theory:**

224 **SANS data modeling:**

225 In a SANS experiment, particles of interest are dispersed in a suitable solvent and bombarded by
 226 a coherent beam of neutrons. The neutron scattering pattern is given as the scattering intensity as a
 227 function of the scattering vector, Q . For a monodisperse solution of isotropically scattering particles, the
 228 resulting scattering intensity profile $I(Q)$ is given by:⁴¹

$$229 \quad I(Q) = nV_p^2(\rho_p - \rho_s)^2 P(Q).S(Q) \quad (1)$$

230 where n is the number density of scatterers having a volume V_p and scattering length density ρ_p dispersed
 231 in a solvent with scattering length density ρ_s . The shape and size of the scatterers is captured in the form
 232 factor $P(Q) = \langle A^2(Q) \rangle$, where $A(Q)$ is the scattering amplitude, and the brackets denote an average over
 233 particle orientations, a requirement for isotropic scattering. The effect of the interparticle interactions is
 234 encoded in the structure factor $S(Q)$. We model the form factor of the rigid rods as straight cylinders.²⁸
 235 For particles with such anisotropic shapes, $S(Q)$ in eq. 1 is approximated by an effective structure factor
 236 $\hat{S}(Q)$ to keep eq. (1) consistent with isotropic scattering.⁴¹ Kotlarchyk and Chen have derived the $\hat{S}(Q)$ for
 237 cylinders, given by:⁴²

$$238 \quad \hat{S}(Q) = 1 + \beta(Q)(S(Q) - 1) \quad (2)$$

239 Where the factor $\beta(Q) = \langle A(Q) \rangle^2 / \langle (A(Q))^2 \rangle$ (see eq. 3 and 4) is the decoupling approximation
 240 expression which accounts for the assumption that the position and orientation of the anisotropic rigid
 241 rods is decoupled. These considerations for model fitting are important also because the 2D scattering
 242 intensity obtained via SANS for all samples were radially symmetric, which is indicative of an isotropic
 243 solution of rod-like particles at corresponding length scales.

244 The bundlemer rigid rods were modeled as straight cylinders having a radius R and length L . The
 245 form factor of an isotropic dispersion of cylinders is given by:⁴¹

$$246 \quad P_{\text{cyl}}(Q) = \langle A^2(Q) \rangle = \int_0^{\pi/2} A^2(Q,\alpha) \sin \alpha \, d\alpha \quad (3)$$

247 Here, $A(Q)$ is the amplitude function of a cylinder, given by:

$$248 \quad A(Q, \alpha) = \frac{\sin(QL \cos(\alpha)/2) 2J_1(QR \sin(\alpha))}{QL \cos(\alpha)/2 \cdot QR \sin(\alpha)} \quad (4)$$

249 Equations 3 and 4 together indicate that the scattering intensity I follows the power law $I \sim Q^{-l}$ for the Q -
 250 range $2\pi L^{-1} < Q < R^{-1}$. If the rods are non-interacting, a Guinier region with $I(Q) = I(0) \exp(-Q^2 R_g^2$
 251 $/3)$ is observed for $QR_g < 1$, where R_g is the radius of gyration of the monodisperse rod particles.⁴¹ For
 252 the rigid rods however, the individual rods interact and form cluster aggregates indicated by the low- Q
 253 upturn, and their R_g can therefore not be determined from the data. Also, the length L cannot be
 254 determined, but, rather, an apparent length, L_a , that gives the characteristic rod length between points of
 255 contact in the cluster. The upturn follows a power law, *i.e.* $I \sim Q^{-D}$ where $2 < D < 3$. This can be attributed
 256 to the presence of mass fractal-like networked cluster of rods with the fractal dimension D . The structure
 257 factor expression used to model this low- Q feature is based on Teixeira's model which was used to
 258 describe aggregation of spherical particles.³⁶ In Teixeira's model, the sphere diameter was applied as a
 259 model parameter giving the length scale for the building blocks of the fractals. Here we have replaced it
 260 by the apparent length L_a of the rods as shown in **Figure 2**. It was not possible to determine the over-all
 261 size of the clusters as it falls outside the length scales probed in the scattering experiment. The model
 262 parameter giving the fractal size, ξ , is therefore fixed at an arbitrary, large value ($\xi = 10000 \text{ \AA}$). The
 263 expression for the fractal structure factor is thus given by:

$$264 \quad S_f(Q) = 1 + \frac{D\Gamma(D-1)\sin[(D-1)\arctan(Q\xi)]}{(QL_a)^D(1+(Q\xi)^2)^{\frac{D-1}{2}}} \quad (5)$$

265 As the peptide concentration increases, a depression or 'correlation hole' in the scattering curve is
 266 observed in the mid- Q regime, corresponding to a depleted concentration of scatterers around a rod
 267 caused by repulsive interrod interactions. We model this feature using a mean-field isotropic structure
 268 factor expression derived by Schneider et al.³⁵ for weakly repulsive, infinitely long thin rods, based on a
 269 modified Yukawa-segment model.

$$270 \quad S_{CH}(Q) = \frac{1}{1 + n \cdot v(Q) P_{thin\ cyl}(Q)} \quad (6)$$

271 Where, the $v(Q)$ is the expression that captures the strength of the interaction, given by:

$$272 \quad v(Q) = \frac{(\mu_L L_a e)^2 / k_B T \epsilon \epsilon_0}{Q^2 + \lambda_D^{-2}} \quad (7)$$

273 The constant parameters k_B , ϵ , ϵ_0 , N_A and e are the Boltzmann constant, permittivity of free space,
274 dielectric constant of water at temperature, T , Avogadro's number, and the charge on an electron,
275 respectively. Here, μ_L is the linear charge density, expressed as the number of charged groups per unit
276 length of rod, and λ_D is the Debye Length in low ionic strength aqueous solution:

$$277 \quad \lambda_D = \left[\frac{8\pi N_A e^2}{k_B T \epsilon \epsilon_0} \right]^{-0.5} = 27.2 \text{ \AA} \quad (8)$$

278 The thin-cylinder form factor used in eq. 6 is given by:⁴¹

$$279 \quad P_{\text{thincyl}}(Q) = \frac{2}{Q L_a} \int_0^{Q L_a} t^{-1} \sin t \, dt - \left[\frac{\sin(Q L_a / 2)}{(Q L_a / 2)} \right]^2 \quad (9)$$

280 The form factor expression, $P_{\text{thincyl}}(Q)$, is introduced in the denominator of the structure factor $S_{CH}(Q)$,
281 which is justified under the assumptions that rods repel only at contact points along their length and in the
282 absence of any end-effects. A similar analytical expression has been employed for describing sterically
283 interacting polymers via the Polymer-Reference Interaction Site Model, i.e., PRISM theory and random-
284 phase approximation (RPA) theory^{41, 43-45}. We chose Schneider et al.'s model over these models due to its
285 simple physical explanation of the system without the introduction of complicated or indirect parameters.
286 Furthermore, using the extracted linear charge density μ_L , the effective diameter (D_{eff}), which is the sum
287 of the cross-section of the rod and the counter-ion double layer around it, can be calculated via Onsager's
288 theory⁴⁶ modified for charged rods⁴⁷:

$$289 \quad D_{\text{eff}} = 2R + \lambda_D \{ \ln(A) + 0.577 + \ln(2) - 0.5 \} \quad (10)$$

290 Where, A is given by Stroobants et al.⁴⁷

$$291 \quad A = 2\pi(\mu_L)^2 \lambda_D \lambda_B \exp(-2R \lambda_D^{-1}) \quad (11)$$

29 And, λ_B is the Bjerrum length in pure water given by:

$$293 \quad \lambda_B = \frac{e^2}{4\pi\epsilon\epsilon_0 k_B T} \sim 7.1 \text{ \AA} \quad (12)$$

294 The total scattering expression is therefore:

$$295 \quad I(Q) = nV_p^2(\rho_p - \rho_s)^2 P_{\text{cyl}}(Q) \hat{S}(Q) \quad (13)$$

296 where, the effective structure factor in **Model 1** i.e. $\hat{S}_1(Q)$ (see eq. 13) is a simple combination of the
297 structure factors for fractal structure and repulsive rods:

$$298 \quad \hat{S}_1(Q) = S_{\text{CH}}(Q) \cdot [1 + \beta(Q)(S_f(Q) - 1)] \quad (14a)$$

299 Here, $\hat{S}_1(Q)$ captures the hierarchy of structures in the semi-dilute and isotropic rod solution, i.e. $S_{\text{CH}}(Q)$
300 is a mean-field isotropic structure factor that describes repulsive inter-rod interactions while $S_f(Q)$ of rods
301 describes a rod cluster that is isotropic by virtue of being corrected by the beta approximation factor $\beta(Q)$
302 . It is important to note that weak anisotropic inter-rod repulsions in semi-dilute solution result in an
303 isotropic distribution of rod particles, which is the simplest model that can be used to satisfactorily
304 describe the SANS data. **Figure 2** illustrates the structure of the rigid rods as described by this combined
305 model. In cases where no correlation hole is observed, such as for neutral rods, low rod concentration, or
306 high salt concentration, we can substitute $S_{\text{CH}}(Q) = 1$ i.e. **Model 2**:

$$307 \quad \hat{S}_2(Q) = 1 + \beta(Q)(S_f(Q) - 1) \quad (14b)$$

308 From **Model 1** (eq. 14 a), we extracted four fit parameters i.e. rod radius R , apparent rod length
309 within cluster L_a , fractal dimension D , and linear charge density μ_L . **Model 2** (eq. 14 b) was used in cases
310 where no correlation hole was apparent, giving three fit parameters, i.e., R , L_a , and D . For both models,
311 the cluster size, ξ , falls outside the length scales that are probed by the experiment and was set to an
312 arbitrary large value of 10000 Å.

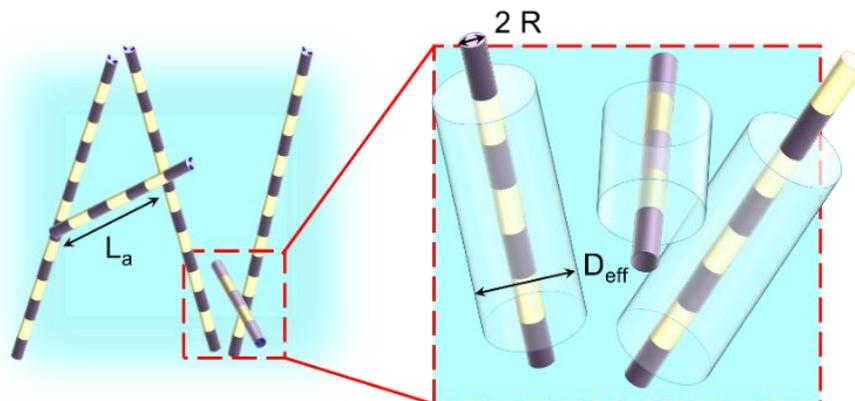


Figure 2: Illustration of the nanostructure of the rigid rod bundlemer chains in low ionic strength solution described by **Model 1**. The rigid rods form fractal networks with an apparent length L_a between contacts. Locally the rods are repulsive with an effective diameter D_{eff} greater than their bare cross section $2R$.

313

314 **Results and discussion**

315 Representative SANS curves for the homotetrameric building block **Peptide 2** versus ultra-long
 316 rigid rods that were constructed using equimolar ratio ($r=1$) of thiol (**Peptide 2**) to maleimide (**Peptide 1**)
 317 groups are shown in **Figure 3 (A)**. TEM analysis of the ultra-long bundlemer chains constructed by the
 318 hybrid physical-chemical assembly pathway resulted in rigid rod-like assemblies that were over one
 319 micron in length (**Figure 3 (B) & 5**), consistent with previous literature.²⁸ Therefore, all SANS data has
 320 been fitted using a cylinder form factor $P_{cyl}(Q)$ (eq. 3). The fits to individual bundlemers (**Peptide 2**)
 321 versus ultra-long rigid rods in **Figure 3 (A)** indicates that both bundlemers and rods have comparable
 322 radii R of $8.0 \text{ \AA} \pm 0.5 \text{ \AA}$ and $9.2 \text{ \AA} \pm 0.3 \text{ \AA}$ respectively, which is also the computationally designed radius
 323 of a single coiled coil bundlemer measured previously.²⁷ Also, the scattering curve for the long rods
 324 follows a power law $I(Q) \sim Q^{-1}$ for $Q < R^{-1}$, indicating a long, cylindrical nanostructure. In contrast, the
 325 scattering curve for the individual bundlemer solution plateaus at low Q due to their short length of $(33 \pm$
 326 $2 \text{ \AA})$. In light of these SANS measurements on the ultra-long rigid rods and since the overall lengths are
 327 beyond the measurable Q -regime using SANS, we focused this solution behavior study on three shorter
 328 length distributions of rigid rods.

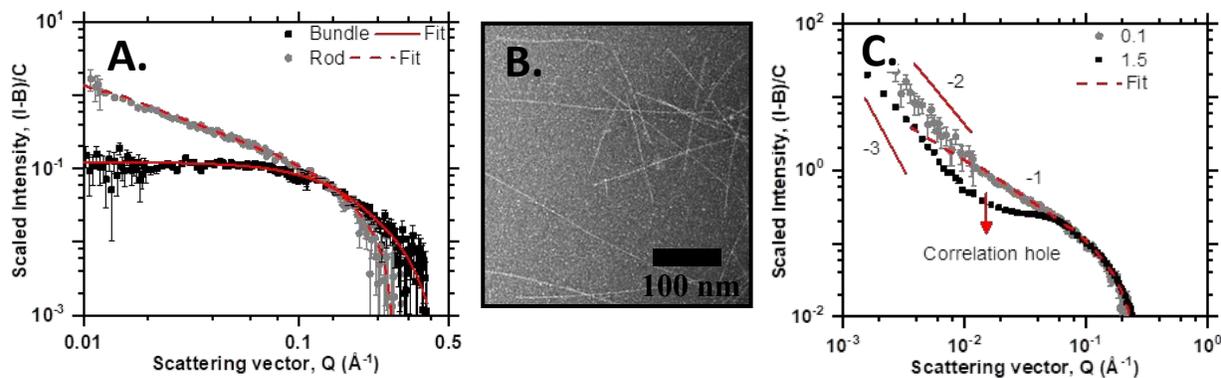


Figure 3: (A) SANS curves for **Peptide 2** versus ultra-long rigid rods of alternating **Peptide 1** and **Peptide 2** bundlemers. The data was modeled using a cylinder form factor plotted as red fit lines. (B) Negatively-stained TEM micrograph of ultra-long rigid rods that confirms their cylindrical shape. (C) Scaled SANS data of rigid rods at low and high concentrations C emphasizing the scattering features, i.e. the mid and high- Q correspond to the shape of the rod, the correlation hole at high concentrations indicate repulsive interactions and the low- Q upturn indicates mass-fractal structures. The legend symbols represents mass by volume fraction (in percent) of peptides in solution. The red solid lines represent slopes on the log-log plot and the red dashed line is a cylinder fit to the 0.1 % data. All scattering intensities I are background subtracted ($-B$) and normalized by sample concentration C .

329

330 As discussed earlier, step-growth polymerization kinetics of the Thiol-Michael *click* reaction
 331 enables the control of the average rod length, L , of bundlemers by varying the stoichiometric ratio, r , of
 332 thiol-to-maleimide functional groups.³⁷ Three rigid rod samples were constructed by increasing r from a
 333 mole ratio of 0.5 to just under 1, resulting in short, medium and long rigid rods. TEM analyses of the
 334 samples confirm that average length of the resulting rod bundlemers increased as r approached unity (see
 335 **Figure 4**) with polydisperse rod samples comprised mainly of short (145 ± 34 Å), medium (528 ± 149 Å)
 336 and long rods (1099 ± 706 Å) (length statistics are summarized in **SI Table S1**). Thus, light scattering
 337 techniques that are sensitive to dispersity in lengths and aggregation were not suitable for extracting
 338 useful information about the sizes and interactions in the rigid rod systems. Also, in order to avoid
 339 artefacts due to beam damage and presence of hydration layer or counterions in solution, we chose
 340 neutrons (SANS) over X-ray (SAXS) as our choice of radiation source to further investigate the solution-
 341 structure and interactions in the rigid rod system.

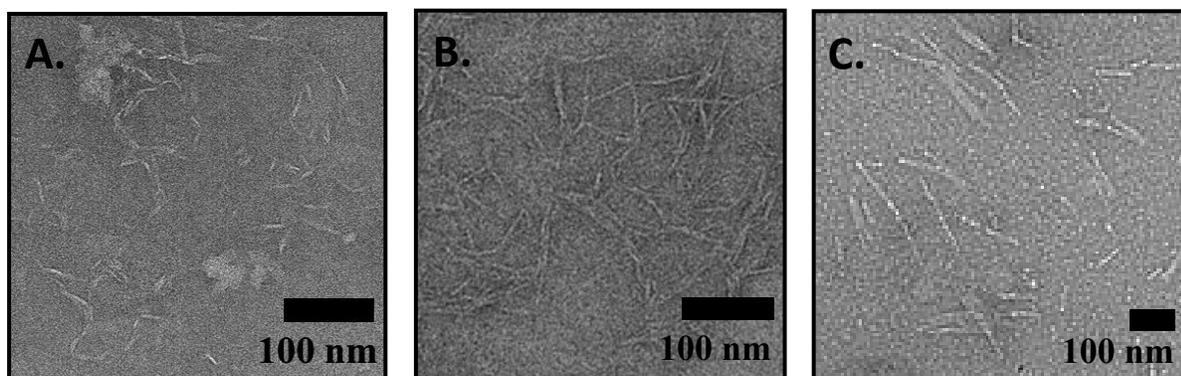


Figure 4: Negatively stained cast film TEM micrographs of 0.1 % (mass by volume) samples of (A) short, (B) medium and (C) long rods constructed by changing the stoichiometric ratio r of thiol containing **Peptide 2** to maleimide decorated **Peptide 1** bundlemers.

342

343 SANS curves and **Model 1** fits for the dilution series of the three samples under salt-free
 344 conditions are presented in **Figure 5**. The SANS curves have four distinct features from which
 345 corresponding model parameters can be independently extracted: the curvature at $Q \approx 0.15 \text{ \AA}^{-1}$
 346 corresponding to the radius of the rigid rods, a Q -regime with an $I(Q)$ slope of -1 indicative of its 1D
 347 shape, an upturn at low- Q having an $I(Q)$ slope between -3 and -2 indicating presence of a mass fractals,
 348 and a depression in scattering signal near the transition between these two slopes that becomes significant
 349 at higher concentrations alluding to presence of a correlation hole. The sample averages for these
 350 parameters are reported in **Table 1** (See **SI Table S3** for all fit results) and each will be discussed in detail
 351 in the rest of the article. It is important to emphasize here that the concentrations probed in this study were
 352 close to or above the critical overlap concentration for dilute to semi-dilute regime for all rigid rod
 353 samples i.e. $C > C^* \approx 1 \text{ rod} / L^3$.³⁵ Scattering data at concentrations in the true dilute regime ($C < C^*$)
 354 were not accessible due to poor counting statistics at lower concentrations. (see **SI Table S1** for C^*
 355 values).

356 In all SANS curves, the scattering feature in the vicinity of $Q \approx 0.15 \text{ \AA}^{-1}$ is related to the rod
 357 cross-section. **Model 1** fit results for the radius R of rigid rods are in good agreement with the coiled coil
 358 radius as discussed before, with a global average value of $11 \text{ \AA} \pm 1 \text{ \AA}$. This corroborates the design
 359 strategy employed for bundlemer assembly, confirming that the rigid rods are indeed formed by an end-

360 to-end assembly of the bundles as depicted in **Figure 1 (C)**. Also, as discussed earlier, for the lowest
 361 concentration samples, the continuation of the scattering feature corresponding to the radius of the rigid
 362 rod into a power law $I \sim Q^{-1}$ ($0.01 \text{ \AA}^{-1} < Q < 0.1 \text{ \AA}^{-1}$) corresponds to the presence of rods in solution.

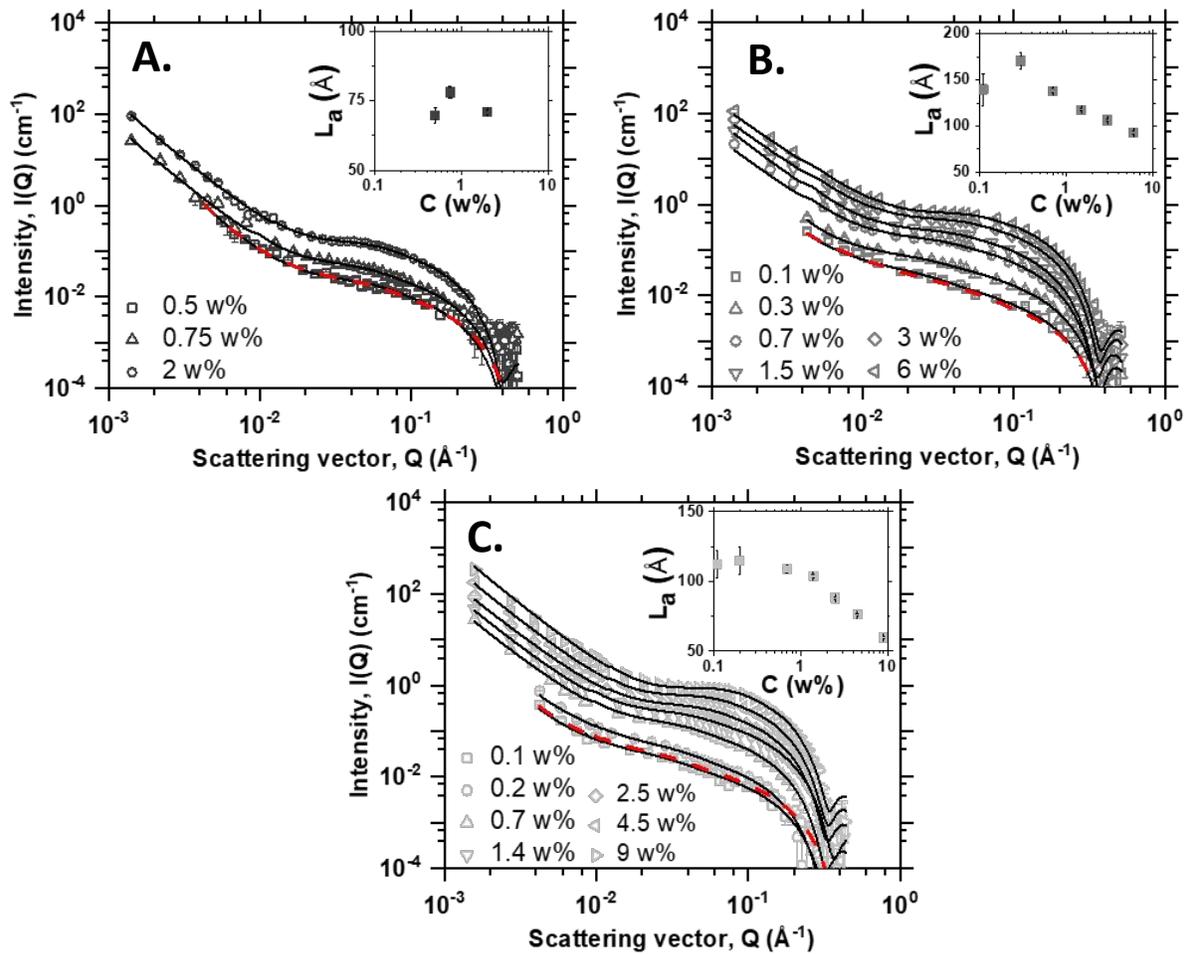


Figure 5: SANS data and corresponding fits for a dilution series of (A) short, (B) medium and (C) long rigid rods in pure D_2O . Here, w% notation in the legends and insets is the mass by volume fraction concentration C of peptides in solution. Black lines are fits using **Model 1**. The lowest concentration fit to **Model 2** is shown as a red dashed line in each case. The inset (top-right) within a graph is a plot of the fitted parameter apparent length L_a versus C that shows the decreasing relation between L_a and C .

363

364 **Table 1: Model 1** fit results for radius, R , charge density, μ_L , and fractal dimension, D , for data shown in
 365 **Figure 5**. Standard error and sample means of R , μ_L , and D are reported here. The number in parentheses
 366 indicates number of individual measurements used for the calculation. The effective diameter, D_{eff} is
 367 calculated via eq. (10) and error propagation is used to calculate uncertainty ($\pm 1\sigma$) in its value (see SI).
 368

Sample	Radius R (\AA)	Charge density μ_L ($e/\text{\AA}$)	Effective diameter D_{eff} (\AA)	Fractal dimension D
Short	9.3 ± 0.4 (3)	0.07 ± 0.01 (2)	71 ± 3	2.6 ± 0.1 (3)

Medium	10.5 ± 0.5 (6)	0.08 ± 0.01 (5)	75 ± 6	2.3 ± 0.1 (5)
Long	11.8 ± 0.5 (7)	0.07 ± 0.01 (5)	71 ± 8	2.5 ± 0.1 (5)

369 We were unable to extract the true length or even the persistence length of the rigid rods via form
 370 factor analysis for all samples since the rods are in the semi-dilute concentration regime ($C > C^*$). This
 371 results in significant overlap of rod excluded volume, giving rise to an upturn in scattering intensity at the
 372 lowest Q values ($Q < 0.01 \text{ \AA}^{-1}$). The upturn follows the power law $I \sim Q^{-D}$ ($D > 1$) which is characteristic
 373 of mass fractal-like cluster aggregates with fractal dimension D , where a larger value of D indicates a
 374 denser fractal structure. Here, we comment on the origin of the low- Q upturn for low-ionic strength rod
 375 solution (see **Figure 5**) measured for samples having a mass by volume fraction concentration C greater
 376 than 0.5 % since for these samples, the fractal dimension was recorded over a decade of Q -values (0.001
 377 $\text{ \AA}^{-1} < Q < 0.01 \text{ \AA}^{-1}$). Phenomenologically, aggregation of rods can take place in semi-dilute concentrations
 378 via the previously reported Diffusion Limited Cluster Aggregation (DLCA) pathway that results in
 379 aggregates having a fractal dimension $D \approx 1.8$ for spheres having aspect ratio of 1 to $D \approx 2.2$ for rods
 380 having an aspect ratio (L/R) of 30.^{29,32} Multiple studies suggest that rod-like colloidal particle suspensions
 381 may also form percolated clusters as a direct result of their anisotropic shape.^{48, 49} However, clustering
 382 due to competing short range attraction versus long range repulsion (SALR)⁵⁰ interactions have been
 383 reported for semi-dilute solutions of polyelectrolytes such as peptide fibrils,⁵¹ DNA⁵² and cellulose
 384 nanocrystals (CNCs)⁵³. Polyelectrolyte clustering is therefore mechanistically different from the
 385 aforementioned entropically-driven DLCA pathways that describe aggregation in colloidal rod-like
 386 particles such as single-walled CNTs.^{34, 54, 55} Since the peptide-based rigid rods investigated here have
 387 multiple solvent-exposed oppositely-charged groups as well as polar and hydrophobic surface patches,
 388 they can interact by a combination of interactions, *i.e.* weak short-range van der Waals or London
 389 dispersive forces and strong long-range electrostatic and entropic (hydrophobic) forces, that can drive
 390 formation of denser cluster aggregates with a larger fractal dimension of $2.3 < D < 2.6$ as observed
 391 here.^{56, 57} Interestingly, while a variation in D with rigid rod sample type (short vs medium vs long) may

392 be attributed to length polydispersity, sample handling, and preparation, the modeling results indicate that
393 D remains unchanged within a dilution series, alluding to a favored packing of sequestered rods within a
394 cluster aggregate.

395 In the mid- to low- Q regime, the cross-over from $I \sim Q^{-1}$, signifying a straight cylinder, to $I \sim Q^{-D}$,
396 signifying a fractal-like object, yields an apparent length, L_a , of rods in the rigid rod sample. The fitted
397 parameter L_a characterizes the average length of rods between contact points within the cluster and is
398 therefore shorter than the actual length L of the rigid rods measured via TEM.^{54, 55, 58} This parameter is
399 also related to the average pore size or mesh size within a network of polyelectrolyte chains.⁵⁹ As the
400 concentration of the rods increases, the fitted L_a decreases (see insets in **Figure 5**). This can be attributed
401 to crowding of rods within a cluster that results in an increase in the number of inter-rod contacts and a
402 decrease in the average distance between these points of contact. For the lowest concentration rigid rod
403 samples, the data can be satisfactorily modeled using **Model 2**, describing a fractal-network structure of
404 rods (see **Figure 2**). However, at higher concentrations, repulsive interactions lead to significant
405 correlation in the spacing of the individual rods. This is manifested as a depression in the scattering
406 intensity in the mid- Q region, i.e. a correlation hole, for $Q \approx 0.01\text{\AA}^{-1}$. The approximate location of the
407 depression increases to larger Q with increasing weight percent of the peptide, paralleling the trend
408 observed for L_a^{-1} and suggesting that this interaction distance corresponds to the mesh size of the
409 networked cluster. While rigid rods constructed using bundlemers may be described as polyelectrolyte-
410 like chains of charged bundlemer units, the correlation hole in the rigid rod systems is distinct from the
411 correlation hole driven ‘polyelectrolyte peak’ that has been widely reported in scattering experiments
412 from low-ionic strength suspensions of polyelectrolytes.³⁴ The correlation peaks at $Q^* \sim C^{1/2}$ in traditional
413 polyelectrolyte systems are indicative of long-range perturbations and preferred distances.^{30, 60} In the
414 present study, no such peak in scattering intensity was recorded even at high peptide concentrations.
415 Sequence-based charge prediction⁶¹ of the peptides in **Peptide 1** and **Peptide 2** bundlemers can give us
416 insight into this behavior of the rigid rod assemblies via charge analysis of the individual bundlemers. At
417 neutral pH, assuming all ionizable groups are fully dissociated, bundlemers of **Peptide 1** and **Peptide 2**

418 can carry a maximum net electronic charge of ca. $-4e$ and $-12e$, respectively. Thus, the rigid rods that
419 comprise of alternating **Peptide 1** and **Peptide 2** bundlemers must have a patchy charge distribution of
420 alternating net charges and can display a maximum net linear charge density of $\approx -0.2 e/\text{\AA}$. However,
421 complete dissociation of oppositely charged groups is unlikely, resulting in weakly repulsive rigid rods
422 with a net charge density smaller than widely studied rod systems such as DNA ($-0.6 e/\text{\AA}$)^{62, 63}, TMV ($-$
423 $1.2 e/\text{\AA}$)⁶⁴ and amyloid fibers ($-0.5 e/\text{\AA}$)⁶⁵. Thus, we infer that the rigid rods are weak polyelectrolytes
424 at neutral pH, i.e., the repulsive interactions between them are not strong enough to cause long-range
425 perturbations typically responsible for the correlation peak observed in other polyelectrolyte-like
426 suspensions.^{30, 45, 52, 60, 66}

427 The scattering profile due to the weak polyelectrolyte character of rigid rods is adequately
428 described by the mean-field structure factor developed by Schneider et al. for weakly-repulsive thin rods,
429 which uses a linear solution to the Poisson-Boltzmann potential, i.e., the Debye-Hückel potential, to model
430 the interrod repulsion.³⁵ This expression gives a physically accurate description of the rigid rod system
431 and has been shown to satisfactorily describe solution structure of weakly-interacting rods observed in
432 both experiments and simulations.^{67, 68} The net linear charge density, μ_L , for the rods is obtained by fitting
433 the data with this structure factor expression and is summarized in **Table 1**. The resulting linear charge
434 density averaged over all rigid rod samples is $-0.08 \pm 0.01 e/\text{\AA}$, which is much smaller than the estimate
435 of $-0.2 e/\text{\AA}$ based on amino acid sequence of the bundlemer. Thus, the model suggests that not all
436 ionizable groups are dissociated under the conditions considered, and the bundlemers carry a weak net
437 charge of approximately $-3e$ per bundlemer along the rod. The magnitude of this effective charge per
438 length may also be attenuated by the presence of counter ions. More information can be extracted via
439 Onsager's theory⁴⁶ modified for charged rods⁴⁷ that gives the effective diameter (D_{eff}) of the rigid rods in
440 solution and is summarized in **Table 1**. The calculated D_{eff} of the rods is $\approx 71 \text{\AA}$, which is larger than their
441 bare rod cross-section $D = 2R$ ($D_{eff} \approx 3.5 D$); this is expected in a low ionic strength solution due to poor
442 screening of net charge on the rigid rod surface, which is effectively the Stern Layer. However, the
443 Oosawa-Manning criterion for counterion condensation ($\lambda_B \mu_L > 1$)⁶⁶, albeit most relevant to solutions

444 with multivalent counterions, is not satisfied for this system ($\lambda_B \mu_L \sim 0.54$). The Stern layer of charge at
445 the rod surface is likely not formed or sparsely populated with counterions. Therefore, we conclude that
446 the electric double layer, i.e., the Gouy-Chapman Double Layer, consists primarily of loosely bound and
447 diffuse counterions under neutral pH conditions.

448 To confirm that the correlation hole results from electrostatic repulsion between like-charged rods
449 at these low ionic strengths, we performed SANS experiments with addition of 200×10^{-3} mol/L NaCl to
450 the long rigid rod sample. The SANS data for these samples are shown in **Figure 6 (B)**. In this case, the
451 correlation hole disappears; the scattering curve shows a direct transition from a $I \sim Q^{-1}$ dependence to a I
452 $\sim Q^{-D}$ dependence without an intermediate depression in scattering. This solution behavior is because the
453 charged side groups are now screened by ions forming a tightly bound electric double layer around the
454 rigid rods. Thus, the effective diameter of the rods decreases ($D_{eff, NaCl} < D_{eff, water}$) and repulsion between
455 rods is screened at contact points within the cluster.^{30, 66} This difference is depicted in the schematic in
456 **Figure 6 (A)**. Consequently, the scattering curve can be modeled more effectively as neutral rods via
457 **Model 2**. The results for fitted parameters obtained from **Model 2** for this case are given in **Table 2**. We
458 also used **Model 1** to fit the data assuming a fixed linear charge density $\mu_L = 0.08 e/\text{\AA}$, corresponding to
459 the value obtained for the samples without added NaCl, and fixed salt concentration of $C_s = 200 \times 10^{-3}$
460 mol/L for comparison (see **Figure 6 (B)**, **SI Table S3** for all fit results). The results for the high ionic
461 strength case are further corroborated by studies at acidic pH, which resulted from the addition of $200 \times$
462 10^{-3} mol/L deuterium chloride (DCl, pH ≈ 1) to the long-rigid rods solution. Although sequence-based
463 charge analysis suggests that the rigid rods should carry a large positive charge, no depression in the
464 curve is evident in this case (see **Figure 6 (C)**). We infer that the presence of ample chloride ions (Cl^-) in
465 this solution effectively screens the positive charge on the rod in a similar manner to the high ionic
466 strength (200×10^{-3} mol/L NaCl) case. We fit this data using **Model 2** and the results are presented in
467 **Table 2**. The similar effect of either salt and pH on the scattering curve confirms the electrostatic origin
468 of the correlation hole, which is sensitive to the presence of electrolytes in solution.

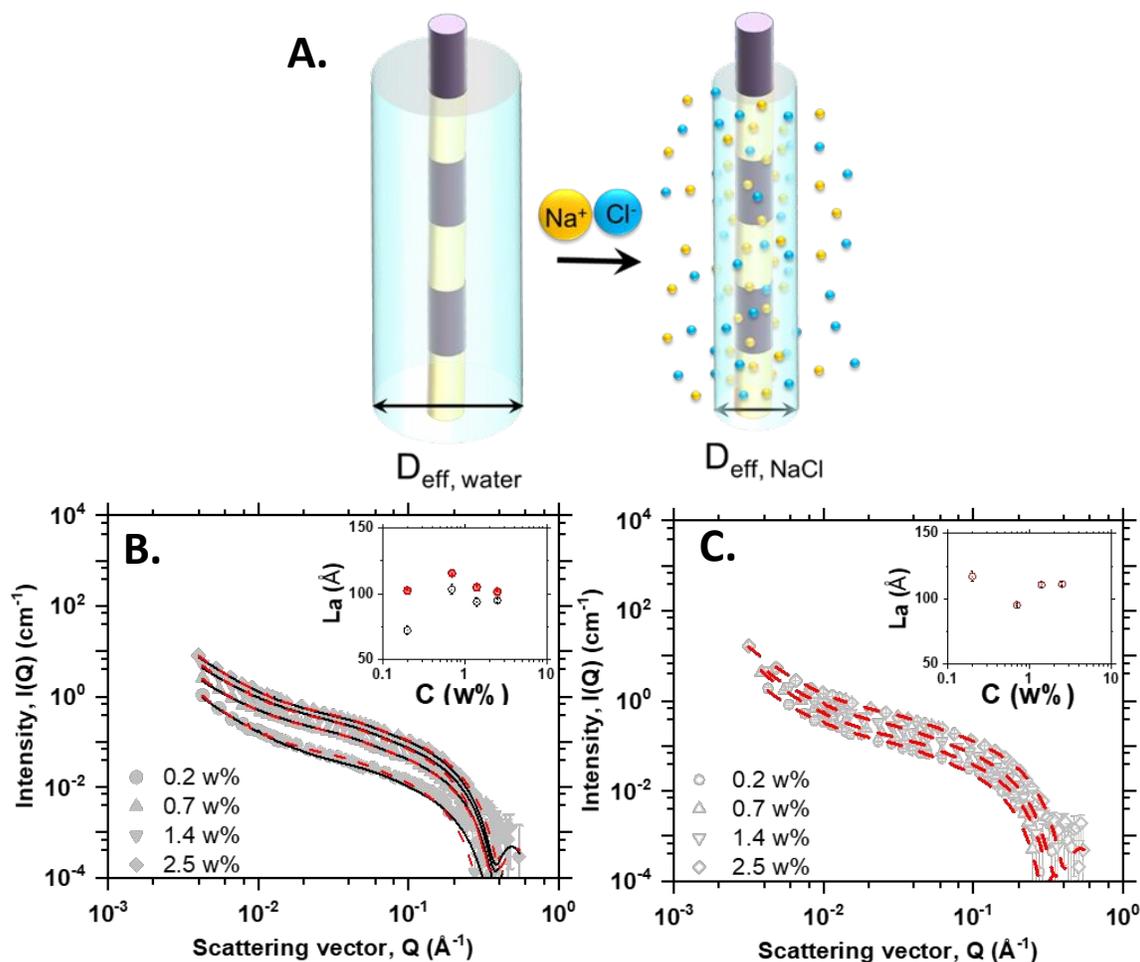


Figure 6: (A) Schematic illustrating the decrease in the effective diameter of the charged rods on addition of monovalent sodium chloride (NaCl) salt. The salt ions screen the net charge on the rods, thus reducing the thickness of the electric double layer. (B) SANS data for long rods in 200 x 10⁻³ mol/L NaCl. The data is fit using Models 1 (black line) and 2 (red dashed line). The apparent length L_a vs rod concentration C is plotted as inset. (C) SANS data for long rods in 200 x 10⁻³ mol/L deuterium chloride (DCl, pH ≈ 1). These data were fit using Model 2 and fitted L_a versus C in this case is shown (inset). The notification of w% in the legend and inset in each plot represents the mass by volume fraction concentration C of the rods.

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Table 2: Results for fits to SANS data shown in **Figure 6** using **Model 2** i.e. long rods with added 200 mM sodium chloride (NaCl salt) and in acidic pH with added 200 mM deuterium chloride (DCl). The sample mean and standard error in radius, R and fractal dimension, D are reported here. The effective diameter, D_{eff} and its uncertainty ($\pm 1\sigma$) are calculated via eq.(10) and error propagation respectively (see SI).

Sample	Radius R (\AA)	Effective diameter $D_{\text{eff,NaCl}}$ (\AA)	Fractal dimension D
Added Salt	11.2 ± 1.0	22 ± 2	2.2 ± 0.1
Acidic pH	11.6 ± 1.4	23 ± 3	2.2 ± 0.1

475

476 We have discussed earlier in this article that **Peptide 1** and **Peptide 2** bundlemers carry different
 477 theoretical net charges at neutral pH, $-4 e$ and $-12 e$ respectively, which alludes to a patchy distribution of
 478 alternating net charge densities along the lengths of the rigid rods. Therefore, the depiction of a
 479 homogenous effective diameter in **Figure 2** and **6 (A)** may be an over-simplification of the true
 480 counterion cloud which may display local counterion distribution and hydration differences. However, the
 481 fit quality of **Model 1** to SANS data in low-ionic strength solution and **Model 2** in high-ionic strength
 482 solution suggests that an average bundlemer charge of $3e$ in neutral pH is sufficient to describe their
 483 solution-state structure and interactions under probed concentration and solution conditions.

484 In our previous publication, we reported that the persistence length l of the rigid rods is
 485 unprecedented, with initial analyses estimating l to be upwards of ten micrometers. Using the linear
 486 charge density μ_L extracted from SANS structure factor fitting, we can estimate the contribution to this
 487 persistence length l by the electrostatic repulsion between bundlemer units, which is expected to result in
 488 stiffening of polyelectrolyte chains in the low salt limit. Using the Odijk-Skonick-Fixman (OSF) model,^{69,}
 489 ⁷⁰ the total persistence length l can be expressed as the sum of the intrinsic persistence length l_o and the
 490 electrostatic persistence length l_e (i.e., $l = l_o + l_e$), where the electrostatic persistence length l_e was
 491 derived using the Debye-Hückel interaction potential between charged monomers along the chain. For the
 492 rigid rods constructed using alternating **Peptide 1** and **Peptide 2** bundlemers, this results in:

$$493 \quad l_e = \frac{\lambda_B \cdot \mu_L^2}{4\lambda_D^{-2}} = 7.6 \text{ \AA}$$

494 Thus, the contribution of the electrostatic persistence length to the total persistence length is small ($l \gg l_e$
 495), implying that the rigid rods are in fact intrinsically stiff polymers of coiled coil bundlemers ($l \sim l_o$).

496 Since the rigid rods are cylinders with a large persistence length, one might expect the formation
 497 of liquid crystalline phases at high rigid rod concentrations. However, we do not observe birefringence in
 498 any rigid rod sample at neutral pH. Onsager's theory for rigid rods that interact via hard-core repulsive
 499 interactions can be used to calculate an approximate onset concentration for liquid crystal formation. For

500 monodisperse stiff rod-like particles, the theoretical cross-over concentration from isotropic to nematic (
501 $I \rightarrow N$) liquid crystal phase is given by the expression:⁷¹

$$502 \quad C_{I \rightarrow N} = 4 \frac{D_{\text{eff}}}{L} \vartheta$$

503 Where, ϑ is the density of the rods ($\vartheta = 1.35 \text{ g/ml}^{72, 73}$). This equation suggests that longer and thinner
504 rods form liquid crystal (LC) phases at lower concentrations. The theoretical $C_{I \rightarrow N}$ for the long rigid rods
505 ($L=1095 \text{ \AA}$, $D_{\text{eff}} = 71 \text{ \AA}$) is calculated to be roughly a mass by volume fraction of 35 %. For the medium
506 and short rigid rod samples, the onset concentration will be even higher. Thus, in all cases, the onset
507 concentrations are much higher than the concentrations that were presently studied. It has also been
508 proposed by Odijk that the $C_{I \rightarrow N}$ in suspensions of charged rods increases due to twisting of charged rods
509 with respect to one another.⁴⁷ These theoretical considerations support the observation that LC phases do
510 not form at neutral pH conditions for the rigid rod samples at probed concentrations. Polydispersity in rod
511 lengths has been reported to result in a broad biphasic concentration regime in which longer rod
512 populations sequentially separate into LC enriched phases at concentrations much lower than the sharp
513 $C_{I \rightarrow N}$ of an equivalent monodisperse rod solution.^{74, 75} The absence of such LC phases in the polydisperse
514 rigid rod solutions studied here may be due to attractive forces that hold together rod clusters and hinder
515 macroscopic rearrangement of the rods into LC phases. This argument is also supported by the constant
516 slope of the upturn in low- Q region of SANS indicating that the internal fractal structure of the cluster
517 aggregate remains unchanged for a given rigid rod sample. Similar arguments can be made for the
518 absence of birefringence in the added salt case wherein the interrod repulsions are further screened by salt
519 ions. Interestingly, lamellar-like liquid crystalline droplets form specifically in acidic pH conditions for
520 the long rigid rod samples, which we have reported elsewhere (data not shown). Overall, a balance of the
521 magnitude of net charge and the balance of opposite charges along the rod length combined with average
522 rod length and distributions inhibits lyotropic liquid crystal formation here.

523

524 **Conclusion:**

525 Computationally designed coiled coils that self-assemble into target nanostructures under mild
526 aqueous conditions provide a toolbox for material construction and discovery. We have utilized a hybrid
527 physical-covalent assembly pathway to construct polymers of bundlemers via thiol-Michael *click*
528 reactions. The resultant bundlemer chains were observable via negatively stained TEM characterization.
529 SANS confirmed that these bundlemer rigid rod chains have the same diameter as the computationally
530 designed coiled coils.

531 The solution structure and interactions in the model rigid rod system were studied in detail via a
532 combination of SANS, TEM and bundlemer-design. The scattering data were modeled using a straight
533 cylinder form factor and a composite structure factor comprised of a fractal scattering function modified
534 for rods coupled with a mean-field structure factor that encodes weak, inter-rod repulsion within the
535 fractal network. The rigid rod solution was heterogeneous wherein the rods formed fractal-like networks
536 or cluster aggregates giving rise to a low- Q upturn in the scattering intensity. The apparent length of the
537 rods within the cluster, related to its mesh size, decreased as the rigid rod concentration increased as more
538 rods were incorporated into the clusters. Furthermore, we showed that the rods are electrostatically
539 repulsive due to the presence of a small net charge as evidenced by the presence of a correlation hole at
540 higher rigid rod concentrations. The presence of a weak net charge is also supported by sequence-based
541 charge analysis of the peptides used to build the rods as well as from fits of an inter-rod structure factor to
542 mid- Q SANS data. Consequently, the addition of monovalent ions to the solution screened the net charge,
543 effectively suppressing the correlation hole. This weak polyelectrolyte character results in formation of
544 fractal clusters of repulsive rods at neutral pH in contrast to other widely reported polyelectrolyte rod
545 materials where a scattering correlation peak, indicating longer-range order, is observed due to larger net
546 charge density.

547 The balance between short-range attraction and long-range repulsion interactions in protein-based
548 biomaterials is a product of their underlying amino-acid sequence that is often over-simplified while
549 describing their structure-property attributes. The model bundlemer rigid rod system studied here is
550 unique in that the rods are intrinsically stiff polymers that are readily viewable and easily modifiable by

551 virtue of their hierarchical assembly and computational design. The patchy local distribution of
552 interaction sites on naturally occurring assemblies can also be readily mimicked on the surface of the
553 bundlemer-based polymers since these are modularly built using orthogonal chemical reactions between
554 designable bundlemer units. The bundlemer-based assemblies can thus provide insights into important
555 sequence-driven structure-property relationships in complex, protein-based materials via future systematic
556 sequence manipulation enabled by their computational design.

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571 **Disclaimer:**

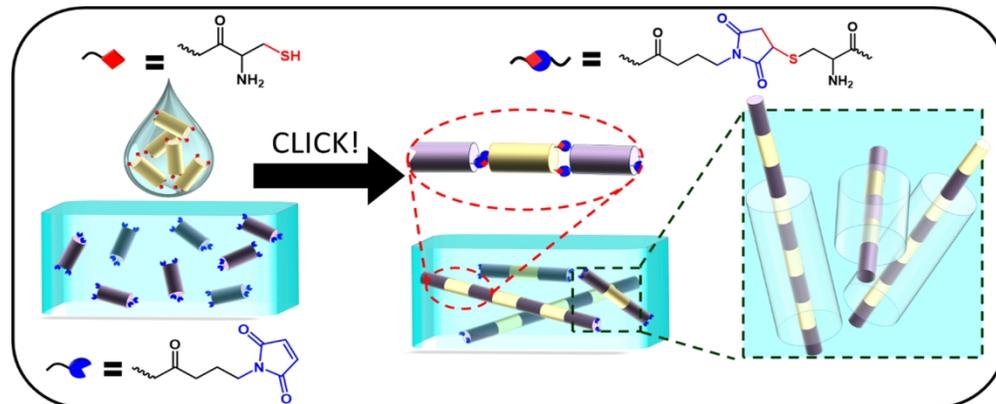
572 The statements, findings, conclusions and recommendations are those of the authors and do not
573 necessarily reflect the view of NIST or the U.S. Department of Commerce. Certain commercial
574 equipment, instruments, materials, suppliers and software are identified in this paper to foster
575 understanding. Such identification does not imply recommendation or endorsement by the NIST, nor does
576 it imply that the materials or equipment identified are necessarily the best available for the purpose.

577 **References:**

- 578 1. D. Philp and J. F. Stoddart, *Angewandte Chemie International Edition in English*, 1996, **35**, 1154-
579 1196.
- 580 2. G. M. Whitesides and M. Boncheva, *Proceedings of the National Academy of Sciences of the*
581 *United States of America*, 2002, **99**, 4769-4774.
- 582 3. S. Zhang, *Nature Biotechnology*, 2003, **21**, 1171.
- 583 4. E. Winfree, F. Liu, L. A. Wenzler and N. C. Seeman, *Nature*, 1998, **394**, 539.
- 584 5. M. Antonietti and S. Förster, *Advanced Materials*, 2003, **15**, 1323-1333.
- 585 6. N. P. King, W. Sheffler, M. R. Sawaya, B. S. Vollmar, J. P. Sumida, I. André, T. Gonen, T. O. Yeates
586 and D. Baker, *Science*, 2012, **336**, 1171.
- 587 7. T. Tørring, N. V. Voigt, J. Nangreave, H. Yan and K. V. Gothelf, *Chemical Society Reviews*, 2011,
588 **40**, 5636-5646.
- 589 8. J. N. Zadeh, C. D. Steenberg, J. S. Bois, B. R. Wolfe, M. B. Pierce, A. R. Khan, R. M. Dirks and N. A.
590 Pierce, *Journal of Computational Chemistry*, 2010, **32**, 170-173.
- 591 9. Y.-T. Lai, N. P. King and T. O. Yeates, *Trends in Cell Biology*, 2012, **22**, 653-661.
- 592 10. T. Kortemme and D. Baker, *Current Opinion in Chemical Biology*, 2004, **8**, 91-97.
- 593 11. B. I. Dahiya and S. L. Mayo, *Science*, 1997, **278**, 82.
- 594 12. G. Colombo, P. Soto and E. Gazit, *Trends in Biotechnology*, 2007, **25**, 211-218.
- 595 13. S. Lyskov and J. J. Gray, *Nucleic Acids Research*, 2008, **36**, W233-W238.
- 596 14. K. A. Dill and J. L. MacCallum, *Science*, 2012, **338**, 1042.
- 597 15. H. Robson Marsden and A. Kros, *Angewandte Chemie International Edition*, 2010, **49**, 2988-
598 3005.
- 599 16. P. Burkhard, J. Stetefeld and S. V. Strelkov, *Trends in Cell Biology*, 2001, **11**, 82-88.
- 600 17. C. Cohen and D. A. D. Parry, *Proteins: Structure, Function, and Bioinformatics*, 1990, **7**, 1-15.
- 601 18. D. N. Woolfson, in *Advances in Protein Chemistry*, Academic Press, 2005, vol. 70, pp. 79-112.
- 602 19. J. W. Bryson, S. F. Betz, H. S. Lu, D. J. Suich, H. X. Zhou, K. T. Neil and W. F. DeGrado, *Science*,
603 1995, **270**, 935.
- 604 20. C. J. Lanci, C. M. MacDermaid, S.-g. Kang, R. Acharya, B. North, X. Yang, X. J. Qiu, W. F. DeGrado
605 and J. G. Saven, *Proceedings of the National Academy of Sciences*, 2012, **109**, 7304.
- 606 21. A. R. Thomson, C. W. Wood, A. J. Burton, G. J. Bartlett, R. B. Sessions, R. L. Brady and D. N.
607 Woolfson, *Science*, 2014, **346**, 485.
- 608 22. C. Xu, R. Liu, A. K. Mehta, R. C. Guerrero-Ferreira, E. R. Wright, S. Dunin-Horkawicz, K. Morris, L.
609 C. Serpell, X. Zuo, J. S. Wall and V. P. Conticello, *Journal of the American Chemical Society*, 2013,
610 **135**, 15565-15578.
- 611 23. E. N. G. Marsh and W. F. DeGrado, *Proceedings of the National Academy of Sciences*, 2002, **99**,
612 5150.
- 613 24. H. V. Zhang, F. Polzer, M. J. Haider, Y. Tian, J. A. Villegas, K. L. Kiick, D. J. Pochan and J. G. Saven,
614 *Science Advances*, 2016, **2**, e1600307.
- 615 25. Y. Tian, H. V. Zhang, K. L. Kiick, J. G. Saven and D. J. Pochan, *Organic & Biomolecular Chemistry*,
616 2017, **15**, 6109-6118.
- 617 26. Y. Tian, F. B. Polzer, H. V. Zhang, K. L. Kiick, J. G. Saven and D. J. Pochan, *Biomacromolecules*,
618 2018, **19**, 4286-4298.
- 619 27. M. J. Haider, H. V. Zhang, N. Sinha, J. A. Fagan, K. L. Kiick, J. G. Saven and D. J. Pochan, *Soft*
620 *Matter*, 2018, **14**, 5488-5496.
- 621 28. D. Wu, N. Sinha, J. Lee, B. P. Sutherland, N. I. Halaszynski, Y. Tian, J. Caplan, H. V. Zhang, J. G.
622 Saven, C. J. Kloxin and D. J. Pochan, *Nature*, 2019, **574**, 658-662.
- 623 29. M. J. Solomon and P. T. Spicer, *Soft Matter*, 2010, **6**, 1391-1400.

- 624 30. M. Muthukumar, *Macromolecules*, 2017, **50**, 9528-9560.
- 625 31. F. Horkay, P. J. Basser, A.-M. Hecht and E. Geissler, *The Journal of Chemical Physics*, 2018, **149**,
626 163312.
- 627 32. A. Mohraz, D. B. Moler, R. M. Ziff and M. J. Solomon, *Physical Review Letters*, 2004, **92**, 155503.
- 628 33. B. A. Schubert, E. W. Kaler and N. J. Wagner, *Langmuir*, 2003, **19**, 4079-4089.
- 629 34. C. E. Williams, M. Nierlich, J. P. Cotton, G. Jannink, F. Boué, M. Daoud, B. Farnoux, C. Picot, P. G.
630 DeGennes, M. Rinaudo, M. Moan and C. Wolff, *Journal of Polymer Science: Polymer Letters*
631 *Edition*, 1979, **17**, 379-384.
- 632 35. J. Schneider, W. Hess and R. Klein, *Journal of Physics A: Mathematical and General*, 1985, **18**,
633 1221.
- 634 36. J. Teixeira, *Journal of Applied Crystallography*, 1988, **21**, 781-785.
- 635 37. P. J. Flory, *Journal of the American Chemical Society*, 1936, **58**, 1877-1885.
- 636 38. C. A. Schneider, W. S. Rasband and K. W. Eliceiri, *Nature Methods*, 2012, **9**, 671.
- 637 39. S. Kline, *Journal of Applied Crystallography*, 2006, **39**, 895-900.
- 638 40. <http://www.sasview.org/>.
- 639 41. T. Zemb and P. Lindner, *Neutrons, X-rays and Light: Scattering Methods Applied to Soft*
640 *Condensed Matter*, Elsevier, 2002.
- 641 42. M. Kotlarchyk and S. H. Chen, *The Journal of Chemical Physics*, 1983, **79**, 2461-2469.
- 642 43. K. S. Schweizer and J. G. Curro, *Physical Review Letters*, 1987, **58**, 246-249.
- 643 44. L. Arleth, M. Bergström and J. S. Pedersen, *Langmuir*, 2002, **18**, 5343-5353.
- 644 45. E. Josef and H. Bianco-Peled, *Soft Matter*, 2012, **8**, 9156-9165.
- 645 46. L. Onsager, *Annals of the New York Academy of Sciences*, 1949, **51**, 627-659.
- 646 47. A. Stroobants, H. N. W. Lekkerkerker and T. Odijk, *Macromolecules*, 1986, **19**, 2232-2238.
- 647 48. L. Wu, C. P. Ortiz and D. J. Jerolmack, *Langmuir : the ACS journal of surfaces and colloids*, 2017,
648 **33**, 622-629.
- 649 49. P. van der Schoot and T. Odijk, *The Journal of Chemical Physics*, 1992, **97**, 515-524.
- 650 50. Y. Liu and Y. Xi, *Current Opinion in Colloid & Interface Science*, 2019, **39**, 123-136.
- 651 51. M. Owczarz, S. Bolisetty, R. Mezzenga and P. Arosio, *Journal of Colloid and Interface Science*,
652 2015, **437**, 244-251.
- 653 52. R. Borsali, H. Nguyen and R. Pecora, *Macromolecules*, 1998, **31**, 1548-1555.
- 654 53. H. Oguzlu, C. Danumah and Y. Boluk, *Current Opinion in Colloid & Interface Science*, 2017, **29**, 46-
655 56.
- 656 54. R. S. Justice, D. H. Wang, L.-S. Tan and D. W. Schaefer, *Journal of Applied Crystallography*, 2007,
657 **40**, s88-s92.
- 658 55. W. Zhou, M. F. Islam, H. Wang, D. L. Ho, A. G. Yodh, K. I. Winey and J. E. Fischer, *Chemical*
659 *Physics Letters*, 2004, **384**, 185-189.
- 660 56. J. Ray and G. S. Manning, *Macromolecules*, 2000, **33**, 2901-2908.
- 661 57. A. Chremos and J. F. Douglas, *The Journal of Chemical Physics*, 2017, **147**, 241103.
- 662 58. D. W. Schaefer, J. Zhao, J. M. Brown, D. P. Anderson and D. W. Tomlin, *Chemical Physics Letters*,
663 2003, **375**, 369-375.
- 664 59. Y. Zhang, J. F. Douglas, B. D. Ermi and E. J. Amis, *The Journal of Chemical Physics*, 2001, **114**,
665 3299-3313.
- 666 60. S. H. Chen, E. Y. Sheu, J. Kalus and H. Hoffman, *Journal of Applied Crystallography*, 1988, **21**, 751-
667 769.
- 668 61. D. S. Moore, *Biochemical Education*, 1985, **13**, 10-11.
- 669 62. M. T. Record, S. J. Mazur, P. Melancon, J. H. Roe, S. L. Shaner and L. Unger, *Annual Review of*
670 *Biochemistry*, 1981, **50**, 997-1024.
- 671 63. R. Gilbert, *Crystallography Reviews*, 2009, **15**, 285-288.

- 672 64. S. Fraden, in *Observation, Prediction and Simulation of Phase Transitions in Complex Fluids*, eds.
673 M. Baus, L. F. Rull and J.-P. Ryckaert, Springer Netherlands, Dordrecht, 1995, DOI: 10.1007/978-
674 94-011-0065-6_3, pp. 113-164.
- 675 65. R. Mezzenga, J.-M. Jung and J. Adamcik, *Langmuir*, 2010, **26**, 10401-10405.
- 676 66. A. V. Dobrynin and M. Rubinstein, *Progress in Polymer Science*, 2005, **30**, 1049-1118.
- 677 67. H. Löwen, *The Journal of Chemical Physics*, 1994, **100**, 6738-6749.
- 678 68. J.-P. Hansen and H. Löwen, *Annual Review of Physical Chemistry*, 2000, **51**, 209-242.
- 679 69. J. Skolnick and M. Fixman, *Macromolecules*, 1977, **10**, 944-948.
- 680 70. T. Odijk, *Journal of Polymer Science: Polymer Physics Edition*, 1977, **15**, 477-483.
- 681 71. G. J. Vroege and H. N. W. Lekkerkerker, *Reports on Progress in Physics*, 1992, **55**, 1241.
- 682 72. I. D. Kuntz and W. Kauzmann, in *Advances in Protein Chemistry*, eds. C. B. Anfinsen, J. T. Edsall
683 and F. M. Richards, Academic Press, 1974, vol. 28, pp. 239-345.
- 684 73. K. Gekko and H. Noguchi, *The Journal of Physical Chemistry*, 1979, **83**, 2706-2714.
- 685 74. T. Odijk and H. N. W. Lekkerkerker, *The Journal of Physical Chemistry*, 1985, **89**, 2090-2096.
- 686 75. P. A. Buining and H. N. W. Lekkerkerker, *The Journal of Physical Chemistry*, 1993, **97**, 11510-
687 11516.
- 688



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