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## Concave Bi<sub>2</sub>WO<sub>6</sub> nanoplates with oxygen vacancies achieving enhanced electrocatalytic oxygen evolution in near-neutral water

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### Abstract:

Developing highly efficient and inexpensive oxygen evolution reaction (OER) electrocatalyst in near-neutral water is of paramount importance for many practical applications. Herein, we report that Bi<sub>2</sub>WO<sub>6</sub> concave nanoplates (CNPs) with oxygen vacancies enable electrocatalytic OER under neutral conditions with high activity and good durability. To the best of our knowledge, this is the first example on W-based electrocatalyst containing no first-row transition metal or precious metal elements for OER in near-neutral water. Experimental results and first-principles calculations revealed that the presence of oxygen vacancies in Bi<sub>2</sub>WO<sub>6</sub> CNPs could significantly decrease charge-transfer resistance and adsorption barrier of H<sub>2</sub>O molecules in the process of electrochemistry, and thus benefiting the improvement of OER activity. The fabrication of concave surfaces with high energy facets could further enhance OER activity of the Bi<sub>2</sub>WO<sub>6</sub> NPs electrocatalysts. As a result, the synergistic effect of oxygen vacancies and concave surfaces endowed impressive performance of Bi<sub>2</sub>WO<sub>6</sub> CNPs for OER, which is comparable to the best electrocatalysts among known inorganic non-precious metal compounds. The present strategy on the combination of defect and crystal facet engineering could open a new avenue to design new and highly efficient OER electrocatalysts.

### 1. Introduction

The depletion of fossil fuels and the associated adverse impact on environment have triggered

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much attention for sustainable energy conversion and storage systems.<sup>1-3</sup> Recently, the oxygen evolution reaction (OER) has attracted great research interest because of its critical role in variously promising energy conversion and storage technologies, such as photocatalytic or electrocatalytic water splitting for hydrogen production,<sup>4</sup> regenerative fuel cells,<sup>5</sup> rechargeable metal-air batteries,<sup>6</sup> and many more.<sup>7</sup> As far as electrocatalytic water splitting is concerned, its reaction rate is mainly determined by complex and kinetically sluggish of OER.<sup>8</sup> Therefore a highly efficient OER electrocatalyst is required to address the kinetically slow process. Although  $\text{IrO}_2$  and  $\text{RuO}_2$  display excellent electrocatalytic performance for OER, severe scarcity and high cost hampered their extensive applications. Accordingly, various non-precious-metal catalysts have been developed as competitive alternatives, typically including first-row spinel and perovskite metal oxides and their derivatives,<sup>9-12</sup> layered double hydroxides,<sup>13</sup> carbon-based nonmetal catalysts,<sup>14</sup> as well as some transition metal complexes.<sup>15</sup> However, usually, the above catalysts need to be performed in concentrated basic ( $\text{pH} \geq 13$ ) or acidic ( $\text{pH} \leq 1$ ) solutions.<sup>16</sup> Up to now, only a few catalysts can efficiently operate OER in near-neutral water under ambient conditions.<sup>17-22</sup> The development of earth-abundant catalysts that run OER under benign conditions remains a great challenge.

As one of the simplest members of the Aurivillius family,  $\text{Bi}_2\text{WO}_6$  with a layered structure is composed of perovskite-like slab of  $[\text{WO}_4]^{2-}$  sandwiched between bismuth oxide  $[\text{Bi}_2\text{O}_2]^{2+}$  units. Relatively narrowed band gap (2.8 eV), appropriate energy band positions, and favorable separation of photogenerated electron-hole pairs caused by internal electric fields between the slabs make  $\text{Bi}_2\text{WO}_6$  particularly suitable for the photocatalytic or photoelectrocatalytic degradation of organics,<sup>23,24</sup> selective organic synthesis and production of fuels.<sup>25</sup> As a result, many efforts have been made to enhance photocatalytic activity of  $\text{Bi}_2\text{WO}_6$  through the fabrication of special nanostructures, doping, substitution, and surface modification.<sup>26</sup> However, until now, concave  $\text{Bi}_2\text{WO}_6$  nanoplates and  $\text{Bi}_2\text{WO}_6$  CNPs as an efficient electrocatalyst for OER in near-neutral water have never been reported.

In some cases, oxygen vacancies can improve the conductivity of the materials and decrease the adsorption barrier of  $\text{H}_2\text{O}$  molecules onto its surfaces and thus will be given enhanced electrocatalytic activity.<sup>27-29</sup> On the other hand, crystals with concave surfaces possess open negative curvatures and high-index facets, which are expected to show significantly enhanced

catalytic activity relative to their flat and convex counterparts.<sup>30-34</sup> Herein, inspired by the above considerations, Bi<sub>2</sub>WO<sub>6</sub> CNPs with oxygen vacancies have been synthesized for the first time. Importantly, the Bi<sub>2</sub>WO<sub>6</sub> CNPs with oxygen vacancies enable electrocatalytic water oxidation under near-neutral conditions with high activity and good stability. Our work opens a new avenue into the design of highly active and stable OER electrocatalysts.

## 2. Experimental Section

**Synthesis of Bi<sub>2</sub>WO<sub>6</sub> CNPs.** In a typical procedure, 0.5 mmol of Bi(NO<sub>3</sub>)<sub>3</sub>·5H<sub>2</sub>O was added to 15 mL of absolute alcohol under stirring for 15 min at room temperature. Afterward, 20 mL of aqueous solution containing 1 mmol of Na<sub>2</sub>WO<sub>4</sub>·2H<sub>2</sub>O and 5 mL of oleylamine were added into the above solution. The mixture was finally transferred into a 50 mL of Teflon-lined autoclave and maintained at 220 °C for 6 h. The reactor was cooled to room temperature naturally. The resulting sample was collected and washed for several times with alcohol and deionized water and dried at 60 °C under vacuum.

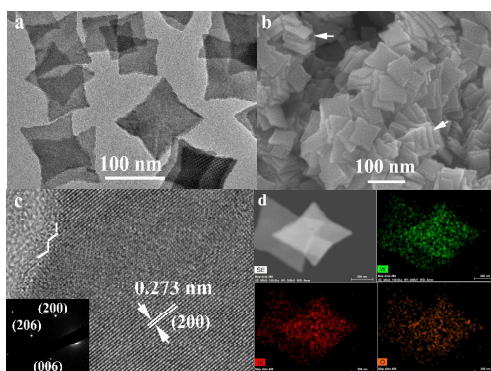
**Materials Characterization.** Powder X-ray diffraction (XRD) was performed on a Bruker D8 Advance X-ray diffractometer using Cu K $\alpha$  radiation ( $\lambda$  = 0.15418 nm) at a scanning rate of 8°/min in the 2 $\theta$  range of 10-70°. Field-emission scanning electron microscopy (FE-SEM) images were taken on a Nova NanoSEM200 scanning electron microscope. Transmission electron microscopy (TEM) image, high-resolution transmission electron microscopy (HRTEM) image, scanning transmission electron microscopy (STEM) image, energy dispersive X-ray spectroscopy (EDS), and elemental mapping were carried out on a JEM-2100F HRTEM, using an accelerating voltage of 200 kV. The UV-vis diffuse-reflectance spectra (UV-DRS) were recorded on a UV2501PC (Shimadzu) using BaSO<sub>4</sub> as a reference. X-ray photoelectron spectroscopy (XPS) measurement was carried out with a Thermo ESCALAB 250 X-ray photoelectron spectrometer with an excitation source of Al K $\alpha$  radiation ( $\lambda$  = 1253.6 eV). The Brunauer-Emmett-Teller (BET) surface area was measured with an ASAP2020 specific surface area analyzer.

**Preparation of Working Electrode.** 5 mg of the as-synthesized Bi<sub>2</sub>WO<sub>6</sub> CNPs was dispersed in 1 mL of isopropyl alcohol and ultrapure water mixture ( $V_{\text{isopropyl alcohol}} : V_{\text{water}} = 2 : 1$ ) by at least 30 min sonication. Then 40  $\mu$ L of naphthol was added in the solution. After another 30 min sonication, a homogenous ink was formed. After that, 5  $\mu$ L of the catalyst ink was loaded onto a

glassy carbon electrode of 3 mm in diameter (loading  $0.34 \text{ mg cm}^{-2}$ ).

**Electrochemical Properties Measurements.** All electrochemical experiments were carried out on an electrochemical system (CHI660D) in a three-electrode cell using  $\text{Bi}_2\text{WO}_6$  CNPs/GC as the working electrode, a Pt wire as the counter electrode and saturated calomel electrode (SCE) as the reference electrode, respectively. Electrolyte was 0.5 M  $\text{Na}_2\text{SO}_4$  whose pH is 6.6. Electrode potential was converted to the RHE scale, using the following equation:  $E(\text{RHE}) = E(\text{SCE}) + 0.242 + 0.059 \times \text{pH}$ . Electrochemical impedance spectra were carried out using above three electrode systems. The frequency range was 100 K Hz to 0.1 Hz, and the amplitude of the applied voltage was 5 mV.

### 3. Results and Discussion

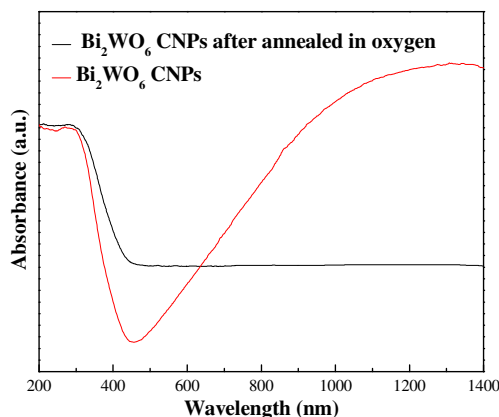


**Fig. 1** TEM image (a) and FE-SEM image (b) of the synthesized  $\text{Bi}_2\text{WO}_6$  CNPs. HRTEM image (c) recorded on the rim of a  $\text{Bi}_2\text{WO}_6$  CNP and its corresponding SAED pattern as inserted. STEM image and elemental mapping of  $\text{Bi}_2\text{WO}_6$  CNPs (d). Green, red, and orange represent W, Bi, and O elements, respectively.

$\text{Bi}_2\text{WO}_6$  CNPs were synthesized by a facile one-pot solvothermal route. As can be seen from TEM image in Fig. 1a, the as-obtained products are CNPs with an average side length of about 80 nm. The X-ray diffraction pattern of the products could be indexed to the orthorhombic phase of  $\text{Bi}_2\text{WO}_6$  (JCPDS No. 73-1126, see Fig. S2). No diffraction peaks from  $\text{Bi}_2\text{O}_3$ ,  $\text{WO}_3$  and other substances were observed, suggesting that the products have high purity. The thickness of  $\text{Bi}_2\text{WO}_6$  nanoplate is *ca.* 20 nm, measured from the nanoplates in FE-SEM image (marked with white arrowheads, Fig. 1b). Both HRTEM image and selected-area electron diffraction pattern recorded

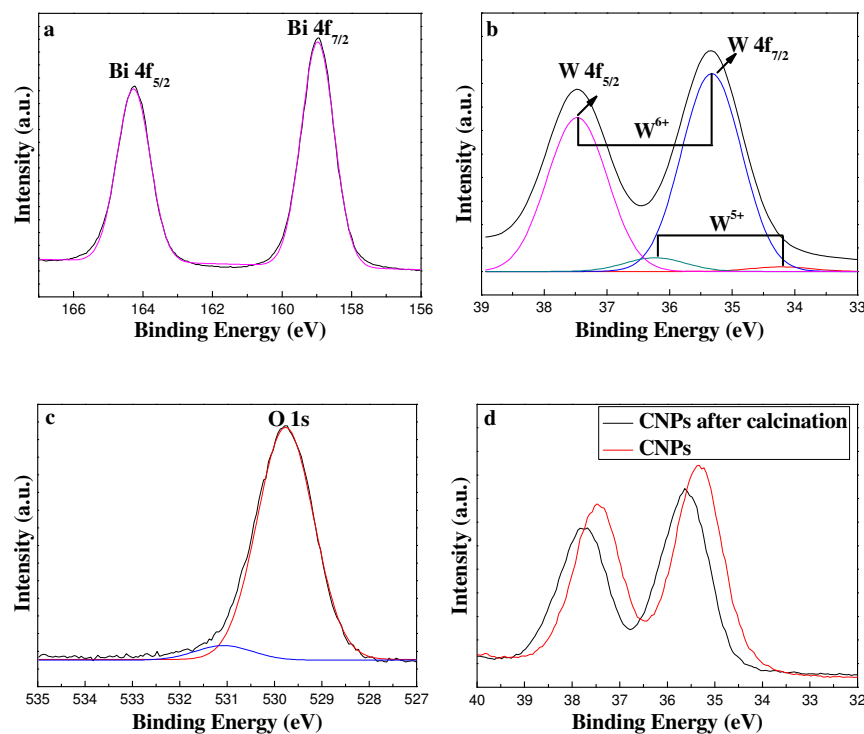
on the individual nanoplate show its single-crystalline structure (Fig. 1c). HRTEM observations also indicate that the nanoplate has good crystallinity. The distinct lattice fringe spacing of 0.273 nm matches well the interplanar separation of (200) planes of orthorhombic  $\text{Bi}_2\text{WO}_6$ . Especially, as seen from HRTEM image of the  $\text{Bi}_2\text{WO}_6$  CNP, there are a number of atomic terraces and steps on its brims (marked with white lines in Fig. 1c), indicating that the  $\text{Bi}_2\text{WO}_6$  nanoplate are enclosed by high-index facets (for detailed proof see Supporting Information, Fig. S3).<sup>31-33</sup> STEM-EDS elemental maps of the nanoplates were further used to identify the element distribution. As shown in Fig. 1d, three elements of W, Bi, and O are homogeneously distributed throughout the  $\text{Bi}_2\text{WO}_6$  CNPs. The quantitative calculation from EDX spectrum of the  $\text{Bi}_2\text{WO}_6$  CNPs shows that the ratio of Bi to W is close to stoichiometric 2 : 1 (Fig. S4).

It has been acknowledged that the formation of crystals with concave surfaces is not favored by thermodynamics because of the higher surface energy and inner strain.<sup>32</sup> In the present reaction system, oleylamine (OLA) plays a key role to control the shape and exposed facets. In the absence of OLA, keeping other synthetic conditions unchanged, only three-dimensional hierarchical microspheres were obtained (Fig. S5). According to time-dependent XRD analyses and SEM observations on the intermediates (Fig. S6 and Fig. S7), the formation of  $\text{Bi}_2\text{WO}_6$  CNPs went through anisotropic growth of nanoparticles and Ostwald ripening process of concave nanoplates. As previous reported, here OLA molecules also played a role of capping reagents.<sup>33</sup> They could selectively adsorb on the surfaces of newborn  $\text{Bi}_2\text{WO}_6$  nanoparticles and induce the preferential overgrowth, thus favoring the formation of CNPs. Interestingly, the morphology of CNPs still kept unchanged after the amounts of reactants were increased by three times and gram-scale products could be easily prepared through one-pot reaction (Fig. S8).



**Fig. 2** UV-visible diffuse reflectance spectra of  $\text{Bi}_2\text{WO}_6$  CNPs before and after annealing in oxygen.

The as-obtained  $\text{Bi}_2\text{WO}_6$  CNPs powder exhibits a unique bluish-gray color, which is quite different from previous report (yellow appearance).<sup>35</sup> UV-visible diffuse reflectance spectroscopy of the sample shows strong absorption in the regions of 200-400 and 550-1400 nm (Fig. 2), which is consistent with its bluish-gray appearance (Fig. S8). Usually, oxygen vacancies can cause some metal oxides strong near-infrared light absorption.<sup>36</sup> Most recently, the theoretical calculation has shown that the 72-atom super cell of  $\text{Bi}_2\text{WO}_6$  possesses three oxygen vacancies.<sup>37</sup> Therefore it suggests that the as-prepared  $\text{Bi}_2\text{WO}_6$  CNPs with strong near-infrared light absorption contain oxygen vacancies. In order to confirm this point, XPS studies on the  $\text{Bi}_2\text{WO}_6$  CNPs were carried out. Fig. 3a, 3b and 3c represent high resolution spectra of Bi 4f, W 4f and O 1s, respectively. As shown in Fig. 3a, the spin-orbit components of Bi 4f<sub>5/2</sub> and Bi 4f<sub>7/2</sub> consist of peaks with binding energy around 164.2 and 159.0 eV, respectively, which is in agreement with  $\text{Bi}^{3+}$  ions in  $\text{Bi}_2\text{WO}_6$ .<sup>38</sup> The W 4f core-level spectrum could be fitted into spin-orbit doublets (Fig. 3b), corresponding to two different oxidation state of W atoms. The dominant peaks at 37.5 and 35.3 eV can be assigned to the emission of W 4f<sub>5/2</sub> and W 4f<sub>7/2</sub> core level from the atoms with  $\text{W}^{6+}$  oxidation state, while the other two weaker peaks located at 36.2 and 34.2 eV correspond to lower +5 valence of W.<sup>37</sup> The binding energy of W 4f for the sample after calcinated in oxygen obviously increases, which also supports the existence of  $\text{W}^{5+}$  in  $\text{Bi}_2\text{WO}_6$  CNPs (Fig. 3d). The presence of low valence W in tungsten oxides is often associated with oxygen vacancies.<sup>36</sup> The further analysis on O 1s core level spectrum could also confirm this point. Specifically, the peak at 529.8 eV is due to oxygen atoms bound to metals, while the peak at 531.1 eV is attributed to a high number of defect sites with a low oxygen coordination, namely, oxygen vacancy.<sup>28</sup> In addition, near-infrared light absorption of  $\text{Bi}_2\text{WO}_6$  CNPs disappeared after they were thermally treated in oxygen atmosphere (Fig. 2), which also supported the presence of oxygen vacancies.

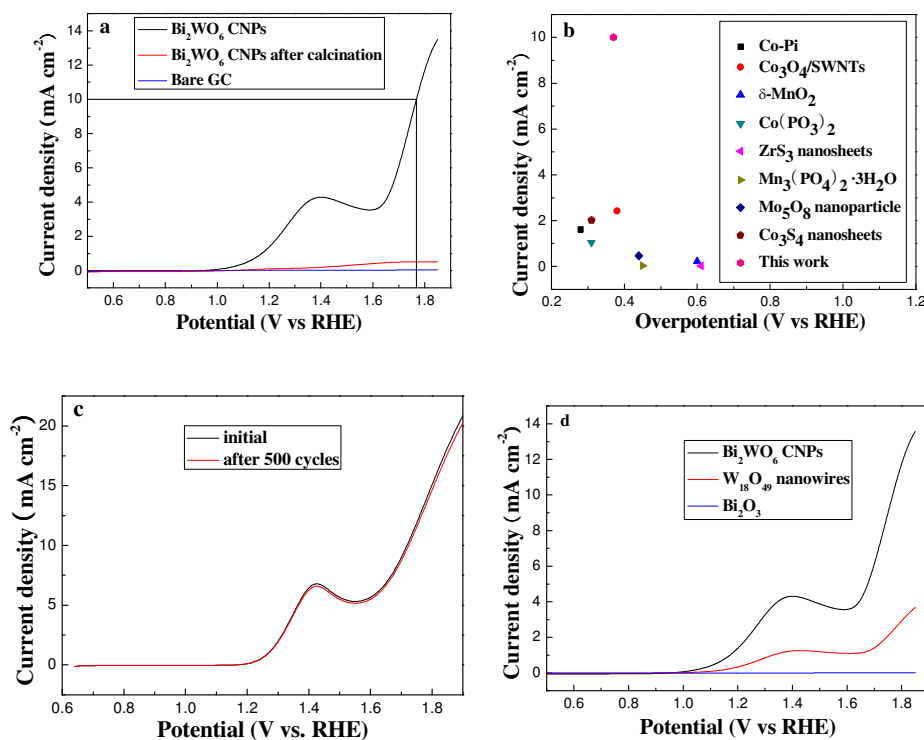


**Fig. 3** High resolution XPS spectra of Bi 4f (a), W 4f (b), and O 1s (c). The comparison of W 4f spectra of Bi<sub>2</sub>WO<sub>6</sub> CNPs and Bi<sub>2</sub>WO<sub>6</sub> CNPs after calcinated in oxygen (d).

The OER activities of the as-prepared Bi<sub>2</sub>WO<sub>6</sub> CNPs were then systematically investigated. The sample was loaded onto glass carbon (GC) electrode (the loading amount was *ca.* 0.34 mg cm<sup>-2</sup>) in 0.5 M Na<sub>2</sub>SO<sub>4</sub> aqueous solution (pH = 6.6) with a scan rate of 10 mV s<sup>-1</sup> using a three-electrode system. For comparison, similar measurements for bare GC and Bi<sub>2</sub>WO<sub>6</sub> CNPs after calcination in oxygen with same loading amounts were also performed. Fig. 4a shows their linear sweep voltammetry (LSV) curves without ohmic potential drop (*i*R) losses correction on the reversible hydrogen electrode (RHE) scale. It can be seen that the Bi<sub>2</sub>WO<sub>6</sub> CNPs/GC electrode exhibits much greater current density compared with Bi<sub>2</sub>WO<sub>6</sub> CNPs after calcination/GC and bare GC ones under a certain applied voltage, directly confirming that Bi<sub>2</sub>WO<sub>6</sub> CNPs possess the highest electrocatalytic activity for OER among them. Specifically, the current density of Bi<sub>2</sub>WO<sub>6</sub> CNPs can reach 10 mA cm<sup>-2</sup> (a metric relevant to solar fuel production) at an applied voltage of 1.77 V vs RHE, which is approximately 20 times larger than that of Bi<sub>2</sub>WO<sub>6</sub> CNPs after calcination counterpart (*ca.* 0.5 mA cm<sup>-2</sup>). Moreover, Bi<sub>2</sub>WO<sub>6</sub> CNPs exhibit a relatively small overpotential of 0.37 V for electrocatalytic O<sub>2</sub> evolution. Considering the onset overpotential and



catalytic current densities together,  $\text{Bi}_2\text{WO}_6$  CNPs represent one of the best electrocatalysts for OER in near-neutral water without involving any noble metal or first-row transition metal elements (Fig. 4b, Table S1). Furthermore, we assessed the long-term electrochemical stability of this electrode for OER under near-neutral condition through cyclic voltammogram tests (Fig. 4c). The polarization curves recorded before and after 500 cycles were compared. As observed, the OER activity of  $\text{Bi}_2\text{WO}_6$  CNPs showed negligible change, indicating good durability of the  $\text{Bi}_2\text{WO}_6$  CNPs catalyst.



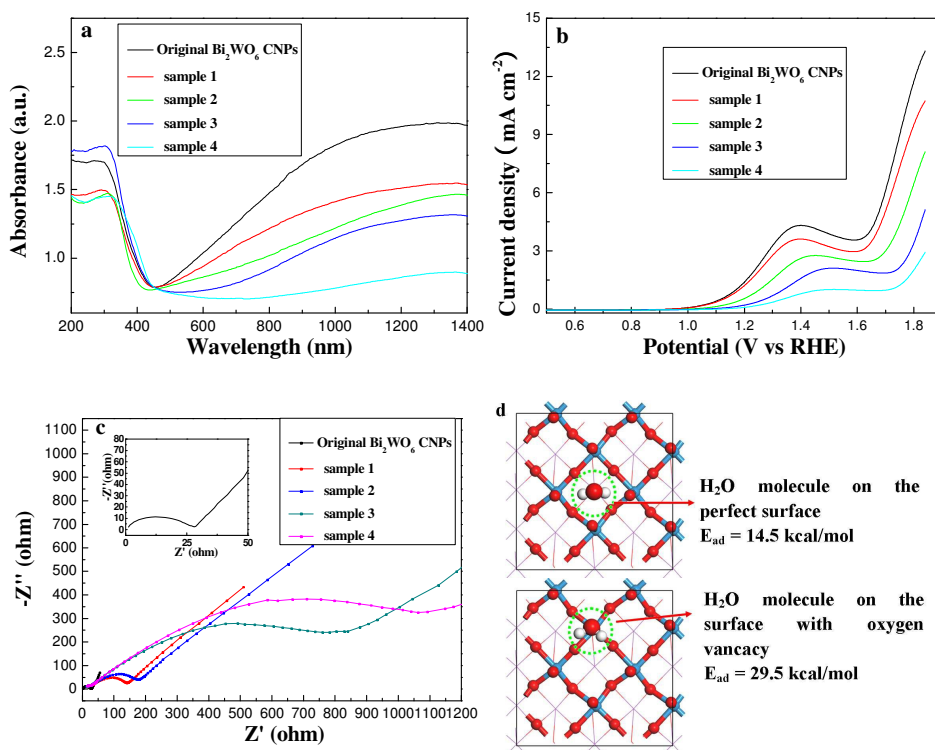
**Fig. 4** LSV curves for  $\text{Bi}_2\text{WO}_6$  CNPs,  $\text{Bi}_2\text{WO}_6$  CNPs after annealing in  $\text{O}_2$ , and bare GC in neutral  $\text{Na}_2\text{SO}_4$  solution with a scan rate of  $10 \text{ mV s}^{-1}$  (a). The current density (at an overpotential of 0.54 V) and onset overpotential of some typical inorganic OER electrocatalysts (b). LSV curves for  $\text{Bi}_2\text{WO}_6$  CNPs/GC electrode before and after 500 cycles (c). LSV curves for  $\text{Bi}_2\text{WO}_6$  CNPs,  $\text{W}_{18}\text{O}_{49}$  nanowires, and rod-like  $\text{Bi}_2\text{O}_3$  (d).

$\text{Bi}_2\text{WO}_6$  is constructed of perovskite-like  $(\text{WO}_4)^{2-}$  and alternating  $(\text{Bi}_2\text{O}_2)^{2+}$  units. In order to understand the active site of  $\text{Bi}_2\text{WO}_6$  CNPs, both rod-like  $\text{Bi}_2\text{O}_3$  and  $\text{W}_{18}\text{O}_{49}$  nanowires containing oxygen vacancies were prepared according to previous reports (Supporting Information, Fig. S9-12).<sup>39, 36</sup> Following the same preparation conditions as that of  $\text{Bi}_2\text{WO}_6$  CNPs/GC electrode, the

obtained  $\text{Bi}_2\text{O}_3$  and  $\text{W}_{18}\text{O}_{49}$  were respectively fabricated into electrodes for further electrochemical studies. As shown in Fig. 4d,  $\text{Bi}_2\text{O}_3$  with oxygen vacancies shows negligible activity for OER, while the current density of  $\text{W}_{18}\text{O}_{49}$  containing oxygen vacancies can reach  $2.42 \text{ mA cm}^{-2}$  at an overpotential of 0.54 V. On the other hand, in LSV curves of  $\text{Bi}_2\text{WO}_6$  CNPs (Fig. 4a) and  $\text{W}_{18}\text{O}_{49}$  nanowires (Fig. 4d), they show similar anodic peaks located at *ca.* 1.41 V vs. RHE, which are probably ascribed to low valent W to high valent one. The above results suggest that the electrocatalytic activity of  $\text{Bi}_2\text{WO}_6$  CNPs originates from perovskite-like slab of  $[\text{WO}_4]^{2-}$ . In order to further support this conclusion, a catalyst poisoning test was carried out on the assumption that the catalytic active center was located at  $(\text{Bi}_2\text{O}_2)^{2+}$  slab. Potassium tartrate can form stable bismuth potassium tartrate complex with  $\text{Bi}^{3+}$  ions on the surfaces of  $\text{Bi}_2\text{WO}_6$  CNPs and thus will poison the OER activity of  $\text{Bi}_2\text{WO}_6$  CNPs. When a certain amount of potassium tartrate was added into  $\text{Na}_2\text{SO}_4$  electrolyte solution, on the contrary, the resultant OER activity almost kept unchanged (Fig. S13). This result shows that the catalytic active center of  $\text{Bi}_2\text{WO}_6$  CNPs originates from  $[\text{WO}_4]^{2-}$  slab rather than  $(\text{Bi}_2\text{O}_2)^{2+}$  layer. Considering to electrochemically active surface area (ECSA) of  $\text{W}_{18}\text{O}_{49}$  nanowires was 2.6-times larger than that of  $\text{Bi}_2\text{WO}_6$  CNPs (see supporting information, Fig. S14), while the catalytic current density for the latter is higher than the former. Therefore we can conclude that the intrinsic electrocatalytic activity of  $\text{Bi}_2\text{WO}_6$  CNPs is obviously superior to that of  $\text{W}_{18}\text{O}_{49}$  nanowires.

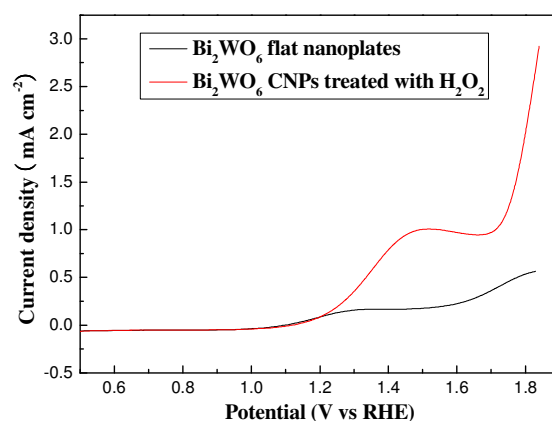
Oxygen vacancies played key roles in enhancing OER activity of  $\text{Bi}_2\text{WO}_6$  CNPs. As above mentioned analyses, oxygen vacancies are related to low valent W, which is manifested by strong absorption in near-infrared region. Therefore the concentration of oxygen vacancies could be adjusted by controlled oxidation reaction between low valent W and appropriate oxidant such as  $\text{H}_2\text{O}_2$ .<sup>36</sup> To be specific, 100 mg of  $\text{Bi}_2\text{WO}_6$  CNPs was treated with 40 mL of  $\text{H}_2\text{O}_2$  solution with concentrations of 3.75 (sample 1), 15 (sample 2), 30 (sample 3), and 60 mg/mL (sample 4) for 1 h, respectively. As UV-vis absorption spectra shown in Fig. 5a, the absorbance in near-infrared region gradually declined for  $\text{Bi}_2\text{WO}_6$  CNPs after treatments with increasing the amount of  $\text{H}_2\text{O}_2$ . The results show that the oxygen vacancies on the surfaces of the  $\text{Bi}_2\text{WO}_6$  CNPs gradually decreased with increasing consumption of  $\text{H}_2\text{O}_2$ . The corresponding  $\text{Bi}_2\text{WO}_6$  CNPs with different amounts of oxygen vacancies displayed distinct OER activities. As shown in Fig. 5b, fewer oxygen vacancies  $\text{Bi}_2\text{WO}_6$  CNPs have, weaker OER activity they show. Usually, more oxygen

vacancies can collect more electrons and lead to enhanced conductivity of the material.<sup>28</sup> The electrochemical impedance of  $\text{Bi}_2\text{WO}_6$  CNPs with different amounts of oxygen vacancies further confirmed this point. As shown in Fig. 5c, the Nyquist plots indicate that the charge-transfer resistance of  $\text{Bi}_2\text{WO}_6$  CNPs varied with the amounts of oxygen vacancies. Untreated  $\text{Bi}_2\text{WO}_6$  CNPs with the largest content of oxygen vacancies exhibited the smallest charge-transfer resistance. With decreasing oxygen vacancies controlled through the treatment of increasing  $\text{H}_2\text{O}_2$  doses, the resultant  $\text{Bi}_2\text{WO}_6$  CNPs showed larger and larger charge-transfer resistance. Therefore the presence of oxygen vacancies really benefits the transfer of charge. On the other hand, the oxygen vacancies on the surfaces of  $\text{Bi}_2\text{WO}_6$  CNPs also favor the adsorption of  $\text{H}_2\text{O}$  molecules. In order to illuminate this point, the adsorption energy of  $\text{H}_2\text{O}$  molecules on a perfect surface and a surface with one oxygen vacancy was calculated based on density functional theory (DFT) model (Fig. 5d, detailed method see Supporting Information). As a result, the adsorption energy of  $\text{H}_2\text{O}$  molecules on a surface with one oxygen vacancy is 29.5 kcal/mol, obviously larger than that of  $\text{H}_2\text{O}$  molecules on a perfect surface (14.5 kcal/mol). This means that  $\text{Bi}_2\text{WO}_6$  CNPs with oxygen vacancies are more favorable for adsorbing  $\text{H}_2\text{O}$  molecules. This feature is conducive to OER because the adsorption of  $\text{H}_2\text{O}$  is a rate-determining process.



**Fig. 5** UV-visible diffuse reflectance spectra (a), LSV curves (b), and electrochemical impedance spectra (c) of original  $\text{Bi}_2\text{WO}_6$  CNPs and  $\text{Bi}_2\text{WO}_6$  CNPs after treated with different concentrations of  $\text{H}_2\text{O}_2$  solution (Sample 1-4). Inset in Fig. 5c: electrochemical impedance spectrum of original  $\text{Bi}_2\text{WO}_6$  CNPs. Adsorption models of  $\text{H}_2\text{O}$  molecule on a perfect surface and a surface with one oxygen vacancy (d).

The good OER activity was also attributed to exposed high energy facets on the surfaces of  $\text{Bi}_2\text{WO}_6$  CNPs except for oxygen vacancies. In order to demonstrate this point, flat  $\text{Bi}_2\text{WO}_6$  nanoplates containing oxygen vacancies were synthesized through a modified hydrothermal route.<sup>40</sup> Compared with  $\text{Bi}_2\text{WO}_6$  CNPs, the obtained flat  $\text{Bi}_2\text{WO}_6$  nanoplates have a similar side length but thinner thickness (15 nm, Fig. S15 and S16). Flat  $\text{Bi}_2\text{WO}_6$  nanoplates have a 4.2-times larger ECSA compared with  $\text{Bi}_2\text{WO}_6$  CNPs (Fig. S17). The flat  $\text{Bi}_2\text{WO}_6$  nanoplates also have higher oxygen vacancies contents than  $\text{Bi}_2\text{WO}_6$  CNPs after treated with 40 mL of  $\text{H}_2\text{O}_2$  (60 mg/mL), roughly estimated from their corresponding UV-vis absorption spectra in near-infrared region (Fig. S18). Although larger ECSA and higher oxygen vacancies contents, flat  $\text{Bi}_2\text{WO}_6$  obviously exhibited lower OER activity than the  $\text{Bi}_2\text{WO}_6$  CNPs sample treated with  $\text{H}_2\text{O}_2$  (Fig. 6). To be specific, at an applied voltage of 1.77 V vs RHE, the current density of  $\text{Bi}_2\text{WO}_6$  CNPs after treated with  $\text{H}_2\text{O}_2$  ( $1.47 \text{ mA cm}^{-2}$ ) is 3 times larger than that of flat  $\text{Bi}_2\text{WO}_6$  nanoplates ( $0.49 \text{ mA cm}^{-2}$ ). These results clearly show that the fabrication of concave facets with high energy facets can significantly enhance OER activity of electrocatalyst.



**Fig. 6** LSV curves of  $\text{Bi}_2\text{WO}_6$  nanoplates and  $\text{Bi}_2\text{WO}_6$  CNPs after treated with 40 mL of  $\text{H}_2\text{O}_2$  (60 mg/mL).

#### 4. Conclusion

In summary, we have reported  $\text{Bi}_2\text{WO}_6$  CNPs with oxygen vacancies as a new electrocatalyst for efficient water oxidation under near-neutral condition for the first time. From experimental results, the presence of oxygen vacancies in  $\text{Bi}_2\text{WO}_6$  CNPs could greatly reduce charge-transfer resistance in the process of electrochemical reaction. Based on the theoretical calculations, the oxygen vacancies on the surfaces of  $\text{Bi}_2\text{WO}_6$  CNPs favor the adsorption of  $\text{H}_2\text{O}$  molecules and decrease the OER barrier, resulting in improved water oxidation activity. On the other hand, the concave surfaces with high energy crystalline facets considerably enhanced electrocatalytic activity of  $\text{Bi}_2\text{WO}_6$  nanoplates. As a result, the synergistic effect of oxygen vacancies and concave surfaces endowed impressive performance of  $\text{Bi}_2\text{WO}_6$  CNPs for OER with a large current density at relatively small overpotential and good durability in near-neutral water, which is comparable to the best electrocatalysts among known inorganic non-precious metal compounds. The strategy based on the combination of high energy crystalline facets and oxygen vacancies could open a new avenue for designing and developing new and highly efficient electrocatalysts.

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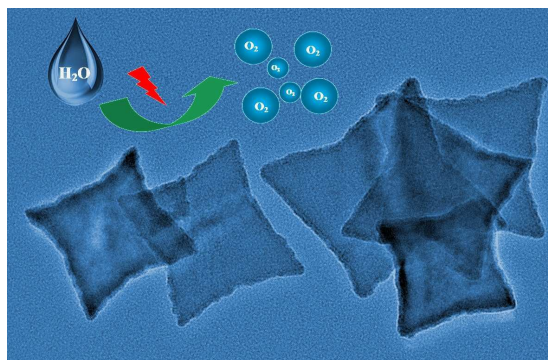
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## Graphical Abstract



Concave  $\text{Bi}_2\text{WO}_6$  nanoplates with oxygen vacancies firstly achieved enhanced electrocatalytic oxygen evolution in near-neutral water.