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**A Three-Dimensional MnO<sub>2</sub>/Graphene hybrid as binder-free supercapacitor  
electrode**

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**Abstract:** Highly aligned manganese dioxide ( $\text{MnO}_2$ ) nanowall arrays electrodeposited onto Ti sheets are used as substrates to grow graphene (GR) through chemical vapor deposition (CVD), thus forming three-dimensional (3D)  $\text{MnO}_2/\text{GR}$  hybrid composite. Furthermore, 3D  $\text{MnO}_2/\text{GR}$  hybrid with different structures and properties has been prepared at different temperatures. The as-prepared hybrid materials could be directly used as supercapacitor electrodes without any binder and conductive additive, and fully maintain the high conductivity and high specific area of GR, large pseudocapacitance of  $\text{MnO}_2$  nanowall arrays. In aqueous electrolytes, the hybrids show a high specific capacitance of  $\sim 326.33 \text{ F g}^{-1}$  with good cycling stability at the scan rate of  $200 \text{ mV s}^{-1}$  and high energy density of  $23.68 \text{ Wh kg}^{-1}$  while maintaining high power density of  $7270 \text{ W kg}^{-1}$ . The preparation method provides a novel road to fabricate 3D graphene-based composite materials, and the so obtained hybrid electrode demonstrates its potential applications in supercapacitors.

**Keywords:** Manganese dioxide nanowall arrays, Electrodeposited, Graphene, Three-dimensional, Energy density, Supercapacitor

## 1. Introduction

In response to the gradual depletion of fossil fuels and the increasing severity of environmental pollution problems, developing sustainable and renewable energy storage devices has become increasingly urgent in order to meet the future demands. Supercapacitors are regarded as an efficient and environmental friendly energy storage and supply technology for its high power density and low pollution. Such “high

power-density” devices, however, has low energy density compared to batteries. To improve the performance and promote the commercialization of supercapacitors, it is important to explore new electrode materials with low cost, high performance and environmental benignity. Carbon-based materials, as supercapacitor electrodes are widely concerned [1-7] due to their low cost and good electrical conductivity. Thereinto graphene (GR) is the ideal electrode materials because of the outstanding electrical conductivity, large specific surface area and mechanical properties. So far many methods have been developed to prepare graphene [8-12] including chemical vapor deposition (CVD) [8], micromechanical cleavage [9,10], and chemical reduction of graphite oxide (GO) [11,12]. Among these methods, chemical reduction of GO is the predominant method in the mass production of graphene for industrial applications. However, GR exhibits an undesired specific surface area ( $200\text{-}500\text{ m}^2\text{ g}^{-1}$ ) far below its theoretical value ( $2675\text{ m}^2\text{ g}^{-1}$ ) due to the intensive stacking among these two dimensional sheets. To solve this problem, 3D GR has become a hot research topic [13,14] because this structure could sufficiently maintain the features of individual GR and generate a large specific surface area. Shi *et al.* reported a 3D mechanically strong, electrically conductive, and thermally stable self-assembled GR hydrogel (SGH) with a high specific capacitance, which was prepared from GO by a one-step hydrothermal method [13], however, the as-prepared composite suffered from poor conductivity with the presence of a large amount of water. Bong *et al.* fabricated assembled embossed chemically modified GR film by using polystyrene colloidal particles as a sacrificial template and explored its use in supercapacitor electrodes [14]. Nevertheless,

this three-dimensional GR structure is unstable and easy to collapse.

Currently, metal oxides have been extensively applied as the spacer to separate GR nanosheets in view of their high pseudocapacitance [15-17]. Among metal oxides, manganese dioxide ( $\text{MnO}_2$ ) is regarded as a promising candidate for supercapacitors due to its low cost, high natural abundance, high theoretical capacity ( $\sim 1370 \text{ F g}^{-1}$ ), and non-toxicity [18,19]. More significantly, unlike other electrode materials, which should be used in strong acidic or alkaline electrolytes,  $\text{MnO}_2$  can be used in neutral aqueous electrolytes, and hence can meet the requirements for “green electrolyte” in supercapacitors. Furthermore, compared to some traditional backing materials such as Ni, Co, Pt, Ru *et al.* [20],  $\text{MnO}_2$  used as a substrate to grow GR is beneficial to reduce CVD reaction temperature, however, the low surface area and poor electronic conductivity ( $10^{-5}$ - $10^{-6} \text{ S cm}^{-1}$ ) remain the major problems. One promising approach is to fabricate 3D  $\text{MnO}_2/\text{GR}$  composite structures, where the hybrid can obtain the synergistic effect of individual constituents. Various approaches to synthesize 3D  $\text{MnO}_2/\text{GR}$  electrodes have been reported [15-17], including physical mixing, microwave-assisted method, chemical co-precipitation and electrochemical deposition. For instance,  $\text{MnO}_2/\text{GR}$  electrodes synthesized by soft chemical route exhibited a specific capacitance (SC) of  $210 \text{ F g}^{-1}$  [15] and those produced by microwave-assisted method displayed SC of  $310 \text{ F g}^{-1}$  [16]. Cheng *et al.* reported a maximum energy density of  $30.4 \text{ Wh kg}^{-1}$  and power density of  $5 \text{ kW kg}^{-1}$  using GR// $\text{MnO}_2/\text{GR}$  hybrid cells [17].

In this work, an effective two-step approach is developed to fabricate high-quality but

low cost 3D MnO<sub>2</sub>/GR hybrids through one-step CVD and an electrodeposition method. The so-obtained 3D MnO<sub>2</sub>/GR hybrid presents a high specific capacitance of 326.33 F g<sup>-1</sup> at the scan rates of 200 mV s<sup>-1</sup> with capacitance retention of ~92% and a high energy density of 23.68 Wh kg<sup>-1</sup> after 1000 cyclic voltammogram (CV) cycles.

## 2. Experimental

### 2.1 Preparation of substrate of aligned MnO<sub>2</sub> nanowall arrays

The precursor solutions were prepared by dissolving manganese acetate, Mn(CH<sub>3</sub>COO)<sub>2</sub>·4H<sub>2</sub>O (99+%, Alfa Aesar) and anhydrous sodium sulfate, Na<sub>2</sub>SO<sub>4</sub> (99%, J. T. Baker) into deionised (DI) water in a concentration of 0.15 M for both precursors. The solution was stirred at room temperature for 1 h to ensure the complete dissolution of both solutes prior to the electrodeposition experiments.

Schematic illustration of electrodeposition of the MnO<sub>2</sub> nanowall arrays was presented in Figure S1. Highly aligned MnO<sub>2</sub> nanowall arrays were obtained by cathodic electrodeposition onto a Ti substrate with an exposed surface area of 1 cm<sup>2</sup>. Ti sheets were rinsed with ethanol and acetone before electrochemical deposition. The anode and cathode of the adjustable constant voltage source is connected to two same Ti sheets, separately (The horizontal distance between the two sheets is 2 cm). The Ti sheets were soak in a mixed aqueous solution of manganese acetate and sodium sulfate. The deposition was induced by a change of the voltage (voltage was maintained between 5.0 - 8.0 V). The deposition time was controlled within 15 mins.

### 2.2 Fabrication of MnO<sub>2</sub>/GR composites

The preparation process was shown in Scheme 1. The aligned MnO<sub>2</sub> nanowall arrays

on the surface of Ti substrate were placed into the tube furnace and heated to 600, 700 °C, individually at the heating rate of 15 °C min<sup>-1</sup> under Ar (200 s.c.c.m.) and H<sub>2</sub> (100 s.c.c.m.). 0.25 mL of benzene used as carbon source was injected into the furnace at the rate of 0.01 mL min<sup>-1</sup> for 30 mins. After 5 min of reaction-gas mixture flow, the samples were rapidly cooled to room temperature at a rate of 100 °C min<sup>-1</sup> under Ar (200 s.c.c.m.) and H<sub>2</sub> (100 s.c.c.m.). A series of 3D MnO<sub>2</sub>/GR composites were prepared and denoted as MnO<sub>2</sub>/GR600, MnO<sub>2</sub>/GR700, MnO<sub>2</sub>/GR800, respectively.

### 2.3. Material characterization

The morphologies and microstructures of the samples were examined by scanning electron microscopy (SEM; XL30), transmission electron microscopy (TEM; Hitachi H-600), and High Resolution TEM (HRTEM). The crystal structures of the as-prepared products were characterized by an X-ray diffractometer (XRD; D/Max 2500 V PC<sup>-1</sup>, Cu-Kα radiation) with a scan speed of 2 min<sup>-1</sup>. Raman spectra were collected through a Renishaw 2000 model confocal microscopy Raman spectrometer with a CCD detector and a holographic notch filter (Renishaw Ltd., Gloucestershire, U.K.) at ambient conditions, using the radiation of 514.5 nm from an air-cooled argon ion laser to excite the SERS.

### 2.4. Electrochemical measurements

The electrochemical experiments were carried out using a two-electrode system at ambient temperature, with 1.5 M Li<sub>2</sub>SO<sub>4</sub> as the electrolyte solution. The as-synthesized materials were directly used as the working electrode and the counter electrode, respectively. Electrochemical measurements were all conducted on a CHI 660D

electrochemical workstation (Shanghai, Chenhua). The CV curves of the electrodes were measured between 0 and 1.2 V at different scan rates from 50 to 400 mV s<sup>-1</sup>. The galvanostatic cycling for each electrode was performed in the potential range from 0 to 1 V at different current densities of 1, 1.5, 2, 2.5 and 3 A g<sup>-1</sup>, and electrochemical impedance spectroscopy (EIS) measurements were carried out in the frequency range of 100 kHz ~ 1 Hz.

### 3. Results and discussion

#### 3.1. Structure and morphology

Fig. 1A presents the photographs of Ti, MnO<sub>2</sub>, MnO<sub>2</sub>700 (MnO<sub>2</sub> heated in 700 °C) and MnO<sub>2</sub>/GR composites at different temperatures. It can be seen that the white original Ti plate changes to yellow after electrodeposition of MnO<sub>2</sub>. After GR grown on the surface of MnO<sub>2</sub> through CVD, the color of the materials changes gradually into black, which indicates that the surface of MnO<sub>2</sub> is coated with a layer of carbon. In order to investigate whether the morphology of MnO<sub>2</sub> is destroyed at high temperature, a comparison of the pristine MnO<sub>2</sub> and heated MnO<sub>2</sub> is necessary. Fig. 1B,C show two SEM images of MnO<sub>2</sub> before and after the heating process (700 °C). Clearly, electrodeposition synthesis of the flower-like MnO<sub>2</sub> nanostructures can be observed before and after the heating process, which demonstrates that the morphology of MnO<sub>2</sub> is not destroyed after the heating process. More significantly, the flower-like MnO<sub>2</sub> nanostructures can provide the high accessibility of electrolytic ions for shorten diffusion paths, which is beneficial to obtain good electrochemical capacitance performance. Unfortunately, when the MnO<sub>2</sub> sample is heated to 800 °C (Figure S2),



the flower-like MnO<sub>2</sub> nanostructures has been destroyed. Fig. 1D,E show the SEM images of GR grown on the surfaces of MnO<sub>2</sub> nanowall substrates at different temperatures (600 and 700 °C), respectively. The surfaces of MnO<sub>2</sub> nanowall substrates are both covered with a layer of GR, which will be further confirmed in the Raman test.

The XRD patterns of the GR grown on the surface of the MnO<sub>2</sub> nanowall substrates at 600 and 700 °C are shown in Fig. 2A. Two characteristic peaks of the MnO<sub>2</sub> at  $2\theta$  around 37° and 66° are displayed, which can be indexed to birnessite-type MnO<sub>2</sub> (JCPDS 42-1317) [21]. Furthermore, from the XRD analysis, the d-spacing of MnO<sub>2</sub> is about 7.0 Å, which is consistent with the high magnification TEM image (Fig. 3B) and the previous report [22]. The diffraction peaks at  $2\theta$  around 43° that is in keeping with the literature report [23] indicates that GR has been obtained successfully on the surface of the MnO<sub>2</sub> nanowall arrays after CVD.

The structure of 3D MnO<sub>2</sub>/GR composites are further confirmed by Raman spectra (Fig. 2B). Except from the D band (about 1350 cm<sup>-1</sup>) and G band (1590 cm<sup>-1</sup>) associated with GR, the major Raman bands located at 500-700 cm<sup>-1</sup> for MnO<sub>2</sub>/GR sample matches well with the major vibrational features of MnO<sub>2</sub> compounds reported previously [22]. Furthermore, compared to the value of  $I_D/I_G$  at 600 °C, the lower value at 700 °C indicates fewer defects. Consequently, the following tests are based on MnO<sub>2</sub>/GR700.

The microstructure of the MnO<sub>2</sub>/GR700 composite is further investigated by TEM (see Fig. 3). Obviously, the GR sheets are closely attached on the surface of the MnO<sub>2</sub> nanowall arrays (Fig. 3A). The flower-like morphology is the characteristic of the

birnessite-type  $\text{MnO}_2$  [24,25]. As shown in Fig. 3B, the interplanar spacing of  $\text{MnO}_2$  nanowall arrays and the GR sheets are measured to be 0.7 and 0.34 nm, respectively, which are in line with the literature report for birnessite-type  $\text{MnO}_2$  [24]. To investigate the morphology of GR, the TEM image of GR is examined by etching away  $\text{MnO}_2$  with 3 M HCl. The ultrathin morphology of the derived GR can be seen clearly in Fig. 3C. The related EDS spectrum (Fig. 3D) reveals the Mn, O signal from  $\text{MnO}_2$  as expected. The C signal is derived from the GR and the Ti signal is derived from the Ti sheets.

### 3.2. Electrochemical properties

To evaluate the electrical behaviour of the as-fabricated 3D  $\text{MnO}_2/\text{GR700}$  hybrids, the electrochemical capacitive performance for each type of electrode is measured by CV, galvanostatic charge-discharge (GCD) and EIS tests. The CV profiles of electrodeposition of  $\text{MnO}_2/700$  nanowall arrays (Figure S3) at different scan rates from 50 to 400  $\text{mV s}^{-1}$  show nearly rectangular shapes, indicating good pseudocapacitive behavior attributed to a continuous and reversible Faradaic redox transition of  $\text{MnO}_2$  over the potential range. Furthermore, Fig. 4A shows the CV curves of the  $\text{MnO}_2/\text{GR700}$  hybrid at various scan rates ranging from 50 to 400  $\text{mV s}^{-1}$  within an electrochemical window from 0 to 1.2 V. As the scan rate increases, the current response also increases without any obvious change in the shape of the CV curves, which presents a good rate performance. The rectangular and symmetric shape of the CV curve is also observed at a high scan rate of 400  $\text{mV s}^{-1}$  due to the low contact resistance of the electrode. Fig. 4B displays a comparison of CV curves of  $\text{MnO}_2$ ,  $\text{MnO}_2/\text{GR600}$ ,  $\text{MnO}_2/\text{GR700}$  and  $\text{MnO}_2/\text{GR800}$  hybrids at a scan rate of 200  $\text{mV s}^{-1}$ . The specific

capacitance can be calculated according to the following equation: [26,27]

$$C_m = \frac{\int_{V_1}^{V_2} |I| dV}{2mv\Delta V}$$

where  $C_m$  is the specific capacitance ( $F g^{-1}$ ),  $I$  is the response current (A),  $m$  ( $\sim 0.1$  mg) is the mass of the electroactive materials (g), and  $\Delta V$  is the potential window during the CV measurements process (V). The specific capacitances of  $MnO_2$ ,  $MnO_2/GR600$ ,  $MnO_2/GR700$  and  $MnO_2/GR800$  electrodes are  $183.23 F g^{-1}$ ,  $208.76 F g^{-1}$ ,  $326.33 F g^{-1}$  and  $136.28 F g^{-1}$ , respectively. Hence, we can confirm that the GR supplies a large double layer capacitance while  $MnO_2$  provides a large pseudocapacitive contribution to the capacitance of  $MnO_2/GR$ . In addition, more than 60% decrease in the capacitance of  $MnO_2/GR800$  supports the SEM analysis of  $MnO_2/GR800$  hybrid (Figure S2). Therefore, the optimal temperature for the growth of GR on  $MnO_2$  substrates should be  $700 ^\circ C$ . Furthermore, the picture in Fig. 4C is the CV curve of  $MnO_2$  at the scan rate of  $100 mV s^{-1}$ . It presents a bigger voltage drop compared with the  $MnO_2/GR$  hybrid, which implies that the internal resistance has decreased largely after the growth of GR on the surface of the  $MnO_2$  nanowall arrays. This is because GR in the hybrid acts as electronic conductive channels to increase the electrical conductivity, which matches well with the GCD curves of the  $MnO_2/GR$  (Fig. 4D). Fig. 4D shows GCD curves for  $MnO_2$ ,  $MnO_2/GR700$  electrodes at the current density of  $3.0 A g^{-1}$ . It can be seen that the curve are highly linear and symmetrical without obvious voltage drop compared with the  $MnO_2$ , indicating a rapid I-V response, an excellent electrochemical reversibility and little overall resistance of this material (more data about the GCD curves of  $MnO_2/GR700$  electrode at different current densities of 1, 1.5, 2, 2.5 and 3 A

$\text{g}^{-1}$  are given in Figure S4 in the support information). It can be seen from Figure S4 that all the GCD curves of  $\text{MnO}_2/\text{GR700}$  at different current densities are highly linear and almost typical isosceles triangular in shape, which further demonstrates that the electrode has ideal capacitive characteristics and excellent electrochemical reversibility. EIS measurements are carried out in a frequency range of 100 kHz to 1 Hz to further evaluate the electrochemical and structural characteristics of the electrode material (Fig. 4E). Compared with the  $\text{MnO}_2$ , the EIS Nyquist plot of the  $\text{MnO}_2/\text{GR700}$  has a shorter arc located at high frequencies followed by an inclined line with a slope about  $45^\circ$ . The span of the arc is indicative of the charge-transfer resistances of the electrode materials [28]. It can be clear from Fig. 4E that, the span of the arc in Nyquist plot decreases when GR is added, which is consistent with the results of the CV and GCD studies. Obviously, the low resistances of  $\text{MnO}_2/\text{GR700}$  electrode should stem from their intrinsic high conductivity, rich porous 3D network and the flower-like morphology of  $\text{MnO}_2$ . Fig. 4F displays the cyclic stability of  $\text{MnO}_2/\text{GR700}$  at the scan rate of  $200 \text{ mV s}^{-1}$  for 1000 cycles. It is clear that the  $\text{MnO}_2/\text{GR700}$  electrode presents long-term cycle stability with capacitance retention of  $\sim 92\%$  after 1000 CV cycles (approximately  $300 \text{ F g}^{-1}$ ). These results fully reflect the merit of the 3D architecture cherished by the  $\text{MnO}_2/\text{GR}$  electrode.

Energy and power densities are two crucial parameters for evaluating the electrochemical performance of supercapacitors. The energy density (  $E$  in  $\text{Wh kg}^{-1}$  ) and power density (  $P$  in  $\text{W kg}^{-1}$  ) of the electrode can be further evaluated from the CV curves at different scan rates using the following equations (1) and (2), respectively:

[29]

$$E = C_m \Delta V^2 / 2 \quad (1)$$

$$P = E / t \quad (2)$$

where  $C_m$  is the specific capacitance ( $F g^{-1}$ ),  $\Delta V$  is the potential window during the CV measurements process (V) and  $t$  is the discharge time of the CV curves (s). Fig. 5 exhibits the energy and power density for the  $MnO_2/GR700$  hybrid electrode, where the reported data of some typical energy storage device such as batteries [30] and  $MnO_2$ -carbon based supercapacitors [18,31-36]. By comparison, our work displays a high energy density of  $23.68 Wh kg^{-1}$  while maintaining high power density of  $7270 W kg^{-1}$  at the scan rate of  $200 mV s^{-1}$ , which can be comparable with the previously reported high-performance energy-storage devices (the related data are displayed in the Table S1 in the support information). In addition, more data for large mass loading (1 and 3 mg) are also given in Figure S5 (A, B) in the support information. Obviously, the results are consistent with the analysis results of Fig. 4A, and the energy and power densities can reach to  $24.86, 25.13 Wh kg^{-1}$  and  $6997, 6812 W kg^{-1}$  at scan rate of  $200 mV s^{-1}$ , respectively, which also displays high-performance electrochemical energy-storage of 3D  $MnO_2/GR700$  electrode. Thus, it is to be expected: the as-fabricated 3D  $MnO_2/GR$  hybrid has enormous potential applications in supercapacitors.

#### 4. Conclusions

In summary, a novel strategy has been put forward to fabricate 3D  $MnO_2/GR$  hybrid through chemical vapor deposition on the substrate of aligned  $MnO_2$  nanowall arrays

that were electrochemically deposited onto Ti sheets. An optimal condition of preparation has been also obtained by comparing the electrical behaviour of MnO<sub>2</sub>/GR hybrid electrodes fabricated at different temperatures. The as-synthesized MnO<sub>2</sub>/GR700 electrode shows a specific capacitance of 326.33 F g<sup>-1</sup> at the scan rate of 200 mV s<sup>-1</sup> and can be cycled reversibly in the voltage region of 0 to 1.2 V. Moreover, it can maintain an energy density of 23.68 Wh kg<sup>-1</sup> even at the power density of 7270 W kg<sup>-1</sup>, which indicates the high-efficiency energy storage of MnO<sub>2</sub>/GR700 hybrid. We expect this work will open up new opportunities for the application of low-cost, high-efficiency 3D GR-based electrodes materials in new energy electric vehicle.

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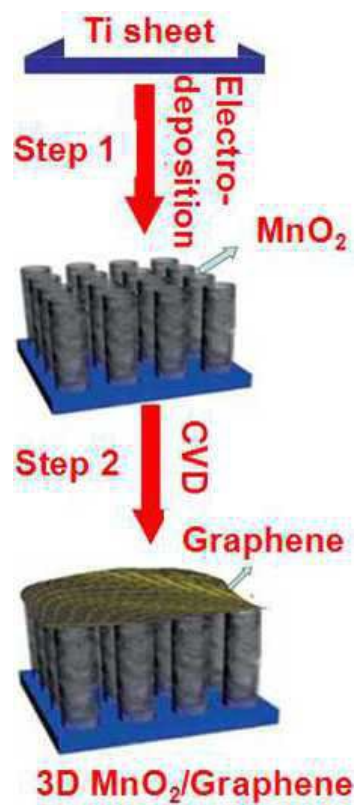
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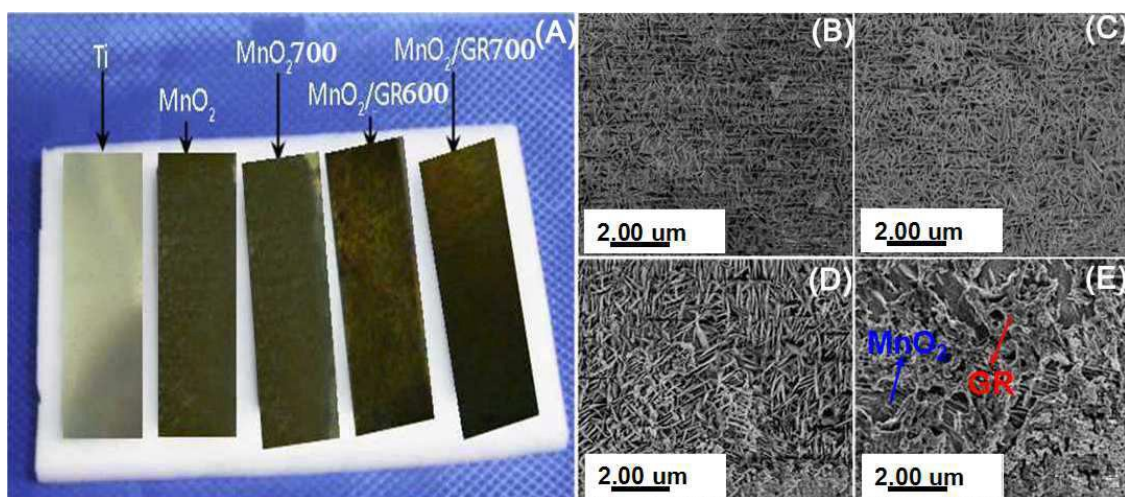
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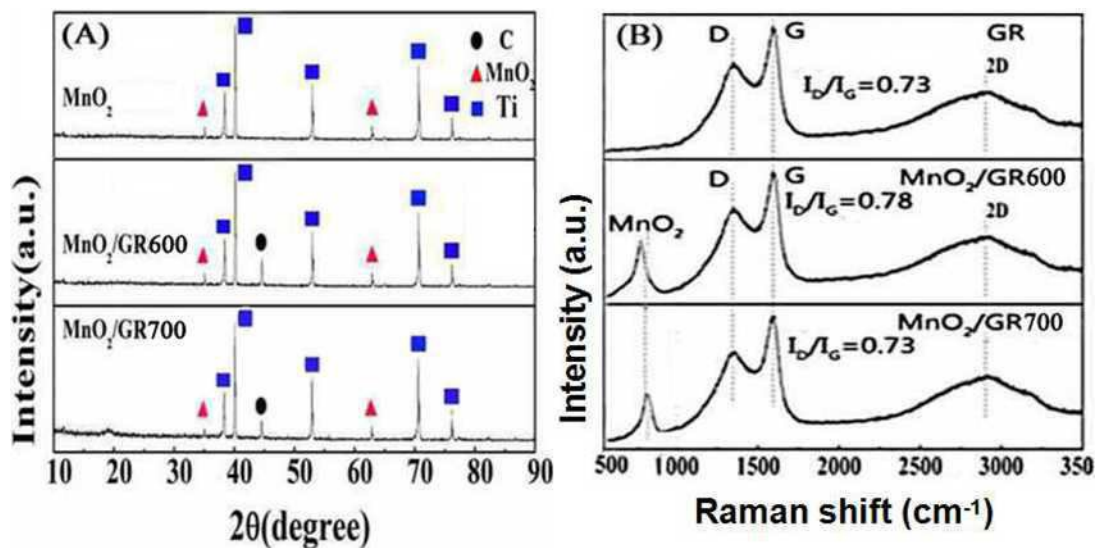




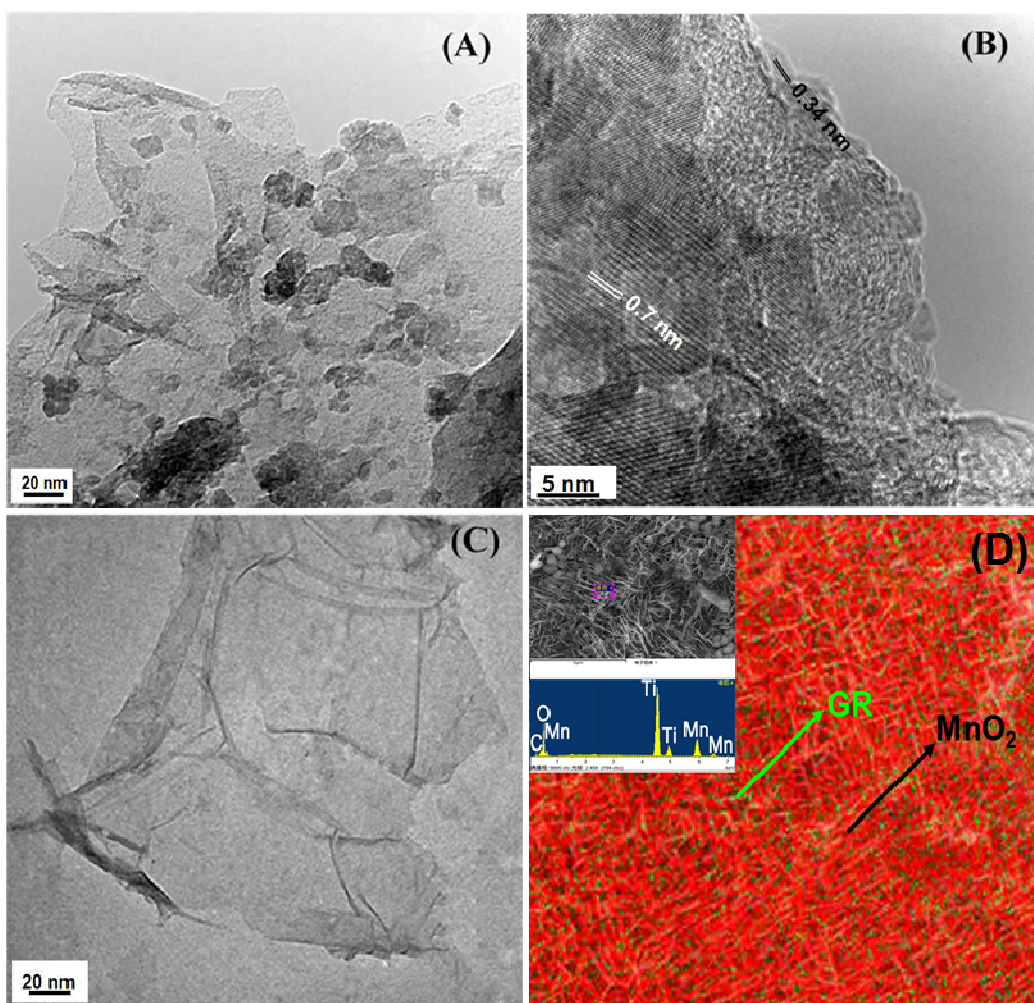
**Scheme 1.** Synthetic scheme of the 3D MnO<sub>2</sub>/GR composite.



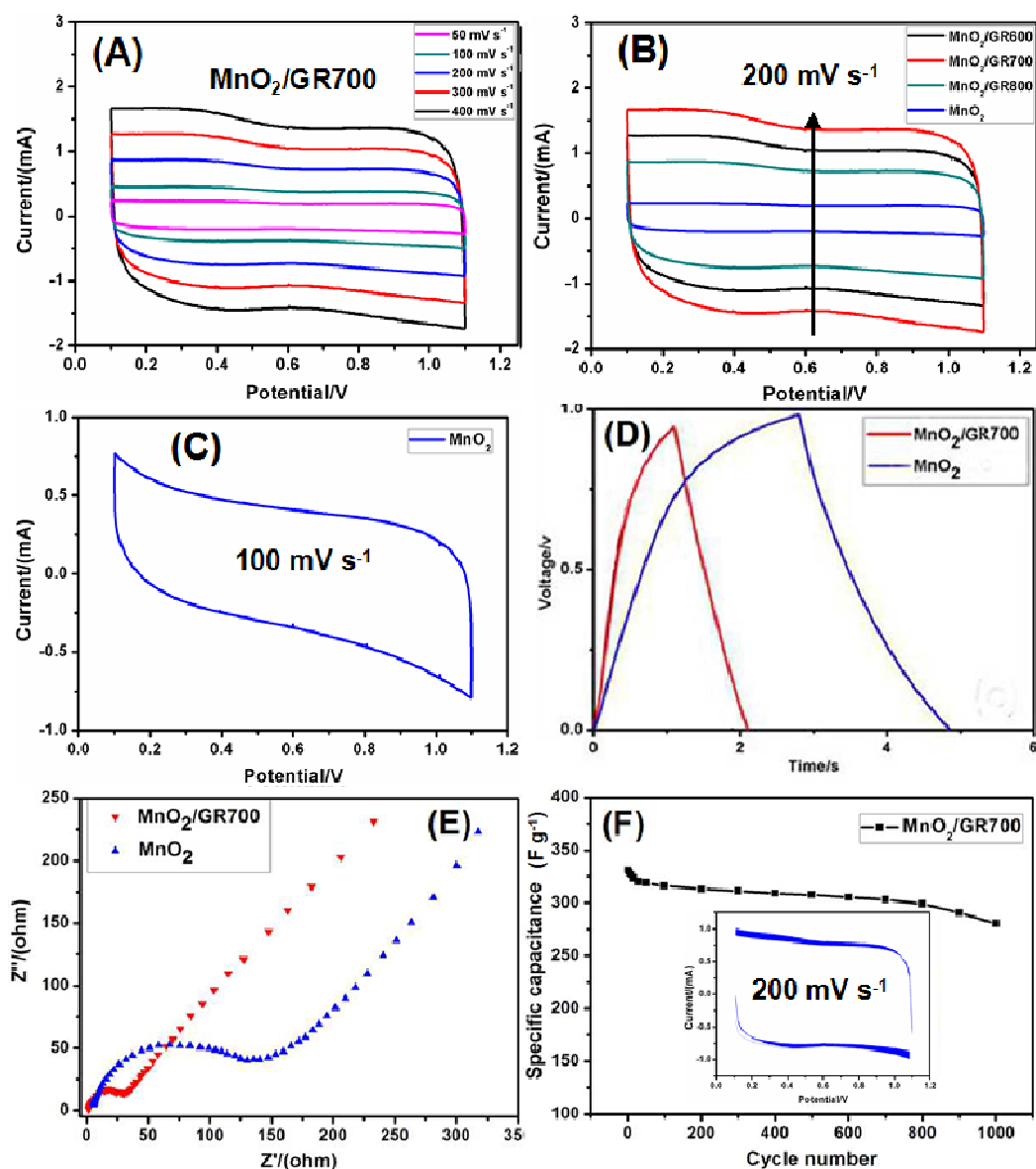
**Fig. 1.** (A) Photographs of Ti, MnO<sub>2</sub>, MnO<sub>2</sub>700 (MnO<sub>2</sub> heated in 700 °C), and MnO<sub>2</sub>/GR composites at different temperature. (B,C) SEM images of MnO<sub>2</sub> before and after the heating. (D,E) SEM images of 3D MnO<sub>2</sub>/GR in 600 and 700 °C, respectively.



**Fig. 2.** (A) XRD patterns of the MnO<sub>2</sub> nanowall electrodeposited and the GR grown on the surface of the MnO<sub>2</sub> nanowall substrates at 600 and 700 °C, respectively. (B) Raman spectra of pure GR (700 °C) after the MnO<sub>2</sub> nanowall was removed and the GR grown on the surface of the MnO<sub>2</sub> nanowall substrates at 600 and 700 °C, respectively.



**Fig. 3.** (A) Low and (B) high magnification TEM images of MnO<sub>2</sub>/GR700 composite, showing the graphene covered on the surface of the MnO<sub>2</sub> nanowall. (C) The TEM images of pure graphene after the MnO<sub>2</sub> nanowall was removed. (D) The scanning electron microscopic elemental map of 3D MnO<sub>2</sub>/GR composite.



**Fig. 4.** (A) CV curves of the MnO<sub>2</sub>/GR700 composite at different scan rates. (B) Comparison CV curves of the MnO<sub>2</sub>/GR600, MnO<sub>2</sub>/GR700, MnO<sub>2</sub>/GR800 and MnO<sub>2</sub> at the scan rate of 200 mV s<sup>-1</sup>. (C) CV curves of MnO<sub>2</sub> with absence of GR at the scan rate of 100 mV s<sup>-1</sup>. (D) Galvanostatic charge-discharge of the MnO<sub>2</sub>/GR700 and MnO<sub>2</sub>. (E) Ragone and Nyquist plots of the supercapacitors based on the MnO<sub>2</sub>/GR700 and MnO<sub>2</sub>. (F) Cycle life curves of the supercapacitors based on MnO<sub>2</sub>/GR700 composites at the scan rate of 200 mV s<sup>-1</sup>.

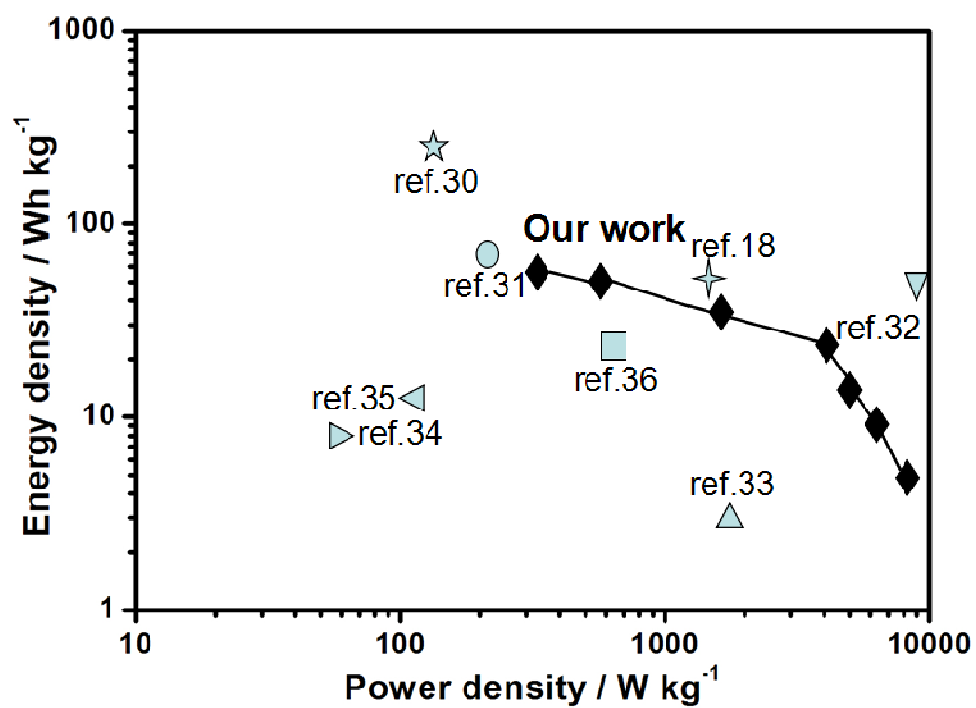


Fig. 5 Ragone plot of the energy density vs. power density for batteries and carbon-based supercapacitors.

## Graphical abstract

