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1 Fabrication of mechanically durable superhydrophobic wood surfaces
2 using polydimethylsiloxane and silica nanoparticles

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8

9 **Abstract:** The excellent properties of wood utilized as an engineering material are detracted by the
10 complex wood-water interactions and the resulting dimensional instability and low durability
11 against biological degradation. Inspired by the lotus effect, mechanically durable
12 superhydrophobic coatings were successfully fabricated on the intrinsically heterogeneous wood
13 substrates by simply dip-coating the suspensions of hydrophobic silica (SiO₂) nanoparticles
14 dispersed in polydimethylsiloxane (PDMS) solution. A subsequent low-surface-energy treatment
15 with some expensive reagents is not necessary owing to the hydrophobic nature of PDMS and the
16 modified silica particles. The surface microstructure, roughness and wetting behavior of the
17 PDMS/silica hybrid coatings on wood surfaces were investigated in relation to the loadings of the
18 silica particles in the PDMS matrix. When the silica particle loading reached a critical level,
19 desirable hierarchical micro/nanostructures were formed on the wood substrate, allowing for the
20 generation of superhydrophobicity with a contact angle of 152° and a sliding angle less than 10°. The
21 fabricated PDMS/silica hybrid coating exhibited desirable durability against mechanical
22 abrasion and high-frequency ultrasonic washing in water whilst basically retaining its
23 microstructure and superhydrophobicity. Such mechanically durable superhydrophobic wood
24 surfaces with self-cleaning properties offer an interesting alternative for wood modification, and
25 could improve the performance of wood as an engineering material.

26

27 **Keywords:** wood, superhydrophobicity, PDMS, silica nanoparticle, hybrid coating, mechanical
28 abrasion

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30 Introduction

31 Wood has long been used as an important engineering material owing to its excellent mechanical
32 properties, aesthetic appeal, and environmental friendliness. However, due to the hygroscopic
33 nature of amorphous cellulose and hemicelluloses in the cell walls, wood shrinks or swells upon
34 changes in moisture contents which can lead to substantial deformations (*e.g.* distortion, twisting
35 and cracking) of construction elements. The presence of high moisture also encourages colonial
36 growth of fungi in wood, resulting in degradation of cell walls and low durability of wood in
37 service.

38 Accordingly, many attempts have been made to improve the hydrophobicity of wood. The
39 transformation of wood from hydrophilicity to hydrophobicity mainly involves using reactive
40 chemicals to block hydroxyl groups of cell wall polymers to reduce water sorption sites in wood,
41 or incorporation of materials into the cell wall to fill microvoids within it, thereby occupying space
42 that would otherwise be available to water molecules, such as acetylation,^{1,2} silanization,^{3,4} *in-situ*
43 polymerization,^{5,6} and flavonoid insertion into cell walls.⁷ Although these methods can be used to
44 reduce or delay water/moisture absorption into wood, they cannot prevent water absorption upon
45 direct exposure of wood to liquid water. Moreover, such bulk hydrophobization of wood in deeper
46 layers is usually complex, costly and undesirable due to the complicated structure of wood. Hence,
47 surface modification and functionalization could be an interesting alternative.

48 Nature has offered many examples of superhydrophobic surfaces exhibiting great
49 water-repellent properties with water contact angles higher than 150°, such as plant leaves,⁸ rose
50 petals,⁹ and water strider's legs.¹⁰ Superhydrophobic surfaces have been attracting considerable
51 attention since they possess not only excellent water-repellency but also show self-cleaning,
52 anti-icing and anti-corrosive properties, and have great potential in various applications.¹¹⁻¹⁵
53 Superhydrophobic surfaces are expected to minimize wood-water interactions and thus avoid
54 problems associated with water absorption. Such system may serve as a desirable solution for
55 wood protection against water. It has been suggested that the superhydrophobic properties are a
56 result of suitable hierarchical micro- and nano-structures of the surface superimposed with
57 materials with low surface free energy.¹⁶

58 Wood substrate is intrinsically heterogeneous due to its surface anatomical structure and

59 porosity, forming primary roughness at the microscale. With the microscale roughness inherent on
60 the wood substrate, it is feasible to develop dual-scale roughness to fabricate superhydrophobic
61 wood surfaces by incorporating nanoscale materials coupled with low-surface-energy treatments.
62 Recently, artificial superhydrophobic coatings have been successfully developed on solid woods
63 by using various techniques to control the surface structure and roughness, such as sol-gel
64 process,^{17,18} hydrothermal method,^{19,20} wet chemical method,^{21,22} and plasma treatment.²³ However,
65 even with attractive nonwetting properties, the fabricated superhydrophobic wood surfaces are not
66 suitable for practical uses in real-life conditions because of the limited mechanical stability and
67 durability of the coatings. Without proper design, the surface hydrophobic layers and the
68 underlying topographical patterns of the superhydrophobic surfaces are prone to be damaged by
69 mechanical abrasion, leading to undesired pinning of water droplets on the surface and loss of
70 water-repellency. The practical application of superhydrophobic surfaces relies on wetting
71 robustness of the products as well as simple and inexpensive fabrication processes.

72 In this study, we present a simple and inexpensive dip-coating method to fabricate mechanically
73 durable superhydrophobic coating on the heterogeneous wood surfaces using
74 polydimethylsiloxane (PDMS) and silica nanoparticles, both of which are inexpensive and
75 environmentally friendly. As illustrated in Fig. 1, hydrophobic silica particles were prepared by
76 hydrolysis and condensation of tetraethoxysilane (TEOS) under an alkaline condition, followed by
77 hydrophobic modification with hexadecyltrimethoxysilane (HDTMS) to graft long-chain alkyl
78 group onto the surface. The HDTMS-coated silica particles were then dispersed into PDMS
79 solutions to form PDMS-silica suspensions, which were subsequently applied onto wood surfaces
80 by dip-coating (other techniques like spray-coating can also be used for large-area fabrication).
81 Heat treatment transformed the applied solutions into organic-inorganic hybrid coatings having
82 hierarchical nanotextured surface morphology, which was readily controlled by adjusting the mass
83 ratio of silica to PDMS. The as-prepared superhydrophobic wood surfaces showed desirable
84 mechanical stability and durability against abrasion damage and high-frequency ultrasonic
85 washing in water.

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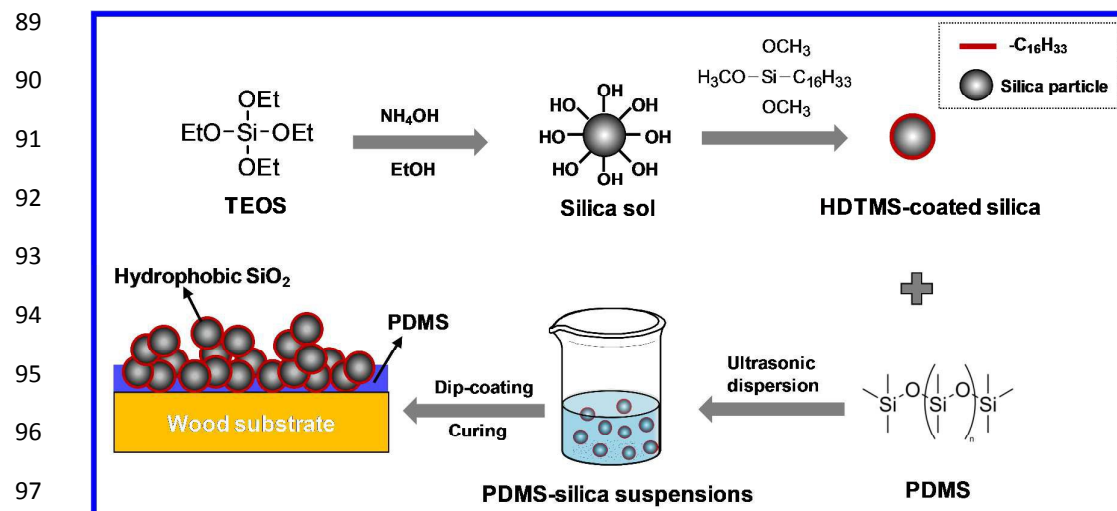


Fig. 1 Schematic illustration of the procedure to fabricate PDMS-silica hybrid superhydrophobic coatings on the wood substrate.

101 Experimental

102 Materials

103 Tetraethoxysilane (TEOS) and hexadecyltrimethoxysilane (HDTMS) were obtained from
 104 Sigma-Aldrich (St. Louis, Minnesota, USA). Ammonium hydroxide (NH₄OH, 25%),
 105 tetrahydrofuran (THF) and ethanol were purchased from Sinopharm Chemical Reagent Co., Ltd
 106 (Shanghai, China). Polydimethylsiloxane (PDMS Sylgard 184) and the corresponding curing agent
 107 were supplied by Dow Corning (Michigan, USA). All chemicals were used as received without
 108 further purification. Wood samples of Chinese fir (*Cunninghamia lanceolata*) were cut parallel to
 109 grain direction and sawn into blocks of 20 mm × 20 mm × 5 mm (radial × longitudinal ×
 110 tangential).

111 Synthesis of hydrophobic silica particles

112 Monodisperse silica sols were firstly synthesized by a typical Stöber method. Briefly, NH₄OH (3
 113 ml) and ethanol (50 ml) were mixed to form a homogeneous solution with magnetic stirring for 30
 114 min at 50 °C, and TEOS (3 ml) was then added dropwise into the above solution while stirring,
 115 which was continued for 2 h to form a transparent silica sol. After that, 1% HDTMS was added
 116 into the sol system in order to modify the hydrophilic silica particles. The reaction was allowed to
 117 continue for another 2 h under magnetic stirring at 50 °C to form a hydrophobic silica sol. The

118 silica particles were then collected by centrifugation (10,000 rpm, 12 min), and re-dispersed in
119 ethanol. After three centrifugation and re-dispersion cycles, the particles were finally
120 vacuum-dried for 10 h.

121 **Preparation of coating solutions**

122 PDMS/THF solution was firstly prepared as follows: PDMS (0.55g) was mixed with THF (30 ml)
123 with magnetic stirring for 30 min at room temperature to form the solution A, and the
124 corresponding curing agent (0.055 g) was dissolved to THF (30 ml) to form the solution B. The
125 solutions A and B were mixed together to form a PDMS/THF solution (1%, w/w). Different
126 amounts of the HDTMS-modified silica particles (with SiO₂/PDMS mass ratio of 1:1, 2:1, 3:1 and
127 4:1, respectively) were then dispersed in the as-prepared PDMS/THF solution with the aid of
128 ultrasonication for 1 h to form coating solutions.

129 **Preparation of superhydrophobic coatings on wood substrates**

130 Prior to the coating treatment, the wood samples were ultrasonically washed with ethanol for 10
131 min and dried at 60 °C for 6 h. The cleaned samples were then dipped into the as-prepared coating
132 solutions for ~10 min, and dried at 103 °C for 1 h. This procedure was repeated for 3 times to
133 allow full deposition of PDMS/SiO₂ composites on wood surfaces and to achieve adequate
134 thickness of the hybrid coatings. The coated wood samples were finally ultrasonically washed in
135 ethanol to remove the free or loosely attached nanoparticles, and dried at 103 °C for 5 h.

136 **Microstructure and morphology of the hybrid coatings**

137 The HDTMS-modified silica particles were observed by transmission electron microscopy (TEM,
138 Tecnai G2 F30, FEI, USA). Fourier transform infrared spectroscopy (FTIR, Nicolet Magna-IR 750,
139 USA) was used to study qualitatively the methyl (-CH₃) and methylene (-CH₂) groups grafted on
140 the modified silica particles. The surface structure and morphology of the hybrid coatings on wood
141 substrates were examined using field-emission scanning electron microscopy (FE-SEM, Zeiss
142 SUPRA 55, Germany). A thin Aurum (Au) layer was sputtered onto the sample surfaces to
143 improve conductivity prior to observation. The surface roughness of the coatings was analyzed by
144 atomic force microscopy (AFM, Dimension Icon, Bruker, Germany) with tapping mode. The
145 scanning scale is 3 μm × 3 μm.

146 **Hydrophobicity measurement**

147 The hydrophobicity of the coated wood samples was evaluated by water static contact angle (CA)

148 and dynamic sliding angle (SA), which were measured by a contact angle meter (Shanghai
149 Zhongchen JC2000D, China) at ambient temperature. The static CA was recorded 60 s after a
150 water droplet (5 μ l) was placed on the sample surface. The SA was determined by the minimum
151 tilt angle at which a water droplet (15 μ l) rolls off the surface. The average CA and SA values were
152 obtained by measuring more than five positions for each sample.

153 **Mechanical durability of the superhydrophobic coatings**

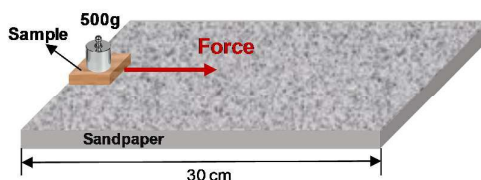
154 In order to examine the abrasion resistance of the superhydrophobic wood coatings, scratch tests
155 were performed according to the reported method.²⁴ As schematically illustrated in Fig. 2, With a
156 pressure of 12.5 kPa applied, the superhydrophobic surface was scratched against sandpaper (1500
157 mesh) with the abrasion length of 30 cm and a speed of 3 cm/s. The scratch tests were conducted
158 for 10 cycles, and CA and SA were measured after each abrasion cycle. Changes in surface
159 morphology and structure of the superhydrophobic coatings during the abrasion procedure were
160 examined by FE-SEM.

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166 **Fig. 2** Sketch of the setup used to determine the mechanical stability of the superhydrophobic
167 surface against abrasion.

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169 The washing durability of the superhydrophobic coatings was examined by submerging the
170 coated wood samples in a volume of distilled water ten times that of the samples for 10 h under
171 ultrasonication with an ultrasonic frequency of 80 kHz and ultrasonic power of 100 W. The
172 samples were collected at certain intervals and dried at 103 °C for 5 h, followed by CA and SA
173 measurements.

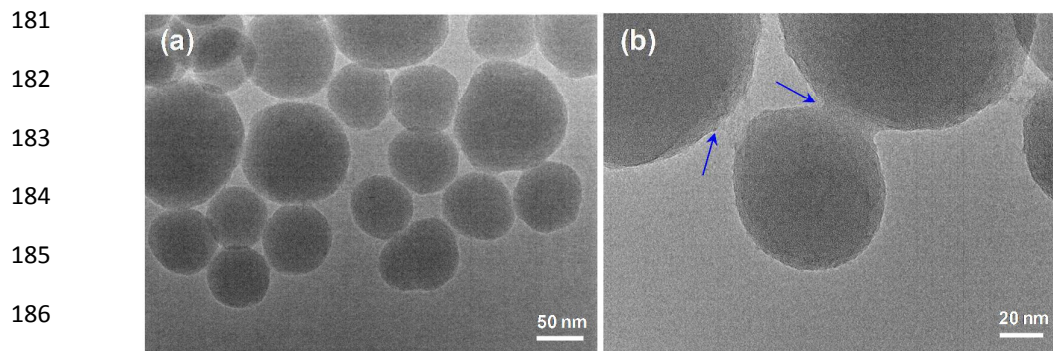
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174 **Results and discussion**

175 **Microstructure and surface morphology**

176 The TEM images of the synthesized silica nanoparticles are shown in Fig. 3. Well-dispersed

177 spherical silica particles of 60~120 nm in diameter were synthesized by the Stöber method. When
178 modified with HDTMS (1%), the silica particles were found to be covered by a thin layer of
179 polymer-like materials (see arrows in Fig. 3b). The neighboring particles were also observed to be
180 connected by the polymer bridge showing an obvious tendency to aggregate with each other.



188 **Fig. 3** TEM images of the (a) synthesized and (b) HDTMS-coated silica nanoparticles.

189 FT-IR was used to gain insights into the chemical structure of the formed polymer covering
190 silica particles (Fig. 4). In the FT-IR spectra of bare silica particles, two absorption peaks were
191 observed at 1055 and 795 cm^{-1} , which are attributed to the stretching vibration of Si-O-Si. The
192 absorption peak at 951 cm^{-1} is ascribed to the stretching vibration of Si-OH. After HDTMS coating,
193 two additional peaks at 2925 and 2854 cm^{-1} appeared, which are assigned to the asymmetric and
194 symmetric stretching of the CH_2 group, respectively.²⁵ This indicates that the long-chain alkyl
195 groups ($-\text{C}_{16}\text{H}_{33}$) of HDTMS were successfully grafted on the silica particle surface, confirming
196 the results of TEM observations.

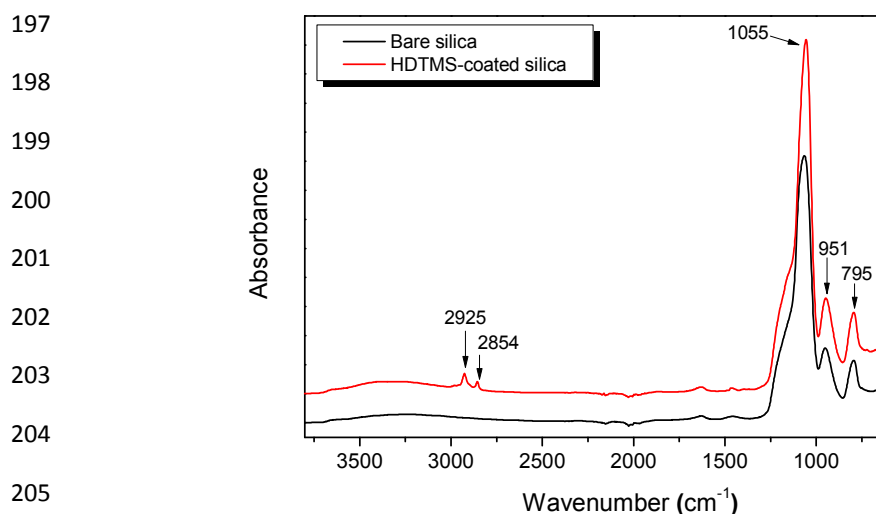
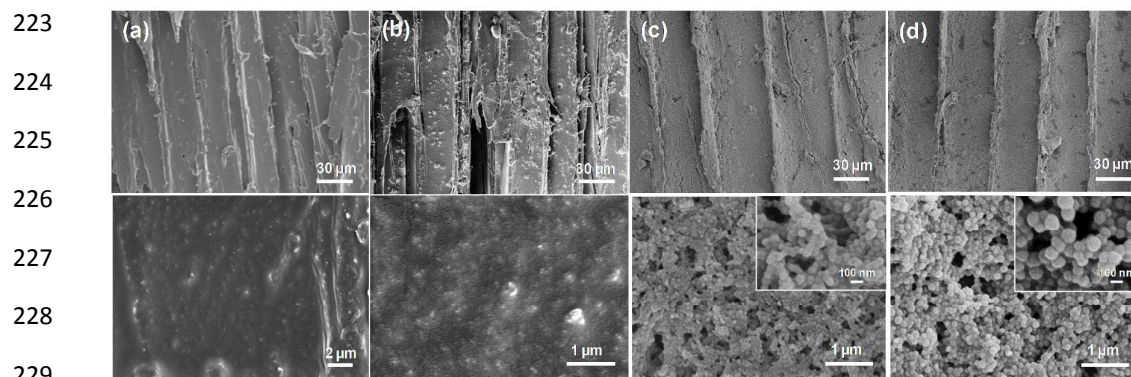


Fig. 4 FTIR spectra of bare and HDTMS-coated silica nanoparticles.

207 The SEM images of the wood surfaces coated with PDMS-silica nanocomposites with various
208 mass ratios of silica to PDMS are shown in Fig. 5. When coated with PDMS only, the wood lumen
209 surface was observed to be covered by a relatively smooth and continuous film at high
210 magnification (Fig. 5a). The wood surface presents a highly-textured cellular structure, forming
211 the primary roughness at the micro-scale. When a small amount of silica particles was
212 incorporated into the coating with a silica/PDMS mass ratio of 1:1, the morphology and roughness
213 of the lumen surface were only slightly changed since most of the silica particles appeared to be
214 embedded in the dense PDMS film (Fig. 5b). In this case, PDMS dictates the structure and
215 appearance of the hybrid coatings. With an increase of silica/PDMS mass ratio to 2:1, a relatively
216 homogeneous layer of PDMS/silica hybrid coating was observed to be coated on the lumen surface,
217 resulting in a highly roughened wood surface (Fig. 5c). The high-magnification image shows that
218 the hybrid coating presents a well-developed hierarchical structure consisting of interconnected
219 microdomains (aggregates of silica particles) and nanopores, in which PDMS acts as a binding
220 agent for connecting and anchoring the silica particles. A further increase of silica/PDMS mass
221 ratio to 3:1 resulted in a similar hybrid coating with a well-developed microstructure and a highly
222 roughened lumen surface (Fig. 5d).

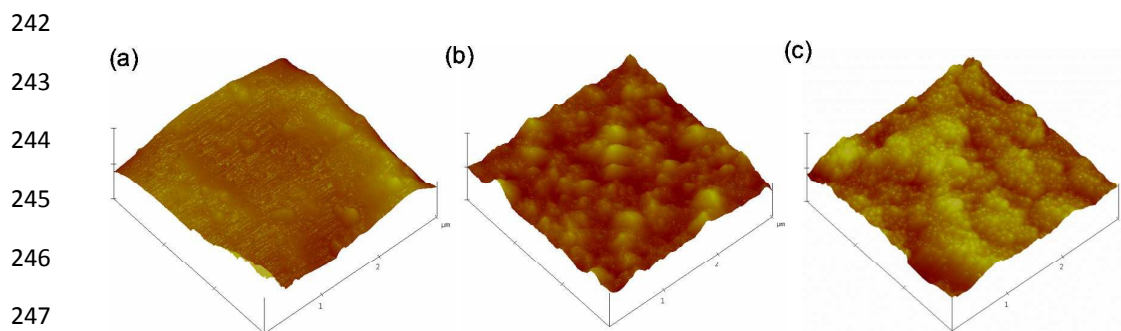


230 **Fig. 5** SEM images of wood surfaces coated with (a) PDMS only and PDMS-silica
231 nanocomposites with SiO₂/PDMS mass ratio of (b) 1:1, (c) 2:1 and (d) 3:1.
232 High-magnification images display details of the coatings.

233

234 The surface morphology and roughness of the coated wood were further examined by atomic
235 force microscopy (AFM). The three-dimensional AFM images of the wood lumen surfaces coated
236 with PDMS-silica nanocomposites are shown in Fig. 6. The PDMS-coated wood surface was

237 rather smooth with R_{rms} (root-mean-square roughness) of 11.8 nm. With a small amount of silica
238 particles incorporated, the hybrid coating surface was slightly roughened with R_{rms} of 23.5 nm (Fig.
239 6b). By contrast, at a relatively high silica particle loading, the well-developed hierarchical
240 structure with microdomains and nanopores can also be observed in the AFM image with
241 relatively high R_{rms} of 58.1 (Fig. 6c).



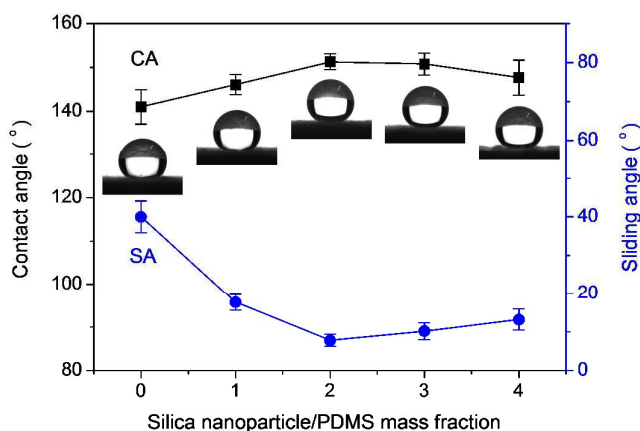
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249 **Fig. 6** Three-dimensional AFM images of the wood lumen surfaces coated with (a) PDMS
250 only and PDMS-silica nanocomposites with SiO_2/PDMS mass ratio of (b) 1:1 and (c) 2:1.

251

252 **Hydrophobicity**

253 The water CAs and SAs were investigated to examine the hydrophobicity of the PDMS/silica
254 hybrid coatings on wood surfaces. As shown in Fig. 7, the PDMS-coated wood already exhibited
255 high water-repellency with a CA of $\sim 140^\circ$, which is of course mainly attributed to the hydrophobic
256 nature of PDMS. However, it has been shown that the maximum CA that can be achieved on a
257 smooth surface is about 120° even when a material with the lowest surface free energy is used.²⁶
258 This indicates that the inherent microscale roughness of wood substrate also contributes to the high
259 hydrophobicity, but applying the hydrophobic PDMS alone is not sufficient to generate
260 superhydrophobicity on wood surfaces. When silica particles were incorporated into the PDMS
261 film, the water-repellent properties of the coated wood were remarkably improved. With
262 increasing silica/PDMS mass fraction from 0 to 4, the CA increased initially from 140° to 152° ,
263 then dropped slightly and remained almost constant around 150° . At low silica particle loadings,
264 the water droplets were also found to adhere strongly to the wood surfaces with high SAs. With
265 increasing silica particle to PDMS ratio, the SAs of the coated wood decreased considerably down
266 to around 10° and the water droplets can roll off the surfaces easily.



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276 **Fig. 7** Static contact angles (CAs) and sliding angles (SAs) of the coated wood as a function
277 of silica nanoparticle/PDMS mass fraction. The typical images of water droplets on the
278 surfaces are presented.

279 As shown in the optical photographs (Fig. 8a and b), spherical water droplets rested steadily on
280 the superhydrophobic wood surface, which is in contrast to the instant penetration of water into the
281 pristine wood. The prepared wood surface was not only superhydrophobic but also showed
282 self-cleaning properties, which was demonstrated by dropping water to the carbon-powder
283 contaminated wood surface (Fig. 8c-f). When dropped on the slightly tilted surface, the water
284 droplets readily rolled off the superhydrophobic wood surface, carrying away the surface
285 contaminants, whereas water droplets merged with the carbon powder and stuck on the pristine
286 wood surface (see Video S1 and S2 in ESI[†]). However, it can be observed from Video S1 that
287 some water droplets adhered to the superhydrophobic wood surface, which is mainly attributed to
288 the structural heterogeneity of wood surfaces consisting of bright region of earlywood alternating
289 with darker regions of latewood (see wood samples in Fig. 8a and b). The water droplets generally
290 slid smoothly on the earlywood region while readily adhered to the narrow latewood region. This
291 highlights the fact that it is more challenging to achieve superhydrophobicity on the heterogeneous
292 wood surface as compared with the homogeneous substrates such as glass.

293 The wettability of a surface is governed by the surface composition as well as the microstructure.
294 When the chemical composition is kept the same, the surface structure is the key factor affecting
295 the surface wettability.²⁷ In the present study, the main function of silica particles in the hybrid
296 coating system is to generate nanoscale roughness. Hence, the silica particle to PDMS mass ratio

297 is very critical to control the microstructure and roughness of the hybrid coatings, resulting in
298 tunable water-repellent properties of the coated wood. The influence of roughness on the
299 wettability of a surface has been primarily discussed in terms of the Wenzel and Cassie-Baxter
300 models.^{28,29} When the silica particle loading was low, the resulting hybrid coatings were not
301 notably roughened, which appeared to impose little impact on the inherent microstructure of the
302 wood surfaces. In this case, water droplets can fill and wet the open lumen space of the
303 longitudinally arranged cells of the wood substrate, resulting in a large contact angle hysteresis,
304 which could be described by the Wenzel model. However, in view of the relatively high CA values,
305 it is reasonable to assume that water droplets may also be in the Cassie state, in which the
306 solid-liquid contact area fraction is high and thus water droplet pinning can occur as indicated by
307 the relatively large SAs.

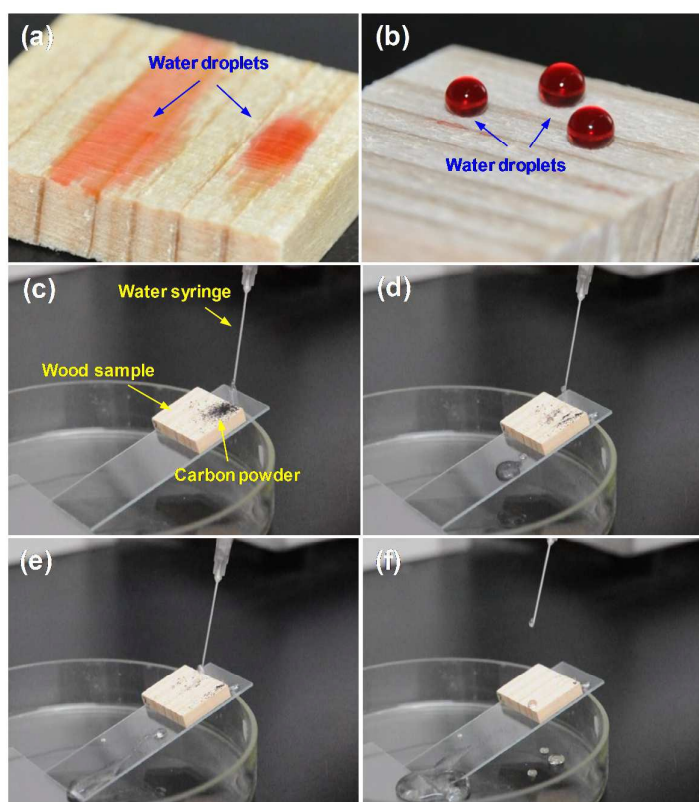


Fig. 8 Optical photographs of water droplets (mixed with a red dye) on the (a) pristine and (b) superhydrophobic wood surfaces; (c-f) snapshots of the self-cleaning process on the superhydrophobic wood surfaces.

327 By contrast, when the silica particle loading was raised to a certain level (*e.g.* silica/PDMS mass
328 ratio is 2:1), the resulting hybrid coating appeared to develop a hierarchical microstructure with
329 interconnected microdomains and nanopores, producing highly roughened wood lumen surfaces.
330 Such a hierarchical structure with nanoscale roughness superimposed on the microscale roughness
331 is known to be an essential feature in generating superhydrophobic properties and more important
332 for achieving low water sliding angles.³⁰⁻³³ Accordingly, water droplets could not wet the surface
333 but are suspended over the hierarchical structure with large fraction of air being entrapped inside,
334 which is a typical wetting behavior described by the Cassie model. Consequently, a higher CA
335 (greater than 150°) was observed, and the adhesion force between the water droplets and the wood
336 substrate was very small. In short, the surface roughness, which is tuned by changing the
337 silica/PDMS ratio, plays a vital role in governing the surface wettability of the coated wood. The
338 surface roughness has to exceed a critical level to transform into the “desired” Cassie state with a
339 small contact angle hysteresis.

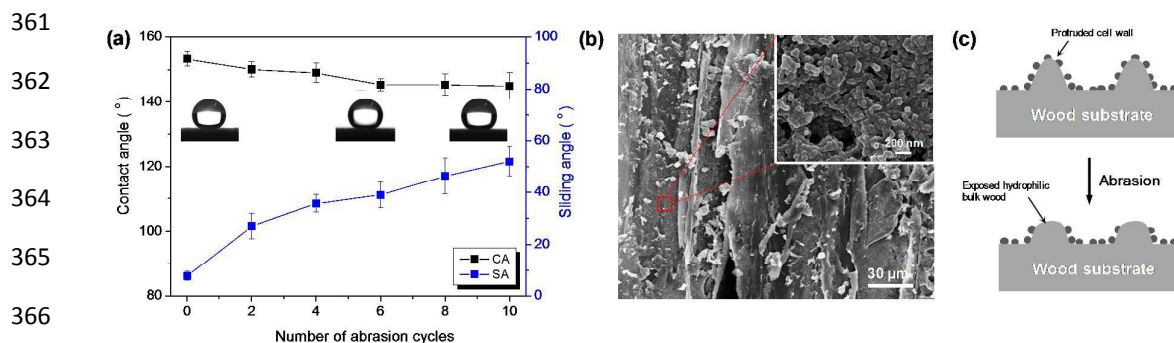
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341 **Mechanical durability of the superhydrophobic coatings**

342 The major issue for the practical application of superhydrophobic wood is that the elaborately
343 fabricated microstructures of the rough surfaces could be easily damaged by mechanical abrasion.
344 Scratch tests were carried out to examine the abrasion resistance of the superhydrophobic coatings
345 (SiO₂/PDMS mass ratio is 2:1) on wood surfaces. Fig. 9a shows the change in CAs and SAs as a
346 function of abrasion cycles for the structured wood surface. It can be seen that the CAs of the
347 coated wood remained almost constant around 150° after being scratched repeatedly, and spherical
348 water droplets can still form on the surface. However, the mechanical abrasion caused an obvious
349 increase in the SAs, and water droplets were not easy to roll, reflecting an increased contact angle
350 hysteresis. SEM images show that the microscale structures on the wood surface were severely
351 damaged by the mechanical abrasion with such a high loading pressure (12.5 kPa), whereas the
352 nanoscale features of the PDMS/silica hybrid coating in the cell lumens was well retained in
353 general (Fig. 9b). As schematically illustrated in Fig. 9c, once the inherent microscale bumps
354 (protruded cell walls) of the wood surface has been partly worn out, the underlying hydrophilic
355 bulk wood would be exposed as a result, introducing water pinning sites that consequently make
356 the surface more sticky towards water. Nevertheless, despite an increase in the contact angle

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357 hysteresis induced by the mechanical abrasion, the wood surface remained non-wettable with the
 358 nanoscale features being kept intact in general. It should be noted that the applied loading pressure
 359 (12.5 kPa) in this study is too harsh for the as-prepared superhydrophobic wood surface since the
 360 inherent microscale features of the surface were severely damaged.



367 **Fig. 9** (a) CAs and SAs as a function of number of abrasion cycles for the superhydrophobic
 368 wood surfaces; (b) SEM images of the wood surface after 10 abrasion cycles; (c) Sketch
 369 illustrating abrasion-induced damage to the microscale features (protruded cell walls) on the
 370 wood surface, exposing the hydrophilic bulk wood.

371

372 We also qualitatively assessed the mechanical stability of the superhydrophobic wood surfaces
 373 against finger touching. As shown in Fig. 10, the wood surface remained superhydrophobic upon
 374 being touched by a finger. Water droplets exhibited a spherical shape on the touched surface with a
 375 CA of $\sim 150^\circ$ and can easily roll off at a small tilting angle. This indicates that the surface textures
 376 were robust enough to withstand the force exerted by finger touching and also exhibited resistance
 377 against grease contamination caused by finger contact, making the surface a finger touchable
 378 superhydrophobic surface. In the hybrid coating system, PDMS was expected to function as a
 379 binding agent to aggregate the silica particles and anchor the particles tightly on the wood
 380 substrate, thus endowing the coatings with good mechanical stability.

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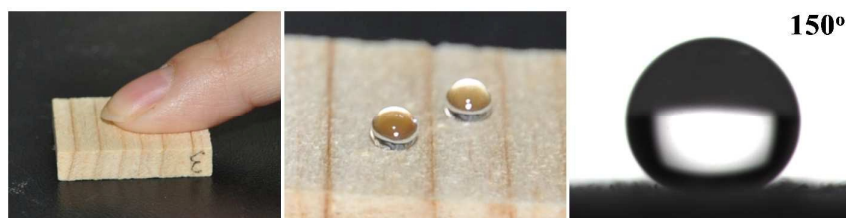
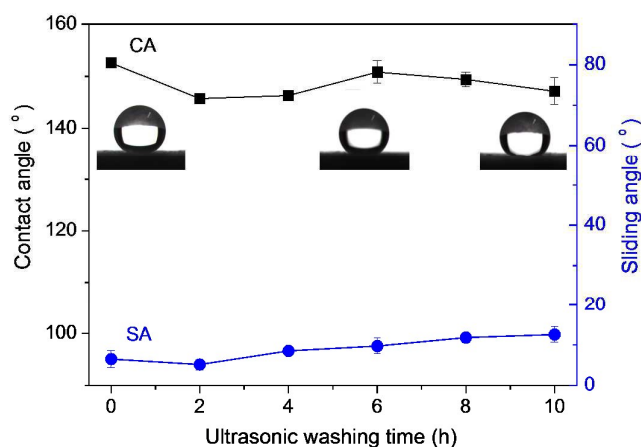


Fig. 10 The wood surface remained superhydrophobic upon being touched by a finger,
 and water droplets exhibited a spherical shape on the touched surface with a CA of 150° . 13

387 The durability of the superhydrophobic coatings was further examined by ultrasonic washing in
388 water with a high ultrasonic frequency of 80 kHz. Fig. 11 shows the change in CAs and SAs as a
389 function of ultrasonic washing time for the coated wood. It can be seen that the CAs of the coated
390 remained almost constant around 150°, and SAs exhibited only a slight increase up to 12° after
391 ultrasonic washing for a duration of 10 h. This indicates that the PDMS-anchored nanoparticles
392 and the well-developed microstructure were resistant against the cavitation damage induced by the
393 high frequency (80 kHz) ultrasound, and the hybrid coatings were therefore durable enough to
394 withstand the ultrasonic impact without impairing its water-repellency.



403
404 **Fig. 11** CAs and SAs as a function of time of ultrasonic washing in water for the
405 superhydrophobic wood surfaces.

406 Conclusions

407 In summary, a simple and inexpensive dip-coating method has been successfully applied to
408 fabricate durable superhydrophobic organic-inorganic hybrid coatings on the intrinsically
409 heterogeneous wood surfaces by using PDMS-silica nanocomposites. The surface morphology and
410 microstructure of the hybrid coatings can be readily controlled by adjusting the silica particle
411 loadings in the PDMS matrix, and the surface roughness and wettability of the coated wood was
412 accordingly tuned. When silica/PDMS mass ratio was raised to 2:1, hierarchical microstructures
413 with interconnected microdomains and nanopores were formed in the coating, allowing the
414 generation of superhydrophobic wood surfaces. The fabricated PDMS/silica hybrid coating also
415 showed desirable mechanical stability and durability against mechanical abrasion and

416 high-frequency ultrasonic washing in water, highlighting their potential in various practical
417 applications.

418

419 **Electronic Supplementary Information (ESI)**

420 Video S1 showing water droplets rolling off the superhydrophobic wood surface, carrying away
421 the surface contaminants, in contrast to Video S2 showing water droplets merged with the
422 contaminants and stuck on the pristine wood surface.

423

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