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Self n-Doped [6,6]-Phenyl-C61-butyric Acid 2-((2-(trimethylammonium)ethyl)-(dimethyl)ammonium) ethyl Ester Diiodides as a Cathode Interlayer for Inverted Polymer Solar Cells

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Abstract

A series of self n-doped fullerene ammonium derivatives have been synthesized and confirmed with
 10 electron paramagnetic resonance and conductivity measurements. The existence of stable $C_{60}O_2^-$ anion
 radical in these materials resulted in intrinsically high conductivities between 1.05×10^{-2} and 1.98×10^{-2}
 S/cm. Among the fullerenes with different number of ammonium and various counter anion, [6,6]-
 phenyl-C61-butyric acid 2-((2-(trimethylammonium)ethyl)-(dimethyl)ammonium)-ethyl ester diiodides
 15 (PCBDANI) showed the best solvent resistance which was confirmed by the measurement of film
 thickness and corresponding UV-vis absorption before and after rinsed with dichlorobenzene. Most
 importantly, the inverted polymer solar cells with the structure of
 ITO/PCBDANI/P3HT:PCBM/MoO₃/Ag remained reasonable high power conversion efficiency even at a
 thickness of 82 nm of the PCBDANI film as the cathode interlayer. This thickness tolerant cathode
 interlayer would enable to achieve good device performance with a wider range of film thicknesses and
 20 thus large-area device printing this interlayer or on this interlayer could become feasible.

Introduction

Polymer solar cells (PSCs) have attracted broad interests both in
 academic and industry as renewable energy sources because of
 their unique advantages of light weight, flexible and low cost
 25 fabrication.¹⁻⁷ So far, tandem solar cells reached a power
 conversion efficiency (PCE) of 10.6 %⁸ while single junction
 cells showed the highest PCE of 9.35 %⁹ based on an inverted
 polymer solar cell (I-PSC) with a cathode interfacial layer
 processed with an alcohol. Cathode interfacial materials are one
 30 of the key components for efficient polymer solar cells.⁸⁻⁵⁰ In
 recent years, water-soluble or alcohol-soluble polar polymers
 have been successfully employed as the cathode interlayers in
 conventional and inverted PSCs including poly(ethylene
 oxide),¹⁴⁻¹⁷ conjugated polyfluorene derivatives with the side
 35 chains containing amino- end groups¹⁸⁻²⁸ or phosphoryl end
 groups²⁹⁻³¹ or metal ion-intercalated crown ether,³² amino-
 containing insulating polyethylenimine derivatives,³³⁻³⁵ a
 hyperbranched conjugated polymer,³⁶ and conjugated polymeric
 zwitterions.³⁷ Moreover, fullerene derivatives and cross-linked
 40 fullerenes have been intensively studied due to their electron
 transport properties.³⁸⁻⁵² The interfacial dipole formation between
 an amine and a metal electrode, and n-doping of conducting

fullerenes are the main reasons for the improvement of device
 performance.

45 Usually, semiconducting cathode interlayers were thin in
 thickness around 5~10 nm in order to achieve optimized device
 performance because thick interlayer will result in a high series
 resistance, while a very thin layer could not provide an ohmic
 contact for electron extraction and transport. However, to print
 50 this thin layer or to print on this thin layer is challenging for large
 area device fabrication.^{21,32,53,54} Thus thickness tolerant
 interlayers which possess excellent electron transport property
 and solvent resistance would have significant advantages.

Recently, Huang *et al.* reported a mercury-containing alcohol
 55 soluble conjugated metallopolymer PFEN-Hg as a cathode
 interlayer in PSCs with enhanced electron transport property by
 improving the main chain interaction.⁵⁵ Thus PFEN-Hg based
 device performance exhibited less stringent dependency on the
 thickness of the interlayer.⁵⁵ However, PFEN-Hg is not an
 60 environmentally friendly material and will limit its use for large-
 area device fabrication. More recently, Li *et al.*⁵⁶ have
 demonstrated that perylene diimide derivatives could be used as a
 thickness-insensitive cathode interlayer for high performance

conventional PSCs in a wide thickness range of 6 to 25 nm.

Electrically doped organic transport layers with high conductivity can facilitate charge carrier transport both in organic light-emitting diodes and organic solar cells.⁵⁷ However, an efficient n-

type molecular doping is intrinsically more difficult than p-type doping since the HOMO level of the dopant must be higher than the LUMO level of the matrix material, which makes such materials vulnerable to oxidation in ambient. In addition, it is difficult to find suitable dopants with increasing HOMO level.

The current n-doping approaches such as thermal evaporation of small molecules and *in situ* creation of volatile dopants from precursors need inert environment and have been hindered by aggregation of small dopants in solid-state.⁵⁷⁻⁵⁸ Therefore, the development of solution processible and stable n-doped cathode interfacial materials for PSCs device fabrication is of great challenge.

More recently, Jen *et al.* demonstrated a self n-doped fullerene with an ammonium salt (FPI)⁵⁹ and further effective doping of fullerenes *via* simple solution-processed blend of [6,6]-phenyl-C61-butyric acid methyl ester (PCBM) and tetrabutylammonium salts⁶⁰ to achieve high conductivity in thin films. Loneragan⁶¹ also confirmed the n-doping but proposed a mechanism involving chemical reaction before the electron transfer induced n-doping.

However, the feasibility of using FPI as a cathode interlayer for I-PSCs is limited because FPI possesses poor orthogonal solvent processability as it can be dissolved in common organic solvents such as chlorobenzene in which the top layer was formed from its solution.⁶² Most interlayers such as poly[(9,9-bis(3'-(N, N-dimethylamino)propyl)-2,7-fluorene)-alt-2,7-(9,9-dioctylfluorene)] (PFN) or fullerene derivatives can be dissolved in common solvents and cross-linking has been used for improving the solvent resistance in organic light emitting diodes^{53,63,64} and PSCs.^{14,28,42,43,55} Similarly, FPI has to be embedded into a thermally cross-linked semiconducting fullerene matrix.⁶² Although doped and solvent resistant interlayer was obtained, it became less conductive by this blending and thus the performance of the solar cells is sensitive to its thickness.⁶²

Previously, we reported an alcohol soluble fullerene derivative, [6,6]-phenyl-C61-butyric acid 2-((2-(trimethylammonium)ethyl)(dimethyl)ammonium)-ethyl ester diiodides (PCBDANI) as a cathode interlayer successfully in conventional PSCs.⁴⁵ In this paper, we report that a few intrinsically conductive fullerene derivatives with different number of ammonium salts and various counter ions are self n-doped, which is confirmed with electron paramagnetic resonance (ESR) and conductivity measurements. PCBDANI is shown to have the best solvent resistance and thickness tolerance as the cathode interfacial layer in a proof-of-concept I-PSCs with device architecture ITO/interlayer/P3HT:PCBM/MoO₃/Ag where P3HT represents poly(3-hexylthiophene), a typical donor for PSCs. Notably, the devices could remain high efficiency even at a thickness of 82 nm of the PCBDANI film.

Experimental

Materials: All solvents were purchased from Sigma Aldrich, P3HT from Merck Chemicals Ltd and PCBM from Nano-C. Inc.

Synthesis and characterization of PCBANI and PCBDANBr

Similar to PCBDANI,⁴⁵ [6,6]-phenyl-C61-butyric acid trimethylammonium ethyl ester iodide (PCBANI) and [6,6]-phenyl-C61-butyric acid 2-((2-(dimethylethanammonium)ethyl)(methyl)amino)ethyl ester bromide (PCBDANBr) in Scheme 1 were synthesized using PCBM as the starting material. [6,6]-Phenyl-C61-butyric acid (PCBA) and [6,6]-phenyl-C61-butyric acid chloride (PCBA-Cl) were prepared as before.⁴⁵ Esterification of PCBA-Cl with 3-(dimethylamino)propan-1-ol and 2-((2-(dimethylamino)ethyl)(methyl)-amino)ethanol afforded [6,6]-phenyl-C61-butyric acid dimethylammonium ethyl ester (PCBAN)⁴² and [6,6]-phenyl-C61-butyric acid 2-((2-(dimethylamino)ethyl)(methyl)amino)ethyl ester (PCBDAN)⁴⁵ separately, which were allowed to react with iodomethane and bromoethane to obtain PCBANI and PCBDANBr accordingly following the method for preparation of PCBDANI.⁴⁵ The structure of PCBANI and PCBDANBr were characterized by using Av400X NMR (Bruker Co.) and Thermo Scientific Q-Exactive FTMS. Both PCBANI and PCBDANI are soluble in methanol and readily soluble in DMF and DMSO. For PCBDANBr, it can be dissolved in DMSO, and mixed solvents of methanol with other solvents such as chloroform or chlorobenzene.

PCBANI. A solution of PCBAN (0.10 g, 0.10 mmol) and methyl iodide (0.5 mL) in chloroform (10 mL) was stirred at room temperature for 24 h. PCBANI precipitated and the suspension was transferred into a centrifuge bottle, centrifuged, decanted and then re-suspended in methanol, centrifuged, decanted, and treated twice with methanol in the same manner. The residue was dried *in vacuo* at room temperature for 48 h to give a brown solid, PCBANI (0.10 g, yield 90 %). ¹H NMR (400 MHz, d₆-DMSO, δ): 1.95-2.07 (m, 2H), 2.54-2.58 (m, 2H), 2.83-2.91 (m, 2H), 3.09-3.21 (m, 15H), 3.67-3.74 (m, 2H), 3.86-3.97 (m, 2H), 4.41-4.48 (m, 2H), 7.43-7.47 (m, 1H), 7.51-7.55 (m, 2H), 7.98 (d, 2H, J = 8.0 Hz). ¹³C NMR (100 MHz, d₆-DMSO, δ): 22.4, 22.8, 33.8, 52.74, 52.95, 61.5, 63.2, 80.6, 129.0, 132.6, 136.8, 137.3, 138.3, 140.7, 140.9, 142.3, 142.4, 142.5, 143.1, 143.9, 144.0, 144.4, 144.6, 144.72, 144.9, 145.2, 145.3, 145.8, 146.4, 148.4, 149.6, 172.9. MALDI-TOF-MS: calcd for [C₇₇H₂₆NO₂]⁺·I⁻, 996.2, found [M-I]⁺, 996.0.

PCBDANBr. A solution of PCBDAN (0.20 g, 0.18 mmol) and ethyl bromide (0.5 mL) in chloroform (10 mL) was stirred at room temperature for 48 h. The product was purified by neutral aluminium oxide column chromatography successively using chloroform and methanol as the eluents to give a brown solid, PCBDANBr (0.16 g, yield 78 %). ¹H NMR (400 MHz, d₆-DMSO, δ): 1.19 (d, 3H, J = 8.0 Hz), 1.97-2.01 (m, 2H), 2.20 (s, 3H), 2.46-2.47 (m, 6H), 2.58-2.60 (m, 2H), 2.70-2.72 (m, 2H), 2.85-2.89 (m, 2H), 2.87-2.98 (m, 6H), 4.06-4.09 (m, 2H), 7.46-7.48 (m, 1H), 7.53-7.57 (m, 2H), 8.01 (d, 2H, J = 8.0 Hz). ¹³C NMR (100 MHz, d₆-DMSO, δ): 8.4, 22.4, 33.4, 33.6, 42.4, 50.3, 51.2, 52.7, 55.7, 59.6, 62.1, 80.5, 128.8, 128.9, 132.5, 136.7, 137.1, 140.5, 140.6, 140.7, 142.0, 142.1, 142.3, 142.87, 142.91, 143.71, 143.74, 144.2, 144.6, 145.1, 145.6, 146.3, 148.4, 149.7, 172.8. MALDI-TOF-MS: calcd. for [C₈₀H₃₃N₂O₂]⁺·Br⁻, 1053.25, found [M-Br]⁺, 1053.25.

ESR measurements

Sample solutions were added into ESR test tubes and then bubbled with nitrogen to form films on the tube walls at room temperature or at 100 °C under nitrogen atmosphere. ESR

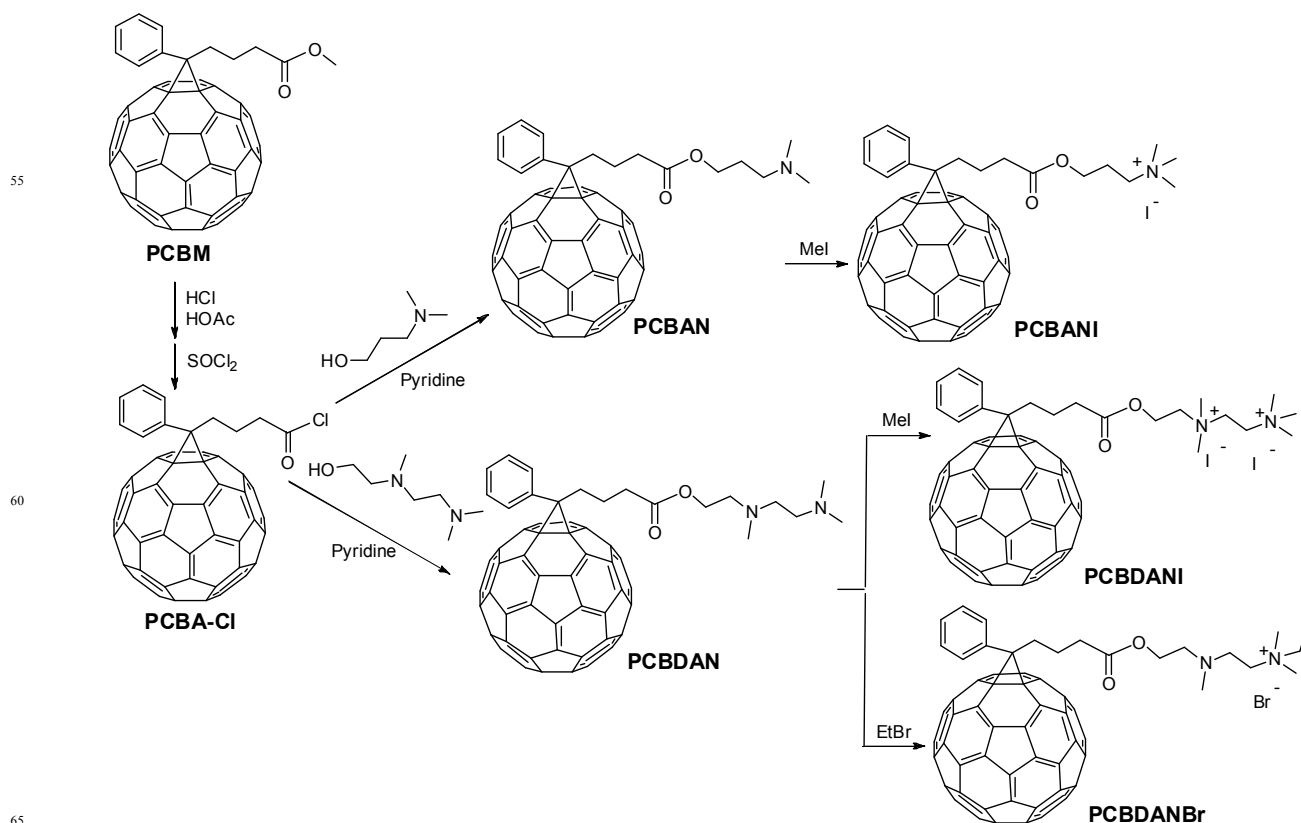
measurements were performed using an ESR-300 electron paramagnetic resonance spectrometer (Bruker Co.) at room temperature under nitrogen.

Device fabrication and characterization

SCLC device. Electron-only device structures were used: ITO/fullerene (80 nm)/Al (100 nm). The ITO-coated glass substrates were firstly cleaned *via* a four-step solvent cleaning procedure under sonicating which started from detergent, then deionized water, acetone and isopropanol, each for 10 min. Then the substrates were dried before undertaking a 10 min ultraviolet-ozone treatment. After that the substrates were transferred into a glove box with nitrogen atmosphere. PCBANI and PCBDANI were dissolved in DMF while PCBDANBr was dissolved in a mixture of methanol and chlorobenzene. Then the solutions were spin-coated (1000 rpm, 60 s) onto the pre-cleaned substrates. A silver layer with a thickness of 100 nm was deposited on the fullerene layers.

I-PSCs. ITO substrates were cleaned with a neutral detergent solution, deionized water, acetone and isopropyl alcohol sequentially. Then they were treated with UV/ozone at 30 °C for 10 min. After that the devices were transferred to a glove-box where the fullerene ammonium layers were spin-coated at 700 rpm for 90 s onto the ITO. P3HT and PCBM were dissolved in dichlorobenzene together in 90 °C for 1 h. Concentration of both P3HT and PCBM was 15 mg/mL. The solution was filtered before being spin coated (1300 rpm, 60 s) on the interlayer coated

substrates. The films were thermally annealed at 150 °C for 10 min. Then a layer of 10 nm MoO₃ was thermally evaporated onto the active layer at a rate of 0.2 Å/s, followed by a layer of 100 nm silver at 3 Å/s at a pressure below 2×10⁻⁶ Torr. A mask defined the active area of the solar cell device as 10 mm². Current density-voltage (J-V) characteristics of the devices were measured using a Keithley 2410 Source Meter Unit. Solar cell performance was measured under an Air Mass 1.5 Global (AM 1.5 G) Oriel solar simulator fitted with a 1000 W Xe lamp, filtered to give an output of 1000 W/m² in a glove-box. The incident photon to current efficiency (EQE or IPCE) was measured under ambient atmosphere at room temperature using a Stanford Research Systems model SR830 DSP lock-in amplifier coupled with a WDG3 monochromator and 500W xenon lamp. The light intensity at each wavelength was calibrated with a standard single-crystal Silicon photovoltaic cell. The absorption was measured by Hitachi U-3010 UV-Vis spectrophotometer. A Dektak 6M stylus profiler (Veeco, Inc.) or an Ambios Technology XP-2 surface profilometer was used to measure the thickness of the films. The ultraviolet photoelectron spectroscopy (UPS) measurements were conducted in ultra-high vacuum (3.0×10⁻⁸ Torr) with a Kratos Axis Ultra DLD ultraviolet photoelectron spectrometer equipped with a monochromatic He I ultraviolet source He I (hν=21.2 eV). To separate the sample and the secondary edge for the analyzer, a sample bias of -9 V was applied.



Scheme 1. Synthetic route of PCBDANI, PCBANI and PCBDANBr.

Results and discussion

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ESR measurements

To confirm the self n-doping of these fullerene derivatives, electron paramagnetic resonance (ESR) were measured. ESR spectra of [6,6]-phenyl-C61-butyric acid trimethylaminoethyl ester iodide (PCBANI), PCBDANI and [6,6]-phenyl-C61-butyric acid 2-((2-(dimethylethanammonium)ethyl)(methyl) amino)ethyl ester bromide (PCBDANBr) films were recorded at room temperature under nitrogen. In Fig. 1a, the paramagnetic signals were observed with g values of 2.0027, 2.0027, and 2.0026 for PCBANI, PCBDANI, and PCBDANBr respectively. In comparison, PCBDAN showed no significant ESR signal, confirming its semiconducting nature. It was reported that g value for PCBM anion radical is 1.998.⁵⁸ The same g value for C₆₀ was reported by Hwang *et al.*⁶⁵ They also observed that upon exposure to molecular oxygen the broad C₆₀ band was slowly converted to band b with $g = 2.0008$ and then further transformed

to more stable band c with $g = 2.0026$. At high temperatures ($T > 355$ K) and in a polar solvent, the band c can be partially reversibly converted back to the band b. Both bands b (g value of 2.0008) and c (g value of 2.0026) were designated to two isomers of the C₆₀O₂ anion radical.⁶⁵ Similarly, both bands b and c for PCBDANBr film were observed when the sample was prepared at 100 °C under nitrogen (Fig. 1b). The existence of band c or band b suggested that fullerene radical anions do exist in these materials through self n-doping of the fullerenes.⁵⁸⁻⁶⁰ We note that FPI also has a similar g -factor of 2.0010,⁵⁹ which may indicate that FPI is also in the form of C₆₀O₂. We believe that insufficient deoxygenation leads to the formation of the C₆₀O₂ anion radical which was readily been observed. Fortunately, these materials are stable under ambient condition for processing and still work well in devices as will be disclosed later.

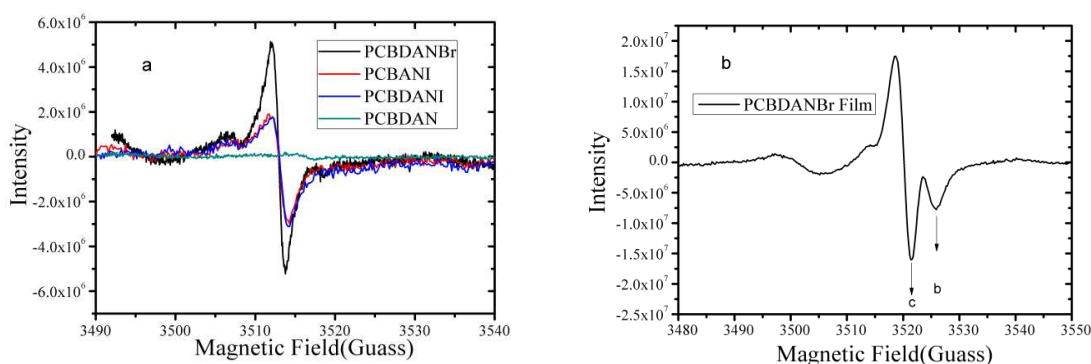


Fig. 1 ESR spectra of PCBANI, PCBDANI, PCBDANBr and PCBDAN films prepared at room temperature (a) and PCBDANBr film prepared at 100 °C under nitrogen (b).

Conductivity measurements

The conductivities of these fullerene ammoniums were tested using space-charge limited current (SCLC) method based on electron-only devices ITO/fullerene (80 nm)/Al (100 nm). We fitted the experimental J - V data in the logarithmical scales and a slope of 1 instead of 2 was obtained, which indicated there is no SCLC but Ohmic's law was followed (Fig. 2). Thus these fullerenes are not semiconductors but conducting materials,

which are in agreement with the ESR measurements. The conductivities for PCBANI, PCBDANI, and PCBDANBr, were 1.50×10^{-2} , 1.98×10^{-2} , and 1.05×10^{-2} S/cm respectively, which are comparable to those for other conducting materials such as FPI (2.0×10^{-2} S/cm),⁵⁹ PEDOT-PSS ($< \sim 10^{-3}$ S/cm for Baytron P VPAI 4083⁶⁶ and 0.4 S/cm for Baytron P V4071⁶⁷) and an anionic narrow band gap conjugated polyelectrolyte (1.5×10^{-3} S/cm).⁶⁸ The intrinsically high conductivity should be ascribed to the effective n-doping of fullerene films.

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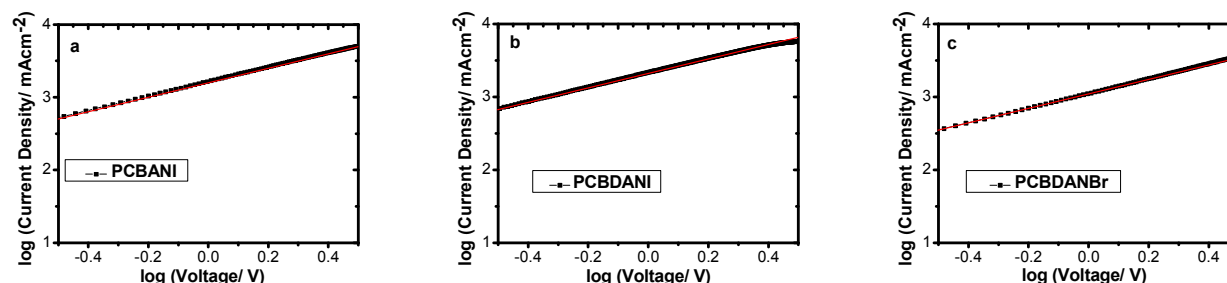


Fig. 2 The current density versus applied voltage (J-V) characteristics of electron-only devices ITO substrate/fullerene/Al. (black line measured and red line fitted from SCLC model): PCBANI (a); PCBDANI (b); PCBDANBr (c).

Solvent resistance test

Solvent resistance of these materials were studied by the measurement of film thickness and corresponding UV-vis absorption before and after rinsed with dichlorobenzene. Spin coating from 40 mg/mL solutions in DMF, PCBDANI and PCBANI films had thickness of 84 and 85.5 nm, respectively. After rinsed with dichlorobenzene, the thickness became 82 and 54.6 nm respectively. For PCBDANBr film, a thickness of 105 nm has been fabricated by multi-layer casting from its methanol

solution and it became 63 nm after rinsed with dichlorobenzene. The measured thicknesses variation was in agreement with the UV-vis absorption (Fig. 3). Thus PCBDANI has the best solvent resistance, and the actual thickness (82 nm) was extraordinarily high for the film prepared at 40 mg/mL. The tuning of the solvent resistance property has been achieved with different numbers of ammonium and various counter ions, and further optimization of solvent resistance could be possible by structure modification.

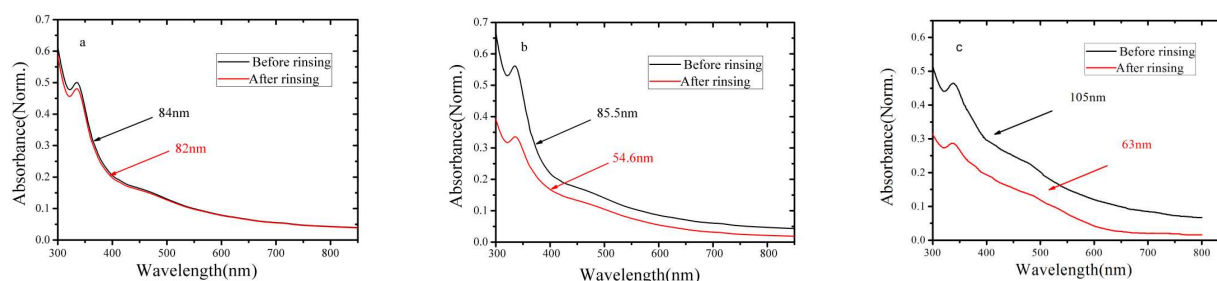


Fig. 3 UV-vis spectra of PCBDANI (a), PCBANI (b) and PCBDANBr (c) films before and after rinsed with dichlorobenzene.

Work function of ITO coated with interlayers

The banding energies of bare ITO and ITO coated with various interlayer films were measured by UPS (Fig. 4). Thus the work function of pristine ITO is decreased from 4.67 to 4.27, 4.23 and 4.19 eV for the ITO coated with PCBANI, PCBDANI and PCBDANBr respectively. These results indicate that all of these materials could be used as cathode interlayer in I-PSC.

Photovoltaic devices study

High conductivities of the self n-doped fullerenes suggested that they would be potentially thickness tolerant as cathode interlayers. To prove this concept, the effect of the thickness of the three fullerene interlayers on the I-PSC performance was examined. The device structure for evaluating the material was ITO/interlayer/P3HT:PCBM/MoO₃/Ag, where ITO was used as

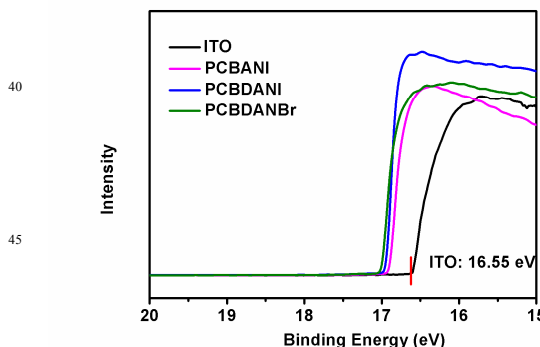


Fig. 4 Ultraviolet photoelectron spectroscopy (UPS) of bare ITO and ITO coated with various interlayer films.

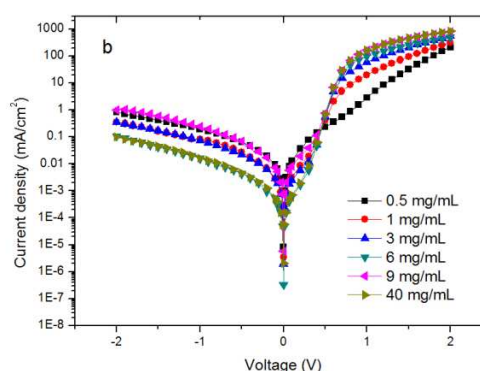
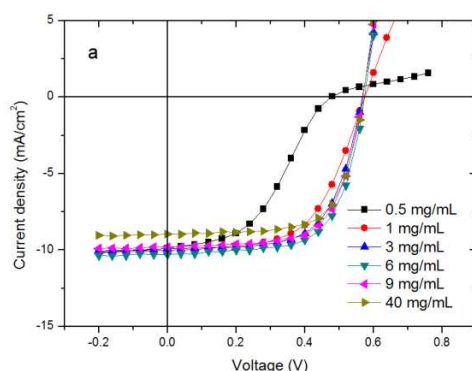
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the cathode, MoO₃/Ag as the anode, P3HT and PCBM used as the donor and acceptor respectively. A series of PCBDANI and PCBANI films with various thicknesses were spin-coated atop ITO substrates from their DMF solutions while PCBDANBr films were spin cast from the solution in mixture solvent of methanol and chlorobenzene. For comparison, a control device without any interlayers was also fabricated. The current density-voltage (*J-V*) curves for the I-PSCs measured under AM 1.5 G irradiation (1000 W m⁻²) and dark are presented in Fig. 5 and the corresponding device characteristics including *J*_{sc}, *V*_{oc}, *FF* and PCE values are summarized in Table 1. The relationship of PCE versus interlayer concentration was plotted in Fig. 6. I-PSC with a bare ITO electrode performed poor with a PCE of 0.32 %, a *J*_{sc} of 6.96 mA cm⁻², a *V*_{oc} of 0.14 V, and a *FF* of 31.7 %. With a thin interlayer, the devices worked reasonable well for all three interlayers. The optimized devices appeared for all the interlayers cast from low concentrations: *V*_{oc} reached 0.58 V, *J*_{sc} reached 10.1~10.2 mA/cm², *FF* reached 64.0~66.4 % and PCE reached 3.62~3.83 %. For the interlayers with iodide counter ions, the PCEs remained as high as over 90% of the maximum PCEs when the interlayers were spin cast from 40 mg/mL concentration, the highest concentration test in this work. These performances are comparable to that from inverted solar cells with other cathode interlayers.^{47,69-71} While, PCBDANBr gave worse performance at high concentrations, due to solubility issue after the material was stored for a period of time. Notably, for PCBDANI film prepared at 40 mg/mL, the thickness reaches 82 nm. At this extraordinarily high thickness, the device remains PCE as high as 90% of the maximum (3.43 % versus 3.82%) with even higher *FF* (66.4 %), along with the reduced *R*_s and increased *R*_{sh}. However, its *J*_{sc} (9.05 mA/cm²) was reduced by 10 % relative to the maximum

value. To understand why the *J*_{sc} is reduced, we measured the UV-vis spectra (Fig. 7) and external quantum efficiency (EQE, Fig. 8.) of the devices with PCBDANI interlayer spin cast from 1, 6 and 40 mg/mL concentration and the control device without PCBDANI interlayer. The average thicknesses of these interlayers are around 5, 23, and 82 nm respectively. Indeed, the interlayer contributed to the absorption at short wavelength peaking at ~330 nm, and extending to long wavelength as seen from both fig. 7 and fig. 3a. From fig. 8, the integrated current is 6.9, 9.8, 10.0, and 8.9 mA/cm² for the device at the interlayer thickness of 0, 5, 23 and 82 nm, respectively. These are very close to the measured *J*_{sc} (Table 1). For thick PCBDANI film of 82 nm, the IPCE is weaker at short wavelength, indicating that the n-doped interlayer absorbed in this region but did not contribute to the photon current, reducing the EQE in this region. In real application, however, this extra-large thickness would be not necessary. The high *FF* along with reduced *R*_s and increased *R*_{sh} indicated that the interface between the interlayer and the active layer remained optimized in the case of thick interlayer. In case of PCBANI as the interlayer, *J*_{sc} decreased only by 2 % relative to the value at its maximal PCE, while the *FF* decreased along with slightly increased *R*_s. The actual PCBDANI thickness (~55 nm) smaller than PCBDANI of 82 nm could be part of the reason for the various *J*_{sc}. In addition, while the conductivities of PCBDANI and PCBANI are similar, PCBDANI has an extra ammonium salt which does not involve in the self n-doping. This additional ammonium iodide may negatively dope the PCBM in the active layer through inter-diffusion,⁷² and made the n-doped region wider than PCBDANI layer itself, and reduced the absorption of the active layer, resulting in lower *J*_{sc} but good *FF*. However, more works are needed to clarify this point.



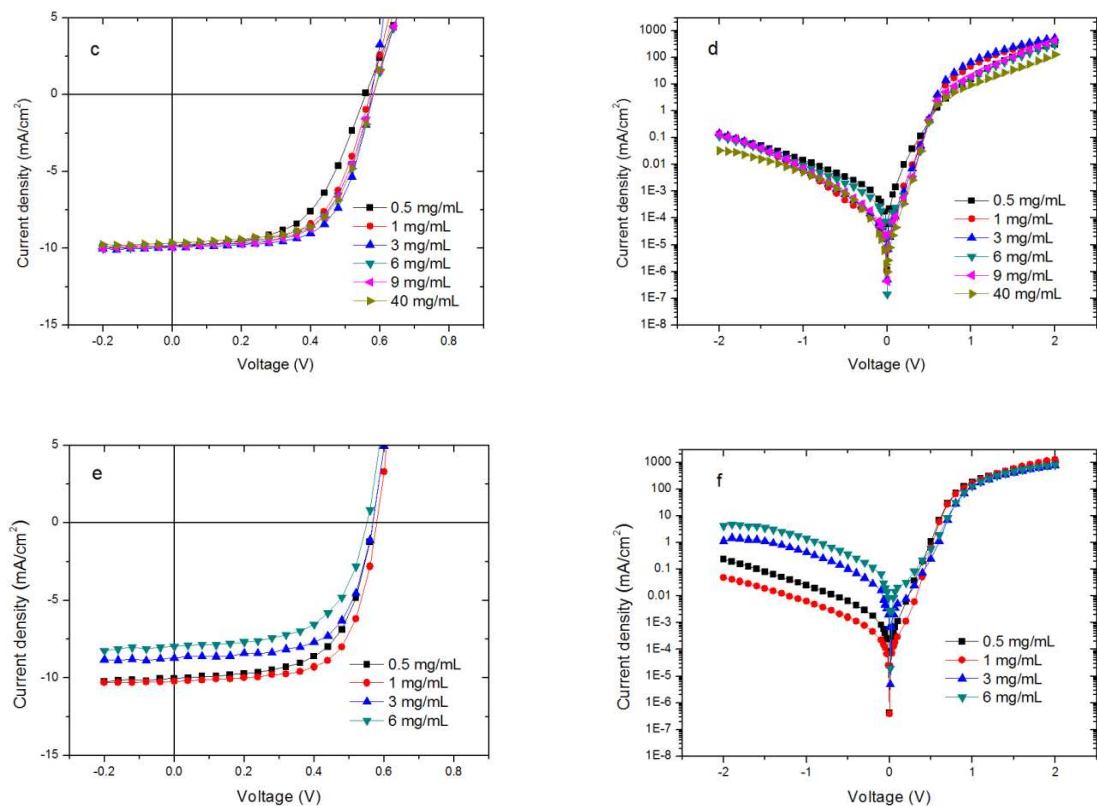


Fig. 5 J - V curves of devices ITO/PCBDANI/P3HT:PC61BM/MoO₃/Ag (a, b), ITO/PCBANI/P3HT:PC61BM/MoO₃/Ag (c, d), and ITO/PCBDANBr/P3HT:PC61BM/MoO₃/Ag (e, f) under 1000 W/m² at AM 1.5 G illumination and dark.

Table 1 Performance of devices ITO/interlayer/P3HT-PCBM/MoO₃/Ag^a.

interlayer	concentration mg mL ⁻¹	V _{oc} V	J _{sc} mA cm ⁻²	FF %	PCE %	R _{sh} kΩ cm ²	R _s Ω cm ²
PCBDANI	0	0.14±0.01	6.96±0.3	31.7±1.4	0.32±0.04	0.03±0.01	5.22±1.4
	0.5	0.47±0.02	9.87±0.21	42.4±1.5	1.97±0.19	6.7±0.6	34.5±5
	1	0.57±0.01	10.1±0.4	56.8±2.9	3.25±0.29	21.7±2.9	11.1±0.9
	3	0.57±0.01	9.75±0.12	64.2±0.7	3.59±0.08	79.6±10.3	5.1±1.1
	6	0.58±0.01	10.1±0.2	65.4±0.5	3.82±0.08	111±7	3.8±0.3
	9	0.57±0.01	9.86±0.14	65.0±2.1	3.66±0.12	136±25	2.0±0.3
	40	0.57	9.05±0.1	66.4±2.5	3.43±0.1	370±53	2.1±0.4
PCBANI	0.5	0.55±0.01	10.1±0.2	54.3±1.8	3.01±0.14	52.6±2.70	5.0±0.2
	1	0.57±0.01	9.88±0.12	58.8±2.9	3.31±0.22	140±15	4.3±0.7
	3	0.57±0.01	9.86±0.11	64.0±1.5	3.62±0.13	425±35	3.0±0.4
	6	0.58±0.01	9.81±0.14	61.1±1.2	3.46±0.09	413±32	11.5±0.1
	9	0.58	9.82±0.12	60.9±0.7	3.47±0.05	734±56	9.3±0.4
	12	0.58±0.01	9.71±0.15	61.5±1.1	3.45±0.08	524±29	11.4±0.7
	15	0.58	9.69±0.2	61.4±1.0	3.45±0.13	660±53	13.6±0.2

	18	0.58	9.68±0.13	61.6±0.8	3.47±0.07	713±28	11.5±0.5
	21	0.58±0.01	9.62±0.11	62.0±0.8	3.46±0.08	708±65	24.1±1.1
	40	0.58	9.65±0.15	61.9±1.4	3.45±0.12	463±38	27.6±2.1
PCBDANBr	0.5	0.56	10.0±0.2	61.4±2.5	3.46±0.16	81.1±10.6	1.80±0.2
	1	0.58	10.2±0.1	65.0±0.6	3.83±0.07	303±44	1.90±0.3
	3	0.57	8.71±0.18	64.5±1.1	3.19±0.04	10.4±0.9	1.7±0.1
	6	0.55	7.83±0.3	60.5±2.4	2.60±0.12	1.99±0.37	1.6±0.1

^a Average and standard deviation based on six 0.10 cm² devices on a single chip.

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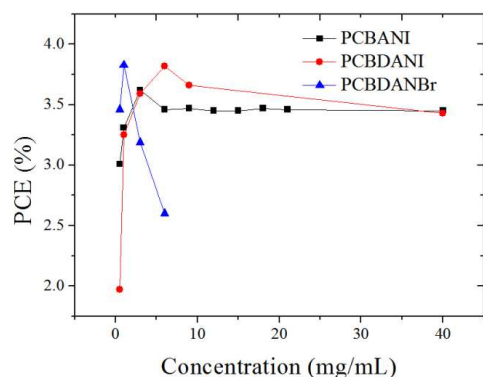


Fig. 6 PCE versus concentrations of interlayers in the I-PSCs.

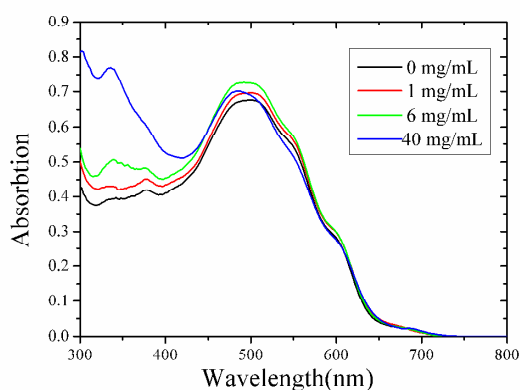


Fig. 7 UV-vis spectra of PCBDANI film spin casted from various concentrations as indicated.

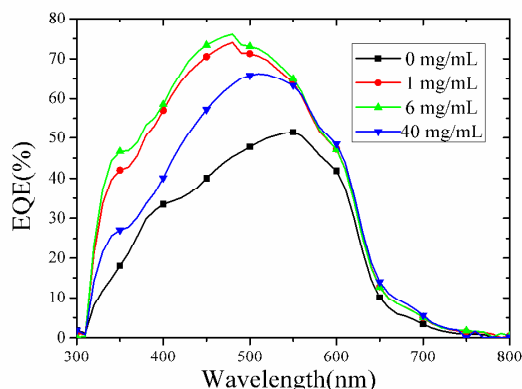


Fig. 8 EQE of I-PSCs devices with various PCBDANI thickness.

Conclusion

Self n-doped fullerene ammonium derivatives: PCBANI, PCBDANI, and PCBDANBr, have been confirmed by ESR, and conductivity measurements. The fullerene anions were in the form of C₆₀O₂⁻ anion radical as prepared. They are stable under ambient condition for processing and possess high conductivities due to self n-doped nature. Among the interlayers with different numbers of ammonium salts and different counter ions, PCBDANI showed the best solvent resistance to dichlorobenzene and the device PCE remained as high as 3.43 % (versus 3.83% maximum value) at the interlayer thickness of 82 nm. Further optimization of the structure and the processing conditions for this kind of interfacial materials would enable to achieve good device performance with a wider range of film thicknesses and large-area device processing possible. In addition, the self n-doped materials with high conductivity could also be applied in other devices such as perovskite solar cells.^{73,74}

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Notes and references

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For Table of Content

Among alcohol soluble self n-doped fullerene ammonium derivatives, the title fullerene showed the best solvent resistance and inverted polymer solar cells remained high efficiency at 82 nm of this interlayer.

Keyword polymer solar cells, n-doped fullerene, cathode interfacial layer, ammonium-modified fullerene

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Self n-Doped [6,6]-Phenyl-C61-butyric Acid 2-((2-(trimethylammonium)ethyl)-(dimethyl)ammonium) ethyl Ester Diiodides as a Cathode Interlayer for Inverted Polymer Solar Cells

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