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# Preparation and in vitro cell-biological performance of sodium alginate/nano-zinc silicate co-modified calcium silicate bioceramics 

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#### Abstract

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Sodium alginate/nano-zinc silicate (SA/nano-ZS) co-modified calcium silicate bioceramics $(\mathrm{Na} / \mathrm{Zn}-\mathrm{CS})$ were prepared via the spin-coating process. X-ray diffraction patterns indicated, the surface components of $\mathrm{Na} / \mathrm{Zn}-\mathrm{CS}$ disks were composed of calcium silicate (pseudowollastonite, CS), sodium calcium silicate and $\mathrm{Na} / \mathrm{Ca}$ co-doped zinc silicate. $\mathrm{Na} / \mathrm{Zn}$ CS disks could induce hydroxyapatite to rapidly deposit on the surface of the disks after being soaked in simulated body fluid (SBF). The pH value of SBF , in which $\mathrm{Na} / \mathrm{Zn}-\mathrm{CS}$ disks were soaked, remained in a relative low range, whereas that of SBF, in which pure CS disks were soaked, increased with the prolongation of soaking time. Whether rat bone marrow mesenchymal stem cells (rBMSCs) were co-cultured with CS disks or with $\mathrm{Na} / \mathrm{Zn}-\mathrm{CS}$ disks, the extracellular matrix mineralization of rBMSCs could be observed, which suggests the released silicon from the samples possesses osteoinductivity. F-actin staining images of rBMSCs revealed that the attachment and the spread of rBMSCs would not be influenced by the inorganic ions released from the samples, but by the interface stability of cell/material. Furthermore, the surface dissolution rate significantly affected the initial proliferation of rBMSCs. After incubation for 14 days, the alkaline phosphatase (ALP) activity of rBMSCs cultured on the surface of CS disk spin-coated with $0.1 \mathrm{~g} / \mathrm{mL}$ of $\mathrm{SA} /$ nano-ZS suspension $(\mathrm{Na} / 0.1 \mathrm{Zn}-\mathrm{CS})$ was much higher than that of rBMSCs cultured on the surface of pure CS disk, which indicates the released zinc ions can promote the differentiation of rBMSCs. In conclusion, the surface co-modification with $\mathrm{SA} /$ nano-ZS is an effective way to improve the dissolution behavior and biological performance of CS bioceramics.


Keywords: zinc silicate, sodium alginate, calcium silicate, surface modification, rBMSCs.

## 1. Introduction

Calcium silicate were widely used as bioceramics, coatings and additives owing to their excellent biocompatibility, superior bioactivity and good biomineralization ability [1-8]. Pseudowollastonite phase calcium silicate (CS) bioceramic not only have higher mechanical property than bioglass, but also show better bioactivity than the stoichiometric hydroxyapatite (HA) bioceramics. Because of its ability to release silicate ions, CS bioceramics also possess osteoinductivity, whereas calcium phosphate based biomaterials do not have such ability[9]. Silicon is one of the vital trace elements in human bone, and it plays an important role in bone formation and bone regeneration [10-12]. Silicate ions can affect osteoblast behavior and further accelerate the restoration of bone tissue. What's more, silicate ions also have the ability to up-regulate the osteogenic-related gene expressions in osteoblast [13, 14]. When calcium silicate degrades, sianols (Si-OH) will form and provide the active sites for the deposition of apatite, which facilitates the calcification of bone matrix [15, 16]. In addition, calcium silicate is capable to induce angiogenesis, which benefits for the bone regeneration [17-19].

However, the overhigh dissolution rate of CS will cause high pH value in the surrounding area, which restricts their further biological or medical applications [20]. Currently, ion doping is an effective way to control the high dissolution rate of CS. But when calcium atom is substituted by other atoms to a certain degree, the phase will change, and the solid solution such as hardystonite $\left(\mathrm{Ca}_{2} \mathrm{ZnSi}_{2} \mathrm{O}_{7}\right)$ [20, 21], akermanite $\left(\mathrm{Ca}_{2} \mathrm{MgSi}_{2} \mathrm{O}_{7}\right)$ [22-24], diopside $\left(\mathrm{CaMgSi}_{2} \mathrm{O}_{6}\right)$ [25-27], sphene $\left(\mathrm{CaTiSiO}_{5}\right)$ [28] or baghdadite $\left(\mathrm{Ca}_{3} \mathrm{ZrSi}_{2} \mathrm{O}_{9}\right)$ [29, 30] will form. Our previous study [31] had proved that the high dissolution rate of CS can be
effectively slowed down by the surface modification with nano-zinc silicate, and there was no phase change occurred in the original CS substrate. Nevertheless, CS bioceramic modified with pure nano-zinc silicate ( $\mathrm{Zn}-\mathrm{CS}$ ) failed to release zinc ions $\left(\mathrm{Zn}^{2+}\right)$ because of its stable surface structure, moreover, only a few mineralized HA crystals could deposit on the surface of $\mathrm{Zn}-\mathrm{CS}$, which is disadvantageous to bone bonding.

Zinc $(\mathrm{Zn})$ plays dual functional roles in bone tissue metabolism, it can not only help to promote bone formation through increasing the activity of osteoblast, but also simultaneously suppress bone resorption via decreasing the activity of osteoclast [32-35]. Dietary Zn supplementation is capable of down-regulating the expression of the tartrate resistant acid phosphatase (TRAP), which is associated with the osteoclast activity and bone resorption [36]. Furthermore, Zn can make osteoclast apoptosis via inhibiting its related gene expressions [33]. $10^{-6}-10^{-4} \mathrm{M}$ of Zn was able to induce mouse osteoblast-like cells (MC3T3E1) to up-regulate the expressions of runt-related transcription factor 2 (Runx-2), osteoprotegerin (OPG) and osteocalcin (OC) [37]. Therefore, it is necessary to improve the surface solubility of $\mathrm{Zn}-\mathrm{CS}$.

As a natural polysaccharide, sodium alginate (SA) has been widely used for the pharmaceutical and food applications [38], and SA is also convenient to be applied in film formation. As reported, zinc silicate nanoparticles with weak crystallinity possess good biocompatibility and biodegradable, and furthermore, zinc silicate is conducive to osteogenesis without cytotoxicity [39, 40]. In this study, sodium alginate/nano-zinc silicate (SA/nano-ZS) suspension was spin-coated on the surface of CS bioceramic disk, and a $(\mathrm{Zn}, \mathrm{Na})$-containing silicate surface layer would form on its surface. We hope $(\mathrm{Zn}, \mathrm{Na})$ -
containing silicate surface layer not only could effectively control the dissolution rate of the CS substrates, but also released a certain amount of Zn . In vitro mineralization behavior as well as biological performance of SA/nano-ZS co-modified $\mathrm{CS}(\mathrm{Na} / \mathrm{Zn}-\mathrm{CS})$ was studied in detail, and a pure CS group was served as control. Rat bone marrow mesenchymal stem cells (rBMSCs) were used to assess the osteogenic performance of $\mathrm{Na} / \mathrm{Zn}-\mathrm{CS}$.

## 2. Materials and Method

### 2.1. Materials

CS precursor powder was prepared via the chemical precipitation method as described in our previous study [31]. Briefly, 0.1 M sodium metasilicate nonahydrate $\left(\mathrm{Na}_{2} \mathrm{SiO}_{3} \cdot 9 \mathrm{H}_{2} \mathrm{O}\right)$ solution was slowly added into 0.1 M calcium chloride $\left(\mathrm{CaCl}_{2}\right)$ solution with a constant stirring. Subsequently, CS precursor powder was collected by centrifugation and washed with deionized water for 3 times, and finally, the powder was oven-dried at $80^{\circ} \mathrm{C}$ for 24 hours. Nano-zinc silicate powder was synthesized at $110{ }^{\circ} \mathrm{C}$ for 40 min by the microwave-assisted hydrothermal method using a microwave accelerated reaction system (Mars-5, CEM, USA) [40]. Zinc nitrate hexahydrate $\left(\mathrm{Zn}\left(\mathrm{NO}_{3}\right)_{2} \cdot 6 \mathrm{H}_{2} \mathrm{O}\right)$ and tetraethyl orthosilicate (TEOS) were used as Zn source and Si source, respectively, and ammonia solution ( $25 \mathrm{wt} . \%$ ) was added to accelerate the hydrolysis of TEOS. All the reagents used for synthesis were analytical pure and purchased from Sinopharm Chemical Reagent Co., Ltd in China.

CS precursor powder was filled in a cylindrical steel mould and pre-pressed at 30 MPa to form the CS precursor disks, then they were vacuum-packaged and transferred into a cold isostatic pressing equipment (LDJ100/320-300, Sichuan Chuanxi Machinery Co., Ltd, China), and isostatically pressed at 200 MPa to obtain the CS green disks. Hereafter, the obtained CS
green disks were sintered at $1150^{\circ} \mathrm{C}$ for 2 h in a furnace (HT16/17, Nabertherm, Germany) to get the densified CS bioceramic disks ( $\Phi 14 \times 2 \mathrm{~mm}$ ). Different concentrations of SA/nano-ZS suspensions were prepared by adding different amounts of nano-zinc silicate powders into the sodium alginate (B.R., Sigma-Aldrich, USA) solutions (3 wt.\%), as shown in Table 1. Finally, SA/nano-ZS suspensions were spin-coated on the surface of the sintered CS disks, and sintered again at $1100^{\circ} \mathrm{C}$ for 2 h to obtain the $\mathrm{Na} / \mathrm{Zn}-\mathrm{CS}$ disks. Furthermore, the definitions of the abbreviations such as $\mathrm{Na} / 0.1 \mathrm{Zn}-\mathrm{CS}, \mathrm{Na} / 0.05 \mathrm{Zn}-\mathrm{CS}$, and $\mathrm{Na} / 0.025 \mathrm{Zn}-\mathrm{CS}$ were listed in Table 2. In addition, all CS and $\mathrm{Na} / \mathrm{Zn}-\mathrm{CS}$ disks were sterilized using the $\gamma$-ray irradiation (20kGy) before the biological assessments.

### 2.2. In vitro mineralization behavior

The in vitro mineralization experiments were carried out by soaking CS and $\mathrm{Na} / \mathrm{Zn}-\mathrm{CS}$ disks in simulated body fluid (SBF) for 4 weeks, respectively, and SBF was replenished once per week. SBF with similar ion concentrations to those in human blood plasma was prepared according to the method described by Kokubo and Takadama [41]. Briefly, reagent-grade components listed in Table 3 (Sinopharm Chemical Reagent Co., Ltd in China) were mixed together in distilled water to yield a final 2L solution containing $142.0 \mathrm{mM} \mathrm{Na}^{+}, 5.0 \mathrm{mM} \mathrm{K}^{+}$, $1.5 \mathrm{mM} \mathrm{Mg}^{2+}, 2.5 \mathrm{mM} \mathrm{Ca}^{2+}, 103.0 \mathrm{mM} \mathrm{Cl}^{-}, 27.0 \mathrm{mM} \mathrm{HCO}^{3-}, 1.0 \mathrm{mM} \mathrm{HPO}_{4}{ }^{2-}, 0.5 \mathrm{mM} \mathrm{SO}_{4}{ }^{2-}$ at $\mathrm{pH}=7.4$. CS disks and $\mathrm{Na} / \mathrm{Zn}-\mathrm{CS}$ disks were soaked in SBF with a surface area to volume ratio of $0.1 \mathrm{~cm}^{2} / \mathrm{mL}$, and they were continuously shaken in a shaking bath (770R, Aplus, USA) at $37^{\circ} \mathrm{C}$ with a speed of 60 rpm .

## 2.3. pH value of the soaking liquid

The SBF solutions in which CS disks and $\mathrm{Na} / \mathrm{Zn}-\mathrm{CS}$ disks had soaked were collected at
the predetermined intervals, respectively. Subsequently, the pH value of the collected SBF solutions was measured by using a pH meter (Sartorius, Germany). Each group has three parallel samples in this study.

### 2.4. Surface phase composition and morphology

After being soaked in SBF for $0,1,2,3$ weeks, the surface phases of CS disks and $\mathrm{Na} / \mathrm{Zn}-\mathrm{CS}$ disks were tested by X-ray diffractometer (XRD, X'Pert PRO, PANalytical, the Netherland) using a $\mathrm{Cu} K \alpha$ source ( $\lambda=1.541874 \AA$ ), and the final components were ascertained by comparing their diffraction patterns with the Joint Committee on Powder Diffraction Standards (JCPDS) cards, respectively. In addition, the surface morphology of CS disks and $\mathrm{Na} / \mathrm{Zn}-\mathrm{CS}$ disks after being soaked in SBF for $0,1,2,3$ weeks was observed by using a fieldemission scanning electron microscope (FE-SEM, Nova NanoSEM 430, FEI, USA) equipped with an energy dispersive X-ray spectrometer (EDS, INCA X-act, Oxford, U.K.), operated at an acceleration voltage of 10 kV . Furthermore, all CS and $\mathrm{Na} / \mathrm{Zn}-\mathrm{CS}$ disks were sputtered twice with gold before FE-SEM observing.

### 2.5. Cell cultures

The primary rBMSCs (Cat.No.CRL-12424) purchased from ATCC were maintained in cell culture flasks, and then they were subcultivated in a cell incubator at $37^{\circ} \mathrm{C}$ with a $5 \% \mathrm{CO}_{2}$ humidified atmosphere. As rBMSCs were almost confluent in the bottom of the cell culture flask, $0.25 \%$ trypsin/EDTA was added to make them detachment. Finally, they were collected and stored in a liquid nitrogen container for use. The rBMSCs after 4 times of passages were used in our following cell experiments. High-glucose Dulbecco's Modified Eagle's Medium (H-DMEM, Gibco, Cat.No.11965-092) with 10 vol.\% fetal bovine serum (FBS, Hyclone,

Cat.No.NWJ0473) was applied for culturing rBMSCs, and the culture medium was refreshed every two days.

### 2.6. Cell viability assessment

The sterilized CS disk and $\mathrm{Na} / \mathrm{Zn}-\mathrm{CS}$ disk were transferred into 12 -well culture plates, and serum-free H-DMEM medium was added to prewet them for 2 h before rBMSC were seeded on their surface. The definitions of the abbreviations such as rBMSCs-CS and rBMSCs- $\mathrm{Na} / \mathrm{Zn}-\mathrm{CS}$ were listed in Table 2. 2 mL H-DMEM medium with $10 \% \mathrm{FBS}$ was added into each well with a cell density of $1.2 \times 10^{5}$ cells/well, subsequently, the culture plates containing rBMSCs- $\mathrm{Na} / \mathrm{Zn}-\mathrm{CS}$ and $\mathrm{rBMSCs}-\mathrm{CS}$ were set in a cell incubator ( $37^{\circ} \mathrm{C}$ with a $5 \%$ $\mathrm{CO}_{2}$ humidified atmosphere) for 1, 3, 7 days. Moreover, culture medium was refreshed every two days. After being cultured for 1 day, rBMSCs-CS and rBMSCs-Na/Zn-CS were rinsed twice with phosphate buffer solution (PBS) and stained with Viability Assay Kit for Animal Live Cells (Calcein-AM, Biotium, USA) according to the instruction of the manufacturer. The fluorescent images of rBMSCs were acquired using a fluorescence microscope equipped with a digital camera (40FL Axioskop, Zeiss, Germany).

Quantitative cell viability assessment was measured by using a Cell Counting Kit-8 (CCK-8, Dojindo, Japan) assay. At the pre-determined intervals, rBMSCs- $\mathrm{Na} / \mathrm{Zn}-\mathrm{CS}$ and rBMSCs-CS were transferred into new 12-well plates, and then $600 \mu \mathrm{~L}$ CCK-8 working solution was added into each well to ensure all rBMSCs-CS and rBMSCs-Na/Zn-CS to be submerged. Subsequently, the plates were set in a cell incubator for 1 h . After the above incubation, $100 \mu \mathrm{~L}$ supernatant was extracted from each well and transferred into a new 96well plate, and the absorbance of the supernatant was read at 405 nm using an enzyme linked
immunoadsorbent assay plate reader (Varioskan Flash, Thermo Scientific, USA).

### 2.7. Cell attachment, morphology and cytoskeletal organization

The rBMSCs were directly seeded on the surface of CS and $\mathrm{Na} / \mathrm{Zn}-\mathrm{CS}$ disks in a 12-well plate with a density of $1.2 \times 10^{5}$ cells/well. After incubation for 24 h , rBMSCs-CS and rBMSCs-Na/Zn-CS were rinsed with PBS twice, and then they were soaked in $2.5 \mathrm{vol} . \%$ glutaraldehyde solution for 4 h to immobilize the rBMSCs. In the following, rBMSCs-CS and rBMSCs- $\mathrm{Na} / \mathrm{Zn}-\mathrm{CS}$ were dehydrated with a graded series of ethanol (30 vol. $\%, 50 \mathrm{vol} . \%, 70$ vol. $\%, 80$ vol. $\%, 90$ vol. $\%, 95$ vol. $\%$ and 100 vol. $\%$ ), and finally dried at room temperature. The attachment and morphology of rBMSCs seeded on the surface of CS and $\mathrm{Na} / \mathrm{Zn}-\mathrm{CS}$ disks were observed by SEM (Quanta 200, FEI, USA).

After rBMSCs were incubated on the surface of CS and $\mathrm{Na} / \mathrm{Zn}-\mathrm{CS}$ disks for 7 days, respectively, rBMSCs-CS and rBMSCs-Na/Zn-CS were rinsed with PBS once, and then soaked in $4 \mathrm{vol} . \%$ formaldehyde solution for 20 min to immobilize rBMSCs. Afterward rBMSCs-CS and rBMSCs-Na/Zn-CS were submerged in Triton X-100 solution (0.1 vol.\%) for 5 min . The cytoskeletal organization of rBMSCs, cultured on the surface of the sample and on the plate margin around the sample in the same well, was investigated by labeling Factin in fixed cells with green fluorescence using cell navigator ${ }^{\mathrm{TM}} \mathrm{F}$-actin labeling kit (AAT Bioquest, USA), which was strictly performed according to the kit protocol. To visualize the nucleus, rBMSCs were counterstained with DAPI staining solution (Beyotime, China). Fluorescence images of rBMSCs were obtained by using an inverted fluorescence microscope (Eclipsc Ti-U, Nikon, Japan).

### 2.8. Alkaline phosphatase (ALP) activity assay

rBMSCs were seeded on the surface of CS disks and $\mathrm{Na} / \mathrm{Zn}-\mathrm{CS}$ disks with a density of $3 \times 10^{5}$ cells/well, respectively. The rBMSCs-CS and rBMSCs-Na/Zn-CS were incubated in 12-well plate and 2 mL osteogenic induction culture medium was added to each well for culturing rBMSCs. H-DMEM containing 10 vol. $\%$ FBS, 10 mM sodium $\beta$-glycerophosphate, $0.1 \mu \mathrm{M}$ dexamethasone and $50 \mathrm{mg} / \mathrm{mL}$ vitamin C were used as the osteogenic induction culture medium, and it was refreshed every other day. The ALP activity assay was performed after rBMSCs were cultured on the surface of CS disks and $\mathrm{Na} / \mathrm{Zn}-\mathrm{CS}$ disks for 7 and 14 days, respectively. At the predetermined intervals, rBMSCs-CS and rBMSCs-Na/Zn-CS were transferred into a new 12-well plate, and then gently rinsed by cold PBS for three times. 10 mM Tris- HCl solution $(\mathrm{pH}=7.4), 0.5 \mathrm{mM}$ magnesium chloride solution $\left(\mathrm{MgCl}_{2}\right), 0.1 \mathrm{vol} . \%$ Triton X-100 solution were used for the preparation of lysis buffer. Subsequently, $600 \mu \mathrm{~L}$ lysis buffer was added to each sample for cell lysis at $4^{\circ} \mathrm{C}$ for 2 h . Afterward cells were sonicated for 20 s and centrifuged at 3000 rpm for 5 min at $4^{\circ} \mathrm{C}$. Then $20 \mu \mathrm{~L}$ of the lysate was added to $200 \mu \mathrm{~L}$ of 5 mM p-nitrophenyl phosphate (p-NPP, Sigma-Aldrich) and incubated for 15 min at $37^{\circ} \mathrm{C}$ with the treatment of light avoidance. The reaction was stopped using 1 mL of 0.1 M NaOH and the absorbance was read at 405 nm by using an enzyme linked immunoadsorbent assay plate reader (Varioskan Flash, Thermo Scientific, USA). The ALP activity was calculated from a standard curve after normalizing to the total protein content, which was measured using Pierce BCA Protein Assay Kit (Thermo Scientific, USA). The results were expressed in millimoles of p-NPP produced per minute per milligram of protein.
2.9. Test of the released zinc ions $\left(\mathrm{Zn}^{2+}\right)$

As the culture medium was refreshed at the pre-determined intervals, the supernatants of
the replaced culture media were collected, respectively. Thereafter, the collected culture medium was digested by boiling it with concentrated nitric acid, and then the well digested culture medium was tested by an inductively coupled plasma atomic emission spectroscope (ICP-AES, Optima 5300DV, Perkin Elmer, USA) to obtain the concentration of $\mathrm{Zn}^{2+}$ released from $\mathrm{Na} / \mathrm{Zn}$-CS disks during the cell culture process.

### 2.10. Statistical Analysis

Quantitative data were presented as mean $\pm$ standard deviation, and statistical analyses were performed using a one-way analysis of variance (one-way ANOVA). A comparison between the two means was made using the turkey's test, with statistical significance set at $P$ $<0.05$.

## 3. Results

### 3.1. Surface phase composition of CS disks and Na/Zn-CS disks

All the diffraction peaks of CS disks (Fig. 1(a)) were in good agreement with those of anorthic pseudowollastonite phase calcium silicate (JCPDS card no. 074-0874). However, obvious phase change could be found after the surface modification of CS disks with different concentrations of SA/nano-ZS suspensions. As shown in Fig. 1(b), the surface composition of $\mathrm{Na} / 0.025 \mathrm{Zn}-\mathrm{CS}$ disks was composed of $\mathrm{Na} / \mathrm{Ca}$ co-doped zinc silicate (Shibkovite, JCPDS card no. 051-1432), sodium calcium silicate $\left(\mathrm{Na}_{4} \mathrm{Ca}_{4} \mathrm{Si}_{6} \mathrm{O}_{18}\right.$, JCPDS card no. 01-079-1089) and pseudowollastonite phase calcium silicate. Moreover, $\mathrm{Na} / 0.05 \mathrm{Zn}$-CS disks (Fig. 1(c)) and $\mathrm{Na} / 0.1 \mathrm{Zn}$-CS (Fig. 1(d)) disks had the same surface composition as $\mathrm{Na} / 0.025 \mathrm{Zn}-\mathrm{CS}$ disks.

### 3.2. Mineralization performance of $N a / Z n$-CS disks after being soaked in SBF

After being soaked in SBF for 1 week, it was found in Fig. 2(a) that $\mathrm{Na} / \mathrm{Zn}$ co-doped HA
(JCPDS card no. 01-070-3371) as well as sodium zinc phosphate hydrate $\left(\mathrm{NaZn}\left(\mathrm{PO}_{4}\right) \cdot \mathrm{H}_{2} \mathrm{O}\right.$, JCPDS card no. 01-089-6669) could deposit on the surface of $\mathrm{Na} / 0.1 \mathrm{Zn}-\mathrm{CS}$ disks. As the soaking time increased, pseudowollastonite phase calcium silicate, sodium calcium silicate and $\mathrm{Na} / \mathrm{Ca}$ co-doped zinc silicate were continuously dissolving. When the soaking time prolonged to 3 weeks, in spite of a small amount of pseudowollastonite phase calcium silicate and sodium calcium silicate, the surface of $\mathrm{Na} / 0.1 \mathrm{Zn}-\mathrm{CS}$ disk was almost covered by $\mathrm{Na} / \mathrm{Zn}$ co-doped HA. Similar to $\mathrm{Na} / 0.1 \mathrm{Zn}-\mathrm{CS}$ disks, the primary $\mathrm{Na} / \mathrm{Ca}$ co-doped zinc silicate, sodium calcium silicate and pseudowollastonite phase calcium silicate progressively dissolved with the prolongation of the soaking time, and $\mathrm{Na} / \mathrm{Zn}$ co-doped HA finally deposited on the surface of $\mathrm{Na} / 0.05 \mathrm{Zn}$-CS disks (Fig. 2(b)), but sodium zinc phosphate hydrate could not be found until $\mathrm{Na} / 0.05 \mathrm{Zn}-\mathrm{CS}$ disks were soaked in SBF for 2 weeks. Its surface was almost covered by $\mathrm{Na} / \mathrm{Zn}$ co-doped HA after $\mathrm{Na} / 0.025 \mathrm{Zn}-\mathrm{CS}$ disk was soaked in SBF for 2 weeks, but no sodium zinc phosphate hydrate could be found on its surface, as shown in Fig. 2(c).

After $\mathrm{Na} / 0.025 \mathrm{Zn}$-CS disks were soaked in SBF for 1 week, the pH value of the collected SBF, raised from 7.4 to 7.7 , whereas it would rapidly declined from the second week, and fluctuated at a narrow range with the increase of soaking time, as illustrated in Fig. 3. $\mathrm{Na} / 0.1 \mathrm{Zn}-\mathrm{CS}$ disks and $\mathrm{Na} / 0.05 \mathrm{Zn}-\mathrm{CS}$ disks had the similar variation trend as that of $\mathrm{Na} / 0.025 \mathrm{Zn}-\mathrm{CS}$ disks. Nevertheless, the pH value of the collected SBF, in which CS disks had been soaked, continued to increase with the prolongation of the soaking time. Fig. 4A displays the surface morphology of $\mathrm{Na} / 0.1 \mathrm{Zn}-\mathrm{CS}$ disk after being soaked in SBF for 1 week, and its partial surface area was covered by $\mathrm{Zn} / \mathrm{Na}$ co-doped HA (the regions where white arrows directed in Fig. 4A). The ( $\mathrm{Zn}, \mathrm{Na}$ )-containing surface layer (the regions where black arrows
directed in Fig. 4B) of $\mathrm{Na} / 0.1 \mathrm{Zn}-\mathrm{CS}$ disk gradually dissolved, and $\mathrm{Zn} / \mathrm{Na}$ co-doped HA deposited on the surface of the CS substrate (the regions where white arrows directed in Fig. 4B). In the same conditions, many pores with a size ranging in $10-100 \mu \mathrm{~m}$ formed on the surface of $\mathrm{Na} / 0.05 \mathrm{Zn}-\mathrm{CS}$ disk (Fig. 4C), but most of the surface of $\mathrm{Na} / 0.025 \mathrm{Zn}-\mathrm{CS}$ disk was covered by $\mathrm{Zn} / \mathrm{Na}$ co-doped HA (Fig. 4D). As the soaking time prolonged to 2 weeks, more $\mathrm{Zn} / \mathrm{Na}$ co-doped HA deposited on the surface of $\mathrm{Na} / 0.1 \mathrm{Zn}-\mathrm{CS}$ disk (Fig. 4E). Owing to its surface dissolution, some big pores formed on the surface of $\mathrm{Na} / 0.1 \mathrm{Zn}-\mathrm{CS}$ disk. In addition, small $\mathrm{Zn} / \mathrm{Na}$ co-doped HA crystals also could be found depositing on the wall of the pores, which spread along the wall of the pores like creepers (Fig. 4F). If all other conditions were equal, a relative dense mineralized $\mathrm{Zn} / \mathrm{Na}$ co-doped HA layer was formed on the surfaces of $\mathrm{Na} / 0.05 \mathrm{Zn}-\mathrm{CS}$ disks and $\mathrm{Na} / 0.025 \mathrm{Zn}$-CS disks after they were soaked in SBF for 2 weeks, respectively, which were shown in Figs. 4G-4H.

### 3.3. Viability and proliferation of rBMSCs cultured on the surfaces of CS disks and Na/Zn-CS disks

In Fig. 5, only a small amount of viable rBMSCs (green fluorescence) could be found on the surface of $\mathrm{Na} / 0.025 \mathrm{Zn}-\mathrm{CS}$ disks and $\mathrm{Na} / 0.05 \mathrm{Zn}-\mathrm{CS}$ disks, and some rBMSCs seemed to sink in the pores. Nevertheless, more viable rBMSCs attached on the surfaces of $\mathrm{Na} / 0.1 \mathrm{Zn}-\mathrm{CS}$ disks and CS disks. After being cultured on the surface of CS disks, $\mathrm{Na} / 0.025 \mathrm{Zn}-\mathrm{CS}$ disks, $\mathrm{Na} / 0.05 \mathrm{Zn}-\mathrm{CS}$ disks and $\mathrm{Na} / 0.1 \mathrm{Zn}-\mathrm{CS}$ disks for 7 days, respectively, rBMSCs significantly proliferated (Fig. 6), which suggests the above bioceramics have good biocompatibility. However, comparing to that of $\mathrm{Na} / 0.025 \mathrm{Zn}-\mathrm{CS}$ disks and $\mathrm{Na} / 0.05 \mathrm{Zn}-\mathrm{CS}$ disks, the rBMSCs cultured on the surface of CS disks for 7 days had a better proliferation. It is well accepted
that OD value is proportional to the quantity of viable rBMSCs. From the first day to the third day, the increased OD value of $\mathrm{Na} / 0.025 \mathrm{Zn}-\mathrm{CS}$ disks, $\mathrm{Na} / 0.05 \mathrm{Zn}-\mathrm{CS}$ disks, $\mathrm{Na} / 0.1 \mathrm{Zn}-\mathrm{CS}$ disks and CS disks was $0.0946,0.0326,0.059$ and 0.238 , respectively, indicating that the rBMSCs cultured on the surface of CS disks proliferated more quickly in this period. But from the third day to the seventh day, the increased OD value of $\mathrm{Na} / 0.025 \mathrm{Zn}-\mathrm{CS}$ disks, $\mathrm{Na} / 0.05 \mathrm{Zn}-\mathrm{CS}$ disks, $\mathrm{Na} / 0.1 \mathrm{Zn}-\mathrm{CS}$ disks and CS disks was $0.5134,0.1238,0.8161$ and 0.6374 , respectively, which indicated the rBMSCs seeded on the surface of $\mathrm{Na} / 0.1 \mathrm{Zn}-\mathrm{CS}$ disks had a faster proliferation in the last four days by comparing with those cultured on the surface of CS disks.
3.4. Attachment, spread and cytoskeletal organization of rBMSCs after being cultured on the surfaces of CS disks and Na/Zn-CS disks

As displayed in Fig. 7, rBMSCs possessed an ellipsoidal shape with a size of $10-15 \mu \mathrm{~m}$, and they had several filopodia. Whether rBMSCs were cultured on the surface of CS disk or $\mathrm{Na} / \mathrm{Zn}-\mathrm{CS}$ disk, they did not attach and spread well on its surface. The rBMSCs cultured on the surface of $\mathrm{Na} / 0.05 \mathrm{Zn}$-CS disks seemed to hang in the sky and supported its own weight only by using several filopodia to adhere to the walls of the pore (Fig. 7C). The rBMSCs surface (Fig. 7D) was covered by many tiny particles, similar to the mineralized HA crystals which deposited on the surface of CS disks and $\mathrm{Na} / \mathrm{Zn}-\mathrm{CS}$ disks. Furthermore, EDS testing result (Fig. 7E) indicated that the surface composition of rBMSCs consisted of Ca and P elements (detected on the surface of rBMSCs in the Fig. 7D where the red * directed), which further confirmed the mineralization occurred on the surfaces of rBMSCs .

In order to demonstrate the presence of actin filaments in rBMSCs, which were cultured
on the surface of $\mathrm{Na} / \mathrm{Zn}-\mathrm{CS}$ disks and CS disks, cytoskeletal organization was determined by staining rBMSCs with iFluor ${ }^{\mathrm{TM}} 488$-phalloidin. After seeded on the surface of CS disks and $\mathrm{Na} / 0.1 \mathrm{Zn}-\mathrm{CS}$ disks for 7 days, rBMSCs did not spread well (Fig. 8), only faint and poorly organized actin stress fibers were found in rBMSCs. Moreover, the actin stress fiber produced by rBMSCs was also less, and no cytoplasmic meshwork could be observed in rBMSCs. However, rBMSCs spread well on the plate margin around the bioceramic disk when all other conditions were equal, and they also produced more distinct and well defined stress fibers as well as actin-containing microfilaments (Fig. 9). Different from the rBMSCs attached on the surface of bioceramic disk (Fig. 8), the rBMSCs with a cytoplasmic meshwork were observed on the plate (red arrows directed region in Fig. 9), and their cytoskeleton obviously expanded.

### 3.5. Differentiation of rBMSCs cultured on the surfaces of CS disks and Na/0.1Zn-CS disks

Cell differentiation ability is closely related with the ALP activity. As shown in Fig. 10, after incubation for 7 days, no significant differences could be found, and the ALP activity of rBMSCs cultured on the surface of CS disks was very close to that cultured on the surface of $\mathrm{Na} / 0.1 \mathrm{Zn}-\mathrm{CS}$ disks. When culture time prolonged to 14 days, the ALP activity of rBMSCs cultured on the surface of both bioceramic disks increased, but the ALP activity of rBMSCs cultured on the surface of $\mathrm{Na} / 0.1 \mathrm{Zn}-\mathrm{CS}$ disks was significantly higher than that cultured on the surface of CS disks. In addition, ICP testing results proved that $\mathrm{Zn}^{2+}$ could be dissolved out from $\mathrm{Na} / 0.1 \mathrm{Zn}-\mathrm{CS}$ disks during the cell culture process (Table 4).

## 4. Discussion

The ionic radius of $\mathrm{Zn}^{2+}(74 \mathrm{pm})$ is smaller than that of $\mathrm{Ca}^{2+}(100 \mathrm{pm})$. According to the Law of Coulomb, the bonding strength of $\mathrm{Zn}-\mathrm{O}$ should be stronger than that of $\mathrm{Ca}-\mathrm{O}$.

Furthermore, different from $\mathrm{Ca}^{2+}, \mathrm{Zn}^{2+}$ exhibit partial covalent interactions in tetrahedral environment [42]. The dissolution rate of "45S5" bioglass obviously decreased with the incorporation of Zn , and almost no apatites could deposit on its surface when $20 \mathrm{wt} . \%$ of Zn were incorporated into "45S5" bioglass [43]. Moreover, the degradation of Zn -containing silica-based bioglass became slower than that without Zn [44]. In addition, hardystonite also showed poor degradability, and $\mathrm{Zn}^{2+}$ was hard to be dissolved out [20]. In fact, our previous study also proved that the structural stability of CS could be improved by only using Zn atom to partially substitute for Ca atom [31]. The ionic radius of $\mathrm{Na}^{+}$is $\operatorname{similar}(102 \mathrm{pm})$ to that of $\mathrm{Ca}^{2+}$, but it has lower valence by comparing with $\mathrm{Ca}^{2+}$ or $\mathrm{Zn}^{2+}$. For the sake of the coordination, more Na atoms have to substitute for Ca or Zn atoms, which will reduce the structural stability of Zn -containing surface layer, so it well explains why ( $\mathrm{Zn}, \mathrm{Na}$ )-containing surface layer had higher dissolution rate than that of Zn -containing surface layer.

Actually, Zn atom played a role in improving the structural stability of CS, whereas Na atom had negative effect on maintaining its structural stability. Therefore, the dissolution rate of $(\mathrm{Zn}, \mathrm{Na})$-containing surface layer is ascribed to its molar ratio of $\mathrm{Na} / \mathrm{Zn}$. The result of EDS (listed in Table 5) indicated, the $\mathrm{Na} / \mathrm{Zn}$ molar ratio on the surface of $\mathrm{Na} / 0.025 \mathrm{Zn}-\mathrm{CS}$ disk, $\mathrm{Na} / 0.05 \mathrm{Zn}-\mathrm{CS}$ disk and $\mathrm{Na} / 0.1 \mathrm{Zn}-\mathrm{CS}$ disk is $5.98,4.39,3.13$, respectively. Therefore, $\mathrm{Na} / 0.025 \mathrm{Zn}-\mathrm{CS}$ disk had a $(\mathrm{Zn}, \mathrm{Na})$-containing surface layer with relatively higher $\mathrm{Na} / \mathrm{Zn}$ ratio, so its surface dissolved faster than that of $\mathrm{Na} / 0.1 \mathrm{Zn}-\mathrm{CS}$ disk, and the fast surface dissolution could lead to rapid deposition of $\mathrm{Na} / \mathrm{Zn}$ co-doped HA on its surface (Fig. 4D), which benefits for its bonding to the bone tissue. However, the interface stability of the cell/material had significant influences on cell attachment, cell spreading and the initial cell
proliferation, and rBMSCs preferred attaching and spreading on a stable contact surface. The composition as well as topography of the hardystonite bioceramic could modulate the attachment and proliferation of human osteoblast-like cells (HOBs) [20]. But the inorganic ions (such as $\mathrm{Zn}^{2+}, \mathrm{Na}^{+}$) released from $\mathrm{Na} / \mathrm{Zn}-\mathrm{CS}$ disks had few effects on the attachment and spreading of rBMSCs, as proved by comparing Fig. 8 with Fig. 9. Zhang N.L. reported that the cell attachment and viability for fine-grained pseudowollastonite (psWf, roughness 0.74 $\mu \mathrm{m}$ ) was lower than that for coarse-grained pseudowollastonite (psWc, roughness $1.25 \mu \mathrm{~m}$ ) surface because of the faster surface dissolution of psWf [45]. Thus, a proper surface dissolution rate of biomaterials may be good for bone bonding and initial cell growth.

Bone mesenchymal stem cell has the ability to differentiate into osteoblast. During the process of bone formation, osteoblast is the primary functional cell which is responsible for the synthesis, secretion and mineralization of bone matrix. Therefore, rBMSCs were used for evaluating the osteoinductivity of CS disk and $\mathrm{Na} / \mathrm{Zn}-\mathrm{CS}$ disk in this study. The extracellular mineralization phenomenon was found on the surface of rBMSCs (Fig. 7), which suggested Si possessed osteoinductivity and rBMSCs had great possibilities to differentiate into the rat osteoblasts. Si released from the siloxane-doped poly(lactic acid)/vaterite composite scaffold could induce the formation of bone nodules on the surface of HOBs [46]. Furthermore, J. R. Jones [47] reported that the released Si was able to stimulate the synthesis of the bone matrix protein collagen I, which would induce the mineralization of HOBs. However, to the best of our knowledge, when cells are cultured with HA, $\beta-\mathrm{TCP}$ and CPC, no extracellular mineralization phenomenon can be observed. Recently, in order to enhance their osteoinductivity, more and more researchers tried to incorporate Si into calcium phosphate
based bone repair materials [48,49]. Nevertheless, the pathway and mechanism of Si to induce bone formation remains unclear up to now, and it will be further studied in the future.

As previously discussed, fast surface dissolution was detrimental to the initial cell growth, so rBMSCs cultured on the surface of $\mathrm{Na} / 0.1 \mathrm{Zn}-\mathrm{CS}$ disks proliferated slower than that cultured on the surface of CS disks in the first three days, as shown in Fig. 6. However, rBMSCs cultured on the surface of $\mathrm{Na} / 0.1 \mathrm{Zn}-\mathrm{CS}$ disks had higher proliferation rate in the last four days by comparing with that cultured on the surface of CS , which further proved the released Zn could indeed promote the proliferation of rBMSCs. Moreover, it was also found that the released Zn had the ability to stimulate rBMSCs to up-regulate the expression of ALP activity. Zn is a key element in ALP which is central to the mineralization of bone matrix [35]. Inhibiting the dissociation of the active Zn center can result in increasing the lifetime of the ALP molecule. Interestingly, Hall S. L. [50] found that the presence of Zn could increase ALP specific activity (at least in a human osteosarcoma cell line), and no similar effect was seen for a variety of other elements, including but not limited to $\mathrm{Ca}, \mathrm{Mg}, \mathrm{Mn}$. Moreover, Zn in TCP helped to enhance the ALP activity of human bone marrow stromal cells (hBMSCs). Calcium phosphate cement implants with $0.03 \mathrm{wt} \%$.Zn can significantly promote the formation of new bone [51]. MC3T3-E1 and mesenchymal stem cells (MSCs) cultured on the $\mathrm{Zn}-\mathrm{HA}$ layer showed better ALP activity and up-regulated gene expression of osteocalcin in comparison with that cultured on the HA layer [52,53]. Wang X. P. [54] reported that Zn released from a Zn -containing bioglass scaffold promoted the ALP activity of MSCs.

## 5. Conclusion

Sodium alginate/nano-zinc silicate co-modified calcium silicate ( $\mathrm{Na} / \mathrm{Zn}-\mathrm{CS}$ ) bioceramic
disks were successfully prepared via the spin-coating process. By comparing with pure pseudowollastonite phase calcium silicate (CS) bioceramic disks, $\mathrm{Na} / \mathrm{Zn}-\mathrm{CS}$ bioceramic disks not only had a proper dissolution rate, but also could induce HA to deposit rapidly on their surfaces. Whether rBMSCs were cultured on the surface of CS disk or $\mathrm{Na} / \mathrm{Zn}-\mathrm{CS}$ disk, the extracellular matrix mineralization could be observed on the surface of rBMSCs, which suggests Si probably possesses osteoinductivity. In addition, the interface stability of the cell/material had great influences on the cell attachment, cell spreading and the initial cell proliferation. In this study, rBMSCs preferred attaching and spreading on a stable surface. Owing to the released Zn , the proliferation and differentiation of rBMSCs cultured on the surface of CS disk spin-coated with $0.1 \mathrm{~g} / \mathrm{mL}$ of $\mathrm{SA} /$ nano-ZS suspension were significantly promoted. In summary, $\mathrm{Na} / \mathrm{Zn}-\mathrm{CS}$ bioceramic has great potentials for the application of bone regeneration.

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## Figure Caption:

Figure 1 XRD patterns of (a) pure CS disk, (b) $\mathrm{Na} / 0.025 \mathrm{Zn}-\mathrm{CS}$ disk, (c) $\mathrm{Na} / 0.05 \mathrm{Zn}-\mathrm{CS}$ disk, and (d) $\mathrm{Na} / 0.1 \mathrm{Zn}-\mathrm{CS}$ disk.

Figure 2 XRD patterns of the surface of (a) $\mathrm{Na} / 0.1 \mathrm{Zn}-\mathrm{CS}$ disk, (b) $\mathrm{Na} / 0.05 \mathrm{Zn}-\mathrm{CS}$ disk and $\mathrm{Na} / 0.025 \mathrm{Zn}-\mathrm{CS}$ disk after being soaked in the SBF solution for $0,1,2,3$ weeks, respectively.

Figure 3 The pH value of the SBF solution in which CS disks and $\mathrm{Na} / \mathrm{Zn}-\mathrm{CS}$ disks have been soaked. (SBF were replenished once per week, $n=3$ )

Figure 4 SEM images of the surface morphology of (A, B) $\mathrm{Na} / 0.1 \mathrm{Zn}-\mathrm{CS}$ disk, (C) $\mathrm{Na} / 0.05 \mathrm{Zn}-\mathrm{CS}$ disk and (D) $\mathrm{Na} / 0.025 \mathrm{Zn}$-CS disk after being soaked in SBF for 1 week; SEM images of (E) the surface morphology of $\mathrm{Na} / 0.1 \mathrm{Zn}-\mathrm{CS}$ disk, ( F ) the wall of the pores in the surface of $\mathrm{Na} / 0.1 \mathrm{Zn}-\mathrm{CS}$ disk, the surface morphology of (G) $\mathrm{Na} / 0.05 \mathrm{Zn}-\mathrm{CS}$ disk and (H) $\mathrm{Na} / 0.025 \mathrm{Zn}$-CS disk after being soaked in the SBF solution for 2 weeks. The black arrows in B indicate the original $(\mathrm{Zn}, \mathrm{Na})$-containing layer on the surface of $\mathrm{Na} / 0.1 \mathrm{Zn}-\mathrm{CS}$ disk, and the white arrows in A and B indicate the deposited $\mathrm{Zn} / \mathrm{Na}$ co-doped HA layer after $\mathrm{Na} / 0.1 \mathrm{Zn}-\mathrm{CS}$ disk being soaked in SBF for 1 week.

Figure 5 The live cell staining fluorescence images of rBMSCs cultured on the surface of $\mathrm{Na} / \mathrm{Zn}-\mathrm{CS}$ disk and CS disk for 1 day, respectively.

Figure 6 The proliferation of rBMSCs cultured on the surface of $\mathrm{Na} / \mathrm{Zn}-\mathrm{CS}$ disk and CS disk for $1,3,7$ days, respectively. ( $\mathrm{n}=4$ ) *The $\mathrm{Na} / \mathrm{Zn}-\mathrm{CS}$ group compared with the CS group, $P<0.05$. CS is used as the control group and $0.64 \%$ phenol solutions were used as the positive control ( $\mathrm{Ctr}^{+}$).

Figure 7 SEM images of rBMSCs attached on the surface of (A) CS disk, (B) $\mathrm{Na} / 0.1 \mathrm{Zn}-\mathrm{CS}$ disk, (C) $\mathrm{Na} / 0.05 \mathrm{Zn}-\mathrm{CS}$ disk and (D) $\mathrm{Na} / 0.025 \mathrm{Zn}-\mathrm{CS}$ disk after being co-cultured for 1 day. (E) EDS test result of the element composition of tiny particles (red *) deposited on the surface of rBMSCs.

Figure 8 The staining images of F-actin (green, left), nucleus (blue, middle) and their merged images (right) of rBMSCs cultured on the surface of CS disk and $\mathrm{Na} / 0.1 \mathrm{Zn}-\mathrm{CS}$ disk for 7 days, respectively.

Figure 9 The staining images of F-actin (green, left), nucleus (blue, middle) and their merged images (right) of rBMSCs attached on the 12-well plate when they were co-cultured with CS disks and $\mathrm{Na} / 0.1 \mathrm{Zn}-\mathrm{CS}$ disk for 3,7 days, respectively. Red arrows directed region refers to the cytoplasmic meshwork, which was formed by the expanding of the cytoskeleton.

Figure 10 The ALP activities of rBMSCs cultured on the surface of CS disk and $\mathrm{Na} / 0.1 \mathrm{Zn}-\mathrm{CS}$ disk for 7,14 days, respectively. $(\mathrm{n}=4) * \mathrm{The} \mathrm{Na} / 0.1 \mathrm{Zn}-\mathrm{CS}$ group compared with the pure CS group after the incubation for 14 days, $P<0.05$.

## Tables

Table 1 The composition of the sodium alginate/nano-zinc silicate suspensions.

| Sample | Nano-Zinc Silicate (g) | Sodium Alginate (g) | Deionized Water (mL) |
| :---: | :---: | :---: | :---: |
| $\mathrm{Na} / 0.1 \mathrm{Zn}$ | 0.4 | 0.1237 | 4 |
| $\mathrm{Na} / 0.05 \mathrm{Zn}$ | 0.2 | 0.1237 | 4 |
| $\mathrm{Na} / 0.025 \mathrm{Zn}$ | 0.1 | 0.1237 | 4 |

Table 2 The definitions of the abbreviations used in this study.

| Abbreviation | Definition |
| :---: | :---: |
| $\mathrm{SA} /$ nano-ZS | Sodium alginate $/$ nano-zinc silicate suspension |
| SA | Sodium alginate |
| CS | Pseudowollastonite phase calcium silicate |
| $\mathrm{Zn}-\mathrm{CS}$ | Pure nano-zinc silicate modified CS |
| $\mathrm{Na} / \mathrm{Zn}-\mathrm{CS}$ | $\mathrm{SA} /$ nano-ZS co-modified CS |
| $\mathrm{Na} / 0.1 \mathrm{Zn}-\mathrm{CS}$ | $\mathrm{SA} / \mathrm{nano-ZS(0.1g/mL)} \mathrm{co-modified} \mathrm{CS}$ |
| $\mathrm{Na} / 0.05 \mathrm{Zn}-\mathrm{CS}$ | $\mathrm{SA} /$ nano-ZS $(0.05 \mathrm{~g} / \mathrm{mL})$ co-modified CS |
| $\mathrm{Na} / 0.025 \mathrm{Zn}-\mathrm{CS}$ | $\mathrm{SA} /$ nano-ZS $(0.025 \mathrm{~g} / \mathrm{mL})$ co-modified CS |
| $\mathrm{rBMSCs}-\mathrm{CS}$ | the cell-bioceramic construct that rBMSCs |
|  | were seeded on the surface of CS disks |
| $\mathrm{rBMSCs}-\mathrm{Na} / \mathrm{Zn}-\mathrm{CS}$ | the cell-bioceramic construct that rBMSCs |
|  | were seeded on the surface of $\mathrm{Na} / \mathrm{Zn}-\mathrm{CS}$ disks |

Table 3 The amounts of each chemical regent in SBF (2L).

| Order | Regent | Amount |
| :---: | :---: | :---: |
| 1 | NaCl | 16.07 g |
| 2 | $\mathrm{NaHCO}_{3}$ | 0.71 g |
| 3 | KCl | 0.45 g |
| 4 | $\mathrm{~K}_{2} \mathrm{HPO}_{4} \cdot 3 \mathrm{H}_{2} \mathrm{O}$ | 0.462 g |
| 5 | $\mathrm{MgCl}_{2} \cdot 6 \mathrm{H}_{2} \mathrm{O}$ | 0.622 g |
| 6 | $\mathrm{HCl}^{(1 \mathrm{M})}$ | 78 mL |
| 7 | $\mathrm{CaCl}_{2}$ | 0.584 g |
| 8 | $\mathrm{Na}_{2} \mathrm{SO}_{4}$ | 0.144 g |
| 9 | Tris | 12.236 g |

Table 4 The contents of $\mathrm{Zn}^{2+}$ released from $\mathrm{Na} / 0.1 \mathrm{Zn}-\mathrm{CS}$ disks as co-cultured with rBMSCs.

| Culture Time (day) | 1 | 3 | 13 | 15 |
| :---: | :---: | :---: | :---: | :---: |
| $\mathrm{Zn}(\mu \mathrm{M})$ | 13.54 | 4.27 | 5.64 | 3.46 |

Table $5 \mathrm{The} \mathrm{Na} / \mathrm{Zn}$ molar ratio on the surface of $\mathrm{Na} / \mathrm{Zn}-\mathrm{CS}$ disks which were surface comodified with different concentrations of SA/nano-ZS suspensions.

| Sample | Na (Atomic Conc \%) | Zn (Atomic Conc \%) | Molar ratio of $\mathrm{Na} / \mathrm{Zn}$ |
| :---: | :---: | :---: | :---: |
| $\mathrm{Na} / 0.025 \mathrm{Zn}-\mathrm{CS}$ | 2.75 | 0.46 | 5.98 |


| $\mathrm{Na} / 0.05 \mathrm{Zn}-\mathrm{CS}$ | 6.63 | 1.51 | 4.39 |
| :---: | :--- | :--- | :--- |
| $\mathrm{Na} / 0.1 \mathrm{Zn}-\mathrm{CS}$ | 9.23 | 2.95 | 3.13 |

## C: Pseudowollastonite phase calcium silicate <br> S : Na /Ca co-doped zinc silicate <br> T: Sodium calcium silicate



Figure 1 XRD patterns of (a) pure CS disk, (b) $\mathrm{Na} / 0.025 \mathrm{Zn}-\mathrm{CS}$ disk, (c) $\mathrm{Na} / 0.05 \mathrm{Zn}-\mathrm{CS}$ disk, and (d) $\mathrm{Na} / 0.1 \mathrm{Zn}-\mathrm{CS}$ disk.
$114 \times 200 \mathrm{~mm}$ ( $300 \times 300$ DPI)

## C: Pseudowollastonite phase calcium silicate <br> H: $\mathrm{Na} / \mathrm{Zn}$ co-doped HA <br> T: Sodium calcium silicate <br> $\mathrm{S}: \mathrm{Na} / \mathrm{Ca}$ co-doped zinc silicate <br> P: Sodium zinc phosphate hydrate



Figure 2 XRD patterns of the surface of (a) $\mathrm{Na} / 0.1 \mathrm{Zn}-\mathrm{CS}$ disk, (b) $\mathrm{Na} / 0.05 \mathrm{Zn}-\mathrm{CS}$ disk and $\mathrm{Na} / 0.025 \mathrm{Zn}-\mathrm{CS}$ disk after being soaked in the SBF solution for $0,1,2,3$ weeks, respectively.

$$
160 \times 127 \mathrm{~mm}(300 \times 300 \text { DPI })
$$



Figure 3 The pH value of the SBF solution in which CS disks and $\mathrm{Na} / \mathrm{Zn}-\mathrm{CS}$ disks have been soaked. (SBF were replenished once per week, $n=3$ ) $61 \times 47 \mathrm{~mm}(300 \times 300$ DPI)


Figure 4 SEM images of the surface morphology of (A, B) Na/0.1Zn-CS disk, (C) $\mathrm{Na} / 0.05 \mathrm{Zn}-\mathrm{CS}$ disk and (D) $\mathrm{Na} / 0.025 \mathrm{Zn}$-CS disk after being soaked in SBF for 1 week; SEM images of (E) the surface morphology of $\mathrm{Na} / 0.1 \mathrm{Zn}-\mathrm{CS}$ disk, (F) the wall of the pores in the surface of $\mathrm{Na} / 0.1 \mathrm{Zn}-\mathrm{CS}$ disk, the surface morphology of (G) $\mathrm{Na} / 0.05 \mathrm{Zn}-\mathrm{CS}$ disk and (H) $\mathrm{Na} / 0.025 \mathrm{Zn}-\mathrm{CS}$ disk after being soaked in the SBF solution for 2 weeks. The black arrows in B indicate the original ( $\mathrm{Zn}, \mathrm{Na}$ )-containing layer on the surface of $\mathrm{Na} / 0.1 \mathrm{Zn}-\mathrm{CS}$ disk, and the white arrows in $A$ and $B$ indicate the deposited $\mathrm{Zn} / \mathrm{Na}$ co-doped HA layer after $\mathrm{Na} / 0.1 \mathrm{Zn}-\mathrm{CS}$ disk being soaked in SBF for 1 week.
$139 \times 60 \mathrm{~mm}$ ( $300 \times 300$ DPI)


Figure 5 The live cell staining fluorescence images of rBMSCs cultured on the surface of $\mathrm{Na} / \mathrm{Zn}-\mathrm{CS}$ disk and CS disk for 1 day, respectively.
$119 x 89 \mathrm{~mm}$ ( $300 \times 300$ DPI)


Figure 6 The proliferation of rBMSCs cultured on the surface of $\mathrm{Na} / \mathrm{Zn}-\mathrm{CS}$ disk and CS disk for $1,3,7$ days, respectively. ( $\mathrm{n}=4$ ) *The $\mathrm{Na} / \mathrm{Zn}$-CS group compared with the CS group, $\mathrm{P}<0.05$. CS is used as the control group and $0.64 \%$ phenol solutions were used as the positive control (Ctr+).
$80 \times 81 \mathrm{~mm}(300 \times 300$ DPI)


Figure 7 SEM images of rBMSCs attached on the surface of (A) CS disk, (B) Na/0.1Zn-CS disk, (C)
$\mathrm{Na} / 0.05 \mathrm{Zn}-\mathrm{CS}$ disk and (D) $\mathrm{Na} / 0.025 \mathrm{Zn}-\mathrm{CS}$ disk after being co-cultured for 1 day. (E) EDS test result of the element composition of tiny particles (red *) deposited on the surface of rBMSCs.
$218 \times 293 \mathrm{~mm}$ ( $300 \times 300$ DPI)


Figure 8 The staining images of F -actin (green, left), nucleus (blue, middle) and their merged images (right) of rBMSCs cultured on the surface of CS disk and $\mathrm{Na} / 0.1 \mathrm{Zn}-\mathrm{CS}$ disk for 7 days, respectively. $121 \times 47 \mathrm{~mm}$ ( $300 \times 300$ DPI)


Figure 9 The staining images of F-actin (green, left), nucleus (blue, middle) and their merged images (right) of rBMSCs attached on the 12-well plate when they were co-cultured with CS disks and $\mathrm{Na} / 0.1 \mathrm{Zn}$-CS disk for 3, 7 days, respectively. Red arrows directed region refers to the cytoplasmic meshwork, which was formed by the expanding of the cytoskeleton.
$307 \times 241 \mathrm{~mm}(300 \times 300$ DPI)


Figure 10 The ALP activities of rBMSCs cultured on the surface of CS disk and $\mathrm{Na} / 0.1 \mathrm{Zn}-\mathrm{CS}$ disk for 7, 14 days, respectively. $(\mathrm{n}=4) *$ The $\mathrm{Na} / 0.1 \mathrm{Zn}-\mathrm{CS}$ group compared with the pure CS group after the incubation for 14 days, $\mathrm{P}<0.05$.
$60 \times 46 \mathrm{~mm}(300 \times 300$ DPI)


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