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1 Enzymatic Degradation and Porous Morphology of the Poly(L-lactide) and 2 Poly(3-hydroxybutyrate-co-4-hydroxybutyrate) Blends

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10 Abstract

11 Fully biodegradable polymer blends based on biosourced polymers, namely poly(L-lactide)
12 (PLLA) and poly(3-hydroxybutyrate-co-4-hydroxybutyrate) (P(3HB-co-4HB)) are prepared
13 by melt compounding. The enzymatic degradation and porous morphology of
14 PLLA/P(3HB-co-4HB) blends are investigated in detail. The lipase from *Pseudomonas*
15 *mendocina* reveals preferred enzymatic degradation of P(3HB-co-4HB) but insignificant
16 attack to PLLA in the blends. At the same time, proteinase K can degrade PLLA, but cannot
17 degrade P(3HB-co-4HB). On account of the surface erosion mechanisms, the enzymatic
18 degradation rates of both the P(3HB-co-4HB) and PLLA in the blends are improved because
19 of the presence of the other component to increase the specific surface area. The results of the
20 ¹H NMR and GPC indicated that there was no more intermediate products formed during the
21 enzymatic degradation of the PLLA and P(3HB-co-4HB). Due to the specificity of the
22 degradation enzymes, selective enzymatic degradation is adopted to degrade and remove one
23 component from the blends, and various porous morphologies are acquired.

24

25 1. Introduction

26 The porous materials have received more and more attention in recent years due to their
27 special surface property. They are widely used in gas-liquid separation, catalyst carrier, tissue
28 engineering, and so on. There are many fabrication methods for the porous polymers, such as
29 fiber bonding,¹ particulate leaching,² solvent casting and selective polymer extraction,³
30 thermally induced phase separation (TIPS),⁴ gas foaming, solid free-form (SFF),⁵ etc. All
31 approaches, however, have limitations. Such as particulate leaching allows for an easy control
32 of pore structure while the using of organic solvent and the residual porogen may be the most
33 disadvantages. Phase separation requires the dissolution of polymers in a solvent which will
34 also be harmful to the cell and organs. Therefore, exploring new methods for the
35 fabrication of porous materials remain to be a challenge. Selective biodegradation was a
36 newer method which was introduced to fabricate the porous polymers in recent years. Liu et
37 al⁶ fabricated poly(L-lactide) (PLLA)/poly(ϵ -caprolactone) (PCL) blend films by solution
38 casting. The different biodegradable porous polymers can be acquired with selectively
39 removing one of the phase by using the selective biodegradation of the proteinase K and
40 *Pseudomonas* lipase to the PLLA and PCL, respectively. A similar method was adopted by
41 Tsuji et al⁷ and Hsiue et al⁸ to prepare the porous films. However, all of the porous materials
42 obtained through the selective biodegradation method were restricted to porous films, and the
43 organic solvent was introduced inevitably, which was not environment-friendly.

44 PLLA, a promising aliphatic polyester derived from renewable resources, has been widely
45 used in biomedical fields due to its biocompatibility, biodegradability, and good mechanical
46 properties. The enzymatic degradation of the PLLA with different treatments was investigated
47 in detail.⁹⁻¹⁴ Tsuji et al revealed that the weight loss rates of the PLLA films after proteinase
48 K-catalyzed hydrolysis increased with increasing the alkaline treatment time⁹ or after
49 hydrophilic polymer coating¹⁰ due to the increased surface hydrophilicity. The

50 poly(3-hydroxybutyrate-*co*-4-hydroxybutyrate) (P(3HB-*co*-4HB)), a member of the
51 poly(hydroxyalkanoates) (PHAs) family which are another type of biosourced polymers, is
52 desirable for applications in biomedical and environmental fields due to the good
53 biodegradability, biocompatibility, relatively good toughness and process properties.¹⁵⁻¹⁷ The
54 biodegradation of P(3HB-*co*-4HB) was investigated extensively¹⁸⁻²³ and the results revealed
55 that the degradation rate of P(3HB-*co*-4HB) depended strongly on its crystallinity and surface
56 morphology due to its surface erosion mechanism. The biodegradation of
57 PLA/P(3HB-*co*-4HB) blends were also investigated. The results revealed P(3HB-*co*-4HB)
58 and PLA had different degradation rate due to different degradation mechanisms.²⁴ The
59 addition of the P(3HB-*co*-4HB) in the PLA could accelerate the biodegradability of the blends
60 whether in laboratory²⁵ or in soil condition.²⁴

61 Up to now, it was confirmed that the lipase from *Pseudomonas mendocina* only degraded
62 P(3HB-*co*-4HB) but did not attack PLA in the blends.²⁵ At the same time, proteinase K only
63 degraded PLA, but not degraded P(3HB-*co*-4HB).^{26,27} Therefore, the selective biodegradation
64 could be used to the blends of PLA/P(3HB-*co*-4HB) to produce the porous materials.

65 In this paper, different ratio of PLLA/P(3HB-*co*-4HB) blends were prepared by melt
66 compounding. *Pseudomonas mendocina* lipase and proteinase K were used as specified
67 degradation enzymes for P(3HB-*co*-4HB) and PLLA chains, respectively, and the effects of
68 one component on the enzymatic degradation behavior of the other component were
69 investigated in detail. Meanwhile, selective enzymatic degradation was adopted to degrade
70 and remove one component from the blend, and porous morphology of PLLA/P(3HB-*co*-4HB)
71 blends were investigated.

72 **2. Experimental Section**

73 **2.1 Materials**

74 The PLLA (Grade 4032D) used in this work was a commercially available product from
75 Natureworks LLC (USA). It exhibited a weight-average molecular weight of $2.07 \times 10^5 \text{ g mol}^{-1}$
76 and polydispersity of 1.74 as determined by gel permeation chromatography (GPC). The
77 P(3HB-*co*-4HB) was provided by Tianjin Guoyun Biotech (Tianjin, P. R. China). It exhibited
78 a weight-average molecular weight of $4.97 \times 10^5 \text{ g mol}^{-1}$ and polydispersity of 1.85 (GPC
79 analysis). The content of 4HB in the copolymer was 6.5 mol% determined by ^1H -nuclear
80 magnetic resonance (NMR) spectroscopy.

81 2.2 Preparation of the Blends

82 Before processing, PLLA and P(3HB-*co*-4HB) were dried at 80 °C in a vacuum oven for 24
83 h. PLLA/P(3HB-*co*-4HB) blends with a series of weight ratios (100/0, 90/10, 80/20, 70/30,
84 60/40, 50/50, 40/60, 30/70, 20/80, 10/90, 0/100)(wt/wt) were prepared by using a Haake batch
85 internal mixer (Haake Rheomix 600, Karlsruhe, Germany) with a batch volume of 50 mL.
86 The melt compounding was performed at 175 °C and a screw speed of 50 rpm during a total
87 mixing time of 8 min, until the viscosity had reached a nearly constant value. After mixing,
88 all the samples were cut into small pieces and then were hot-pressed at 180 °C for 3 min
89 followed by cold-press at room temperature to form the sheets with thickness of 3 mm or 0.1
90 mm. The compression molding steps were carried out carefully in order to obtain the same
91 treatment for every sample.

92 2.3 Enzymatic Degradation

93 The enzymatic degradation of the blend films for the P(3HB-*co*-4HB) was carried out in
94 phosphate buffer (pH=8.0) containing *Pseudomonas mendocina* at 30 °C with shaking at 140
95 rpm. The composites films from the pressed sheets with thicknesses of 0.1 mm were chopped
96 into square with gauge dimensions of $10 \times 10 \text{ mm}^2$. Then all samples were placed in small

97 glass bottles containing the buffer and *Pseudomonas mendocina*. The samples were picked up
98 after a fixed time interval, washed with distilled water, and dried to constant weight in a
99 vacuum, and then the weights of the films were measured.

100 The enzymatic degradation of the blend films for PLLA was carried out in Tris-HCl
101 buffered solution (pH=8.0) containing 0.2 mg mL⁻¹ of proteinase K (Genview) at 45 °C with
102 shaking at 140 rpm. Sample films (10×10×0.1 mm³) were placed in small glass bottles filled
103 with 1.5 mL Tris-HCl buffered solution containing proteinase K. The films were periodically
104 removed, washed with distilled water, and dried to constant weight in a vacuum, and then the
105 weights of the films were measured.

106 Selective enzymatic degradation was used to completely degrade and remove the
107 P(3HB-co-4HB) or PLLA from the cryo-fractured surfaces of the blends to obtain the
108 corresponding porous materials. The cryo-fractured surfaces were obtained through that the
109 blends from the pressed sheets with thicknesses of 3 mm were immersed in liquid nitrogen for
110 about 5 min, and then broken. The selective enzymatic degradation of the blends was carried
111 out in the same conditions as above. When the P(3HB-co-4HB) or PLLA component in the
112 sample was degraded completely, the sample was removed, washed with distilled water, and
113 dried to constant weight in a vacuum.

114 **2.4 Measurement and Observation**

115 Thermal analysis was performed using a TA Instruments differential scanning calorimeter
116 DSC Q20 (USA) under N₂ atmosphere. The specimens were crimp sealed in aluminum
117 crucibles and had a nominal weight of about 4~6 mg. The specimens before and after
118 enzymatic degradation were heated from the ambient temperature to 190 °C at a heating rate
119 of 10 °C min⁻¹ in the determination of the melting enthalpy (ΔH_m) of the blends. The degree
120 of crystallinity of the P(3HB-co-4HB) was calculated by the following equations:

$$\chi_{c, P(3HB-co-4HB)} (\%) = \Delta H_{m, P(3HB-co-4HB)} \cdot 100 / [X_{P(3HB-co-4HB)} \cdot 146] \quad (1)$$

Where $X_{P(3HB-co-4HB)}$ is the weight ratio of P(3HB-co-4HB) in the blends and will be shown as follows. $\Delta H_{m, P(3HB-co-4HB)}$ ($J g^{-1}$ of polymer) is the melting enthalpy of P(3HB-co-4HB), and 146 ($J g^{-1}$ of P(3HB-co-4HB)) is the melting enthalpy of P(3HB-co-4HB) with 100% crystallinity reported by Barham P.J..²⁸

Percentage weight loss was calculated according to the following equation using the weights of a film before and after degradation (W_{before} and W_{after} respectively)

$$\text{Non-normalized weight loss (wt\%)} = 100(W_{before} - W_{after})/W_{before} \quad (2)$$

The non-normalized weight loss obtained by Equation (2) was normalized by degradable PLLA or P(3HB-co-4HB) weight fraction using the equations:

$$\begin{aligned} &\text{Normalized P(3HB-co-4HB) weight loss (wt\%)} \\ &= \text{Non-normalized weight loss}/X_{P(3HB-co-4HB)} \end{aligned} \quad (3)$$

(for *Pseudomonas mendocina* lipase-catalyzed enzymatic degradation).

$$\begin{aligned} &\text{Normalized PLLA weight loss (wt\%)} \\ &= \text{Non-normalized weight loss}/(1-X_{P(3HB-co-4HB)}) \end{aligned} \quad (4)$$

(for proteinase K-catalyzed enzymatic degradation), and

$$X_{P(3HB-co-4HB)} = W_{P(3HB-co-4HB)} / (W_{PLLA} + W_{P(3HB-co-4HB)}) \quad (5)$$

Where $W_{P(3HB-co-4HB)}$ and W_{PLLA} are the weights of P(3HB-co-4HB) and PLLA, respectively, in a film. The experimental weight loss values represent averages of measurements from the three replicate specimens.

The 1H NMR spectra were recorded using a Bruker 300 MHz spectrometer with $CDCl_3$ as solvent and tetramethylsilane (TMS) as an internal standard.

The weight-average molecular weight (M_w), number-average molecular weight (M_n) and molecular weight distribution were determined by Gel Permeation Chromatography (GPC) conducted in $CHCl_3$ at $35^\circ C$, at a flow rate of $1 mL min^{-1}$ using a Waters 515 HPLC pump

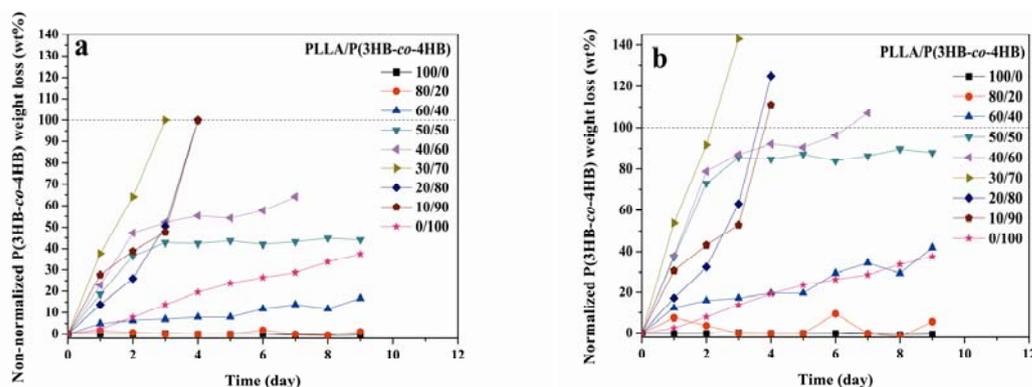
146 solvent delivery system with a set of two Waters Styragel HT4 and HT3 column and Waters
 147 2414 refractive index detector. Polystyrene standards with narrow molar mass distribution
 148 were used to generate a calibration curve.

149 The cryo-fractured surfaces after degraded and removed the one component from the
 150 blends were coated with a thin layer of gold and then they were observed with a field
 151 emission scanning electron microscopy (XL30 ESEM FEG, FEI Co., Eindhoven, The
 152 Netherlands) at an accelerating voltage of 10 kV to obtain the porous morphology of the
 153 PLLA and P(3HB-*co*-4HB), respectively. The size of the pore for the porous polymers was
 154 acquired with the Image Analysis.

155 3. Results and Discussions

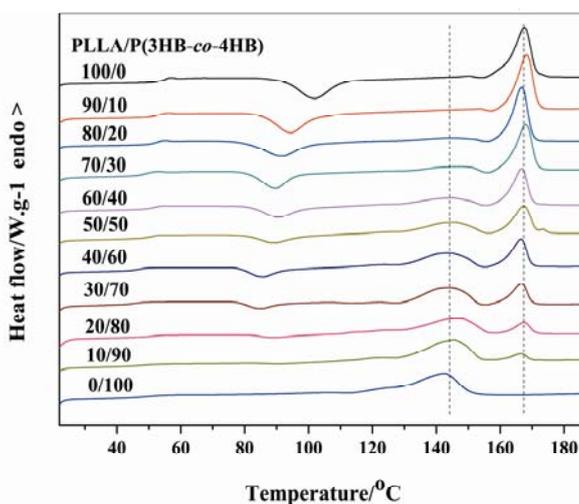
156 3.1 Weight Loss of PLLA/P(3HB-*co*-4HB) Blends after Enzymatic Degradation

157 3.1.1 *Pseudomonas Mendocina* Lipase-Catalyzed Degradation of P(3HB-*co*-4HB)
 158 Component



159
 160 **Figure 1** The non-normalized (a) and normalized (b) P(3HB-*co*-4HB) weight loss profiles of
 161 PLLA/P(3HB-*co*-4HB) blend films as a function of time during the *Pseudomonas mendocina*
 162 lipase-catalyzed degradation.

163 **Figure 1** shows the non-normalized and normalized weight loss profiles of the films as a
164 result of *Pseudomonas mendocina* lipase-catalyzed degradation. The rate of enzymatic
165 degradation could be determined from the slope of the weight loss against time. Here, only
166 the P(3HB-*co*-4HB) in the films was degraded since the neat PLLA had little weight loss. As
167 is shown in Figure 1, the weight loss of the PLLA/P(3HB-*co*-4HB) blend was increased as a
168 function of time during the enzymatic degradation. Neat P(3HB-*co*-4HB) showed a gently
169 curve which indicated a slow enzymatic degradation rate. However, the blend with the
170 addition of 10 wt% PLLA into the P(3HB-*co*-4HB) exhibited greatly increasing of the
171 enzymatic degradation rate. The samples with PLLA content of 30 wt%, 40 wt%, 50 wt% had
172 the most fast enzymatic degradation rate and these samples could be completely biodegraded
173 within four days. A lot of evidence demonstrated that the rate of enzymatic hydrolysis of
174 biodegradable materials was dependent on the degree of crystallinity^{18, 21, 22} and the surface
175 area of the polymer^{18, 19} exposed to enzymatic hydrolysis. In order to clarify the reason of the
176 increased enzymatic degradation rate for the blends, the thermal properties of the blends were
177 first investigated to determine the degree of crystallinity of the P(3HB-*co*-4HB) in the blends.



178
179 **Figure 2** The DSC thermograms of the first heating traces for PLLA/P(3HB-*co*-4HB) blends
180 with different compositions.

181 **Figure 2** shows the DSC thermograms of the first heating traces for PLLA/P(3HB-*co*-4HB)
 182 blends with different compositions. The melting enthalpy (ΔH_m) of the P(3HB-*co*-4HB)
 183 derived from the DSC curves and the corresponding calculated degree of crystallinity of the
 184 P(3HB-*co*-4HB) are summarized in **Table 1**. In Figure 2, the neat P(3HB-*co*-4HB) and neat
 185 PLLA exhibited only one melting temperature (T_m) of 142.9 °C and 167.5 °C, respectively.
 186 While for the blends of PLLA/P(3HB-*co*-4HB) with various compositions, there were two
 187 melting temperatures corresponding to the P(3HB-*co*-4HB) and PLLA, respectively, and the
 188 melting enthalpy of the two components varied with the content of the component.

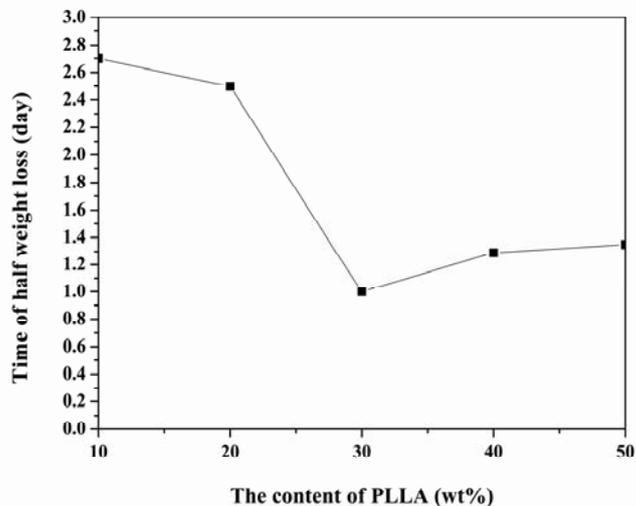
189 **Table 1** The parameters of the PLLA/P(3HB-*co*-4HB) blends

| PLLA/ P(3HB- <i>co</i> -4HB) | $\Delta H_{m,P(3HB-co-4HB)}^a)$ (J g ⁻¹) | $\chi_{c,P(3HB-co-4HB)}$ (%) | Size of porous PLLA (μm) | Size of porous P(3HB- <i>co</i> -4HB) (μm) |
|---------------------------------|---|---------------------------------|-----------------------------|--|
| 100/0 | - | - | - | - |
| 90/10 | 41.04 | 28.1 | 0.39 ± 0.15 | - |
| 80/20 | 40.13 | 27.5 | 1.08 ± 0.39 | - |
| 70/30 | 43.07 | 29.5 | 2.65 ± 1.76 | - |
| 60/40 | 42.53 | 29.0 | 13.15 ± 6.41 | - |
| 50/50 | 41.24 | 28.2 | - | - |
| 40/60 | 42.83 | 29.3 | - | 15.91 ± 8.29 |
| 30/70 | 40.87 | 28.0 | - | 3.14 ± 4.40 |
| 20/80 | 41.20 | 28.2 | - | 1.73 ± 0.70 |
| 10/90 | 42.11 | 28.8 | - | 0.61 ± 0.23 |
| 0/100 | 41.14 | 28.2 | - | - |

190 ^{a)} $\Delta H_{m,P(3HB-co-4HB)}$ are corrected for the content of P(3HB-*co*-4HB) in the blends.

191 From Table 1, it was clear that the addition of PLLA showed little effect on the variation of
 192 the degree of crystallinity of P(3HB-*co*-4HB) component, indicating that the enzymatic
 193 degradation result of the P(3HB-*co*-4HB) was not attributed to the change of the
 194 crystallization. On the other hand, enzymatic hydrolysis started on the surface and at physical
 195 lesions on the polymer and proceeded to the inner part of the material.²² The addition of the
 196 PLLA increased the specific surface area of the P(3HB-*co*-4HB). According to Han et al.,²⁵
 197 the PLLA and P(3HB-*co*-4HB) were immiscible, therefore, the diffusion rate of the enzymatic

198 molecules into the interface between the two phase could be improved. Thus the
199 biodegradation rate of P(3HB-*co*-4HB) could be increased. The normalized weight loss of the
200 films with 10 wt%, 20 wt%, and 30 wt% content of PLLA increased to over 100 wt% for
201 degradation times longer than 4d. This may be explained by the release of PLLA into the
202 media with the removal of P(3HB-*co*-4HB) molecules. In other words, the dispersed PLLA
203 molecules contained in the continuous P(3HB-*co*-4HB)-rich phase may be liberated by the
204 enzymatic hydrolysis and removal of P(3HB-*co*-4HB). For the samples with content of 60
205 wt%, 80 wt% PLLA, the degradation rate was decreased to a very low value. This may be
206 ascribed to that with the increase of the PLLA content, the P(3HB-*co*-4HB) was trapped in the
207 continuous of PLLA domain, and the diffusion of the *Pseudomonas mendocina* lipase into the
208 interfaces of the two polymer phases was hindered, thus the P(3HB-*co*-4HB) could not get in
209 touch with the enzymatic molecules sufficiently.

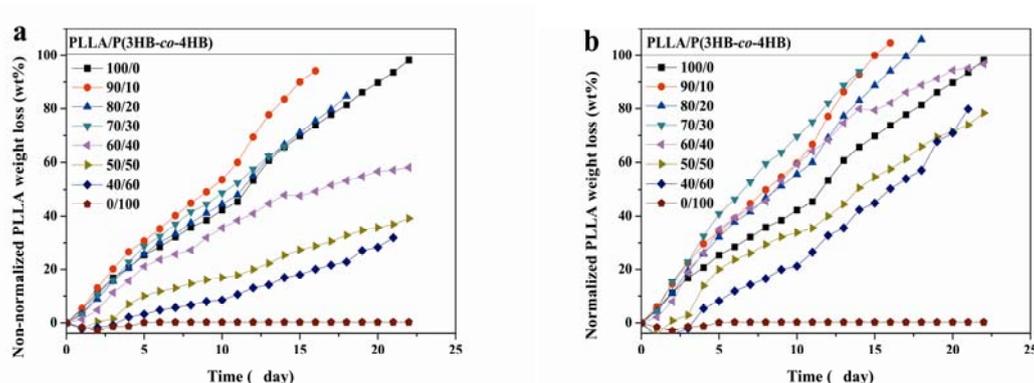


210
211 **Figure 3** The time of half P(3HB-*co*-4HB) weight loss profiles of the PLLA/P(3HB-*co*-4HB)
212 blend films versus the content of the PLLA during the *Pseudomonas mendocina*
213 lipase-catalyzed degradation.

214 Time of half weight loss is an important parameter to determine and compare degradation

215 behaviors. **Figure 3** shows the time of half P(3HB-*co*-4HB) weight loss profiles of the films
 216 versus the content of the PLLA. It showed that the time of half weight loss of the
 217 P(3HB-*co*-4HB) decreased with increasing the content of the PLLA, indicating the
 218 accelerated enzymatic degradation of the P(3HB-*co*-4HB). While there was an optimum
 219 PLLA content of 30 wt% for the enzymatic degradation of P(3HB-*co*-4HB), after this peak,
 220 the enzymatic degradation of P(3HB-*co*-4HB) began to decline with increasing the content of
 221 the PLLA due to the wraparound effect of the PLLA.

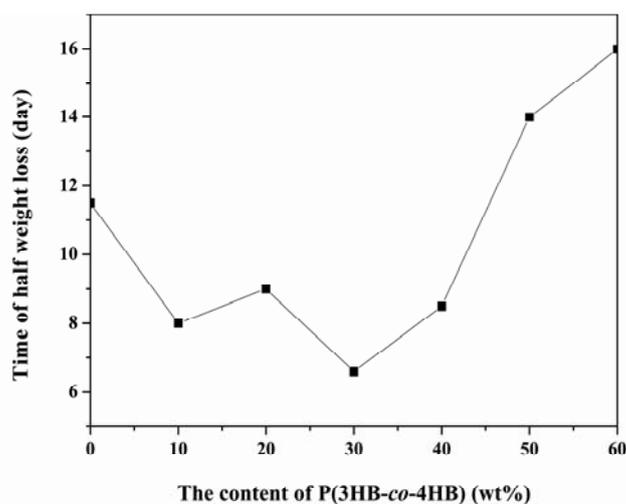
222 3.1.2 Proteinase K-Catalyzed Degradation of PLLA Component



223
 224 **Figure 4** The non-normalized (a) and normalized (b) PLLA weight loss profiles of
 225 PLLA/P(3HB-*co*-4HB) blend films as a function of time during the proteinase K-catalyzed
 226 degradation.

227 **Figure 4** shows the non-normalized and normalized weight loss profiles of
 228 PLLA/P(3HB-*co*-4HB) blend films as a function of time during the proteinase K-catalyzed
 229 degradation. The rate of enzymatic degradation could be determined from the slope of the
 230 weight loss against time. As evident from Figure 4, for all the samples, the weight losses of
 231 the films increased with time due to the enzymatic degradation of the PLLA except for the
 232 neat P(3HB-*co*-4HB) which almost had no weight loss. It confirmed that the proteinase K can

233 only degrade PLLA, but not degrade P(3HB-*co*-4HB). For the blends with 10 wt%, 20 wt%,
234 30 wt% content of P(3HB-*co*-4HB) that the rates of enzymatic degradation were fast even at
235 the initial stage whether they were normalized or not. **Figure 5** shows the time of half PLLA
236 weight loss profiles of the films versus the content of the P(3HB-*co*-4HB) during the
237 proteinase K-catalyzed degradation.

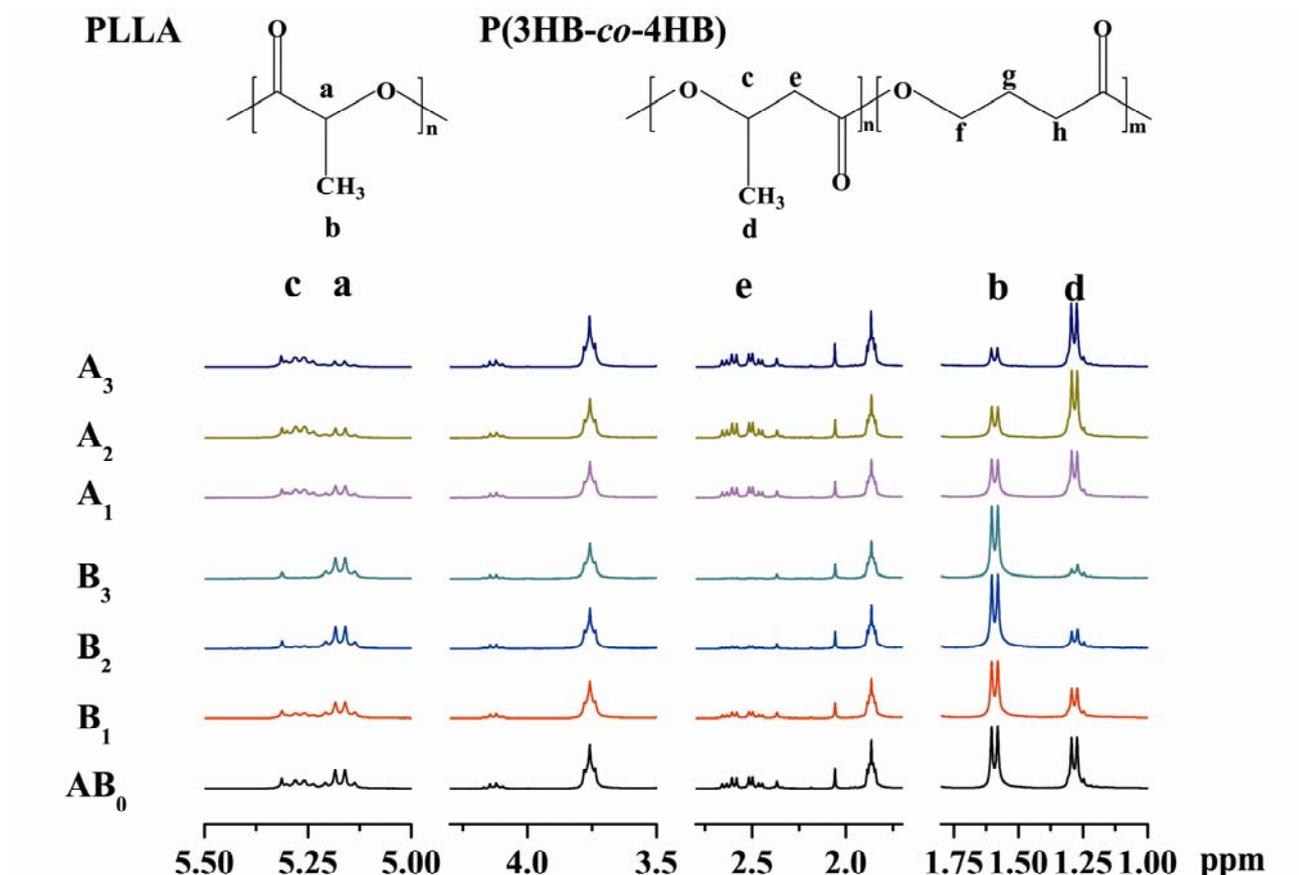


238
239 **Figure 5** The time of half PLLA weight loss profiles of the PLLA/P(3HB-*co*-4HB) blends
240 films versus the content of the P(3HB-*co*-4HB) during the proteinase K-catalyzed
241 degradation.

242 As is shown in Figure 5, The time of half PLLA weight loss decreased with the addition of
243 the P(3HB-*co*-4HB) when the P(3HB-*co*-4HB) content was lower than 40 wt%, indicating the
244 addition of P(3HB-*co*-4HB) accelerated the degradation of PLLA by proteinase K. An
245 optimum P(3HB-*co*-4HB) content for the enzymatic degradation of PLLA was 30 wt%. When
246 the P(3HB-*co*-4HB) content was more than 40 wt%, the enzymatic degradation of PLLA was
247 restricted. As mentioned above, the rate of enzymatic hydrolysis of biodegradable materials
248 was dependent on the degree of crystallinity^{18, 21, 22} and the surface area of the polymer^{18, 19}
249 exposed to enzymatic hydrolysis. From Figure 2, it should be noted that there was an obvious

250 cold crystallization peak for the PLLA in neat PLLA and its blends. In addition, the cold
251 crystallization enthalpy was very similar with the melting enthalpy of the PLLA for each
252 sample, which indicated that the PLLA was primarily amorphous and the enzymatic
253 degradation result of the PLLA was not influenced by its crystallization. Since the
254 degradation of PLLA was mainly started with a hydrolysis, the water absorption was very
255 important in the degradation process. The two-phase nature of the blends due to the
256 immiscibility resulted in a higher water absorption and proteinase K diffusion from the
257 surface to the inside of the blends than that of neat PLLA, thus led to the acceleration of
258 PLLA hydrolysis there, thereby accelerating degradation.^{7, 25, 29, 30} In other words, the
259 degradable surface areas of PLLA domains per unit mass were increased by phase separation
260 and particulate domain formation. A counter-example shows that the addition of
261 poly[(L-lactide)-*co*-(ϵ -caprolactone)] as a compatibilizer between PLLA and PCL reduced
262 proteinase K-catalyzed enzymatic degradation rate due to the decreased hydrolysable
263 interfacial area between the PLLA-rich and PCL-rich phase.³¹ That is to say that the
264 decreasing voids and cavities between the phase interface bring down the proportion of the
265 entrance of the proteinase K then reduce the enzymatic degradation rate of PLLA. The
266 mechanism of the accelerated enzymatic degradation rate of PLLA is the same as that of
267 P(3HB-*co*-4HB). When the content of P(3HB-*co*-4HB) increased to 50 wt% and 60 wt%, the
268 rate of enzymatic degradation were decreased compared to the neat PLLA. This may be due
269 to that the PLLA was contained by P(3HB-*co*-4HB)-rich phase and cannot get in touch with
270 the enzymatic molecules sufficiently. The normalized weight loss of the films with 10 wt%,
271 20 wt% content of P(3HB-*co*-4HB) increased to over 100 wt% for hydrolysis times longer
272 than 15d. This also can be explained by the release of P(3HB-*co*-4HB) into the hydrolysis
273 media with the removal of PLLA molecules just as the enzymatic degradation of the
274 P(3HB-*co*-4HB).

275 3.2 The Composition Changes of PLLA/P(3HB-co-4HB) Blends characterized by NMR



276

277 **Figure 6** Selected regions of the ^1H NMR spectra of PLLA/P(3HB-co-4HB) blends films
 278 (AB₀) before and after *Pseudomonas mendocina* lipase catalyzed degradation of
 279 P(3HB-co-4HB) for (B₁) 1day; (B₂) 2day; (B₃) 9day and proteinase K catalyzed degradation
 280 of PLLA for (A₁) 5day; (A₂) 10day; (A₃) 22day.

281 The ^1H NMR experiment of the remaining PLLA/P(3HB-co-4HB) blend films with 50 wt%
 282 content of P(3HB-co-4HB) before and after specific times of degradation by *Pseudomonas*
 283 *mendocina* lipase and proteinase K, respectively, was carried out and **Figure 6** depicted the
 284 most informative regions of the spectra (which are most important for determination of
 285 chemical structure). In Figure 6, AB₀ shows the ^1H NMR spectrum of PLLA/P(3HB-co-4HB)
 286 blend films before enzymatic degradation, and the selected regions of ^1H NMR spectrum of

287 the investigated samples corresponding to the occurrence of signals ascribed to PLLA ($\delta =$
288 5.16, 1.58 ppm) and P(3HB-*co*-4HB) ($\delta = 5.27, 2.51, 1.28$ ppm). The integral area of the
289 characteristic peaks was almost 1:1 at $\delta = 1.58$ ppm and $\delta = 1.28$ ppm corresponding to the
290 methyl group protons of PLLA and P(3HB-*co*-4HB), respectively, indicating the composition
291 ratio of PLLA and P(3HB-*co*-4HB) was 1:1. When the blend film was degraded by
292 *Pseudomonas mendocina* lipase for 1 day, the intensities of the signals characteristic for the
293 methyl, methylene, methine ($\delta = 1.28, 2.51, 5.27$ ppm) of the P(3HB-*co*-4HB) were decreased
294 while the characteristic peak corresponding to the PLLA ($\delta = 5.16, 1.58$ ppm) remained
295 unchanged. By comparing the integral area of the characteristic peaks between the $\delta = 1.58$
296 ppm and $\delta = 1.28$ ppm, the content of the P(3HB-*co*-4HB) in the remained films could be
297 calculated as 36%, almost matching with the non-normalized weight loss of 18% for the blend
298 films with the P(3HB-*co*-4HB) content of 50 wt% degraded for 1 day. When the degradation
299 was proceeded for 9 day, the signals for P(3HB-*co*-4HB) were almost vanished from sight,
300 implied that the P(3HB-*co*-4HB) could be removed efficiently from the PLLA.
301 Correspondingly, after the proteinase K catalyzed degradation, the characteristic peaks
302 corresponding to the PLLA ($\delta = 5.16, 1.58$ ppm) decreased little by little while those of the
303 P(3HB-*co*-4HB) ($\delta = 1.28, 2.51, 5.27$ ppm) kept constant. At the end of the degradation, the
304 signals of PLLA almost disappeared. The composition changes in the PLLA and
305 P(3HB-*co*-4HB) could be matched well with the corresponding weight loss of the
306 PLLA/P(3HB-*co*-4HB) films after enzymatic degradation.

307 It should be emphasized that there were no new signals appeared during the process of the
308 enzymatic degradation either by the *Pseudomonas mendocina* lipase or by proteinase K
309 compared to the ^1H NMR spectrum before enzymatic degradation, which indicated that there
310 was no other form of materials fabricated. In other words, there were no intermediate products
311 formed during the enzymatic degradation process for both the PLLA and P(3HB-*co*-4HB).

312 According to Albertsson et al.,³² the final degradation products of the pure PLLA catalyzed
 313 by proteinase K were water-soluble lactic acid and its oligomers. For the *Pseudomonas*
 314 *mendocina* lipase-catalyzed degradation of the P(3HB-co-4HB),³³ the degradation products
 315 were a mixture of oligomers and finally were metabolized by the *Pseudomonas mendocina* to
 316 CO₂ and H₂O. This may be attributed to the surface erosion mechanisms of the enzymatic
 317 degradation for both the PLLA and P(3HB-co-4HB). In short, only the surface of the samples
 318 was eroded, the macromolecular chains were cut into the corresponding soluble oligomers
 319 immediately while the internal remained almost unchanged during the enzymatic degradation
 320 process, which could also be confirmed by the results of the GPC (shown as follows).

321 3.3 Molecular Weight Change of PLLA/P(3HB-co-4HB) Blends before and after 322 Enzymatic Degradation

323 **Table 2.** Molecular weights and dispersity index of PLLA/P(3HB-co-4HB) (50/50 wt/wt)
 324 before and after enzymatic degradation

| PLLA/P(3HB-co-4HB) (50/50 wt/wt) | | $M_n/10^5$ (g mol ⁻¹) | $M_w/10^5$ (g mol ⁻¹) | M_w/M_n |
|--|--------|--------------------------------------|--------------------------------------|-----------|
| Before enzymatic degradation | | 0.61 | 1.18 | 1.95 |
| After <i>Pseudomonas mendocina</i> lipase catalyzed degradation | 1 day | 0.87 | 1.32 | 1.52 |
| | 2 day | 0.89 | 1.36 | 1.53 |
| | 9 day | 0.97 | 1.47 | 1.51 |
| After Proteinase K catalyzed degradation | 5 day | 0.74 | 1.21 | 1.64 |
| | 10 day | 0.79 | 1.33 | 1.69 |
| | 22 day | 0.71 | 1.13 | 1.58 |

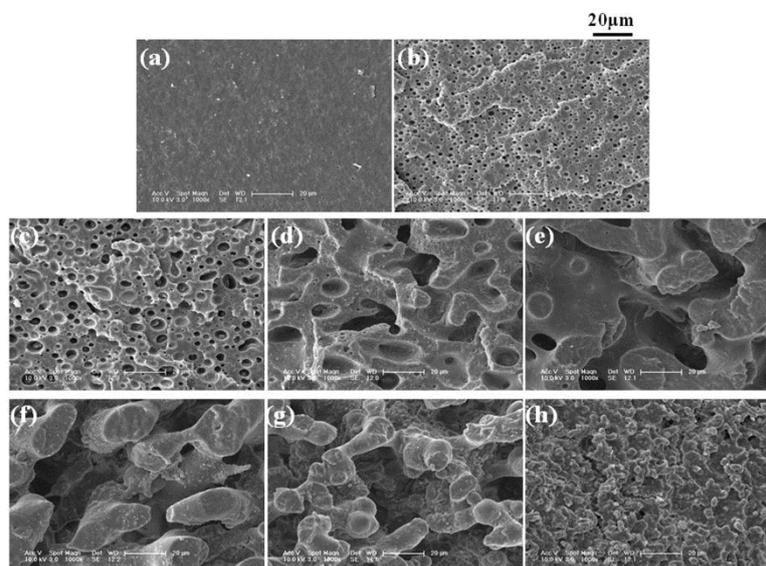
325 The GPC traces of the remaining PLLA/P(3HB-co-4HB) blend films with 50 wt% content
 326 of PLLA after specific times of degradation by *Pseudomonas mendocina* lipase and proteinase
 327 K, respectively, were performed and the detailed data of M_w , M_n and M_w/M_n before and after
 328 enzymatic degradation are summarized in **Table 2**. The results revealed that there was only
 329 one retention volume peak (not shown here) before enzymatic degradation for the
 330 PLLA/P(3HB-co-4HB) (50/50 wt/wt) blend films and the value of M_w was around 1.18×10^5 g
 331 mol⁻¹ shown as Table 2. It should be noted that the results of GPC analyses presented in Table

332 2 were calculated for the blend which contained two components, PLLA and P(3HB-*co*-4HB).
333 Thus in Table 2, the presented molecular mass and dispersity values were apparent. The only
334 one retention volume peak may be due to that the PLLA and P(3HB-*co*-4HB) suffered from
335 certain thermal degradation after the blending and molding process and the molecular weight
336 of them became too close to distinct from each other. What's more, the PLLA/P(3HB-*co*-4HB)
337 blend films showed almost the same molecular weight and molecular weight distributions
338 before and after either the *Pseudomonas mendocina* lipase or proteinase K catalyzed
339 degradation shown as Table 2. The M_w of all the PLLA/P(3HB-*co*-4HB) blend films remained
340 almost unchanged after the enzymatic degradation either by the *Pseudomonas mendocina*
341 lipase or proteinase K catalyzed, implying no accumulation of the low-molecular weight
342 enzymatic degraded components. This was accordance with the results of the ^1H NMR and
343 further confirmed the surface erosion mechanisms of the enzymatic degradation for both the
344 PLLA and P(3HB-*co*-4HB). At the same time, the M_w/M_n list in Table 2 was 1.95 before
345 enzymatic degradation and had a little decrease both after the *Pseudomonas mendocina* lipase
346 and proteinase K catalyzed degradation, which may be due to that with the process of the
347 degradation, one of the component was removed gradually and there was almost only one
348 component after the enzymatic degradation eventually, therefore the molecular weight
349 distribution became narrower.

350 3.4 Porous Morphology of PLLA/P(3HB-*co*-4HB) Blends after Degradation

351 The previous section investigated the influence of one component on the biodegradation
352 properties of the other component and the results revealed that the enzymatic degradation
353 process of the *Pseudomonas mendocina* lipase and the proteinase K for the blends with
354 different ratio of PLLA/P(3HB-*co*-4HB) could be finished completely within 9 and 22 days,
355 respectively. Therefore, selective enzymatic degradation for the biodegradation of the

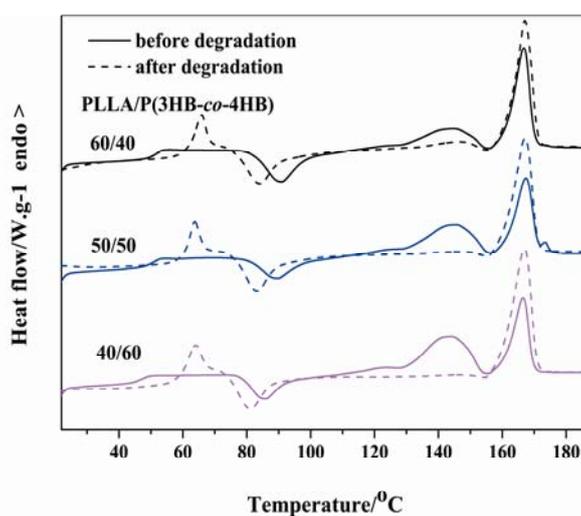
356 P(3HB-co-4HB) was proceeded for 9 days, and the biodegradation of the PLLA was 22 days
357 to confirm the complete removal of the corresponding component thus to acquire the certain
358 porous structure.



359
360 **Figure 7** The porous PLLA with selectively removal of the P(3HB-co-4HB), the content of
361 the P(3HB-co-4HB): (a) 10 wt%; (b) 20 wt%; (c) 30 wt%; (d) 40 wt%; (e) 50 wt%; (f) 60
362 wt%; (g) 70 wt%; (h) 80 wt%.

363 **Figure 7** presents the porous PLLA with selectively removal of the P(3HB-co-4HB) by
364 *Pseudomonas mendocina* after selective enzymatic degradation for 9 days, and the contents of
365 the P(3HB-co-4HB) were (a) 10 wt%; (b) 20 wt%; (c) 30 wt%; (d) 40 wt%; (e) 50 wt%; (f) 60
366 wt%; (g) 70 wt%; (h) 80 wt%, respectively. The size of the porous PLLA obtained was listed
367 in Table 1. Because *Pseudomonas mendocina* can only degraded P(3HB-co-4HB), but not
368 degraded PLLA, it can be concluded that the black pores in the cryo-fractured surfaces of the
369 blends must be formed by the removal of P(3HB-co-4HB) and the pore morphology should
370 agree with that of P(3HB-co-4HB) in the initial blends. As is shown in Figure 7, when the
371 content of the P(3HB-co-4HB) was 10 wt%, the PLLA had a pore of $0.39 \pm 0.15 \mu\text{m}$ and the
372 pore in the PLLA become more and more bigger with the increasing content of the

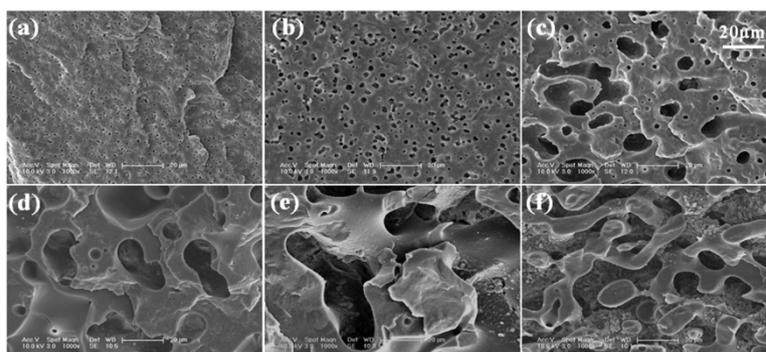
373 P(3HB-*co*-4HB). Interestingly, when the content of the P(3HB-*co*-4HB) increased to 40~60
374 wt%, there were interconnected pores appeared due to the co-continuous phase formed. These
375 samples that have interconnected pores are biodegradable, biocompatible and innocuous,
376 which will be appropriate for the application of tissue engineering. Further increasing the
377 content of the P(3HB-*co*-4HB), the blends evolved into the sea-island structure, and the PLLA
378 phase became the “island” of sea-island structure. When the content of the P(3HB-*co*-4HB)
379 increased to 80 wt%, the remainder PLLA phase were particles therefore the sample could not
380 keep its integrity.



381
382 **Figure 8** The DSC thermograms of the cryo-fractured surfaces of the PLLA/P(3HB-*co*-4HB)
383 blends with the content of 40 wt%, 50 wt%, 60 wt% of the PLLA before and after
384 *Pseudomonas mendocina* lipase-catalyzed degradation of P(3HB-*co*-4HB).

385 In order to confirm that the P(3HB-*co*-4HB) in the porous PLLA was removed completely,
386 the differential scanning calorimeter (DSC) was used to study the thermal properties of the
387 blends after enzymatic degradation. **Figure 8** shows the DSC thermograms of the
388 cryo-fractured surfaces of the PLLA/P(3HB-*co*-4HB) blends with the content of 40 wt%, 50
389 wt%, 60 wt% of the PLLA before and after *Pseudomonas mendocina* lipase-catalyzed

390 degradation of P(3HB-*co*-4HB). It can be found that there was only the T_m peak of the PLLA
391 for the blends containing 40 wt%, 50 wt%, 60 wt% PLLA after enzymatic degradation, this
392 confirmed that the P(3HB-*co*-4HB) was removed completely for the samples by selective
393 enzymatic degradation. It provided an accessible method for the fabrication of porous
394 polymers through selective biodegradation.



395
396 **Figure 9** The porous P(3HB-*co*-4HB) with selectively removal of the PLLA, the content of
397 the PLLA: (a) 10 wt%; (b) 20 wt%; (c) 30 wt%; (d) 40 wt%; (e) 50 wt%; (f) 60 wt%.

398 **Figure 9** shows the corresponding porous P(3HB-*co*-4HB) with selectively removal of the
399 PLLA, the contents of the PLLA were (a) 10 wt%; (b) 20 wt%; (c) 30 wt%; (d) 40 wt%; (e)
400 50 wt%; (f) 60 wt%; respectively. There was a similar phase inversion in the P(3HB-*co*-4HB).
401 When the content of PLLA was as low as 40 wt%, the holes of the porous P(3HB-*co*-4HB)
402 became bigger and bigger with the increasing content of PLLA. When the content of the
403 PLLA increased to 40~60 wt%, there was a co-continuous phase appeared. However, the
404 morphology of the PLLA in the P(3HB-*co*-4HB) was slightly different from that of
405 P(3HB-*co*-4HB) in PLLA. When the content of the PLLA was 10 wt%, 20 wt%, and 30 wt%,
406 the size of the porous P(3HB-*co*-4HB) was about 0.61, 1.73, and 3.14 μm , respectively. While
407 when the content of the P(3HB-*co*-4HB) was 10 wt%, 20 wt%, and 30 wt%, the size of the
408 porous PLLA was about 0.39, 1.08, and 2.65 μm , respectively. The size of the porous
409 P(3HB-*co*-4HB) was slightly bigger than that of the porous PLLA. That meant the

410 P(3HB-*co*-4HB) was preferentially distributed in the PLLA. When the content of the
411 P(3HB-*co*-4HB) were 40~60 wt%, the interconnected porous structure could be acquired
412 through the selective enzymatic degradation and the morphology can be different due to the
413 different content of P(3HB-*co*-4HB).

414 Accordingly, we can acquire the tunable porous materials with well-defined pore sizes,
415 controlled pore size distributions, and interconnectivities through changing the composition
416 and the content of the biodegradable component in the blends by selective biodegradation.
417 The porous materials obtained are biodegradable, biocompatible and innocuous, which is
418 expected to apply in tissue engineering area in future.

419 **4. Conclusions**

420 Fully biodegradable polymer blends of PLLA and P(3HB-*co*-4HB) were prepared by melt
421 compounding. The enzymatic degradation and porous morphology of PLLA/P(3HB-*co*-4HB)
422 blends were investigated in detail. The enzymatic degradation rate of P(3HB-*co*-4HB) was
423 accelerated due to the addition of the PLLA which increased the specific surface area of the
424 P(3HB-*co*-4HB), and the blends containing 30 wt%, 40 wt%, 50 wt% of PLLA had the most
425 fast enzymatic degradation rate. For the samples with the content of 60 wt%, 80 wt% PLLA,
426 the degradation rates were decreased due to the package effect of PLLA-rich phase on the
427 P(3HB-*co*-4HB). For the enzymatic degradation of PLLA, the degradation rate was also
428 improved due to the similar effect. The composition changes in the PLLA and P(3HB-*co*-4HB)
429 characterized by ^1H NMR could be matched well with the corresponding weight loss of the
430 PLLA/P(3HB-*co*-4HB) films after the enzymatic degradation either by the *Pseudomonas*
431 *mendocina* lipase or proteinase K catalyzed. The results of the ^1H NMR and GPC indicated
432 that there were no intermediate products formed during the enzymatic degradation both of the
433 PLLA and P(3HB-*co*-4HB), which confirmed the surface erosion mechanisms for the

434 degradation of the PLLA and P(3HB-*co*-4HB) catalyzed by the proteinase K and
435 *Pseudomonas mendocina* lipase, respectively. The selective enzymatic degradation results
436 revealed that when the P(3HB-*co*-4HB) component was degraded, the pore in the PLLA
437 became more and more bigger with increasing the P(3HB-*co*-4HB) content, and it became
438 interconnected pores when the content of the P(3HB-*co*-4HB) increased to 40~60 wt%. There
439 was a similar phase inversion in the porous P(3HB-*co*-4HB). However, it should be noted that
440 the size of the porous PLLA was slightly smaller than that of the porous P(3HB-*co*-4HB),
441 indicating the P(3HB-*co*-4HB) was preferentially distributed in the PLLA. Consequently, the
442 porous materials could be obtained through the selective biodegradation method and the
443 morphology of the pore can be controlled through changing the composition and the content
444 of the biodegradable component in the blends. The porous materials fabricated have tunable
445 mechanical properties, well-defined pore sizes, and controlled pore size distributions and
446 interconnectivities. Since both of the materials for blending are environmentally friendly, the
447 porous materials obtained are biodegradable, biocompatible and innocuous whether the
448 second component is removed completely or not, which are suitable for the application of
449 tissue engineering. Such studies may be of great interest and importance for the development
450 of scaffolds for regeneration of tissues from biodegradable polymers.

451 **Acknowledgements**

452 The authors are grateful for the support from the National Natural Science Foundation of
453 China (51021003).

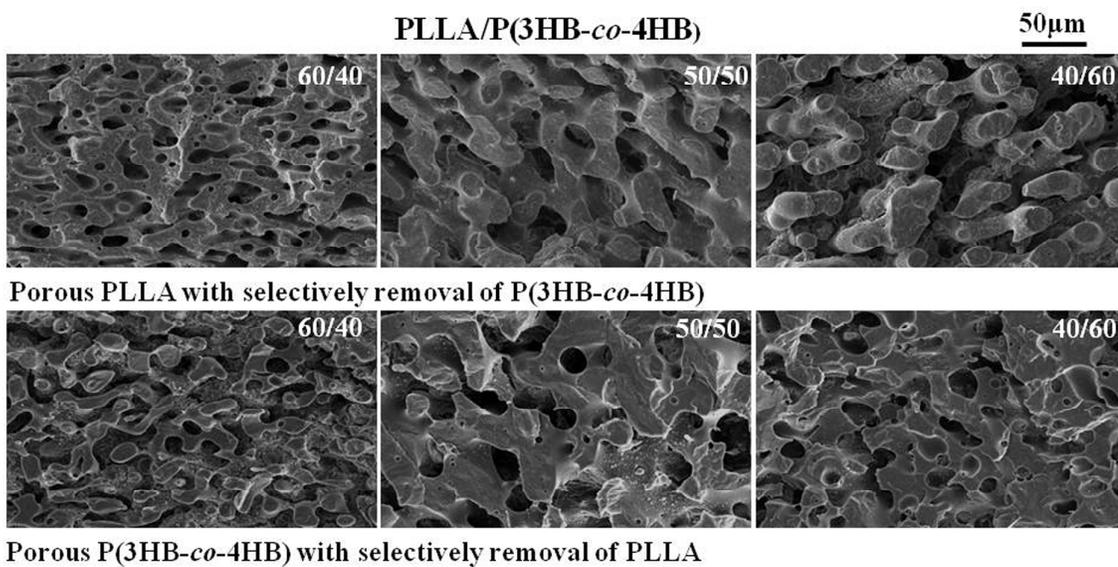
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Table of Contents Graphic



Due to the specificity of the degradation enzymes, the porous PLLA and the porous P(3HB-*co*-4HB) could be acquired by selectively removal of the P(3HB-*co*-4HB) component and the PLLA component, respectively.