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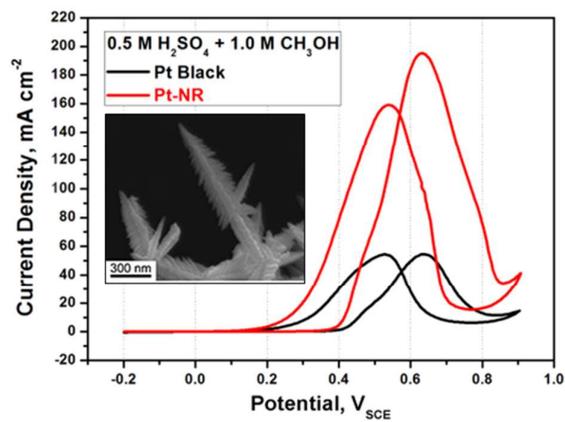
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The pulse-mode potentiostatic electrodeposited single-crystal fern-like Pt nanorods enhanced the electrocatalytic activities for the methanol oxidation reaction.



COMMUNICATION

Enhanced electrocatalytic activities of pulse-mode potentiostatic electrodeposited single-crystal, fern-like Pt nanorods †

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Huei-Yu Chou,^a Ming-Chi Tsai,^a Sung-Yen Wei,^b Yu-Hsuan Wei,^{ad} Tsung-Kuang Yeh,^a Fu-Rong Chen,^a Chen-Chi M. Ma,^c Chuen-Hong Tsai^{*a} and Chien-Kuo Hsieh^{*d}

Single-crystal and fern-like Pt nanorods (Pt-NR) were electrodeposited on carbon paper. The electrocatalytic activity of prepared Pt-NR electrodes, as determined by the catalyst mass activity (MA) and specific activity (SA) for the methanol oxidation reaction, are respectively 4.59 times and 10.54 times better than that of a commercial Pt-black catalyst.

Platinum (Pt), which is a very important material for many applications, syntheses of size- and shape-controlled Pt or Pt-embedded nanostructures are among today's most important research topics. Because of their excellent physical and chemical characteristics, they have been widely used as catalysts in many areas such as fine chemical synthesis, hydrogen production, gas sensors and electrocatalysis.^{1,2}

Among the important applications, Pt is the key catalyst in proton exchange membrane (PEM) fuel cells.³ The core component in such a fuel cell is the membrane electrode assembly (MEA), which generally comprises a thin, flat proton exchange membrane with catalyzed electrodes bonded to both sides. Conventionally, the electrodes are made of electronconducting carbon cloth or carbon paper (CP) plated with a mixture of carbon-black-supported catalyst particles and a proton-conducting ionomer. For a PEM fuel cell to operate efficiently, the catalyst must be both highly dispersed and nanoscale in size so as to optimize the rate of hydrogen or methanol oxidation reaction at the anode. Supporting evidence for these requirements is that the activity of a nanocatalyst is known to correlate strongly with its morphology and crystallinity.^{4,5}

For the electrocatalyst in a fuel cell to exhibit high efficiency, three qualities are required: (1) large specific area, (2) three phase contact (including electronic conductivity through the catalyst support), and (3) high durability during and after MEA assembling. In our previous study,⁶ we demonstrated that carbon nanotubes (CNT) grown directly on carbon cloth could possess high specific surface area and display uniform catalyst (Pt or its alloy) dispersion, and therefore exhibit high catalytic efficiency. However, damage to (and indeed, collapse of) the catalytic CNT layer is likely to occur during the MEA assembly, suggesting the importance of an alternative assembly method.⁷

It has been found that Pt has distinct electrocatalytic activity on different crystalline planes,⁸ suggesting the maximum

electrocatalytic activity may be achieved by controlling single crystallinity in fabricating the catalysts. To date, most relevant research has focused on syntheses of Pt nanocrystals with specific crystallinities,⁹ mainly polycrystalline structure.^{1,10} For example, tetrahedral Pt nanocrystals with high-index facets have been synthesized by electrochemical approaches.⁹ Very recently, however, some single-crystal Pt structures with unique anisotropic structures, such as one-dimensional nanowires and nanorods, have been reported and have attracted considerable attention because of their attractive surface properties and superior catalytic efficiencies.¹¹⁻¹³ For example, Xia and coworkers generated single-crystal Pt nanowires by a polyol process combined with the introduction of a trace amount of ionic iron species with PVP (used as a surfactant).¹¹ Song and coworkers produced Pt nanowires networks by chemical reduction using a soft template.¹² Sun and coworkers synthesized Pt nanowires and/or nanoflowers by aging hexachloroplatinic acid in a formic acid aqueous solution for several hours to 3 days under ambient conditions.^{13,14}

A challenge to synthesize single-crystal Pt nanowires or nanorods by surfactant- and template-free approaches under mild conditions still remains. Another key challenge is to increase catalyst dispersion and electronic conductivity between the gas diffusion layer and the catalysts themselves. In this communication, we address both these challenges by proposing a novel pulse-mode potentiostatic electrodeposition (PPE) technique using a simple aqueous electrolyte to directly synthesize single-crystal, fern-like Pt nanorods (Pt-NR) on a CP surface. Then, we examine the electrocatalytic activity of such a system for the oxidation of formic acid (CH₂O₂), formaldehyde (HCHO), and methanol (CH₃OH).

To control the Pt-NR crystal structure during synthesis, we investigated the effect of non-hydrogen-bubble formation during electrodeposition. An applied potential of less than $-0.241 V_{SCE}$ during electrodeposition is known to cause the generation of hydrogen bubbles by the standard hydrogen reduction reaction (SHRR).¹⁵ Such hydrogen-bubble formation apparently alters the morphologies of Pt that has been reduced on the CP surface, in our case, causing the reduced Pt to assume a petal-like structure.

We used the PPE technique by applying periodic cycles of high positive potential ($1.2 V_{SCE}$), followed by the negative potential ($-0.24 V_{SCE}$), slightly higher than the value known to induce the SHRR. As expected, hydrogen bubbles were not generated during

electrodeposition, and Pt-NR was electrodeposited on the CP surface. Obviously, then, as has previously been reported, hydrogen-bubble adsorption on the electrode surface in an aqueous sulphuric electrolyte during deposition can affect the final morphology of the reduced Pt metal.¹⁶

Figure 1(a) shows a typical SEM micrograph of the fern-like Pt-NR so created. Each Pt-NR is approximately 300 nm to 1 μm long. Figure 1(b) shows a TEM micrograph of the Pt-NR. Their length is approximately 600 nm, consistent with that determined by SEM observation. Figures 1(c) and (d) show HR-TEM micrographs of the regions indicated by circles in Fig. 1(b). The Pt-NR end and branch both possess a single-crystal structure. The insets in Figs. 1(c) and (d) show respectively the selected-area electron diffraction (SAED) patterns at corresponding region, both these SAED from [001] electron beam zone axis show the typical F.C.C. diffraction, not only the end of stem and branch but also whole Pt-NR show the same diffraction pattern, which indicates that the single crystallinity of the fern-like Pt-NR.

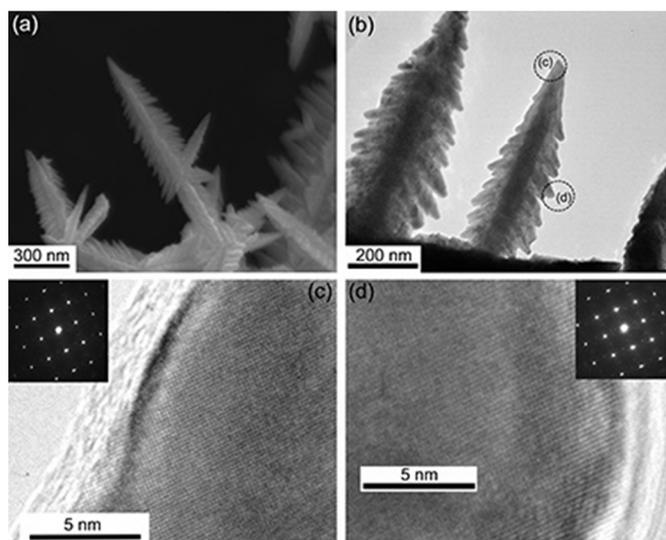


Fig. 1 (a) SEM micrograph of Pt-NR electrodeposited on carbon paper; (b) typical TEM micrograph of prepared Pt-NR; (c) HR-TEM micrograph of the tip region circled in Fig. 1(b) and the related SAED pattern; (d) HR-TEM image of the branch region circled in Fig. 1(b) and the related SAED pattern.

Thus, the PPE technique can be used to synthesize the single-crystal Pt-NR. We herein suggest the associated mechanism is as follows. During the application of $-0.24 V_{\text{SCE}}$, reaction S1 and S2 occur (ESI^\dagger). Moreover, during the application of $1.2 V_{\text{SCE}}$, other reactions may occur, as previously suggested.⁹ Specifically, reduced Pt may react with H_2O to form the adsorption product $\text{Pt}(\text{O})_{\text{ad}}$ (reaction S4 in ESI^\dagger), which in turn, again reacts with H_2O to form PtO_2 (reaction S5 in ESI^\dagger).¹⁷⁻¹⁹ During the next application of $-0.24 V_{\text{SCE}}$, these resultants then form pure Pt by reactions (S1), (S2), and (S6)–(8) in ESI^\dagger . In other words, during PPE, the nucleation/growth of Pt atoms and subsequent structural modifications on the Pt surface occur continuously. It has been shown that the PPE technique without the influence of hydrogen bubbles enables the synthesis of fern-like single crystals in an aqueous solution containing sulphuric acid and Pt precursor salt.

To determine the catalytic efficiency of the as-prepared Pt-NR catalyst and compare it with that of a commercial Pt-black catalyst for the reaction of methanol oxidation, we performed cyclic voltammetry (CV) measurements in an Ar-saturated $0.5 \text{ M H}_2\text{SO}_4$ + (a) $1 \text{ M CH}_3\text{OH}$, (b) $0.1 \text{ M CH}_2\text{O}_2$, and (c) 0.1 M HCHO in aqueous solution, respectively.

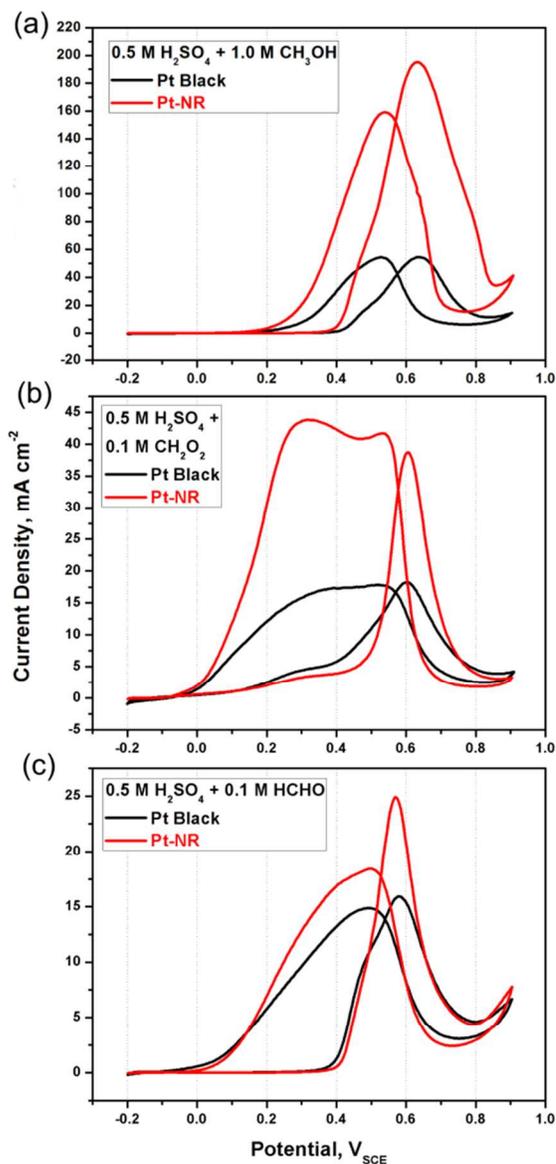


Fig. 2 Cyclic voltammograms of the Pt-NR catalyst (red curve) and Pt-black catalyst (black curve) tested in Ar-saturated $0.5 \text{ M H}_2\text{SO}_4$ for the oxidation reactions of (a) $1 \text{ M CH}_3\text{OH}$, (b) $0.1 \text{ M CH}_2\text{O}_2$, and (c) 0.1 M HCHO in aqueous solutions. The scan rate was 20 mV s^{-1} . The test temperature was controlled at 30°C .

Figure 2(a) shows the voltammograms from fifth cycle for each catalyst. Table 1 lists the following electrochemical characteristics derived for each catalyst: forward peak potential (V_p), forward peak current density (i_f), ratio of i_f to reverse peak density (i_f/i_b), and mass activity (MA), the peak current density of methanol oxidation obtained from cyclic voltammogram per unit of Pt loading mass). Compared with Pt black, Pt-NR exhibits a distinctly higher i_f value for the methanol oxidation reaction. The peaks that appeared during the reverse scans signify the desorption of intermediate products generated by methanol oxidation during the forward scans. It has been suggested that these intermediates might be associated with the adsorption of CO (an intermediate during methanol oxidation) on the Pt surfaces, a process that has been termed “CO poisoning.”^{14, 20} Consequently, the i_f/i_b value can be considered to represent the CO tolerance of the catalysts. From Table 1, for Pt-NR, we see that the i_f

i_b value is significantly better (23%) than that for Pt black, suggesting more efficient CO desorption on Pt-NR. From Fig. 2(a) and Table 1, we see that the V_p value of Pt-NR (0.63 V_{SCE}) is lower than that of Pt black (0.64 V_{SCE}), indicating that Pt-NR exhibits superior electrocatalytic activity. In addition, at a given potential of 0.5 V_{SCE} , i_p of Pt-NR (76.1 mA cm^{-2}) is 3.59 times better than that of Pt black (21.2 mA cm^{-2}) of Pt black catalysts. Similarly, at a given current density of 40 mA cm^{-2} , the potential of Pt-NR (0.45 V_{SCE}) is 120 mV lower than that of Pt black (0.57 V_{SCE}), resulting in lower overpotential during methanol oxidation.

Thus, compared with Pt black, Pt-NR exhibits not only a higher i_p/i_b value, but also a higher i_f value, indicating its considerably superior catalytic efficiency for methanol oxidation reaction. To verify this superior catalytic efficiency, we also examined the performance of both catalysts for CH_2O_2 and HCHO oxidation. From Fig. 2(b) for CH_2O_2 , we see the following: (1) At 0.2 V_{SCE} , the oxidation reaction begins, resulting in the decomposition of CH_2O_2 to CHOO or COOH.⁴ (2) At 0.6 V_{SCE} , forward peaks appear, caused by the formation of CO_2 with i_f of 40 mA cm^{-2} on the Pt surface. (3) During the reverse scan from 0.65 to near 0 V_{SCE} , peaks appear, caused by the desorption of CO_2 gas from the Pt surfaces (desorption bubbles can be clear seen during the experiments). The i_f value of Pt-NR is higher than that of Pt black, suggesting a higher rate of catalytic oxidation of formic acid and intermediates.^{21, 22} Similar results have been found that Pt-NR again has a higher i_f value (decomposition of HCHO forms HCH or HCO and further formation of CO on Pt surfaces at about 0.58 V_{SCE}) and both the reverse peak current densities are caused by CO desorption from the Pt surfaces. For Pt black, a shoulder appears at 0.5 V_{SCE} during the forward scan, caused perhaps by indirect HCHO decomposition and CO formation, which may be a rate-determining-step for methanol oxidation. In contrast, for Pt-NR, no such reaction occurs, suggesting that single-crystal Pt-NR exhibits better electrocatalytic activity for methanol oxidation than a commercial Pt black.

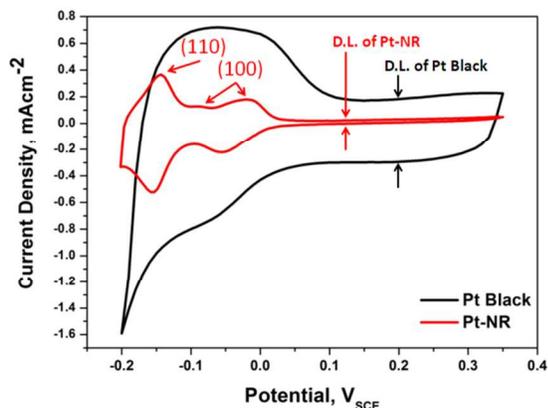


Fig. 3 ECSA of cyclic voltammograms of the Pt-NR catalyst (red curve) and Pt black catalyst (black curve) tested in Ar-saturated 0.5 M H_2SO_4 . The scan rate was 20 mV s^{-1} . The test temperature was controlled at 30 $^\circ\text{C}$.

Figure 3 shows the electrochemical surface area (ECSA) measurements of both Pt-NR and Pt black in 0.5 M H_2SO_4 at 30 $^\circ\text{C}$. As shown in Fig. 3, the significantly strong peak appeared at -0.15 V_{SCE} indicated that our synthesized Pt-NR showed ECSA features characteristic of Pt(110) surfaces rich (ESI^\dagger).^{23, 24} A lower poisoning effect of $\text{CO}_{(\text{ad})}$ will rarely occur on the Pt(110) due to the Pt catalyst surface atomic configuration (ESI^\dagger).²⁵ By comparison, the current density of Pt black without peak probably because the carbon may wrap around Pt to reduce the effective surface planes.²⁶ Due to the high surface area carbon used as the support for Pt black, a wider double layer (D.L.) region can be seen of Pt black in Fig. 3. The

electrochemical characteristics of specific activity (SA) derived from the H_{ad} area of ECSA listed in table 1, are 9.036 mA cm^{-2} and 0.857 mA cm^{-2} for Pt-NR and Pt black, respectively. The catalyst specific activity (SA) of Pt-NR is 10.54 times better than that of a commercial Pt black. We herein suggest the Pt(110) rich of Pt-NR enhanced the electrocatalytic activity.

Table 1 Estimated Pt loadings and electrochemical characteristics of specimens during CV analyses in Fig. 2(a) and Fig. 3.

Specimen	Loadings ^a (mg cm^{-2})	V_p ^b (V_{SCE})	i_f ^b (mA cm^{-2})	i_p/i_b ^b	MA ^c ($\text{mA mg}^{-1}_{\text{Pt}}$)	SA ^d ($\text{mA cm}^{-2}_{\text{Pt}}$)
Pt-NR	0.55	0.63	195	1.23	354.5	9.036
Pt black	0.65	0.64	54	1	77.1	0.857

^a ICP-MS analysis of Pt catalyst prepared sample.

^b Forward peak current density obtained from the voltammograms (Fig. 2(a)) per unit Pt loading mass.

^{cd} The mass and specific activities are obtained by the normalizing the peak current (i_p) to the Pt loadings and electrochemical surface active area (ECSA), respectively (ESI^\dagger).

To investigate the effect of Pt loading on the efficiency of methanol oxidation, we calculated the associated MA and SA values (also listed in Table 1). For Pt-NR, which has less loading mass, the MA value is 4.59 times higher (354.5 mA mg^{-1}) than that for Pt black (77.1 mA mg^{-1}), the SA value is 10.54 times higher (9.036 mA cm^{-2}) than that for Pt black (0.857 mA cm^{-2}), indicating that our prepared single-crystal, fern-like Pt-NR exhibits superior electrocatalytic activity for methanol oxidation. These results suggest the benefit of the PPE technique for the preparation of single-crystal Pt catalysts.

Conclusions

We synthesized single-crystal, fern-like Pt nanorods on the surface of carbon paper by a pulse-mode potentiostatic electrodeposition technique without use of either surfactant or template. Compared with a commercial Pt-black catalyst, the prepared Pt nanorods exhibit definitely improved electrocatalytic activity (4.59 times better mass activity, and 10.54 times better specific activity) for methanol oxidation. More detailed electrochemical and physical characterizations of the prepared nanorods are in progress, with the purpose of optimizing their use as electrodes in PEM-based fuel cells and other electrochemical applications.

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Notes and references

^a Department of Engineering and System Science, National Tsing Hua University, Hsingchu, 30013, Taiwan, ROC.

^b Department of Advanced Optoelectronic Materials and Devices, Industrial Technology Research Institute, Chutung 31040, Taiwan, ROC.

^c Department of Chemical Engineering, National Tsing Hua University, Hsingchu 30013, Taiwan, ROC.

^d Department of Materials Engineering, Ming Chi University of Technology, New Taipei City 24301, Taiwan, ROC. E-mail: jack_hsieh@mail.mcut.edu.tw

[†] Electronic Supplementary Information (ESI) available: Experimental details and supplementary reactions. See DOI: 10.1039/b000000x/

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1. Z. Y. Zhou, N. Tian, Z. Z. Huang, D. J. Chen and S. G. Sun, *Faraday Discuss*, 2008, **140**, 81-92.
2. C. K. Hsieh, M. C. Tsai, C. Y. Su, S. Y. Wei, M. Y. Yen, C. C. M. Ma, F. R. Chen and C. H. Tsai, *Chem Commun*, 2011, **47**, 11528-11530.
3. C. Koenigsmann and S. S. Wong, *Energ Environ Sci*, 2011, **4**, 1161-1176.
4. P. Ferrin and M. Mavrikakis, *J Am Chem Soc*, 2009, **131**, 14381-14389.
5. K. Yamamoto, T. Imaoka, W. J. Chun, O. Enoki, H. Katoh, M. Takenaga and A. Sono, *Nat Chem*, 2009, **1**, 397-402.
6. M. C. Tsai, T. K. Yeh and C. H. Tsai, *Int J Hydrogen Energ*, 2011, **36**, 8261-8266.
7. M. C. Tsai, T. K. Yeh, C. Y. Chen and C. H. Tsai, *Electrochem Commun*, 2007, **9**, 2299-2303.
8. E. Herrero, K. Franaszczuk and A. Wieckowski, *J Phys Chem-Us*, 1994, **98**, 5074-5083.
9. N. Tian, Z. Y. Zhou, S. G. Sun, Y. Ding and Z. L. Wang, *Science*, 2007, **316**, 732-735.
10. Y. J. Song, S. B. Han and K. W. Park, *Mater Lett*, 2010, **64**, 1981-1984.
11. J. Y. Chen, T. Herricks, M. Geissler and Y. N. Xia, *J Am Chem Soc*, 2004, **126**, 10854-10855.
12. Y. Song, R. M. Garcia, R. M. Dorin, H. R. Wang, Y. Qiu, E. N. Coker, W. A. Steen, J. E. Miller and J. A. Shelnutt, *Nano Lett*, 2007, **7**, 3650-3655.
13. S. Sun, D. Yang, G. Zhang, E. Sacher and J. P. Dodelet, *Chem Mater*, 2007, **19**, 6376-6378.
14. S. H. Sun, G. X. Zhang, D. S. Geng, Y. G. Chen, M. N. Banis, R. Y. Li, M. Cai and X. L. Sun, *Chem-Eur J*, 2010, **16**, 829-835.
15. D. Dobos, *Electrochemical Data: A Handbook for Electrochemists in Industry and Universities*, Elsevier, 1975.
16. M. C. Tsai, T. K. Yeh, Z. Y. Juang and C. H. Tsai, *Carbon*, 2007, **45**, 383-389.
17. V. I. Birss, M. Chang and J. Segal, *J Electroanal Chem*, 1993, **355**, 181-191.
18. G. Jerkiewicz, G. Vatankhah, J. Lessard, M. P. Soriaga and Y. S. Park, *Electrochim Acta*, 2004, **49**, 1451-1459.
19. K. A. Friedrich, K. P. Geysers, A. J. Dickinson and U. Stimming, *J Electroanal Chem*, 2002, **524**, 261-272.
20. H. S. Liu, C. J. Song, L. Zhang, J. J. Zhang, H. J. Wang and D. P. Wilkinson, *J Power Sources*, 2006, **155**, 95-110.
21. B. E. Conway, *Journal of Electroanalytical Chemistry*, 2002, **524**, 4-19.
22. G. Q. Lu, A. Crown and A. Wieckowski, *Journal of Physical Chemistry B*, 1999, **103**, 9700-9711.
23. N. Furuya and S. Koide, *Surf Sci*, 1989, **220**, 18-28.
24. J. Solla-Gullon, F. J. Vidal-Iglesias, A. Lopez-Cudero, E. Garnier, J. M. Feliu and A. Aldaza, *Phys Chem Chem Phys*, 2008, **10**, 3689-3698.
25. M. Hepel, I. Dela, T. Hepel, J. Luo and C. J. Zhong, *Electrochim Acta*, 2007, **52**, 5529-5547.
26. J. Xie, X. G. Yang, B. H. Han, Y. Shao-Horn and D. W. Wang, *Acs Nano*, 2013, **7**, 6337-6345.