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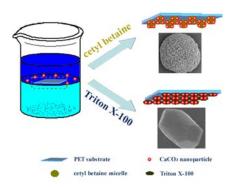
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In the present work, spherical and hexagonal CaCO3 were fabricated on different surfactants micelles modified PET substrates at liquid-liquid interfaces. The results revealed a same nanoparticle-mediate self-organization process in which the surfactants act not only as the regulators but also as templates.

Cite this: DOI: 10.1039/c0xx00000x

www.rsc.org/xxxxxx

ARTICLE TYPE

Morphology Controlling of Calcium Carbonate by Self-Assembled **Surfactant Micelles on PET Substrate**

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Received (in XXX, XXX) Xth XXXXXXXXX 20XX, Accepted Xth XXXXXXXX 20XX 5 DOI: 10.1039/b000000x

Spherical and hexagonal CaCO₃ with hierarchical structures were prepared on surfactant-modified PET substrate between the interfaces of saturated Ca(OH)₂ solution and n-hexane. The results show that CaCO₃ synthesis follows the same nanoparticle-mediated self-organization processes, but the morphology of products strongly depends on the properties of surfactant modification. The synthetic procedure offers 10 several important characteristics for CaCO₃ superstructures fabrication. The surfactants in this experimental system acts not only as templates, but also as regulators for the fabrication of superstructures. This biomimetic system will be promising in synthesizing other functional nanoparticles or nanodevices.

Introduction

15 Hierarchical structure of biomaterials that exhibit uniform morphology and outstanding mechanical properties are universal in living organisms. One secret of biomaterials' superior properties is their organic-inorganic hybrid structure whereby precise arrangement of building blocks is achieved over several 20 length scales.² The organic components, biomacromolecules and organic matrices, can regulate crystal nucleation and growth, modulate crystal shape and size, and control the organization of building blocks into complex structures, having been an active area of research during the past few years.³⁻⁵

25 Calcium carbonate is one of the most abundant biomaterials in nature, mainly as exoskeleton in shells and cell walls or as mechanical support in spicules and spines,1 exhibiting different structures and properties to carry out specific functions respectively. Its crystal structure, size, and shape are the major 30 concerns in crystallization, and have recently received a great deal of attention. Many efforts have been made to mimic the biomineralization process in order to prepare CaCO₃ crystals with defined characteristics.⁵⁻⁷ These biological strategies can be concluded into two categories: (1) mineralization in insoluble 35 matrices such as polymeric scaffolds⁸, biopolymer gels⁹⁻¹¹ or selfassembled monolayers (SAMs)12-14, (2) crystal growth with the help of soluble additives such as polymers¹⁵, copolymers¹⁶, or surfactants¹⁷. And the latter soluble molecules are believed to participate in both the nucleation and growth processes by either 40 binding selectively to certain families of crystal planes^{18, 19} or interacting with the charged ions to stabilize the unstable phases²⁰. which are thought to be an effective and versatile method. Therefore, most researchers put emphasis on finding new molecules with regulating effect and how the regulating effect 45 works. For instance, calcium carbonate crystal with spiky dumbbell-like superstructure was synthesized in the presence of

casein, a typical phosphoprotein.²¹ Wei et al. used a non-ionic peptide type block copolymer as the structure-directing agent to produce calcite with stepped (104) face at the air-solution 50 interface. 16 In most of the works, the soluble additives are present in the form of molecules. However, these amphipathic molecules are easy to assemble to different shapes of micelles. And the micelle itself is a good template. So these micelles are better candidates to regulate the shape of CaCO₃ crystals. But there 55 have been very few researches using soluble additive micelles to direct the growth of the crystals. 22, 23

As most of the crystallizations take place in solution, the biggest problem to this issue is the dynamic nature of the micelles in solution. The shape of the micelles in solution is strongly 60 affected by the solution environment. As the time changing, the properties of the solution where the crystallization reaction happens are also changed. Thus, the shape of the micelles is different. Even the micelles may disappear. Compared with bulk solution, solid-liquid interface provide a unique environment for 65 additive molecules assembly. The assembled patterns are mainly determined by the substrate and additive properties (such as hydrophobic property, surface charge density), which are always different from those in free solution.²⁴⁻²⁸ This makes it possible to control the morphology of crystals by anchoring the self-70 assembled micelles on solid substrate. 29-31

In the previous study, ¹⁴ we revealed the basic rules of forming the superstructure materials via a nanoparticle-mediated selforganization process. We found that the synergetic effects between the organic-inorganic interfaces and the self-assembled 75 monolayers (SAMs) are the essential factors for the control of the hierarchical structures. The SAMs can induce the directions and arrangement of the building blocks. Indeed, the regulation effect of the substrate will be enhanced if the surfactant micelles are further modified on the SAMs. The stronger effect of the 80 surfactant micelles modified substrate relies not only on the larger amount of functional groups, but also the template effect of the micelles.

Therefore, in this work, the regulation effects of the surfactant micelles modified poly (ethylene terephthalate) (PET) substrate 5 on CaCO₃ morphology were studied. And spherical and hexagonal CaCO3 hierarchical structures were produced respectively when cetyl betaine and Triton X-100 were induced. Herein, the organic-inorganic interface provides an independent synthetic environment from bulk solution. The surfactants act not 10 only as structure-directing agents inducing the orientation of the particles, but also as templates guiding them to the final morphology. In both conditions, the formation of CaCO₃ hierarchical structures follows the same nanoparticle-mediated self-organization process, which provides us with a facile way to 15 produce hierarchical nanomaterials with defined structures.

Experimental

Materials

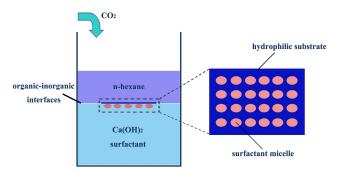
Calcium hydroxide, ammonium bicarbonate, n-hexane, ethanol, acetone, toluene and p-octyl polyethylene glycol phenyl ether 20 (Triton X-100) were purchased from Beijing Chemical Reagents Company. Aminopropyl triethoxysilane (APTES) was purchased from Alfa Aesar. Octadecyltrichlorosilane (OTS) was purchased from Acros Organics. Cetyl betaine was purchased from Chem Service Company. All chemicals were of analytical grade and 25 used as received without further purification. All the glassware was cleaned by sonication in water for 5 min, and rinsed with distilled water, finally dried at 105 °C.

Fabrication of hydrophobic and hydrophilic surface of the polymer substrate

30 By assembling octadecyltrichlorosilane (OTS) on PET substrate, the hydrophobic surface of PET substrate was fabricated according to references³²⁻³⁴. In a typical experiment, APTES was dissolved in acetone (1 vol%) and aged for 240 h. PET substrate was ultrasonically cleaned for 5 min successively in distilled 35 water, ethanol, and acetone and then dried at 105 °C for 5 min. The dried substrate was immersed in the acetone solution of APTES for 5 min and baked at 105 °C for 5 min to form a buffer layer. After that, the substrate grafted by buffer layers was immersed in the OTS solutions (1 vol % in toluene) under a 40 nitrogen atmosphere for 2 min and then rinsed with anhydrous toluene and baked at 105 °C for 5 min to form the hydrophobic surface of PET substrate. The hydrophilic surface of PET substrate was obtained by irradiating the hydrophobic PET substrate with ultraviolet (UV) light. The UV-irradiated surface 45 became hydrophilic because of the formation of hydroxyl groups.

Self-assembly of surfactant on the modified PET substrate

The PET substrates with hydrophobic/hydrophilic surface were immerged in the saturated solution of surfactant for 30 min at 80 °C, and slightly rinsed with deionized water, then dried at 50 105 °C. In this way, the surfactant molecules assembled at the solid-liquid interface. Here, the cetyl betaine and Triton X-100 were chosen for the study.



Scheme 1. Design of the experiment. Surfactant micelles modified 55 substrate was put at the organic-inorganic interface with the modified side towards the inorganic phase. CO2 came from the decomposition of NH₄HCO₃.

CaCO₃ growth on the surfactant-assembled substrate

Room-temperature mineralization of CaCO₃ on the PET substrate 60 was performed at the organic-inorganic liquid interface by a slow gas-diffusion procedure. The organic phase was n-hexane while the inorganic phase was Ca(OH)₂ saturated solution with 500 µL saturated surfactant solution. The nucleation and growth of CaCO₃ were induced by carbon dioxide from the decomposition 65 of NH₄HCO₃ diffusing into the system. (Scheme 1) After a period of time, the substrates covered with CaCO₃ were taken out, and the overgrown specimens were slightly rinsed with deionized water. Then the as-grown CaCO₃ on the substrate was dried in air for further characterization.

70 Characterization of CaCO₃ Deposited on the substrate

The processes of silane and surfactants self-assembly were characterized by water contact angle through JC2000C1 (POWEREACH). The small pieces of as-prepared CaCO₃ on the modified PET substrates were coated with gold for scanning 75 electron microscopy (SEM) measurements on a JMS-6301F microscope. An X-ray diffractometer (XRD, MSAL XD-2 powder X-ray diffractometer), operating at 30 kV, and a current of 30 mA with Cu K α radiation, λ =1.54056 Å, also was used for characterizing the samples. Infrared spectroscopic analysis was 80 performed in transmission mode (FT-IR) using a VERTEX 70. with scanning from 4000 to 500 cm⁻¹.

Results and Discussion

Surfactant assembly on the PET substrate

In this research, zwitterionic surfactant cetyl betaine and nonionic 85 surfactant Triton X-100 were chosen to modify both the hydrophilic and the hydrophobic substrates. The surface properties of substrates were characterized with the water contact angle device. And the results are shown in Figure 1. The contact angles of cetyl betaine micelles modified hydrophilic substrate 90 and Triton X-100 one are 5° and 7° respectively. Comparing with the hydrophilic surface of blank PET, the contact angle became smaller after assembling either cetyl betaine or Triton X-100. While for the hydrophobic substrates, the contact angle did not change after surfactants assembly, still remaining 105°.

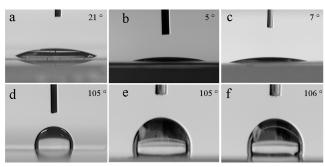


Figure 1. Water contact angles (°) of surfactant micelles assembled substrates and the blank substrates. a) the hydrophilic surface of blank PET; b) the cetyl betaine micelles were assembled on the hydrophilic 5 surface; c) Triton X-100 micelles were assembled on the hydrophilic surface; d) the hydrophobic surface of blank PET; e) cetyl betaine was assembled on the hydrophobic surface; and f) Triton X-100 was assembled on the hydrophobic surface. The contact angle value was shown on the top right corner of each figure.

The drop of the contact angle of the modified hydrophilic substrates results from the change of the functional groups on the surface, indicating the formation of the surfactant micelles on the substrate. In the saturated aqueous solutions, whose concentrations were above the critical micelle concentration 15 (CMC), surfactants existed almost in the form of micelles, with head groups outward to form hydrophilic membrane. Since the head groups of surfactant have a strong affinity to the hydrophilic substrate, the micelles could be assembled onto the substrate easily when the substrate terminated with hydroxyl groups is

20 immerged into the saturated aqueous solutions of surfactants. 35-37 Figure 2 shows the SEM images of PET substrate before and after modification. Before modification, both the hydrophilic and hydrophobic substrates have a smooth surface. The bright spots on the substrates are the sputtered Au nanoparticles. After 25 modification, cetyl betaine micelles and Triton X-100 micelles are formed only on the hydrophilic substrates. Figure 2b shows cetyl betaine micelles modified substrates are spherical while the Triton X-100 micelles have spindle morphology as shown in Figure 2c. What should point out here is that since the drying 30 process will affect the micelle structure, the SEM images may not reflect the structures formed in situ. Although the SEM data is speculative, these surfactant aggregates on the substrates also suggested a successful modification. Given the experimental results mentioned above, we proposed a possible modification 35 process with detailed description which was shown in Figure 3. After the assembly of cetyl betaine micelles, the terminal groups of the substrate become -COO while the surface groups of the blank PET are hydroxyl. The difference between the deprotonated carboxyl groups against hydroxyl groups benefits 40 the improvement of hydrophilicity, and the water contact angle become smaller. When it comes to Triton X-100, the surface groups are still hydroxyl. But as the assemblies of the micelles, more hydroxyl groups are formed on the surface of the substrate. Thus the modified substrate has a denser hydroxyl groups than 45 that of the blank PET, becoming a more hydrophilic substrate.

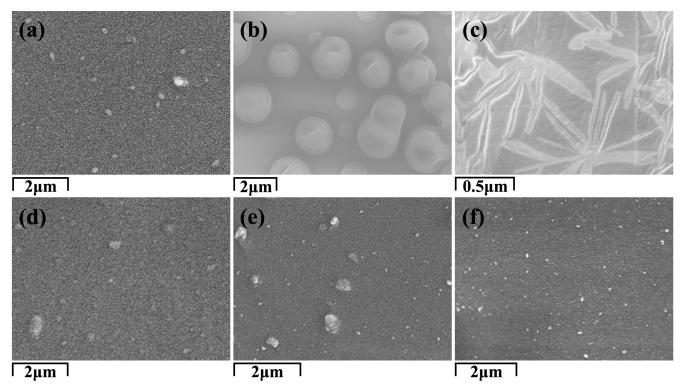


Figure 2. SEM images of hydrophilic PET (a), treated with cetyl betaine (b) and Triton X-100 (c). SEM images of hydrophobic PET (d), treated with cetyl betaine (e) and Triton X-100 (f).

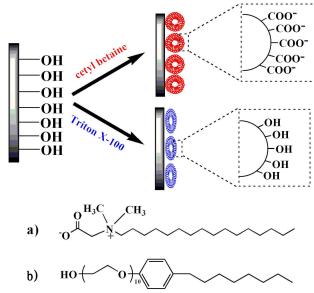


Figure 3. The formation of surfactant micelles on the hydrophilic substrates. a) The molecular formula of cetyl betaine; b) the molecular formula of Triton X-100.

5 And the water contact angle reduces.

The final state of surfactants aggregates was a balance of the interactions between surfactant, solvent, and a solid substrate on the aggregation. To make it simple, we have studied only net neutral surfactants. (For zwitterionic surfactant cetyl betaine, the 10 negative charged carboxyl was neutralized by quaternary ammonium cation.) The strong monopolar charge-charge interactions that occur for anionic or cationic surfactants were minimized. Under this condition, the surfactants aggregates were more strongly affected by solution conditions, substrate property 15 and neighboring aggregates. 30 In this research, the authors chose saturated aqueous solutions for the surfactant assembly in which the concentration was higher than that of the bulk CMC. Therefore, most of the surfactant in solution formed spherical aggregates, and the surfactant micelles' morphology was no 20 longer a function of concentration. Herein, the difference between the assembled aggregates of cetyl betaine and Triton X-100 was contributed to their different head groups. So, it is reasonable to consider the structure change of the micelles from solution to substrate due to the replacement of water-headgroup interactions 25 with surfactant-solid interactions. There is a correlation between the surfactant curvature and the affinity of the head group for the solid substrate. On the one hand, a high density of surface sites that can form strong bonds between the head groups and the substrate will lead to a strong perturbation of micelle structure. 30 Thus a less curved micelle would form as the density of available strong-bonding sites increases. As for Triton X-100, its head groups are hydroxyl, which can hydrogen-bond with the hydroxyl groups on the substrate. This strong interaction may be the main contributor to the structural change after assembling. It is 35 therefore, the curvature becomes smaller, forming the spindleshaped surfactant micelles on the substrate. On the other hand, if the substrate forms low energy bonds with the surfactant head groups, it is reasonable to have slightly perturbed micelles attached to the solid. The head groups of cetyl betaine are 40 carboxyl groups and quaternary ammonium cations, which can

interact with the substrate also by hydrogen bonds. But the hydrogen bonds between the cetyl betaine and the substrate are weaker than that of the Triton X-100. This relatively weak interaction would not change the shape of the micelles after 45 assembling, maintaining spherical. Furthermore, the repelling interaction between carboxyl groups causes the distances between the cetyl betaine aggregates units larger than that of Triton X-100 (Figure 2b~c).

While for the hydrophobic substrates, the contact angle had 50 nearly no change after immerging into the surfactant saturated aqueous solutions (Figure 1d~f). There was an alkyl chain layer on the surface of the OTS-modified hydrophobic substrate, so that the solid substrate cannot form hydrogen bonds with the surfactant head group. In this situation, a large free energy 55 reduction of the system was possible by covering up the solid substrate with surfactant hydrocarbon tails. So the free surfactant monomers or hemicylinder micelles could be assembled onto the substrates by the hydrophobic interactions.³⁷ However, the results showed that the water contact angle of hydrophobic substrate was 60 not changed. The SEM images of this kind of substrate before and after treating with surfactant solutions (Figure 1d~f) had also shown that the assembly of surfactants did not take place. This indicated that there were few free surfactant monomers or hemicylinder micelles in the saturated solutions, and the 65 reduction of free energy was not sufficient to perturb the structure of spherical micelle in the bulk solution.

It is worth noting that surfactant micelles can only selfassembled on hydrophilic substrate. Since the hydrophilic substrates were obtained by irradiating the hydrophobic PET 70 substrates with UV light. If we expose only one side of the substrate to UV light, this side becomes hydrophilic while the other side keeps hydrophobic. When the substrate was immerged into surfactant solution, the micelles assembled only on the hydrophilic side, and the other side remained hydrophobic. The 75 amphipathy of the substrate allows it to float between the organic-inorganic interfaces, which make the following study possible.

Morphology of CaCO₃ on the substrates

Cetyl betaine and Triton X-100 surfactant micelles modified 80 substrates were put at the organic-inorganic interface separately with the hydrophilic side towards the aqueous phase. After 48 hours of reaction, different shapes of CaCO₃ crystals were grown on these substrates respectively. When cetyl betaine micelles modified substrates were induced in the reaction system, 85 spherical particles with about 30 µm in diameter were fabricated (Figure 4a). The inset picture of Figure 4a is the lowmagnification image of CaCO₃ samples, which shows that the samples are composed of a large quantity of well-dispersed spherical particles with uniform diameter. The surface appearance 90 of sphere was clearly visible in Figure 4c, showing triangular pyramidal morphology radiating from the centre of the crystals. These pyramids were 1µm height with similar shape and smooth surfaces. The typical image of the cracked sample in Figure 4d shows that the CaCO₃ spheres are composed of some small sticks. 95 And we should also notice the hole (indicated by the black arrow) in the centre of the cracked sphere suggesting a core in the spherical crystal. The size of the hole is about 1-2 µm which is just the same as the size of the cetyl betaine micelle. These results

reveal that the micelles have a template role in the growing process. More detailed information about the spherical structure can be further provided by XRD analyses. In Figure 4b, the XRD spectrum clearly indicates that the as-prepared products are 5 composed of a calcite phase with the characteristic (104), (110), (113), (202) and (116) planes (JCPDS card No. 05-0586). The two broad peaks observed at 47.2 ° and 54.1 ° were consistent with the XRD pattern of blank PET shown in Figure S1. The random orientations of the crystals suggest that cetyl betaine 10 molecules have relative weak interaction with the planes of calcite crystals. Thus, the adsorption on specific planes did not take place. Controlled experiment was taken when cetyl betaine micelles modified substrate was removed from the system. The obtained products showed a rhombohedral calcite phase with an 15 average size over 50µm (Figure S2), revealing the surfactant has an important influence on the morphology of CaCO₃.

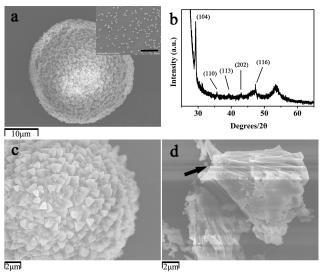
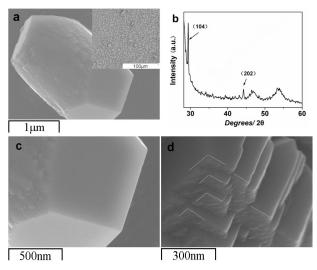


Figure 4. a, c, d) The typical SEM images and b) corresponding XRD pattern of CaCO3 which were fabricated on the cetyl betaine-assembled 20 hydrophilic substrate between the *n*-hexane-aqueous interfaces. The inset in (a) is the low-magnification image of the specimen presented in Figure 2a.

On the other hand, the introduction of Triton X-100 micelles onto the substrate resulted in the formation of hexagonal CaCO₃ 25 crystals. In Figure 5a, the hexagon was 4µm in length and 3µm in width. The inset picture of Figure 5a shows a large number of hexagonal CaCO3 with high nucleation densities and narrow spread particle size on the substrate. The rough surface of the hexagonal crystals can be seen at high magnification in Figure 30 5c~d. The corresponding XRD spectrum in Figure 5b indicated that the as-synthesized products were composed of a calcite phase with the characteristic (104) and (202) planes. Similarly, the two broad peaks at 47.2 $^{\circ}$ and 54.1 $^{\circ}$ belonged to blank PET (Figure S1). Without the Triton X-100 micelles, CaCO₃ crystals grow in 35 random directions resulting in many other planes in the XRD curve. 14 This means Triton X-100 molecules have strong interaction with the (104) and (202) planes. When absorbing on the planes, they can inhibit the crystals from growing vertical to the directions. Thus these planes are remained while others 40 disappeared. CaCO₃ that mineralized without Triton X-100 results in the formation of large rhombohedral calcite crystals with an average size over 50µm (Figure S2), revealing the surfactant has a significant influence on the crystalline CaCO₃.



45 Figure 5. a, c, d) The typical SEM images and b) corresponding XRD pattern of CaCO3 which were fabricated on the Triton X-100-assembled hydrophilic substrate between the *n*-hexane-aqueous interfaces. The inset in (a) is the low-magnification image of the specimen presented in Figure

Both the spherical and hexagonal products are different from those obtained from the organic-inorganic interface. In addition, CaCO₃ crystals grow evenly on the hydrophilic side of the substrate. While on the hydrophobic side, we can only find very few rhombohedral crystals at the edge of the substrate. All these 55 evidence shows that the surfactant micelles modified substrates play a key role in modulating the morphology of the crystals.

The roles of surfactant on CaCO3 morphology

To further demonstrate the important role of the surfactant, the morphological development of CaCO₃ at different crystallization 60 times was examined. The representative SEM images of the products obtained at 2h, 12h, 24h and 48h are presented in Figure 6 and Figure 7.

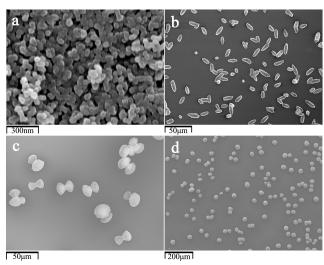


Figure 6. a) SEM image of the products obtained directly from the 65 organic-inorganic interface (not on the substrate) after crystallization for 2

h; b, c, d) SEM images of CaCO₃ prepared on the cetyl betaine-assembled substrate for 12 h, 24 h and 48 h respectively.

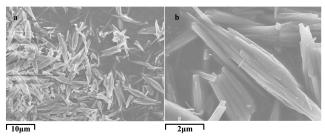


Figure 7. SEM images of rod-like CaCO3 crystals obtained from the 5 substrate free system. The surfactant in the inorganic phase is cetyl

Figure 6 shows the whole growing process when cetyl betaine was induced in the system. The products of Figure 6a were collected directly from the organic-inorganic interfaces (not on 10 the substrate) after 2h of reaction. It is clear that the first species to form are amorphous calcium carbonate sphere particles with about 50nm in diameter. This is in good agreement with previous works. 14, 38, 39 When the reaction time is prolonged to 12h, the asprepared products are rod-like crystals with 20µm in length and 15 5µm in diameter. When crystallized for 24h, sheaf-like CaCO₃ hierarchical structures appear. Finally, spherical CaCO₃ crystals are formed. The whole process of the crystals growth on the substrate is just like flowers in bloom. In addition, the size of the crystals did not change from 12h to 48h, which means the further 20 growing process is just the continually assembly of the building blocks. This result also reminds us of the earlier findings in Figure 4d that the CaCO₃ spheres are composed of small sticks. To obtain more evidence for the formation of these building blocks, the morphology of CaCO₃ grown in the same but 25 substrate free system was examined. In this system, the modulate effect of the functional substrates was removed, but the surfactant in the aqueous phase will still regulate the crystal growth. The corresponding images in Figure 7 show the products have the similar size and shape to that of the building blocks. This is direct 30 evidence that the CaCO₃ sticks which finally assembled into spherical CaCO₃ hierarchical structure was formed with the help of the surfactant in the aqueous phase.

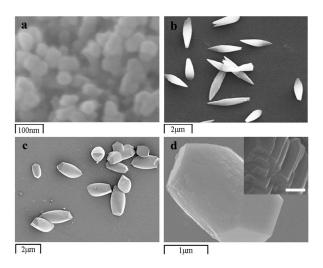


Figure 8. a) SEM image of the products at the n-hexane-aqueous interface 35 after crystallization for 2 h; b, c, d) SEM images of CaCO3 prepared on the Triton X-100-assembled substrate for 12 h, 24 h and 48 h respectively. The scale bar of the inset in (d) is 300 nm.

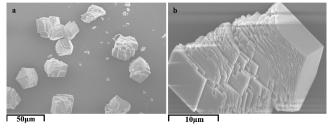
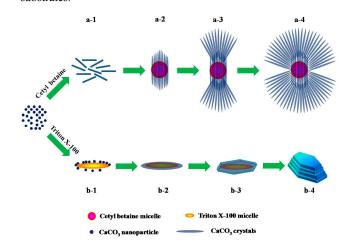


Figure 9. SEM images of rod-like CaCO₃ crystals obtained from the 40 substrate free system. The surfactant in the inorganic phase is Triton X-

When it comes to the Triton X-100-modified substrate, the growth of the hexagonal CaCO₃ followed a similar process. Figure 8a shows the microstructure of CaCO₃ obtained from the 45 organic-inorganic interface (not on the substrate) after 2h of crystallization. It can be seen that the products are consisted of 30nm nanoparticles, revealing that the early stage of the hexagonal products are also amorphous calcium carbonate. With the time passing, spindle-shaped and hexagonal CaCO3 were 50 fabricated on the substrate. Comparing Figure 8b-d, we can find a smooth surface is gradually forming at the end of the spindle, which reveals a ripening process in the crystal growth. In order to get more details about the formation mechanism, we construct a same but substrate free system. After 24h of reaction, calcite 55 crystals with layered structure have been found as shown in Figure 9. It is worth noting that the products grown on the Triton X-100 micelles modified substrate (Figure 5d) have similar structures, which means the surfactant molecules in the solution play a significant role in forming the structure and morphology of 60 the hexagonal CaCO₃. And this layered structure comes from the specific adsorption of Triton X-100 on (104) and (202) planes (see the XRD pattern in Figure 5b).^{4, 16}

Besides, in both conditions, the density of the crystals on the substrates is high in the edges while low in the middle. This 65 phenomenon reveals CaCO₃ growth begins from the edges of the substrates, also indicating a self-organization process on the substrates.



Scheme 2. The possible formation process of $CaCO_3$ crystals on cetyl betaine micelles modified substrate (a-1~a-4) and Triton X-100 micelles modified substrate (b-1~b-4).

Based on the experiment results mentioned above, it could be 5 suggested that CaCO₃ synthesis follows the same nanoparticlemediated self-organization processes. However the final morphology of products depends on the shapes of the micelles on the substrate. Scheme 2 shows the possible formation process of CaCO₃ crystals. At the very beginning of CaCO₃ growth, carbon 10 dioxide from the decomposition of ammonium bicarbonate reacted with H₂O to enrich the carbonate ions and then induce heterogeneous precipitation to form nanoparticles between the organic-inorganic interfaces. In this process, the terminal groups of the surfactant in the aqueous solution may affect the particles 15 formation. As for cetyl betaine, the deprotonated -COO groups have the ability to bind calcium ions by electrostatic interaction. After that, these nanoparticles are likely to transfer to the substrate, interacting with the surface groups of the cetyl betaine micelles through electrostatic interactions, hydrogen bonds or 20 Van der Waals forces. At the same time, these nanoparticles with random crystallographic orientations and high internal energy are growing gradually to fuse with each other and tending to lower the energy by arranging orderly and assembling into CaCO3 sticks (Scheme 2 a-1). Consequently, these sticks may rotate to 25 get the same orientation with driving force coming from specific absorption of the surfactant and the removing of surface energy which is shown in Scheme 2 a-2.40 Meanwhile, some other CaCO₃ nanoparticles attach to the sticks by Brownian motion and form more sticks in the same manner. With an increasing of 30 crystallization time, these building blocks (the sticks) begin to aggregate at both ends of the CaCO₃ nanorods. (Scheme 2 a-3) Finally, CaCO₃ hierarchical structures with unusual morphologies are formed by continually attaching sticks or nanoparticles. (Scheme 2 a-4) When cetyl betaine is replaced by a nonionic 35 surfactant (Triton X-100), CaCO₃ growth suffers the same process. CaCO3 nanoparticles formed at the interfaces will transfer to the substrate and absorb on the micelles. (Scheme 2 b-1) These nanoparticles tends to fuse with each other and crystallize into hexagonal calcite crystals. (Scheme 2 b-2~b-3) At 40 this time, surfactant molecules (Triton X-100) in the saturated Ca(OH)₂ solution may have strong interaction with some specific planes of CaCO₃ and absorbed on these planes. So the crystal growth along this direction will be constrained. When other nanoparticles absorbed on the hexagonal crystals, they can only 45 grow vertical to this direction. Finally, hexagonal CaCO3 crystals with layered structure are formed after hours of reaction.

For most of the time, the balance between the crystal growth and mass transport is an important factor in controlling the morphology of crystals. But herein, the surfactant micelles assembled on the substrate play the leading role in influencing the final morphologies. And the size of CaCO₃ crystals is also affected by the micelles. The cetyl betaine micelle (Figure 1b) has a bigger size than the Triton X-100 micelle (Figure 1c). Then, after crystallization, the spherical CaCO₃ flowers (Figure 4a) are therefore bigger than the hexagonal crystals (Figure 5a). Moreover, the surfactant micelles in the biomimetic mineralization system are acting not only as a template, but also as an inhibitor preventing CaCO₃ crystallizing. In the FTIR

spectra (Figure 10), the presence of the peak at 1082 cm⁻¹ attributed to symmetric stretch of carbonate ion in plane (v₁), and the split peaks at around 1387 could be seen as its asymmetric stretch in plane (v₃) which can be indexed as the characteristic peaks for amorphous calcium carbonate (ACC). ^{7, 21, 41, 42} This result is consistent with our previous work ¹⁴ that nanoparticles obtained by an ACC-to-calcite transformation. It also shows that the surfactant micelles have the ability to stabilize ACC as ACC can be found even after 48h mineralization.

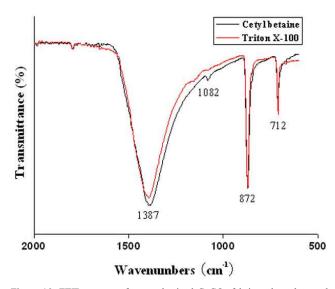


Figure 10. FTIR spectra of as-synthesized CaCO₃ fabricated on the cetyl betaine-assembled hydrophilic substrate (black line) and the Triton X-100-assembled hydrophilic substrate (red line) after 48h of mineralization.

What should also be mentioned is the n-hexane-aqueous interfaces we constructed. The amphipathic surfactants in the aqueous solution tend to aggregate at the interfaces to minimize 75 the interfacial free energy. The concentration of the surfactants at the interfaces is higher than that of the bulk solution. Therefore, the surfactants micelles on the substrates will be kept intact when the surfactants micelles modified substrates were put at the interfaces. Besides, the use of the organic-inorganic interfaces is 80 a way to mimic the bio systems because many organisms, including mollusks, echinoderms, calcisponges, corals, form their hierarchical mineral skeleton out of calcium carbonate minerals in the organic-inorganic interfaces. Also, the organic-inorganic interfaces have been proved to have outstanding advantages over 85 the bulk aqueous solution in controlling the assembly of inorganic nanoparticles in the previous works. 43,44 Thus, this biomimetic approach will be promising in synthesizing other functional nanoparticles or nanodevices.

Conclusions

Ocombining the regulatory effect and the template role of surfactants, we use the surfactant micelles modified PET substrate to modulate the morphologies of CaCO₃ at the n-hexane-Ca(OH)₂ saturated solution interfaces. After 48h of crystallization, spherical and hexagonal CaCO₃ crystals with hierarchical structure were obtained separately by using different kinds of surfactants. During the crystal growth process, the assembled surfactant micelles not only contributed to stabilizing

the assembly of nanoparticles, but also exerted vigorous control over the morphology of the products. From the time dependent experiments, we suggest that CaCO3 synthesis follows the nanoparticle-mediated self-organization processes. The organic-5 inorganic liquid interfaces brought about the controlled

nucleation of calcite from ACC. The surfactant in solution induced the formation of CaCO₃ with well-defined orientation and morphology by specific adsorption. And the surfactant micelles on the substrate act as a template which allows further 10 growth and decides the final hierarchical structure of the crystals.

These findings will help for better understanding of the formation mechanism of the unique CaCO₃ morphologies. The biomimetic system will also contribute some valuable information to biomineralization researches.

15 Acknowledgment

This work is supported by the National Basic Research Program of China (973 Program) with No. 2011CB706900, the National Nature Science Foundation of China (nos. 50872149 and 50502003), the Scientific Research Foundation for Returned 20 Scholars within the Ministry of Education of China, and the

President Foundation of the Graduate University of the Chinese Academy of Sciences.

Notes and references

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- 30 † Electronic Supplementary Information (ESI) available: The XRD pattern of blank PET substrate. The SEM image of crystals grown at the interface without surfactant in the inorganic phase for 48h. See DOI: 10.1039/b000000x/
- ‡ Footnotes should appear here. These might include comments relevant 35 to but not central to the matter under discussion, limited experimental and spectral data, and crystallographic data.
 - 1. R. A. Metzler, I. W. Kim, K. Delak, J. S. Evans, D. Zhou, E. Beniash, F. Wilt, M. Abrecht, J.-W. Chiou, J. Guo, S. N. Coppersmith and P. U. P. A. Gilbert, Langmuir, 2008, 24, 2680-2687.
 - 2. H. Colfen, Nat Mater, 2010, 9, 960-961.
 - W. Dong, C. Tu, W. Tao, Y. Zhou, G. Tong, Y. Zheng, Y. Li and D. Yan, Cryst. Growth Des., 2012, 12, 4053-4059.
- 4. J. W. Nielsen, K. K. Sand, C. S. Pedersen, L. Z. Lakshtanov, J. R. Winther, M. Willemoës and S. L. S. Stipp, Cryst. Growth Des., 2012, **12**. 4906-4910.
- G. Begum and R. K. Rana, Chem. Commun., 2012, 48, 8216-8218.
- Y. Fukui and K. Fujimoto, J. Mater. Chem., 2012, 22, 3493-3499.
- J. Ihli, Y.-Y. Kim, E. H. Noel and F. C. Meldrum, Adv. Funct. Mater., 2013, 23, 1575-1585.
- 8. L. Liu, D. He, G.-S. Wang and S.-H. Yu, Langmuir, 2011, 27, 7199-7206.
- D. C. Bassett, B. Marelli, S. N. Nazhat and J. E. Barralet, Adv. Funct. Mater., 2012, 22, 3460-3469.
- 55 10. E. Asenath-Smith, H. Li, E. C. Keene, Z. W. Seh and L. A. Estroff, Adv. Funct. Mater., 2012, 22, 2891-2914.
 - 11. H. Li and L. A. Estroff, J. Am. Chem. Soc., 2007, 129, 5480-5483.
 - 12. J. Aizenberg, A. J. Black and G. M. Whitesides, J. Am. Chem. Soc., 1999, 121, 4500-4509.
- 60 13. B. Pokroy and J. Aizenberg, Crystengcomm, 2007, 9, 1219-1225.
 - 14. X. Liang, J. Xiang, F. Zhang, L. Xing, B. Song and S. Chen, Langmuir, 2009, 26, 5882-5888.

- 15. A. S. Schenk, I. Zlotnikov, B. Pokroy, N. Gierlinger, A. Masic, P. Zaslansky, A. N. Fitch, O. Paris, T. H. Metzger, H. Cölfen, P. Fratzl and B. Aichmayer, Adv. Funct. Mater., 2012, 22, 4668-4676.
- 16. X. Wei, Y. Su, T. Wen, Z. Li, J. Yang and D. Wang, Crystengcomm, 2013, 15, 3417-3422.
- 17. Z. Zhao, L. Zhang, H. Dai, Y. Du, X. Meng, R. Zhang, Y. Liu and J. Deng, Microporous and Mesoporous Materials, 2011, 138, 191-199.
- 70 18. J. Aizenberg, N. Ilan, S. Weiner and L. Addadi, Connect. Tissue Res., 1996, 35, 17-23.
- 19. S. Elhadj, E. A. Salter, A. Wierzbicki, J. J. De Yoreo, N. Han and P. M. Dove. Cryst. Growth Des., 2005, 6, 197-201.
- 20. K. Naka, S.-C. Huang and Y. Chujo, Langmuir, 2006, 22, 7760-7767.
- 75 21. Y. Liu, Y. Cui, H. Mao and R. Guo, Cryst. Growth Des., 2012, 12, 4720-4726.
 - 22. Y.-Y. Kim, K. Ganesan, P. Yang, A. N. Kulak, S. Borukhin, S. Pechook, L. Ribeiro, R. Kröger, S. J. Eichhorn, S. P. Armes, B. Pokroy and F. C. Meldrum, Nat Mater, 2011, 10, 890-896.
- 80 23. B. Xie, H. Shi, G. Liu, Y. Zhou, Y. Wang, Y. Zhao and D. Wang, Chem. Mat., 2008, 20, 3099-3104.
 - 24. Y. Hou, M. Cao, M. Deng and Y. Wang, Langmuir, 2008, 24, 10572-
 - 25. K. S. Varanasi and S. Garoff, Langmuir, 2005, 21, 9932-9937.
- 85 26. H. Bock and K. E. Gubbins, Phys. Rev. Lett., 2004, 92, 135701.
- 27. R. Atkin and G. G. Warr, J. Am. Chem. Soc., 2005, 127, 11940-11941
- 28. W. Mamdouh, H. Uji-i, J. S. Ladislaw, A. E. Dulcey, V. Percec, F. C. De Schryver and S. De Feyter, J. Am. Chem. Soc., 2005, 128, 317-
- 29. A. Guerrero-Martínez, J. Pérez-Juste, E. Carbó-Argibay, G. Tardajos and L. M. Liz-Marzán, Angew. Chem.-Int. Edit., 2009, 48, 9484-9488.
- 30. K. S. Choi, E. W. McFarland and G. D. Stucky, Adv. Mater., 2003, **15**, 2018-2021.
- 31. K.-S. Choi, H. C. Lichtenegger, G. D. Stucky and E. W. McFarland, J. Am. Chem. Soc., 2002, 124, 12402-12403.
- 32. P. Zhu, M. Teranishi, J. Xiang, Y. Masuda, W.-S. Seo and K. Koumoto, Thin Solid Films, 2005, 473, 351-356.
- 100 33. J. Xiang, P. Zhu, Y. Masuda and K. Koumoto, Langmuir, 2004, 20, 3278-3283.
- 34. M.-H. Park, Y. Ofir, B. Samanta and V. M. Rotello, Adv. Mater., 2009, 21, 2323-2327.
- 35. S. Manne and H. E. Gaub, Science, 1995, 270, 1480-1482.
- 105 36. Y. I. Rabinovich, S. Pandey, D. O. Shah and B. M. Moudgil, Langmuir, 2006, 22, 6858-6862.
 - 37. K. Shah, P. Chiu and S. B. Sinnott, Journal of Colloid and Interface Science, 2006, 296, 342-349.
 - 38. J. Ihli, A. N. Kulak and F. C. Meldrum, Chem. Commun., 2013.
- 110 39. P. Bots, L. G. Benning, J.-D. Rodriguez-Blanco, T. Roncal-Herrero and S. Shaw, Cryst. Growth Des., 2012, 12, 3806-3814.
 - 40. N. Gehrke, H. Cölfen, N. Pinna, M. Antonietti and N. Nassif, Cryst. Growth Des., 2005, 5, 1317-1319.
 - 41. L. Addadi, S. Raz and S. Weiner, Adv. Mater., 2003, 15, 959-970.
- 115 42. A. Sarkar, K. Dutta and S. Mahapatra, Cryst. Growth Des., 2012, 13, 204-211.
 - 43. Y. Lin, H. Skaff, T. Emrick, A. D. Dinsmore and T. P. Russell, Science, 2003, 299, 226-229.
- 44. M. X. Luo, Y. M. Song, and L. L. Dai, J. Chem. Phys., 2009,131, 194703.