



**A comparative study for the characterization of polyaniline based nanocomposites and membrane properties**

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Dear Dr Suzanne Howson MRSC  
Publishing Editor.

**Reply to reviewer's comments on the Manuscript ID RA-ART-02-2014-001040.R1**

Dear Professor,

Thank you very much for your mail regarding above manuscript along with the reviewers comments, questions as well as suggestions. We are really appreciated for the insightful comments from anonymous reviewers, which are believed to improve the quality of the manuscript considerably. We have tried our best efforts to revise the manuscript as per their suggestions and guidelines. The changes done in the manuscript are marked by the **green text**. The replies to the referee comments are given below:

**Referee: 2**

Comments to the Author

In my opinion, the article has considerably improved its quality in the present form. I think it can be accepted for publication. I only have some minor comments:

1. Despite the fact that the procedure presented for the synthesis of composite membranes could imply an interesting advance in the field, I am not sure about the possibility of using the membranes to detect  $K^+$ , since the difference in transport numbers and effective fixed charge densities between  $K^+$  and  $Na^+$  are very low. Moreover, could this slight difference has more to do with the mobilities of the ions itself rather than with the selectivity of the membranes?

**Answer: We agreed with the comments. In future we will try to synthesize the materials that show large difference in transport numbers and effective fixed charge densities among different cations.**

2. Line 141, page 6: In my opinion "That's why..." seems to informal for a scientific paper.

**Answer: Modified sentence has been mentioned (Line 141-142, page 6)**

With these clarifications I hope the present manuscript is now suitable for publication in your esteemed journal.

Yours Sincerely  
Prof. Tjoon Tow Teng

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1 A comparative study for the characterization of polyaniline based  
2 nanocomposites and membrane properties

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8

9 Abstract

10 Polyaniline (PANI) based novel nanocomposites have been fabricated by using a  
11 simple sol-gel method. The experimental conditions including mixing volume ratio of  
12 reactants, pH and temperature were established for their synthesis. For the comparative  
13 studies, characterizations were performed by using FTIR, TGA, SEM, TEM and XRD  
14 analyses. The structural studies expose amorphous and semi-crystalline nature with the  
15 particle size in nano range. PANI supported nanocomposites exhibit improved conducting  
16 behavior, higher ion uptake capacity along with chemical and thermal stability. Membrane  
17 potentials of monovalent electrolytes namely KCl, NaCl and LiCl have been studied at  
18 different concentrations to examine the relationship between effective fixed charge density  
19 and transport properties of the membrane. Present study revealed that among Li<sup>+</sup>, K<sup>+</sup> and Na<sup>+</sup>  
20 ions, PANI-Zr(IV)sulphosalicylate nanocomposite membrane shows more cation-selective  
21 behaviour towards K<sup>+</sup> ions. Effective fixed charge densities were found to follow the order  
22 K<sup>+</sup> > Na<sup>+</sup> > Li<sup>+</sup>.

23 Key word: *Synthesis, characterization, nanocomposite, thermal and chemical stability,*  
24 *membrane properties.*

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## 26 1.0 Introduction

27 Polyaniline (PANI) is one of the potential conducting polymers owing to its simple  
28 preparation, excellent environmental stability.<sup>1-4</sup> Recently PANI based nanostructures such  
29 as nanofibre, nanowire, nanotubes have established great interest because of their successful  
30 applications in diverse fields.<sup>5-10</sup> In addition, PANI nanoparticles can be synthesized either by  
31 chemically or electrochemically through the polymerization or self assemble process.<sup>11, 12</sup>  
32 However, there are some drawbacks associated with pure PANI due to its intractable nature  
33 as it is usually obtained as an insoluble powder and electrochemically thin brittle film.<sup>13</sup> To  
34 overcome limitations and improve the performance of the conducting polymer, organic-  
35 inorganic hybrid material was synthesized by the reaction of organic polymer (PANI) with  
36 inorganic precipitate. The conjugation of inorganic precipitate with organic polymer  
37 improves the performance of both the 'host' and the 'guest' thereby leads to interesting  
38 physical and chemical properties. In this regard, a number of investigations concerning the  
39 development of the techniques for the incorporating of nanoparticles into the matrix of PANI  
40 have been published.<sup>14-19</sup> These composite materials possess interesting physical and  
41 chemical properties and showed interesting applications in diverse fields.<sup>20-26</sup> PANI based  
42 organic-inorganic composites materials (PANI-Ti(IV)arsenate, PANI-Ti(IV)tungstate and  
43 PANI-Zr(IV)S) have been successfully used as ion exchangers, conducting materials and  
44 photocatalyst as reported earlier.<sup>27-30</sup> To apply these composite materials in diverse field work  
45 was continued and it was found that these materials can also be effectively used as an ion  
46 selective electrode membrane. A number of papers related to PANI based membrane  
47 electrodes have been published however, no work associated with membrane properties of  
48 PANI-Zr(IV)S nanocomposite has not been reported yet. The present work reports a  
49 comparative study for the characterizations of PANI based organic-inorganic composite  
50 materials and their ion selective membrane potential. The ion exchange and transport

51 properties of composite materials in the form of membrane have also been explored. The  
52 electric potential of the composite membrane, in contact with different monovalent  
53 electrolyte solutions, has been measured to evaluate the thermodynamically effective fixed  
54 charge density by means of different proposed approaches.<sup>31-34</sup> The experimental values of  
55 potential have also been used to calculate the mobility ratio, transport numbers of counter-  
56 ions and permselectivity of the polystyrene blended hybrid membrane.

57

## 58 2.0. Experimental

### 59 2.1. Materials and Methods

60 **All reagents and chemicals were of analytical grade and used without any treatment.**

61 The main reagents for the synthesis: aniline, potassium persulphate, sodium tungstate and, 5-  
62 Sulphosalicylic acid and zirconium(IV)oxychloride were procured from Loba Cheme Pvt.  
63 Ltd. Titanium tetrachloride, carbon tetrachloride and sodium arsenate were from CDH, India.  
64 For the characterization of all three nanocomposites the instruments such as pH meter,  
65 Fourier Transform-IR Spectrometer, Thermogravimetric Analysis/Differential Thermal  
66 Analysis (TGA/DTA), X' Pert PRO analytical diffractometer, Scanning Electron Microscope  
67 (SEM; LEO, 435 VF), Transmission Electron Microscopy (TEM) Jeol H-7500 Microscope,  
68 temperature controlled shaker and Muffle furnace were used as described in previous paper.<sup>35</sup>  
69 A digital potentiometer model 118 (Electronics India) was employed for measuring the  
70 membrane potential.

71

#### 72 2.2.0. Synthesis of PANI and inorganic precipitates

73 The synthesis of PANI and inorganic precipitates of Ti(IV)arsenate, Ti(IV)tungstate  
74 and Zr(IV)sulphosalicylate were carried out by using the sol gel method as described in  
75 previous reported paper.<sup>27</sup> An inorganic precipitate of Ti(IV)arsenate was prepared by the

76 mixing solutions of titanium tetrachloride (0.1M) and sodium arsenate (0.25M) gradually  
77 with constant stirring at  $25\pm 2^\circ\text{C}$  for 1h whereby gel type slurries were obtained. The pH of  
78 the solution was maintained by adding dilute solution of HCl or  $\text{HNO}_3$ . The resulting white  
79 precipitate so formed was kept overnight in the mother liquor for digestion. For the synthesis  
80 of Ti(IV)tungstate and Zr(IV)sulphosalicylate, the similar method was applied as mentioned  
81 above except the change in concentration of zirconium oxychloride (0.25M) and 5-  
82 sulphosalicylic acid (0.5 M). The resulting white precipitate (of each solution) was kept  
83 overnight in the mother liquor for digestion. After decanting off the supernatant liquid, the  
84 remaining precipitate was washed with demineralized water to remove any excess reagent.

85

86 2.2.1. Synthesis of PANI-Ti(IV)arsenate, PANI-Ti(IV)tungstate and PANI-  
87 Zr(IV)sulphosalicylate composite materials

88 Each composite material was prepared by adding gel of PANI individually to each  
89 inorganic precipitate in a volume ratio of 1:1, by mixing thoroughly with continuous stirring  
90 for 1 h at  $25\pm 2^\circ\text{C}$ . The resultant green gel obtained was kept for 24 h at room temperature for  
91 digestion. The supernatant liquid was decanted and the gel was filtered under suction. The  
92 excess acid was removed by washing with DMW and each sample of material was dried in an  
93 oven at  $50\pm 2^\circ\text{C}$ . The dried material was ground into small granules, sieved and converted  
94 into  $\text{H}^+$  form by treating with 1M nitric acid solution for 24 h with occasional shaking  
95 intermittently replacing the supernatant liquid with fresh acid. The excess of acid was  
96 removed after several washings with DMW and finally the granules of composite materials  
97 were dried at  $50\pm 2^\circ\text{C}$ . Thus, under varying conditions of mixing ratio, pH and temperature,  
98 various samples of each composite material were synthesized (as shown in supplementary  
99 Tables (S-1, S-2 and S-3)). Among the synthesized samples of composite materials, on the  
100 basis of optimum ion uptake capacity, physical appearance of the beads and percentage yield,

101 three samples were selected (as shown in Table 1) that were used for exhaustive studies. The  
102 comparative studies of composite materials were carried out in terms of characterization, ion  
103 exchange capacity and electrical conductivity. The ion exchange capacity and electrical  
104 conductivity of selected samples were determine by using same method as described in  
105 reported paper.<sup>36,37</sup> Among all samples of Table 1, the sample that showed higher ion  
106 exchange capacity and electrical conductivity was chosen for the measurement of membrane  
107 potential in detail.

108

#### 109 2.2.2. Fabrication of hybrid membranes

110 The composite precipitates were powdered and sieved through 200 meshes (Granule  
111 size < 0.07 mm). To improve the stability of the membrane, polystyrene has been used as a  
112 binder as its cross-linked rigid framework provides adequate adhesion to the molecules of  
113 composite which accounts for the high mechanical stability to the membrane over other  
114 binders like polyvinyl chloride (PVC), cellulose acetate, etc. Pure crystalline polystyrene was  
115 also ground to fine powder and sieved through 200 meshes. The precipitates were then mixed  
116 with granulated polystyrene with the help of a pestle and mortar to get ion-exchange  
117 membranes<sup>38</sup> having a varying percentage (by mass) of polystyrene (15-35%). The  
118 thicknesses of the membranes were found to be 0.08 cm when it is applied with a pressure of  
119 146 M.Pa. The membranes having 25% polystyrene were only selected for electrochemical  
120 studies because the membranes containing less or more than 25% polystyrene was found to  
121 be mechanically less stable.

122

#### 123 2.4. Measurement of potential

124 The freshly prepared inorganic-organic nanocomposite membrane was cemented in a  
125 Pyrex glass tube cell having two compartments in which a saturated calomel electrode was

126 placed for measuring the membrane potential. The schematic diagram of the constructed  
127 electrochemical cell of the above type is shown in **Fig. 1.**<sup>39</sup> The monovalent electrolytes of  
128 concentrations  $c_1$  and  $c_2$  in both the compartments of the cell, where the ratio  $\gamma(=c_2/c_1)$  is  
129 fixed at 10, were vigorously stirred by a magnetic stirrer to minimize the effects of boundary  
130 layers on the membrane potential ( $\text{mV} \pm 0.5$ ). The experiment was conducted at room  
131 temperature and atmospheric pressure.

132

### 133 3. Results and Discussion

#### 134 3.1 Synthesis

135 All nanocomposite materials were prepared by employing a very simple sol-gel  
136 chemical route. The optimum conditions for the synthesis of all three composite materials  
137 have been shown in supplementary Tables (S-1, -S2 and S-3). Among all three samples,  
138 PANI-Zr(IV)S (sample V<sub>3</sub>) showed higher ion exchange capacity and electrical conductivity  
139 (Table 1). **The higher ion uptake capacity and electrical conductivity of this material is due to**  
140 **the existence of more number of ionogenic groups as compared to PANI-Ti(IV)W and PANI-**  
141 **Ti(IV)As. On the basis of higher ion exchange capacity and electrical conductivity, PANI-**  
142 **Zr(IV)S nanocomposite was selected for the measurement of membrane potential.**

#### 143 3.2. Characterization

144 To identify the presence of functional groups in all materials FTIR studies were  
145 carried out at room temperature ( $25 \pm 2^\circ\text{C}$ ). The appended FTIR spectra of the composite  
146 materials are shown in **Fig. S-1**. In all the spectra a strong extensive band in the range 3550 to  
147  $3000\text{ cm}^{-1}$  corresponds to the presence of hydroxyl groups.<sup>40</sup> A broadband in the region (1600  
148 to  $1630\text{ cm}^{-1}$ ) is ascribed to aquo H–O–H bending and C–C in plane stretching. The peaks  
149 near  $1442\text{ cm}^{-1}$  correspond to the presence of considerable amounts of polyaniline while the  
150 peaks at  $1385$ ,  $1136$  and  $614\text{ cm}^{-1}$  are recognized as in plane bending vibration of the -CH

151 bands.<sup>41</sup> In each spectrum, a band close to  $500\text{ cm}^{-1}$  may be due to the presence of metal  
152 oxide stretching. In the spectrum of PANI Ti(IV)W, the peak at  $1013\text{ cm}^{-1}$  is owing to  
153 vibration of O=W=O bond.<sup>42</sup> More intense bands at  $811\text{ cm}^{-1}$  may be due to stretching  
154 vibration of Ti–O–W bond. The spectrum of PANI Ti(IV)As, show a medium intensity broad  
155 band at  $871\text{ cm}^{-1}$  is assigned to  $\square$  (As–O–Ti) vibration.<sup>43</sup> In the spectrum of PANI Ti(IV)S  
156 two sharp bands at  $1160$  and  $1030\text{ cm}^{-1}$  are attributed to the  $\text{SO}_3$  asymmetric and symmetric  
157 stretching vibrations of sulphonic acid salts, respectively.<sup>44</sup>

158 In order to get an idea concerning thermal stability of all composite materials, the  
159 TGA analyses were performed (**Fig. S-2**). All spectra show a continuous weight loss of mass  
160 (4, 25 and 10%, respectively) up to  $100^\circ\text{C}$ .<sup>45</sup> The weight loss (4, 7 and 6%, respectively) from  
161  $100^\circ\text{C}$  to  $200^\circ\text{C}$  is by the condensation of -OH groups. Further, weight loss (1, 1 and 6%,  
162 respectively 4.3%) from  $300^\circ$  to  $500^\circ\text{C}$  may be due to complete decomposition of the organic  
163 part of the materials. A gradual decrease in weight loss beyond  $500^\circ\text{C}$  may be due to the  
164 formation of metal oxide. TGA analysis studied shows that among all three materials, PANI-  
165 Ti(IV)W was found to be more thermally stable, due to the presence of tungsten metal which  
166 has a high melting point.

167 X-ray diffraction patterns of PANI-Ti(IV)W and PANI-Ti(IV)As (**Fig. S-3**) show no  
168 sharp peaks of high intensity which suggests amorphous nature while the spectra of PANI-  
169 Zr(IV)S demonstrate some weak intensity peaks which indicate slight crystalline character.  
170 SEM image (**Fig. S-4**) also confirms the amorphous nature of PANI-Ti(IV)W and PANI-  
171 Ti(IV)As and semicrystalline morphology of PANI-Zr(IV)S. TEM studies of each composite  
172 material (**Fig. S-5**) show particles size is in the range of 20-50 nm. To investigate the surface  
173 morphology, thickness, homogeneity, pore structure and cracks of nanocomposite PANI-  
174 Zr(IV)S membrane, SEM with EDEX studies were performed. SEM images (**Fig. 2A, B**)  
175 demonstrate the uniform distribution of organic and inorganic substrates on the surface

176 without any cracks with 2  $\mu\text{m}$  surface thickness. In addition, SEM images do not show  
177 translucent characteristics which indicate heterogeneous nature of the composite membrane.<sup>46</sup>

178 The existence of cavities on the surface of membrane (**Fig. 2A**) suggests the porous  
179 nature of membrane on which  $\text{Li}^+$ ,  $\text{Na}^+$  and  $\text{K}^+$  ions can be easily adsorbed resulting ion  
180 selective nature of the membrane. The cross section photograph of nanocomposite membrane  
181 together with EDEX spectrum and weight percentage of membrane elements are shown in  
182 **Fig. 2 (BCD)**.

### 183 3.3 Measurements of membrane transport potential

184 For the transport properties of the composite membrane, membrane potential values  
185 have been measured. When electrolyte solutions (KCl, NaCl and LiCl) of different  
186 concentrations are separated by a membrane, an electrical potential difference develops  
187 across the membrane due to the development of charge on the membrane surface which is  
188 because of the diffusion of oppositely charged ions through it with different mobilities. This  
189 charge imparts some important electrochemical properties of the membrane, the most  
190 important being the differences in the permeabilities of co-ions, counter ions and neutral  
191 molecules. The quantity of charge required to generate the potential is small, especially when  
192 dilute solutions are used. Membrane potential has been measured for the composite  
193 membranes equilibrated with various monovalent electrolyte solutions of concentrations  
194 ranging between 1 and  $0.007\text{mol/dm}^3$ . The membrane potential,  $\psi_m$  data obtained with the  
195 composite membrane using different electrolytes are plotted as a function of  $1/c_1$  (**Fig. 3A**).  
196 The magnitude of the membrane potential is dependent on many factors such as the applied  
197 pressure at the membrane preparation stage, concentration of electrolyte solutions used, the  
198 counter-ion to co-ion mobility ratio, the exchange characteristics of the membrane material  
199 for various cations. It was observed that the hybrid membrane prepared at higher applied  
200 pressure exhibited higher membrane potential for a fixed  $c_2/c_1$ . The values of membrane

201 potentials observed across the membrane in contact with various monovalent electrolytes  
202 show the potential order to be  $K^+ > Na^+ > Li^+$ .

203 It was observed that during the experiment, the membrane potential increased with  
204 time up to a maximum value and then decreased gradually. Such variation was also reported  
205 by other workers.<sup>47</sup> The time taken for the attainment of maximum potential is found to be  
206 concentration dependent, more time being required for concentrated solution, and it differs  
207 with different electrolytes. Membrane potential has been regarded as a measure of membrane  
208 selectivity for a long time. The measurement of ion activity by means of a membrane  
209 electrode is most successful in the concentration range over which the membrane behaves as  
210 ideally permselective and obeys the Nernst equation. An ideally permselective membrane is  
211 one in which the permeability for co-ions is negligible as compared to that for counter-ions.

212 The deviation from Nernstian behavior is due to the co-ion transference<sup>48</sup> and the  
213 dependence of the exchange of cations between the solution and the membrane phase and on  
214 the electrolyte concentration. The values in **Fig. 3A** show that the potential values increase  
215 with decrease in concentration of all the tested electrolyte solutions and are of the order of  
216 positive mV indicating that the membrane is negatively charged i.e. cation selective. The  
217 selectivity increases with a decrease in the concentration because of the structural changes  
218 occurring in the electrical double layer at the membrane-solution interface<sup>49</sup>. The increase in  
219 selectivity with dilution is also supported by the increasing value of the counter-ion transport  
220 numbers,  $t_+$  (**Table 2**).

221 When an ionic gradient is maintained by using two solutions of different  
222 concentrations of same electrolyte on both sides of the membrane, the mobile species  
223 infiltrate the membrane at different magnitudes inducing various transport phenomena into  
224 the system, including the development of potential across it. The influence is greater in case  
225 of counter-ions than in the co-ions. The ratio of the molar mobilities of the cation and anion

226 ‘ $u_+ / u_-$ ’ is defined as the mobility ratio ( $\bar{\omega}$ ) of the membrane. The values of the mobility  
 227 ratio calculated for the composite membrane are also represented in **Table 2**. The values of  
 228  $\bar{\omega}$  in the membrane phase were found increasing with decrease in concentration for all the  
 229 monovalent electrolytes used (KCl, NaCl and LiCl). The high mobility is attributed to a  
 230 higher transport number of comparatively free cations as compared to the anion of  
 231 electrolytes.

232 When a negatively charged membrane is imposed between two solutions of a  
 233 monovalent electrolyte of unequal concentrations  $c_1$  and  $c_2$  ( $c_2 > c_1$ ), the observed membrane  
 234 potential,  $\psi_m$  is related with electrolyte concentration by the following equation.<sup>31,33</sup>

$$235 \quad -\psi_m = \frac{RT}{F} \left( \frac{\gamma}{\gamma-1} \right) \left( \frac{\phi X}{2} \right) \frac{1}{c_1} \quad (1)$$

236 Here,  $\phi X$  is the effective fixed charge density of the negatively charged membrane. Eqn. (1)  
 237 indicates that the plot of  $\psi_m$  against  $1/c_1$  will be linear (**Fig. 3A**) with a slope equal to  
 238  $\frac{RT}{F} \left( \frac{\gamma}{\gamma-1} \right) \left( \frac{\phi X}{2} \right)$  from which the values of  $\phi X$  for different electrolytes have been evaluated.

239 The calculated values are represented in **Table 3**.

240 Another widely accepted approach to calculate the fixed charge density was derived  
 241 by Kobatake *et al.*<sup>32,33</sup> who suggested that when two solutions of an electrolyte of different  
 242 concentrations  $c_1$  and  $c_2$  are separated by a membrane, the inverse of the apparent transport  
 243 number of the anion,  $t_{-app}$  in a high salt concentration range could be expressed by the  
 244 following equation:

$$245 \quad \frac{1}{t_{-app}} = \frac{1}{1-\alpha} + \frac{\alpha(\gamma-1)}{(1-\alpha)\gamma \ln \gamma} \left( \frac{\theta \bar{X}_c}{c_1} \right) \quad (2)$$

246 Here,  $\alpha$  is the ratio of molar mobility of cation to the sum of molar mobilities of cation and  
 247 anion,  $\theta\bar{X}_c$  is the effective fixed charge density of the membranes under investigation and  $c_1$   
 248 (in mol/dm<sup>3</sup>) the concentration of the monovalent electrolyte in the lower concentration side  
 249 of the cell.

250 The apparent transport number of the anion  $t_{-app}$  is defined by the Nernst equation:

$$251 \quad \psi_m = (RT/F)(1 - 2t_{-app}) \ln \frac{c_2}{c_1} \quad (3)$$

252 Eqn. (2) indicates that the values of  $\alpha$  and  $\theta\bar{X}_c$  can be evaluated by using the values of  
 253 intercept ( $\frac{1}{1-\alpha}$ ) and slope ( $\frac{\alpha(\gamma-1)}{(1-\alpha)\gamma \ln \gamma} \theta\bar{X}_c$ ) from the linear plot of  $1/t_{-app}$  against  
 254  $1/c_1$  (**Fig. 3B**). The calculated values of the fixed charge densities  $\theta\bar{X}_c$  of the polystyrene  
 255 blended calcium tungstate composite membrane are also given in **Table 3**. The fixed charge  
 256 density of the hybrid membrane under different electrolytic environments is found to be in  
 257 the order KCl > NaCl > LiCl. The data in **Table 3** show that the fixed charge density is  
 258 highest for KCl and lowest in the case of LiCl for the same electrolytic concentration,  
 259 indicating that the composite membrane shows higher cation selectivity towards K<sup>+</sup> ions. The  
 260 same result has also been explained in terms of counter-ion transport numbers.

#### 261 4. Conclusions

262 PANI based nanocomposite materials have been successfully synthesized by using a  
 263 simple sol-gel route. XRD and SEM analysis revealed that PANI-Ti(IV)W and PANI-  
 264 Ti(IV)As showed amorphous while PANI-Zr(IV)S demonstrated semi crystalline nature. The  
 265 particle size of each material was found to be in the range of 20-50 nm by TEM analysis.  
 266 Thermally and chemically stable materials covered semiconductor range together with good  
 267 ion uptake capacity and can be used as an ion exchanger as well as conducting materials.  
 268 Among three materials, PANI-Zr(IV)S demonstrated higher ion exchange capacity together

269 with electrical conductivity. The results of transport properties of the composite membrane  
270 indicate that the fixed charge density values were in the order  $\text{KCl} > \text{NaCl} > \text{LiCl}$  and the  
271 membrane was found to be selective for  $\text{K}^+$  ions. Membrane transport results indicate that  
272 besides behaving as ion exchanger, conducting material, photocatalyst and antimicrobial  
273 agent, PANI-Zr(IV)sulphosalicylate can also be successfully used as an ion selective  
274 membrane for the detection of  $\text{K}^+$  ions.

275

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279

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281

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370 **Table 1: Selected samples of PANI based nanocomposite materials at 25±2°C (pH= 1.0).**

S.No	Composite materials	Colour	IEC (meq g <sup>-1</sup> ) for Na <sup>+</sup> ions	Conductivity (S cm <sup>-1</sup> )
V <sub>1</sub>	PANI Ti(IV)W	Green granules	0.85	0.168
V <sub>2</sub>	PANI Ti(IV)As	Green granules	1.37	0.178
V <sub>3</sub>	PANI Zr(IV)S	Green granules	1.80	0.278

371 Ion exchange Capacity (IEC)

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384 **Table 2:** Calculated values of transport number ( $t_+$ ) and mobility ratio ( $\bar{\omega}$ ) of the  
 385 nanocomposite membrane from the measured membrane potential values with various  
 386 monovalent electrolytes at different concentrations with  $c_2/c_1 = 10$  at an applied pressure  
 387 146 MPa at  $25 \pm 1$  °C.

Electrolyte	$c_2$ (mol/dm <sup>3</sup> )	$t_+$	$\bar{\omega}$
<b>KCl</b>	1	0.628	1.6893
	0.7	0.641	1.7870
	0.5	0.656	1.9103
	0.25	0.682	2.1492
	0.1	0.778	3.5050
	0.07	0.848	5.5588
<b>NaCl</b>	1	0.609	1.5550
	0.7	0.617	1.6130
	0.5	0.630	1.7051
	0.25	0.661	1.9475
	0.1	0.754	3.0672
	0.07	0.826	4.7409
<b>LiCl</b>	1	0.602	1.5131
	0.7	0.613	1.5836
	0.5	0.617	1.6130
	0.25	0.652	1.8739
	0.1	0.739	2.8304
	0.07	0.804	4.1044

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394 **Table 3:** Comparison of the values of the effective fixed charge densities, calculated  
395 from different methods, of the nanocomposite membrane in contact with different  
396 electrolytes.

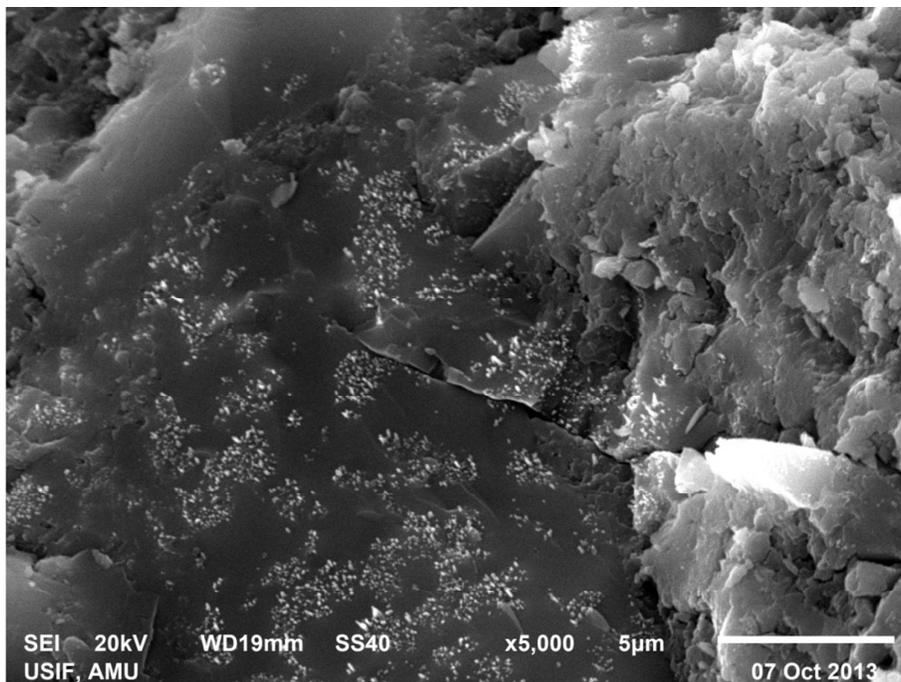
Electrolyte	KCl	NaCl	LiCl
$\phi X^{\text{c}}$	0.0126	0.0125	0.0108
$\theta X_c^{\text{@}}$	0.0165	0.0161	0.0138

397 <sup>c</sup> from the slope of **Fig. 7A** with **Eqn. (1)**.

398 <sup>@</sup> from the slope of **Fig. 7B** with **Eqn. (2)**.

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SEM image of PANI-Zr(IV)S nanocomposite membrane

Figure captions:

Fig. 1 Schematic diagram for the measurement of membrane potential.

Fig. 2 SEM images of nanocomposite membrane. Surface image (A), cross sectional image (B), EDAX spectrum (C) and percentage content (D) of PANI-Zr(IV)S nanocomposite membrane.

Fig. 3 Plots of (A) potential values,  $\psi_m$  and (B)  $1/t_{app}$  against  $1/c_1$  of polystyrene based composite membrane for different monovalent electrolytes.

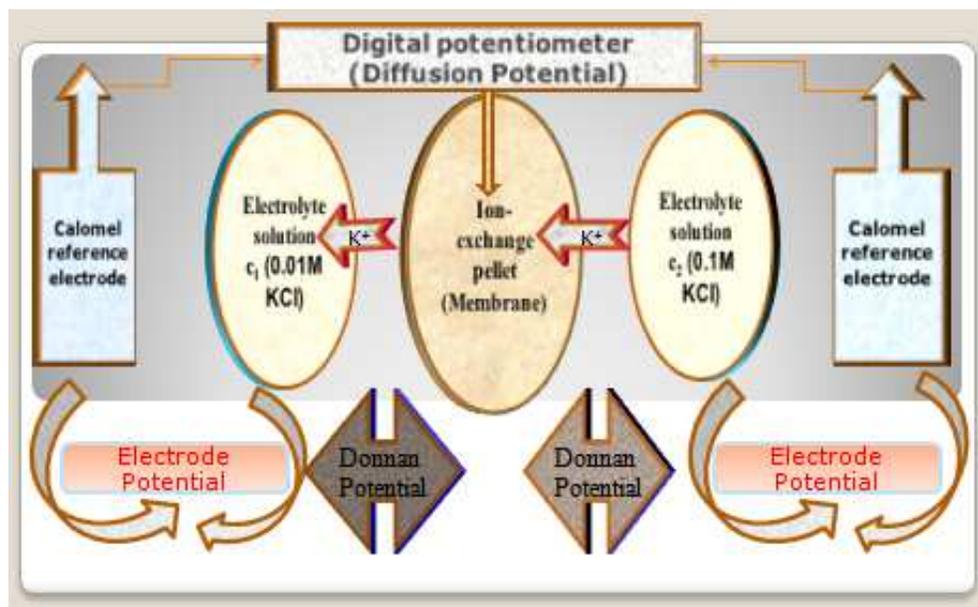


Fig. 1

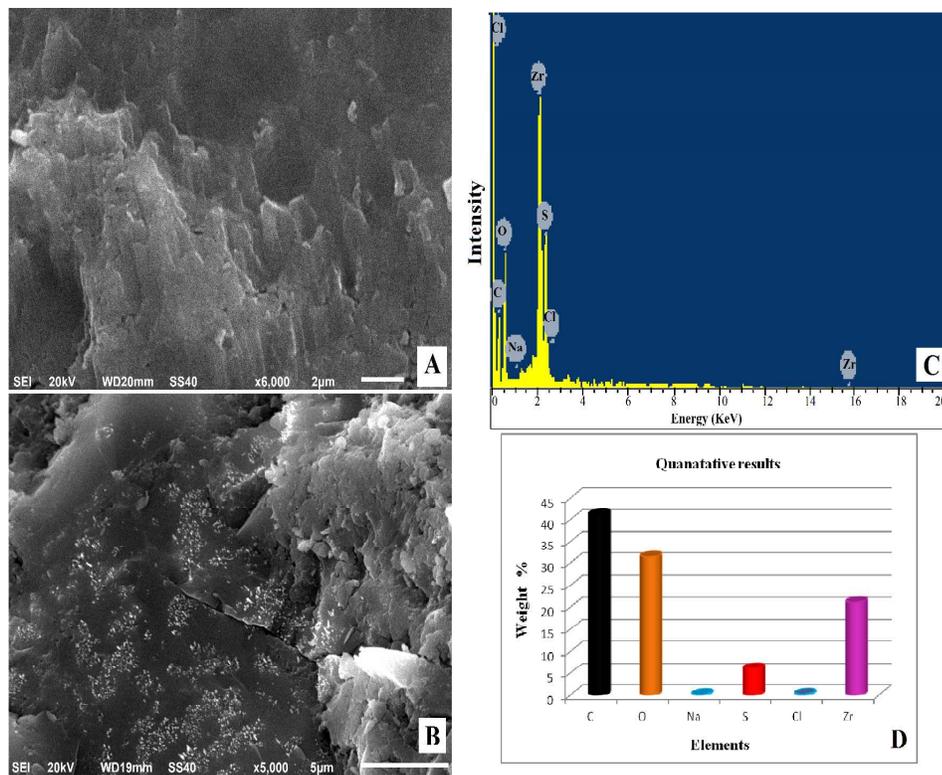


Fig. 2 (ABCD)

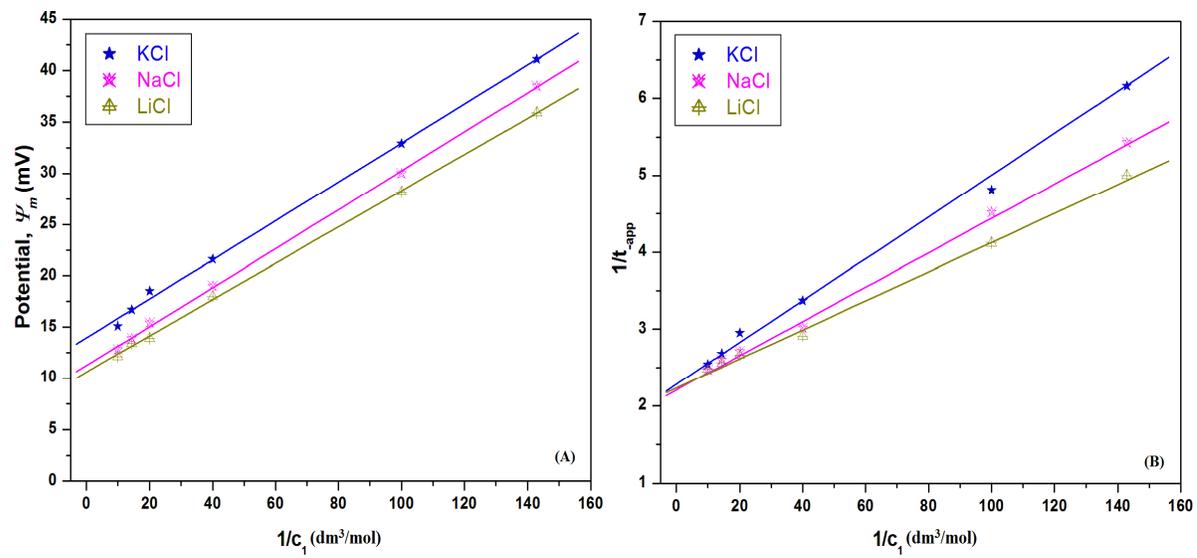


Fig. 3 (A, B)

**Supporting Information****A comparative study for the characterization of polyaniline based nanocomposites and membrane properties**

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Table S-1 Conditions for the synthesis of PANI-Ti(IV)W cation exchanger.

S.No	A	B	C (%) v/v	Mixing ratio v/v/v	Temperature (°C)	pH	Appearance of bead	IEC for Na <sup>+</sup>	Yield (g)
T-1	0.25	0.1	--	1:1	25±2	0.5	White granular	0.25	1.68
T-2	0.25	0.1	--	1:1	25±2	1.0	White granular	0.60	1.96
T-3	0.25	0.1	--	1:1	100±2	1.0	White granular	0.40	1.85
T-4	0.25	0.1	10	1:1	25±2	1.5	Green granular	0.35	2.95
T-5	0.25	0.1	10	1:1:1	25±2	0.5	Green granular	0.65	2.33
T-6	0.25	0.1	10	1:1:1	25±2	1.0	Green granular	0.85	2.99
T-7	0.25	0.1	10	1:1:1	100±2	1.0	Green granular	0.70	3.00
T-8	0.25	0.1	10	1:1:1	25±2	1.5	White granular	0.55	3.25

Sodium tungstate (Mol L-1), B. Titanium tetrachloride (Mol L-1), C. (stock solution of 10 % PANI), IEC. Ion Exchange Capacity (meq g-1)

Tables S-2 Conditions for the synthesis of PANI-Ti(IV)As cation exchanger.

S.No	A	B	C (%) v/v	Mixing ratio v/v/v	Temperature (°C)	pH	Appearance of bead	IEC for Na <sup>+</sup> ions	Yield (g)
A-1	0.25	0.1	--	1:1	25±2	0.5	White granular	0.75	1.62
A-2	0.20	0.1	--	1:1	25±2	1.0	White granular	0.99	1.96
A-3	0.25	0.1	--	1:1	100	1.0	White granular	0.70	2.01
A-4	0.25	0.1	--	1:1	25±2	1.5	Green granular	1.00	2.90
A-5	0.25	0.1	10	1:1:1	25±2	0.5	Green granular	1.10	2.30
A-6	0.25	0.1	10	1:1:1	25±2	1.0	Green granular	1.37	3.99
A-7	0.25	0.1	10	1:1:1	100	1.0	Green granular	1.00	3.30
A-8	0.25	0.1	10	1:1:1	25±2	1.5	White granular	0.90	3.45

A. Sodium arsenate (Mol L<sup>-1</sup>), B. Titanium tetrachloride (Mol L<sup>-1</sup>), C. (stock solution of 10 % PANI), IEC: Ion Exchange Capacity (meq g<sup>-1</sup>)

Table S-3 Conditions for the synthesis of nanocomposite PANI-Zr(IV)sulphosalicylate cation exchanger at 25± 2 °C.

S. No	A	B	C (%)	Mixing ratio v/v/v	pH	Appearance of bead	IEC for Na <sup>+</sup> ions	Yield (g)
S-1	0.25	0.5	--	1:1	0.5	White granular	---	----
S-2	0.25	0.5	--	1:1	1.0	White granular	1.05	2.46
S-3	0.25	0.5	--	1:1	1.5	White granular	0.70	2.50
S-4	0.25	0.5	10	1:1:1	0.5	Green granular	1.25	2.54
S-5	0.25	0.5	10	1:1:1	1.0	Green granular	1.80	3.54
S-6	0.25	0.5	10	1:1:1.5	1.0	Green granular	1.24	2.94
S-7	0.25	0.5	10	1:1:1	1.5	Green granular	1.20	3.90
S-8	0.25	0.5	12	1:1:1	1.0	Light green granular	1.54	2.90
S-9	0.25	0.5	15	1:1:1	1.0	Dark green granular	1.40	2.98

A. Zirconium oxychloride (Mol L<sup>-1</sup>), B. Sulphosalicylate (Mol L<sup>-1</sup>), C. (stock solution of 10 % PANI), IEC: Ion Exchange Capacity (meq

g<sup>-1</sup>)

**Figure captions:**

**Fig. S-1 FTIR spectra of nanocomposite materials.**

**Fig. S-2 TGA curves of nanocomposite materials.**

**Fig. S-3 Powder XRD patterns of nanocomposite materials.**

**Fig. S-4 SEM images of nanocomposite materials.**

**Fig. S-5 TEM images nanocomposite materials.**

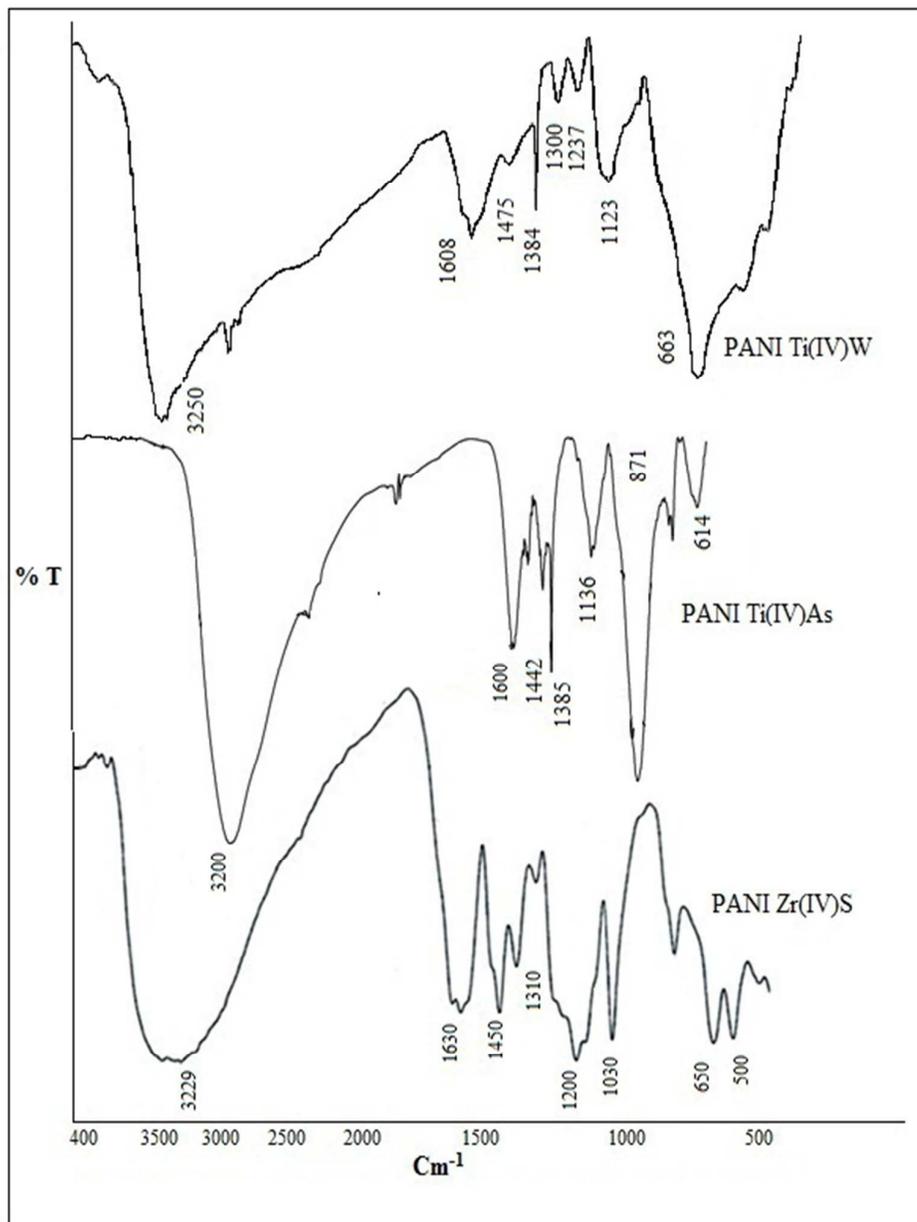


Fig. S-1

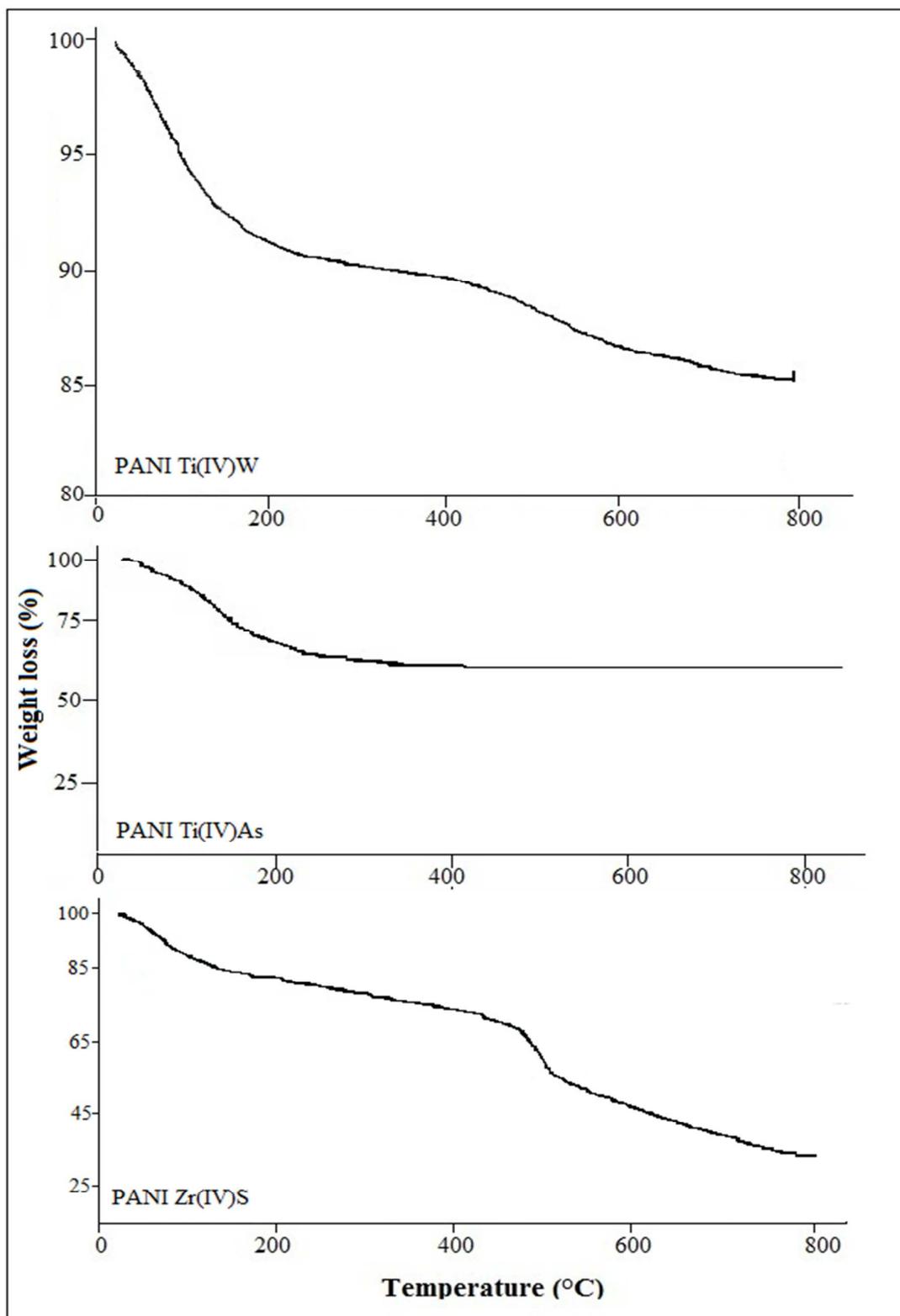


Fig. S-2

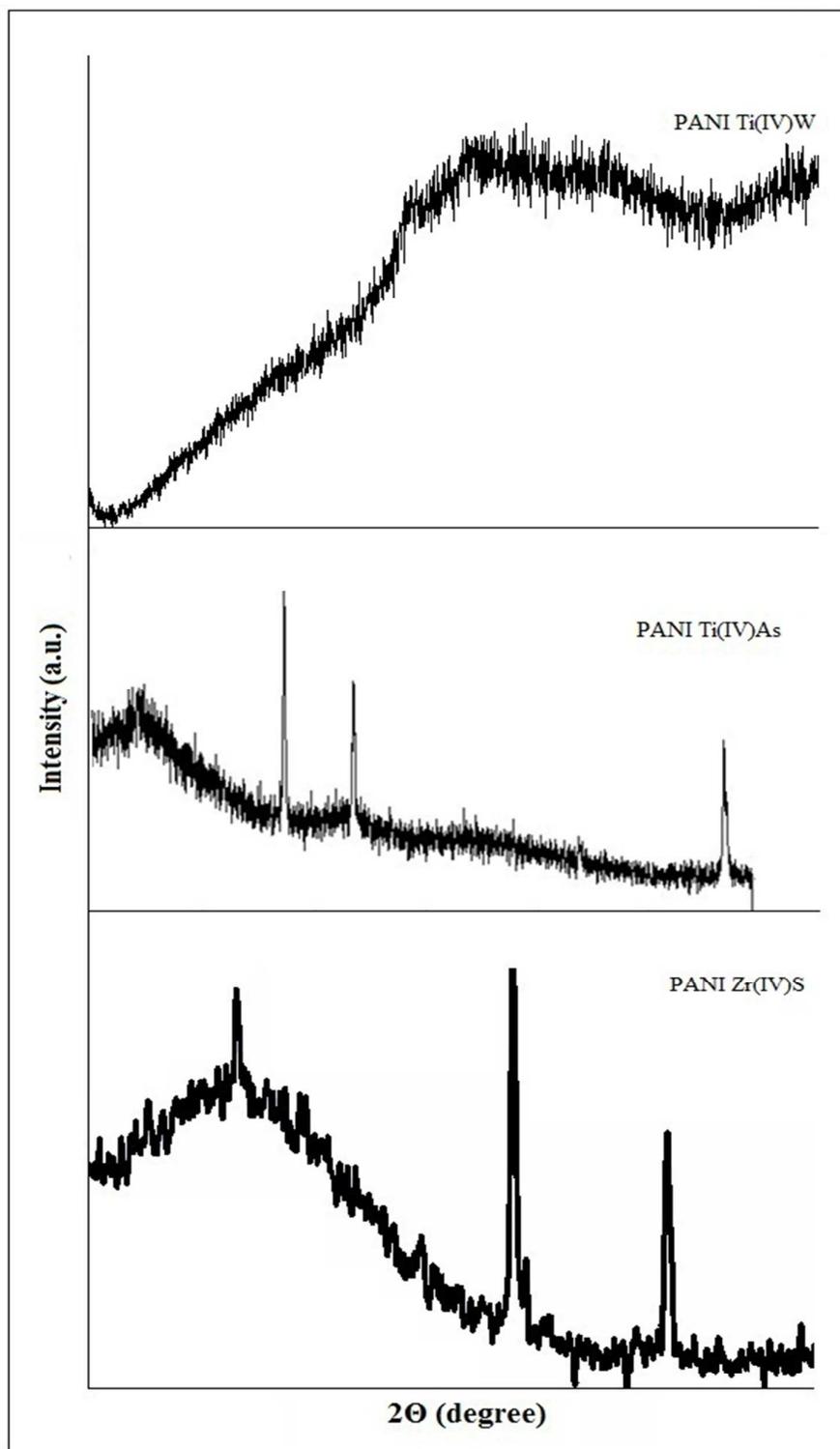


Fig. S-3

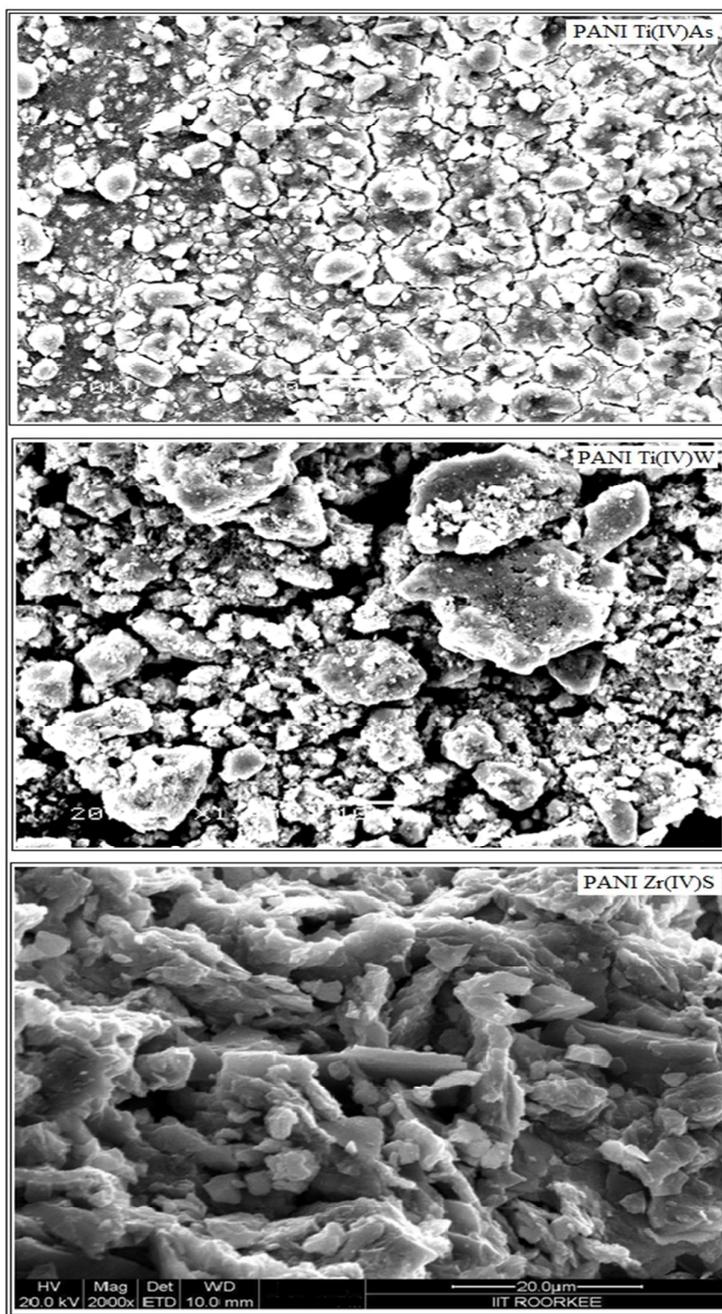


Fig. S-4

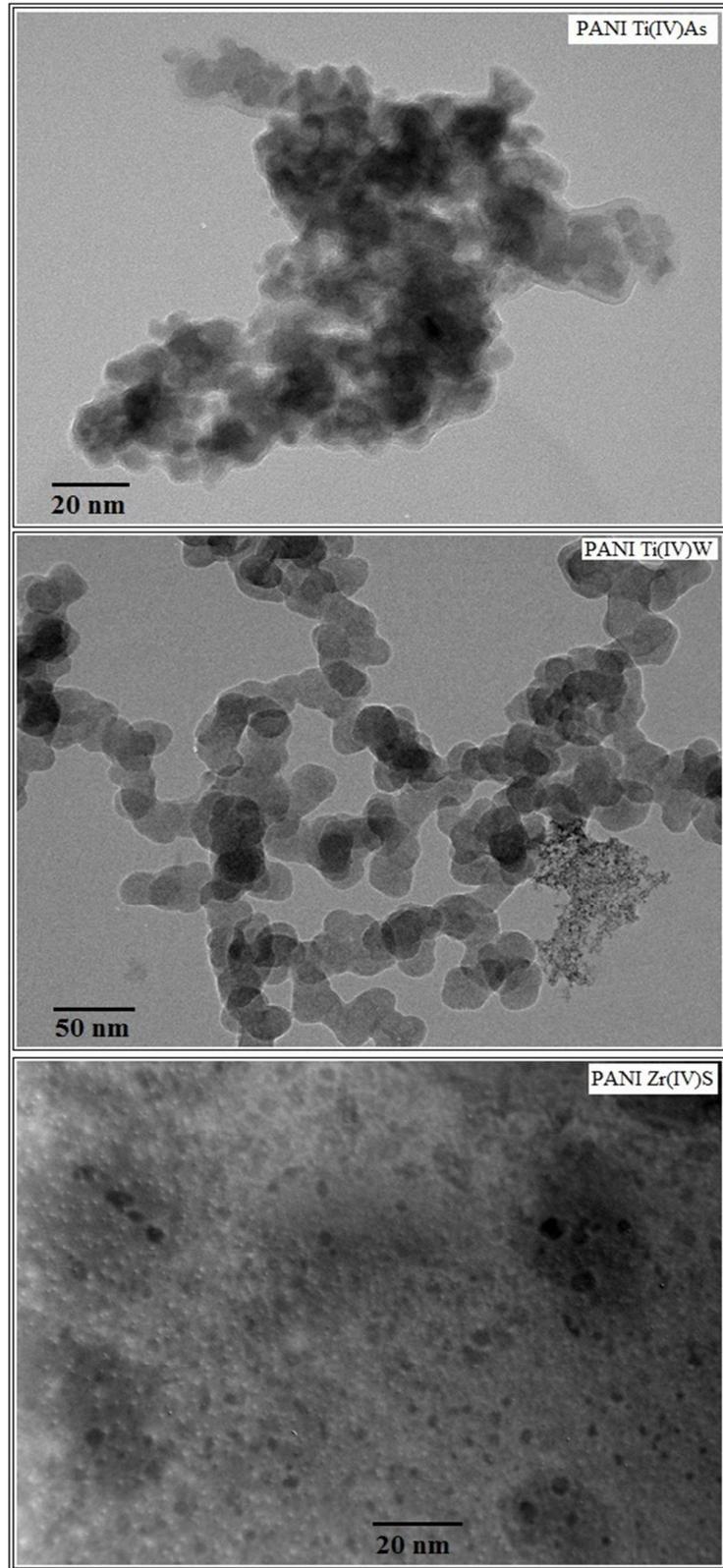


Fig. S-5