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Exploring Mechanochemical Reactions at the Nanoscale: Theory Versus Experiment

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Abstract

Mechanochemical reaction pathways are conventionally obtained from the force-displaced stationary points on the potential energy surface of the reaction. This work tests a postulate that the steepest-descent pathway (SDP) from the transition state to the reactant can be used instead with reasonably accurately to calculate mechanochemical reaction kinetics. This method is much simpler because the SDP and the associated reactant and transition-state structures can be obtained relatively routinely. Experiment and theory are compared for the normal-stress-induced decomposition of methyl thiolate species on Cu(100). The mechanochemical reaction rate is calculated by compressing the initial- and transition-state structures by a stiff copper counter-slab to obtain plots of energy versus slab displacement for both structures. The reaction rate is also measured experimentally under compression using a nanomechanochemical reactor comprising an atomic-force-microscope (AFM) tip compressing a methyl thiolate overlayer on Cu(100), the same system for which the calculations were carried out. The rate is measured from the indent created on a defect-free region of the methyl thiolate overlayer, which also enables the contact area to be measured. Knowing the force applied by the AFM tip yields a reaction rate as a function of the contact stress. The result agrees well with the theoretical prediction without the use of adjustable parameters. This confirms that the postulate is correct and will facilitate the calculation of the rates of more complex mechanochemical reactions. An advantage of this approach, in addition to the results agreeing with experiment, is that it provides insights into the effects that control mechanochemical reactivity that will assist in the targeted design of new mechanochemical syntheses.

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Introduction

The rates of chemical reactions can be accelerated in several ways: by heating, by photons,² by electrons,³ or by supplying mechanical energy in a field known as mechanochemistry. Despite mechanochemical reactions having been known for millennia,⁴ and having been studied by luminaries such as Michael Faraday,⁵ the subfield of mechanochemistry remains much less well developed than the rest. However, over the past few years, a large number of active and selective organic and inorganic syntheses have been discovered by trial and error.^{6, 7} Methods that combine mechanochemical activation with catalysis have also achieved remarkable results such as the ability to synthesize ammonia from hydrogen and nitrogen at room temperature and modest pressures⁸⁻¹¹ to mimic the Haber-Bosch process that typically occurs at pressures of 200 to 400 atmospheres and temperatures of ~500 °C.^{12, 13}

Single-molecule mechanochemical experiments have been carried out by attaching the reacting system (known as a mechanophore) to an atomic force microscopy (AFM) tip and by measuring the force needed to extend, and finally cleave, one or more bonds.¹⁴⁻²¹ Because of the ability to accurately measure these forces, this approach can lead to precise measurements of force-induced bond scission rates. Many biological processes also involve forces exerted by or acting on molecular systems.^{22, 23 24-26}

One of the first theoretical approaches to analyzing the results of single-molecule experiments was the <u>constrained geometries simulate external force (CoGEF)²⁷ method, which displaces the attachment (AP) and pulling points (PP) and then allows the perturbed system to relax to a new stable configuration to calculate the energy as a function of the force.²⁸⁻³⁰ Analogous approaches include the <u>force-modified potential energy surface (FMPES) method</u>,^{31, 32} in which the PES is modified by directly including the effect of the forces on the AP and PP. Mechanochemical</u>

reaction rates under hydrostatic pressure have been analyzed by the generalized force-modified potential energy surface (G-FMPES)³²⁻³⁴ and the EFEI (<u>external force is explicitly included</u>) methods.³⁵

Chemical reactions are analyzed in the framework of a Born-Oppenheimer potential-energy surface (BOPES). This complicated, multi-dimensional surface contains local energy minima due to atomic configurations that correspond to stable chemical compounds. The conversion between them constitutes chemical reactions and requires the transition over an energy barrier otherwise the reaction would occur spontaneously. The lowest-energy route between two metastable states (a reactant and product) passes over a saddle point, known as the transition state, via a so-called activated complex. This concept allows a complicated multidimensional problem to be reduced to one in one dimension. The quickest reaction path from the activated complex to the reactant is called a reaction coordinate and is taken to be the steepest-descent pathway (SDP). Methods for calculating transition-state structures and the SPD are common and easy to implement.³⁶⁻⁴⁰

The general physical concepts that underpin mechanochemistry are based on the idea that the imposed force modifies the Born-Oppenheimer potential-energy surface (BOPES) of the chemical reaction to change the energies and locations of the initial and transition states. This modifies the activation energy, thereby changing the reaction rate. Similar concepts also explain friction and viscosity.⁴¹ The way in which the BOPES is modified by the force depends on the direction and magnitude of the force. The reaction then follows the SDP on this modified BOPES along the so-called force-displaced stationary points (FSDPs).⁴²⁻⁴⁵ However, calculating a FDSP for the chemical reaction requires knowing the shape of the BOPES, which can be tedious to compute. It was recently proposed that mechanochemical reactions could be analyzed with good accuracy using the (SDP) and experimental evidence was presented to support this postulate.⁴⁶ Since the

SDP is relatively easy to compute, this provides an efficient approach to calculating mechanochemical reaction rates. This idea is tested in this paper.

An additional challenge for mechanochemical theories, in particular, for stress-induced reactions as opposed to those carried out by pulling single molecules, is to exert precise stresses on well-defined systems and to accurately measure their reaction rates. We address this issue by using a nanomechanochemical reactor of about 60 nm² in size ⁴⁷ formed by compressing a reactant-covered single-crystal substrate using an atomic force microscope (AFM) tip.^{41, 48} This allows elastic contact mechanics to be used to calculate the stresses at the center of the contact ⁴⁹ and to ensure that the reaction rate is measured on pristine regions of the surface.^{1, 47} We study the surface mechanochemical decomposition of methyl thiolate (CH₃–S) on Cu(100) where the mechanical reaction pathway has been extensively investigated under ultrahigh-vacuum conditions, which we have previously demonstrated can decompose under the influence of normal stresses.^{1, 50, 51} This model system is sufficiently simple that it is amenable to analysis using first-principles density-functional theory (DFT) calculations. The results of calculations obtained by using the SDP successfully reproduce the experimental results, confirming the validity of our initial postulate.

Background: The Mechanochemical Reaction Pathway

The mechanochemical reaction pathway for methyl thiolate decomposition on clean Cu(100) has been extensively investigated in ultrahigh vacuum (UHV).^{50, 52-56} The methyl thiolate species is grafted to Cu(100) by exposure to dimethyl disulfide,⁵⁷ and is thermally stable up to ~450 K. Lateral shear and normal stresses accelerate the reaction rate by lowering the activation barrier from ~100 kJ/mol to a value that is low enough that the reaction proceeds at room temperature.^{1, 47} The calculated energy profile for methyl thiolate decomposition is shown in Figure 1, where the

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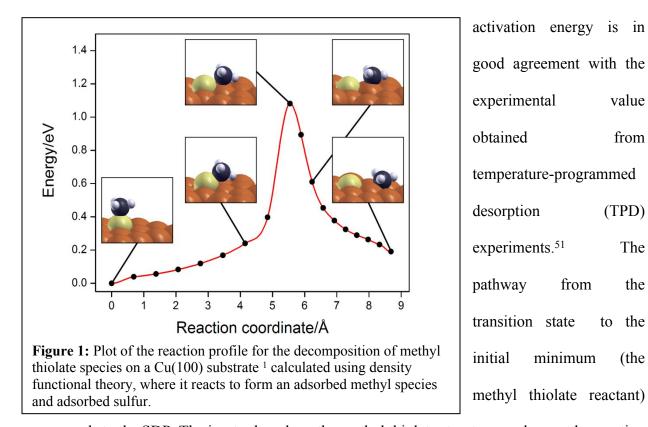
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corresponds to the SDP. The insets show how the methyl thiolate structure evolves as the reaction proceeds. This is also shown as a movie in Figure S1. This process occurs by the terminus of the methyl thiolate mechanophore moving both laterally and vertically with respect to the surface. Note that the reaction is still thermally driven and mechanochemical reaction rates depend on temperature. This predicts that both normal and lateral stresses should be effective in lowering the energy barrier; this work investigates the influence of normal stresses. Note that the evolution of the electronic structure during the course of a mechanochemical reaction is the same as for the thermal one.14

This work focusses just on normal-stress accelerated reactions, not those that are induced by a shear stress. The latter are more complicated because, unless the experiment is designed to impose a static shear, the rates will be velocity (as well as stress and temperature) dependent.⁵⁸⁻⁶⁰

Theoretical and Experimental Methods

Density functional theory (DFT) calculations of methyl thiolate overlayers on Cu(100) were performed using the projector augmented wave method $^{61, 62}$ as implemented in the Vienna *ab initio* simulation package, VASP.⁶³⁻⁶⁵ The exchange-correlation potential was described using the generalized-gradient approximation of Perdew, Burke and Ernzerhof.⁶⁶ A cutoff of 400 eV was used for the planewave basis set, and the wavefunctions and electron density were converged to within 1×10^{-6} eV. The first Brillouin zone was sampled with a $4 \times 4 \times 1$ Monkhorst-Pack grid.⁶⁷ Geometric relaxations were considered to be converged when the force was less than 0.01 eV/Å on all unrestricted atoms. Van der Waals' interactions were implemented using the DFT-D3 method as described by Grimme et al.⁶⁸ The reaction profile and the transition-state structure (as shown in Figure 1) were calculated by the climbing nudged-elastic band (cNEB) method.^{36, 37, 40}

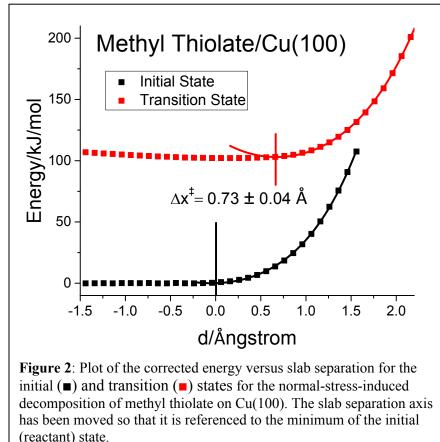
Normal loads were exerted on the initial- and transition-state structures of methyl thiolate on a Cu(100) slab with a lattice constant of 3.575 Å as shown in the movie in Figure S2. The system consisted of two (2×2) Cu(100) slabs to mimic the experimentally measured saturation coverage of alkyl thiolates on Cu(100).⁶⁹ The initial- and the transition-state alkyl thiolate structures were adsorbed onto the bottom 6-layer slab. The bottom three layers of that slab were frozen, while the top three layers were allowed to relax. A counterface slab located above the adsorbate-covered substrate was used to exert a force on the thiolate overlayer to mimic the AFM compression experiment. The upper slab was three copper atomic layers thick and was passivated by hydrogen atoms located in the four-fold hollow sites of the slab to render the interacting surface chemically inert. All atoms in this slab were kept frozen to simulate an infinitely stiff surface. The initial coordinates of the system at a large initial separation are include in the Supporting Information section. Compression of the methyl thiolate molecule was simulated by translating the hard slab

in 0.1 Å steps along a direction perpendicular to the plane of the methyl-thiolate-covered slab. Decompression was simulated by raising the slab, also in 0.1 Å steps. Because the slabs were not identical, the surface charges induced by the different locations of the Fermi energies caused a long-range electrostatic interaction that resulted in an approximately 1/d energy dependence, where *d* is the slab separation. The resulting energy versus distances curves are shown in Figure S3 prior to the removal of the background, which was too large to be removed using the dipole-correction subroutine in the VASP software. Thus, the background was removed by carrying out separate single-point calculations of approaching slabs without an adsorbed methyl thiolate overlayer. Calculations also determined the strength of the interaction between the adsorbed methyl thiolate and the top slab. These interactions are discussed in greater detail in the Supporting Information section. The sum of the interaction energies was subtracted from the approach curves with methyl thiolate present to lead to flat regions as the two slabs approached until they reached a repulsive region for which the energy increased approximately quadratically with decreasing slab separation. These calculations were performed for both the reactant and transition-state structures.

The extent of reaction is measured from the depths of the indents formed by pressing the AFM tip on the methyl-thiolate-saturated Cu(100) surface as a function of time, where the surface was imaged at a low, non-perturbative load.¹ Previous work has shown that similar methyl thiolate reaction pathways as discussed above are also induced by an AFM tip on a Cu(100) surface in UHV.⁴⁶ An example plot is shown in Fig. S8, which shows that the maximum depth is ~200 nm, similar to the height of a methyl thiolate species on copper. A fit to the data yields a first-order, stress-dependent rate constants, $k(\sigma)$, identical to the reaction order found for ball-on-flat sliding in UHV.⁵⁶ The normal stress is calculated from the diameter of the indent to gauge the contact area and the normal force exerted on the tip as described in the Supporting Information section.

Results: Normal-Stress Induced Decomposition of Methyl Thiolates on Cu(100)

We tested the postulate that the reactant and transition-state structures connected by the SDP can be used to calculate mechanochemical reaction rates. The resulting calculated energies of the compressed methyl thiolate reactant (**■**) and the transition-state (**■**) structures are plotted as a



function of slab separation 2, where the in Fig. abscissa is shifted so that the origin coincides with the point of contact of the initial state (methyl thiolate) with the counterface. A video of the structural evolution is depicted in Figure S2.

The energy difference at large separations is ~104

kJ/mol, in agreement with the calculated (Fig. 1) and the experimental reaction activation energies measured by TPD.⁵¹ The energies stay constant until the rigid slab encounters the adsorbed species where the methyl thiolate reactant contacts the counterface slab at larger separations than the geometrically lower transition-state structure at $x = 0.73 \pm 0.04$ Å. The applied normal force multiplied by this distance is the work done in going from the reactant to the activated complex and corresponds to an activation length, $\Delta x^{\ddagger.48}$ This is conceptually analogous to the definition used in single-molecule pulling experiments where it is the difference between the AP and PP in the initial and transition states.²⁶

This leads to a formula for the force-dependent activation energy: $E_{act}(F_N) = E_{act}^0 + \Delta x^{\dagger} F_N$, where F_N is the normal force and E_{act}^0 is the intrinsic activation energy. This is known as the Bell model.⁷⁰ Note that the value of Δx^{\dagger} depends on the direction of the force relative to reactant and transition-state structures.

The results in Fig. 2 indicate that these interactions are compliant and that the activation length itself depends on the force. Assuming that the distortion is harmonic, as shown by the solid lines fitted to the theoretical results in Fig. 2, results in the so-called extended-Bell model:^{48, 71-73}

$$E_{act}(F_N) = E_{act}^0 + \Delta x^{\dagger} F_N + \frac{F_N^2}{2} (\chi_T - \chi_I)$$
(1),

where χ_T (= 1/ k_T) and χ_I (= 1/ k_I) are the transition- and initial-state compliances and k_T and k_I are the corresponding force constants, where $k_I = 16.8 \pm 0.2$ N/m and $k_T = 25.8 \pm 0.2$ N/m. The average normal force exerted on each reactant, F_N , is calculated from the normal stress σ_{zz} , where the *z* axis is taken to be perpendicular to the surface, by $\sigma_{zz} = F_N/A_C$ where A_C is the area subtended by each methyl thiolate adsorbate over which the normal stress acts. Since $\Delta x^{\pm}F_N = \Delta x^{\pm}A_C\sigma_{zz}$, this yields a value of the activation volume $\Delta V^{\pm} = A_r\Delta x^{\pm}$, as first proposed by Stearn and Eyring.⁷⁴ A value of $A_C = 26.12$ Å² is calculated from the area occupied by a (2×2) methyl thiolate overlayer on Cu(100)⁶⁹ and leads to $\Delta V^{\pm} = \Delta x^{\pm}A_C = -19.1 \pm 1.0$ Å³/molecule. This indicates that the activation volume does not just depend on the nature of the reaction pathway, but also on the direction of the force relative to it. In this case, the activation volume comprises a cylinder of length ~0.73 Å perpendicular to the surface with a cross-sectional area of ~26 Å² parallel to it. The form of Eqn. 1 is similar to those derived using the FDSP. However, the formula in Eqn. 1 is calculated from data for the steepest-descent pathway. Eqn. 1 can written directly in terms of the normal stress, σ_{zz} to mimic the experimental conditions by using the Stearn-Eyring postulate to give:

$$E_{act}(\sigma_{zz}) = E_{act}^{0} + \Delta x^{\dagger} A_{C} \sigma_{zz} + \frac{A_{C}^{2} \sigma_{zz}^{2}}{2} (\chi_{T} + \chi_{I}) = E_{act}^{0} + \Delta V^{\dagger} \sigma_{zz} + \frac{A_{C}^{2} \sigma_{zz}^{2}}{2} (\chi_{T} + \chi_{I})$$
(2)

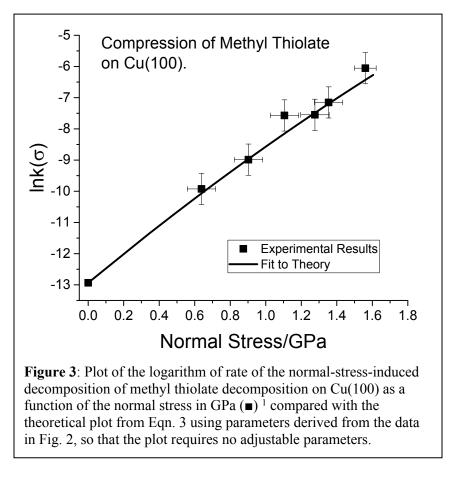
Note that all the parameters required to evaluate Eqn. 2 are available from the plots in Fig. 2.

The results of these calculations are compared with the experimental data and a plot of $lnk(\sigma)$ versus the normal stress (**•**) is shown in Figure 3,¹ which is an almost straight line with just a slight curvature. Note that stresses up to ~ 1.6 GPa can routinely be obtained using the AFM nanoreactor and that this stress reduces the activation energy by ~15 kJ/mol from the intrinsic value.⁴⁶

To compare directly with experiment, Eqn. 2 is written as:

$$\ln(k(\sigma_{zz})) = \ln(k_0) - \frac{\Delta V^{\ddagger}}{k_B T} \sigma_{zz} - \frac{A_r^2 \sigma_{zz}^2}{2k_B T} (\chi_T + \chi_I)$$
(3).

where k_0 is the zero-pressure rate constant, k_B is the Boltzmann constant and T is the absolute temperature. The value of k_0 (the value at zero stress) is obtained from an Arrhenius equation: k_0 $= Aexp(-\frac{E_{act}^0}{k_BT})$, where A is a pre-exponential factor that is set to a value of 1×10^{13} s⁻¹ that was used to calculate the activation energy from the peak temperature in TPD experiments for methyl thiolate decomposition on Cu(100).⁵¹ Note that the lack of data between ~0.6 GPa and this data point is due to the fact that mechanochemical reaction rates are too slow to be measured for lower stresses. The fact that the mechanochemically measured reaction rates extrapolate almost linearly to the rate at zero stress measured using a completely different methods confirms that we are correctly measuring the mechanochemical reaction for the same reaction. The prediction from Eqn. 3 is also plotted in Fig. 3 and the agreement between the experimental data (\blacksquare) and the theory (solid line) is very good.⁵⁶ It should be emphasized that the theory involves no adjustable parameters and uses only values obtained from Fig. 2, a value of the pre-exponential factor (*A*) to relate the activation energy to the rate constant, and the value of A_C used in the Stearn-Eyring postulate. Note that the activation volume here is larger than previously reported ¹ because



of refinements in the calibration of the contact areas and forces as described in the Supporting Information section.

The good agreement between experiment and theory over a relatively wide stress range confirms that (i) the SDP can be used with good accuracy to calculate mechanochemical reaction rates and (ii) the Stearn-

Eyring postulate can be used to calculate an activation volume. The results show that, at least in this case, a linear force dependence reproduces the experimental results quite well and obeys the Bell model, suggesting that the F^2 contributions are small, although a slight curvature is noticeable in the data in Fig. 3.

It also emphasizes that both the value of the activation length and the area over which the stresses act can influence the activation volume. The latter parameter can be controlled by modifying the structure of the mechanophore in a targeted way to modulate the force exerted along the SDP. The results also illustrate how the direction of the stress relative to the structure of the mechanophore influences the mechanochemical reactivity and can induce reactions that are not thermally accessible.⁷⁵

Conclusions

This work tests the postulate that mechanochemical reaction rates can be calculated using the steepest-decent pathway⁴⁶ using the example of the normal-stress-induced decomposition of methyl thiolate on Cu(100). The reaction rate was measured using a nanomechanochemical reactor using an AFM tip that allows the contact stresses and the reaction rates to be measured.

The calculation was carried out by compressing the initial- and transition-state structures by a rigid counterface to provide values of the activation length and the compliances of the initial- and transition-states for use as parameters in an extended-Bell model. The activation volume was calculated from the product of the activation length and the area over which the force acts as first suggested by Stearn and Eyring. The results agreed very well with experiment confirming the postulate that using the SDP as the mechanochemical reaction pathway provides a good approximation that yields stress-dependent energies that are in good agreement with those measured experimentally. The advantage of this approach is that the calculations are easy to perform and produce results that are in good agreement with experiment. This, of course, needs to be tested for other, more complex mechanochemical reactions.

Supporting Information

Calibration of normal stresses and reaction rates; Atomic coordinates for the compression of methyl thiolate species on Cu(100); Background subtraction methods for calculating the energy versus slab separation. The Supporting Information is available free of charge at.....

Author contributions

All authors contributed equally to the paper.

Conflicts of interest

There are no conflicts to declare.

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