

# An Effective Pathway to Design and Synthesize UV Birefringent Crystals via Rational Assemble of π-Conjugated [CO3]2- and [NO3]- Triangles

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# ARTICLE

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# An Effective Pathway to Design and Synthesize UV Birefringent Crystals via Rational Assemble of $\pi$ -Conjugated [CO<sub>3</sub>]<sup>2-</sup> and [NO<sub>3</sub>]<sup>-</sup> Triangles

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Planar MO<sub>3</sub> (M=B, C, N) units have frequently been considered as the important structural components of novel birefringent crystal materials. An efficient approach for creating new functional crystals is to simultaneously assemble multiple structural motifs together. Two compounds, Na<sub>3</sub>Rb<sub>6</sub>(CO<sub>3</sub>)<sub>3</sub>(NO<sub>3</sub>)<sub>2</sub>X·6H<sub>2</sub>O (X = Br and Cl), were synthesized by the integration of three kinds of anionic groups. More interestingly,  $[CO_3]^{2-}$  and  $[NO_3]^-$  groups in Na<sub>3</sub>Rb<sub>6</sub>(CO<sub>3</sub>)<sub>3</sub>(NO<sub>3</sub>)<sub>2</sub>X·6H<sub>2</sub>O are all coplanar under the guidance of  $[NaO_7]^{13-}$  polyhedra, which can enhance the anisotropic polarizability. Na<sub>3</sub>Rb<sub>6</sub>(CO<sub>3</sub>)<sub>3</sub>(NO<sub>3</sub>)<sub>2</sub>X·6H<sub>2</sub>O have large theoretical birefringence of ~0.165 at 1064 nm and possess short UV cut-off edge of ~230 nm. Additionally, the two compounds exhibit good crystal growth habits. These properties illustrate that Na<sub>3</sub>Rb<sub>6</sub>(CO<sub>3</sub>)<sub>3</sub>(NO<sub>3</sub>)<sub>2</sub>X·6H<sub>2</sub>O are promising UV birefringent crystals.

#### Introduction

UV birefringent crystals, as a type of important functional material for optoelectronics, have important applications in optical deflection, optical modulation, and nonlinear optics.<sup>1-5</sup> At present, commercial UV birefringent crystals, such as CaCO<sub>3</sub>,<sup>6</sup>  $\alpha$ -BaB<sub>2</sub>O<sub>4</sub> ( $\alpha$ -BBO)<sup>7</sup> and MgF<sub>2</sub><sup>8</sup>, have been facing kinds of problems, like difficulty in crystal growth, small birefringence, insufficient band gap, which seriously limit their practical applications. Therefore, new high-performance UV birefringent crystals are still needed.

Previous studies mainly focused on borate UV birefringent crystals,<sup>9-12</sup> such as SrAlB<sub>3</sub>O<sub>6</sub>F<sub>2</sub> (0.078 at 1064 nm, 7.65 eV),<sup>13</sup> Li<sub>3</sub>La<sub>2</sub>(BO<sub>3</sub>)<sub>3</sub> (0.078 at 1064 nm, 5.22 eV),<sup>14</sup> Ca(BO<sub>2</sub>)<sub>2</sub> (0.247 at 193 nm, 7.34 eV),<sup>15</sup> Ba<sub>2</sub>Zn<sub>2</sub>B<sub>6</sub>O<sub>13</sub> (0.085 at 532 nm, 6.12 eV),<sup>16</sup> NaMgBe<sub>2</sub>(BO<sub>3</sub>)<sub>2</sub>F (0.078 at 589 nm, 4.19 eV),<sup>17</sup> which is mainly attributed to the excellent properties of the planar triangle [BO<sub>3</sub>]<sup>3-</sup> group, including relatively strong anisotropic polarizability and high UV transmittance.<sup>18-21</sup> It is well known that [CO<sub>3</sub>]<sup>2-</sup> and [NO<sub>3</sub>]<sup>-</sup> groups possess the similar  $\pi$ -conjugated electronic configuration and planar triangle structure as [BO<sub>3</sub>]<sup>3-</sup> group.<sup>22,23</sup> In terms of the anisotropic polarizability of anionic

groups, it generally follows the trend of  $[NO_3]^- > [CO_3]^{2-} > [BO_3]^{3-.24}$  Therefore,  $[CO_3]^{2-}$  and  $[NO_3]^-$  groups are also ideal fundamental building blocks (FBBs) for designing birefringent materials, and they have advantages over  $[BO_3]^{3-}$  group when building birefringent materials.<sup>25</sup>

After reviewing the inorganic crystal structure database (ICSD with the version of 4.9.0. the last release of ICSD-2022.2), we find that there have been relatively more borate carbonates reported, but fewer nitrate borates and nitrate carbonates. Through extensive literature research, it is noticed that the pH value for synthesizing nitrate is often different from that for borate/carbonate when using solution methods. Many of the reported nitrates are synthesized under acidic conditions, for example, La(OH)<sub>2</sub>NO<sub>3</sub> (pH 4-6),<sup>26</sup> [Pb<sub>4</sub>(OH)<sub>4</sub>](NO<sub>3</sub>)<sub>4</sub> (pH 4-5),<sup>27</sup> BiGeO<sub>2</sub>(OH)<sub>2</sub>NO<sub>3</sub> (pH  $\approx$  0.2),<sup>28</sup> etc. In contrast, hexagonal H<sub>3</sub>BO<sub>3</sub> crystals will deposit from borate system and [CO<sub>3</sub>]<sup>2-</sup> will be decomposed into CO<sub>2</sub> molecule from carbonate system under acidic environment. Therefore, the coexistence of these three groups is difficult to reconcile. In recent years, there are several compounds recorded in BO<sub>3</sub>-CO<sub>3</sub> and BO<sub>3</sub>-NO<sub>3</sub> systems, such as Pb<sub>7</sub>O(OH)<sub>3</sub>(CO<sub>3</sub>)<sub>3</sub>(BO<sub>3</sub>),<sup>29</sup> Rb<sub>9</sub>[B<sub>4</sub>O<sub>5</sub>(OH)<sub>4</sub>]<sub>3</sub>(CO<sub>3</sub>)Br·7H<sub>2</sub>O,<sup>30</sup>  $Ba_3(BO_3)(CO_3)F$ ,<sup>31</sup>  $K_3(B_6O_{10})(NO_3)$ ,<sup>32</sup>  $Pb_6O_4(BO_3)(NO_3)^{33}$  and  $Pb_2(BO_3)(NO_3)$ ,<sup>34</sup> while carbonate nitrates are rarely reported. In the ICSD, there are only ten compounds containing both [CO<sub>3</sub>]<sup>2-</sup> and [NO<sub>3</sub>]<sup>-</sup> groups been reported before 2022, and their physical and chemical properties are less studied (shown in Table 1). In particular, their optical properties (like birefringence, SHG effect) have not been systematically studied.

Metal-centered polyhedra also have impacts on the structure and optical properties of crystals. Like stereochemically active lone pair (SCALP) cations or d<sup>0</sup> metal cations, they can positively affect the

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Table 1 Compounds in CO<sub>3</sub>-NO<sub>3</sub> system before 2022.

Compounds	Space group	Year
[Co(NH <sub>3</sub> ) <sub>4</sub> (CO <sub>3</sub> )(NO <sub>3</sub> )] <sub>2</sub> ·H <sub>2</sub> O <sup>35</sup>	P21/n	1990
$[Pt(NH_3)_2]_4(CO_3)_2(NO_3)_4 \cdot 3H_2O^{36}$	рĪ	1993
$(TI_{0.85}Cr_{0.15})Sr_4Cu_2(CO_3)_{0.5}(NO_3)_{0.5}O_7^{37}$	P4/mmm	1997
(NH <sub>3</sub> ) <sub>3</sub> Co(OH) <sub>2</sub> (CO <sub>3</sub> )Co(NH <sub>3</sub> ) <sub>3</sub> (NO <sub>3</sub> ) <sub>2</sub> ·H <sub>2</sub> O <sup>38</sup>	рĪ	1999
$Co(NH_3)_5(CO_3)NO_3 \cdot H_2O^{39}$	P2 <sub>1</sub>	1999
(Pb <sub>6</sub> O <sub>4</sub> )(OH)(NO <sub>3</sub> )(CO <sub>3</sub> ) <sup>40</sup>	Pnma	2000
PbTiO <sub>2</sub> (CO <sub>3</sub> ) <sub>0.3</sub> (NO <sub>3</sub> ) <sub>0.35</sub> (OH) <sup>41</sup>	р <u>3</u> 1т	2001
$Ca_5(UO_2(CO_3)_3)_2(NO_3)_2 \cdot 10H_2O^{42}$	P2 <sub>1</sub> /n	2002
Co(CO <sub>3</sub> )(NH <sub>3</sub> ) <sub>4</sub> NO <sub>3</sub> <sup>43</sup>	P21/c	2013
(NH <sub>3</sub> ) <sub>6</sub> Rh <sub>2</sub> (CO <sub>3</sub> )(OH) <sub>2</sub> (NO <sub>3</sub> ) <sub>2</sub> ·H <sub>2</sub> O <sup>44</sup>	рĪ	2016

birefringence,<sup>45,46</sup> such as SCALP Sn<sup>2+</sup> atom in Sn<sub>2</sub>PO<sub>4</sub>I (0.664 at 532nm),<sup>47</sup> [WO<sub>5</sub>F]<sup>5-</sup> octahedron in KWO<sub>3</sub>F (0.088 at 1064nm),<sup>48</sup> etc. However, these metal cations play a negative role in the transmission of UV light.<sup>49</sup> During the design and syntheses of UV birefringent crystals, alkali or alkaline earth metals are the preferred elements owing to the absence of electron transition of *d-d* and *f-f* orbitals, which can effectively extend the band gap.<sup>50,51</sup> Furthermore, compared with O<sup>2-</sup> ion, halogen anions (F<sup>-</sup>, Cl<sup>-</sup>, Br<sup>-</sup>) possess distinctive electronegativity, coordination ability and polarizability. So halogenated oxypolyhedra (like [BO<sub>3</sub>F]<sup>4-</sup>, [TiO<sub>5</sub>F]<sup>7-</sup>, [WO<sub>5</sub>F]<sup>5-</sup> groups) always exhibit larger polarizability than oxypolyhedra (like [BO<sub>4</sub>]<sup>5-</sup>, [TiO<sub>6</sub>]<sup>8-</sup>, [WO<sub>6</sub>]<sup>6-</sup> groups). Additionally, the addition of halogen anions usually can make the UV cutoff edge blue-shift.<sup>52</sup>

Guided by above ideas, we successfully synthesized two mixed-alkali metal carbonate nitrate halides,  $Na_3Rb_6(CO_3)_3(NO_3)_2X \cdot 6H_2O$  (X = Br and Cl), through hydrothermal method. We noticed during our manuscript preperation that, quite recently. the Na<sub>3</sub>Rb<sub>6</sub>(CO<sub>3</sub>)<sub>3</sub>(NO<sub>3</sub>)<sub>2</sub>Cl·6H<sub>2</sub>O was synthesized via evaperation method by Pan's group.<sup>53</sup> Therefore, in this work, we will mainly focus on the synthesis and properties of Na<sub>3</sub>Rb<sub>6</sub>(CO<sub>3</sub>)<sub>3</sub>(NO<sub>3</sub>)<sub>2</sub>Br·6H<sub>2</sub>O, but Na<sub>3</sub>Rb<sub>6</sub>(CO<sub>3</sub>)<sub>3</sub>(NO<sub>3</sub>)<sub>2</sub>Cl·6H<sub>2</sub>O is also briefly described and characterized for comparison. And Na<sub>3</sub>Rb<sub>6</sub>(CO<sub>3</sub>)<sub>3</sub>(NO<sub>3</sub>)<sub>2</sub>Br·6H<sub>2</sub>O is the first crabonate nitrate bromide. Na<sub>3</sub>Rb<sub>6</sub>(CO<sub>3</sub>)<sub>3</sub>(NO<sub>3</sub>)<sub>2</sub>X·6H<sub>2</sub>O contains four different functional primitives, which includes alkali metals (Na<sup>+</sup> and Rb<sup>+</sup>), [CO<sub>3</sub>]<sup>2-</sup>, [NO<sub>3</sub>]<sup>-</sup> and halogen anions. Their synergistic effect gives  $Na_3Rb_6(CO_3)_3(NO_3)_2X\cdot 6H_2O$  excellent linear optical properties: large birefringences and wide UV optical band gaps. Besides, they have no layered growth habit and it is feasible to obtain millimeter sized crystals, ensuring the availability of large-scale single crystals. Therefore, they have the potential to serve as excellent UV birefringent crystals.

## Experimental

**Reagents.** Ce(NO<sub>3</sub>)<sub>3</sub>·6H<sub>2</sub>O(99%), Rb<sub>2</sub>CO<sub>3</sub>(99%), NaOH(99%), NaBr(99%), NaCl(99%) were purchased from Taitan and used as received.

Synthesis. Considering that the [CO<sub>3</sub>]<sup>2-</sup> group is not stable in acidic solutions, we will try to combine [CO<sub>3</sub>]<sup>2-</sup> and [NO<sub>3</sub>]<sup>-</sup> groups under neutral or alkaline conditions. Although nitrates tend to form under acidic conditions, there are still examples synthesized under alkaline conditions, such as  $Pb_2O(OH)NO_3$  (pH = 7~8) and  $Pb_3O_2(OH)NO_3$  (pH = 11).<sup>27</sup> After constant adjustment of raw materials and ratios,  $Na_3Rb_6(CO_3)_3(NO_3)_2X\cdot 6H_2O$ was finally synthesized using hydrothermal method. In the synthesis of  $Na_3Rb_6(CO_3)_3(NO_3)_2Br\cdot 6H_2O$ , the reaction mixture of  $Ce(NO_3)_3\cdot 6H_2O$ (1.520 g, 3.5 mmol), Rb<sub>2</sub>CO<sub>3</sub> (3.002 g, 13 mmol), NaOH (0.500 g, 12.5 mmol), NaBr(1.029 g, 10 mmol) and 5 mL deionized water were sealed in an autoclave with a 23 mL Teflon linear. After heating at 220 °C for 2 days, the reactant was cooled to room temperature at a rate of 3 °C/h. Similarly, single crystals of Na<sub>3</sub>Rb<sub>6</sub>(CO<sub>3</sub>)<sub>3</sub>(NO<sub>3</sub>)<sub>2</sub>Cl·6H<sub>2</sub>O were synthesized with a reaction mixture of  $Ce(NO_3)_3 \cdot GH_2O$  (1.086 g, 2.5 mmol), Rb<sub>2</sub>CO<sub>3</sub> (3.464 g, 15 mmol), NaOH (0.800 g, 20 mmol), NaCl (0.584 g, 10 mmol) and 5 ml deionized water. Colorless and block crystals (Figure S1) were obtained after washing by deionized water or ethanol. The polycrystalline powder samples of  $Na_3Rb_6(CO_3)_3(NO_3)_2Br\cdot 6H_2O$  and  $Na_3Rb_6(CO_3)_3(NO_3)_2CI\cdot 6H_2O$  were obtained directly from grinding the as-grown crystals, and their purity was checked by powder X-ray diffraction patterns (see Figure S2).

Single crystal X-ray diffraction. Single crystal X-ray diffraction data of Na<sub>3</sub>Rb<sub>6</sub>(CO<sub>3</sub>)<sub>3</sub>(NO<sub>3</sub>)<sub>2</sub>Br·6H<sub>2</sub>O and Na<sub>3</sub>Rb<sub>6</sub>(CO<sub>3</sub>)<sub>3</sub>(NO<sub>3</sub>)<sub>2</sub>Cl·6H<sub>2</sub>O were collected at room temperature by single-crystal XRD on an APEX II CCD diffractometer using monochromatic Mo-Ka radiation ( $\lambda$ =0.71073 Å). A transparent block of crystal was mounted on a glass fiber with epoxy for structure determination. Moreover, absorption corrections were carried out using the SCALE program for area detector and integrated with the SAINT program.<sup>54</sup> Absorption corrections based on the multi-scan technique were also applied. The structure was solved by the direct method and refined by fullmatrix least-squares fitting on F<sup>2</sup> by SHELX-97.55 All H atoms are located at geometrically calculated positions and refined with isotropic thermal parameters. All the other atoms were refined anisotropically. All the structural data were also checked for possible missing symmetry with the program PLATON,56 and no higher symmetries were found. Relevant crystallographic data and details of the experimental conditions for Na<sub>3</sub>Rb<sub>6</sub>(CO<sub>3</sub>)<sub>3</sub>(NO<sub>3</sub>)<sub>2</sub>Br·6H<sub>2</sub>O and Na<sub>3</sub>Rb<sub>6</sub>(CO<sub>3</sub>)<sub>3</sub>(NO<sub>3</sub>)<sub>2</sub>Cl·6H<sub>2</sub>O are summarized in Table S1. Further details about atomic coordinates and isotropic displacement coefficients are supplied in Tables S2 - S3. Selected bond lengths are listed in Tables S4 - S5. Hydrogen coordinates and isotropic displacement parameters are listed in Tables S6 - S7.

**Powder X-ray diffraction (PXRD).** PXRD patterns of polycrystalline materials were obtained on a TD-3500 X-ray diffractometer by using Cu K $\alpha$  radiation ( $\lambda$ =1.540598 Å) at room temperature in the angular range of 2 $\theta$  = 10 - 70° with a scan step width of 0.05° and a fixed time

of 0.2 s. The PXRD patterns for the pure samples of Na<sub>3</sub>Rb<sub>6</sub>(CO<sub>3</sub>)<sub>3</sub>(NO<sub>3</sub>)<sub>2</sub>Br·6H<sub>2</sub>O and Na<sub>3</sub>Rb<sub>6</sub>(CO<sub>3</sub>)<sub>3</sub>(NO<sub>3</sub>)<sub>2</sub>Cl·6H<sub>2</sub>O show good agreement with the calculated PXRD patterns from the single-crystal models (see Figure S2).

**Thermal analysis.** Thermogravimetric analysis (TG) and differential scanning calorimetry (DSC) were conducted on Mettler Toledo unit. The crystal samples (5-12 mg) were enclosed in  $Al_2O_3$  crucibles and heated from room temperature to 800 °C at a rate of 10 °C/min under a constant flow of nitrogen gas.

**Infrared spectroscopy.** IR spectra were recorded on a Thermo Fisher Scientific Fourier transform infrared (FT-IR) spectrometer as KBr pellets in the range of 4000-400 cm<sup>-1</sup>.

**UV-vis diffuse reflectance spectroscopy.** The UV-vis diffuse reflection data were recorded at room temperature using powder samples on a Varian Cary 5000 UV/vis/NIR spectrophotometer and scanned at 200-800 nm. BaSO<sub>4</sub> acts as the standard.

**Birefringence.** The birefringence ( $\Delta n$ ) of Na<sub>3</sub>Rb<sub>6</sub>(CO<sub>3</sub>)<sub>3</sub>(NO<sub>3</sub>)<sub>2</sub>Br·6H<sub>2</sub>O was characterized by using a polarization microscope equipped (ZEISS Axio Scope. 5 pol) with a Berek compensator. The average wavelength of the light source was 546 nm. The equation for calculating the birefringence is listed below,

$$R = |N_g - N_p| \times d = \Delta n \times d$$

Here, *R* represents the optical path difference;  $N_{g}$ ,  $N_{p}$  and  $\Delta n$  mean the refractive index of fast light, slow light, and the difference value of the refractive index, respectively; *d* denotes the thickness of the crystal.

Computational descriptions. The first-principles calculations for  $Na_3Rb_6(CO_3)_3(NO_3)_2X \cdot 6H_2O$  (X = Br and Cl) crystal were performed by the plane-wave pseudopotential method implemented in CASTEP57 program based on density functional theory (DFT)<sup>58</sup> with the normconserving pseudopotentials (NCP).<sup>59</sup> The exchange-correlation (XC) functional were described by the Perdew-Burke-Ernzerhof(PBE) functional within the generalized gradient approximation (GGA).<sup>60</sup> The crystal structure of Na<sub>3</sub>Rb<sub>6</sub>(CO<sub>3</sub>)<sub>3</sub>(NO<sub>3</sub>)<sub>2</sub>X·6H<sub>2</sub>O was optimized until the total energy, the maximum stress, the maximum atomic displacements and the maximum force were smaller than 5×10<sup>-6</sup> eV/atom,  $2 \times 10^{-2}$  GPa,  $5 \times 10^{-4}$  Å, less than  $1 \times 10^{-2}$  eV/Å, respectively. The following valence electron configurations were used: H(1s<sup>1</sup>), C(2s<sup>2</sup>2p<sup>2</sup>), N(2s<sup>2</sup>2p<sup>3</sup>), O(2s<sup>2</sup>2p<sup>4</sup>), Na(2s<sup>2</sup>2p<sup>6</sup>3s<sup>1</sup>), Cl(3s<sup>2</sup>3p<sup>5</sup>), Rb(4s<sup>2</sup>4p<sup>6</sup>5s<sup>1</sup>) for Na<sub>3</sub>Rb<sub>6</sub>(CO<sub>3</sub>)<sub>3</sub>(NO<sub>3</sub>)<sub>2</sub>Cl·6H<sub>2</sub>O and H(1s<sup>1</sup>), C(2s<sup>2</sup>2p<sup>2</sup>),  $N(2s^22p^3)$ ,  $O(2s^22p^4)$ ,  $Na(2s^22p^63s^1)$ ,  $Br(4s^24p^5)$ ,  $Rb(4s^24p^65s^1)$  for  $Na_3Rb_6(CO_3)_3(NO_3)_2Br\cdot 6H_2O$ . And the plane-wave energy cut-off was both set at 830.0 eV. The k-point of Monkhorst-Pack grid in the Brillouin zone for Na<sub>3</sub>Rb<sub>6</sub>(CO<sub>3</sub>)<sub>3</sub>(NO<sub>3</sub>)<sub>2</sub>X·6H<sub>2</sub>O were set as 3×3×2 and the span of the Brillouin zone were selected as 0.04 Å<sup>-1</sup> which ensure the sufficient accuracy of the calculated results. It is well-known that GGA method always underestimates the bandgap owing to the discontinuity of exchange-correlation energy functional, hence a socalled scissors operator was used to calculate the optical properties.

#### Results and discussion

Structure description. Na<sub>3</sub>Rb<sub>6</sub>(CO<sub>3</sub>)<sub>3</sub>(NO<sub>3</sub>)<sub>2</sub>Br·6H<sub>2</sub>O crystallizes in the centrosymmetric space group P6<sub>3</sub>/mcm (No. 194, Table S1). In its asymmetric unit, there is one Na atom, one Rb atom, one N atom, two C atoms, four O atoms, one Br atom and one H atom (Figures S3 and S4). Na<sub>3</sub>Rb<sub>6</sub>(CO<sub>3</sub>)<sub>3</sub>(NO<sub>3</sub>)<sub>2</sub>Br·6H<sub>2</sub>O exhibits a three-dimensional (3D) crystal structure consisting of [NaO<sub>7</sub>]<sup>13-</sup>, [BrRb<sub>6</sub>]<sup>5+</sup>, [CO<sub>3</sub>]<sup>2-</sup>, [NO<sub>3</sub>]<sup>-</sup> and H<sub>2</sub>O units (Figure 1a). Each Na atom is coordinated to seven O atoms as [NaO<sub>7</sub>]<sup>13-</sup> pentagon double cone, five of which are equatorial O atoms and the rest two are apical O atoms (Figure S5). In [NaO<sub>7</sub>]<sup>13-</sup>, the five equatorial O atoms are connected to two edgeshared  $[NO_3]^-$  and one corner-shared  $[C(1)O_3]^{2-}$ , and the two apical O atoms are linked to H atoms to form two H<sub>2</sub>O groups. And then the  $[Na(CO_3)(NO_3)_2 \cdot 2H_2O]^{4-}$  polyhedron are formed, which extends infinitely in ab plane to create the 2D  $[Na_3(CO_3)(NO_3)_2 \cdot 2H_2O] - \infty$ layer (Figure 1b). The distance between adjacent  $[Na_3(CO_3)(NO_3)_2 \cdot 2H_2O] - \infty$  layers  $(d_{Br})$  is 7.93 Å. Each Br atom is surrounded by six Rb atoms to form [BrRb<sub>6</sub>]<sup>5+</sup> cationic octahedron (Figure S6). [BrRb<sub>6</sub>]<sup>5+</sup> octahedra and [C(2)O<sub>3</sub>]<sup>2-</sup> units are further connected with each other to compose the  $[(Rb_6Br)(CO_3)_2] + \infty$  layers via Rb-O bonds (Figure 1c). Moreover,  $[C(2)O_3]^{2-}$  groups are also parallel to the ab plane under the guidance of [BrRb<sub>6</sub>]<sup>5+</sup>. Eventually,  $[Na_3(CO_3)(NO_3)_2 \cdot 2H_2O] - \infty$  layers and  $[(Rb_6Br)(CO_3)_2] + \infty$  layers are stacked alternately along the c-axis to construct the whole structure of  $Na_3Rb_6(CO_3)_3(NO_3)_2Br\cdot 6H_2O$ . In connectivity terms, the material may be described as  $3[\text{Na}(\text{CO}_3)_{1/3}2(\text{NO}_3)_{1/3}\cdot2\text{H}_2\text{O}]^{1/3-}$  anion, with charge balance maintained by  $[(Rb_6Br)6(CO_3)_{1/3}]^+$  cation. Compared with  $Na_3Rb_6(CO_3)_3(NO_3)_2Br\cdot 6H_2O$ , the layer spacing of isostructural  $Na_3Rb_6(CO_3)_3(NO_3)_2Cl \cdot 6H_2O$  (d<sub>cl</sub>, Figure 1d) decreases from 7.93 Å to 7.91 Å due to the replacement of the Br<sup>-</sup> ion by smaller Cl<sup>-</sup> ion.

In Na<sub>3</sub>Rb<sub>6</sub>(CO<sub>3</sub>)<sub>3</sub>(NO<sub>3</sub>)<sub>2</sub>Br·6H<sub>2</sub>O, Na-O distances are in the range of 2.303(4) - 2.683(5) Å. The C-O bond distances range from 1.281(2) to 1.285(4) Å. The N-O bond distance is 1.254(4) Å. The Rb<sup>+</sup> cation is surrounded by eight O atoms and one Br atom, with Rb-O distances of 3.012(2) - 3.413(3) Å and Rb-Br distance of 3.344(1) Å. All bond lengths are within reasonable limits. The bond valence of the crystalline elements is determined by the following equation,

$$V_i = \sum_j s_{ij} = \sum_j exp\{(r_0 - r_{ij})/B\}$$

where  $S_{ij}$  is an empirical determined parameter related to the bond lengths  $r_0$ ,  $r_{ij}$  and B (usually 0.37).<sup>61,62</sup> Bond valence sums (BVSs) calculations indicate that BVSs for Na, Rb, C(1) - C(2), N, O(1) - O(4) and Br are +1.178, +1.130, +3.992 - +4.282, +4.817, -0.463 - -2.105, -1.307, respectively (Table S2). Wherein, O(1) has smaller BVS value (-0.463), so it is determined to be connected with H atoms to form H<sub>2</sub>O molecule. The elemental bond valence of Na<sub>3</sub>Rb<sub>6</sub>(CO<sub>3</sub>)<sub>3</sub>(NO<sub>3</sub>)<sub>2</sub>Cl·6H<sub>2</sub>O is listed in Table S3.

Structure-property relationship.  $\pi$ -conjugated groups, MO<sub>3</sub> (M=B, N and C), are ideal functional motifs for birefringent materials because of their large anisotropic polarizability. The appropriate arrangement of MO<sub>3</sub> groups has a strong influence on the macroscopic optical properties of crystals. For example, compounds NaZnCO<sub>3</sub>F and Na<sub>4</sub>Zn(CO<sub>3</sub>)<sub>3</sub>, reported by Ye's group, have the same functional unit of  $\pi$ -conjugated [CO<sub>3</sub>]<sup>2–</sup> group. In terms of their crystal structures, the

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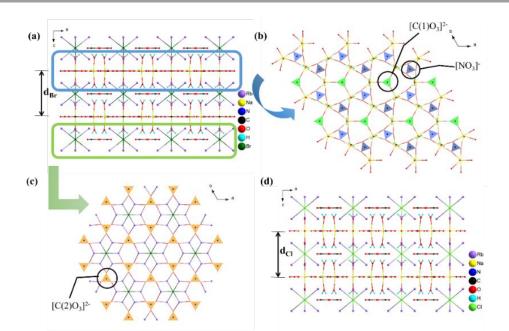
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 $[CO_3]^{2-}$  in NaZnCO<sub>3</sub>F are arranged in coplanar manner, but  $[CO_3]^{2-}$  groups in Na<sub>4</sub>Zn(CO<sub>3</sub>)<sub>3</sub> point in different directions.<sup>63</sup> The different arrangements of  $[CO_3]^{2-}$  groups make NaZnCO<sub>3</sub>F have a larger birefringence of 0.171 at 1064 nm, and Na<sub>4</sub>Zn(CO<sub>3</sub>)<sub>3</sub> have a relatively smaller birefringence of 0.138 at 1064 nm. Therefore, coplanar arrangement is the optimum choice, which can promote effective superposition of microcosmic anisotropic polarizability.

Then, how to achieve the coplanar arrangement of MO<sub>3</sub> group? From the well-known ABCO<sub>3</sub>F system, it can be realized that the equatorial O atoms can efficiently direct  $MO_3$  for coplanar arrangement. Moreover, we found that the oxypolyhedra formed by divalent metal cations can provide equatorial O atoms, such as Sr<sup>2+</sup> in RbSrCO<sub>3</sub>F<sup>64</sup> (Figure 2a) and Sr<sub>2</sub>(OH)<sub>3</sub>NO<sub>3</sub> (Figure 2b),<sup>65</sup> Mg<sup>2+</sup> in RbMgCO<sub>3</sub>F (Figure 2c),<sup>66</sup> Cd<sup>2+</sup> in and KCdCO<sub>3</sub>F (Figure 2d),<sup>67</sup> Pb<sup>2+</sup> in CsPbCO<sub>3</sub>F (Figure 2e).68 But it is rarely reported that alkali metal-centered oxypolyhedra offer equatorial O atoms. Interestingly, [NaO7]13polyhedron in  $Na_3Rb_6(CO_3)_3(NO_3)_2Br\cdot 6H_2O$  (this work) has five equatorial O atoms, and same situation happened in  $Na_3Rb_6(CO_3)_3(NO_3)_2Cl \cdot 6H_2O$ , which prompts us to study related systems (Figure 1a). We investigated two systems including mixedalkali metal carbonates and mixed-alkali metal nitrates to study the phenomenon of alkali metal cations supplying equatorial O atoms. There are eleven compounds in the two systems according to ICSD database (listed in Table 2), and only five of them possess equatorial O atoms:  $NaRb_2(NO_3)_3$ ,<sup>69</sup>  $NaA_2(HCO_3)(CO_3) \cdot 2H_2O$  (A = K and Rb)<sup>70</sup> and Na<sub>3</sub>Rb<sub>6</sub>(CO<sub>3</sub>)<sub>3</sub>(NO<sub>3</sub>)<sub>2</sub>Cl·6H<sub>2</sub>O,<sup>53</sup> Na<sub>3</sub>Rb<sub>6</sub>(CO<sub>3</sub>)<sub>3</sub>(NO<sub>3</sub>)<sub>2</sub>Br·6H<sub>2</sub>O (this work). In NaRb<sub>2</sub>(NO<sub>3</sub>)<sub>3</sub>, Na<sup>+</sup> is connected with eight O atoms, six of which are equatorial O atoms from three bidentate [NO<sub>3</sub>]<sup>-</sup>. While in NaA<sub>2</sub>(HCO<sub>3</sub>)(CO<sub>3</sub>)·2H<sub>2</sub>O, Na<sup>+</sup> is bonded to six O atoms, and only four of them are equatorial O atoms linking with two bidentate  $[CO_3]^{2-}$ . However, Na<sup>+</sup> in Na<sub>3</sub>Rb<sub>6</sub>(CO<sub>3</sub>)<sub>3</sub>(NO<sub>3</sub>)<sub>2</sub>X·6H<sub>2</sub>O has five equatorial O

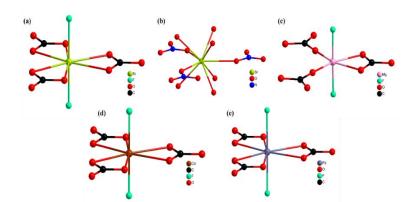
atoms, four of which are from two bidentate  $[NO_3]^-$  groups and the rest one is from a monodentate  $[CO_3]^{2-}$  group. Through literature research and structural comparison, it was found that Na<sup>+</sup> in Na<sub>3</sub>Rb<sub>6</sub>(CO<sub>3</sub>)<sub>3</sub>(NO<sub>3</sub>)<sub>2</sub>X·6H<sub>2</sub>O (this work) is similar to Ca<sup>2+</sup> in RbCaCO<sub>3</sub>F<sup>64</sup> from ABCO<sub>3</sub>F family, and both of them are seven coordinated and have five equatorial O atoms to direct MO<sub>3</sub> groups to be coplanar (Figure 3).

A proper ionic radius has an essential effect on the formation of equatorial O atoms. In Table 2, it is noticed that the cation with too large or small radius does not have equatorial atoms. For example, Li<sup>+</sup> ions in LiACO<sub>3</sub> (A=Na, K, Rb, Cs) are four or five-coordinated, and these [LiO<sub>4</sub>]<sup>7-</sup> or [LiO<sub>5</sub>]<sup>9-</sup> groups do not have equatorial atoms, which is probably because Li atom is too small to bond with enough O atoms to form the equatorial plane. If the cation radius is too large, it will impede the formation of equatorial plane, because large cations usually may attract too many O atoms as ligands and make the O atoms deviate from the equatorial plane, like A" atoms in Table 2. Therefore, the cations with moderate radius can promote the construction of equatorial O plane and then guide the arrangement of planar MO<sub>3</sub> groups. It is found that all the equatorial O atoms appear in the Na-centered polyhedra in Table 2, which demonstrates that Na<sup>+</sup> possesses a suitable ionic radius. Inspired by Zou's work,<sup>64</sup> we speculated that the number of equatorial O atoms is related to the radius ratio of different cations. So, we calculated the cation radius ratio (A"/A' ratio) of the compounds listed in Table 2. It illustrates that when A"/A' ratio is between 1.386 - 1.509, equatorial O atoms will appear, otherwise, they will not exist. In addition, the number of equatorial O decreases with the increase of the A"/A' ratio. Therefore, the choice of cation is crucial in design and synthesis of planar group optical materials, which has a significant effect on the structure and properties of materials.



**Figure 1** (a) The crystal structure of  $Na_3Rb_6(CO_3)_3(NO_3)_2Br\cdot 6H_2O$ ; (b)  $[Na_3(CO_3)(NO_3)_2 \cdot 2H_2O] - \infty$  layer; (c)  $[(Rb_6Br)(CO_3)_2] + \infty$  layer; (d) The crystal structure of  $Na_3Rb_6(CO_3)_3(NO_3)_2CI\cdot 6H_2O$ . (Rb-O bonds are omitted for clarity)

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**Figure 2** The coordination environments of  $Sr^{2+}$  in  $RbSrCO_3F$  (a),  $Sr^{2+}$  in  $Sr_2(OH)_3NO_3$  (b),  $Mg^{2+}$  in  $RbMgCO_3F$  (c),  $Cd^{2+}$  in and  $KCdCO_3F$  (d) and  $Pb^{2+}$  in  $CsPbCO_3F$  (e)

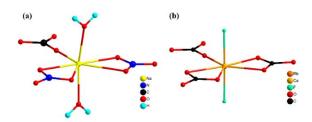
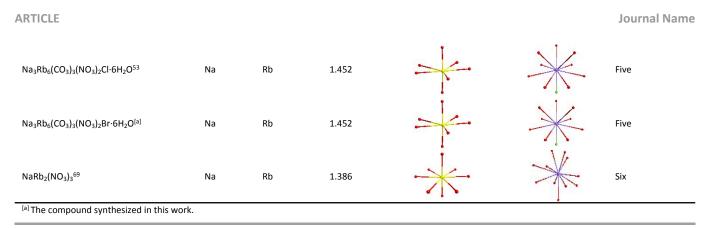


Figure 3 The coordination environments of (a)  $Na^+$  in  $Na_3Rb_6(CO_3)_3(NO_3)_2Br\cdot 6H_2O$  and (b)  $Ca^{2+}$  in  $RbCaCO_3F$ .

Table 2 Mixed-alkali metal carbonates and mixed-alkali metal nitrates. (A'= alkali metals with smaller atomic numbers; A''= alkali
metals with bigger atomic numbers)

Molecular formula	Α'	Α''	A''/A'	A'O <sub>n</sub> unit	A''O <sub>n</sub> unit	Equatorial O atoms
KRb <sub>2</sub> CO <sub>3</sub> F <sup>71</sup>	К	Rb		Disor	dered structure	
LiNaCO <sub>3</sub> <sup>72</sup>	Li	Na	1.562	$\succ$	X	None
LiKCO3 <sup>73</sup>	Li	К	2.192	Y	×	None
LiRbCO <sub>3</sub> <sup>73</sup>	Li	Rb	2.329	Y	×	None
LiCsCO <sub>3</sub> <sup>73</sup>	Li	Cs	2.671	Y	×	None
NaKCO <sub>3</sub> ·6H <sub>2</sub> O <sup>74</sup>	Na	К	1.310	4	$\neq$	None
NaK <sub>2</sub> (HCO <sub>3</sub> )(CO <sub>3</sub> )·2H <sub>2</sub> O <sup>70</sup>	Na	К	1.422		×	Four
NaRb <sub>2</sub> (HCO <sub>3</sub> )(CO <sub>3</sub> )·2H <sub>2</sub> O <sup>70</sup>	Na	Rb	1.509		×	Four



Thermal Behaviours. The TG-DSC curves of Na<sub>3</sub>Rb<sub>6</sub>(CO<sub>3</sub>)<sub>3</sub>(NO<sub>3</sub>)<sub>2</sub>Br·6H<sub>2</sub>O are shown in Figure S7a. It is stable until ~100 °C, and the significant weight loss is presumably attributed to the absence of crystal water. The graph shows a weight loss ratio of 9.76%, which is basically consistent with the theoretical calculation 10.06%. The decomposition of Na<sub>3</sub>Rb<sub>6</sub>(CO<sub>3</sub>)<sub>3</sub>(NO<sub>3</sub>)<sub>2</sub>Br·6H<sub>2</sub>O has also occurred at the same time. The remaining chemicals are mainly RbBr,  $RbNO_3$  and  $Na_2CO_3$  (Figure S8a). And the peak around 350  $^\circ C$ represents the melting point. The weightlessness after 600 °C is due to volatilization of residues. Na<sub>3</sub>Rb<sub>6</sub>(CO<sub>3</sub>)<sub>3</sub>(NO<sub>3</sub>)<sub>2</sub>Cl·6H<sub>2</sub>O exhibits similar trends (Figure S7b).

**IR and UV-vis diffuse reflectance spectra.** The IR spectrum of Na<sub>3</sub>Rb<sub>6</sub>(CO<sub>3</sub>)<sub>3</sub>(NO<sub>3</sub>)<sub>2</sub>Br·6H<sub>2</sub>O is shown in Figure S9a. The stretching of O-H leads to large broad bands at 3104 cm<sup>-1</sup> and around 2423 cm<sup>-1</sup>. The stretching vibrations of C-O and N-O can be reflected by the peaks around 1400 cm<sup>-1</sup>. The out-of-plane and bending vibrations of C-O-C can be observed at 864 cm<sup>-1</sup> and 702 cm<sup>-1</sup>, and the peaks at 825 cm<sup>-1</sup> and 794 cm<sup>-1</sup> reflect the nonplanar bending vibration of [NO<sub>3</sub>]<sup>-</sup> triangular group. Na<sub>3</sub>Rb<sub>6</sub>(CO<sub>3</sub>)<sub>3</sub>(NO<sub>3</sub>)<sub>2</sub>Cl·6H<sub>2</sub>O also possesses similar peaks (Figure S9b).

The UV-vis diffuse reflectance spectra of Na<sub>3</sub>Rb<sub>6</sub>(CO<sub>3</sub>)<sub>3</sub>(NO<sub>3</sub>)<sub>2</sub>X·6H<sub>2</sub>O were measured by using the powder samples. As shown in Figure 4, the reflectance for Na<sub>3</sub>Rb<sub>6</sub>(CO<sub>3</sub>)<sub>3</sub>(NO<sub>3</sub>)<sub>2</sub>Br·6H<sub>2</sub>O at 230 nm is about 13%, and the value for Na<sub>3</sub>Rb<sub>6</sub>(CO<sub>3</sub>)<sub>3</sub>(NO<sub>3</sub>)<sub>2</sub>Cl·6H<sub>2</sub>O at 228 nm is about 5%, which illustrates that the UV cut-off edges of Na<sub>3</sub>Rb<sub>6</sub>(CO<sub>3</sub>)<sub>3</sub>(NO<sub>3</sub>)<sub>2</sub>Br·6H<sub>2</sub>O and Na<sub>3</sub>Rb<sub>6</sub>(CO<sub>3</sub>)<sub>3</sub>(NO<sub>3</sub>)<sub>2</sub>Cl·6H<sub>2</sub>O are shorter than 230 nm and 228 nm, corresponding energy band gaps of 4.80 eV and 4.90 eV respectively. The above data indicate that Na<sub>3</sub>Rb<sub>6</sub>(CO<sub>3</sub>)<sub>3</sub>(NO<sub>3</sub>)<sub>2</sub>X·6H<sub>2</sub>O are potentially capable of UV region applications, and may have relatively large laser damage threshold.

**Theoretical Calculations.** To further understand the relationship between the electronic structure and optical properties of Na<sub>3</sub>Rb<sub>6</sub>(CO<sub>3</sub>)<sub>3</sub>(NO<sub>3</sub>)<sub>2</sub>X·6H<sub>2</sub>O, theoretical calculation was performed based on DFT. The calculated band structure (Figure S10) shows that Na<sub>3</sub>Rb<sub>6</sub>(CO<sub>3</sub>)<sub>3</sub>(NO<sub>3</sub>)<sub>2</sub>Br·6H<sub>2</sub>O and Na<sub>3</sub>Rb<sub>6</sub>(CO<sub>3</sub>)<sub>3</sub>(NO<sub>3</sub>)<sub>2</sub>Cl·6H<sub>2</sub>O are direct band-gap compounds with the valence band maximum (VBM) and the conduction band minimum (CBM) at G point. The GGA-PBE band gaps of Na<sub>3</sub>Rb<sub>6</sub>(CO<sub>3</sub>)<sub>3</sub>(NO<sub>3</sub>)<sub>2</sub>Br·6H<sub>2</sub>O and Na<sub>3</sub>Rb<sub>6</sub>(CO<sub>3</sub>)<sub>3</sub>(NO<sub>3</sub>)<sub>2</sub>Cl·6H<sub>2</sub>O are calculated to be 1.77 eV and 1.85 eV, respectively, which is smaller than that of the measured value because of the discontinuity of the exchange-correlation functional.

To understand the composition and origination of the electronic structures, the total density of states (TDOS) and projected density of states (PDOS) are obtained (Figure S10). As shown in Figure S10, for Na<sub>3</sub>Rb<sub>6</sub>(CO<sub>3</sub>)<sub>3</sub>(NO<sub>3</sub>)<sub>2</sub>Br·6H<sub>2</sub>O, O-2p, C-2p, Rb-4p, Rb-4s and Br-3p states mainly occupy the gap from -6 to -1 eV. The VBM of Na<sub>3</sub>Rb<sub>6</sub>(CO<sub>3</sub>)<sub>3</sub>(NO<sub>3</sub>)<sub>2</sub>Br·6H<sub>2</sub>O is mainly composed of O-2p, N-2p, Rb-4s, Rb-4p, Na-2p, and a small amount of Br-s orbitals; the CBM of Na<sub>3</sub>Rb<sub>6</sub>(CO<sub>3</sub>)<sub>3</sub>(NO<sub>3</sub>)<sub>2</sub>Br·6H<sub>2</sub>O is mainly composed of Rb-s as well as Brs orbitals. For Na<sub>3</sub>Rb<sub>6</sub>(CO<sub>3</sub>)<sub>3</sub>(NO<sub>3</sub>)<sub>2</sub>Cl·6H<sub>2</sub>O (Figure S10), it is mainly occupied by the O-2p, C-2p, Rb-4p, Rb-4s and Cl-3p states from -6 to -1 eV. The VBM of Na<sub>3</sub>Rb<sub>6</sub>(CO<sub>3</sub>)<sub>3</sub>(NO<sub>3</sub>)<sub>2</sub>Cl·6H<sub>2</sub>O is dominated by O-2p and H-1s states. The CBM of Na<sub>3</sub>Rb<sub>6</sub>(CO<sub>3</sub>)<sub>3</sub>(NO<sub>3</sub>)<sub>2</sub>Cl·6H<sub>2</sub>O is mainly dominated by N-2p, O-2p and a little part from Na-2p, Rb-4p, and Rb-4s orbitals. It is well known that, the electronic transition from the VBM to the CBM near the Fermi level determines the optical performance of optical materials. Thus, the linear optical properties of  $Na_3Rb_6(CO_3)_3(NO_3)_2X\cdot 6H_2O$  are mainly determined by the  $\pi$ conjugated [CO<sub>3</sub>]<sup>2-</sup> and [NO<sub>3</sub>]<sup>-</sup> groups.

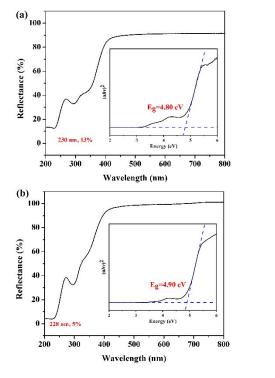
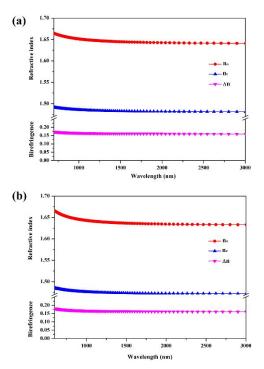
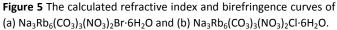
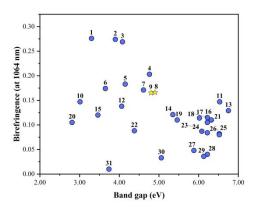


Figure 4UV-visdiffusereflectancespectraof(a) $Na_3Rb_6(CO_3)_3(NO_3)_2Br\cdot 6H_2O$ and(b) $Na_3Rb_6(CO_3)_3(NO_3)_2Cl\cdot 6H_2O.$ 







No.	Molecular formula	No.	Molecular formula
1	$Pb_6O_4(BO_3)(NO_3)$	17	NaZnCO <sub>3</sub> (OH)
2	KPb <sub>2</sub> (CO <sub>3</sub> ) <sub>2</sub> F	18	LiKCO <sub>3</sub>
3	$In(IO_3)_2(NO_3)$	19	KCdCO <sub>3</sub> F
4	La(OH) <sub>2</sub> NO <sub>3</sub>	20	$Cs_3VO(O_2)_2CO_3$
5	CsPbCO <sub>3</sub> F	21	KY(CO <sub>3</sub> ) <sub>2</sub>
6	Pb <sub>2</sub> (BO <sub>3</sub> )(NO <sub>3</sub> )	22	Lu <sub>8</sub> O(CO <sub>3</sub> ) <sub>3</sub> (OH) <sub>15</sub> Br
7	NaZnCO <sub>3</sub> F	23	Zn(NH <sub>3</sub> )CO <sub>3</sub>
8	$Na_3Rb_6(CO_3)_3(NO_3)_2CI \cdot 6H_2O$	24	$Na_2Gd(CO_3)F_3$
9	$Na_3Rb_6(CO_3)_3(NO_3)_2Br\cdot 6H_2O$	25	Ca <sub>2</sub> Na <sub>3</sub> (CO <sub>3</sub> ) <sub>3</sub> F
10	Cs <sub>2</sub> Pb(NO <sub>3</sub> ) <sub>2</sub> Br <sub>2</sub>	26	Ba <sub>2</sub> NO <sub>3</sub> (OH) <sub>3</sub>
11	LiZn(OH)CO <sub>3</sub>	27	Ba <sub>3</sub> (BO <sub>3</sub> )(CO <sub>3</sub> )F
12	Na₄Zn(CO₃)₃	28	$(NH_4)_2Ca_2Y_4(CO_3)_9 \cdot H_2O$
13	RbMgCO <sub>3</sub> F	29	Cs <sub>3</sub> B <sub>8</sub> O <sub>13</sub> (NO <sub>3</sub> )
14	RbCdCO₃F	30	$Ba_2B_5O_8(OH)_2(NO_3)\cdot 3H_2O$
15	Sr <sub>3</sub> [SnOSe <sub>3</sub> ][CO <sub>3</sub> ]	31	$Na_{10}Cd(NO_3)_4(SO_3S)_4$
16	RbCaCO <sub>3</sub> F		

**Figure 6** Birefringences and band gaps statistics of selected carbonates and nitrates.

**Linear optical properties.** The linear optical properties can be obtained by using the dielectric function,  $\varepsilon(\omega)=\varepsilon_1(\omega)+i\varepsilon_2(\omega)$ . The obtained curve of birefringence versus wavelength is shown in Figure 5. The calculated refractive indices and birefringence curves reveal that Na<sub>3</sub>Rb<sub>6</sub>(CO<sub>3</sub>)<sub>3</sub>(NO<sub>3</sub>)<sub>2</sub>X·6H<sub>2</sub>O are negative uniaxial crystals. Their birefrigence value are 0.165 and 0.166 at 1064 nm for Na<sub>3</sub>Rb<sub>6</sub>(CO<sub>3</sub>)<sub>3</sub>(NO<sub>3</sub>)<sub>2</sub>Br·6H<sub>2</sub>O and Na<sub>3</sub>Rb<sub>6</sub>(CO<sub>3</sub>)<sub>3</sub>(NO<sub>3</sub>)<sub>2</sub>Cl·6H<sub>2</sub>O.

A real-space atom-cutting (RSAC) method has been used to intuitively exhibit contribution of  $[CO_3]^{2-}$  and  $[NO_3]^-$  groups. Following the rule of keeping the cutting spheres in contact and avoiding overlap, the cutting radii are set to be 1.48 Å (Rb), 1.25 Å (Na), 1.60 Å(C), 1.23 Å (N), 0.35 Å (H), 1.10 Å (O), 1.55 Å (Br) and 1.50 Å (Cl). Table S8 shows that the main contribution to the birefringence is derived from  $[CO_3]^{2-}$  and  $[NO_3]^-$  groups. From the birefringence of the decomposed groups,  $[NO_3]^-$  exhibits a birefringence (0.121 at 1064 nm) larger than  $[CO_3]^{2-}$  (~0.090 at 1064 nm).

The birefringence of Na<sub>3</sub>Rb<sub>6</sub>(CO<sub>3</sub>)<sub>3</sub>(NO<sub>3</sub>)<sub>2</sub>Br·6H<sub>2</sub>O was preliminarily evaluated using the polarized light microscope (Figure S11). The measured direction was chosen randomly and the average thickness of selected crystal is ~7 µm with an optical path difference of 855 nm at 546 nm. So, the corresponding birefringence of Na<sub>3</sub>Rb<sub>6</sub>(CO<sub>3</sub>)<sub>3</sub>(NO<sub>3</sub>)<sub>2</sub>Br·6H<sub>2</sub>O along the measured direction is ~0.12, close to the value of Na<sub>3</sub>Rb<sub>6</sub>(CO<sub>3</sub>)<sub>3</sub>(NO<sub>3</sub>)<sub>2</sub>Cl·6H<sub>2</sub>O reported in reference 53.

Birefringence and band gap are important criteria for UV birefringent crystals, however, they are essentially opponent. Figure 6 shows the birefringences and band gaps of some carbonates and nitrates (see Table S9 for detailed values), and it can be seen that both properties of  $Na_3Rb_6(CO_3)_3(NO_3)_2X\cdot 6H_2O$  have relatively well balanced linear optical properties.

## Conclusions

In summary, we succeeded in enabling [CO<sub>3</sub>]<sup>2-</sup> and [NO<sub>3</sub>]<sup>-</sup> groups to co-exist in one structure, and obtained two isostructural compounds Na<sub>3</sub>Rb<sub>6</sub>(CO<sub>3</sub>)<sub>3</sub>(NO<sub>3</sub>)<sub>2</sub>Br·6H<sub>2</sub>O and Na<sub>3</sub>Rb<sub>6</sub>(CO<sub>3</sub>)<sub>3</sub>(NO<sub>3</sub>)<sub>2</sub>Cl·6H<sub>2</sub>O. Although both compounds were characterized in this work, the synthesis and properties of  $Na_3Rb_6(CO_3)_3(NO_3)_2Br\cdot 6H_2O$  were paid more attention. In the structure of Na<sub>3</sub>Rb<sub>6</sub>(CO<sub>3</sub>)<sub>3</sub>(NO<sub>3</sub>)<sub>2</sub>Br·6H<sub>2</sub>O, [NaO<sub>7</sub>]<sup>13-</sup> provides five equatorial O atoms, which help [CO<sub>3</sub>]<sup>2-</sup> and [NO<sub>3</sub>]<sup>-</sup> groups arrange in the coplanar manner, resulting in the most favorable superposition of their anisotropic polarization. Benefiting from the optimized crystal structure, Na<sub>3</sub>Rb<sub>6</sub>(CO<sub>3</sub>)<sub>3</sub>(NO<sub>3</sub>)<sub>2</sub>Br·6H<sub>2</sub>O owns a large birefringence of 0.165 at 1064 nm based on theoretical calculations. And the experimental birefringence is about 0.12 at 546 nm. In addition, its UV cut-off edges can reach 230 nm. The desired balance between the birefringence and band gap proves that Na<sub>3</sub>Rb<sub>6</sub>(CO<sub>3</sub>)<sub>3</sub>(NO<sub>3</sub>)<sub>2</sub>Br·6H<sub>2</sub>O could be promising UV birefringent crystal. Furthermore, it is noticed that a moderate ionic radius and a suitable radius ratio of cations are essential for the formation of equatorial O atoms, so the selection of an appropriate cation plays a pivotal role in design and synthesis of optical materials based on planar

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groups. The successful synthesis of the carbonate nitrate gives more possibilities for exploring new compounds, and we expect that our study on the structure-properties relationship will contribute to the discovery of more novel functional crystal materials.

# **Conflicts of interest**

There are no conflicts to declare.

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