

# Lignin-derived Porous Graphene for Wearable and Ultrasensitive Strain Sensors

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1	Lignin-derived Porous Graphene for Wearable and Ultrasensitive Strain Sensors
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9	Abstract
10	This study aimed to explore lignin as a naturally occurring aromatic precursor for the
11	synthesis of LIG and further fabrication of ultrasensitive strain sensors for the detection of small
12	deformations. One-step direct laser writing (DLW) induced high quality porous graphene, so
13	called laser induced graphene (LIG), from kraft lignin under the conditions optimized for laser
14	power, focus distance, and lignin loading. An electrode based on the resulting LIG was facilely
15	fabricated by transferring LIG onto an elastomeric substrate (i.e., Dragon Skin <sup>TM</sup> ). The novel
16	LIG transfer was facilitated by spinning coating followed by water lifting, leading to full
17	retention of porous graphene onto the elastomeric substrate. The strain sensor was shown to be
18	highly sensitive to small human body motions as well as tiny deformations caused by vibrations.
19	It had a working range of up to 14% strain with a gauge factor of 960 and showed high stability
20	as evidenced by repetitive signals over 10,000 cycles at 4% strain. The sensor was also
21	successfully demonstrated for detecting human speaking and movement of pulse and eye.
22	Overall, the lignin-derived LIG can serve as excellent piezoresistive materials for wearable,

stretchable, ultrasensitive strain sensors with applications in human body motion monitoring andsound-related applications.

Keywords: laser-induced graphene, direct laser writing, lignin, strain sensor, human body
motion, vibration detection.

#### 27 1. Introduction

Strain sensors can be used as wearable and healthcare-oriented electronics for human 28 29 health monitoring. In general, there are two main categories of human body motion detections, 30 including large scale (bending, jumping, movements of arms and legs, etc.) and small scale (pulse, breath, and other tiny muscle movements).<sup>1-6</sup> A high-performance strain sensor would 31 require large working range, high sensitivity, long durability, low fabrication cost, etc. Numerous 32 methods, such as carbon nanotube (CNT),<sup>7-9</sup> nanoparticles,<sup>10</sup> graphene,<sup>1,11</sup> and laser-induced 33 graphene (LIG),<sup>12-17</sup> have been reported for developing high-performance strain sensor to fulfill 34 35 the increasing demands for human health monitoring. However, most strain sensors have limitations in sensing. For example, a CNT-based strain sensor with PDMS as an elastomeric 36 substrate had a low gauge factor ( $0.82 \sim 14$ ) although it exhibited excellent stretchability.<sup>8,9</sup> 37 38 Similarly, the strain sensor based on CVD-grown graphene showed high sensitivity in detection of small-scale human motions, but it can only be stretched at 7% strain or below, with a gauge 39 factor only up to 35.<sup>1</sup> Sensor configuration like sandwiched structure can significantly improve 40 gauge factors,<sup>18, 19</sup> but correlation between signals and applied strains would need to be further 41 improved. Therefore, further efforts would be needed to explore new piezoresistive materials and 42 43 advance sensor fabrication techniques for developing ultrasensitive, wearable, and stretchable 44 strain sensors with low cost.

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45	LIG has been explored as active materials for strain sensor fabrication, as
46	aforementioned. LIG can be facilely synthesized from various precursors using single-step direct
47	laser writing (DLW), exhibiting excellent electrical, chemical, and mechanical properties. <sup>14, 16, 20,</sup>
48	<sup>21</sup> More strikingly, it has intrinsic porous structures which typically cannot be achieved by
49	traditional techniques for graphene synthesis. Such attribute makes LIG a unique active element
50	for strain sensor fabrication. The LIG-based strain sensors particularly derived from polyimide
51	(PI) have been reported to outperform CVD-grown graphene sensors, <sup>22</sup> and show high sensitivity
52	to capture various human body motions (e.g., speaking, pulse detection, arm movements). <sup>12, 16, 17,</sup>
53	<sup>23, 24</sup> However, those sensors had the limited detection strain ranges (below 1% strain) and gauge
54	factors below 30 if the LIG-embedded PI film was used for a substrate for sensor fabrication. <sup>12</sup>
55	While stretchable strains can be realized by transferring PI-derived LIG onto an elastomer, the
56	gauge factors were still below 200 with significantly decreased sensitivity in small strain
57	detection. One major reason for comprised sensitivity should be drastically increased sheet
58	resistance (>20 k $\Omega$ ) associated with LIG transfer. <sup>15</sup> LIG synthesized from other precursors such
59	as cellulose paper was also used to fabricate on-chip, flexible strain sensors; however, the as-
60	prepared sensors had poor stretchability and a very low working range (0~0.3% strain) as well as
61	low gauge factors (only up to 42). <sup>25</sup> To fully explore the potential of LIG to be an active material
62	for strain sensors, technical barriers like LIG quality, sensitivity, and transferability would need
63	to be tackled.
64	Among various precursors (polyimide, wood, etc.) explored for LIG, <sup>25-29</sup> lignin stands out

to a unique natural precursor for graphene synthesis given its high-carbon and aromatic
structures. Compared to cellulose and hemicellulose, lignin is the most favorable constituent in
lignocellulosic biomass for LIG formation. Lignin separated from biomass, so called technical

68 lignin, is more processable than raw biomass like wood to form a suitable substrate for DLW and 69 yield higher quality of LIG. On the other hand, in puling mills and biorefinery facilities lignin is 70 generated at large quantities (>100 million dry tons annually) as waste streams.<sup>30</sup> Therefore, 71 upgrading lignin into graphene-based materials can add additional revenues and be of great 72 significance to paper and pulp and biorefinery industries. As lignin is typically recovered in a 73 powder form from biorefinery, lignin-based films can be made using a binder such as PVA, 74 PEO, and cellulose nanofibers (typically not contributing to LIG formation) to well suit direct 75 laser writing.<sup>31-33</sup> This strategy would enable mass production of lignin derived LIG as renewable 76 advanced carbon materials via roll-to-roll manufacturing process. Prior studies reported that lignin derived LIG has high potential in energy storage and sensing applications.<sup>26, 28, 32</sup> 77 78 However, research on lignin to LIG and subsequent applications is still limited compared to PI, a 79 synthetic polymer precursor. For strain sensing, little research has been done to explore lignin derived LIG as an active material. A recent study reported a strain sensor based on lignin derived 80 81 LIG/PDMS for human motion monitoring but with low sensitivity as reflected by a quite low 82 gauge factor of 20 because of low LIG quality.<sup>34</sup> Therefore, more research efforts are needed for 83 tailoring properties of lignin derived LIG as an excellent piezoresistive material. Further 84 exploration of lignin-to-LIG upgrading pathway would need to address technical barriers to both 85 LIG and lignin-specific precursor. For example, for sensing application, the sensor fabrication 86 could involve LIG transfer from a lignin-based substrate (e.g., film); however, carbon loss, a 87 common issue with LIG regardless of precursors, should be mitigated. Another consideration is to obtain high quality LIG like crystalline graphene since abundant oxygen-containing 88 89 functionalities in lignin and biomass would contribute to noticeable defects in LIG. To this end, 90 it would be very necessary to tune lasing conditions to ensure LIG quality.

91	In this study, we demonstrated the feasibility to fabricate a stretchable and wearable
92	strain sensor using lignin derived LIG. The LIG was synthesize from kraft lignin by DLW and
93	further transferred onto Dragon Skin <sup>TM</sup> for sensor fabrication. The LIG transfer was facilitated
94	by a novel technique based on spinning coating coupled with water lifting, without noticeable
95	modification on 3D structure of LIG. The key parameters optimized for lasing included precursor
96	loading, laser power, and defocus distance. The resultant LIG was characterized for
97	microstructure and graphene domains. The fabricated strain sensor showed ultrasensitive
98	response to external stimuli including strains up to 14% and vibrations (5-135 Hz) with high
99	stability. The excellent sensing performance underscored that lignin derived LIG can serve as a
100	highly promising piezoresistive material for detecting tiny deformations induced by strains
101	(below 6%) and vibrations.

## **102 2. Methods**

# 103 **2.1. LIG synthesis**

104 A lignin-based film was prepared by mixing kraft lignin with polyvinyl alcohol (PVA). In 105 brief, PVA (Mw=146,000-186,000) was first dissolved at 9 wt% in deionized (DI) water with 106 continuous stirring at 90 °C for 40 min. Kraft lignin of 0.5, 1.0, or 1.5 g was dissolved in 10 g of 2 wt% NaOH solution. Then, the kraft lignin solution was mixed with the PVA solution with 1:1, 107 108 w/w until clear, dark brown liquid was obtained. The liquid was cast onto a plastic Petri dish 109 with 9 cm in diameter and dried at 35 °C in an incubator with humidity of 25-50% for about 30 h. 110 After drying, the film was peeled off from the Petri dish and attached to a glass slide using 111 adhesive tape for laser writing.

A CO<sub>2</sub> laser (30 W, 10.6 μm, Universal Laser Systems VLS3.50 Laser Cutter/Engraver)
 was used for DLW. The process was conducted under ambient conditions, with 1000 pulses per

114 inch, 10% speed, varying z-axis distance from the focal plane (0.0, 1.0, 2.0, and 3.0 mm), and 115 varying power (20%, 30%, and 40% of the upper limit of laser power setting). An area of 116  $3.0 \times 0.5$  cm<sup>2</sup> on the lignin-based film was subjected to laser writing. LIG samples were denoted 117 as LxPy with 2 mm z-axis defocus distance unless stated otherwise, where x stands for the initial 118 mass loading of lignin in a film (i.e., 0.5, 1.0, or 1.5 g) and y for laser power level (i.e., 20%, 119 30%, or 40%).

#### 120 2.2. Fabrication of strain sensors

A strain sensor was fabricated by transferring LIG onto Dragon Skin<sup>TM</sup> (Dragon Skin<sup>TM</sup> 121 10 FAST Silicones), an elastomeric substrate. First, Dragon Skin<sup>TM</sup> was prepared by mixing two 122 123 prepolymers (Part A: Part B = 1:1, w/w) and then coated to LIG embedded onto the lignin film 124 by a spin coater (Model WS-650MZ-23NPPB) with a spin speed of 3000 rpm and an accelerated speed of 500 rpm/sec. After spinning coating, Dragon Skin<sup>TM</sup> was cured in a convection oven at 125 35 °C for 30 min. The Dragon Skin<sup>TM</sup>-coated film was then immersed in DI water for about 10 126 127 min to facilitate the detachment of LIG from the lignin film. The LIG/Dragon Skin<sup>TM</sup> composite 128 was then dried at 35 °C in a convection oven for 20 min, cut into a dimension of 3.8×0.85 cm<sup>2</sup>, 129 and used for the sensor fabrication. The composite was further attached with two copper wires on 130 both ends by conductive epoxy adhesive followed by air drying. Fig. 1 illustrates the sensor 131 fabrication from lignin derived LIG.

#### 132 **2.3.** Characterization

LIG embedded onto a film was used for characterization, unless stated otherwise. Raman
spectra were acquired on inVia<sup>™</sup> confocal Raman microscope at a wavelength of 633 nm. Sheet
resistance (Rs) was measured using Pro-4 Four Point Resistivity Systems (Lucas Labs, Gilroy,
CA). Morphology of LIG samples was examined using a scanning electron microscope (SEM,

FEI Quanta 600 FEG) equipped with a Bruker Quantax 200 Silicon Drift Detector and operated
at a 15 kV and 100 pA. High resolution transmission electron microscopy (HRTEM) images
were acquired on a JEOL-2100FFEI Tecnai G2 F30 300 kV microscope, and LIG power was
sonicated in ethanol for 10 min and imaged. X-ray diffraction (XRD) scanning was conducted
for LIG power at 0.02° step size with a scanning rate of 12 s/0.02° using Bruker SMART CCD
system.

## 143 2.4. Sensing evaluation

144 Signal patterns for sensing were acquired on an LCR meter (HIOKI IM3523 LCT meter) 145 operated at a DC voltage of 0.1 V. For strain sensing, the sensor was stretched at a varying strain 146 with a rate of 0.2 mm/min by LCR for testing. To detect vibration, a Permanent Magnet Shaker 147 LDS (B&K V203) was used to vibrate the sensor at a frequency of 0-135 Hz or an input voltage 148 of 0-2 V. The invoked amplitudes were proportional to the input voltage. At each level of 149 frequency or input voltage, the strain was kept being vibrated for 10 s. Small-scale human 150 motion detections were also performed, including speaking, eyeball movement, heartbeat (pulse), 151 and breath. The sensor was attached onto a tester's throat, wrist, belly using a biocompatible 152 adhesive gel for signal collecting.

### 153 **3. Results and Discussions**

## 154 **3.1.** Characterization of lignin derived LIG

Raman spectra reveal that all the LIG samples had three dominant peaks including D, G, and 2D peaks (Fig. S1). Specifically, a D peak at ~1360 cm<sup>-1</sup> reflects defects and bending of sp<sup>2</sup> carbon bonds; a G peak at ~1580 cm<sup>-1</sup> arises from the first order band of all sp<sup>2</sup> hybridized

bonds; and a 2D peak at  $\sim 2670 \text{ cm}^{-1}$  originates from the second order zone boundary phonons.<sup>35</sup>

High  $I_G/I_D$  and  $I_{2D}/I_G$  ratios indicate high degree of graphitization and quality graphene,

160 respectively. Laser power showed different effects on the films with different lignin loading. For 161 L0.5, the higher laser power better induced graphene (Fig. S1a). For L1.0, 30% laser power 162 appeared to be sufficient for inducing higher quality LIG with the  $I_G/I_D$  and  $I_{2D}/I_G$  of 2.7 and 0.77 163 (Fig. S1d&e), respectively, indicating the formation of few-layer graphene with higher quality 164 compared to the counterparts derived from other precursors (e.g., PI, polyetherimide, wood, cellulose paper) and even lignin reported in prior studies.<sup>25-29</sup> Further increasing lignin loading in 165 166 the film (L1.5) did not necessarily improve LIG quality regardless of laser power applied (Fig. 167 S1c). In general, high laser power can lead to sufficient carbonization and further graphitization.<sup>26, 29</sup> However, laser power, if too high, can cause detrimental effects on LIG 168 169 quality due to severe photothermal irradiation. It should be noted that PVA in the lignin-based composite film would be decomposed and released as gaseous products (H<sub>2</sub>O, CO<sub>2</sub>, etc.) during 170 171 laser scribing rather than contribute to LIG synthesis. The Raman results show that the threshold 172 of laser power for inducing porous graphene would also be related to lignin loading in a 173 substrate. In the case of L1.5, laser parameter would need to be further optimized due to higher 174 lignin loading. In consistence with Raman analysis, the LIG resulting from L1.0 films exhibited 175 at least 2-fold lower Rs than that from the other two films and synthetic polymers-/biomass-176 based precursors previously reported (Table S1), with the lowest Rs (4.5  $\Omega/sq$ ) being observed 177 with 30% power level (P30) (Fig. S1f). In addition to optimal laser power, defocused lasing 178 appeared to favor LIG formation compared to focused lasing. Fig. S2 depicts the effects of laser 179 scribing as a function of z-axis defocus when 30% power level was applied. With the z-axis 180 defocus distance increased from 0 (on the focal plane) to 2 mm, the  $I_G/I_D$  and  $I_{2D}/I_G$  ratios reached the highest values for L1.0 (Fig.S2e). Further increase in the defocus distance to 3 mm led to 181 182 comprised LIG quality regardless of lignin loading in the films.

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183	The LIG resulting from L1.0P30 was further examined for ultrastructure given its
184	impressive Raman spectrum and low Rs. The HRTEM images showed fringe-like pattens of LIG
185	with <i>d</i> -spacing of 0.355 nm between two neighboring (002) planes in graphitic carbon (Fig. 2d).
186	This evidence further supported the presence of graphene and suggested few-layer graphene
187	domain randomly oriented in the LIG. The XRD pattern showed characteristic peaks centering at
188	$2\theta=25.9^{\circ}$ (assigned to (002) reflection) and $2\theta=42.9^{\circ}$ (assigned to (100) reflection) (Fig. 2b),
189	with an interlayer spacing of 0.344 nm, indicating that L1.0P30 was well graphitized with high
190	crystallinity. Overall, well-identified graphene domains in L1.0P30 proved successful
191	transformation of lignin into graphene.

## 192 **3.2.** Characterization of LIG/Dragon Skin<sup>TM</sup> composites

193 L1.0P30 was selected for the fabrication of strain sensors by transferring the LIG to Dragon Skin<sup>TM</sup>. LIG transfer was assisted by spinning coating followed by water lifting. 194 Specially, after immersing the whole piece of Dragon Skin<sup>TM</sup> spin-coated film in water for about 195 196 10 min, the uncarbonized lignin film was automatically detached from LIG. No residual LIG was 197 found in either the detached lignin film or water, while the transferred LIG showed well retained hierarchical microstructure onto Dragon Skin<sup>TM</sup> as discussed below. The LIG before and after 198 199 the transfer was characterized and compared. The SEM images show that rigid and clear lines 200 were formed orderly along with traces of CO<sub>2</sub> laser on the surface of pre-transferred LIG (Fig. 3a 201 &S3a&e). At higher magnification, LIG showed foam-like microstructure as typically observed 202 with LIG synthesis from other precursors (Fig. 3d &S3). The porous structure resulted from 203 rapid release of gaseous molecules (CO<sub>2</sub>, H<sub>2</sub>O, etc.) as byproducts generated from laser-assisted 204 oxidation and decomposition of lignin film. Water lifting enabled facile transfer of LIG without 205 noticeable loss of LIG. The cross-sectional SEM images also suggested complete transfer of LIG

layer onto Dragon Skin<sup>TM</sup> with similar thickness (115µm) before and after transfer (Fig. 3c&f)
and the total thickness of the LIG/Dragon Skin<sup>TM</sup> composite of ~150µm. Moreover, this finding
suggested sufficient infiltration and curing of Dragon Skin<sup>TM</sup> into interconnected network of
LIG. Overall, water lifting method can overcome common issues associated with LIG transfer
onto an elastomeric substrate, such as carbon loss and disruption of microstructure. As a result,
LIG/Dragon Skin<sup>TM</sup> can well preserve intrinsic functional properties of LIG and exhibit excellent
sensing capability as discussed below.

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# **3 3.3. Sensing strains and vibrations**

214 The fabricated strain sensors were evaluated for sensitivity in response to two external 215 stimuli, i.e., strains and vibrations. Like many LIG-based strain sensors, the working mechanism 216 of the proposed strain sensor is proposed to be resistance change caused by increasing gap 217 distance while applying strains. Strikingly, the strain sensor demonstrated ultrasensitivity 218 compared to those based on PI-derived LIG and many other materials (Table S2). For example, 219 the gauge factors of strain sensors based on pristine PI-derived LIG reached only up to 160, about 17% of the maximum value given by the proposed strain sensor.<sup>12, 15, 36</sup> The R<sub>0</sub> of the strain 220 221 sensor was around 300  $\Omega$ , which made it suitable for low power applications. Fig. 4a depicts that 222 the sensor experienced three stages throughout the stretching and the photo in Fig. 4b shows how 223 the sensor was stretched vertically. Its gauge factor was initially about 100 when the strain was 224 below 4%. The sensor also showed good sensitivity even when a strain was applied as low as 0.1% (Fig. S4a). The gauge factor became 160 when the strain was increased from 4% to 9%. 225 226 When the sensor was stretched further from 9% to 14%, the sensor exhibited a superb gauge factor of ~960. Notably, the signals at three stages had good linearity, i.e.,  $R^2 > 99$  for Stages 227 1&2 and  $R^2 > 98$  for Stage 3. The sensor was also tested for reversibility in a stretch-release 228

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cycle for hysteresis test. It showed a low degree of hysteresis (<7%) at 9% strain, suggesting

230	good reversibility (Fig. S4b). Compared to strain sensor based on conventional piezoresistive
231	materials such as CNT and nanowires, the LIG-based strain sensors showed much lower
232	hysteresis degree (Table S2).
233	With the strain range below 9%, the two responding stages can be explained by the fact
234	that with larger strains being applied, the crack between graphene flakes would keep expanding
235	and change into a new junction consisting of reduced conductive paths, leading to the increased
236	gauge factor. The evolution of the crack structure in LIG at different strains applied is also
237	shown in Fig. 4d-g. Within the strain range of 9%-14%, the gauge factor was remarkably
238	increased due to the more enlarged cracks causing more irregular signals. However, when the
239	crack was too large, the electrical connection between responding stages would be broken, with
240	the maximum strain being applied at 14%. The stability of signal was tested by applying
241	different strains (i.e., 2%, 4%, 6%, 9%) (Fig. 4b). For each applied strain, the device kept being
242	stretched for 50 s. It was found that the signals of relative resistance change ( $\Delta R/R_0$ ) were stable
243	and repeatable at a given strain. Compared to start-of-art strain sensors especially based on LIG
244	(Table S2), the proposed strain sensor exhibited significantly improved gauge factor (as high as
245	960) and markedly lower detection limits (as low as 0.1%) for small-scale strain detections (Fig.
246	S4a). Considering excellent sensing linearity, it would be interesting to test and compare with
247	other device configurations like sandwiched structures for ultrasensitivity. <sup>18, 19</sup> The sensor was
248	further evaluated for stability over 10,000 stretch-release cycles at 4% strain. There were no
249	significant signal fluctuations being observed after 10,000 cycles (Fig. 4c). The sensor's
250	resistance only increased from around 300 to 450 $\Omega$ along with ~10% increase in the gauge
251	factor. The findings demonstrated the strain's superb stability of conductive network and

reliability for repeated uses, which was shown to be superior to most start-of-art strain sensorsfabricated from PI-derived LIG and CNT (Table S2).

254 In addition to strain sensing, the sensor was also shown to be ultrasensitive to vibration, 255 another form of tiny deformation. The resistance change of the strain sensor was continuously 256 monitored throughout the vibration. Fig. 5a&b shows resistance in response to a frequency 257 ranging from 5 to 135 Hz with an increment of 5 Hz when the input voltage was kept at 2V. 258  $\Delta R/R_0$  first increased along with the increasing frequency until reaching the highest value at 85 259 Hz. When the frequency was increased further,  $\Delta R/R_0$  drastically dropped to 35% at 100 Hz and 260 then increased. The spike in  $\Delta R/R_0$  at 85 Hz suggested that this frequency could be at or near 261 resonant frequency/natural frequency of the sensor. As a result, the sensor can vibrate with a 262 larger amplitude than that at non-resonant frequencies. Larger vibration amplitude can lead to 263 larger cracks and subsequently more significant signal changes. The electrical resistance of 264 sensor was further measured against amplitudes range controlled by input voltages from 0.2-3V 265 when applying a frequency of 85 Hz (Fig. 5c). An excellent linear correlation between  $\Delta R/R_0$ 266 and input voltage was observed ( $R^2 > 99$ ) (Fig. 5d). These results indicate that lignin derived LIG has exceptional piezoresistive property that can also be explored for sound-related applications. 267

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# 3.4. Sensors for detecting speaking

The sensor was used as a speaking detector given the frequency of human speaking ranging from 50 to 300 Hz. Fig. 6&S5 depicts representative strain's performance on detecting human speaking by sensing throat muscle movements and vibrations (Fig. S5a). Swallow, one of typical muscle movements, was also detected, showing distinguishing patterns (Fig. 6a). When one was asked to speak two example words "today" and "beautiful" as well as the whole sentence "Today is beautiful" 4-5 times, the signature peaks corresponding to two individual

275 words were well identified in the whole sentence (Fig. 6b-d). More intriguingly, when the 276 speaking testing was extended to a longer sentence "Laser-induced graphene from lignin-based precursor has many good properties." three times, the signature pattens of individual words were 277 278 clearly detected, with good repeatability (Fig. S5b-m&S7a). These findings suggested that the 279 sensor can distinguish tiny deformations caused by speaking, with reliable and repeatable 280 performance. The sensing mechanism for speaking is a combined effect of strains and vibrations. 281 On one hand, the muscle movements of throat during speaking are unique and repeatable, which 282 can be captured as a strain stimulus by the sensor. On the other hand, sound generated during 283 speaking causes vibrations to the sensor. The sensor's exceptional sensitivity to strains, 284 vibrations, or combined suggests its great potential to be used as a sophisticated speaking 285 detector. One example would be to aid people with disability in speaking by coupling artificial 286 intelligence.

# 287 3.5. Sensor for small-scale human motion detection

288 The sensor's performance in detecting human motions, including eyeball movement, 289 heartbeat (pulse), breath, and seismocardiography (SCG), was depicted in Fig. 7. The sensor 290 showed excellent and stable responses. For eyeball movement, the sensor was horizontally attached at the lower part of one's eves for eve blink, fast eve blinks, looking up, looking down, 291 292 looking left, and looking right in order, each three times except fast eye blinks. Fig. 7a shows 293 that the strain sensor can provide instant and clear feedback with the unique, stable, and 294 repeatable resistance change patterns for different eve movements. Muscle movements 295 controlling an eyeball to different directions were at tiny scales. Compared to reported eyeball 296 movement sensors.<sup>37, 38</sup> the proposed strain sensor offers a simpler way to monitor real-time eve movement detection for human health monitoring and virtual reality system.<sup>38, 39</sup> Furthermore, 297

298 the strain sensor was attached to one's chest for breath detection, including belly breath (deep 299 breath) and chest breath (normal breath) (Fig. 7b). As breath disorder is usually related to 300 pulmonary diseases (asthma, COVID-19, etc.), the strain sensor's ability to distinguish breath 301 patterns would be highly useful for respiratory system monitoring. 302 SCG is a non-invasive technique used for cardiovascular disease diagnostics by 303 measuring vibrations around chest caused by one's heartbeats. The strain sensor was also 304 evaluated for its performance in SCG signal recording. Fig. 7c depicts the signal patterns to body 305 microvibrations during one cycle of holding breath including systole phrase and diastole phrase. 306 The corresponding region of signal for two phrases as well as their related heart activity are 307 shown in the inset of Fig. 7c. Specially, mitral valve closure (MC), aortic valve opening (AO), 308 and rapid blood ejection (RE) form systole phrase, as well as a rtic valve closure (AC), mitral 309 valve opening (MO), and rapid blood filling (RF) from diastole phrase are annotated. The SCG 310 acquired from different positions around the chest including its low part was distinguishing given 311 different structure of human body (Fig. 7c&S6). The signals showed good repeatability across all 312 the testing cycles based on an error band analysis (Fig. S7b). Although the relationship between 313 SCG and electrocardiogram (ECG) is not well understood, recent advances, including more 314 accessible sensors and better algorithms enhanced by machine learning, have demonstrated the 315 potential of SCG in monitoring the health of patient with heart disease.<sup>40</sup> The proposed strain 316 sensor, which can be facilely fabricated from lignin derived LIG at low cost, can serve the 317 purpose for easy and rapid SCG detection. 318 Pulse wave detection is another key parameter in human health monitoring. As shown in

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Fig. 7d, the sensor can distinguish the deformations caused by percussion wave (P), tidal wave

(T), and diastolic wave (D), which are important indicators to one's cardiovascular health
 monitoring.<sup>41</sup>

# 322 4. Conclusions

323 Lignin-derived strain sensor had excellent performance in detection of small deformations caused by strains and vibrations. It exhibited a large working range for sensing up 324 325 to 14% strain, with gauge factor as high as 960 and low hysteresis. A good linear relationship 326 between the relative resistance changes and strains was observed. It was also sensitive to 327 vibrations in response to change in frequency and amplitudes. Moreover, the strain sensor can 328 capture tiny deformations based on human speaking, eyeball movements, breath, 329 seismocardiography, and pulses. In all the tests, the strain sensor showed excellent stability and 330 repeatability and was able to generate distinguishing signal patterns of deformations caused by 331 various external stimuli. This work indicated that LIG induced from lignin can be used as an 332 excellent piezoresistive material to fabricate wearable, high-performance, ultrasensitive strain sensors for detection of strain-/vibration-induced deformations toward applications such as 333 334 healthcare monitoring and sound sensing.

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## **Figure captions**

Fig. 1. Schematic diagram of strain sensor fabrication from lignin derived LIG.

**Fig. 2.** Characteristics of LIG (L1.0P30). (a) Raman spectrum. (b) XRD spectra of LIG and pristine film (L.0). (c, d) TEM images: scale bars of 20 and 5 nm, respectively.

**Fig. 3.** SEM images of L1.0P30 before and after transfer. (a, d) Top view before transfer: scale bars of 100 and 20  $\mu$ m, respectively. (b, e) Top view after transfer: scale bars of 100 and 20  $\mu$ m, respectively. (c, f) Cross-sectional views before and after transfer, respectively: scale bar of 100  $\mu$ m.

**Fig. 4.** Strain sensor's sensitivity to strain. (a)  $\Delta R/R_0$  in response to up to 14% strain, with the inset demonstrating the signal change at a stain up to 1%. (b)  $\Delta R/R_0$  over the course of testing at different strain levels, with the inset showing the photo of strain sensor stretching. (c) Sensing stability at 4% strain over 10,000 cycles, with the inset showing the first and last 10 cycles (left and right). (d-g) SEM images of evolution of microgap with an increase in strain: scale bar of 100 µm.

**Fig. 5.** Strain sensor's sensitivity to vibration. (a)  $\Delta R/R_0$  at different vibration frequencies (5-135 Hz) when the input voltage for vibration was applied at 2 V. (b) Averaged  $\Delta R/R_0$  at different frequencies (5-135 Hz). (c)  $\Delta R/R_0$  at different input voltages (0.2-3 V) for vibration when a vibration frequency was applied at 85 Hz. (d) Linear correlation between  $\Delta R/R_0$  and input voltages. The photo in (a) shows the testing setup in which the sensor was attached to a permanent magnetic electrodynamic shaker where input voltages were provided.

**Fig. 6.** Strain sensor working as a speaking detector. (a)  $\Delta R/R_0$  of detecting swallow. (b-d)  $\Delta R/R_0$  of detecting speaking "beautiful," "today," and "today is beautiful", respectively. The photo in (a) shows the sensor's placement for detecting swallowing and speaking.

**Fig. 7.** Strain sensor for human motion monitoring. (a) Signal patterns of monitoring eyeball movements. (b) Signal pattern of monitoring belly and chest breath. (c) Signal pattern of seismocardiography. (d) Signal pattern of pulses. The photos show the sensor's placement at different positions for testing. The inset in (c) annotates mitral valve closure (MC), aortic valve opening (AO), and rapid blood ejection (RE) form systole phrase, as well as aortic valve closure (AC), mitral valve opening (MO), and rapid blood filling (RF) from diastole phrase. The inset in (d) shows the signal changes caused by percussion wave (P), tidal wave (T), and diastolic wave (D).



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