



# **Exploiting Heat Transfer to Achieve Efficient Photoelectrochemical CO2 Reduction under Light Concentration**







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### 14 **Abstract**

15 Photoelectrochemical (PEC) conversion of carbon dioxide into valuable chemicals and fuels 16 represents a promising path towards combating anthropogenic  $CO<sub>2</sub>$  emissions. However, the 17 limited conversion efficiencies, operation lifetimes and  $CO<sub>2</sub>$  utilization efficiencies of PEC 18 devices currently prohibit their application beyond the laboratory scale. Here, a wireless device 19 converting  $CO<sub>2</sub>$  and water into carbon monoxide and hydrogen at a peak solar conversion 20 efficiency exceeding 16% under an illumination intensity of 5 suns is demonstrated. A CO/H<sub>2</sub> 21 product ratio between 10 - 20 is measured during a 17 h stability test. Fluctuations in device 22 performance are rigorously analyzed via deconvolution of electrochemical and photoabsorber 23 contributions. It is demonstrated that beneficial heat dissipation is enabled by wireless 24 integration of the photoabsorber and electrocatalyst components, accounting for roughly 10% 25 of the achieved conversion efficiency, an achievement unattainable with physically separated 26 photoabsorber and electrolyzer components.

# 27 **Introduction**

28 While mean planetary temperatures continue to rise due to increasing anthropogenic 29 carbon dioxide emissions, many countries still depend heavily on fossil fuels.<sup>1,2</sup> Converting  $30$  exhaust  $CO<sub>2</sub>$  into fuels and other valuable chemicals provides a promising pathway to reduce 31 the impacts of climate change and move away from fossil fuels. Electrochemical (EC)  $CO<sub>2</sub>$ 32 reduction is one of several options for industrial-scale  $CO_2$  conversion.<sup>3,4</sup> Ideally, such a process 33 will be driven by renewable energy sources such as wind or sunlight.

34 Solar-driven CO2 reduction may be facilitated by connecting a photovoltaic (PV) 35 element to a dark electrolyzer. Yet, these designs generally do not exploit the potential heat 36 exchange between the photoabsorber and electrolyte, which is expected to improve the overall 37 efficiency in fully-integrated, photoelectrochemical (PEC) cells. Such heat exchange is 38 especially important when concentrated sunlight is used to illuminate the photoabsorber.<sup>5</sup> 39 Liquid electrolyte that flows through the electrolyzer cools the PV, significantly increasing its 40 efficiency at elevated light concentrations. The heated electrolyte subsequently raises the 41 temperature of catalytically active sites, improving the reaction kinetics. While conceptually 42 promising, to date this beneficial heat exchange has rarely been demonstrated in PEC cells, due 43 to system complexity and insufficient data collection and analysis.<sup>5</sup>

44 Until now, systems consisting of physically separated PV and electrolyzer components 45 have outperformed wireless, PEC devices. Solar-to-fuel (STF) conversion efficiencies up to 46 20% have been demonstrated with the former at 1 sun illumination intensity,  $6-13$  while the latter 47 have reached only 10%.14–17 Integration of the photoabsorber into the strongly alkaline 48 environment usually present in  $CO<sub>2</sub>$  electrolyzers introduces durability challenges, hindering 49 the employment of beneficial device design strategies developed for dark  $CO<sub>2</sub>$  electrolysis. So 50 far there have been very few reports of membrane-electrode assembly (MEA)-type, PEC 51 devices for  $CO_2$  reduction,<sup>14,18</sup> despite persuasive arguments that MEA devices can offer lower

52 operating voltages than corresponding flow cells, with potentially increased lifetimes.<sup>19,20</sup> 53 Similarly, there have been very few demonstrations of solar-driven  $CO<sub>2</sub>$  electrolysis using 54 concentrated sunlight, even though semiconductor photoabsorbers are generally more efficient 55 under modest light concentration if sufficient cooling is provided and the series resistance is 56 low enough. $21-26$ 

 $57$  Here, we describe a fully-integrated, wireless, PEC device converting  $CO<sub>2</sub>$  to carbon 58 monoxide at a peak solar-to-CO (STCO) efficiency of 15%, while producing hydrogen at a 59 solar-to-H<sub>2</sub> (STH) efficiency of 1%, resulting in a combined STF efficiency of 16%. This marks 60 a 60% improvement in STF efficiency relative to previous reports of fully-integrated, wireless, 61 PEC devices for  $CO_2$  conversion.<sup>14–17</sup> During a 17 h trial, the  $CO/H_2$  product ratio ranges 62 between 20:1 and 10:1 with no permanent degradation observed in device performance. 63 Fluctuations in the combined STF efficiency between 6% and 16% are shown to be the result 64 of bubble accumulation in the relatively thick anode layer. Stability tests revealed that the 65 photoabsorber temperature plays an important role for the STF efficiency, especially when the 66 operating point lies near the maximum power point (MPP) of the PV. Consequently, by 67 analyzing the operating current and voltage, we demonstrate how PV cooling via liquid 68 electrolyte enabled noteworthy device performance enhancements.

69

### 70 **Results and Discussion**

#### 71 **Device architecture**

72 At the center of our device lies the PV-integrated membrane (PIM), consisting of a 73 triple-junction PV surrounded by a Selemion membrane, as detailed in previous reports<sup>14,27</sup> 74 (**[Fig. 1](#page-6-0)**a and **[Fig. 1](#page-6-0)**c). Similar to a MEA, the PIM is sandwiched by two carbon paper substrates 75 coated with catalysts, leaving the PV unobstructed. On the cathode side, a gold catalyst is 76 deposited on a micro-porous layer (MPL) containing polytetrafluoroethylene, which creates a 77 hydrophobic micro-environment on the gas diffusion electrode.<sup>28</sup> Humidified  $CO<sub>2</sub>$  flows 78 through the serpentine channels in the acrylic front endplate and diffuses through the carbon 79 paper towards the catalyst. On the catalyst surface,  $CO_2$  and  $H_2O$  are converted to CO and  $H_2$ , 80 depending on local reaction conditions,  $14,28-31$  while generating hydroxide ions that diffuse 81 through the anion exchange membrane and react to form  $O_2$  on the anodic catalyst surface. On 82 the anode side, the Ni catalyst is directly sputtered on the carbon fibers (no MPL), creating a 83 more hydrophilic environment than at the cathode. This allows the potassium hydroxide anolyte 84 to wet the surface of the catalyst as it flows through channels machined in the anodic endplate. 85 The PV sits in the middle of the catalytic structure and can be directly illuminated 86 through the acrylic endplate (cathode side), reducing path-dependent light attenuation 87 compared to devices that include catalysts or electrolytes in the illumination path. However, 88 since the catalyst is placed around the PV, the effective illumination area (with respect to the 89 overall device footprint) is reduced unless a lens is used to concentrate the light that would 90 otherwise fall on the carbon paper onto the PV (**[Fig. 1](#page-6-0)**b). Furthermore, employment of a lens 91 may significantly reduce the material costs, as the area of the high-efficiency photoabsorber can 92 be reduced compared to the total illumination area. The light concentration factor using a 93 Fresnel lens is adjusted by changing the distance between the lens and the PEC cell (**Fig. S1**, 94 ESI).



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<span id="page-6-0"></span>**Fig. 1**. Fully-integrated, wireless, PEC device used for light concentration testing. (a) PIM device performing PEC  $CO_2$  reduction to  $CO$  and  $H_2$ . We note that the cathode outlet stream also contains unreacted  $CO<sub>2</sub>$  and the device design and operating conditions dictate the overall  $CO<sub>2</sub>$  utilization. (b) Fresnel lens concentrating the incoming light onto the PV of the PIM device. Moving the (green) cathodic inlet and outlet to the side of the cell enabled closer placement of the Fresnel lens to the PEC device. (c) Cross-section image showing the operating principle of the PIM device. The chemical reactions are shown without coefficients and the thickness of the layers are not to scale.

95

## 96 **Electrochemical optimization for operation under light concentration**

97 During PEC operation, the catalytic, EC components are coupled directly to the PV. 98 Electroneutrality demands that the flux of electrical currents through series-linked components 99 must be equal. As a result, the measured device current will be determined by the current-100 limiting component (PV or EC). To avoid significant efficiency penalties in a PEC device due 101 to the shape of the PV polarization curve, the achievable EC current should equal or exceed the

102 PV short circuit current  $(I_{\rm sc})$  at a voltage slightly lower than at the MPP. Here, we define a 103 critical voltage  $(V_{\text{crit}})$ , to prevent current drops in the PV polarization curve larger than 0.2% of 104 *I*<sub>sc</sub>. In other words,  $V_{\text{crit}}$  marks the point at which the PV polarization curve transitions from a 105 flat line to an exponentially falling curve (see Fig. S2, ESI). In our case  $V_{\text{crit}}$  is ~2.1 V.

106 As mentioned above, illumination of the non-photoactive carbon paper should be 107 avoided to maximize fuel production per total device area. Concentrating the light onto the 108 photoabsorber increases its total output current provided to the catalytically active sites. 109 Therefore, increasing the light concentration while keeping the catalyst area unchanged, 110 requires optimizing the electrochemical activity of the catalyst to enable higher catalytic current 111 densities at the same operating voltage.

112 The minimum concentration factor (*c*) to avoid losses of illumination area can be 113 determined according to the respective areas of the catalyst  $(A_{cat})$  and PV  $(A_{PV})$ :

114 
$$
c = \frac{A_{\rm PV} + A_{\rm cat}}{A_{\rm PV}} \cdot \#(1)
$$

115 This results in the expected PV current under light concentration:

$$
I_{\rm PV} = j_{\rm sc, 1\,sun} \times c \times A_{\rm PV} \, \text{#}(2)
$$

117 with  $j_{\rm sc, 1 \, sun}$  describing the maximum achievable PV current density under short circuit 118 conditions and 1 sun illumination intensity (no concentration).

119 Since the PV (equation 2) and EC ( $I_{EC} = j_{EC} \times A_{cat}$ ) currents need to be equal, the EC 120 current density  $(j_{EC})$  can be calculated as a function of the PV and catalyst areas:

121 
$$
j_{EC} = j_{sc, 1 \text{ sun}} \times \left(\frac{A_{PV}}{A_{cat}} + 1\right) \cdot \#(3)
$$

122 Therefore,  $j_{\text{EC}} \ge j_{\text{sc, 1 sum}}$  and using the parameters of the chosen PV ( $A_{\text{PV}}$  = 0.94 cm<sup>2</sup>,  $j_{\text{sc, 1 sum}}$ )  $123 = 12.22$  mA cm<sup>-2</sup>) results in the blue current density as a function of the catalyst area curve 124 shown in **[Fig. 2](#page-8-0)**a.



<span id="page-8-0"></span>**Fig. 2**. Optimization of the EC performance. (a) Operating voltage and current density of the catalysts as a function of the geometric area. Optimization of the operating conditions for the baseline device was necessary to avoid current drops larger than 0.2% of *I*sc by decreasing the operating voltage below  $V_{\text{crit}}(2.1 \text{ V})$ . (b) EC CV curves of the baseline and optimized device, corresponding to the orange voltage curves in (a).

125 With a cyclic voltammogram (CV) measurement, the EC current density can be 126 determined as a function of the applied voltage (**[Fig. 2](#page-8-0)**b). With the data from the CV and the 127 pre-determined current density curve from **[Fig. 2](#page-8-0)**a, the expected operating voltage can then be 128 determined as a function of the catalyst area (orange curves in **[Fig. 2](#page-8-0)**a). Even though a larger 129 catalyst area significantly reduces the operating voltage, using operating conditions similar to 130 those described previously<sup>14</sup> (baseline device) will always lead to an operating point above  $V_{\text{crit}}$ . 131 Since this would result in significant efficiency penalties, the operating conditions required 132 optimizing catalyst activity to lower the operating voltage.

133 Similar to previous studies,  $32-34$  we found that Ir and Ni perform equally well as oxygen 134 evolution reaction catalysts under the anodic conditions used in this study (**Fig. S3**, ESI). Hence, 135 Ni catalysts supported on carbon paper were used for all subsequent experiments. The cathode's 136 catalytic environment was tailored through the use of an MPL on the gas diffusion electrode, 137 which resulted in favorable performance and shifted the product distribution towards CO

138 (**Fig. S4**, ESI). Moreover, the cathodic product ratio became independent of the cathodic inlet 139 humidity when using the MPL (not shown here), in contrast to our previous findings without 140 the MPL, which showed that higher humidity skewed the product distribution towards  $H_2$ .<sup>14</sup> 141 The hydrophobic nature of the MPL prevents water molecules entering the cathode chamber 142 from reaching the catalyst sites, resulting in high CO/H<sub>2</sub> product ratios throughout a broad range 143 of cathodic humidities.

144 In addition, raising the concentration and flow rate of the KOH anolyte increased the 145 current at an applied potential of 2 V (**Fig. S5 and S6**, ESI). Even though 5м KOH showed the 146 highest currents, 2м KOH provided sufficient electrochemical performance at higher flow rates, 147 while keeping the concentration of KOH at a minimum. However, it is important to note that 148 higher KOH flow rates lead to increased  $CO<sub>2</sub>$  crossover through the membrane, accelerating 149 acidification of the anolyte. Such acidification may result in a rapid current degradation, when 150 a small anolyte reservoir is used (**Fig. S7**, ESI).

151 In summary, the addition of the MPL to the cathodic gas diffusion electrode, combined 152 with a higher KOH anolyte concentration and flow rate, increased the current by a factor of 1.6- 153 1.7 in the region of 2-2.1 V (optimized device in [Fig. 2](#page-8-0)b). For a fixed PV area of 0.94 cm<sup>2</sup>, this 154 optimized performance now allows an expected operating voltage below  $V_{\text{crit}}$  for any catalyst 155 area larger than 2.1 cm<sup>2</sup>, with no significant voltage reductions occurring for areas beyond 156 5 cm<sup>2</sup>. Given these considerations, we chose a catalyst area of 4 cm<sup>2</sup> to allow for small voltage 157 fluctuations. The relative PV and EC areas thus require a light concentration factor of  $\sim$ 5 to 158 minimize illumination of the non-photoactive EC element, while ensuring homogeneous 159 illumination of the PV component in the integrated assembly.

160

#### 161 **Photoelectrochemical operation**

162 Prior to PEC operation, the PV polarization response of a fully-assembled PIM device 163 was evaluated using an external circuit at 1, 3 and 5 suns. The short circuit current density

164 scaled linearly with the light intensity (12.34, 36.23, and 62.07 mA cm-2, respectively), the open 165 circuit voltage increased from 2.65 V to 2.74 V, and only a minor reduction in fill factor 166 (< 2% sun-1) was observed (**Fig. S8**, ESI). The PV efficiency was between 27-28% for all three 167 illumination intensities. We then added electrolyte (2м KOH, 5.9 mL min-1) and humidified 168  $CO<sub>2</sub>$  (60 sccm) and conditioned the catalysts by biasing the system in the dark at 2 V for 50 min 169 to achieve EC performance equilibration. This step allowed for the membrane to become 170 equilibrated with anolyte and fully wet the cathode catalyst, which increased its 171 electrochemically active surface area (**Fig. S9**, ESI).14,35–37

172 After equilibration, we established a direct connection between the PV and EC 173 components to operate the PIM device in PEC mode, without external bias, at the three 174 illumination intensities. At 1 sun, the current was stable at 12 mA during a 1 h trial, with a low 175 operating voltage near 1.7 V due to the low EC current density needed at this light intensity 176 (Fig. S10, ESI). As expected from the low operating voltage,<sup>14</sup> the product distribution heavily 177 favored CO production, with a CO/H2 product ratio near 30:1. By increasing the light intensity 178 to 3.02 suns, the PEC current increased to 36 mA at 1.9 V (**Fig. S11**, ESI) and was again stable 179 for a period of 1 h. The CO/H<sub>2</sub> product ratio was approximately the same at 30:1 as with 1-sun 180 illumination.

181 At 5.05 suns, the optimized illumination intensity, the PIM device yielded a peak current 182 near the short circuit current of the PV (**[Fig. 3](#page-11-0)**a and **Fig. S12**a, ESI). The peak current was 183 maintained for about 3 h before the current and voltage began to fluctuate. As the operating 184 voltage rose above the voltage at the MPP  $(V_{\text{MPP}})$ , the current dropped significantly to about 185 50% of its peak value due to the large gradient of the PV polarization curve in this voltage 186 region (**Fig. S12**a, ESI). Subsequent characterization strongly suggested that these increases in 187 operating voltage are caused by  $O_2$  bubble accumulation in the relatively thick carbon paper 188 layer at the anode (see **Fig. S13**, ESI, Supporting Video and Experimental Section). Specifically, 189 as the amount of trapped  $O_2$  bubbles at the anode rises, more active sites for  $O_2$  evolution are

190 blocked, effectively reducing the catalyst surface area and increasing the operating voltage (see 191 **[Fig. 2](#page-8-0)**a). When these trapped bubbles are suddenly released and swept away by the flowing 192 electrolyte, the operating voltage quickly decreases and the current recovers to its peak value 193 (**[Fig. 3](#page-11-0)**a). Overall, there is no irreversible current degradation observed during the 17 h trial at 194 5.05 suns, with a peak current density normalized by the illumination intensity of 195 12.2 mA cm-2 sun-1 at an operating voltage of 2.05 V to 2.10 V. The excellent stability is further 196 evidenced by the CVs of the PEC device before and after operation (**Fig. S12**b, ESI).



<span id="page-11-0"></span>**Fig. 3**. PEC device operation for 17 h at 5.05 suns illumination intensity without external bias. (a) Operating current and voltage. (b) STF energy conversion efficiencies.

197 The combined faradaic efficiency for CO and H<sub>2</sub> remained near 1 throughout the 17 h 198 test, with a CO/H2 product ratio ranging between 10:1 and 20:1 and an energy efficiency 199 between 51-65% (**Fig. S12**c-e, ESI and Experimental Section). The slightly increased fraction 200 of produced  $H_2$  can be explained by the higher operating voltage compared to the 1- and 3-sun 201 trials.<sup>14</sup> In particular, **Fig. S14** (ESI) shows the voltage dependence of the product ratio during 202 this 17 h test. As the voltage increases, the product ratio moves towards increased  $H_2$  generation 203 until a turning point is reached near the *V*<sub>MPP</sub> when the current starts to drop quickly. Therefore, 204 for achieving both high currents and high  $CO/H<sub>2</sub>$  product ratios, it is important to maintain a 205 low operating voltage.

206 With the amount of produced  $H_2$  and CO, the respective solar conversion efficiencies 207 can be calculated. The resulting STH efficiency is stable around 1% and the STCO efficiency 208 ranges between 5.5-15.5% with a combined STF efficiency peak at 16.4% (**[Fig. 3](#page-11-0)**b). The 209 measured STF peak value marks a 60% relative increase compared to previous reports of fully-210 integrated, wireless, PEC devices.<sup>14–17</sup> In addition, it lies just below the theoretical maximum 211 of 16.6%, as given by the PV short circuit current density measured directly after the 17 h trial 212 and assuming a faradaic efficiency for CO of 100% (see Experimental Section and **Fig. S12**a, 213 ESI).

214 Due to the high  $CO<sub>2</sub>$  flow rate used for this experiment,  $CO<sub>2</sub>$  utilization efficiencies were 215 below 1%. However,  $CO<sub>2</sub>$  utilization efficiencies above 20% were achieved by reducing the 216 CO<sub>2</sub> flow rate, with minimal reductions to the device current  $(\sim 5\%)$  (Fig. S15, ESI). Dilution 217 of the  $CO_2$  stream with N<sub>2</sub> at 10 sccm total flow rate also increased the  $CO_2$  utilization by a 218 factor of 5, with a 5%  $CO_2$  feed showing the highest  $CO_2$  utilization of 11% (**Fig. S16**, ESI).

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### 220 **Performance gains revealed through current-voltage analysis**

221 As shown above in **[Fig. 3](#page-11-0)**a, the operating current and voltage fluctuate considerably 222 during the 17 h of PEC operation. Simultaneous collection of current and voltage data during 223 steady-state operation (rather than merely logging device current over time) provides extra 224 insight into device behavior and loss contributors.<sup>38</sup> In particular, the intersection of PV and EC 225 performance curves indicates the expected operating point of the full PEC device at the 226 beginning and end of operation (**[Fig. 4](#page-16-0)**a and **Fig. S17**a, ESI). Similarly, concurrent 227 measurement of voltage and current enables live tracking of the PEC operating point during 228 unbiased operation. In this work, current logging is enabled by rerouting the electrons collected 229 at the PV front surface through a potentiostat (shunt path)<sup>27</sup> before they hit the cathodic catalyst 230 layer. The operating voltage can be measured between the PV front and back surface.

231 Any current deviation from the  $I_{\rm sc}$  results in a performance decrease of the system. 232 Similar to a previously disclosed method,<sup>38</sup> the encountered current losses may be deconvoluted 233 into EC- and PV-related sources, provided that current and voltage are logged simultaneously, 234 as described below. We note that, in contrast to our previously reported method, here the 235 operating point is matched with a fitted EC curve instead of fitting a PV curve, which would 236 require an assumption of unchanging, characteristic PV parameters such as shunt and series 237 resistance during PEC device operation. PV fitting works well if it can be assumed that the PV 238 temperature is constant and if its performance is mostly affected by partial shading, but EC 239 fitting is preferred when the PV temperature over time is unknown.

240 As described above, bubble accumulation in the anode is likely the primary driving force 241 for the elevated operating voltages. An increased amount of trapped bubbles will therefore 242 reduce the usable area for catalysis, resulting in EC-related losses. Consequently, as a first 243 deconvolution step, an EC curve is fitted through every measured operating point. For this 244 fitting process, the EC curve shown in **[Fig. 4](#page-16-0)**a is taken as a reference and then scaled by a factor 245 that represents the percentage of usable EC surface area compared to the reference case, 246 yielding EC curves intersecting with the extracted operating points. The resulting percentage of 247 usable EC area was found to range from 25-125% (**Fig. S17**b, ESI). It should be noted that the 248 reference case for the EC curve does not represent the highest possible polarization response 249 for the electrocatalyst layer, allowing for the determination of relative, active areas over 100%.

250 After fitting the operating point to an appropriate EC curve, PV current losses may be 251 determined by the difference of the measured current during operation and the corresponding 252 current on the same EC curve but at the intersection with the initial PV curve  $(I<sub>init</sub>,$  **[Fig. 4](#page-16-0)**b). EC 253 losses are then calculated by subtracting  $I_{\text{init}}$  from the current on the initial PV curve at 1.33 V, 254 which is the minimum (thermodynamic) potential required for reducing  $CO<sub>2</sub>$  to  $CO<sub>39,40</sub>$  While 255 an increase in operating voltage naturally results in higher EC losses, it also makes the PEC

256 device performance more susceptible to changes in the PV fill factor due to temperature 257 fluctuations.

258 During the 17 h of operation, the operating point fluctuates in the range bounded by the 259 PV curves measured before and after operation (**[Fig. 4](#page-16-0)**a). While it is difficult to determine the 260 temperature of the PV in the PEC cell assembly, measurements of the PV temperature in the 261 reference cell indicate a logarithmic increase over time as soon as the PV is illuminated, 262 concurrent with a drop in PV fill factor (**Fig. S18**, ESI). Since both EC and PV losses are found 263 to be essentially reversible, PV losses are most likely caused by PV heating, while EC losses 264 are the result of catalyst overpotential increases due to bubble accumulation, as described above. 265 As shown in **[Fig. 4](#page-16-0)**c, the major loss process comes from the EC component, while the PV loss 266 contribution makes up only roughly 5% of the total during most of the duration of the 267 experiment (**Fig. S17**c, ESI).

268 Interestingly, when the operating points of the PEC device during the 17 h trial are 269 compared to the PV curve measured after the trial, the majority lie above the final PV curve. 270 The origin of this behavior is the reduction of the PV operating temperature by convective heat 271 transfer to the liquid electrolyte flowing through the anolyte chamber. This increases the PV 272 efficiency in accordance with the detailed balance model developed by Shockley and Queisser<sup>41</sup> 273 (**Fig. S19**, ESI). The resulting current gains due to lower PV temperatures may be calculated 274 by the difference of the operating current and the corresponding current on the same EC curve 275 but at the intersection with the final PV curve (*I*final, **[Fig. 4](#page-16-0)**b). According to the current increases, 276 up to 5% of the achieved STF efficiency was enabled by the lower PV temperatures during 277 operation compared to the final PV temperature after the experiment (**Fig. S20**, ESI). However, 278 to capture the full effect of electrolyte cooling, the operating point of the PIM device should be 279 compared to a PV curve measured without active cooling under 5 suns illumination intensity 280 (**Fig. S21**, ESI). This comparison highlights that more than 10% of the STF efficiency was 281 enabled by cooling effects only available in a fully-integrated PEC device (**[Fig. 4](#page-16-0)**d).

282 In summary, EC losses were the dominating factor for the observed current losses. 283 Therefore, avoiding bubble accumulation in the anode chamber is the most effective way to 284 stabilize device efficiencies. For instance, the flow channels and thickness of anodic carbon 285 paper layer could be optimized to aid  $O_2$  bubble transport. Furthermore, comparing the PEC 286 operating point to a PV curve without active cooling highlights that more than 10% of the 287 achieved solar conversion efficiency was enabled by cooling of the illuminated photoabsorber 288 via electrolyte. While not described in detail here, the EC performance is also expected to 289 benefit, in some measure, from the convective heat transfer between photoabsorber and catalyst 290 surface facilitated by the electrolyte. Higher temperatures generally reduce the catalyst 291 overpotential at a constant current due to the Arrhenius-type dependencies of reaction kinetics.<sup>5</sup> 292 However, it is worth nothing that the electrolyte temperature increase due to illumination is 293 rather small for the moderate concentration levels explored by this study (**Fig. S19**, ESI). Such 294 temperature effects on reaction kinetics may become more relevant at higher levels of 295 illumination.

296 Finally, it should be noted that efficiency gains from PV cooling are most significant at 297 voltages near the MPP of the PV due to the shape of the PV polarization curve. Therefore, PV 298 cooling will be most effective when the catalyst overpotentials are relatively high. However, 299 industrially-relevant devices can be expected to be highly optimized and will likely operate 300 close to the MPP of the PV to reduce costs and maximize efficiency (otherwise the photovoltage 301 would be underutilized). Consequently, small fluctuations in catalyst overpotential will have a 302 larger effect on overall efficiency without PV cooling due to reduced PV fill factors. As 303 demonstrated in this report, beneficial heat transfer between electrolyzer and photoabsorber 304 components can stabilize device performance while operating near the MPP, by keeping the PV 305 fill factor high and reducing the catalyst overpotential.

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<span id="page-16-0"></span>**Fig. 4**. Analysis of current fluctuations. (a) Comparison of the operating point during PEC operation to the PV and EC CV curves (see **Fig. S17**a, ESI for full-scale CVs). (b) Fitting of the EC curve to intersect with an operating point which subsequently allows specific determination of losses originating in the fully-integrated PV and EC elements. (c) Deconvoluted PV and EC losses during the 17 h trial. (d) Solar conversion efficiency of the PIM device enabled by cooling via electrolyte flow when compared to a PV without active cooling.

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## 307 **Surface analysis after operation**

308 After the 17 h stability testing, we analyzed the surfaces of the catalyst-coated carbon 309 papers using scanning electron microscopy (SEM) and energy dispersive X-ray spectroscopy 310 (EDS). The Ni-coated carbon paper did not show any signs of degradation but the Au-coated 311 MPL delaminated from most of the carbon paper substrate upon device disassembly and 312 adhered to the membrane (**Fig. S22**, ESI). However, as long as electrical contact between the 313 catalyst and carbon paper substrate is maintained via mechanical compression, catalyst 314 adhesion to the membrane is not expected to result in performance degradation. Indeed, the 315 enhanced, interfacial contact between membrane and carbon paper substrate likely improves 316 device performance. This accords with previous findings that anion-exchange membrane 317 electrolyzers benefit from a catalyst coating directly on the membrane due to improved ionic 318 transport.<sup>42</sup> Finally, the range of measured  $CO/H<sub>2</sub>$  product ratios did not change much during 319 the 17 h trial (**Fig. S12**, ESI), further indicating a stable, catalytic environment.

320

## 321 **Conclusion**

322 We demonstrated 16% STF energy conversion efficiency in a fully-integrated, wireless, 323 PEC device at an elevated illumination intensity of 5 suns. No permanent performance 324 degradation could be detected during a 17 h stability test. Notably, analysis of the device 325 polarization behavior over time revealed that, on average, 10% of the measured STF efficiency 326 was enabled by convective PV cooling in this fully-integrated structure. Such benefits would 327 have been inaccessible when using the same materials but physically separating the 328 photoabsorber and electrochemically active components (PV-electrolyzer configuration).  $329$  Finally, with this compact device structure,  $CO<sub>2</sub>$  utilization efficiencies can be increased from 330 1% to 20% by simply adjusting the  $CO<sub>2</sub>$  flow rate, an important step towards commercial 331 viability.

332

### 333 **Experimental Section**

#### 334 **Deposition of catalysts**

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335 Untreated Toray TGP-H-60 carbon paper (Alfa Aesar) with a thickness of 200 μm served as 336 substrate for the anodic Ni catalyst, while a 275 um thick carbon paper coated with a MPL 337 (AvCarb GDS2230) was used as substrate for the cathodic Au catalyst. The MPL contains 338 polytetrafluoroethylene particles, introducing a strongly hydrophobic characteristic to the 339 cathodic catalyst environment. The area of the carbon paper substrates was 4 cm<sup>2</sup> shaped 340 similarly to a window frame, leaving a gap in the center to allow for illumination of the 341 photoabsorber. All catalysts were deposited by radio frequency sputtering until a thickness of 342 100 nm was reached with catalyst loadings of 0.09 mg cm−2 for Ni and 0.19 mg cm−2 for Au.

343

### 344 **Fabrication of the PIM and cell assembly**

345 The fabrication of the PIM has been detailed previously.14,27 Briefly, a square hole is cut into 346 the membrane to make room for the PV. Then, a combination of epoxies is used to secure the 347 PV in the membrane and to attach Ta metal strips which enable current-voltage analysis during 348 operation. It should be noted that the PEC cell can operate without the Ta strips, which only 349 serve analytical purposes. Since caustic KOH was used in the anode compartment, the PV back 350 was covered with chemically-resistant epoxy. For this study, we used Selemion AMV 351 membranes in the chloride form (~0.1 mm thick, AGC Engineering) and III-V triple-junction 352 PVs from Spectrolab (XTE family). The PVs show a short circuit current density of 353  $\sim$  14.07 mA cm<sup>-2</sup>, open circuit voltage of  $\sim$ 2.69 V, fill factor of  $\sim$ 0.86 and efficiency of  $\sim$ 32.38% 354 at 1 sun illumination intensity and before encapsulation. The PV front and back contacts 355 consisted of 200 μm thick Au layers and the PV front was covered by an anti-reflective coating. 356 Non-transparent epoxy covered the edges of the PV, likely reducing the illuminated area of the 357 PV to slightly less than 0.94 cm<sup>2</sup> . In fact, the short circuit current density dropped from  $358 \sim$  -14 mA cm<sup>-2</sup> to 12.62 mA cm<sup>-2</sup> after epoxy encapsulation. This drop in current density may be 359 lowered by optimizing the amount of used epoxy or by deploying larger PVs where the area 360 near the edges represents a smaller fraction of the total area. Despite some expected area losses, 361 an illuminated area of 0.94 cm<sup>2</sup>was considered for all efficiency calculations.

362 All cells were assembled by stacking the following layers from bottom to top: acrylic 363 endplate with straight flow channels for the anolyte, carbon paper coated with the anodic 364 catalyst and surrounded by a silicone gasket, PIM, Au-coated carbon paper surrounded by a 365 silicone gasket, acrylic endplate with serpentine gas flow channels. For the PEC cell, three extra 366 layers of uncoated carbon paper were placed in between the acrylic endplate and the Ni-coated 367 carbon paper to make enough space for the epoxy-coated PV. Alternatively, a slot can be 368 machined into the endplates. However, if the PV does not sit perfectly in that slot, nonuniform 369 compression can easily damage the protective coating at the edges of the PV and promote 370 pinhole formation. Therefore, we chose to create this slot with compressible carbon paper which 371 allows for more flexibility during cell assembly. This also required a slightly higher torque 372 (0.18 N m) when tightening the screws of the PEC cell compared to the EC cell (0.11 N m) to 373 prevent leaks. However, the thick layer combined with the higher torque can hinder  $O_2$  bubbles 374 from leaving the anode catalyst surface and cause bubble accumulation.

375

### 376 **Calibration of concentrated light intensity**

377 A Newport Oriel Sol3A solar simulator equipped with a Xe lamp and AM1.5G filter was used 378 for all illuminated experiments. To calibrate the light intensity under concentrated sunlight a 379 Fresnel lens (2" diameter, Thorlabs FRP251) was first positioned at 1 sun illumination intensity 380 using a monocrystalline silicon reference solar cell (Newport 91150V). The increased light 381 intensity behind the Fresnel lens was then measured with a home-made reference cell, which 382 mimics the composition, size and position of the photoabsorber in the PEC cell. The calibration 383 of the home-made reference cell yielded a short circuit current density  $(j_{\rm sc})$  of 12.62 mA cm<sup>-2</sup> 384 at 1 sun and 25 °C with a temperature dependence of 0.009 mA cm<sup>-2</sup> °C<sup>-1</sup>. The measured  $j_{sc}$  as 385 a function of the distance between the reference cell and Fresnel lens can then be correlated to

386 the light concentration factor, assuming direct proportion and correcting for the temperature (*T*)

387 as the reference cell heats up:

388 
$$
c = \frac{j_{\text{sc}} - 0.009 \text{ mA cm}^{-2} \text{ °C}^{-1} \times (T - 25 \text{ °C})}{12.62 \text{ mA cm}^{-2}} \text{ H}(4)
$$

389

#### 390 **Measurement conditions**

391 If not otherwise mentioned, the following experimental conditions were used. A peristaltic 392 pump pushed 2м KOH at a flow rate of  $~6$  mL min<sup>-1</sup> through the anode chamber of the PEC 393 cell. KOH electrolytes were prepared using ACS reagent grade pellets from Sigma-Aldrich. For 394 all light-driven experiments, a KOH reservoir volume of ~200 mL was chosen to prevent 395 acidification of the anolyte via  $CO<sub>2</sub>$  crossover during long-term experiments.  $CO<sub>2</sub>$  was first 396 flowed at 60 sccm through a bubble humidifier filled with Milli-Q water (resistivity > 18.2 M $\Omega$ 397 cm) kept at room temperature ( $\sim$ 25 °C) and then into the cathode chamber. CVs were measured 398 in a two-electrode configuration at scan rates of 10 mV  $s^{-1}$  for EC and PEC tests and 200 mV 399 s-1 for PV tests. Dark EC stability measurements were carried out at a constant full cell potential 400 of 2 V, while illuminated PEC tests were not biased by a potentiostat (0 V).

401

### 402 **Product analysis**

403 The cathode outlet was fed directly into a gas chromatograph from SRI Instruments (Multiple 404 Gas Analyzer #5, 8610C) equipped with a thermal conductivity detector and flame ionization 405 detector (FID). Prior to the FID, all products went through a methanizer to increase the 406 minimum detection limit of the carbon-based products. The highest CO concentration available 407 for calibration of the FID was 7980 ppm. If  $CO<sub>2</sub>$  utilization efficiencies surpass 1%, the CO 408 concentration in the product stream will likely exceed the calibration limits. Therefore, the 409 cathode outlet needs to be diluted with an inert gas such as argon to stay within the limits and



420 calculated by dividing the produced  $H_2$  through the illumination power density at 1 sun 421 (0.1 W cm<sup>-2</sup>), the size of the PV (0.94 cm<sup>2</sup>) and the concentration factor:<sup>6</sup>

422

423 
$$
\text{STH} = \frac{1.23 \text{ V} \times I_{\text{PV}} \times \text{faradaic efficiency (H}_2)}{0.1 \text{ W cm}^{-2} \times A_{\text{PV}} \times c} \text{#(5)}
$$

424

425 Similarly, the STCO efficiency is determined using the minimum cell potential of 1.33 V

426 required for the cathodic evolution of CO:

427

428 
$$
STCO = \frac{1.33 \text{ V} \times I_{\text{PV}} \times \text{faradaic efficiency (CO)}}{0.1 \text{ W cm}^{-2} \times A_{\text{PV}} \times c} \text{#(6)}
$$

429

430 Since no other products were observed (cumulative faradaic efficiency of measured products is

432 Further, the energy/energetic efficiency (EE) towards  $H<sub>2</sub>$  and CO products is calculated via 433 dividing the amount of energy used to form the aforementioned products by the net electrical 434 energy supplied to the system:<sup>2</sup>

435 
$$
EE_{H_2} = \frac{1.23 \text{ V} \times \text{faradaic efficiency (H}_2)}{\text{full cell voltage}}, \#(7)
$$

436 
$$
EE_{CO} = \frac{1.33 \text{ V} \times \text{faradaic efficiency (CO)}}{\text{full cell voltage}} \cdot \text{#(8)}
$$

437

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446

# 447 **Conflict of Interest**

448 The authors declare no competing interests.

449

# 450 **References**

- 451 1 R. U. Ayres and E. H. Ayres, *Crossing the Energy Divide. Moving from Fossil Fuel*
- 452 *Dependence to a Clean-Energy Future*, Prentice Hall, Upper Saddle River, NJ, 2010.
- 453 2 R. Küngas, *J. Electrochem. Soc.*, 2020, **167**, 44508.
- 454 3 J. Zhang, W. Cai, F. X. Hu, H. Yang and B. Liu, *Chem. Sci.*, 2021, **12**, 6800–6819.
- 455 4 S. Garg, M. Li, A. Z. Weber, L. Ge, L. Li, V. Rudolph, G. Wang and T. E. Rufford, *J.*  456 *Mater. Chem. A*, 2020, **8**, 1511–1544.
- 457 5 S. Tembhurne, F. Nandjou and S. Haussener, *Nat. Energy*, 2019, **4**, 399–407.
- 458 6 W.-H. Cheng, M. H. Richter, I. Sullivan, D. M. Larson, C. Xiang, B. S. Brunschwig and 459 H. A. Atwater, *ACS Energy Lett.*, 2020, **5**, 470–476.
- 460 7 M. Asadi, M. H. Motevaselian, A. Moradzadeh, L. Majidi, M. Esmaeilirad, T. V. Sun, C. 461 Liu, R. Bose, P. Abbasi, P. Zapol, A. P. Khodadoust, L. A. Curtiss, N. R. Aluru and A. 462 Salehi-Khojin, *Adv. Energy Mater.*, 2019, **9**, 1803536.
- 463 8 M. Schreier, F. Héroguel, L. Steier, S. Ahmad, J. S. Luterbacher, M. T. Mayer, J. Luo and 464 M. Grätzel, *Nat. Energy*, 2017, **2**, 17087.
- 465 9 M. Schreier, L. Curvat, F. Giordano, L. Steier, A. Abate, S. M. Zakeeruddin, J. Luo, M. T. 466 Mayer and M. Grätzel, *Nat. Commun.*, 2015, **6**, 7326.
- 467 10 G. Gurudayal, J. Bullock, D. F. Srankó, C. M. Towle, Y. Lum, M. Hettick, M. C. Scott, A. 468 Javey and J. Ager, *Energy Environ. Sci.*, 2017, **10**, 2222–2230.
- 469 11 F. Urbain, P. Tang, N. M. Carretero, T. Andreu, L. G. Gerling, C. Voz, J. Arbiol and J. R. 470 Morante, *Energy Environ. Sci.*, 2017, **10**, 2256–2266.
- 471 12 Y. Xiao, Y. Qian, A. Chen, T. Qin, F. Zhang, H. Tang, Z. Qiu and B.-L. Lin, *J. Mater.*  472 *Chem. A*, 2020, **8**, 18310–18317.
- 473 13 B. Kim, H. Seong, J. T. Song, K. Kwak, H. Song, Y. C. Tan, G. Park, D. Lee and J. Oh, 474 *ACS Energy Lett.*, 2020, **5**, 749–757.
- 475 14 T. A. Kistler, M. Y. Um, J. K. Cooper, I. D. Sharp and P. Agbo, *Adv. Energy Mater.*, 476 2021, **11**, 2100070.
- 477 15 V. Andrei, B. Reuillard and E. Reisner, *Nat. Mater.*, 2020, **19**, 189–194.
- 478 16 T. Arai, S. Sato and T. Morikawa, *Energy Environ. Sci.*, 2015, **8**, 1998–2002.
- 479 17 M. Asadi, K. Kim, C. Liu, A. V. Addepalli, P. Abbasi, P. Yasaei, P. Phillips, A. 480 Behranginia, J. M. Cerrato, R. Haasch, P. Zapol, B. Kumar, R. F. Klie, J. Abiade, L. A. 481 Curtiss and A. Salehi-Khojin, *Science*, 2016, **353**, 467–470.
- 482 18 C. E. Creissen and M. Fontecave, *Adv. Energy Mater.*, 2020, **11**, 2002652.
- 483 19 L.-C. Weng, A. T. Bell and A. Z. Weber, *Energy Environ. Sci.*, 2019, **12**, 1950–1968.
- 484 20 C. M. Gabardo, C. P. O'Brien, J. P. Edwards, C. McCallum, Y. Xu, C.-T. Dinh, J. Li, E. 485 H. Sargent and D. Sinton, *Joule*, 2019, **3**, 2777–2791.
- 486 21 J. F. Geisz, R. M. France, K. L. Schulte, M. A. Steiner, A. G. Norman, H. L. Guthrey, M. 487 R. Young, T. Song and T. Moriarty, *Nat. Energy*, 2020, **32**, 326–335.
- 488 22 A. Royne, C. J. Dey and D. R. Mills, *Sol. Energy Mater. Sol. Cells*, 2005, **86**, 451–483.
- 489 23 A. Aldossary, S. Mahmoud and R. AL-Dadah, *Appl. Therm. Eng.*, 2016, **100**, 490–500.
- 490 24 G. Siefer and A. W. Bett, *Prog. Photovolt: Res. Appl.*, 2014, **22**, 515–524.
- 491 25 P. Singh and N. M. Ravindra, *Sol. Energy Mater. Sol. Cells*, 2012, **101**, 36–45.
- 492 26 A. Luque, *Solar cells and optics for photovoltaic concentration*, Adam Hilger, Bristol, 493 1989.
- 494 27 T. A. Kistler, N. Danilovic and P. Agbo, *J. Electrochem. Soc.*, 2019, **166**, H656.
- 495 28 L.-C. Weng, A. T. Bell and A. Z. Weber, *Phys. Chem. Chem. Phys.*, 2018, **20**, 16973– 496 16984.
- 497 29 J. S. DuChene, G. Tagliabue, A. J. Welch, X. Li, W.-H. Cheng and H. A. Atwater, *Nano*  498 *Lett.*, 2020, **20**, 2348–2358.
- 499 30 D. M. Koshy, S. Chen, D. U. Lee, M. B. Stevens, A. M. Abdellah, S. M. Dull, G. Chen, D. 500 Nordlund, A. Gallo, C. Hahn, D. C. Higgins, Z. Bao and T. F. Jaramillo, *Angew. Chem.*, 501 2020, **132**, 4072–4079.
- 502 31 A. Goyal, G. Marcandalli, V. A. Mints and M. T. M. Koper, *J. Am. Chem. Soc.*, 2020, **142**, 503 4154–4161.
- 504 32 M. Tahir, L. Pan, F. Idrees, X. Zhang, L. Wang, J.-J. Zou and Z. L. Wang, *Nano Energy*, 505 2017, **37**, 136–157.
- 506 33 N.-T. Suen, S.-F. Hung, Q. Quan, N. Zhang, Y.-J. Xu and H. M. Chen, *Chem. Soc. Rev.*, 507 2017, **46**, 337–365.
- 508 34 C. C. L. McCrory, S. Jung, I. M. Ferrer, S. M. Chatman, J. C. Peters and T. F. Jaramillo, *J.*  509 *Am. Chem. Soc.*, 2015, **137**, 4347–4357.
- 510 35 P. Connor, J. Schuch, B. Kaiser and W. Jaegermann, *Z. Phys. Chem.*, 2020, **234**, 979–994.
- 511 36 S. Trasatti and O. A. Petrii, *Pure Appl. Chem.*, 1991, **63**, 711–734.
- 512 37 M. C. O. Monteiro, F. Dattila, B. Hagedoorn, R. García-Muelas, N. López and M. T. M. 513 Koper, *Nat. Catal.*, 2021.
- 514 38 T. A. Kistler and P. Agbo, *APL Mater.*, 2020, **8**, 31107.
- 515 39 S. Nitopi, E. Bertheussen, S. B. Scott, X. Liu, A. K. Engstfeld, S. Horch, B. Seger, I. E. L. 516 Stephens, K. Chan, C. Hahn, J. K. Nørskov, T. F. Jaramillo and I. Chorkendorff, *Chem.*  517 *Rev.*, 2019, **119**, 7610–7672.
- 518 40 K. P. Kuhl, E. R. Cave, D. N. Abram and T. F. Jaramillo, *Energy Environ. Sci.*, 2012, **5**, 519 7050.
- 520 41 W. Shockley and H. J. Queisser, *J. Appl. Phys.*, 1961, **32**, 510–519.
- 521 42 G. Lindquist, S. Oener, Q. Xu, A. R. Motz, A. Keane, C. Capuano, K. E. Ayers and S. W. 522 Boettcher, *Meet. Abstr.*, 2020, **MA2020-02**, 2446.
- 523

#### **Broader Context**

Current trends in global mean temperatures indicate that mere reductions in carbon dioxide emissions will not be enough to combat global warming. Instead, it will eventually become necessary to deploy carbon-negative and carbon-neutral technologies such as electrochemical CO<sup>2</sup> reduction driven by renewable energy sources like solar. Respective devices generally consist of two main components: the first one converts solar energy into electrical energy which is then converted into chemical energy (such as fuels) by the second. The second component is usually made more efficient by employing catalysts. However, these catalytic materials generally deactivate over time, resulting in performance drops of electrochemical and photoelectrochemical (PEC) devices. In this report, we highlight how the performance of fully-integrated, PEC devices can be stabilized via thermal management relative to devices consisting of physically-separated solar conversion and fuel production components. We demonstrate this using a cell that employs beneficial design elements and materials developed for dark electrochemistry, resulting in the highest efficiency for fully-integrated PEC CO<sub>2</sub> reduction devices reported to date.