



Long-range mechanical signaling in biological systems

| | |
|-------------------------------|---|
| Journal: | <i>Soft Matter</i> |
| Manuscript ID | SM-REV-08-2020-001442.R1 |
| Article Type: | Review Article |
| Date Submitted by the Author: | 11-Oct-2020 |
| Complete List of Authors: | Alisafaei, Farid; University of Pennsylvania, Center for Engineering Mechanobiology, Department of Materials Science and Engineering Chen , Xingyu; University of Pennsylvania, 1Center for Engineering Mechanobiology, Department of Materials Science and Engineering Leahy, Thomas; University of Pennsylvania, Department of Bioengineering, McKay Orthopaedic Research Laboratory Janmey, Paul; University of Pennsylvania, Department of Bioengineering and Institute for Medicine and Engineering Shenoy, Vivek; University of Pennsylvania, Department of Materials Science and Engineering |
| | |

1 Long-range mechanical signaling in biological systems

2
3 Farid Alisafaei^{1,2}, Xingyu Chen^{1,2}, Thomas Leahy^{1,3,4}, Paul A. Janmey^{1,5,6}, Vivek B. Shenoy^{1,2*}

4
5 ¹Center for Engineering Mechanobiology, University of Pennsylvania, Philadelphia, PA 19104

6 ²Department of Materials Science and Engineering, School of Engineering and Applied Science,
7 University of Pennsylvania, Philadelphia, PA 19104

8 ³Department of Bioengineering, School of Engineering and Applied Science, University of
9 Pennsylvania, Philadelphia, PA 19104

10 ⁴McKay Orthopaedic Research Laboratory, University of Pennsylvania, Philadelphia, PA 19104

11 ⁵Institute for Medicine and Engineering, University of Pennsylvania, 3340 Smith Walk,
12 Philadelphia, PA 19104

13 ⁶Departments of Physiology, and Physics & Astronomy, University of Pennsylvania, Philadelphia,
14 PA 19104

15 16 **Abstract**

17 Cells can respond to signals generated by other cells that are remarkably far away. Studies
18 from at least the 1920's showed that cells move toward each other when the distance between
19 them is on the order of a millimeter, which is many times the cell diameter. Chemical signals
20 generated by molecules diffusing from the cell surface would move too slowly and dissipate too
21 fast to account for these effects, suggesting that they might be physical rather than biochemical.
22 The non-linear elastic responses of sparsely connected networks of stiff or semiflexible filament
23 such as those that form the extracellular matrix (ECM) and the cytoskeleton have unusual
24 properties that suggest multiple mechanisms for long-range signaling in biological tissues.
25 These include not only direct force transmission, but also highly non-uniform local deformations,
26 and force-generated changes in fiber alignment and density. Defining how fibrous networks
27 respond to cell-generated forces can help design new methods to characterize abnormal tissues
28 and can guide development of improved biomimetic materials.

29
30
31

32 **1. Introduction**

33 The idea that cells can signal to other cells at a distance and that the basics of this signal might
34 be mechanical rather than chemical can be traced back a century ¹. This article provides some
35 examples in which long-range force transmission is an important factor in tissue morphogenesis
36 and other biological processes. In contrast to the strain fields in simple elastic continuum
37 materials such as those formed by flexible polymers, where the strain magnitude decays rapidly
38 from the point of force following a power law, the force transmission in biological materials relies
39 on the presence of fibrous networks with large mesh sizes and stiff filaments. The physical
40 properties of these dilute networks include shear strain-stiffening ^{2, 3}, alignment in the stress
41 direction ^{4, 5}, non-affine deformations ^{6, 7}, and anomalous, strain-dependent Poisson's ratios ⁸,
42 each of which can contribute to force transmission. These effects are considered in a summary
43 of the key theoretical models that can account for long-range force transmission in networks
44 formed by semiflexible or stiff biopolymers.

45
46

47 **2. Background**

48 **2.1. Experimental evidence for long-range force transmission.** Studies published in the
49 1920s showed that when nerves were severed and then placed into cell culture media of
50 various kinds, the cells emerged from the damaged nerve and spread or grew in a random
51 radial fashion if the nerve end was placed in liquid or if a single nerve was placed in a dilute
52 blood clot. However, if two nerve ends were placed near each other in a blood clot, the cells at
53 first emerged randomly, but then rapidly moved toward each other to make a line of new tissue
54 connecting the two previously separated nerve ends. Even earlier there was evidence that the
55 growth of neural tissue was influenced by a stimulatory fibrillation⁹ and various studies at that
56 time tested the hypotheses that the signals leading to spatial guidance of nerve cells were
57 primarily chemical, electrical, or mechanical (reviewed in¹⁰). The possibility of mechanical
58 guidance was not limited to neural cells, and these early studies showed that two triangular
59 islands of fibroblasts, placed mm away from each other within blood plasma clots acted as
60 “suction pumps” (“saugenpumpen”) to draw cells from each island to the other¹.

61 Later studies showed that the traction stresses exerted by different cell types in collagen gels
62 varied over a large range and that, perhaps paradoxically, the fastest moving cells, such as
63 neutrophils or neuronal growth cones, exerted the least force, whereas fibroblasts generated
64 much more force than was required for them to locomote. As a result, explants of fibroblasts
65 distant from each other could reorganize and align collagen fibers between them over a
66 distance of a cm^{11, 12}.

67 An example of the pattern formed by cells, largely fibroblasts, emerging from two severed
68 nerves placed in a blood clot is shown in Figure 1. Although the magnification of this image is
69 not given in the original report, the diameter of a typical adult rat nerve is approximately 0.5 mm,
70¹³ so the distance between the two cut nerves is more than 1 mm. Immediately between the
71 nerve ends, the cells grew toward each other; in other positions where the side of one nerve
72 end faced away from the other, the growth was random. This pattern of growth was described
73 as being due to an “attraction field” emanating from the cluster of cells at the nerve ends
74 growing into the matrix. The nature of this attraction has been the subject of much debate^{14, 15}.
75 A related quantitative study placed pairs of small embryonic chick heart pieces, consisting
76 mainly of fibroblasts, at different distances to each other within a mixture of embryonic fluid and
77 a fibrin gel formed from chicken blood plasma. This study showed that the fibroblasts placed
78 tension on the fibrin strands within the clot, and that as the tissue pieces grew, the cells
79 preferentially moved to the space between adjacent tissue pieces and aligned the fibers in
80 between¹⁴. Calculating the probability that cells from adjacent tissue pieces made oriented
81 bridges between them led to a measure of the attraction field incidence, I , as a function of the
82 initial distance, d , between tissues pieces within the clot. Remarkably, I depended inversely on
83 d^2 , and approached zero only at d between 3.5 and 4 mm. This distance is far too large to
84 support spatial gradients of chemical signals that might be generated between the cell clusters.
85 The large length scale and the power-law decay suggested that the signal might be physical.
86 Whether this signal is the force that the cells exert on the matrix and transmit to the distant cell
87 or spatial patterning of the matrix as cells pull on the fibrin or collagen fibers in the extracellular
88 matrix (ECM) is not obvious, since cells can respond to both forces at the membrane and to the
89 topography and the stiffness of the fibers in their substrate.

90
91 Measurements of individual cells on the surface of thin collagen gels have revealed more clearly
92 the distances over which a cell can sense mechanical signals and how the contractile energy of
93 the cell, as well its ability to chemically modify the matrix, reorient the fiber network structure¹⁶.
94 Figure 2A shows the morphology of a single fibroblast, of average diameter approximately 50
95 μm , placed on collagen gels contained within rigid square frames of length 200 μm , 500 μm , or
96 1700 μm ¹⁷. The cells within the 200 μm x 200 μm frame extend multiple processes toward all

97 sides of the frame. The number of extensions decreases when the frame length is 500 μm and
98 is close to 2 when the frame length is 1700 μm , similar to the shape of the cell in an infinitely
99 large gel. These results are consistent with the hypothesis that the cell extends protrusions
100 toward a rigid boundary that is near enough to the force it develops on the network so that the
101 strain field propagates to the rigid boundary, and therefore the cell feels more resistance in that
102 direction and moves toward it. If the boundary is more than $\sim 800 \mu\text{m}$ away, the cell no longer
103 feels resistance from the boundary, and the number of branches decreases, leading to a bipolar
104 cell. During the hours that the cell accommodates to its substrate, it is also remodeling it.
105 Figure 2B shows how the collagen gel surrounding the contractile cell is reorganized. The
106 collagen fibers tend to concentrate near the cell edges and to align in parallel with the cell
107 extensions ¹⁷.

108
109 Cells are capable of altering their surrounding mechanical environment, which can alter the
110 perceived mechanical force transduction of surrounding cells. Specifically, the local stiffness
111 near a contractile cell in a collagen or fibrin gel can be higher than the average stiffness far
112 away from the cell ^{16, 18, 19}. Since many cell types respond to substrate stiffness ²⁰, often by
113 moving to areas of increased stiffness (see section 3.2), these changes in surrounding matrix
114 mechanical properties due to local stiffening may directly alter nearby cell behavior. A cell's
115 ability to sense long-range forces from other cells is also modulated by its environment. For
116 example, if a cell in a fiber network can feel a rigid boundary, then it is likely also to respond to
117 another cell pulling within the same matrix. When mesenchymal stem cells were sparsely
118 cultured on fibrin gels, they generated strain fields larger than 5 times the cell diameter, similar
119 to the field generated by fibroblasts in collagen gels, and they oriented their long axes toward
120 each other if they were less than 400 μm away. On the surface of the gel, they formed ribbon-
121 like aggregates, whereas on rigid substrates they aggregated randomly ¹⁸.

122
123 **2.2. Models for long-range force transmission.** Multiple mechanisms can explain the
124 apparent traction field around cells in a fibrous matrix. The simplest might be that a single fiber
125 connects two cells, and as one cell pulls on the fiber, the adjacent cell immediately feels the
126 force when the fiber is pulled taught. This is unlikely to be the case in biomimetic systems,
127 because the mesh size of collagen and fibrin gels at physiologically realistic concentrations is
128 less than one micron, and fibers long enough to directly connect two distant cells have not been
129 identified, and if they existed would be part of a 3D network, rather than free long filaments. If
130 the cell responds to a force, then that force is propagated through a series of fibers and
131 crosslinks that form a force chain long enough to span between cells. This mechanism is well
132 supported theoretically ²¹, and predicts that long-range strain fields are possible only in stiff
133 polymer networks and not in hydrogels formed by flexible polymers.

134
135 Alternatively, the cell responds to the alignment in the fiber networks caused by the neighboring
136 contractile cell. The reorganization has two spatial aspects. The fiber density increases when
137 the fibers align, thereby providing a higher concentration of adhesive sites for cell receptors,
138 and the directionality of the aligned fiber bundles provides a spatial cue for the adhesive steps
139 during cell motility ²². An additional mechanism involves the nonlinear elasticity of fibrous
140 networks. Unlike linear elastomers, for which the elastic modulus is independent of strain,
141 networks of semiflexible and rigid biopolymers stiffen with increasing shear strain ^{3, 23}, as
142 caused by the contractile cell ²⁴. Whether long-range mechanical signaling results from strain-
143 stiffening *per se* ²⁵ or requires the long fibers typically present in strain-stiffening materials ²⁶ is
144 still unresolved and might depend on the specific system.

145
146 One recent study shows that, despite the doubts raised by the originator of the attraction field
147 hypothesis ¹⁵ it is in some cases the force itself to which a cell responds to initiate its movement

148 toward a point of local force generation. Pakshir et al ²⁷ studied how macrophages respond
149 when a contractile fibroblast deforms a collagen matrix on which both cell types are placed.
150 They found that macrophages migrated persistently toward the contractile cell even when they
151 were hundreds of microns, or many cell diameters away. Initial studies placed a single
152 myofibroblast in the middle of a mm scale collagen matrix and monitored how macrophages that
153 were initially distributed throughout the matrix moved. When macrophages were within 600
154 microns they moved persistently toward the contracting cell. This study alone does not
155 unambiguously imply reaction to a force, because chemical gradients and fiber alignments are
156 still possible attracting stimuli. However, if the matrix was aligned by the cell and then
157 chemically fixed before the macrophages were deposited, they no longer moved persistently
158 toward the cell, even in the presence of some fiber alignment. Even more strikingly, the
159 contractile fibroblast in the center of the matrix could be replaced by a microneedle that applied
160 directional forces of the same magnitude as the myofibroblast. This force was sufficient to
161 create strain fields that extended hundreds of microns away from the point of force, and
162 macrophages within this strain field moved persistently toward the force, as seen in Figure 3. It
163 was proposed that macrophages mechanosense the velocity of matrix local displacement as
164 supported by the following evidence. (i) Fibrous matrices enable long-range transmission of
165 tensile forces generated by contractile fibroblasts, which in turn triggers migration of
166 macrophages over distances 20-40 times larger than their diameters. (ii) Static mechanical
167 cues, such as pre-aligned collagen or collagen condensation are neither required nor sufficient
168 to trigger the migration of macrophages. (iii) Dynamic changes in the deformation of the
169 collagen matrix are required to attract migratory macrophages above a critical matrix strain
170 velocity.

171

172

173 **3. Long-range force transmission in biological materials: tissues,** 174 **cells, and artificial matrices**

175 In this section, we review long-range force transmission in the contexts of various physiological
176 tissues, in cells, as well as in artificial matrices and biomaterials.

177

178 **3.1. Tissues**

179 Within biological tissues, long-range force transmission becomes necessary for physiological
180 processes early in development. A well-conserved example is the mechanical stimulation that is
181 necessary for generating epithelial tubule branching structures, such as in the case of the
182 mammalian lungs, intestines, or kidney ^{28, 29}. For example, branching behavior of the developing
183 lung epithelium is synchronized between distant parts of the lung ²⁹. This process is carefully
184 coordinated by contractions of the developing smooth muscle surrounding the airway epithelium
185 and the resulting fluctuations in transmural pressure within the epithelial tubules. This leads to
186 regulated pressures experienced by the airway epithelium that regulate the synchronized
187 branching morphogenesis ^{30, 31}. Similar sorts of patterning are possible in generating other
188 epithelial patterns. For instance, *in vitro* studies have demonstrated that epithelial cells maintain
189 and contract type I collagen within the ECM to successfully transmit forces between cells up to
190 600 μm away to generate and maintain a tubule-like patterning ³². A similar dependence on Col-
191 I fiber orientation is shown in branching morphogenesis in mammary gland maturation, as
192 epithelial cells migrate along axially oriented collagen fibers in the stromal fat pad. *In vitro*
193 experiments further suggest that this epithelial cell–type I collagen fiber relationship is both
194 causal, as aligned Col-I fibers are necessary to direct epithelial cell orientation, and
195 interdependent, as the epithelial cells are also capable of axially aligning the fibers of their
196 substrate via RhoA/ROCK-mediated contractions ³³. Following development, these matrix-

197 aligning forces must then be carefully regulated for epithelial patterning to be maintained, as
198 uncontrolled epithelial cell contractility can lead to tumor initiation and progression ³⁴.

199
200 In addition to playing a role in developing tissue structures, long-range force transmission can
201 be involved in normal tissue function and homeostasis. This is perhaps best exemplified in
202 musculoskeletal tissues, where mechanical loads are transmitted to allow for locomotion of the
203 body. The cells within these tissues experience these loads as well, as mechanical strain is
204 transmitted to the resident fibroblasts and fibroblast nuclei ^{35, 36}. However, tendons also exhibit
205 the ability to transmit forces from the cell to the macroscale tendon ECM as unloaded tendons
206 are able to contract the macroscale tendon ECM to restore tension ^{37, 38}. The specific ECM
207 components and organization in addition to cell types within different musculoskeletal tissues
208 result in tissue-specific macro- to micro-scale strain transfer ¹⁶. Force transmission within
209 musculoskeletal tissues is disrupted by tissue injury, either through overloading or a puncture
210 injury ^{35, 36}. Alterations in force transmission alone can lead to disease progression in these
211 tissues. For example, increasing collagen crosslinks within the cartilage extracellular matrix via
212 lxyloxidase overexpression can directly lead to osteoarthritis progression at a similar scale and
213 rate to surgically-induced osteoarthritis progression ³⁹.

214
215 Long-range mechanical force transmission plays a role in the progression of various diseases,
216 such as cancer ⁴⁰. For example, cancer cells are capable of generating sufficiently high force to
217 align the nearby ECM fibrils, which promotes cell migration and diffusion of cancer growth
218 factors away from the tumor microenvironment (Figure 4) ⁴¹. This effect was validated by
219 growing cancer cell spheroids on collagen gels to observe the mechanical effect the spheroids
220 had on the surrounding ECM and fibroblasts and by investigating how matrix alignment alters
221 diffusion, as shown in Figure 4. Also, the rearrangement of ECM fibers further increases cancer
222 cell stiffness and, therefore, the traction forces that the cell puts on the surrounding ECM,
223 creating a positive feedback loop ⁴².

224
225 While the importance of long-range force transmission within tissues is becoming more
226 appreciated, continued understanding of how long-range force transmission guide tissue
227 development, homeostasis, and disease progression is necessary for the development of future
228 beneficial therapies and tissue engineering solutions that recapitulate normal tissue mechanical
229 behavior.

230

231 **3.2. Cells**

232 Assessing long-range force transmission to cells is important for understanding how cells within
233 tissues interpret their mechanical environment and use it to regulate their behavior. Cells
234 transduce mechanical force from their surroundings via integrins, cytoskeleton filaments, and
235 cytoskeletal-nucleus mechanical tethers, such as the LINC complex ^{43, 44}. A cell's interpretation
236 of its mechanical surrounding is not a passive process. Rather, the cells are constantly probing
237 their surrounding ECM by pulling it with actomyosin fibers anchored via focal adhesions to the
238 matrix ⁴⁵. Moreover, cells maintain a significant amount of prestress within themselves in order
239 to prime themselves for understanding their mechanical environment ⁴⁵.

240

241 Cell interpretation of their mechanical environment is necessary for guiding and regulating cell
242 behavior. For instance, the mechanical properties of the environment alone can lead to altered
243 differentiation states in stem cells ⁴⁶. This regulatory role occurs most directly because varying
244 ECM stiffnesses and applied mechanical forces are transmitted to the nucleus resulting in
245 shape changes that alter gene transcription ⁴⁷. In addition to matrix stiffness alone, anisotropy of
246 the substrate also directs cell phenotype and stem cell fate towards an anisotropic (i.e., fibrillar
247 collagen-producing) lineage ⁴⁸. The ECM mechanical environment regulates how the cells

248 interact with their substrate by increasing focal adhesion and stress fiber density on stiffer
249 substrates^{49,50}. Beyond focal adhesion and stress fiber density and organization, there is a lack
250 of understanding of the mechanisms by which cells interpret mechanical cues from the ECM.
251 However, it has been hypothesized that substrate stiffness is estimated by cells probing
252 deformation fields in the surrounding fibrous ECM, whereby fiber buckling would lead to
253 decreased interpreted compressive stiffness⁵¹. This fiber buckling amplifies cell contraction and
254 increases their mechanosensitivity⁵².

255
256 While the mechanism of cell transduction of long-range forces is not fully understood, it is
257 known that it plays a role in cell processes through direct involvement in the process or in a
258 regulatory role. One such example of a cell process is cell migration, where cells apply forces to
259 their substrate in order to move themselves along. Specifically, long-range tensile forces are
260 necessary to coordinate collective cell migration, as tensile forces at the front of invasive cell
261 cohorts displace and align the ECM in order to create tracks along which the cells can migrate
262⁵³. Long-range forces can also be transmitted intracellularly to drive collective cell migration
263 during development, as forces at the rear of a neural crest cell group work to push the cell
264 collective forward (Figure 5)⁵⁴. In addition to coordinating cell migration patterns, force
265 transmission directly regulates this process. Durotaxis is the migration of cells as guided by
266 rigidity gradients, whereby cells generally migrate in the direction of greater matrix stiffness in a
267 cell type-specific manner⁵⁵. In addition to relatively static rigidity gradients, cells can also be
268 guided along migratory paths by application of mechanical strain, which elicits a non-monotonic
269 migration response in the direction of applied strain⁵⁶. Thus far, durotaxis is less understood
270 than other methods of guided cell migration such as chemotaxis. Continued investigation of
271 durotaxis is essential for basic science understanding of cell behaviors but also has direct
272 clinical relevance, as migration in response to mechanical stiffness gradients play a large role in
273 cancer cell migration/metastasis as described previously^{41,42}. Specifically, cancer cells exhibit
274 increased durotactic migratory potential on softer substrates, possibly reminiscent of the
275 increased migratory capability of cancer cells as they metastasize away from the primary tumor
276⁵⁷.

277
278 Another example of long-range forces playing a role in cell behavior is in distant cell
279 communication, as cells are capable of communicating via mechanical signals transmitted
280 through the extracellular matrix⁵⁸. Specifically, the nonlinear elastic nature of fibrous matrices
281 has been demonstrated to be a necessary ECM component for this communication to take
282 place²¹. One example of such communication is exemplified in the macrophage-fibroblast
283 relationship, as fibroblast signal through force perturbations in the ECM to the local resident
284 macrophages. Interestingly, application of forces to the ECM is sufficient to initiate macrophage
285 migration in the direction of these forces, as discussed previously²⁷. It is also worth noting that
286 long-range force transmission is necessary to elicit the assembly of multicellular structures and
287 patterns^{32,59}. Long-range force transmission can also affect intercellular biochemical
288 communication. Specifically, it has been shown that long-range forces are capable of altering
289 the physical structure of the ECM to increase rates of diffusion and, therefore, enhance cell-cell
290 biomechanical communication^{60,61}.

291 292 **3.3. Artificial Matrices and Biomaterials**

293 After addressing long-range force transmission within cells and tissues, it is necessary to
294 acknowledge how these concepts are translated to artificial matrices and biomaterials. Artificial
295 matrices include materials that are largely or entirely synthetic, such as self-assembling block
296 copolymer networks⁶², with biomaterials being engineered materials made primarily from
297 biological macromolecules such as fibrin, collagen, or glycosaminoglycans. Matrix stiffness and
298 organization can be carefully modulated to observe the effects of these parameters on force

299 transmission across matrices via fiber buckling and tensioning ⁶³. The stiffness of the individual
300 fibers can also be tuned, whereby fibers of lower stiffness are more easily recruited by cellular
301 traction forces, which promotes focal adhesion formation ⁶⁴. It is important to note that the
302 process of focal adhesion formation is multi-faceted and complex, as it is a dynamic process
303 that is regulated by signaling cascades that are modulated by the cell's surrounding mechanical
304 environment ⁶⁵. Moreover, these processes also guide the formation of different types of stress
305 fibers (i.e., dorsal or ventral), which are determined by spatial relation to the cell nucleus. These
306 different types of stress fibers also have differing roles in cell contractility, as dorsal stress fibers
307 typically do not contain myosin while ventral stress fibers do ⁶⁶.

308
309 Given that many in vitro experiments are performed on artificial matrices, it is also important to
310 understand how long-range force transmission may play a role in these experiments. When
311 culturing cells on matrices of specific stiffness, it is possible that the cells modulate the matrix
312 stiffness by pulling on their local fibers and causing them to stiffen with increasing strain.
313 Moreover, this result may be compounded as the resulting stiffer fibrous matrix promotes
314 greater cell force generation ⁶⁷. The porosity of the matrix can also affect what the cell is
315 sensing, and the density of adhesion sites on artificial matrices might affect the interpreted
316 mechanical stiffness ⁶⁸. Relatedly, it is known that shorter fiber lengths can limit the amount of
317 traction a cell can generate, leading to altered force generation and, therefore, altered cell
318 spreading and migration ⁶⁹. In addition to static mechanical cues, it is also important to consider
319 how dynamic matrix loading is attenuated as it reaches the level of the cell, though this is
320 dependent on the type of strain that is being applied to the sample ⁷⁰. Moreover, there is
321 continued debate over how the matrix allows for strain attenuation at the level of the cell ⁷¹.
322 While cells may misinterpret mechanical cues that the artificial matrix is designed to impart to
323 them, it is also important to consider that these cells may not directly sense these mechanical
324 cues as the cells degrade and remodel matrix as well as deposit new ECM in the surrounding
325 area within hours of being seeded on the substrate ^{72, 73}. It is also possible that cells generate
326 strain fields that go beyond the matrix in their immediate vicinity, and so respond to barriers at
327 the distal side of matrices, such as the stiff frames present in Figure 2, or a rigid surface like
328 bone or tissue culture plastic that underlies the ECM or a gel. Therefore, the appropriate
329 thickness of a fibrous gel requires the consideration of long-range force transmission ^{74, 75}.

330
331 Overall, artificial matrices and biomaterials provide a tool for increased understanding of how
332 mechanical forces are transmitted through fibrous networks. They also provide a tool for
333 culturing cells within environments that closely recapitulate their physiological mechanical
334 environment. Continued use and understanding of force transmission within these artificial
335 matrices and biomaterials will allow for mechanistic understanding of long-range force
336 transmission in physiological cells and tissues.

337
338

339 **4. Modeling the mechanical behavior of biomaterials**

340 In native states, cells of different types are usually surrounded by a three-dimensional (3D)
341 fibrous microenvironment whose local physical properties can impact many important cellular
342 functions including migration and proliferation ⁷⁶. The local physical properties of the fibrous
343 microenvironment, in turn, depend on different factors including the collagen concentration,
344 initial stiffness, degree of strain stiffening, pore size, cross-linking, degradability, viscosity, and
345 plasticity ^{8, 67, 77-81}. In experimental systems, it is often difficult to isolate the potential contribution
346 of each factor, and thus the impact of each factor cannot be separately investigated. To fill this
347 gap, many computational models have been developed. In silico models offer the following

348 features that can help us to better understand the mechanics of fibrous networks: (1) each
 349 physical parameter can be independently varied, allowing decoupling of different mechanisms
 350 and assessing the contribution of each of them to the overall mechanical behavior, (2)
 351 simulations can be carried out much faster compared with experiments and they can be easily
 352 shared and replicated, (3) computational models enable us to measure the cell-generated force
 353 from the experimentally measured displacement field, and (4) simulations can reveal new
 354 perspectives of biological phenomena and therefore suggest new experiments.

355

356 4.1. Linear analysis

357 In this section, we first present the theoretical prediction from the linear elastic framework on
 358 how the strain field generated by a contractile cell decays with distance from the cell. We will
 359 then compare the strain field with the one generated within a fibrous nonlinear network to show
 360 the effect of material nonlinearity on the range of displacement propagation. Assume a spherical
 361 cell with a radius r_0 within a linear elastic matrix. Assuming that u_0 is the cell-generated radial
 362 displacement at the cell-matrix interface ($r = r_0$), our goal is to determine the matrix
 363 displacement field u as a function of the distance from the cell center $r = r_0$. To this end, we
 364 need to solve the mechanical equilibrium in the matrix

$$365 \frac{d\sigma_r}{dr} + \frac{2}{r}(\sigma_r - \sigma_\theta) = 0 \quad (1)$$

366 where σ_r and σ_θ are the radial stress and hoop stress, respectively. For linear elastic materials,
 367 σ_r and σ_θ are related to the radial and hoop strains ε_r and ε_θ as follows

$$368 \sigma_r = \frac{E}{(1+\nu)(1-2\nu)}[(1-\nu)\varepsilon_r + 2\nu\varepsilon_\theta] \quad (2)$$

$$369 \sigma_\theta = \frac{E}{(1+\nu)(1-2\nu)}[\varepsilon_\theta + \nu\varepsilon_r] \quad (3)$$

370 where E and ν are the elastic modulus and Poisson's ratio of the matrix, respectively. The
 371 strains ε_r and ε_θ for a linear material are defined in terms of the radial displacement u as follows

$$372 \varepsilon_r = \frac{du}{dr}, \quad \varepsilon_\theta = \frac{u}{r} \quad (4)$$

373 Substituting equations (2-4) into equation (1), the mechanical equilibrium can be written in the
 374 following form

$$375 \frac{d^2u}{dr^2} + \frac{2du}{rdr} - \frac{2u}{r^2} = 0 \quad (5)$$

376 To solve the above differential equation, we need two boundary conditions. Considering that the
 377 displacement u at the cell-matrix interface and far from the cell are respectively u_0 and zero, the
 378 two boundary conditions are given as follows

$$379 u(r_0) = u_0, \quad u(\infty) = 0 \quad (6)$$

380 which yields the following solution for equation (5)

$$381 u = u_0 \left(\frac{r_0}{r} \right)^2 \quad (7)$$

382 With the displacement field at hand from equation (7), the strain and stress fields can be
 383 determined from equations (2-4)

$$384 \varepsilon_r = -2 \frac{u_0(r_0)}{r} \left(\frac{r_0}{r} \right)^3, \quad \varepsilon_\theta = \frac{u_0(r_0)}{r} \left(\frac{r_0}{r} \right)^3 \quad (8)$$

$$385 \sigma_r = -\frac{2E}{(1+\nu)r} \frac{u_0(r_0)}{r} \left(\frac{r_0}{r} \right)^3, \quad \sigma_\theta = \frac{2E}{(1+\nu)r} \frac{u_0(r_0)}{r} \left(\frac{r_0}{r} \right)^3 \quad (9)$$

386 Equation (7) shows that the displacement decay in linear materials is proportional to $1/r^2$ and
387 the stress/strain decay is proportional to $1/r^3$ ^{5, 51, 82, 83}. However, experimental results from 3D
388 particle tracking microscopy experiments reveal that the cell-generated displacement field
389 decays significantly slower within collagen fibrous matrices ⁶⁷ due to the long-range
390 transmission of mechanical forces within these matrices which will be later discussed.

391

392 **4.2. Nonlinear (strain-stiffening) response of fibrous matrices**

393 A large fraction of biological materials is composed of fibrous networks whose mechanical
394 properties, unlike linear hydrogels, change as they are deformed under cell-generated forces.
395 When fibrous networks are mechanically loaded, forces are carried by individual fibers, which
396 can lead to translation, rotation, and deformation of each fiber ⁸⁴. As a result, the deformation
397 field of fibrous networks can be highly nonaffine, *i.e.*, the displacement field at the microscale
398 does not match the deformation field at the scale of the bulk material ⁷ which in turn generates a
399 nonhomogeneous local strain field entirely different from the far-field imposed strain ^{6, 84-86}. This
400 feature of fibrous networks leads to unique behaviors in tension, compression, and shear.
401 Specifically, when loaded, individual fibers in the network tend to rotate and align along the
402 direction of the maximum principal strain. The rotation and alignment of fibers can cause
403 unusual behaviors in fibrous network materials including strain stiffening and long-range force
404 transmission which distinguish them from linear elastic hydrogels. For example, as an initially
405 isotropic fibrous collagen network undergoes large deformations, there is a set of collagen fibers
406 that is reorganized and aligned in the direction of the maximum principal strain when the matrix
407 is stretched in this direction beyond a critical strain ^{5, 87} (Fig. 6). While this set of fibers reorients
408 and aligns in the maximum principal stretch direction causes strain stiffening, there is another
409 set of fibers that experiences compression and buckles in the minimum principal stretch
410 direction ^{5, 67}. The stress-strain relationship becomes even more complicated with the presence
411 of cells within the network and/or when the network is loaded multiaxially ^{8, 88}. Note that the
412 alignment of collagen fibers can lead to local stiffening of the matrix, while cells sense and
413 actively respond to this local stiffening by promoting their contractility leading to a positive
414 feedback loop between cells and the ECM ^{67, 76, 89, 90}.

415

416 As the stress and strain fields generated by a contractile cell decay with distance from the cell, it
417 is clear that large fiber alignment in the collagen matrix is confined to a region surrounding the
418 cell, while far away from the cell the stress and strain fields are small enough to be
419 approximated with linear elasticity. Sander ⁹¹ determined the critical distance from the cell above
420 which the cell-generated stress and strain fields can be approximated using linear elasticity.
421 Below the critical distance, collagen fibrous networks exhibit significant nonlinear strain
422 stiffening behavior that cannot be captured by linear elastic models as shown in Figure 7. Using
423 two-dimensional discrete fiber network simulations, Onck et al. ⁴ showed that the nonlinear
424 strain stiffening behavior of fibrous networks lies in the rearrangement of the network rather than
425 in its constituent fibers. Similarly, using realistic network architectures of collagen-I networks,
426 Stein et al., ⁹² demonstrated that the nonlinear behavior of collagen fibrous networks can be
427 entirely explained by the alignment of collagen fibers in the direction of tensile stress, as
428 opposed to entropic stiffening of individual collagen fibers.

429

430 Note that while individual collagen fibers show significant strain stiffening in tension to resist
431 extension, they buckle and soften in compression ^{25, 93, 94}. The stiffening of collagen fibrous
432 networks in tension and their softening in compression ^{88, 95} can also lead to negative normal
433 stresses when collagen networks undergo shear deformations ⁹⁴. When an initially isotropic
434 fibrous network (with fibers equally distributed in all directions) undergoes shear deformations,
435 we can assume that an equal number of fibers are stretched and compressed. If the fibrous
436 network is made of linear fibers that show the same resistance against tension and

437 compression, sliding one plate with respect to the other in shear deformations only generates
438 shear (tangential) stresses and not normal stresses (that tend to pull the plates together or push
439 them apart). However, if the fibrous network is made of collagen fibers, since the tensile force
440 generated by the stretched fibers is significantly higher than the compressive force of those
441 under compression, a net tensile force is generated that tends to pull the plates together⁹⁴. This
442 negative normal stress can be also observed in discrete fiber simulations of collagen networks
443 in shear tests where the negative normal stress increases quadratically with shear strain^{92, 96}.

444
445 Another striking property of fibrous networks is their capability to transmit forces over relatively
446 long distances. The alignment of collagen fibers in the direction of tension and the subsequent
447 stiffening of the network^{80, 97, 98} can lead to long-range transmissions of mechanical forces
448 within fibrous collagen matrices^{26, 52}. For example, when cells contract in a fibrous network, the
449 displacement can be felt as far as 20 times the cell size, which is significantly high compared
450 with the force-transmission range in linear hydrogels. As a result, cells can sense other cells
451 located at distances ~ 20 times their size in 3D collagen fibrous matrices. Note that the
452 alignment of collagen fibers by the cellular tensile forces and the subsequent long-range force
453 transmission can be even lead to the formation of collagen tracts between neighboring cells
454 through which cells can mechanically interact with each other within the matrix^{61, 99, 100}. To
455 capture the above physical behaviors of fibrous network materials, there are mainly two schools
456 of models: (i) discrete fiber network models, and (ii) continuum models. As their names imply,
457 the major difference between these two types of models is whether the fibers are treated
458 discretely or as a continuum. In the following section, we look into these two different types of
459 models and review their strengths and weaknesses.

461 **4.3. Discrete fiber networks**

462 Discrete fiber networks explicitly consider the geometry of individual fibers and the
463 microstructure of the network (Figure 8. (a-d)). Fibers in the model are connected to each other
464 when they intersect. This construction mimics the structure of natural fibrous networks. When a
465 discrete fiber network is loaded, mechanical forces are transmitted through fibers and
466 crosslinks, leading to displacement and rotation of individual fibers. The discrete fiber network
467 intrinsically captures the non-affine deformation of the fibrous network and is therefore widely
468 used to study the impact of fiber microstructure on the mechanical behavior of fibrous network
469 materials. To construct a discrete fiber network, the following two major specifications of
470 networks should be considered: (i) the microstructure of the network, and (ii) the constitutive
471 models of individual fibers.

472
473 **4.3.1. Network generation.** Since the topology of in vivo fibrous networks (e. g., collagen, fibrin
474 networks) are not well established, many models have either employed imaging-based networks
475 or artificially generated networks. Using images of a 3D collagen network, Stein et al. confirmed
476 that the alignment of fibers, instead of nonlinearity of individual fibers, lead to the strain-
477 stiffening of the whole network⁹². Ma et al. used confocal reflectance microscopy images of
478 cells and their surrounding network of collagen fibers to generate the structure of the fibrous
479 network and identified that the presence of fibers is critical for the long-range force transmission
480²⁶. Sander et al. used confocal microscopy data for a collagen-I network to propose a critical
481 radius within which the fibers are aligned due to the cell contraction⁹¹. While using real network
482 images has a clear advantage in clinical relevance, it suffers in practice from artifacts from
483 imaging techniques and segmentation algorithms. For example, fibers at different depths could
484 be misidentified to be crosslinked. Imaging at the nano- and micro- scales are also difficult to
485 segment due to limits on the resolution.

486

487 Due to these difficulties in imaging-based models, many studies use models that are artificially
 488 created. The networks can be generated by either introducing randomness in a periodic
 489 network, or randomly placing fibers in a domain according to a preset rule. In the first category,
 490 for example, Arzash et al. studied the fiber networks in the ropelike limit using periodic 2D
 491 triangular and hexagonal lattices (Figures 9a and 9b) ¹⁰¹. They eliminated fibers randomly to
 492 match the connectivity (*i. e.*, the number of fibers joined at one crosslink) with the real
 493 biopolymer networks, and to remove the unphysical effects of network-spanning fibers. In the
 494 second category, for example, the Delaunay networks are constructed by placing N random
 495 points in a box and triangulating them in a way that there is no point inside the
 496 circumferencecircle of any triangle, which maximizes the smallest angle among all triangulations
 497 of the given point set (Figure 9c) ⁸⁴. Another example is the Voronoi network which is a
 498 derivative of the Delaunay networks by connecting the circumcircles (Figure 9d) ¹⁰¹.

499
 500 With many discrete models developed for fibrous network materials, the freedom of choice in
 501 network geometry raises potential issues on the clinical relevance of the results and their
 502 implications. Humphries et al. compared dual, Voronoi, growth, and perturbed networks and
 503 found all these network geometries are able to capture the long-range mechanical
 504 communications ⁶¹. However, the response heterogeneity, fiber alignment, and substrate
 505 displacement fields are sensitive to the network choice. Aghvami et al. showed that low
 506 connectivity and rotational freedom of the fibers in the network is critical for the enhanced long-
 507 range mechanosensing ¹⁰². As the networks are generated randomly, larger variations of
 508 mechanical response were also observed with the same type of networks. This shows the
 509 importance of the choice of network geometry and further validation of the model by comparing
 510 it with experiments in multi-axial testing.

511
 512 **4.3.2. Fiber mechanical properties.** In addition to the geometry of the network, the constitutive
 513 model of individual fibers also plays an important role in the mechanical response of the
 514 network. One of the most frequent choices is linear elastic beams. When deformed, the strain
 515 energy is given by

$$516 \quad U = \frac{1}{2} \int EI(\nabla^2 u)^2 ds + \frac{1}{2} \int EA(dl/ds)^2 ds \quad (10)$$

517 where E denotes the Young's modulus, A represents the cross-sectional area of the beam, and
 518 I indicates the moment of inertia. The ratio I/A indicates the easiness of bending the fiber.
 519 When the fiber is long and thin with large I and small A , it is easier to bend the fiber than to
 520 stretch it. When compressed, the fibers (modeled as elastic beams) will buckle due to instability,
 521 leading to the softening of the whole network. In some studies, the fibers are modeled as wavy
 522 structures with curvatures. This resembles the shape of fibers observed in many experiments.
 523 These filaments are assumed to be stress-free in the initial wavy state and when loaded, the
 524 work required for the deformation of the network is stored as bending strain energy in each
 525 fiber. Onck et al. studied modeling wavy fibers and concluded that despite quantitative
 526 differences, the general behavior is qualitatively similar ⁴.

527 528 **4.4. Anisotropic strain-stiffening continuum models**

529 Recently, several continuum models have been developed to capture the long-range force
 530 transmission in fibrous networks (Figure 8(e-i)). While the discrete fiber networks can explicitly
 531 illustrate the mechanism of fiber realignment, they are computationally complex. Moreover,
 532 since the networks are generated randomly, the results are statistical, making it difficult to
 533 reproduce the results. Continuum models are simpler with fewer parameters and deterministic
 534 without randomness, making them a convenient tool to model experiments. Wang et al. ⁵
 535 developed a constitutive continuum model by incorporating the fact that the fibrous materials
 536 stiffen preferentially along the directions of tensile principal stretches. The model is developed

537 based on discrete fiber simulations that show aligned fibers stiffen the network anisotropically
 538 along the loading direction (Figure 6). The strain energy density of the matrix in this model can
 539 be written as

$$540 \quad W = \underbrace{\frac{\mu}{2}(\bar{I}_1 - 3)}_{\text{randomly aligned fibers}} + \underbrace{\frac{k}{2}(J - 1)^2 + \sum_{a=1}^3 f(\lambda_a)}_{\text{aligned fibers}} \quad (11)$$

541 where the first part captures the isotropic mechanical behavior of randomly distributed fibers
 542 using a hyperelastic neo-Hookean material, and the second part captures the alignment of
 543 fibers which causes strain-stiffening along the principal stretch directions. μ and k respectively
 544 denote the initial shear and bulk moduli, \bar{I}_1 is the first invariant of the deviatoric part of the
 545 Cauchy-Green tensor, J denotes the determinant of the deformation gradient tensor, and λ_a
 546 ($a = 1, 2, 3$) represents the principal stretches. In equation (11), f is a non-linear function which
 547 rises sharply as λ_a increases, capturing the anisotropic strain-stiffening induced by fiber
 548 alignment. Wang et al.⁵ showed that the ability of the material to anisotropically stiffen along the
 549 loading direction is essential to capture the long-range force transmission. However, the specific
 550 form of the constitutive equation is not crucial as long as it captures the orientational anisotropy
 551 and stiffening that naturally arise along the principal directions upon loading. Using the strain
 552 energy function in equation (11), the radial stress in equation (2) can be obtained in the
 553 following form (see reference⁵)

$$554 \quad \sigma_r = \frac{E}{(1 + \nu)(1 - 2\nu)}[(1 - \nu)\varepsilon_r + 2\nu\varepsilon_\theta] + E_f \varepsilon_r \quad (12)$$

555 where E_f represents the stiffening response of collagen matrices in tension. Substituting σ_r (from
 556 equation (12)) and σ_θ (from equation (3)) into the equilibrium equation (1) yields the following
 557 equation

$$558 \quad \left[1 + \frac{(1 + \nu)(1 - 2\nu)E_f}{(1 - \nu)E} \right] \left(\frac{d^2 u}{dr^2} + \frac{2 du}{r dr} \right) - \frac{2u}{r^2} = 0 \quad (13)$$

559 Solving equation (13) with the boundary conditions (6) yields the following solution

$$560 \quad u = u_0 \left(\frac{r_0}{r} \right)^n \quad (14)$$

561 where

$$562 \quad n = \frac{1}{2} \left(\sqrt{\frac{9 + \chi}{1 + \chi}} + 1 \right) \quad , \quad \chi = \frac{(1 + \nu)(1 - 2\nu)E_f}{(1 - \nu)E} \quad (15)$$

563 Note that for $E_f/E \gg 1$ (strong fibrous response), the exponent $n \rightarrow 1$ and therefore equation (14)
 564 shows a slow decay of displacement, whereas for an isotropic material ($E_f/E \ll 1$), $n \rightarrow 2$ which
 565 yields equation (7).

566

567 This continuum model has been successfully used to explain and predict the force transmission
 568 in collagen matrices with different microstructures¹⁰⁰. Hall et al.⁶⁷ used single-cell traction force
 569 measurements for breast cancer cells embedded within 3D collagen matrices. As expected, the
 570 displacements are highest in the matrix near the two tips of the cell along the long axis of the
 571 cell. While the isotropic neo-Hookean hyperplastic model predicted a quick decay of the
 572 displacement field with distance from the cell, the experimentally measured displacement field
 573 decays significantly slower and can be only captured by the above continuum model (Figure
 574 7B). With the help of the computational model, Hall et al.⁶⁷ identified that the cells are able to
 575 generate sufficient strain to locally align and stiffen the surrounding collagen matrix, which in
 576 turn positively feeds back to the cell to enhance the generation of cell contractile force.
 577

578 In addition to discrete fiber network and continuum models, multiscale models have been also
579 used to study the mechanics of fibrous matrices^{103, 104}. These multiscale models use both
580 continuum and discrete fiber network frameworks to simulate material behavior at different
581 scales. At the macroscopic scale, these multiscale models use a continuum framework, but
582 instead of using a constitutive equation to relate the stress to the strain, discrete fiber network
583 simulations at the microscopic scale are used at the locations where the stress-strain
584 relationships needed for the continuum simulation¹⁰⁵. Note that continuum and multiscale
585 models can also enable us to approximate cell-generated traction forces within fibrous collagen
586 environments^{67, 83, 100}. Historically, in methods for measuring cell-generated forces, cells are
587 cultured on a linear elastic hydrogel with known mechanical properties and we use the
588 experimentally-measured displacement field generated on the surface of the hydrogel together
589 with a linear elastic constitutive model to calculate cell traction forces. As discussed earlier, the
590 linear elastic model, however, cannot capture the mechanical behavior of collagen fibrous
591 matrices and thus cannot be used to measure cell-generated forces within these physiologically
592 more relevant environments.

593

594

595 **5. Conclusions**

596 Physical signals allow cells to sense the presence of other cells at distances much larger than
597 are possible by diffusing chemical signals. These physical signals include direct transmission of
598 force from one cell to another, as well as cell traction-generated changes in the alignment,
599 density and stiffness of the extracellular matrix. Long-range force transmission in biological
600 materials appears to require the unique, nonlinear responses of fibrous networks such as those
601 that form the extracellular matrix and the intracellular cytoskeleton. There is much still to learn,
602 both experimentally and theoretically, about how fibrous networks respond to the forces
603 generated in biological tissues, and understanding these principles can lead to better methods
604 for characterizing soft tissues and to improved biomimetic materials.

605

606 **Figure Legends**

607

608 Figure 1. Two rat nerves were severed and then placed in a blood plasma clot. The regenerating cell at
609 the top form a bridge from one nerve end to the other. From¹⁵.

610

611 Figure 2. (A) Morphology of 3T3 fibroblasts in grids with opening widths of 200 μm , 500 μm , and 1700 μm
612 visualized by rhodamine phalloidin staining for actin filaments. (B) Cell-induced alignment of collagen
613 networks. After remodeling by cells, collagen fibers imaged by confocal reflectance microscopy were
614 aligned parallel to cell extensions. Scale bar: 20 μm . From¹⁷.

615

616 Figure 3. Macrophages ($\text{M}\phi$) are attracted by local pulling events in collagen ECM. (A) $\text{M}\phi$ were seeded
617 onto collagen ECM with microneedles inserted 5 μm into the 200 μm thick collagen gel. Lateral collagen
618 deformation was performed by using negative pressure to pull collagen fibers into the tip. $\text{M}\phi$ migration
619 was tracked from phase contrast movies. Scale bar: 100 μm . (B) Deformation field growth with time. (C)
620 $\text{M}\phi$ trajectories are plotted with respect to distance from the microneedle.

621

622 Figure 4. A. Morphology of collagen ECM and fibroblasts surrounding a non-metastatic EpH4-Ev spheroid
623 and a metastatic 67NR spheroid, demonstrating increased alignment surrounding the metastatic
624 spheroid. B,C. Magnetically-controlled increased fiber alignment to model the effect of the cancerous
625 spheroid results in increased rates of diffusion of exosome-sized particles. Scale bars: 200 μm . From⁴¹.

626

627 Figure 5. A. Neural crest cell group treated with SDF1 gradient to induce migration, with migratory
628 behavior abolished via relaxing contractility at the rear of the cell group via optoGEF-relax. B. Neural crest
629 cell group without SDF1 begins to directionally migrate when contractility at the rear side of the group is
630 induced via optoGEF-contract. From⁵⁴.

631
 632 Figure 6. Discrete fiber simulations of an initially random (isotropic) fiber network before (a) and after (b)
 633 50% shear strain. The inset in (a) shows that fibers are isotropically distributed in all directions in the initial
 634 configuration. The inset in (b) shows that after the shear deformation, more fibers are aligned in the 45°
 635 orientation which coincides with the direction of the maximum principal stretch⁵.

636
 637 Figure 7. Long-range force transmission within a three-dimensional collagen network. (A) Deformation
 638 field generated by an MDA-MB-231 breast cancer cell within a three-dimensional collagen network. Each
 639 arrow represents the displacement of a fluorescent bead covalently bonded to collagen fibers. 4,000 of
 640 12,000 tracked bead displacements are shown. Arrows are rendered at four times their true size. The cell
 641 is shown in magenta. The inset shows a zoomed-in view where all displacement vectors are rendered at
 642 their true scale. (B) Bead displacements along the long axis of the cell are plotted as a function of their
 643 position along the long axis of the cell. Coordinate (0,0) represents the center of the cell. Solid lines are
 644 fits to the experimental data (circles) using three different material models: fibrous model (red)⁶⁷,
 645 nonlinear hyperelastic neo-Hookean model (black), and linear elastic model (blue).

646
 647 Figure 8. (a-d) Numerical results from discrete fiber network simulations show the interaction between two
 648 cells with different center-to-center distances at 90% cell contraction⁸⁷. When the distance is 50 μm, cells
 649 of all aspect ratios mechanically interact by forming collagen tracts (a and c). However, as the separation
 650 distance increases, only cells with high aspect ratios (d) can mechanically interact with each other, while
 651 no visible collagen tracts are observed for circular cells (b). (e-i) Numerical results from continuum models
 652⁵. Contour plots of the maximum principal strain in three-dimensional matrices for linear isotropic
 653 materials (e) and fibrous materials (f). Vector plots of the maximum principal strain which coincides with
 654 the orientation of the collagen fibers after cellular contraction (g). Contour plots of the maximum principal
 655 strain on two-dimensional matrices for linear isotropic materials (h) and fibrous materials (i).

656
 657 Figure 9. Different networks for discrete fiber simulations. (a) A triangular lattice network. The arc denotes
 658 that one of the three crossing fibers is detached from the cross-link which reduces the local connectivity
 659 from 6 to 4. (b) A hexagonal lattice which has a local connectivity of 3. (c) A Delaunay network with a
 660 nonuniform local connectivity which has the average local connectivity of 6. (d) A Voronoi network which
 661 has a local connectivity of 3.

662
 663
 664

665 References

- 666
- 667 1. P. Weiss, *Roux Archiv für Entwicklungsmechanik der Organismen*, 1929, 116, 438–554.
 - 668 2. P. A. Janmey, E. J. Amis and J. D. Ferry, *Journal of Rheology*, 1983, 27, 135-153.
 - 669 3. C. Storm, J. J. Pastore, F. C. MacKintosh, T. C. Lubensky and P. A. Janmey, *Nature*, 2005,
 670 435, 191-194.
 - 671 4. P. R. Onck, T. Koeman, T. van Dillen and E. van der Giessen, *Physical Review Letters*,
 672 2005, 95.
 - 673 5. H. Wang, A. S. Abhilash, Christopher S. Chen, Rebecca G. Wells and Vivek B. Shenoy,
 674 *Biophysical Journal*, 2014, 107, 2592-2603.
 - 675 6. P. L. Chandran and V. H. Barocas, *Journal of Biomechanical Engineering*, 2006, 128, 259-
 676 270.
 - 677 7. Q. Wen, A. Basu, P. A. Janmey and A. G. Yodh, *Soft Matter*, 2012, 8, 8039-8049.
 - 678 8. E. Ban, H. Wang, J. M. Franklin, J. T. Liphardt, P. A. Janmey and V. B. Shenoy, *Proceedings*
 679 *of the National Academy of Sciences*, 2019, 116, 6790-6799.
 - 680 9. T. Bok, *Folia neurobiologica*, 1915, 9, 475.

- 681 10. P. Weiss, *Journal of Experimental Zoology*, 1934, 68, 393-448.
- 682 11. A. K. Harris, D. Stopak and P. Wild, *Nature*, 1981, 290, 249-251.
- 683 12. D. Stopak and A. K. Harris, *Dev Biol*, 1982, 90, 383-398.
- 684 13. J. C. Mira, *J Anat*, 1979, 129, 77-93.
- 685 14. A. A. Katzberg, *Science*, 1951, 114, 431-432.
- 686 15. P. Weiss, *Science*, 1952, 115, 293-295.
- 687 16. Y. L. Han, P. Ronceray, G. Xu, A. Malandrino, R. D. Kamm, M. Lenz, C. P. Broedersz and
688 M. Guo, *Proceedings of the National Academy of Sciences of the United States of*
689 *America*, 2018, 115, 4075-4080.
- 690 17. H. Mohammadi, P. A. Janmey and C. A. McCulloch, *Biomaterials*, 2014, 35, 1138-1149.
- 691 18. J. P. Winer, S. Oake and P. A. Janmey, *PLoS One*, 2009, 4, e6382.
- 692 19. S. van Helvert and P. Friedl, *ACS Appl Mater Interfaces*, 2016, 8, 21946-21955.
- 693 20. P. A. Janmey, D. A. Fletcher and C. A. Reinhart-King, *Physiol Rev*, 2020, 100, 695-724.
- 694 21. A. Mann, R. S. Sopher, S. Goren, O. Shelah, O. Tchaicheeyan and A. Lesman, *Journal of*
695 *the Royal Society Interface*, 2019, 16.
- 696 22. X. Li, R. Balagam, T.-F. He, P. P. Lee, O. A. Igoshin and H. Levine, *Proceedings of the*
697 *National Academy of Sciences of the United States of America*, 2017, 114, 8974-8979.
- 698 23. H. Kang, Q. Wen, P. A. Janmey, J. X. Tang, E. Conti and F. C. MacKintosh, *J Phys Chem B*,
699 2009, 113, 3799-3805.
- 700 24. J. V. Shah and P. A. Janmey, *Rheologica Acta*, 1997, 36, 262-268.
- 701 25. X. Xu and S. A. Safran, *Physical Review E*, 2015, 92.
- 702 26. X. Ma, M. E. Schickel, M. D. Stevenson, A. L. Sarang-Sieminski, K. J. Gooch, S. N. Ghadiali
703 and R. T. Hart, *Biophysical Journal*, 2013, 104, 1410-1418.
- 704 27. P. Pakshir, M. Alizadehgiashi, B. Wong, N. M. Coelho, X. Chen, Z. Gong, V. B. Shenoy, C.
705 McCulloch and B. Hinz, *Nature Communications*, 2019, 10.
- 706 28. J. Davies, in *Principles of Developmental Genetics: Second Edition*, Elsevier Inc., 2015, pp.
707 255-264.
- 708 29. D. Warburton, A. El-Hashash, G. Carraro, C. Tiozzo, F. Sala, O. Rogers, S. D. Langhe, P. J.
709 Kemp, D. Riccardi, J. Torday, S. Bellusci, W. Shi, S. R. Lubkin and E. Jesudason, *Current*
710 *Topics in Developmental Biology*, 2010, 90, 73-158.
- 711 30. K. Goodwin, S. Mao, T. Guyomar, E. Miller, D. C. Radisky, A. Košmrlj and C. M. Nelson,
712 *Development (Cambridge)*, 2019, 146.
- 713 31. C. M. Nelson, J. P. Gleghorn, M. F. Pang, J. M. Jaslove, K. Goodwin, V. D. Varner, E.
714 Miller, D. C. Radisky and H. A. Stone, *Development (Cambridge)*, 2017, 144, 4328-4335.
- 715 32. C.-L. Guo, M. Ouyang, J.-Y. Yu, J. Maslov, A. Price and C.-Y. Shen, *Proceedings of the*
716 *National Academy of Sciences of the United States of America*, 2012, 109, 5576-5582.
- 717 33. D. G. Brownfield, G. Venugopalan, A. Lo, H. Mori, K. Tanner, D. A. Fletcher and M. J.
718 Bissell, *Current Biology*, 2013, 23, 703-709.
- 719 34. Q. M. Shi, R. P. Ghosh, H. Engelke, C. H. Rycroft, L. Cassereau, J. A. Sethian, V. M. Weaver
720 and J. T. Liphardt, *Proceedings of the National Academy of Sciences of the United States*
721 *of America*, 2014, 111, 658-663.
- 722 35. B. R. Freedman, A. B. Rodriguez, C. D. Hillin, S. N. Weiss, B. Han, L. Han and L. J.
723 Soslowsky, *Journal of the Royal Society Interface*, 2018, 15, 20170880.

- 724 36. B. R. Freedman, A. B. Rodriguez, R. J. Leiphart, J. B. Newton, E. Ban, J. J. Sarver, R. L.
725 Mauck, V. B. Shenoy and L. J. Soslowsky, *Scientific Reports*, 2018, 8, 1-13.
- 726 37. K. Gardner, M. Lavagnino, M. Egerbacher and S. P. Arnoczky, *Journal of Orthopaedic*
727 *Research*, 2012, 30, 1695-1701.
- 728 38. M. Lavagnino, A. E. Brooks, A. N. Oslapas, K. L. Gardner and S. P. Arnoczky, *Journal of*
729 *Orthopaedic Research*, 2017, 35, 573-579.
- 730 39. J. H. Kim, G. Lee, Y. Won, M. Lee, J. S. Kwak, C. H. Chun and J. S. Chun, *Proceedings of the*
731 *National Academy of Sciences of the United States of America*, 2015, 112, 9424-9429.
- 732 40. D. Wirtz, K. Konstantopoulos and P. C. Searson, in *Nature Reviews Cancer*, Nature
733 Publishing Group, 2011, vol. 11, pp. 512-522.
- 734 41. W. H. Jung, N. Yam, C. C. Chen, K. Elawad, B. Hu and Y. Chen, *Biomaterials*, 2020, 234,
735 119756.
- 736 42. D.-H. Kim, A. J. Ewald, J. Park, Kshitiz, M. Kwak, R. S. Gray, C.-Y. Su, J. Seo, S. S. An and A.
737 Levchenko, *Scientific Reports*, 2018, 8.
- 738 43. A. J. Maniotis, C. S. Chen and D. E. Ingber, *Proc Natl Acad Sci U S A*, 1997, 94, 849-854.
- 739 44. A. E. Miller, P. Hu and T. H. Barker, *Advanced Healthcare Materials*, 2020, 9.
- 740 45. S. S. Ng, C. Li and V. Chan, *Interface Focus*, 2011, 1, 777-791.
- 741 46. A. J. Engler, S. Sen, H. L. Sweeney and D. E. Discher, *Cell*, 2006, 126, 677-689.
- 742 47. N. Wang, J. D. Tytell and D. E. Ingber, *Nat Rev Mol Cell Biol*, 2009, 10, 75-82.
- 743 48. A. Islam, M. Younesi, T. Mbimba and O. Akkus, *Advanced Healthcare Materials*, 2016, 5,
744 2237-2247.
- 745 49. Y. G. Kang, H. Jang, T. D. Yang, J. Notbohm, Y. Choi, Y. Park and B.-M. Kim, *Journal of*
746 *Biomedical Optics*, 2018, 23, 1.
- 747 50. I. Lelidis and J. F. Joanny, *Soft Matter*, 2013, 9, 11120-11128.
- 748 51. J. Notbohm, A. Lesman, P. Rosakis, D. A. Tirrell and G. Ravichandran, *Journal of the Royal*
749 *Society Interface*, 2015, 12, 20150320.
- 750 52. P. Ronceray, C. P. Broedersz and M. Lenz, *Proceedings of the National Academy of*
751 *Sciences*, 2016, 113, 2827-2832.
- 752 53. N. Gjorevski, A. S. Piotrowski, V. D. Varner and C. M. Nelson, *Scientific Reports*, 2015, 5.
- 753 54. A. Shellard, A. Szabó, X. Trepát and R. Mayor, *Science*, 2018, 362, 339-343.
- 754 55. H. Ebata, K. Moriyama, T. Kuboki and S. Kidoaki, *Biomaterials*, 2020, 230, 119647.
- 755 56. M. Dietrich, H. Le Roy, D. B. Brückner, H. Engelke, R. Zantl, J. O. Rädler and C. P.
756 Broedersz, *Soft Matter*, 2018, 14, 2816-2826.
- 757 57. B. J. DuChes, A. D. Doyle, E. K. Dimitriadis and K. M. Yamada, *Biophysical Journal*, 2019,
758 116, 670-683.
- 759 58. L. Sapir and S. Tzlil, *Seminars in Cell & Developmental Biology*, 2017, 71, 99-105.
- 760 59. C. D. Davidson, W. Y. Wang, I. Zaimi, D. K. P. Jayco and B. M. Baker, *Scientific Reports*,
761 2019, 9.
- 762 60. D. Gomez, S. Natan, Y. Shokef and A. Lesman, *Advanced Biosystems*, 2019.
- 763 61. D. L. Humphries, J. A. Grogan and E. A. Gaffney, *Bulletin of Mathematical Biology*, 2017,
764 79, 498-524.
- 765 62. K. Liu, S. M. Mihaila, A. Rowan, E. Oosterwijk and P. H. J. Kouwer, *Biomacromolecules*,
766 2019, 20, 826-834.
- 767 63. B. Burkel and J. Notbohm, *Soft Matter*, 2017, 13, 5749-5758.

- 768 64. B. M. Baker, B. Trappmann, W. Y. Wang, M. S. Sakar, I. L. Kim, V. B. Shenoy, J. A. Burdick
769 and C. S. Chen, *Nature Materials*, 2015, 14, 1262-1268.
- 770 65. S. Tojkander, G. Gateva and P. Lappalainen, *Journal of Cell Science*, 2012, 125, 1855-
771 1864.
- 772 66. S. Tojkander, G. Gateva, G. Schevzov, P. Hotulainen, P. Naumanen, C. Martin, P. W.
773 Gunning and P. Lappalainen, *Current Biology*, 2011, 21, 539-550.
- 774 67. M. S. Hall, F. Alisafaei, E. Ban, X. Feng, C.-Y. Hui, V. B. Shenoy and M. Wu, *Proceedings of
775 the National Academy of Sciences of the United States of America*, 2016, 113, 14043-
776 14048.
- 777 68. J. H. Wen, L. G. Vincent, A. Fuhrmann, Y. S. Choi, K. C. Hribar, H. Taylor-Weiner, S. Chen
778 and A. J. Engler, *Nature Materials*, 2014, 13, 979-987.
- 779 69. J. Xie, M. Bao, S. M. C. Bruekers and W. T. S. Huck, *Acs Applied Materials & Interfaces*,
780 2017, 9, 19630-19637.
- 781 70. M. E. Wall, P. S. Weinhold, T. Siu, T. D. Brown and A. J. Banes, *Journal of Biomechanics*,
782 2007, 40, 173-181.
- 783 71. M. S. Rudnicki, H. A. Cirka, M. Aghvami, Edward A. Sander, Q. Wen and Kristen L. Billiar,
784 *Biophysical Journal*, 2013, 105, 11-20.
- 785 72. S. A. Ferreira, M. S. Motwani, P. A. Faull, A. J. Seymour, T. T. L. Yu, M. Enayati, D. K.
786 Taheem, C. Salzlechner, T. Haghighi, E. M. Kania, O. P. Oommen, T. Ahmed, S. Loaiza, K.
787 Parzych, F. Dazzi, O. P. Varghese, F. Festy, A. E. Grigoriadis, H. W. Auner, A. P. Snijders, L.
788 Bozec and E. Gentleman, *Nature Communications*, 2018, 9.
- 789 73. A. Malandrino, X. Trepas, R. D. Kamm and M. Mak, *Plos Computational Biology*, 2019,
790 15.
- 791 74. S. Sen, A. J. Engler and D. E. Discher, *Cell Mol Bioeng*, 2009, 2, 39-48.
- 792 75. A. Buxboim, K. Rajagopal, A. E. X. Brown and D. E. Discher, *Journal of Physics-Condensed
793 Matter*, 2010, 22.
- 794 76. S. van Helvert, C. Storm and P. Friedl, *Nature Cell Biology*, 2018, 20, 8-20.
- 795 77. O. Chaudhuri, L. Gu, D. Klumpers, M. Darnell, S. A. Bencherif, J. C. Weaver, N. Huebsch,
796 H.-p. Lee, E. Lippens, G. N. Duda and D. J. Mooney, *Nature Materials*, 2016, 15, 326-334.
- 797 78. B. P. Flynn, A. P. Bhole, N. Saeidi, M. Liles, C. A. DiMarzio and J. W. Ruberti, *PLoS One*,
798 2010, 5, e12337.
- 799 79. J. Kim, J. Feng, C. A. R. Jones, X. Mao, L. M. Sander, H. Levine and B. Sun, *Nature
800 Communications*, 2017, 8, 842.
- 801 80. D. Shakiba, B. Babaei, F. Saadat, S. Thomopoulos and G. M. Genin, *Proceedings of the
802 National Academy of Sciences*, 2017, 114, 5772-5774.
- 803 81. E. Ban, J. M. Franklin, S. Nam, L. R. Smith, H. Wang, R. G. Wells, O. Chaudhuri, J. T.
804 Liphardt and V. B. Shenoy, *Biophysical Journal*, 2018, 114, 450-461.
- 805 82. Y. Shokef and S. A. Safran, *Physical Review Letters*, 2012, 108, 178103.
- 806 83. J. Steinwachs, C. Metzner, K. Skodzek, N. Lang, I. Thievessen, C. Mark, S. Münster, K. E.
807 Aifantis and B. Fabry, *Nature Methods*, 2016, 13, 171-176.
- 808 84. R. C. Picu, *Soft Matter*, 2011, 7, 6768.
- 809 85. D. A. Head, A. J. Levine and F. C. MacKintosh, *Physical Review Letters*, 2003, 91, 108102.
- 810 86. J. Wilhelm and E. Frey, *Physical Review Letters*, 2003, 91, 108103.

- 811 87. A. S. Abhilash, B. M. Baker, B. Trappmann, C. S. Chen and V. B. Shenoy, *Biophysical*
812 *Journal*, 2014, 107, 1829-1840.
- 813 88. A. S. van Oosten, M. Vahabi, A. J. Licup, A. Sharma, P. A. Galie, F. C. MacKintosh and P. A.
814 Janmey, *Sci Rep*, 2016, 6, 19270.
- 815 89. F. Alisafaei, D. S. Jokhun, G. V. Shivashankar and V. B. Shenoy, *Proceedings of the*
816 *National Academy of Sciences*, 2019, 116, 13200-13209.
- 817 90. H. Ahmadzadeh, M. R. Webster, R. Behera, A. M. Jimenez Valencia, D. Wirtz, A. T.
818 Weeraratna and V. B. Shenoy, *Proc Natl Acad Sci U S A*, 2017, 114, E1617-E1626.
- 819 91. L. M. Sander, *Journal of Biomechanical Engineering*, 2013, 135.
- 820 92. A. M. Stein, D. A. Vader, D. A. Weitz and L. M. Sander, *Complexity*, 2011, 16, 22-28.
- 821 93. C. P. Broedersz and F. C. MacKintosh, *Reviews of Modern Physics*, 2014, 86, 995-1036.
- 822 94. P. A. Janmey, M. E. McCormick, S. Rammensee, J. L. Leight, P. C. Georges and F. C.
823 MacKintosh, *Nature Materials*, 2007, 6, 48-51.
- 824 95. M. Vahabi, A. Sharma, A. J. Licup, A. S. van Oosten, P. A. Galie, P. A. Janmey and F. C.
825 MacKintosh, *Soft Matter*, 2016, 12, 5050-5060.
- 826 96. E. Conti and F. C. MacKintosh, *Physical Review Letters*, 2009, 102, 088102.
- 827 97. B. Lee, X. Zhou, K. Riching, K. W. Eliceiri, P. J. Keely, S. A. Guelcher, A. M. Weaver and Y.
828 Jiang, *PLoS One*, 2014, 9, e111896.
- 829 98. T. Stylianopoulos and V. H. Barocas, *Computer Methods in Applied Mechanics and*
830 *Engineering*, 2007, 196, 2981-2990.
- 831 99. M. Aghvami, V. H. Barocas and E. A. Sander, *Journal of Biomechanical Engineering*, 2013,
832 135.
- 833 100. D. Shakiba, F. Alisafaei, A. Savadipour, R. A. Rowe, Z. Liu, K. M. Pryse, V. B. Shenoy, E. L.
834 Elson and G. M. Genin, *ACS Nano*, 2020.
- 835 101. S. Arzash, J. L. Shivers, A. J. Licup, A. Sharma and F. C. MacKintosh, *Physical Review E*,
836 2019, 99.
- 837 102. M. Aghvami, K. L. Billiar and E. A. Sander, *Journal of Biomechanical Engineering-*
838 *Transactions of the Asme*, 2016, 138.
- 839 103. M. Yasodharababu and A. K. Nair, *Cell Mol Bioeng*, 2020, 13, 229-245.
- 840 104. V. K. Lai, M. F. Hadi, R. T. Tranquillo and V. H. Barocas, *J Biomech Eng*, 2013, 135, 71007.
- 841 105. E. A. Sander, T. Stylianopoulos, R. T. Tranquillo and V. H. Barocas, *Proc Natl Acad Sci U S*
842 *A*, 2009, 106, 17675-17680.

843

844

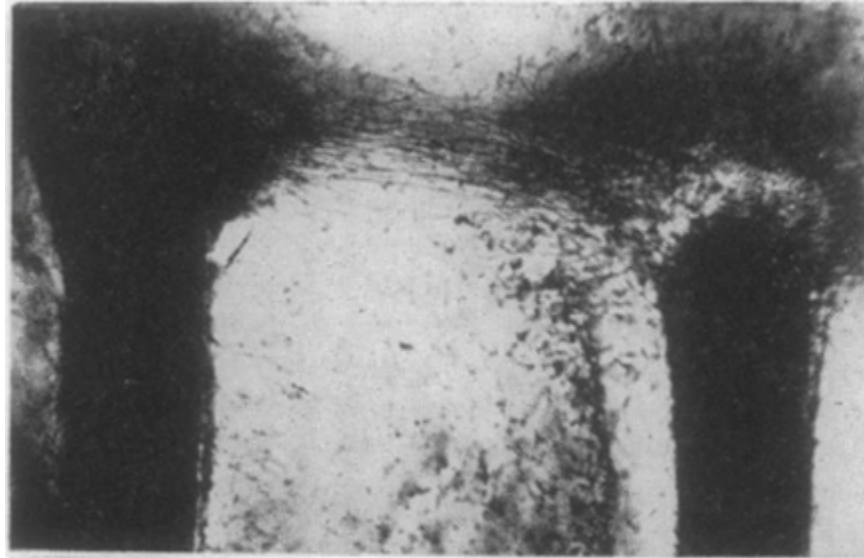


Figure 1. Two rat nerves were severed and then placed in a blood plasma clot. The regenerating cell at the top form a bridge from one nerve end to the other. From 1.

114x73mm (96 x 96 DPI)

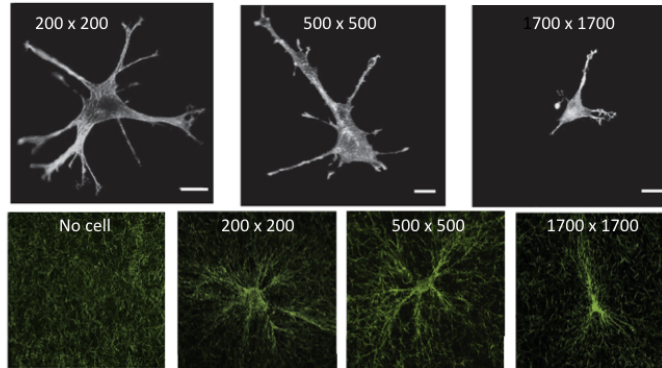


Figure 2. (A) Morphology of 3T3 fibroblasts in grids with opening widths of 200 μm , 500 μm , and 1700 μm visualized by rhodamine phalloidin staining for actin filaments. (B) Cell-induced alignment of collagen networks. After remodeling by cells, collagen fibers imaged by confocal reflectance microscopy were aligned parallel to cell extensions. Scale bar: 20 μm . From 17.

338x190mm (72 x 72 DPI)

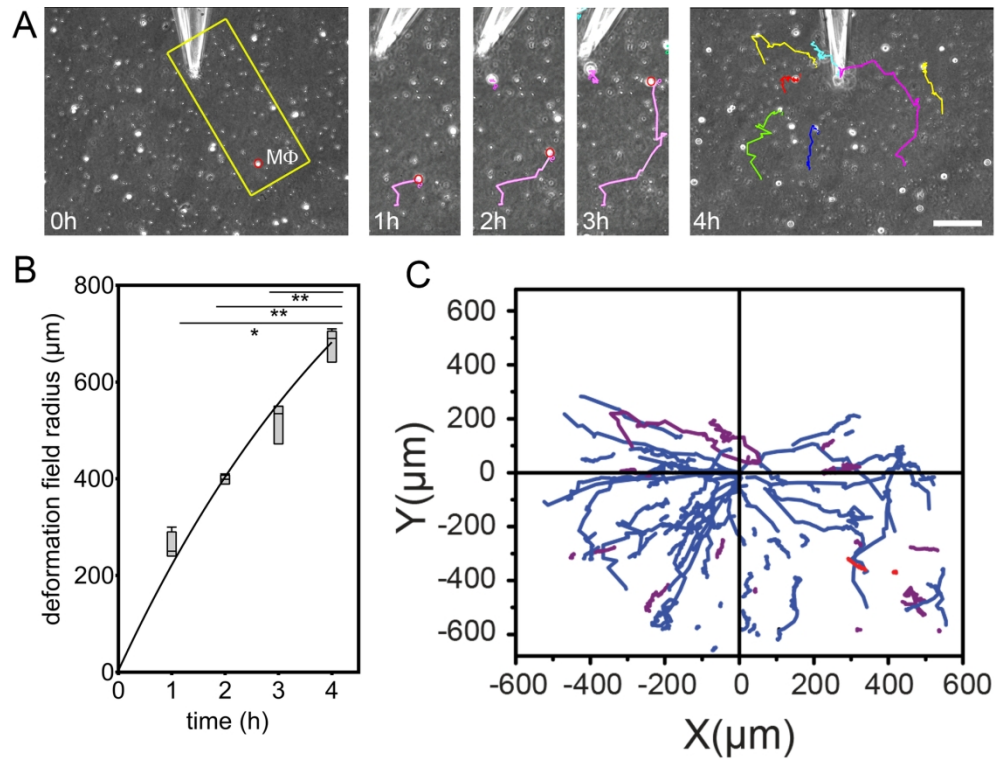


Figure 3. Macrophages (M ϕ) are attracted by local pulling events in collagen ECM. (A) M ϕ were seeded onto collagen ECM with microneedles inserted 5 μm into the 200 μm thick collagen gel. Lateral collagen deformation was performed by using negative pressure to pull collagen fibers into the tip. M ϕ migration was tracked from phase contrast movies. Scale bar: 100 μm . (B) Deformation field growth with time. (C) M ϕ trajectories are plotted with respect to distance from the microneedle.

175x136mm (300 x 300 DPI)

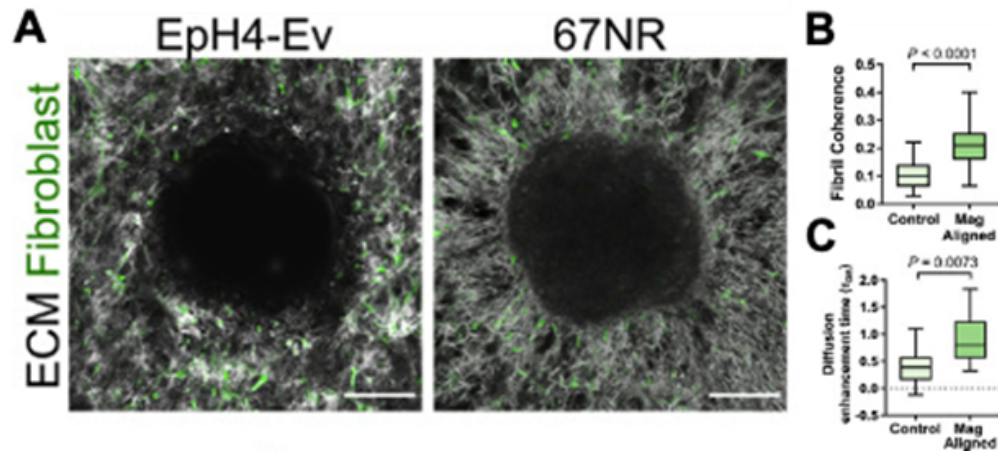


Figure 4. A. Morphology of collagen ECM and fibroblasts surrounding a non-metastatic EpH4-Ev spheroid and a metastatic 67NR spheroid, demonstrating increased alignment surrounding the metastatic spheroid. B,C. Magnetically-controlled increased fiber alignment to model the effect of the cancerous spheroid results in increased rates of diffusion of exosome-sized particles. From 3.

161x72mm (96 x 96 DPI)

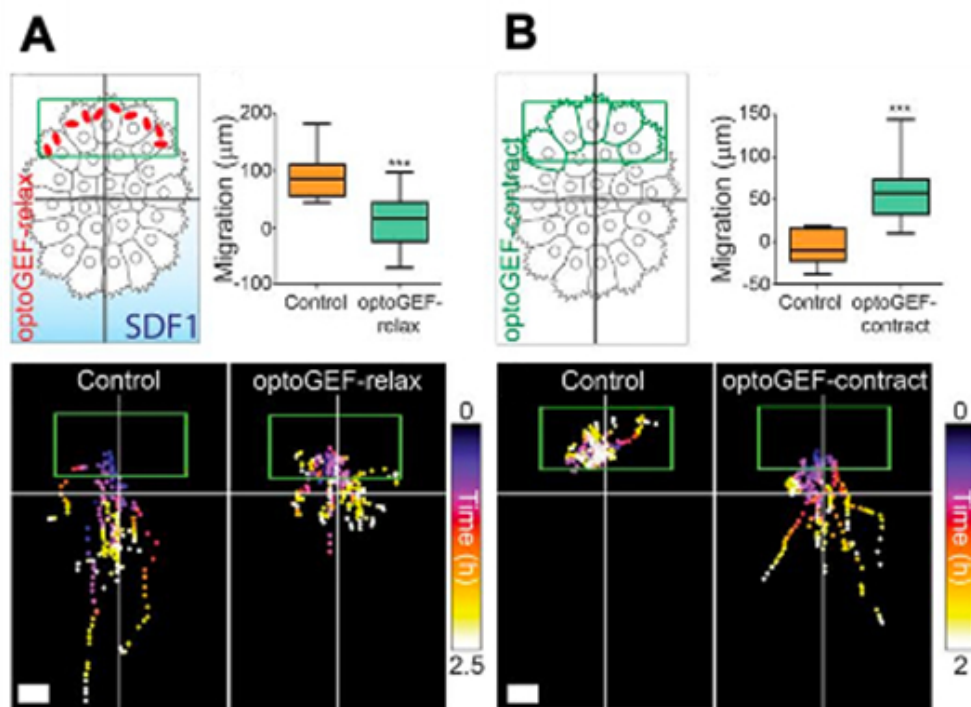


Figure 5. A. Neural crest cell group treated with SDF1 gradient to induce migration, with migratory behavior abolished via relaxing contractility at the rear of the cell group via optoGEF-relax. B. Neural crest cell group without SDF1 begins to directionally migrate when contractility at the rear side of the group is induced via optoGEF-contrast. From 2.

142x104mm (96 x 96 DPI)

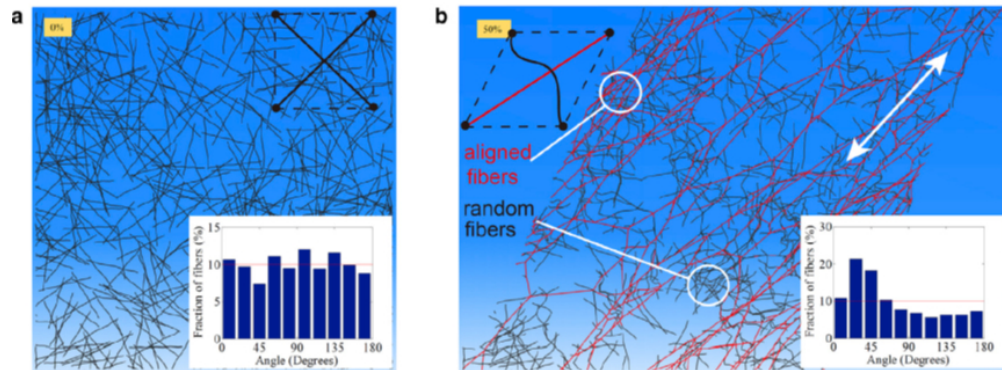


Figure 6. Discrete fiber simulations of an initially random (isotropic) fiber network before (a) and after (b) 50% shear strain. The inset in (a) shows that fibers are isotropically distributed in all directions in the initial configuration. The inset in (b) shows that after the shear deformation, more fibers are aligned in the 45° orientation which coincides with the direction of the maximum principal stretch ϵ_1 .

228x85mm (96 x 96 DPI)

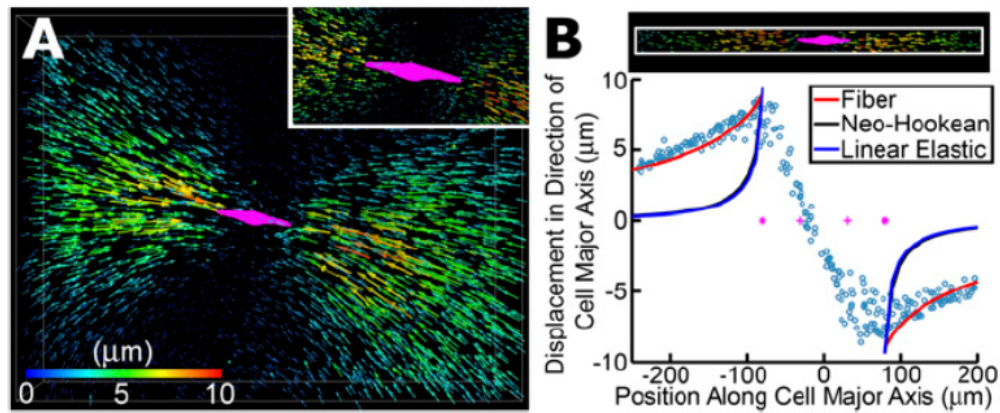


Figure 7. Long-range force transmission within a three-dimensional collagen network. (A) Deformation field generated by an MDA-MB-231 breast cancer cell within a three-dimensional collagen network. Each arrow represents the displacement of a fluorescent bead covalently bonded to collagen fibers. 4,000 of 12,000 tracked bead displacements are shown. Arrows are rendered at four times their true size. The cell is shown in magenta. The inset shows a zoomed-in view where all displacement vectors are rendered at their true scale. (B) Bead displacements along the long axis of the cell are plotted as a function of their position along the long axis of the cell. Coordinate (0,0) represents the center of the cell. Solid lines are fits to the experimental data (circles) using three different material models: fibrous model (red) 57, nonlinear hyperelastic neo-Hookean model (black), and linear elastic model (blue).

203x85mm (96 x 96 DPI)

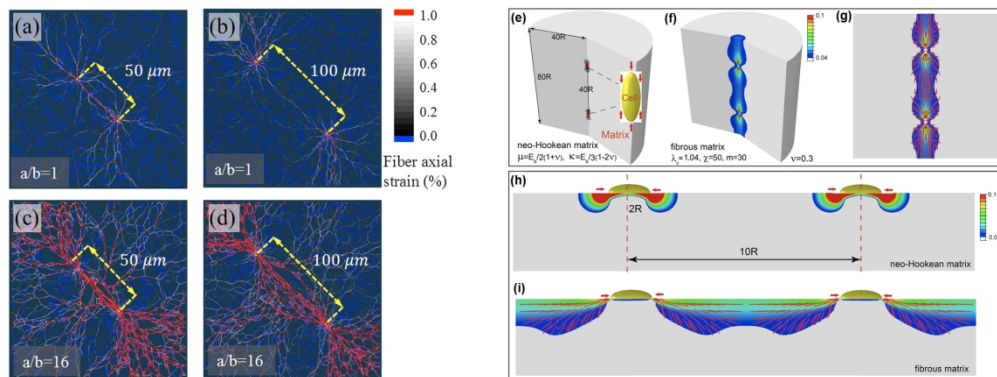


Figure 8. (a-d) Numerical results from discrete network fiber simulations show the interaction between two cells with different center-to-center distances at 90% cell contraction $\gamma = 77$. When the distance is $50 \mu\text{m}$, cells of all aspect ratios mechanically interact by forming collagen tracts (a and c). However, as the separation distance increases, only cells with high aspect ratios (d) can mechanically interact with each other, while no visible collagen tracts are observed for circular cells (b). (e-i) Numerical results from continuum models 8. Contour plots of the maximum principal strain in three-dimensional matrices for linear isotropic materials (e) and fibrous materials (f). Vector plots of the maximum principal strain which coincides with the orientation of the collagen fibers after cellular contraction (g). Contour plots of the maximum principal strain on two-dimensional matrices for linear isotropic materials (h) and fibrous materials (i).

516x193mm (96 x 96 DPI)

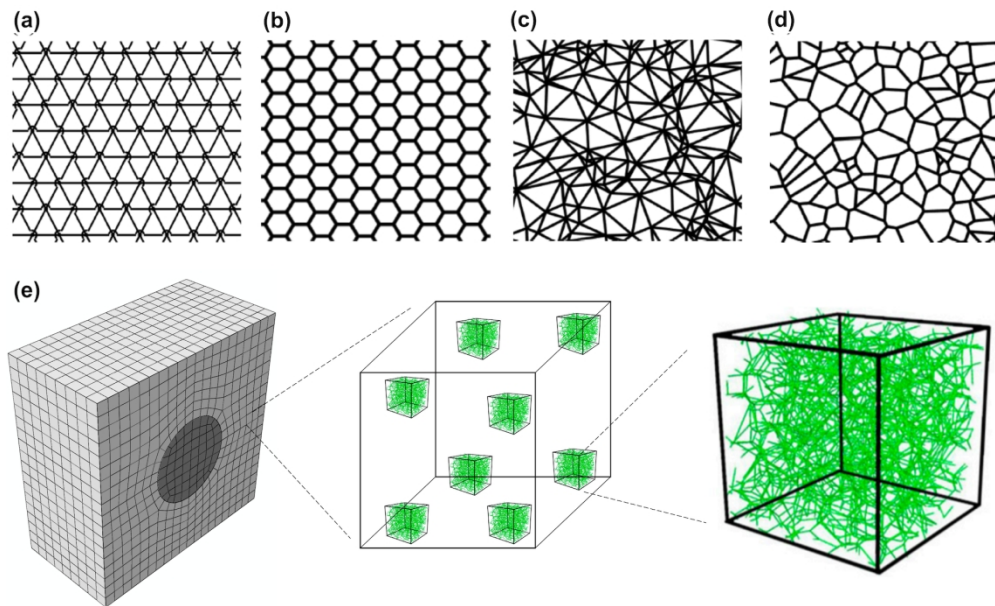


Figure 9. Different networks for discrete fiber simulations. (a) A triangular lattice network. The arc denotes that one of the three crossing fibers is detached from the cross-link which reduces the local connectivity from 6 to 4. (b) A hexagonal lattice which has a local connectivity of 3. (c) A Delaunay network with a nonuniform local connectivity which has the average local connectivity of 6. (d) A Voronoi network which has a local connectivity of 3.

203x123mm (256 x 256 DPI)