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Charging-driven coarsening and melting of a colloidal nanoparticle monolayer at an ionic liquid-vacuum interface[†]Connor G. Bischak,^{a*} Jonathan G. Raybin,^{a‡} Jonathon W. Kruppe^{b‡} and Naomi S. Ginsberg^{abcdef}

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We induce and investigate the coarsening and melting dynamics of an initially static nanoparticle colloidal monolayer at an ionic liquid-vacuum interface, driven by a focused, scanning electron beam. Coarsening occurs through grain interface migration and larger-scale motions such as grain rotations, often facilitated by sliding dislocations. The progressive decrease in area fraction that drives melting of the monolayer is explained using an electrowetting model whereby particles at the interface are solvated once their accumulating charge recruits sufficient counterions to subsume the particle. Subject to stochastic particle removal from the monolayer, melting is recapitulated in simulations with a Lennard-Jones potential. This new driving mechanism for colloidal systems, whose dynamical timescales we show can be controlled with the accelerating voltage, opens the possibility to manipulate particle interactions dynamically without need to vary particle intrinsic properties or surface treatments. Furthermore, the decrease in particle size availed by electron imaging presents opportunities to observe force and time scales in a lesser-explored regime intermediate between typical colloidal and molecular systems.

1 Introduction

Colloidal physics has provided an extensive and readily accessible platform to mimic interatomic and intermolecular interactions and associated phase behavior and to discover emergent phenomena distinct from those occurring in traditional condensed matter systems.^{1–8} More recently, the exploration of soft matter driven away from equilibrium by various means has opened new possibilities to study and create systems in novel dynamic regimes. Irrespective of whether the individual constituents are active or passive, their dynamics are primarily visualized with optical microscopy,^{9–14} requiring feature sizes larger than the optical diffraction limit. Studying colloidal behavior in the limit

of molecular-scale balances between attractive and repulsive interparticle interactions would require much smaller particle sizes and a different observation modality.

Although often intended as a means to passively image materials with high spatial resolution, electron irradiation often plays an active role in driving dynamic nanoscale or atomic processes. For instance, electron irradiation has been used to controllably etch nanoparticles,¹⁵ induce crystallization in two-dimensional materials,¹⁶ drive particle assemblies at the nanoscale,^{17,18} and manipulate the positions of individual atoms in a lattice¹⁹. Using electron irradiation for such processes provides high spatial and temporal resolution and precise control over electron-sample interactions by modulating the current and accelerating voltage of the electron beam. For recording or inducing nanoparticle dynamics, electron imaging allows for the precise mapping of particle trajectories leading to the formation of higher-order assemblies. Although in situ investigations of particle assemblies in liquid environments are becoming more prevalent,^{20,21} the exact impact of electron irradiation on the dynamics of these assemblies is not well understood.

Merging high resolution and deliberate manipulation of colloidal systems with electron microscopy provides the opportunity both to explore interactions among smaller particles and to leverage and elucidate the influence of an electron beam on a colloidal system while employing it to drive different behaviors. Both the particle size and the electron beam enable interparticle interac-

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tions on different, shorter time scales that would otherwise be challenging to observe. While most colloidal studies are performed on monolayers in aqueous solutions, ionic liquids (IL)—replete with enigmatic properties in their own right—are interesting alternative media, especially for higher-resolution electron microscopy studies, owing to their low vapor pressures.²² Recent efforts have shown that colloidal nanoparticles can be incorporated into ILs²³ or self-assembled at IL interfaces.²⁴ For example, Kim et al. recently showed that scanning electron microscopy (SEM) can image few-nanoparticle dynamics at the IL-vacuum interface and map interaction potentials between particles.²¹

Here we present the dynamics of a 2D colloidal polycrystalline monolayer at an IL-vacuum interface driven by the irradiation of a focused, scanning electron beam. By introducing the combination of an IL interface and the charge-mediated actuation of the nanoparticles, we show that progressive charging of the 300-nm-diameter insulating silica nanoparticles by the electron beam causes a series of dynamic processes. First, particle charging, combined with compensation by IL screening, increases particle kinetic energy to mobilize the particles, resulting in annealing of the monolayer. Second, the particles detach from the IL-vacuum interface, resulting in the formation of point vacancies in the 2D monolayer, which leads to a melting phase transition. We first describe the dynamics that we induce and observe, then explore the macroscopic and microscopic dynamics of each stage, and finally describe a model to consistently explain the sum total of the observations. This newfound understanding of how the electron beam and IL cooperate to drive particle dynamics will enable a powerful new approach to manipulating colloidal systems without the need to vary their intrinsic properties.

2 Results and Discussion

2.1 Driven colloidal monolayer dynamics

Figure 1A shows the experimental setup, consisting of a polycrystalline colloidal monolayer that self-assembles at the interface between an IL droplet and vacuum. The self-assembled colloidal monolayer is formed by combining an aqueous solution of colloidal nanospheres (silica, 300 nm) with an IL (1-ethyl-3-methylimidazolium methyl sulfate). The mixture is pipetted onto a silicon substrate and placed in a vacuum chamber (40 mtorr). As the water evaporates, the beads assemble at the interface between the IL and vacuum environment. The polycrystalline monolayer is formed on a droplet of IL (1-3 mm diameter and ~ 100 -200 μm height) and placed inside a SEM. The monolayer, which covers a large region, primarily forms a hexagonal lattice although it contains many point vacancies and disordered interfaces between crystalline grains (Figure S1). The formation of a polycrystalline monolayer suggests an attractive potential between the colloids. Upon repeated irradiation by a scanning, focused electron beam, the particles first begin to fluctuate, and then the monolayer begins to coarsen. Eventually, individual particles disappear from the interface, reducing the area fraction (ϕ_A) until the monolayer melts (Figure 1B). The inset to Figure 1B shows a single particle disappearing from a crystalline region of the monolayer upon electron beam exposure. Most often, the

particle completely disappears within a single frame, but in some cases, the particle disappears more gradually (Figure S2).

Figure 1C shows SEM images of the particle monolayer as a function of time, as well as corresponding images false colored by the magnitude of the single particle hexagonal bond order parameter ($|\Psi_{6,j}|$). The images are acquired at an accelerating voltage of 5 kV, a beam current of 100 pA, and a scan rate of 2.56 s/frame. Movie S1 includes all recorded frames. Here, this bond order parameter is defined as:

$$\Psi_{6,j} = \frac{1}{N_{nm}} \sum_{k=1}^{N_{nm}} e^{6i\theta_{j,k}} \quad (1)$$

where N_{nm} is the number of neighboring particles within 0.4 μm , $\theta_{j,k}$ is the angle between the bond vector connecting particle j and neighboring particle k . $\Psi_{6,j}$ quantifies the crystallinity of individual particles based on the spatial configuration of their nearest neighbors. The magnitude, ranging between 0 and 1, specifies the local order of the particles, while the phase provides the local orientation of a crystalline region that includes the particle, referenced to the horizontal direction in this case. We note that under this convention for $\Psi_{6,j}$, a particle next to a vacancy that is otherwise configured with perfect hexagon symmetry with its five remaining neighbors retains a high magnitude of $\Psi_{6,j}$. The false color aids in visualizing disordered regions (blue) and crystalline regions (yellow) of the film. Based on the time series of images, once the particles begin to fluctuate, we observe two primary regimes of dynamics. First, we observe that the number of blue disordered particles decreases with time, indicating coarsening of the monolayer. Second, after some time, individual particles begin to disappear, causing the area fraction to decrease and resulting in a higher fraction of particles with a low $|\Psi_{6,j}|$ value (bluer), and ultimately to melting of the 2D solid, followed by conversion to a 2D vapor at low enough area fraction. These stages are all captured in $g(r)$, the radial distribution function of the monolayer as a function of time (Figure 1D), which describes the average spatial correlations between the colloidal particles and can be used to identify the structural phase²⁵. At early times, $g(r)$ shows sharp peaks and long-range correlations characteristic of a solid hexagonal lattice (Figure 1D left). Over time, the peaks broaden and damp out to show a function representative of a liquid (Figure 1D middle). At the longest times, the disappearance of additional particles gives rise to a noisier liquid-like $g(r)$ profile (Figure 1D right). Similar dynamics to those depicted in Figure 1 have been measured reproducibly in separate trials, an example of which is shown in Figure S3. We next independently characterize the two primary regimes of particle dynamics, coarsening and melting, to better understand how the electron beam impacts particle motion in the self-assembled monolayer.

2.1.1 Coarsening

Using single particle tracking, we obtain the dynamics of the coarsening at the level of the entire several-hundred nanoparticle field of view to elucidate how the degree of crystallinity increases over time, and we also examine smaller fields of view to follow individual events contributing to coarsening, such as grain merging and disordered interface area reduction. To visualize

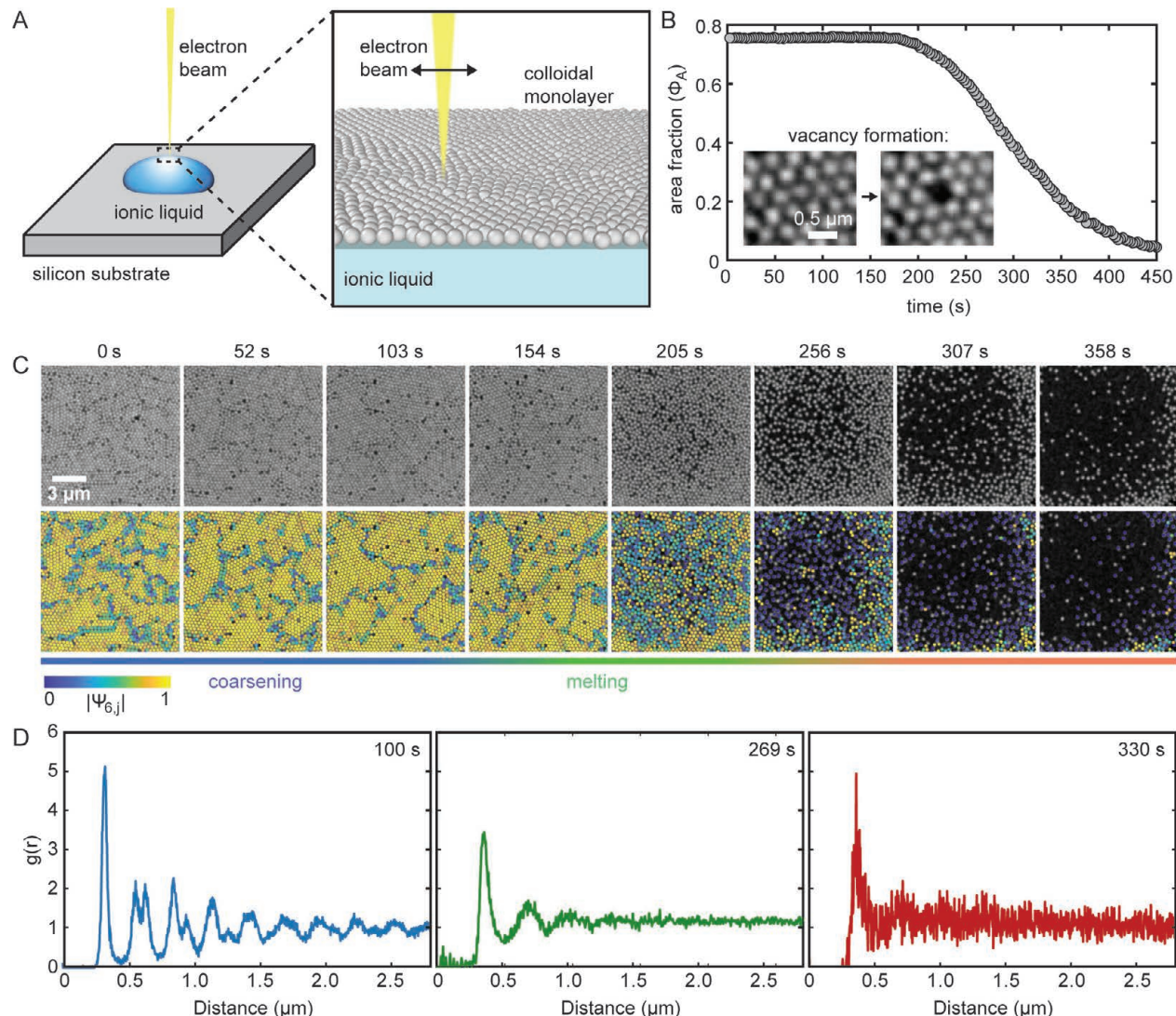


Fig. 1 Overview of coarsening, melting, and subsequent thinning upon electron beam irradiation. (A) Experimental setup showing the colloidal monolayer at the IL-vacuum interface and the electron beam scanning over the monolayer. (B) Area fraction of the monolayer as a function of time. The inset shows a single vacancy forming upon electron beam irradiation. (C) Time series of raw electron beam images and correlated image with false color indicating the magnitude of the single particle hexagonal bond order parameter ($|\Psi_{6,j}|$) with increasing electron beam irradiation. It shows coarsening and then melting of the colloidal monolayer. (D) Radial distribution function $g(r)$ at 100 s, 222 s, and 330 s illustrates the progression from hexagonal solid to liquid to lower density liquid.

how the grain structure changes with electron beam irradiation, we calculate θ_j , the phase of $\Psi_{6,j}$. The value of θ_j describes the orientation of single particles relative to their neighbors and is a convenient way to visualize the evolution of grain structure. Figure 2A shows the evolution of the colloidal monolayer upon electron beam irradiation over the initial coarsening period. Initially at 0 s, the film is composed of many small crystalline grains, which merge into larger grains as the electron beam exposure time increases. This merging of grains occurs through a number of processes, including grain interface motion, grain rotations, and other more complex correlated particle motions. As the small grains merge, we observe a decrease in the number of disordered particles. To separately visualize disordered versus ordered particles, we false color particles with $|\Psi_{6,j}| < 0.7$ in blue and particles with $|\Psi_{6,j}| > 0.7$ in yellow (Figure 2B). We observe that

the number of blue particles decreases substantially with time, indicating that disordered particles are being transferred to the crystalline phase. We also map rapid changes in particle crystallinity by plotting the change in $|\Psi_{6,j}|$ from frame to frame ($\Delta|\Psi_{6,j}|$), with $\Delta|\Psi_{6,j}| > 0.2$ in purple and $\Delta|\Psi_{6,j}| < -0.2$ in cyan overlaid on the SEM image (Figure S4). We find that the most substantial fluctuations occur at interfaces between grains. In Figure 2C we plot the number of crystalline ($|\Psi_{6,j}| > 0.7$) and disordered ($|\Psi_{6,j}| < 0.7$) particles. After about 20 s, the number of crystalline particles increases, while the number of disordered particles decreases. After approximately 80 s, both populations have nearly plateaued. Overall, we find that the electron beam anneals the particle monolayer over the first ~ 100 s of imaging.

More microscopically, this coarsening occurs through both grain interface migration and larger-scale correlative motions,

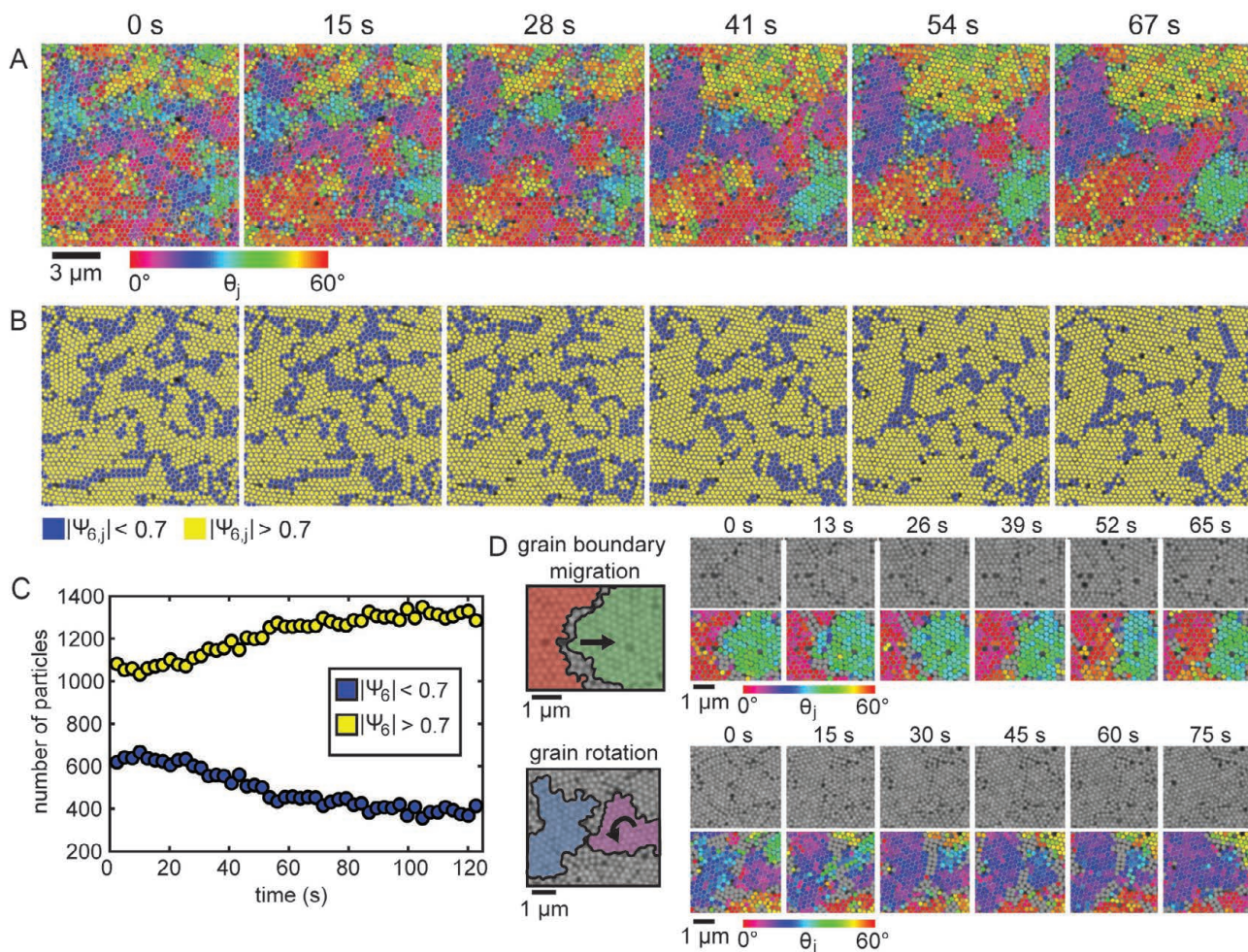


Fig. 2 Coarsening dynamics. (A) Time series of the colloidal monolayer false colored by the phase of the bond order parameter (θ_j). (B) Number of disordered (blue, $|\Psi_{6,j}| < 0.7$) and bulk crystalline (yellow, $|\Psi_{6,j}| > 0.7$) particles as a function of time. (C) Average $|\Psi_6|$ as a function of time for crystalline (yellow, $|\Psi_6| > 0.7$) and disordered (blue, $|\Psi_6| < 0.7$) particles. (D) Two types of correlated motion of particles, grain interface migration and grain rotation, false colored by θ_j .

such as grain rotations. An example of each type of motion is included in Figure 2D. We show the SEM electron images and the corresponding color maps of θ_j . In the first case, the red grain takes over a larger portion of the field of view as the interface between it and the green domain migrates to the right. In the second case, the purple grain rotates until the two grains merge into a single grain (Figure S5A). At the single particle level, particles typically hop from one grain to the other to facilitate grain interface motion, yet in other cases, we find that sliding dislocations play a prevailing role. These sliding dislocations, shown schematically in Figure S5B, are observed often upon electron beam exposure. One function of these sliding dislocations is to shuttle point vacancies to disordered interfaces, as shown in Figure S5C. In this case, sliding dislocations serve as a mechanism to transport vacancies over relatively long distances from highly ordered regions to more disordered regions, such as grain interfaces. The shuttling of defects can increase the fluidity of the interfaces and contribute to their motions. Sliding dislocations also underlie grain rotations (Figure S5D). In Figure S5D, we observe a sliding dislocation forming in the middle of a grain. This

sliding dislocation changes the overall grain orientation, which is also helped by relatively disordered regions surrounding the single crystal grain. The above examples emphasize the importance of correlated motions, such as sliding dislocations, of particles in the coarsening process. We next focus on the second regime of electron beam-induced particle dynamics, melting of the monolayer due to particle vacancy formation under electron beam irradiation.

2.1.2 Melting

We observe that a liquid phase emerges through the formation of vacancies in the crystal lattice. Figure 3A shows a time series of the particle monolayer after the initial coarsening occurs, between 115 and 205 s, false colored by $|\Psi_{6,j}|$, as in Figure 1C. As the electron beam scans, particles recede into the IL, effectively lowering the area fraction of the colloidal monolayer. Particles begin disappearing, and the area fraction decreases as shown in Figure 1B. We find that particles disappear both within crystalline and disordered regions. While particles are observed to disappear less frequently toward the edges of the field of view, this is only a boundary effect, with the irradiated particles being close

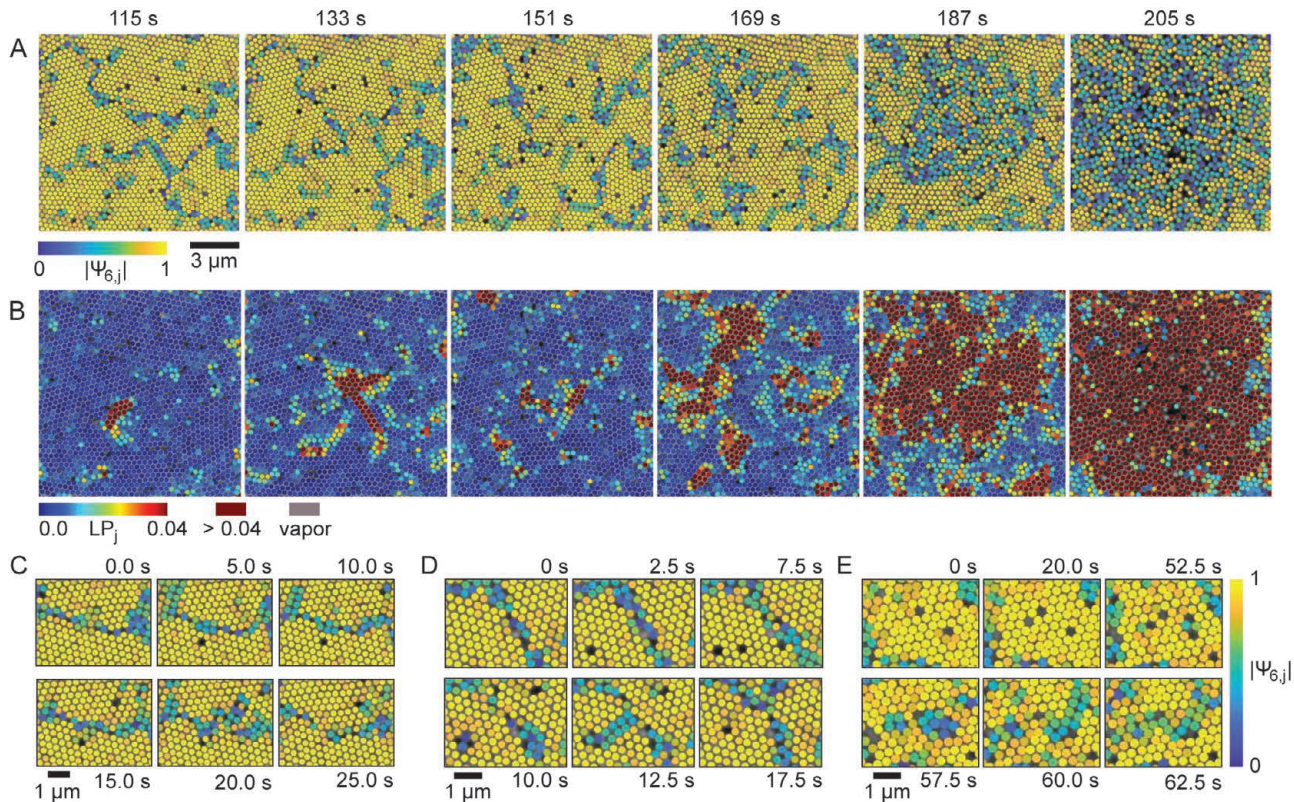


Fig. 3 Microscopic dynamics of melting and thinning. (A) Time series of particles false colored by the magnitude of the bond orientational order parameter ($|\Psi_{6,j}|$). (B) Time series of particles false colored by the single particle instantaneous Lindemann parameter, LP_j , with vapor particles (no nearest neighbors) colored grey. (C) Example of a vacancy generated and then incorporated into a grain interface with false color showing $|\Psi_{6,j}|$. (D) Example of two neighboring vacancies generated, merged, and then incorporated into a grain interface through a sliding dislocation with false color showing $|\Psi_{6,j}|$. (E) Example of a series of generated vacancies forming a more disordered region with false color showing $|\Psi_{6,j}|$.

to unirradiated ones beyond the field of view.

To gain further insights into the dynamics of the system upon particle deletion, we use single particle tracking and calculate an instantaneous 2D Lindemann parameter^{26–30} for each particle j as a function of time t (Figure 3B), which we define as:

$$LP_j(\tau_0, t) = \frac{\left(\frac{1}{N_{nn}} \sum_k^{N_{nn}} (\Delta r_j(\tau_0, t) - \Delta r_k(\tau_0, t))\right)^2}{2a^2} \quad (2)$$

where τ_0 represents the time between successive frames, N_{nn} is the number of nearest neighbors, a corresponds to the average spacing between particles at time $t = 0$, and $\Delta r_j(\tau_0, t) = r_j(\tau_0 + t) - r_j(t)$ indicates the displacement of particle j between successive frames. (See the Supplementary Information for details on how this expression is adapted from Ref.²⁸.) The instantaneous Lindemann parameter associated with a given particle j captures the extent of correlated motions with its nearest neighbors, which therefore decreases upon melting. We distinguish solid-like and liquid-like particles at any given time point by specifying a threshold value for the divergence of LP_j of 0.04. Particles with $LP_j > 0.04$, colored in dark red in Figure 3B, are designated as liquid-like particles.^{29,30} The liquid-like particles are initially, e.g. at 151 s, few in number. As more particles disappear, the 2D system enters a state of coexistence between a solid and liquid phase, with particles rapidly converting between solid and liquid states (e.g., 169 s). After enough particles disappear (e.g., 187 s),

the liquid phase emerges.

Figure 3C-E show different microscopic mechanisms for the emergence of the liquid phase. In Figure 3C, a vacancy forms adjacent to a grain interface. As the interface moves, the vacancy is incorporated into it. Overall, the incorporation of vacancies leads to larger, more mobile disordered regions. In Figure 3D, two adjacent vacancies are formed. These vacancies combine and then are transported to more disordered regions through a sliding dislocation. These sliding dislocations, common motifs in the data, relay vacancies to disordered regions over a larger distance. Finally, Figure 3E shows how the formation of multiple vacancies can produce a new disordered region.

Interestingly, the melting transition observed upon repeated electron beam irradiation can be recapitulated with simple Brownian dynamics simulations. Rather than explicitly including progressive particle charging, we need only use a Lennard-Jones interparticle potential with fixed parameters and a phenomenological, randomized particle deletion rate that follows the experimentally obtained area fraction versus time (Figure 1B). We carry out our simulations in reduced units and find that a temperature $T = 0.75$ yields optimal qualitative agreement with the experiment, indicating that the system undergoes a solid to fluid melting transition upon progressive particle disappearance³¹. The simulation is initiated at a point corresponding to ~ 100 s of evolution experimentally, indicated as a dashed vertical line in Figure 4C-F,

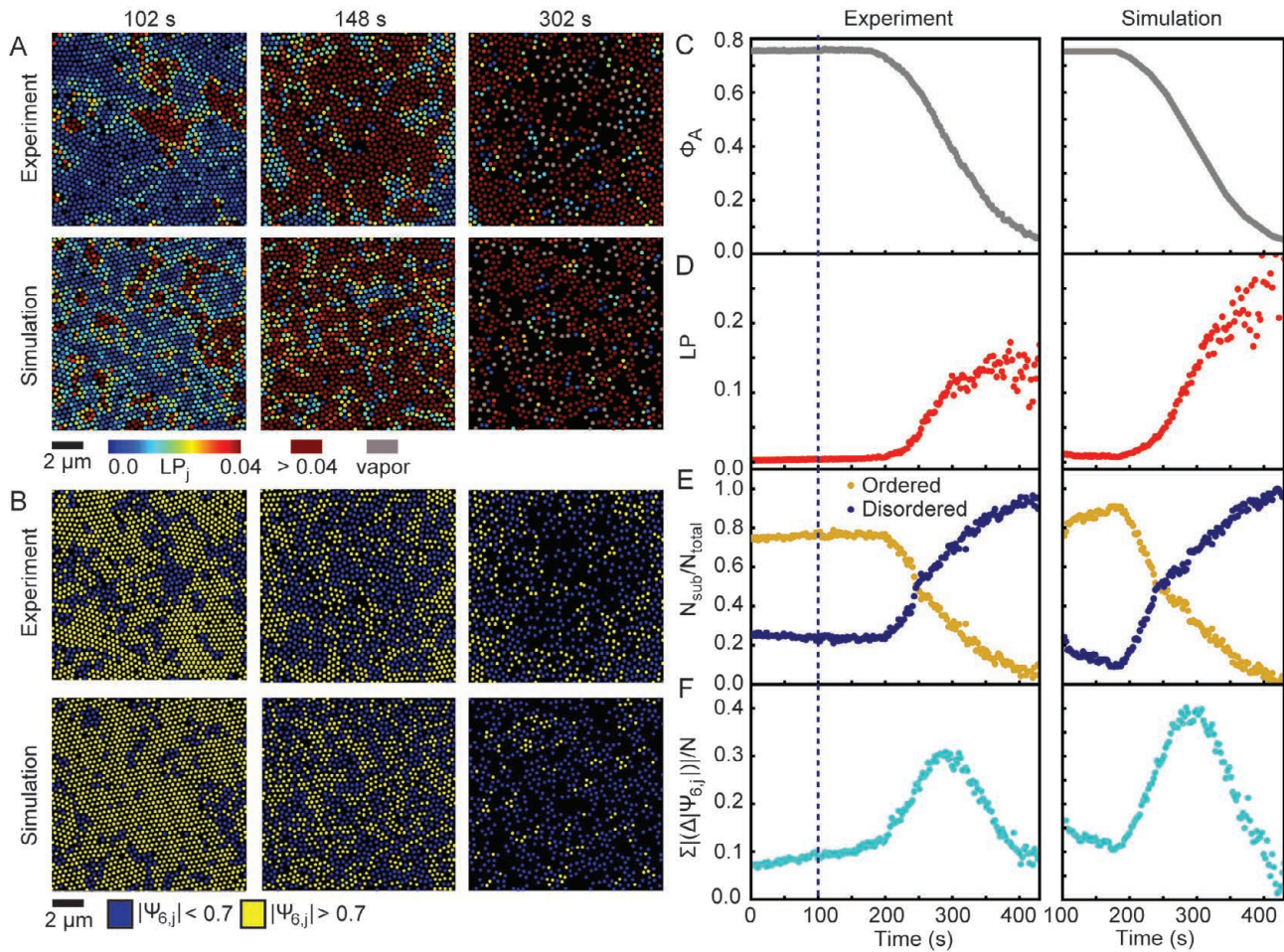


Fig. 4 Melting is consistent with simulations of monolayer dynamics. (A) Comparison of the single particle instantaneous Lindemann parameter LP_j for the experiment and simulation as a function of time. (B) Comparison of $|\Psi_{6,j}|$ as a function of time for experiment and simulation and simulation with $|\Psi_{6,j}| > 0.7$ in yellow and $|\Psi_{6,j}| < 0.7$ in blue. (C) Area fraction of particles as a function of time for experiment and simulation. Average instantaneous Lindemann parameter (LP) (D), fraction of particle with $|\Psi_{6,j}| > 0.7$ in yellow and $|\Psi_{6,j}| < 0.7$ in blue (E), and average of frame-to-frame single particle bond order parameter fluctuations, $\Sigma(|\Delta|\Psi_{6,j}|)/N$, as a function of time (F) for both experiment and simulation.

by using the experimental configuration of particles at that point as the initial condition. Before this time point, the primary role of the electron beam is to increase particle fluctuations without decreasing the area fraction (see Movie S1). Figure 4A compares 102, 148, and 302 s snapshots of experimental and simulation data false colored by the single particle instantaneous Lindemann parameter, LP_j . Figure 4B compares the same experimental and simulation frames, false colored instead by $|\Psi_{6,j}|$, using the same false coloring scheme as in Figure 2B. These images show qualitative similarities between experiment and simulation.

To provide a more quantitative comparison, we evaluate a series of parameters while the area fraction of particles changes upon electron beam-induced particle disappearance for both the experiment and simulation (Figure 4C). These include the instantaneous Lindemann parameter of the entire system (Figure 4D), the number fraction of particles with $|\Psi_{6,j}|$ both above and below a value of 0.7 (Figure 4E), and a representation of the average particle fluctuations (Figure 4F) via the average frame-to-frame change of the magnitude of the bond order parameter. A comparison of $g(r)$ at various times between simulation and experi-

ment is shown in Figure S6. The good match in these parameters between the simulation and experiment suggests that the simulation captures the same dynamic processes as observed experimentally and that the experimental system is well described as a 2D Lennard-Jones system, which is known to undergo a density controlled melting phase transition³¹.

We first investigate the evolution of the system's instantaneous Lindemann parameter (Figure 4D). Upon a melting phase transition, the average value of the Lindemann parameter should diverge.²⁶ Through a similar trend in the experiment and simulation we find that the system's instantaneous Lindemann parameter begins to diverge shortly after the particles begin disappearing, at an area fraction still very close to its original value. This point in the evolution indicates the beginning of the melting transition and can be used as a reference point for other average properties. Next, we evaluate the number of non-crystalline (disordered) particles versus crystalline (ordered) particles as the system evolves using the same definition of crystalline versus noncrystalline particles as in Figure 4B ($|\Psi_{6,j}| < 0.7$ are non-crystalline and $|\Psi_{6,j}| > 0.7$ are crystalline). Before the instan-

taneous Lindemann parameter begins to diverge, the number of crystalline and non-crystalline particles remains roughly constant (Figure 4E). After melting begins (~ 200 s), the number of crystalline particles sharply decreases, whereas the number of non-crystalline particles increases. This observation is consistent with an expansion of the disordered regions as particles are removed from the system. Last, we investigate the average frame-to-frame magnitude change in $|\Psi_{6,j}|$, i.e., $\langle |\Delta|\Psi_{6,j}| \rangle$, which shows how the magnitude of fluctuations between crystalline and non-crystalline states changes as particles disappear (Figure 4F). After melting begins, this ensemble averaged measure of fluctuations increases as more of the fluid phase emerges and then decreases as the fluid density continues to decrease. This steady increase occurs during coexistence between solid and fluid phases, in which particles are rapidly fluctuating between the two phases, and the rate is controlled by the rate at which the area fraction decreases. The subsequent decrease follows the further density reduction of the fluid phase. That the decrease begins before a plateau in $\langle |\Delta|\Psi_{6,j}| \rangle$ is obtained indicates that the area fraction is decreasing sufficiently quickly that the melting transition does not complete at a fixed fluid density because the decreasing area fraction protocol proceeds too quickly. The effects revealed by this ensemble averaged measure of fluctuations are also consistent between simulation and experiment. Together, these parameters demonstrate that the system undergoes a solid to fluid melting transition upon progressive particle disappearance that involves the expansion of disordered regions and large fluctuations in crystallinity.

2.2 Modeling observed behavior

The experimental observations, taken together, raise the question of how the repeated scanning of the electron beam under unchanging conditions fosters a sequence of both initial mobilization and coarsening and also eventual melting and thinning. Furthermore, it is striking that a Lennard-Jones potential combined with the randomized particle deletion without explicit inclusion of beam interactions is able to recapitulate experimental dynamics, despite the intrinsic, perturbative role of the electron beam. We address these points with a self-consistent model. We first treat the progressively charged single particle interaction with the IL and vacuum and, second, address the use of the Lennard-Jones interparticle interaction.

2.2.1 Single particle interaction with the ionic liquid

Particles initially decorate the IL-vacuum interface to minimize free energy.³² At equilibrium, the energy balance for interfacial adsorption can be described in terms of the interfacial pinning energy, the work required to remove a single particle from the interface:^{33,34}

$$\Delta E = \frac{\pi r^2}{\gamma_{L-V}} [\gamma_{L-S} - (\gamma_{V-S} - \gamma_{L-V})]^2, \quad (3)$$

where r represents the radius of the particle cross-section at the interface, and the set of γ_{i-j} refer to the surface tensions between the three different media—IL (L), vacuum (V), and solid particle (S) (Figure S7A). This pinning energy may be reexpressed

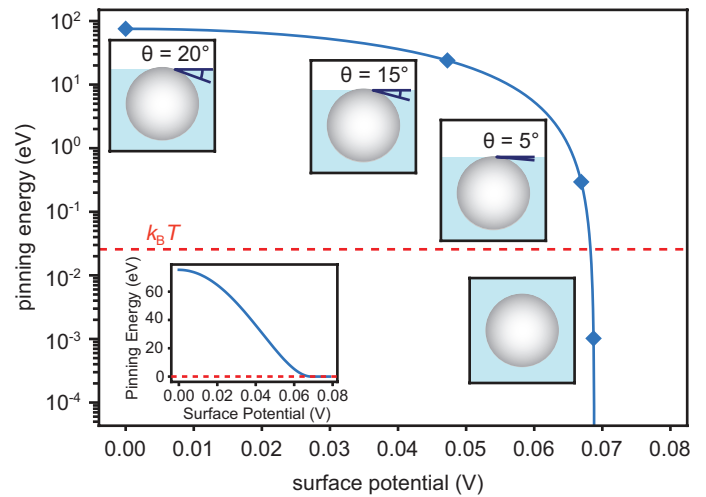


Fig. 5 Log plot of the interfacial pinning energy for a single particle as a function of surface potential. Schematics of the decreasing particle contact angle are shown at points along the curve. Particles thermally detach from the surface when the pinning energy is of order $k_B T$ (red dashed line). Inset: Plot of pinning energy versus surface potential on linear axes.

in terms of the contact angle, θ , using Young's equation, which describes the contact angle strictly in terms of the three surface tensions, as:

$$\Delta E = \pi r^2 \gamma_{L-V} (1 - \cos \theta)^2. \quad (4)$$

The initial contact angle, calculated from known values for the media employed, and assuming zero charge on the silica particle, should be in the range of $\theta_0 = 15\text{-}20^\circ$.²¹ In this range, interfacial pinning energies are of order $10^3 \times$ the room temperature thermal energy ($k_B T$); Only when the particles are nearly completely wetted with $\theta < 2^\circ$ does interfacial pinning become comparable to the thermal energy. We elaborate on this treatment in the Supplementary text and Figure S7A-B. SEM imaging causes each particle to charge over time, modifying the balance of interfacial interactions. In contrast with suspensions in polar solvents, where particle charging modulates electrostatic interparticle interactions,^{35,36} in an IL particle charging primarily influences screening by the solvent. More specifically, progressive particle charging stabilizes the particle-IL interaction, γ_{L-S} , and destabilizes the particle-vacuum interaction, γ_{V-S} , reducing the particle affinity for the interface and lowering the particle-IL contact angle. Equivalently viewed from the perspective of solvation, as a function of charging, more and more IL counterions must be recruited to the particle surface to screen a particle's electric field, forming a capacitive double-layer.³⁷ The counterions progressively cover the particle surface until none of the silica is exposed to vacuum, and the particle is no longer pinned to the interface. This electrowetting behavior as a function of the surface potential, ψ , is described by the Young-Lippmann equation:^{38,39}

$$\Delta \cos \theta = \frac{\epsilon_0 \epsilon_r}{\lambda_D \gamma_{L-V}} \frac{(2k_B T)^2}{e^2} \left(\cosh\left(\frac{e\psi}{k_B T}\right) - 1 \right). \quad (5)$$

Here, ϵ_0 and ϵ_r are the vacuum permittivity constant and the IL

dielectric constant, respectively, λ_D is the IL Debye length, and e is the electron charge. There is debate over the particular choice of Debye length, and we select a value of 3 to reflect the capacitive screening length scale of the double-layer.^{40–42} By parameterizing the contact angle in Equations 4 and 5, we can directly assess the particle pinning energy as a function of surface potential as plotted in Figure 5. When the pinning energy is comparable to the thermal energy, the particle is able to desorb from the interface and be completely solvated by the IL. While the inset of Figure 5 show the same model plotted on a linear scale, it is apparent on either plot that, at the point of unpinning, modest fluctuations of the surface pinning energy over a very narrow range in surface potential (~ 70 mV) are able to destabilize the pinning interaction. (See Table S1 for all model parameters used to obtain this estimate of the surface potential.) As the contact angle decreases, interfacial fluctuations enable more facile 2D translational motion and eventual unpinning.^{43,44} Particle unpinning induces a solid-to-liquid phase transition in the 2D monolayer expressly by lowering the area fraction and corresponding 2D pressure. Given that the charging protocol for any given particle has a very low duty cycle while the electron beam scans the entire remainder of the field of view, we expect that the particles remain in equilibrium with their surrounding media though not necessarily with one another. In fact, the steeper changes in the simulation data seen between 100 and 200 s in the ordered and disordered particle fractions in Figure 4E likely arise because the Lennard-Jones simulation is more readily able to equilibrate than the interparticle interactions allow for in the experiment.

2.2.2 Interparticle interactions

Having provided an explanation for particle unpinning, we turn to describing the effect of progressive particle charging on interparticle interactions within the monolayer at the interface and further assess the validity and limitations of employing the Lennard Jones model. We suspect that Coulomb interactions are well screened by the IL and that they do not directly influence interparticle interactions.^{45,46} This assumption is supported by our determination that a Lennard-Jones potential with fixed parameters is able to recapitulate the interparticle dynamics in the monolayer once they are set in motion and while they continue to be further charged by the electron beam. This recapitulation is illustrated by the good agreement in Figure 4 between experimental and simulation data. The exception noted above—that before any substantial decrease in area fraction the experimental system is not at equilibrium—indicates that additional interparticle interactions or particle-interface interactions likely do exist. *structured colloids* The recovery of experiment-model agreement once the area fraction more precipitously decreases suggests that the area fraction changes dominate the simulation dynamics during this stage. To gain insight into the physical origin of the interaction between particles within the monolayer, we varied the interaction potential. Using a Weeks-Chandler-Anderson (WCA) potential⁴⁷ in order to eliminate attractive interparticle interactions and approximate a hard spheres model yielded generally faster dynamics and qualitative disagreement with experimental data (Figure S8). Since our model for particle unpinning suggests that the IL

effectively screens the progressively increasing charge on the particles, the interparticle potential appears to be rather invariant to charge build-up from repeated irradiation after the initial stage in which the particles are made to fluctuate. The exact nature of the interparticle forces presents an exciting avenue of future research. The strong qualitative agreement of the Lennard-Jones simulation with the experimental data suggests that van der Waals forces may be at play. Prior work, however, suggests that lateral capillary forces may also be present in our system.^{48–50} This effective attractive interaction would be mediated by a combination of charging and the dielectric mismatch across the IL-vacuum interface.⁵¹

2.3 Control over the particle charging rate

The model that we have established to explain the observed behavior of the electron beam-induced monolayer evolution, diagrammed in Figure 6A, is further supported by additional studies of the same monolayer system performed at progressively lower accelerating voltages. As a prelude to these experiments we describe the anticipated effect of lowering the accelerating voltage from the 5 kV that is used in all previously described experiments. Standard Monte Carlo simulations⁵² of our electron beam irradiation of a silica particle at 2, 3, 4, and 5 kV yield the distributions of energy deposited in the particle shown in Figure 6B. The same scale is used in all four simulation results. We take the energy deposited to correlate to the inelastic scattering events of the beam's electrons in the particle, which should also be a good proxy for the extent of charging. Altogether, particle charging reflects the balance of trapped primary electrons and of holes generated concomitantly with secondary electrons (Figure S9). Because they are insulating we expect the charge distribution created in the particles by the electron beam to persist because there is neither a mechanism to transport charge within the particle nor to carry the charge out of the particle. The simulation results show that at 5 kV, the highest accelerating voltage explored, energy is deposited relatively uniformly over an entire particle. Decreasing the beam voltage decreases the size of the scattering volume to the point where at 2 kV it is concentrated near the surface in the top third of the particle. Additionally, because the scattering volume is largely confined to the particle, relatively few beam electrons are transmitted through the particle, leading to higher rates of energy deposition and charging. Although the panels in Figure 6B show asymmetric energy distributions, we suspect that repeated irradiation might distribute charge in the particle with spherical symmetry if the particle were able to fluctuate and hence rotate or revolve in place. This action, which could occur at other accelerating voltages as well, would serve to minimize the free energy of the particle by allowing the charge initially deposited at the vacuum interface to be best solvated by the IL.

Returning to the experimental data, Figure 6C shows the corresponding time series of SEM images at 2 kV, 3 kV, 4 kV, and 5 kV, keeping the current constant at ~ 100 pA and the scanning time per frame constant at 2.6 s. In all cases, the particles ultimately disappear into the IL. The particles disappear at an earlier time at lower accelerating voltages than at higher accelerating

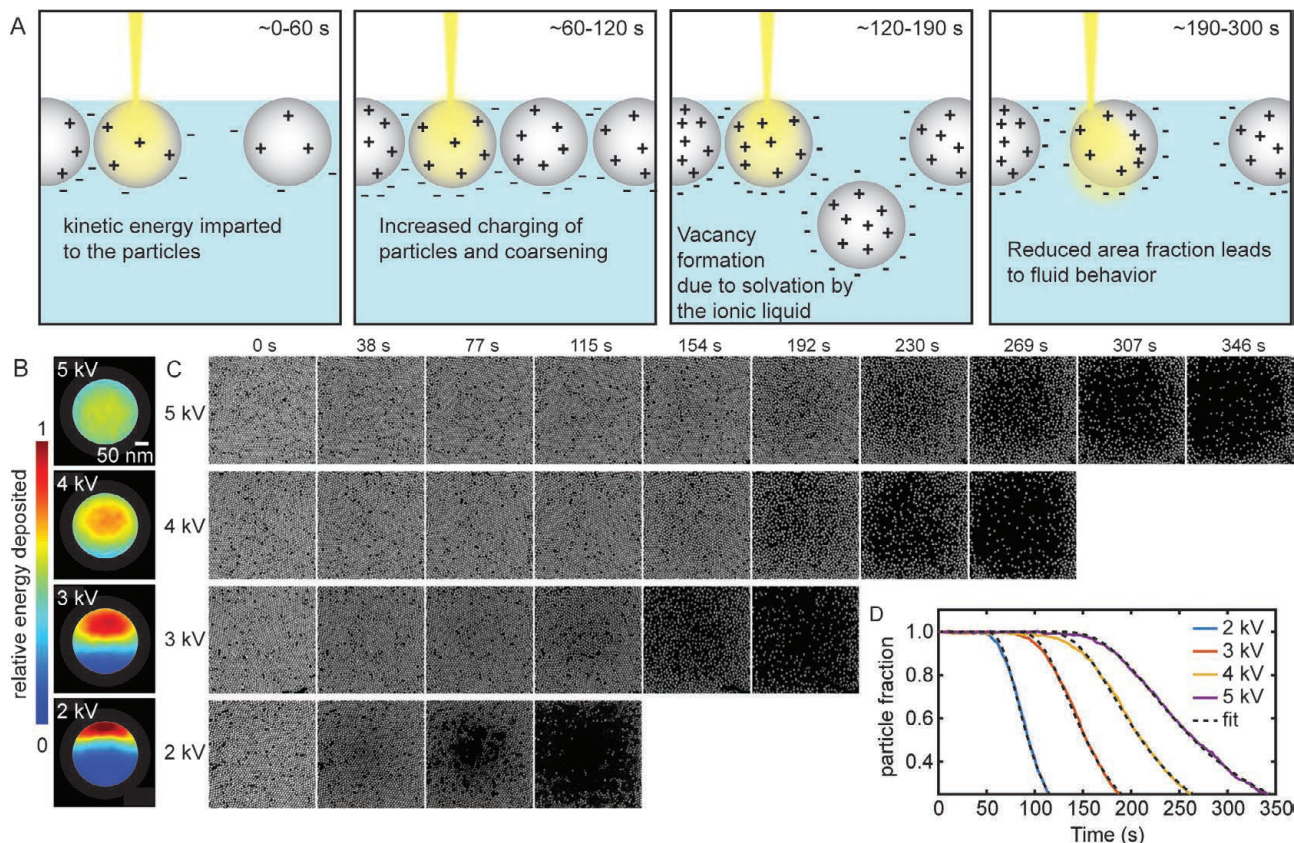


Fig. 6 Mechanism of particle coarsening and sinking and accelerating voltage dependence. (A) Schematic showing how the electron beam charges the silica particles to induce coarsening and then sinking into the IL layer. While the contact angle decreases with charging, it is not exaggerated, in order to most accurately depict the true scenario. (B) Simulated energy deposition of primarily electrons in a 300 nm silica sphere by electron beam as a function of accelerating voltage. The same energy scale is employed for all four cases to facilitate direct comparison. (C) Time series of SEM images showing the accelerating voltage dependence of coarsening and sinking. (D) Particle fraction as a function of time for 2-5 kV.

voltages. The solid curves in Figure 6D shows the fraction of particles remaining at the interface as a function of time for 2-5 kV. Not only do the particles disappear from the monolayer sooner at lower accelerating voltages, but the distribution of disappearance times is also narrower (Figure S10). The median sinking time (t_s in Table S2) in Figure 6D scales roughly linearly with accelerating voltage, starting from ~ 93 s at 2 kV and going up to ~ 265 s at 5 kV. The unpinning model proposed above is consistent with this trend in accelerating voltage, presumably due to a fairly linear relationship between the surface potential that results from a particular accelerating voltage and the time taken for each particle to achieve that potential. We expect that the lower electron transmission, and consequently higher charging rate, at lower accelerating voltage increases the surface potential to recruit more countercharge from the IL and to lower the contact angle more quickly, promoting unpinning of the particles from the interface. Interestingly, the number of particles remaining at the interface as a function of time in addition to its accelerating voltage dependence can be captured with a phenomenological model in which particle charge is normally distributed with a standard deviation that linearly increases as a function of time (dotted curves in Figure 6D). The trajectories measured at the higher accelerating voltages that have later average disappearance times feature intrinsically broader distributions, as variation in the charge increases

over time. Discussion of the fit is elaborated in the Supplementary Information (see fit parameters in Table S2). The finite slope of the curves indicates a broad distribution of unpinning times, which we attribute to variations in charging rate associated with local particle environment, the effective dose across the imaging field of view, and variations in electron beam exposure as particles fluctuate at the liquid-vacuum interface. Beyond these hypotheses to explain the particle disappearance profiles, the accelerating voltage dependence shows that changes to the properties of the self-assembled particles. The longer wait time before particles begin to disappear at higher accelerating voltages allows more time to first reorganize, such that ordering is more substantial than at lower accelerating voltages.

3 Conclusion

We have examined and elucidated the mechanisms for charging-driven coarsening and melting of an initially static nanoparticle colloidal monolayer at an IL-vacuum interface by prompting and probing with a scanning electron beam. The coarsening of the initially defected 2D hexagonal lattice occurs through grain interface migration and larger-scale motions such as grain rotations, which are both often underlied more microscopically by sliding dislocations. The melting transition appears to be driven by a progres-

sive decrease in the area fraction of the monolayer and is well-described by Lennard-Jones interparticle interactions, combined with a phenomenological protocol for particle deletion. We have found that particle disappearance from the monolayer, which is at the heart of the decreasing area fraction that drives the monolayer phase transition, can be explained by considering the contact angle and associated pinning energy of a particle at the interface as a function of a surface potential that increases with increased charging by the electron beam. Although the particles are initially strongly pinned at the interface, charging recruits progressively more counterions from the IL, decreasing the pinning energy. Full wetting occurs when the particle surface potential is substantial enough to weaken the pinning interaction to the point where it is comparable to thermal fluctuations. Exploring the dynamics as a function of accelerating voltage and concomitant particle charge density profiles further supports our model of the phenomenon by showing how the characteristic time scales in the dynamics can be varied as a function of charging rate. We have thus elucidated a new mechanism by which to externally drive a colloidal system, with the exciting possibility to control behaviors without having to change the intrinsic properties of the particles being studied.

Looking forward, combining the independently rich parameter spaces of colloidal particles, ILs, and electron beam attributes stands to generate a wide range of new driven, emergent phenomena. First, in addition to exploring particle shapes, symmetries, and sizes, different electronic properties should strongly impact their interaction with the IL, with the electron beam, and therefore also with one another. For example, Janus-like insulating/conducting particles and other anisotropic forms should yield orientation-dependent, bond-like interactions, more closely approximating molecular systems, and should access new interaction regimes, some of which will be strongly dominated by electrostatics.⁵³ Second, since IL properties strongly affect the tendency toward monolayer assembly, exploring their composition could present an interesting platform to probe the frontiers of lateral capillary forces, as a high monolayer density could lead to many body effects theorized to strongly enhance capillary forces.⁵⁴ Third, manipulating electron beam current, accelerating voltage, and X-Y scanning trajectories could readily be used to controllably perturb colloidal systems in more sophisticated ways as well, including tuning the accelerating voltage to preserve particle charge neutrality. This control could prepare initial or dynamically modulated conditions with specific surface pressures by corralling particles, could be used to selectively remove particles, by analogy with optical tweezers, and could generate more sophisticated time-dependent interparticle interactions. Finally, further decreasing particle sizes beneath what are typically employed in colloidal experiments while interrogating them non-perturbatively is still important to pursue in order to more closely approximate the interactions of molecular or atomic systems and to access shorter dynamical time scales. Developing other schemes that leverage electron microscopy resolution but that can non-invasively study more nanoscopic systems will be a valuable complement to electron beam-induced characterizations.^{55,56} Non-invasive imaging approaches offer the chance to compare the dynamics of a given nanoparticle system in the ab-

sence of charging to the same system upon direct electron beam irradiation.

Experimental

Colloid Monolayer Preparation

The colloid monolayer was prepared by mixing 20 μL stock colloid solution (300 nm diameter 10 mg/mL in water, nanocomposix) with 20 μL 1-ethyl-3-methylimidazolium ethyl sulfate (EMIM $\text{CH}_3\text{CH}_2\text{SO}_4$, Aldrich). 10 μL of the solution was then deposited on a $\sim 1 \times 1$ cm Si substrate (Virginia Semiconductor), which was first cleaned by sonication in isopropyl alcohol (Aldrich) and acetone (Aldrich) for 5 min each. The sample was then placed in a vacuum chamber at 40 mtorr for 20 min to evaporate the water.

Scanning Electron Microscopy

Colloidal monolayers on Si were grounded with a copper clip and loaded into a Zeiss Gemini SUPRA 55 S2 scanning electron microscope (SEM). Movies of electron beam-induced colloid motion were acquired using the in-lens detector with a 5 kV accelerating voltage (unless otherwise specified), a beam current of ~ 100 pA, a scan rate of 0.01 s/line, and a field-of-view of approximately 12×12 μm ($8000\times$ magnification, 256×256 pixels). Movies of colloid dynamics were acquired using custom python software⁵⁷ and saved as HDF5 files.

Single Particle Tracking

Single particle tracking of the electron beam-induced colloidal monolayer dynamics was performed using the TrackMate plugin⁵⁸ of ImageJ.⁵⁹ HDF5 files were imported using the HDF5 Plugin for ImageJ and Fiji. The movies were binarized using the Otsu method and then loaded into TrackMate. The parameters for the single particle tracking using TrackMate were a blob diameter of 0.3 μm , a threshold of 0.1, a maximum linking distance of 0.250 μm , a gap-closing maximum distance of 0.1 μm , and a gap-closing frame gap of 1.

Analysis of Single Particle Tracking Outputs

Output files from TrackMate were loaded into a custom Matlab code. The custom Matlab code was used to compute the bond order parameter magnitude ($|\Psi_{6,j}|$), bond order parameter phase (θ_j), and instantaneous 2D Lindemann parameter (LP_j) for each particle j . Movies and times series of frames were generated with false colors that indicate $|\Psi_{6,j}|$, θ_j , LP_j , and $\Delta|\Psi_{6,j}|$. Disordered and crystalline particles were distinguished by $|\Psi_{6,j}| < 0.7$ and $|\Psi_{6,j}| > 0.7$, respectively, and plotted as blue and yellow false-colored particles, respectively. A threshold value of 0.7 was chosen to distinguish crystalline and disordered particles by plotting the distribution of $|\Psi_{6,j}|$ for the initial frames and finding the midpoint between the two subpopulations.

Monte Carlo Simulations of Electron Trajectories

Monte Carlo simulations of electron energy deposition were performed using the CASINO v3.2 program.⁵² For each simulation,

the 3D trajectories of 10,000 electrons were simulated. Simulations were carried out at accelerating voltages ranging from 2.0 kV to 5.0 kV. For each accelerating voltage, energy deposited in a 300 nm-diameter sphere with a density of $\sim 2.5 \text{ g/cm}^3$ was calculated and integrated along one axis in order to represent the results in Figures 6B and S9.

Molecular Dynamics Simulations

The simulations presented in the main text were performed using LAMMPS software.⁶⁰ The initial conditions were extracted from the experimental data. Each simulation was carried out in two dimensions, and periodic boundary conditions were used. All simulations were carried out in reduced Lennard-Jones units with a fundamental unit of time given by $\tau_{LJ} = \frac{\sigma^2}{D}$, where σ and D are given in non-reduced units, and are the diameter and diffusion coefficient for an isolated colloidal particle, respectively. The diffusion coefficient was calculated using the Stokes-Einstein relation for a particle in an ionic-liquid (IL) solution⁶¹. Each simulation is carried out with a time step of $\delta t = 0.001$, and we use a Langevin thermostat with a relaxation time of $200\delta t$. Each simulation was initialized by using the experimental data frame at 100 seconds of experimental time, as this is the time required for the electron beam to drive the colloidal particles into a well defined state of motion.

Conflicts of interest

There are no conflicts to declare.

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Author Contributions

C.G.B., J.G.R., J.W.K. and N.S.G. wrote the manuscript. C.G.B. and N.S.G. conceptualized the experiment. C.G.B. performed the electron microscopy experiments and single particle tracking analysis. C.G.B. and J.W.K. performed the molecular dynamics simulations. J.G.R. developed the unpinning model of the electron beam-irradiated colloids.

Notes and references

1 A. J. Liu and S. R. Nagel, *Annu. Rev. Condens. Matter Phys.*, 2010, **1**, 347–369.

- 2 K. L. Kohlstedt and S. C. Glotzer, *Phys. Rev. E*, 2013, **87**, 032305.
- 3 T. D. Edwards, Y. Yang, D. J. Beltran-Villegas and M. A. Bevan, *Sci. Rep.*, 2014, **4**, 1–8.
- 4 V. N. Manoharan, *Science*, 2015, **349**, 6251.
- 5 F. A. Lavergne, D. G. Aarts and R. P. Dullens, *Phys. Rev. X*, 2017, **7**, 041064.
- 6 M. Girard, S. Wang, J. S. Du, A. Das, Z. Huang, V. P. Dravid, B. Lee, C. A. Mirkin and M. O. d. l. Cruz, *Science*, 2019, **364**, 1174–1178.
- 7 V. Soni, E. S. Bililign, S. Magkiriadou, S. Sacanna, D. Bartolo, M. J. Shelley and W. T. M. Irvine, *Nat. Phys.*, 2019, **15**, 1188–1194.
- 8 T. Hueckel, G. M. Hocky, J. Palacci and S. Sacanna, *Nature*, 2020, **580**, 487–490.
- 9 E. R. Weeks, J. C. Crocker, A. C. Levitt, A. Schofield and D. A. Weitz, *Science*, 2000, **287**, 627–631.
- 10 A. M. Alsayed, M. F. Islam, J. Zhang, P. J. Collings and A. G. Yodh, *Science*, 2005, **309**, 1207–1210.
- 11 P. Lipowsky, M. J. Bowick, J. H. Meinke, D. R. Nelson and A. R. Bausch, *Nat. Mater.*, 2005, **4**, 407–411.
- 12 B. Li, F. Wang, D. Zhou, Y. Peng, R. Ni and Y. Han, *Nature*, 2016, **531**, 485–488.
- 13 B. Li, D. Zhou and Y. Han, *Nat. Rev. Mater.*, 2016, **1**, 1–13.
- 14 A. L. Thorneywork, J. L. Abbott, D. G. Aarts and R. P. Dullens, *Phys. Rev. Lett.*, 2017, **118**, 158001.
- 15 X. Ye, M. R. Jones, L. B. Frechette, Q. Chen, A. S. Powers, P. Ercius, G. Dunn, G. M. Rotskoff, S. C. Nguyen, V. P. Adiga, A. Zettl, E. Rabani, P. L. Geissler and A. P. Alivisatos, *Science*, 2016, **354**, 874–877.
- 16 B. C. Bayer, R. Kaindl, M. Reza Ahmadvour Monazam, T. Susi, J. Kotakoski, T. Gupta, D. Eder, W. Waldhauser and J. C. Meyer, *ACS Nano*, 2018, **12**, 8758–8769.
- 17 J. Kim, Z. Ou, M. R. Jones, X. Song and Q. Chen, *Nat. Commun.*, 2017, **8**, 1–10.
- 18 B. Luo, J. W. Smith, Z. Ou and Q. Chen, *Acc. Chem. Res.*, 2017, **50**, 1125–1133.
- 19 C. Su, M. Tripathi, Q.-B. Yan, Z. Wang, Z. Zhang, C. Hofer, H. Wang, L. Basile, G. Su, M. Dong, J. C. Meyer, J. Kotakoski, J. Kong, J.-C. Idrobo, T. Susi and J. Li, *Sci. Adv.*, 2019, **5**, eaav2252.
- 20 P. Y. Kim, A. E. Ribbe, T. P. Russell and D. A. Hoagland, *ACS Nano*, 2016, **10**, 6257–6264.
- 21 P. Y. Kim, Y. Gao, Y. Chai, P. D. Ashby, A. E. Ribbe, D. A. Hoagland and T. P. Russell, *ACS Nano*, 2019, **13**, 3075–3082.
- 22 T. Torimoto, T. Tsuda, K.-i. Okazaki and S. Kuwabata, *Adv. Mater.*, 2010, **22**, 1196–1221.
- 23 H. Zhang, K. Dasbiswas, N. B. Ludwig, G. Han, B. Lee, S. Vaikuntanathan and D. V. Talapin, *Nature*, 2017, **542**, 328–331.
- 24 D. S. Frost, E. M. Nofen and L. L. Dai, *Adv. Colloid Interface Sci.*, 2014, **206**, 92–105.
- 25 P. M. Chaikin and T. C. Lubensky, *Principles of Condensed Matter Physics*, Cambridge University Press, 1995.

- 26 V. M. Bedanov, G. V. Gadiyak and Y. E. Lozovik, *Phys. Lett. A*, 1985, **109**, 289–291.
- 27 K. Zahn, R. Lenke and G. Maret, *Phys. Rev. Lett.*, 1999, **82**, 2721–2724.
- 28 C. P. Kelleher, R. E. Guerra, A. D. Hollingsworth and P. M. Chaikin, *Physical Review E*, 2017, **95**, 022602.
- 29 Y. Peng, F. Wang, Z. Wang, A. M. Alsayed, Z. Zhang, A. G. Yodh and Y. Han, *Nat. Mater.*, 2015, **14**, 101–108.
- 30 Y. Peng, W. Li, F. Wang, T. Still, A. G. Yodh and Y. Han, *Nat. Commun.*, 2017, **8**, 1–12.
- 31 B. Smit and D. Frenkel, *J. Chem. Phys.*, 1998, **94**, 5663.
- 32 P. Pieranski, *Phys. Rev. Lett.*, 1980, **45**, 569–572.
- 33 D. F. Williams and J. C. Berg, *J. Colloid Interface Sci.*, 1992, **152**, 218–229.
- 34 A. Böker, J. He, T. Emrick and T. P. Russell, *Soft Matter*, 2007, **3**, 1231–1248.
- 35 F. Reincke, W. K. Kegel, H. Zhang, M. Nolte, D. Wang, D. Vanmaekelbergh and H. Möhwald, *Phys. Chem. Chem. Phys.*, 2006, **8**, 3828–3835.
- 36 M. Luo, G. K. Olivier and J. Frechette, *Soft Matter*, 2012, **8**, 11923–11932.
- 37 R. M. Lynden-Bell, A. I. Frolov and M. V. Fedorov, *Phys. Chem. Chem. Phys.*, 2012, **14**, 2693–2701.
- 38 S. Millefiorini, A. H. Tkaczyk, R. Sedev, J. Efthimiadis and J. Ralston, *J. Am. Chem. Soc.*, 2006, **128**, 3098–3101.
- 39 H. Horiuchi, A. Nikolov and D. T. Wasan, *J. Colloid Interface Sci.*, 2012, **385**, 218–224.
- 40 H. Weingärtner, *Angew. Chem. Int.*, 2008, **47**, 654–670.
- 41 S. Perkin, L. Crowhurst, H. Niedermeyer, T. Welton, A. M. Smith and N. N. Gosvami, *Chemical Communications*, 2011, **47**, 6572.
- 42 M. A. Gebbie, A. M. Smith, H. A. Dobbs, A. A. Lee, G. G. Warr, X. Banquy, M. Valtiner, M. W. Rutland, J. N. Israelachvili, S. Perkin and R. Atkin, *Chem. Commun.*, 2017, **53**, 1214–1224.
- 43 G. Boniello, C. Blanc, D. Fedorenko, M. Medfai, N. B. Mbarek, M. In, M. Gross, A. Stocco and M. Nobili, *Nat. Mater.*, 2015, **14**, 908–911.
- 44 X. Hua, M. A. Bevan and J. Frechette, *Langmuir*, 2016, **32**, 11341–11352.
- 45 K. Ueno, A. Inaba, M. Kondoh and M. Watanabe, *Langmuir*, 2008, **24**, 5253–5259.
- 46 Z. He and P. Alexandridis, *Phys. Chem. Chem. Phys.*, 2015, **17**, 18238–18261.
- 47 J. D. Weeks, D. Chandler and H. C. Andersen, *J. Chem. Phys.*, 1971, **54**, 5237–5247.
- 48 P. A. Kralchevsky and K. Nagayama, *Adv. Colloid Interface Sci.*, 2000, **85**, 145–192.
- 49 M. G. Nikolaidis, A. R. Bausch, M. F. Hsu, A. D. Dinsmore, M. P. Brenner, C. Gay and D. A. Weitz, *Nature*, 2002, **420**, 299–301.
- 50 M. Oettel, A. Domínguez and S. Dietrich, *Phys. Rev. E*, 2005, **71**, 051401.
- 51 D. M. Kaz, R. McGorty, M. Mani, M. P. Brenner and V. N. Manoharan, *Nat. Mater.*, 2012, **11**, 138–142.
- 52 H. Demers, N. Poirier-Demers, A. R. Couture, D. Joly, M. Guilmann, N. d. Jonge and D. Drouin, *Scanning*, 2011, **33**, 135–146.
- 53 J. Zhang, B. A. Grzybowski and S. Granick, *Langmuir*, 2017, **33**, 6964–6977.
- 54 V. M. Pergamenschchik, *Phys. Rev. E*, 2009, **79**, 011407.
- 55 C. G. Bischak, C. L. Hetherington, Z. Wang, J. T. Precht, D. M. Kaz, D. G. Schlom and N. S. Ginsberg, *Nano Lett.*, 2015, **15**, 3383–3390.
- 56 C. G. Bischak, R. B. Wai, C. Cherqui, J. A. Busche, S. C. Quillin, C. L. Hetherington, Z. Wang, C. D. Aiello, D. G. Schlom, S. Aloni, D. F. Ogletree, D. J. Masiello and N. S. Ginsberg, *ACS Nano*, 2017, **11**, 10583–10590.
- 57 C. G. Bischak, E. M. Sanehira, J. T. Precht, J. M. Luther and N. S. Ginsberg, *Nano Lett.*, 2015, **15**, 4799–4807.
- 58 J.-Y. Tinevez, N. Perry, J. Schindelin, G. M. Hoopes, G. D. Reynolds, E. Laplantine, S. Y. Bednarek, S. L. Shorte and K. W. Eliceiri, *Methods*, 2017, **115**, 80–90.
- 59 J. Schindelin, I. Arganda-Carreras, E. Frise, V. Kaynig, M. Longair, T. Pietzsch, S. Preibisch, C. Rueden, S. Saalfeld, B. Schmid, J.-Y. Tinevez, D. J. White, V. Hartenstein, K. Eliceiri, P. Tomancak and A. Cardona, *Nat. Methods*, 2012, **9**, 676–682.
- 60 S. Plimpton, *J. Comput. Phys.*, 1995, **117**, 1–19.
- 61 P. F. Requejo, E. J. González, E. A. Macedo and A. Domínguez, *J. Chem. Thermodyn.*, 2014, **74**, 193–200.

