



Advanced Manufacturing for Electrosynthesis of Fuels and Chemicals from CO₂

Journal:	<i>Energy & Environmental Science</i>
Manuscript ID	EE-ART-11-2020-003679.R1
Article Type:	Paper
Date Submitted by the Author:	26-Jan-2021
Complete List of Authors:	<p>Corral, Daniel; Stanford University, Chemical Engineering; Lawrence Livermore National Laboratory, Materials Engineering Division</p> <p>Feaster, Jeremy; Lawrence Livermore National Laboratory, Materials Science Division</p> <p>Sobhani, Sadaf; Lawrence Livermore National Laboratory, Computational Engineering Division</p> <p>DeOtte, Joshua; LLNL, Center for Micro and Nano Technology</p> <p>Lee, Dong Un; Stanford University, Chemical Engineering</p> <p>Wong, Andrew; Lawrence Livermore National Laboratory, Materials Engineering Division</p> <p>Hamilton, Julie; Lawrence Livermore National Laboratory, Materials Engineering Division</p> <p>Beck, Victor; Lawrence Livermore National Laboratory, Computational Engineering Division</p> <p>Sarkar, Amitava; Total SA, EP Research & Technology; Lawrence Livermore National Laboratory, Materials Science Division; Stanford University, Chemical Engineering</p> <p>Hahn, Christopher; Stanford University, Chemical Engineering; SLAC National Accelerator Laboratory, SUNCAT Center for Interface Science and Catalysis</p> <p>Jaramillo, Thomas; Stanford University, Assistant Professor of Chemical Engineering</p> <p>Baker, Sarah; Lawrence Livermore National Lab,</p> <p>Duoss, Eric; Lawrence Livermore National Laboratory, Materials Engineering Division</p>

1 **Advanced Manufacturing for Electrosynthesis of Fuels** 2 **and Chemicals from CO₂**

3 Daniel Corral^{1,2,†}, Jeremy T. Feaster^{3,†}, Sadaf Sobhani⁴, Joshua R. DeOtte¹, Dong Un Lee²,
4 Andrew A. Wong¹, Julie Hamilton¹, Victor A. Beck⁴, Amitava Sarkar^{2,3,5}, Christopher Hahn^{6,*},
5 Thomas F. Jaramillo^{2,6,*}, Sarah E. Baker^{3,*}, and Eric B. Duoss^{1,*}

6 ¹ Materials Engineering Division, Lawrence Livermore National Laboratory, Livermore,
7 CA 94550, United States

8 ² SUNCAT Center for Interface Science and Catalysis, Department of Chemical
9 Engineering, Stanford University, Stanford, CA 94305, USA

10 ³ Materials Science Division, Lawrence Livermore National Laboratory, Livermore, CA
11 94550, United States

12 ⁴ Computational Engineering Division, Lawrence Livermore National Laboratory,
13 Livermore, CA 94550, United States

14 ⁵ Total EP Research & Technology USA, LLC., Houston, TX 77002, USA

15 ⁶ SUNCAT Center for Interface Science and Catalysis, SLAC National Accelerator
16 Laboratory, Menlo Park, CA 94025, USA

17 † These authors contributed equally to this work.

18 *Correspondence: chahn@slac.stanford.edu, jaramillo@stanford.edu, baker74@llnl.gov,
19 and duoss1@llnl.gov □

20

21 **Abstract**

22 Advanced manufacturing (AM) represents an appealing approach for creating novel
23 electrochemical systems for chemicals synthesis. In this work, we demonstrate AM for rapid
24 development and testing for improved performance for the carbon dioxide reduction reaction
25 across an evolution of vapor-fed reactor designs. In our final design, we observe activation- and
26 mixed- control regimes for a variety of operating conditions via inlet CO₂ flow rate and
27 electrochemical potential. Furthermore, we define a dimensionless number (Da) to identify mass
28 transport regimes by exploring the impact of hypothesized underlying mass transport mechanisms,
29 including consumption of CO₂ via OH⁻, increased local temperatures, and partial penetration of
30 electrolyte into gas diffusion layer. The accelerated pace of reactor design and development led to
31 high geometric current densities (500 mA cm⁻²), heightened selectivity (85.5% FE C₂₊ products),
32 and increased carbon dioxide conversion (16.6%) and cathodic energy efficiency (56.3% CO₂R).
33 Using AM vapor-fed reactors, we attain high ethylene (3.67%) and record ethanol (3.66%) yields
34 compared to the literature. This work underscores the promise of AM for accelerating reactor
35 design, understanding of governing phenomena, and improving the performance of catalytic
36 systems.

37 **Broader Impact**

38 Advanced manufacturing as a method to create novel electrochemical reactors opens the
39 door to many different energy technologies ahead. In combination with CO₂ electroreduction,
40 excess renewable energy can effectively be stored as fuels and additives, as well as non-fossil
41 derived consumer and industrial chemicals (e.g. monomers for polymerization and constructions
42 materials) for longer-term sequestration of CO₂ out of environment. Reactor design and
43 manufacturing is central to achieving progress toward these goals; understanding of the mass
44 transport phenomena occurring within the reactor must be developed to build more efficient and
45 effective systems. The high activity, selectivity, cathodic energy efficiency, CO₂ single-pass
46 conversion and product yields achieved in this work were accomplished with a generic Cu catalyst
47 in bulk neutral pH. This highlights the potential of this methodology in design optimization and
48 enhanced system performance when sophisticated catalysts and/or reaction systems are employed.
49 AM reactors represent a significant step towards de-risking scale-up and optimization of this game-
50 changing technology while developing design metrics that can promote commercial feasibility.
51 This work also validates that cost-effective AM techniques can be utilized for rapid production of
52 modular system components with high precision and reproducibility that are necessary for future
53 industrial deployment.

54

55

56 **Keywords**

57 Advanced manufacturing, CO₂ reduction, 3D printing, electrocatalysis, reactor design,
58 ethylene, ethanol

59 Introduction

60 The accelerating need to mitigate global emissions of CO₂ has driven significant
61 advancements in designing systems and strategies that transition towards a sustainable future.¹⁻³
62 However, a growing global population continues to rely on fossil-derived sources for power,
63 chemicals, agriculture, and transportation sectors.⁴ Converting this CO₂ to fuels and chemical
64 feedstocks can be an effective way to reduce the CO₂ footprint of many conventional chemical
65 processes, thus promoting a circular carbon-economy.⁵ Considering the cheap (~ \$0.02 kWh⁻¹) and
66 sometimes free renewable energy generated in certain parts of the world, such energy can be stored
67 by converting CO₂ to synthesis gas, future energy carriers (e.g. fuels and additives), and monomers
68 for polymerization. Thus, electrochemical CO₂ reduction (CO₂R) to produce non-fossil sourced
69 chemical products can be a significant milestone in decarbonizing the chemicals industry.

70 While there are several challenges associated with scaling up CO₂R, much progress has
71 been made to better understand a variety of complex phenomena and coexisting processes.
72 Aqueous-phase CO₂R, performed in liquid-fed reactors (LFRs) leveraging electrochemical⁶⁻⁸ as
73 well as photocatalytic^{9,10} driving forces, served as the primary platform for investigating selectivity
74 trends and discovering new materials.^{11,12} Furthermore, studies in LFRs elucidated the effects of
75 various intermediates, including *COOH^{8,13-15}, *CO^{16,17}, and even *C₂O₄¹⁸ in light driven
76 conditions, which aided in our understanding of reaction mechanisms. The transition from LFRs
77 to vapor-fed reactors (VFRs) utilizing gas diffusion electrodes (GDEs) yielded higher performance
78 metrics by minimizing the distance between the electrode and bulk vapor CO₂ stream.¹⁹ Previous
79 studies achieved high activity and selectivity towards the production of CO, HCOO⁻, C₂H₄, and
80 C₂H₅OH by capitalizing on the increased complexity and tunability of GDEs.²⁰ Despite recent

81 advances, it is unclear whether the understanding of kinetic mechanisms and mass transport
82 phenomena in LFRs translates to industrially relevant current densities observed in VFRs.

83 To date, published reports on electrochemical reactors for CO₂ reduction have all focused
84 on components that rely on traditional manufacturing methods which are time- and resource
85 intensive.^{21,22} Industrial deployment of CO₂R technology to replace current production volumes of
86 the chemical and petrochemical industry will require manufacturing millions of these system
87 components with high precision and reproducibility for modular electrolyzers. Advanced
88 manufacturing (AM) is a promising approach to overcome these limitations, as it allows for rapid
89 prototyping and iteration of reactor components and parts with submicron control.^{23,24} These
90 advantages of AM have led to novel materials and devices for energy storage,²⁵ optics,²⁶ drug
91 delivery²⁷, and carbon capture applications.²⁸ However, there have been few studies to date that
92 have employed AM for electrochemical reactor design,²⁹ and none to date for electrochemical CO₂
93 reduction.

94 For the first time to date, we demonstrate AM for fabrication of CO₂ electrolyzers with
95 increased precision. This enables the rapid prototyping of different system components with
96 conceptual design aspects and corresponding assessment of important design parameters. AM
97 VFRs showcase promising results for CO₂ reduction with an unoptimized electrode, membrane,
98 and electrolyte, including high selectivity for multi-carbon products at high current densities. We
99 identify and evaluate mass transport regimes for various flow rate and electrochemical potential
100 conditions and investigate underlying mass transport mechanisms. Consequently, we propose that
101 for a given reactor and system configuration, the use of the dimensionless Damköhler (Da) number
102 can elucidate regimes based on the proposed mechanisms. Lastly, we introduce yield as a key
103 metric for evaluating performance of CO₂ electrolyzers.

104 **Experimental methods**

105 **Reactor manufacturing and catalyst characterization**

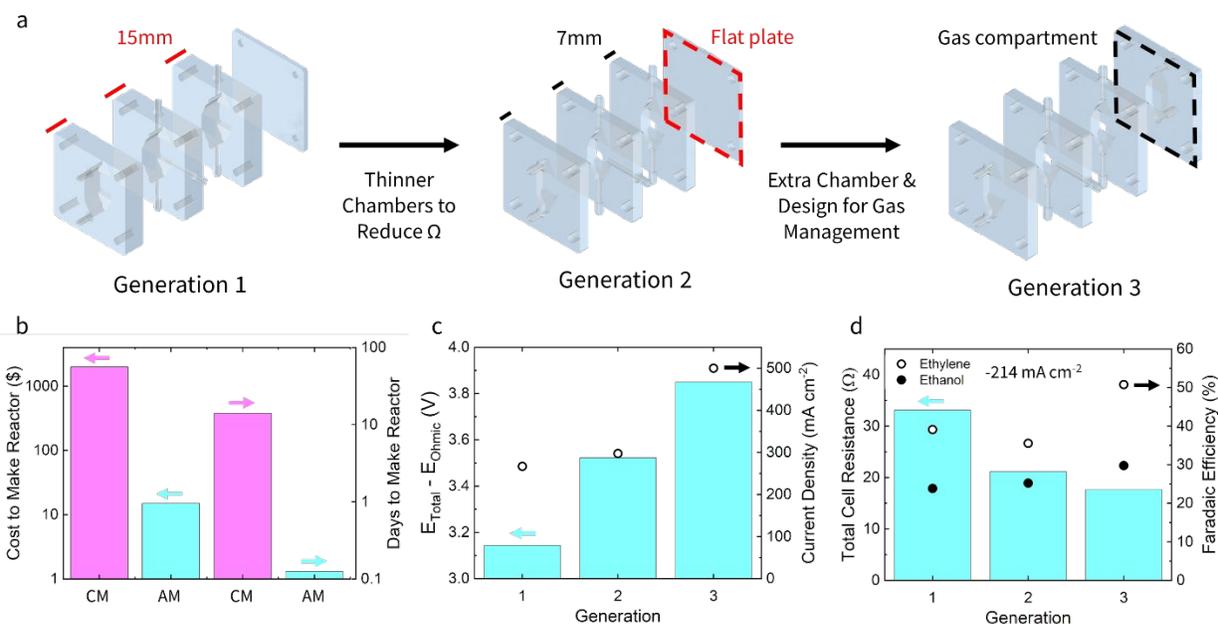
106 All experimental data was obtained from 3D-printed VFRs and components based on an informed
107 and iterative reactor design process. Electrochemical reactors (Fig. S1) and electrolyte sparging
108 chambers were first drafted using computer-aided design (CAD) software (AutoCAD Inventor).
109 Each VFR was designed with a 1.4 cm² geometric surface area for the cathode, anode, and
110 membrane windows. As gas bubbles formed on the electrodes can influence mass transport and
111 electric field uniformity within the reactor, compartments were designed with a tapered geometry
112 to direct fluid flow for enhanced bubble management (Fig. S3). All components were then printed
113 using a stereolithography (SLA) 3D printer (FormLabs Form2, 25 μm step size); all components
114 were made from a polyacrylate-based commercial resin (Formlabs). After printing, each reactor
115 component was sonicated in an IPA bath for at least 30 minutes to remove any excess or uncured
116 resin, then cured with UV light at 60°C for 60 minutes. The components were then rinsed
117 thoroughly in deionized water before use and rinsed thoroughly between experimental runs.

118 A 275 nm thick layer of Cu was deposited onto ePTFE (Sterlitech) using e-beam physical
119 vapor deposition in a rotating chamber (see Supplemental Information for full details). Scanning
120 electron microscopy (SEM) images reveals the coating of Cu catalyst on top of the ePTFE fibers
121 (Fig. S6). X-ray photoelectron spectroscopy (XPS) validated both cleaned ePTFE and Cu/ePTFE
122 samples (Fig. S7). To identify the facets post-deposition, we performed X-ray diffraction (XRD)
123 and observed Cu(111), Cu(200) and Cu(220) peaks (Fig. S8). Double-layer capacitance was used
124 to measure the electrochemically active surface area (ECSA) of the catalyst (Figure S10); the
125 ECSA measured was similar to those measured other reports.^{30–32}

126

127 **Electrochemical experiments**

128 All electrochemical experiments were carried out using a multichannel potentiostat (Bio-
129 Logic). An IrO₂ catalyst on a carbon paper electrode (Dioxide Materials) was cut and used as the
130 anode. A Low-Profile silver/silver chloride (Ag/AgCl) electrode (Pine Instruments) was used as
131 the reference electrode and connected to the catholyte compartment of the reactor. A 50µm
132 Sustainion anion exchange membrane (Dioxide Materials) was inserted between the catholyte and
133 anolyte chambers to mitigate the cross-over of impurities and products between the electrodes.
134 Both catholyte and anolyte were recirculated using external 3D-printed chambers (each with a
135 volume of 25 mL) to minimize local accumulation of products. Potentiostatic electrochemical
136 impedance spectroscopy (PEIS) was used to measure the solution resistance (1) between the
137 cathode and the reference electrode, and (2) between the cathode and the anode (Fig. S9).
138 Chronoamperometry (CA) was performed for 5 minutes at a soft overpotential of -0.75 V vs.
139 Ag/AgCl to reduce any oxide on the Cu surface. Finally, chronopotentiometry (CP) was performed
140 at a given current density; the solution resistance was measured every 3 minutes to allow for
141 correction of the voltage drop due to changes in electrolyte conductivity. The average resistance
142 and potential from each run were used for analysis. Each experiment was duplicated to ensure each
143 result was reproducible.

144 **Results and discussion**

145

146 **Fig. 1.** Advantages of AM for electrochemical reactor design. (a) Schematic representation of three generations of
 147 reactors designed for CO₂R with brief description of changes between generations. (b) Order of magnitude advantages
 148 of AM over conventional manufacturing (CM) with regards to cost and time to produce electrolyzer. (c) Potential
 149 realized at electrodes (represented as the difference of the cell potential and the potential lost to ohmic drop) and
 150 maximum current density in each generation. (d) Total cell resistance and Faradaic efficiencies of ethylene and ethanol
 151 at a fixed current density for all three generations.

152

153

154 We employed stereolithography 3D printing to develop several generations of vapor-fed

155 electrochemical systems (Fig. 1a). Using AM instead of conventional manufacturing (CM), we

156 reduced the cost and the amount of time to produce each reactor component by several orders of

157 magnitude (Fig. 1b). Several metrics informed the rapid development from Generation 1 to

158 Generation 3: the total cell resistance measured between the cathode and anode, the total cell

159 potential to operate the reactor at 214 mA cm⁻², and the current density achieved at a compliance

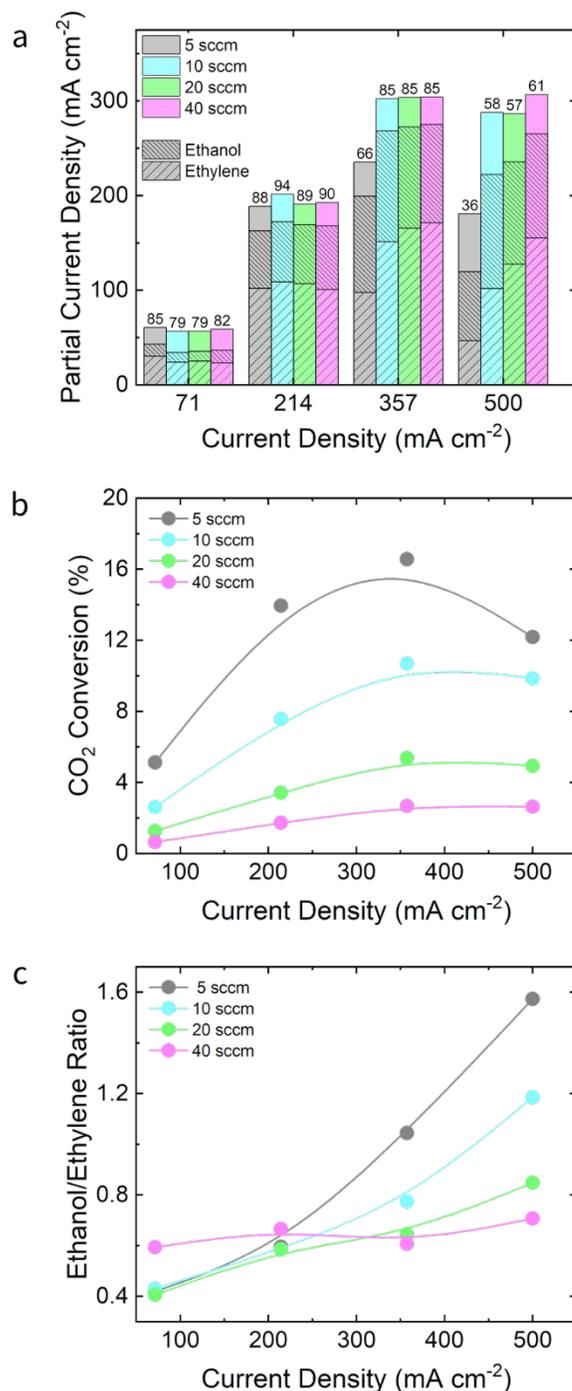
160 voltage of ±10V (Fig. 1, c and d). Generation 1 reactors had a total cell resistance of 33.1Ω, split

161 between the electrolyte (22.3Ω) and the membrane (10.7Ω), which led to a large voltage drop

162 (~10V at 214 mA cm⁻²) and a maximum current density of 266 mA cm⁻². Informed by these

163 findings, Generation 2 reactors decreased the distance between the electrodes by over half, which
164 consequently reduced the ohmic resistance. Accordingly, Generation 2 reactors exhibited a total
165 cell resistance of 21.2Ω , which represents a 34% decrease from Generation 1; however, the
166 maximum current density achieved only increased to 297 mA cm^{-2} , indicating that ohmic
167 resistance of the electrolyte was not the only limiting factor.

168 The Generation 3 system design connected the cathode gas and catholyte compartments in
169 a 3D-printed sparging chamber, saturating the electrolyte with CO_2 and reducing pH gradients and
170 concentration polarization (Fig. S2). This also eliminated pressure differences between the gas
171 compartments and the liquid compartments, minimizing breakthrough of electrolyte and flooding
172 due to a pressure gradient. Additionally, Generation 3 reactors featured a fourth compartment for
173 removal of oxygen and degassed carbon dioxide³³ from the anode at high current densities and
174 maintained the same distance between the cathode and anode. With these changes to the reactor
175 and system design, Generation 3 reactors achieved a lower cell resistance (17.7Ω), which is
176 primarily the result of a lower bulk pH due to continuous CO_2 saturation. At 214 mA cm^{-2} ,
177 Generation 3 reactors resulted in a $\sim 5\text{V}$ drop due to ohmic losses, almost 50% less than that of
178 Generation 1 reactors; this generation also exhibited higher Faradaic efficiencies (FE) for ethylene
179 and ethanol. This system reached current densities greater than 500 mA cm^{-2} , establishing the
180 viability of AM for high-rate electrosynthesis from CO_2 .

181 Evaluating performance of AM VFRs for various inlet CO₂ flow rates

182

183 **Fig. 2.** (a) CO₂R partial current density as a function of total current density for various flow rates. Ethylene and
 184 ethanol fractions are shaded. Other CO₂R products (CO, HCOO⁻, CH₄, CH₃COO⁻, CH₃CHO, and C₃H₇OH) are
 185 grouped within the unshaded region. The CO₂R Faradaic efficiency corresponding to condition is shown above each
 186 stacked column. (b) Electrochemical CO₂ conversion as a function of total current density for various flow rates. (c)
 187 Ratio of oxygenates to hydrocarbons as a function of flow rate and current density.

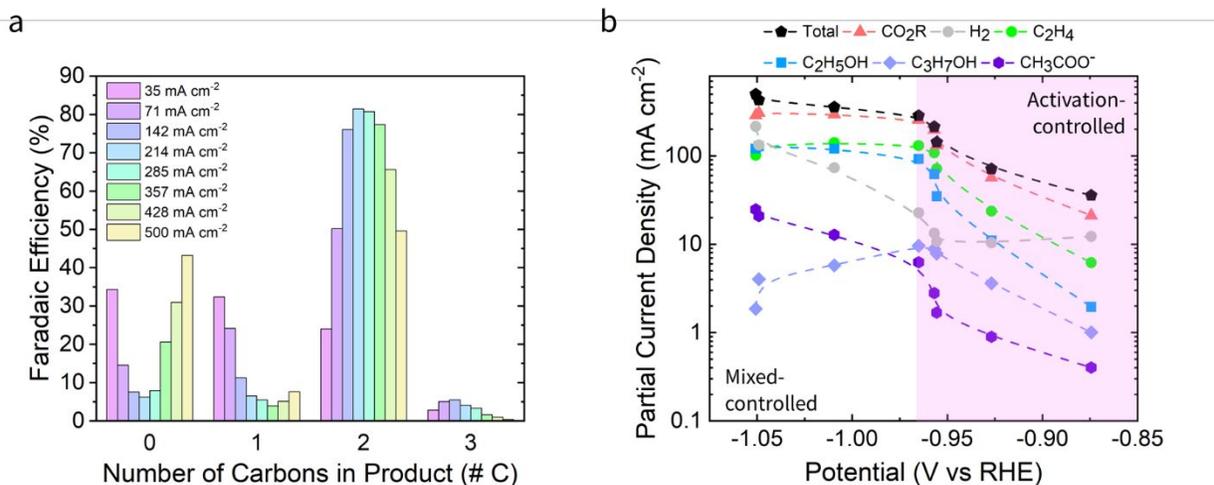
188
189 Building upon the advancements of the Generation 3 reactor, we explored the CO₂R performance
190 in depth for this AM system. We covered a range of CO₂ inlet flow rates of 5, 10, 20, and 40 sccm
191 on a simple Cu/PTFE catalyst and in 1 M KHCO₃. For these conditions, we display the current
192 density towards CO₂-reduced products (Fig. 2a) and the corresponding electrochemical CO₂
193 conversion (Fig. 2b). Peak conversion occurs for all flow rate conditions at 357 mA cm⁻², with a
194 maximum of 16.5% at 5 sccm. We observed two distinct regimes regarding CO₂ mass transport
195 effects: an activation-controlled regime, which has sufficient transport of CO₂ for electroreduction,
196 and a mixed-controlled regime, where both kinetic and mass transport effects lead to insufficient
197 concentrations of CO₂ throughout catalyst layer.

198 We observe little to no effect of CO₂ flow rate on the activity and selectivity of products at
199 lower current densities (≤ 214 mA cm⁻²), indicating this regime is activation controlled for all
200 tested flow rates. At higher current densities, electrolytic performance differs with CO₂ flow rate,
201 suggesting that some conditions operate in the mixed-controlled regime. At 357 mA cm⁻², the total
202 CO₂R current (~ 235 mA cm⁻²) at 5 sccm is lower than that of the other flow rates (~ 300 mA cm⁻²).
203 For > 10 sccm, there is a positive correlation between ethylene production and CO₂ flow rate,
204 while ethanol production remains unaffected, thus increasing the ratio of ethanol to ethylene (Fig.
205 2c). This trend is opposite of that observed in aqueous systems¹⁶; we attribute this to dramatic
206 differences in the local environment and available CO₂ between VFRs and LFRs. The selectivity
207 for ethylene at the catalyst surface could be influenced by the diffusion of the gaseous products
208 into the PTFE and out with the effluent gas stream; this diffusion depends on the flow rate of CO₂
209 in the reactor. The diffusion of ethanol into the electrolyte is not sharply influenced by the CO₂
210 gas flow, suggesting the ethanol pathway is preferred when CO₂R is either partially or fully

211 diffusion-controlled. Furthermore, the consistent activity towards ethanol may be influenced by
212 the heterogeneity of active sites³⁴ and increased absorption of H on Cu.³⁵

213 At 500 mA cm⁻², we observe a drop in CO₂R current density across flow rates < 40 sccm;
214 this is an extension of the effect seen at 5 sccm and 357 mA cm⁻². A decrease in ethylene production
215 corresponds to an increase production of hydrogen and methane, accounting for the remainder of
216 CO₂R current. At higher cathodic potentials, increased adsorption of hydrogen (*H) can suppress
217 conversion of CO₂ to multi-carbon products.¹⁷ Feeding CO₂ at 40 sccm resulted in reaching the
218 mass transport limited current density previously observed at 357 mA cm⁻². This clear distinction
219 between activation- and mixed-controlled regimes highlights the importance of both CO₂ inlet
220 flow and applied current density on mass transport effects within electrochemical reactors.

221 Mass transport limited current density in AM VFRs



222

223 **Fig. 3.** (a) Selectivity towards 0-carbon (H₂), 1-carbon (CO, HCOO⁻, CH₄), 2-carbon (C₂H₄, C₂H₅OH, CH₃COO⁻,
 224 CH₃CHO), and 3-carbon (C₃H₇OH) products at different current densities. (b) Partial current densities of select
 225 products and CO₂R, as well as total current density, as a function of applied cathodic potential.
 226

227 We also investigated the effects of the electrode potential on electrocatalytic performance at 10
 228 sccm CO₂ flow rates. The system demonstrated remarkable selectivity for CO₂R across a wide
 229 range of current densities (Fig. 3a), including over 90% FE for CO₂R from 142 mA cm⁻² to 285
 230 mA cm⁻²; at 142 mA cm⁻² we observed the maximum CO₂R cathodic energy efficiency and FE for
 231 the system at 49.6% (Fig. S13) and 92.8%, respectively. Total C₂₊ FE reached a maximum of
 232 85.9% FE at 214 mA cm⁻², which is one of the highest reported to multi-carbon products for CO₂R;
 233 the two major products formed were ethylene (50.7% at 214 mA cm⁻²) and ethanol (33.9% at 357
 234 mA cm⁻²). Strikingly, we show here that high selectivity for CO₂R and CO₂ conversion can be
 235 achieved in bulk pH-neutral environments with no modifications or additives to either the catalyst
 236 or electrolyte. This compares favorably to previous reports using further-developed Cu electrodes
 237 and/or operating in alkaline conditions,^{32,35-42} while mitigating issues associated with bulk
 238 electrolyte consumption of CO₂ that is particularly problematic in alkaline environments and can
 239 lead to lower overall system efficiency.³³

240 Hydrogen, the only zero-carbon product, remained below 8% from 142 to 285 mA cm⁻²,
241 agreeing with previous reports depicting a preference for CO₂R within this current density
242 range.^{33,43} At current densities \leq 285 mA cm⁻², hydrogen decreased in FE from 34% to 8% while
243 2-carbon products rapidly increased from 24% to 81%; formation of single-carbon products were
244 consistently low in this regime. This inverse relationship is a result from several factors, including
245 sufficient adsorption of *CO^{17,43} and *COOH^{8,13-15} intermediates on the catalyst surface and
246 favorable kinetics for carbon-carbon coupling.^{7,8,44} Furthermore, suppression of HER occurs via
247 an increasing pH in the local environment,^{32,33} we predict an increase up to 5 units for pH in our
248 system (Fig. S21).

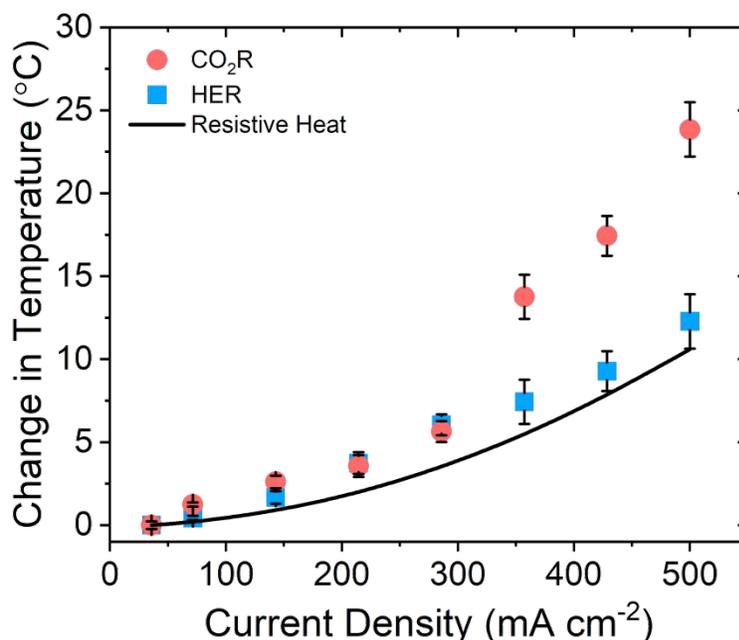
249 Similar to the results from the flow rate study, we observed both activation-controlled and
250 mixed-controlled regimes based on partial current densities and the corresponding electrochemical
251 potential (Fig. 3b). The potential range from -0.87V to -0.97V vs RHE corresponds to the
252 activation-controlled regime; CO₂R current density increases exponentially as a function of
253 potential, with product activities tracing the behavior of Butler-Volmer kinetics.⁴⁵ The partial
254 current densities of the 2-carbon products (C₂H₄, C₂H₅OH, and CH₃COO⁻) increase at similar rates
255 in this regime; their slopes suggest the sharing of some reaction pathways and carbonaceous
256 intermediates.^{7,12,17,46}

257 A plateau in CO₂R current density of around \sim 300 mA cm⁻² is a result of a transition from
258 activation- to mixed-controlled regimes (\leq -0.97V vs. RHE). While ethylene and ethanol current
259 densities reach an asymptote, H₂ and CH₄ partial current densities increase and agree with previous
260 literature (Fig. S12).^{12,47,48} This relationship suggests a higher *H coverage on the catalyst surface,
261 increasing the probability of protonating adsorbed single-carbon species. Interestingly, acetate
262 partial current density increases throughout both regimes, suggesting acetate formation may

263 correlate with concentration of OH^- .⁴⁹ As the total current density approaches 500 mA cm^{-2} , total
264 CO_2 conversion decreases, suggesting a modified reaction environment that favors hydrogen
265 evolution. At higher overpotentials, an high local pH and lower concentrations of KHCO_3 likely
266 result in a dominant proton-donor shift to H_2O .⁵⁰ The CO_2 mass transport limitation observed in
267 this region is analogous to results that have been observed in aqueous, liquid-fed CO_2R systems.¹²

268 **CO₂ mass transport-limiting phenomena**

269 There are several mass transport phenomena that control how CO₂ reaches the catalyst surface in
270 aqueous electrolyte VFRs: diffusion of CO_{2(g)} through the PTFE layer to the catalyst layer,
271 dissolution of CO_{2(g)} into the electrolyte to form CO_{2(aq)}, and diffusion of CO_{2(aq)} to the catalyst
272 surface. Many studies show that OH⁻ production at the cathode and consequent homogenous
273 consumption of CO_{2(aq)} to form carbonate can result in variable reactant concentrations, alter
274 product selectivity, and limit CO₂ transport to the catalyst layer.^{32,51–54} However, high current
275 densities (> 1 A cm⁻²) to reduced products in alkaline conditions^{40,42} suggest additional mass
276 transport limiting phenomena must be present in these bulk neutral-media, AM VFR systems.

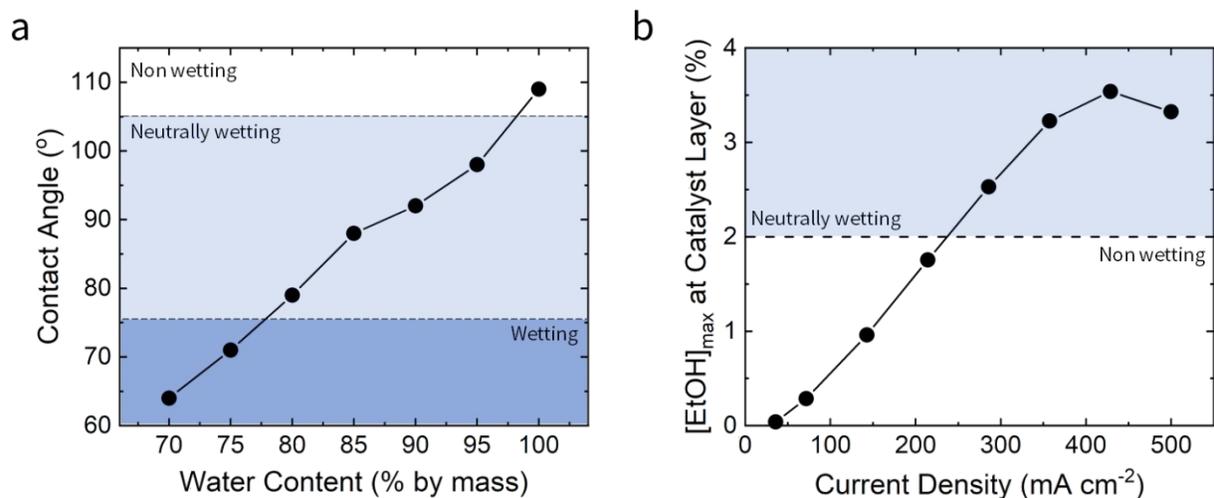


277
278 **Fig. 4.** Measured cathode temperature during electrolysis for CO₂R and HER at 10 sccm. Modeling of the expected
279 increase in temperature based on resistive heating is represented as the solid black curve.
280

281 We propose that temperature can change the concentration of CO₂ at the gas-liquid
282 interface, contributing to CO₂ mass transport limitations. While there is previous literature on the

283 effect of temperature on CO₂R,^{55–57} there remains a lack of experimental understanding on how
284 temperature changes within VFRs during CO₂ electrolysis. We measured the temperature change
285 of the catholyte near the cathode surface for both HER and CO₂R (Fig. 4). At current densities <
286 300 mA cm⁻², the increase in temperature across the cathode is < 7°C for all experiments; this
287 results from resistive heating of the electrolyte, as seen by the simulated temperature change from
288 the model. As resistive heating is dependent on the applied current and electrolyte resistance, the
289 measured increase in temperature for HER and CO₂R are similar in this range. At current densities
290 > 300 mA cm⁻², the increase in temperature is higher in CO₂R (max of ~24°C at 500 mA cm⁻²)
291 than in HER (max of ~12°C at 500 mA cm⁻²). The exothermic reaction of OH⁻ with CO₂ and the
292 heats of reaction from all electrochemical reactions at the catalyst surface contribute significantly
293 to the difference in temperature increase.

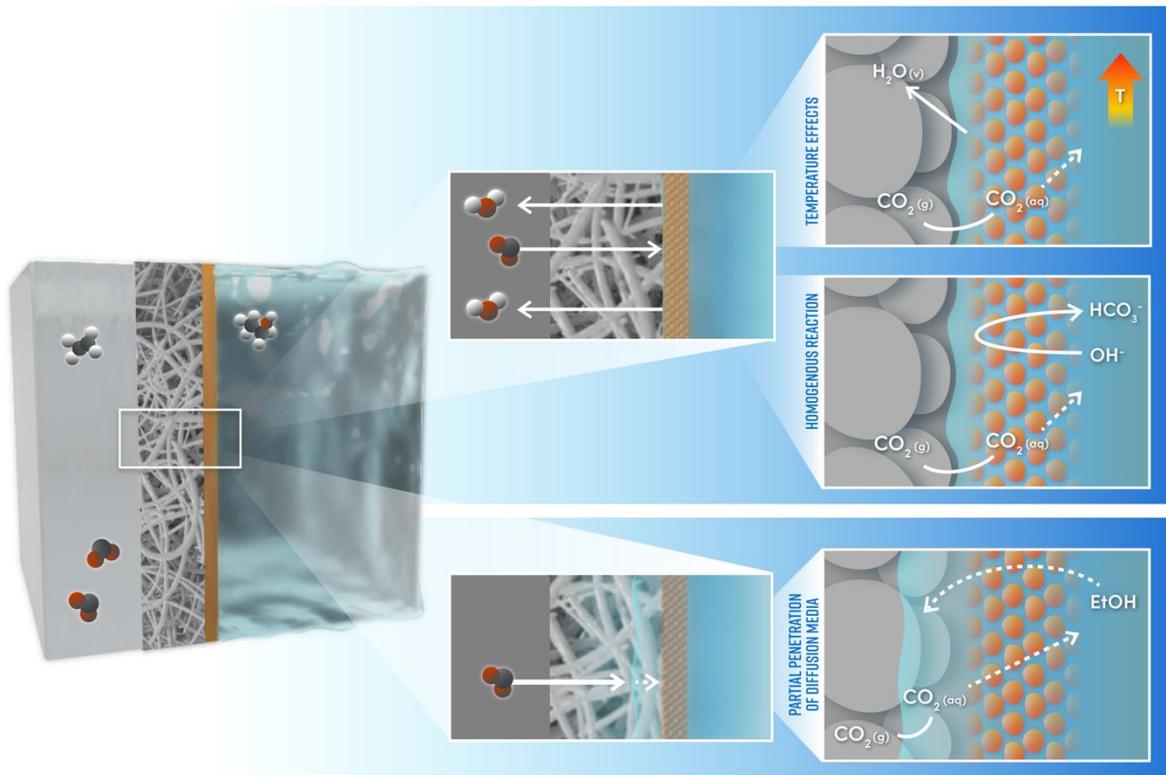
294 Dynamic changes in temperature change several key variables that can impact CO₂ mass
295 transport. At higher temperatures, the solubility of CO₂ decreases from 33.4 mM at 25°C to 20.0
296 mM at 45°C. Coupled with higher concentrations of OH⁻ in the electrolyte, this can reduce the
297 amount of CO₂ that is available to the catalyst, leading to a mass transport limited current density
298 for CO₂R. Additionally, temperature increases can change the conductivity of the electrolyte and
299 the diffusion of various species in the aqueous phase; it can also influence the kinetic barriers for
300 CO₂R reaction pathways, which can alter the selectivity of the catalyst and limit the rate of CO₂R
301 at high current densities.



302

303 **Fig. 5.** (a) Contact angle measurements for ethanol/water mixtures on bulk PTFE. This represents a conservative
 304 estimate for wetting of electrodes, as the Cu catalyst is more hydrophilic than PTFE. (b) Calculated maximum ethanol
 305 concentrations at the gas-liquid-solid interface as a function of total current density. Shaded and unshaded zone
 306 represents neutrally wetted and non-wetted regions, respectively.
 307

308 Additionally, we hypothesize that changes in wettability of the PTFE support can
 309 contribute to CO₂ mass transport limitations.⁵⁸ Solvent water does not wet the PTFE support and
 310 the triple phase interface remains abrupt and close to the cathode.³² Liquid products
 311 (particularly ethanol) that are produced in the catalyst layer can wet the PTFE support by
 312 changing the solvent surface tension.⁵⁸ Based on contact angle measurements (Fig. 5a), only a
 313 small concentration of ~2% (by mass) of ethanol is sufficient to change the hydrophobicity of the
 314 PTFE from non-wetting ($\Theta > 105^\circ$) to neutrally wetting ($75^\circ < \Theta < 105^\circ$).⁵⁹ We calculate a
 315 maximum ethanol concentration of > 3.5% for a range of current densities in our reactor (Fig. 5b);
 316 this suggests that the electrolyte could begin penetrating the neutrally wet PTFE interface.
 317 Electrolyte penetrating the PTFE could block pores and extend the gas-electrolyte interface,
 318 inhibiting CO₂ mass transfer.^{51,58} It is challenging to determine the exact depth of electrolyte
 319 penetration due to species transport, local pressure, capillary forces, and the specific geometry of
 320 PTFE fibers.

321 **Insights from dimensionless number analysis**

322

323 **Scheme 1.** Proposed phenomena that can impact CO₂ mass transport to the catalyst surface: temperature effects,
 324 homogenous reaction, and partial penetration into diffusion media.

325

326

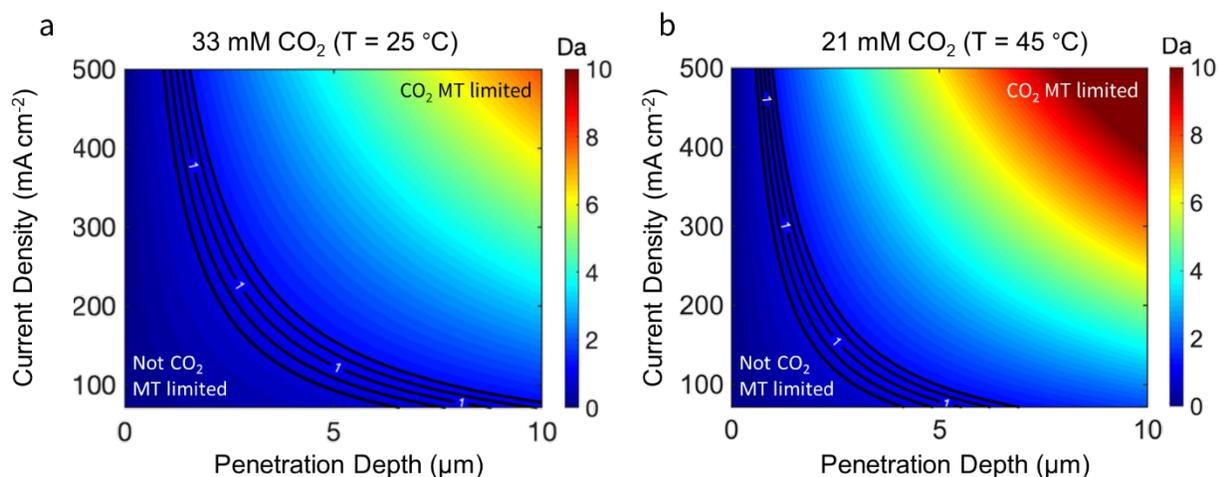
We sought to use a single parameter that incorporates the aforementioned mass transport
 327 mechanisms (Scheme 1) to better understand operating regimes. The Damköhler number, Da , is a
 328 dimensionless quantity that represents the ratio between diffusion and reaction time scales. We
 329 define Da to solicit a regime diagram for transport within the GDE (Eq. 1):

$$330 \quad Da_{diff} = \frac{\tau_{DMI}}{[CO_{2,0}]P/m} = \frac{\nu P}{[CO_{2,0}]D_{CO_2}} \quad (1)$$

331

Where m represents the molar consumption per unit area from reduction ($m = j/nF$, where
 332 j is the current density), and P is the liquid penetration depth into the ePTFE. We performed a
 333 timescale analysis representing diffusion of CO₂ to decouple activation- and mixed-controlled
 334 regimes. When the timescale for CO₂ diffusion to the catalyst layer is greater than the timescale

335 for surface reactions, we define the system to be operating in a CO₂ mass transport limited regime
 336 (Da > 1). As higher current densities are reached, the combination of mass transport limiting
 337 mechanisms can push the system towards the Da > 1 regime.



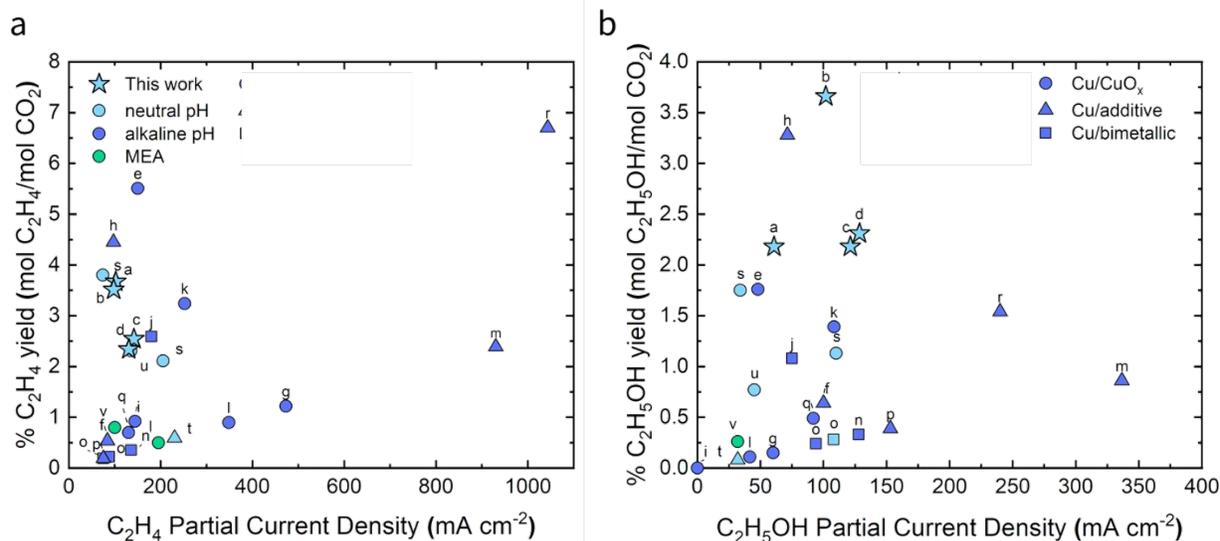
338
 339 **Fig. 6.** Contour plots of Da, with contour lines at near the mass transport limitation boundary of Da~1 (Da=0.75-1.25),
 340 representing two conditions for CO₂ concentration at the gas-liquid interface corresponding to (a) 33 mM (25°C) and
 341 (b) 21 mM (45°C).
 342

343 Contour plots (Fig. 6) elucidate different mass transport regimes based on CO₂ gas-liquid
 344 interface concentrations of 33 (25°C) and 21 mM (45°C). For a system operating ≥ 100 mA cm⁻²
 345 with an interfacial CO₂ concentration of 33 mM (Fig. 6a), we predict that ≤ 5 μ m of electrolyte
 346 penetration will push the system towards a Da > 1 regime. As current density increases, this effect
 347 is exacerbated due to increased OH⁻ generation. When extending this to lower interfacial CO₂
 348 concentrations, we see the quantity of penetration to induce mass transport limitations is further
 349 decreased. At 21 mM – a concentration corresponding to the temperature measurements during
 350 electrolysis – the system enters the mass transport limited regime at ≤ 2 μ m penetration at ≥ 100
 351 mA cm⁻² (Fig. 6b).

352 This model provides the framework for using Da as a relevant dimensionless number to
 353 distinguish when a VFR is CO₂ mass transport limited for any reactor and system design. Other

354 additional variables, including the contributions from catalyst deformation, are not accounted in
355 this model; however, those effects are less likely to lead to our observations. Using Da to
356 understand the primary drivers and bottlenecks for species transport can lead to better design and
357 operation of reactors to overcome these limitations. Furthermore, AM presents a promising
358 platform for coupling with computational fluid dynamics for greater understanding of these
359 electrochemical systems.

360 Achieving high yield of ethanol and ethylene in AMVFRs



361 **Fig. 7.** Comparison of literature and this work for (a) % ethylene and (b) % ethanol yield as a function of partial
 362 current density. Indigo colored data represents VFRs with bulk alkaline pH including [e] Ma et al.,³⁶ [f] Zhuang et
 363 al.,⁶⁰ [g] Dinh et al.,³² [h] Hoang et al.,⁶¹ [i] Wang et al.,³⁷ [j] Hoang et al.,³⁸ [k] Lv et al.,³⁹ [l] Wang et al.,⁶² [m]
 364 Arquer et al.,⁴⁰ [n] Luo et al.,³⁵ [o] Li et al.,⁶³ [p] Wang et al.,⁴¹ [q] Yang et al.,⁶⁴ [r] Ma et al.,⁴² Light blue colored
 365 data represents VFRs with bulk neutral pH including [a-d] this work; [o] Li et al.,⁶³ [s] Tan et al.,⁴³ [t] Li et al.,⁶⁵ [u]
 366 Ma et al.,³³ Green colored data represents MEA VFRs including [l] Wang et al.,⁶² [v] Garbarido et al.,⁶⁶ Point shapes
 367 represent catalyst layer composition class: Cu/Cu_x, Cu modified with a non-metal additive, and Cu-based bimetallics.
 368
 369

370 When assessing the feasibility of these systems, optimization of operating conditions and
 371 system components toward specific products has seen significant advancements. However, this
 372 does not account for the flowrate of CO₂ into the cathodic gas compartment; maximizing product
 373 formation per molecule of CO₂ is imperative for improving the economic viability of CO₂
 374 electrolysis. This ratio, known as yield, is defined for an CO₂R process below:

$$375 \text{ Yield} = \frac{\text{products out}}{\text{reactants in}} = \frac{\text{mol } C_xH_yO_z \text{ produced}}{\text{mol } CO_2 \text{ in}} \quad (2)$$

376 Product yield is critical as a primary metric to evaluate the reactor performance in many
 377 industries. Considering the differences in catalyst composition and reaction area, local electrolyte
 378 environment, and reactor and system design, we employ it here as a quantitative metric that can
 379 be used to compare results from many reports across the field.

380 We compare the ethylene (Fig. 7a) and ethanol (Fig. 7b) yield for the highest performing
381 set of conditions from various reports. We classify the results from literature into three categories
382 for the electrolyte environment (neutral pH, alkaline pH, and membrane electrode assembly or
383 MEA) and three categories for the catalyst composition (Cu/CuO_x, Cu/additive, and bimetallics).
384 The AM VFRs in this work achieve high yield towards ethylene (3.67%), which is amongst the
385 highest performing systems for neutral electrolytes in literature (Fig. 7a). As expected, there is an
386 inverse relationship between product yield and CO₂ flow rate (Fig. S28).

387 On a partial current density basis, ethanol production is limited by challenging kinetics and
388 a focus on conditions that favor the competing ethylene pathway.^{32,40,65} While various reports have
389 shown improved selectivity toward ethanol,^{35,39,41} these findings are attributed to advanced catalyst
390 formulation and/or tuning of electrolyte and environment. With consistent production of ethanol
391 in AMVFRs across a range of CO₂ flow rates, we report the highest yield to date towards ethanol
392 (3.66%). This work affirms that high yield towards multi-carbon products from CO₂ can be
393 achieved with generic Cu catalysts in moderate electrolyte conditions.

394 **Conclusions**

395 In this work, we introduce a new paradigm for chemical production using advanced
396 manufacturing. We highlight prospects of AM for accelerated improvement and development of
397 CO₂ electrolyzers via reactor design; using 3D printing instead of conventional machining can lead
398 to significant improvements in activity, selectivity, yield, and energy efficiency through rapid
399 iteration. We realize reactor design, in addition to electrode and electrolyte engineering, as a key
400 variable for CO₂ reduction. Insights gained from this work can be employed for all vapor-fed
401 reactors to improve performance and build understanding.

402 The potential of this design strategy can lead to well-characterized and scalable systems,
403 especially when coupled with computational efforts and technoeconomic analyses. Furthermore,
404 we anticipate the implementation of AM for deterministic electrode, catalyst and membrane
405 development. This work demonstrates, for the first time, AM as a platform for creating robust
406 vapor-fed reactors and components for high-performance CO₂ reduction. We are optimistic that
407 AM will transform our approach towards designing systems for catalysis and propel the design of
408 novel reactors for the next generation of chemicals synthesis.

409

410 **Author Contributions**

411 D.C. and J.T.F. contributed significantly and equally to this work. *Writing – Original Draft*: D.C.
412 and J.T.F.; *Writing – Review & Editing*: D.C., J.T.F., S.S., J.D., D.L., A.A.W., A.S., C.H., T.F.J.,
413 S.B., and E.D.; *Conceptualization*: J.T.F., D.C., S.B., and E.D.; *Investigation*: D.C., J.T.F., J.D.,
414 D.L., and A.A.W.; *Methodology*: J.T.F., D.C., S.S., J.D., J.H. and V.A.B.; *Formal Analysis*: S.S.,
415 J.D., and V.A.B.; *Software*: V.A.B., S.S., J.D.; *Validation*: D.C., J.T.F., S.S., J.D., D.L., A.A.W.,
416 J.H., and V.A.B.; *Project Administration*: J.T.F., D.C., V.A.B., C.H., T.F.J., S.B., and E.D.;
417 *Funding Acquisition*: A.S., C.H., T.F.J., S.B., and E.D.

418 **Acknowledgements**

419 This work was performed under the auspices of the U.S. Department of Energy by
420 Lawrence Livermore National Laboratory under Contract DE-AC52-07NA27344 and 19-SI-005
421 with IM release number LLNL-JRNL-1025388-DRAFT and a cooperative research and
422 development agreement with TOTAL American Services, Inc. (affiliate of TOTAL SE) under
423 agreement number TC02307. DC thanks the GEM Fellowship at Stanford University and
424 Lawrence Livermore National Laboratory, as well as the Stanford Graduate Fellowship.

425 The authors thank Veronica Chen and TID Team at Lawrence Livermore National
426 Laboratory for preparation of figures. The authors thank and appreciate guidance from several
427 colleagues at Lawrence Livermore National Laboratories including Dr. April Sawvel, Hannah
428 Eshelman, Cameron Cornell, Dr. Zhen Qi, Dr. Nikola Dudukovic, Dr. Thomas Moore, Dr. Jon
429 Lee, Dr. Jonathan Davis, Dr. David Wakerley, Dr. Sarah Lamaison, and Dr. Lei Wang. The authors
430 thank members from the Sargent Group at the University of Toronto for helpful discussions. The
431 authors thank the Nuclear Magnetic Resonance Spectroscopy and Cleanroom facilities at
432 Lawrence Livermore National Laboratory, the Stanford Nano-Shared Facilities at Stanford
433 University, and the contact angle facilities at SLAC National Laboratory.

434

435

436 **Declaration of Interests**

437 The authors declare no competing interests.

438 **References**

- 439 1. Obama, B. (2017). The irreversible momentum of clean energy. *Science* 355, 126–129.
- 440 2. Intergovernmental Panel on Climate Change (2014). *Clim. Change 2014 Synth. Rep.*
- 441 3. Chu, S., Cui, Y., and Liu, N. (2017). The path towards sustainable energy. *Nat. Mater.* 16, 16.
- 442 4. International Energy Outlook 2019 (2019).
- 443 5. De Luna, P., Hahn, C., Higgins, D., Jaffer, S.A., Jaramillo, T.F., and Sargent, E.H. (2019).
444 What would it take for renewably powered electrosynthesis to displace petrochemical
445 processes? *Science* 364, eaav3506.
- 446 6. Hori, Y., Wakebe, H., Tsukamoto, T., and Koga, O. (1994). Electrocatalytic process of CO
447 selectivity in electrochemical reduction of CO₂ at metal electrodes in aqueous media.
448 *Electrochimica Acta* 39, 1833–1839.
- 449 7. Kuhl, K.P., Cave, E.R., Abram, D.N., and Jaramillo, T.F. (2012). New insights into the
450 electrochemical reduction of carbon dioxide on metallic copper surfaces. *Energy Environ. Sci.*
451 5, 7050.
- 452 8. Peterson, A.A., Abild-Pedersen, F., Studt, F., Rossmeisl, J., and Nørskov, J.K. (2010). How
453 copper catalyzes the electroreduction of carbon dioxide into hydrocarbon fuels. *Energy*
454 *Environ. Sci.* 3, 1311–1315.
- 455 9. Zhang, M., Cheng, J., Xuan, X., Zhou, J., and Cen, K. (2017). Pt/graphene aerogel deposited
456 in Cu foam as a 3D binder-free cathode for CO₂ reduction into liquid chemicals in a TiO₂
457 photoanode-driven photoelectrochemical cell. *Chem. Eng. J.* 322, 22–32.
- 458 10. Zhang, M., Xuan, X., Wang, W., Ma, C., and Lin, Z. (2020). Anode Photovoltage
459 Compensation-Enabled Synergistic CO₂ Photoelectrocatalytic Reduction on a Flower-Like
460 Graphene-Decorated Cu Foam Cathode. *Adv. Funct. Mater.* 30, 2005983.
- 461 11. Resasco, J., and Bell, A.T. (2020). Electrocatalytic CO₂ Reduction to Fuels: Progress and
462 Opportunities. *Trends Chem.* 2, 825–836.
- 463 12. Nitopi, S., Bertheussen, E., Scott, S.B., Liu, X., Engstfeld, A.K., Horch, S., Seger, B.,
464 Stephens, I.E.L., Chan, K., Hahn, C., et al. (2019). Progress and Perspectives of
465 Electrochemical CO₂ Reduction on Copper in Aqueous Electrolyte. *Chem. Rev.* 119, 7610–
466 7672.
- 467 13. Mauri, R. (2015). *Transport phenomena in multiphase flows* (Springer).
- 468 14. Higgins, D., Hahn, C., Xiang, C., Jaramillo, T.F., and Weber, A.Z. (2019). Gas-Diffusion
469 Electrodes for Carbon Dioxide Reduction: A New Paradigm. *ACS Energy Lett.* 4, 317–324.

- 470 15. Liang, S., Altaf, N., Huang, L., Gao, Y., and Wang, Q. (2020). Electrolytic cell design for
471 electrochemical CO₂ reduction. *J. CO₂ Util.* *35*, 90–105.
- 472 16. Vennekoetter, J.-B., Sengpiel, R., and Wessling, M. (2019). Beyond the catalyst: How
473 electrode and reactor design determine the product spectrum during electrochemical CO₂
474 reduction. *Chem. Eng. J.* *364*, 89–101.
- 475 17. Gibson, I., Rosen, D., and Stucker, B. (2010). *Additive Manufacturing Technologies.* *238*.
- 476 18. Duoss, E.B., Weisgraber, T.H., Hearon, K., Zhu, C., Small IV, W., Metz, T.R., Vericella, J.J.,
477 Barth, H.D., Kuntz, J.D., Maxwell, R.S., et al. (2014). Three-dimensional printing of
478 elastomeric, cellular architectures with negative stiffness. *Adv. Funct. Mater.* *24*, 4905–4913.
- 479 19. Zhu, C., Liu, T., Qian, F., Chen, W., Chandrasekaran, S., Yao, B., Song, Y., Duoss, E.B.,
480 Kuntz, J.D., Spadaccini, C.M., et al. (2017). 3D printed functional nanomaterials for
481 electrochemical energy storage. *Nano Today* *15*, 107–120.
- 482 20. Nguyen, D.T., Meyers, C., Yee, T.D., Dudukovic, N.A., Destino, J.F., Zhu, C., Duoss, E.B.,
483 Baumann, T.F., Suratwala, T., Smay, J.E., et al. (2017). 3D-printed transparent glass. *Adv.*
484 *Mater.* *29*, 1701181.
- 485 21. Clark, E.A., Alexander, M.R., Irvine, D.J., Roberts, C.J., Wallace, M.J., Yoo, J., and Wildman,
486 R.D. (2020). Making tablets for delivery of poorly soluble drugs using photoinitiated 3D inkjet
487 printing. *Int. J. Pharm.* *578*, 118805.
- 488 22. Vericella, J.J., Baker, S.E., Stolaroff, J.K., Duoss, E.B., Hardin, J.O., Lewicki, J., Glogowski,
489 E., Floyd, W.C., Valdez, C.A., Smith, W.L., et al. (2015). Encapsulated liquid sorbents for
490 carbon dioxide capture. *Nat. Commun.* *6*, 1–7.
- 491 23. C. Bui, J., T. Davis, J., and V. Esposito, D. (2020). 3D-Printed electrodes for membraneless
492 water electrolysis. *Sustain. Energy Fuels* *4*, 213–225.
- 493 24. Yang, K., Kas, R., Smith, W.A., and Burdyny, T. (2021). Role of the Carbon-Based Gas
494 Diffusion Layer on Flooding in a Gas Diffusion Electrode Cell for Electrochemical CO₂
495 Reduction. *ACS Energy Lett.* *6*, 33–40.
- 496 25. Zhang, J., Luo, W., and Züttel, A. (2019). Self-supported copper-based gas diffusion
497 electrodes for CO₂ electrochemical reduction. *J. Mater. Chem. A* *7*, 26285–26292.
- 498 26. Ma, M., Clark, E.L., Therkildsen, K.T., Dalsgaard, S., Chorkendorff, I., and Seger, B. (2020).
499 Insights into the carbon balance for CO₂ electroreduction on Cu using gas diffusion electrode
500 reactor designs. *Energy Environ. Sci.* *13*, 977–985.
- 501 27. Hahn, C., Hatsukade, T., Kim, Y.-G., Vailionis, A., Baricuatro, J.H., Higgins, D.C., Nitopi,
502 S.A., Soriaga, M.P., and Jaramillo, T.F. (2017). Engineering Cu surfaces for the
503 electrocatalytic conversion of CO₂: Controlling selectivity toward oxygenates and
504 hydrocarbons. *Proc. Natl. Acad. Sci.* *114*, 5918–5923.

- 505 28. Wang, L., Nitopi, S.A., Bertheussen, E., Orazov, M., Morales-Guio, C.G., Liu, X., Higgins,
506 D.C., Chan, K., Nørskov, J.K., Hahn, C., et al. (2018). Electrochemical Carbon Monoxide
507 Reduction on Polycrystalline Copper: Effects of Potential, Pressure, and pH on Selectivity
508 toward Multicarbon and Oxygenated Products. *ACS Catal.* *8*, 7445–7454.
- 509 29. Luo, M., Wang, Z., Li, Y.C., Li, J., Li, F., Lum, Y., Nam, D.-H., Chen, B., Wicks, J., Xu, A.,
510 et al. (2019). Hydroxide promotes carbon dioxide electroreduction to ethanol on copper via
511 tuning of adsorbed hydrogen. *Nat. Commun.* *10*, 1–7.
- 512 30. Liu, X., Schlexer, P., Xiao, J., Ji, Y., Wang, L., Sandberg, R.B., Tang, M., Brown, K.S., Peng,
513 H., Ringe, S., et al. (2019). pH effects on the electrochemical reduction of CO (2) towards C
514 2 products on stepped copper. *Nat. Commun.* *10*, 32.
- 515 31. Ma, S., Sadakiyo, M., Luo, R., Heima, M., Yamauchi, M., and Kenis, P.J. (2016). One-step
516 electrosynthesis of ethylene and ethanol from CO₂ in an alkaline electrolyzer. *J. Power
517 Sources* *301*, 219–228.
- 518 32. Dinh, C.-T., Burdyny, T., Kibria, M.G., Seifitokaldani, A., Gabardo, C.M., de Arquer, F.P.G.,
519 Kiani, A., Edwards, J.P., De Luna, P., Bushuyev, O.S., et al. (2018). CO₂ electroreduction to
520 ethylene via hydroxide-mediated copper catalysis at an abrupt interface. *Science* *360*, 783–
521 787.
- 522 33. Wang, Y., Shen, H., Livi, K.J., Raciti, D., Zong, H., Gregg, J., Onadeko, M., Wan, Y., Watson,
523 A., and Wang, C. (2019). Copper Nanocubes for CO₂ Reduction in Gas Diffusion Electrodes.
524 *Nano Lett.* *19*, 8461–8468.
- 525 34. Hoang, T.T., Verma, S., Ma, S., Fister, T.T., Timoshenko, J., Frenkel, A.I., Kenis, P.J., and
526 Gewirth, A.A. (2018). Nanoporous Copper–Silver Alloys by Additive-Controlled
527 Electrodeposition for the Selective Electroreduction of CO₂ to Ethylene and Ethanol. *J. Am.
528 Chem. Soc.* *140*, 5791–5797.
- 529 35. Lv, J.-J., Jouny, M., Luc, W., Zhu, W., Zhu, J.-J., and Jiao, F. (2018). A highly porous copper
530 electrocatalyst for carbon dioxide reduction. *Adv. Mater.* *30*, 1803111.
- 531 36. de Arquer, F.P.G., Dinh, C.-T., Ozden, A., Wicks, J., McCallum, C., Kirmani, A.R., Nam, D.-
532 H., Gabardo, C., Seifitokaldani, A., Wang, X., et al. (2020). CO₂ electrolysis to multicarbon
533 products at activities greater than 1 A cm⁻². *Science* *367*, 661–666.
- 534 37. Wang, X., Wang, Z., García de Arquer, F.P., Dinh, C.-T., Ozden, A., Li, Y.C., Nam, D.-H.,
535 Li, J., Liu, Y.-S., Wicks, J., et al. (2020). Efficient electrically powered CO₂-to-ethanol via
536 suppression of deoxygenation. *Nat. Energy* *5*, 478–486.
- 537 38. Ma, W., Xie, S., Liu, T., Fan, Q., Ye, J., Sun, F., Jiang, Z., Zhang, Q., Cheng, J., and Wang,
538 Y. (2020). Electrocatalytic reduction of CO₂ to ethylene and ethanol through hydrogen-
539 assisted C–C coupling over fluorine-modified copper. *Nat. Catal.* *3*, 478–487.
- 540 39. Tan, Y.C., Lee, K.B., Song, H., and Oh, J. (2020). Modulating Local CO₂ Concentration as a
541 General Strategy for Enhancing C–C Coupling in CO₂ Electroreduction. *Joule*.

- 542 40. Yang, H., Hu, Y., Chen, J., Balogun, M.-S. (Jie T., Fang, P., Zhang, S., Chen, J., and Tong, Y.
543 (2019). Intermediates Adsorption Engineering of CO₂ Electroreduction Reaction in Highly
544 Selective Heterostructure Cu-Based Electrocatalysts for CO Production. *Adv. Energy Mater.*
545 *9*, 1901396.
- 546 41. Feaster, J.T., Shi, C., Cave, E.R., Hatsukade, T., Abram, D.N., Kuhl, K.P., Hahn, C., Nørskov,
547 J.K., and Jaramillo, T.F. (2017). Understanding selectivity for the electrochemical reduction
548 of carbon dioxide to formic acid and carbon monoxide on metal electrodes. *Acs Catal.* *7*, 4822–
549 4827.
- 550 42. Luo, W., Nie, X., Janik, M.J., and Asthagiri, A. (2016). Facet Dependence of CO₂ Reduction
551 Paths on Cu Electrodes. *ACS Catal.* *6*, 219–229.
- 552 43. Bagotsky, V.S. (2005). *Fundamentals of Electrochemistry* (John Wiley & Sons).
- 553 44. Lum, Y., Cheng, T., Goddard, W.A., and Ager, J.W. (2018). Electrochemical CO Reduction
554 Builds Solvent Water into Oxygenate Products. *J. Am. Chem. Soc.* *140*, 9337–9340.
- 555 45. Kuhl, K.P., Hatsukade, T., Cave, E.R., Abram, D.N., Kibsgaard, J., and Jaramillo, T.F. (2014).
556 Electrocatalytic Conversion of Carbon Dioxide to Methane and Methanol on Transition Metal
557 Surfaces. *J. Am. Chem. Soc.* *136*, 14107–14113.
- 558 46. Wang, X., Xu, A., Li, F., Hung, S.-F., Nam, D.-H., Gabardo, C.M., Wang, Z., Xu, Y., Ozden,
559 A., Rasouli, A.S., et al. (2020). Efficient methane electrosynthesis enabled by tuning local
560 CO₂ availability. *J. Am. Chem. Soc.*
- 561 47. Luc, W., Fu, X., Shi, J., Lv, J.-J., Jouny, M., Ko, B.H., Xu, Y., Tu, Q., Hu, X., Wu, J., et al.
562 (2019). Two-dimensional copper nanosheets for electrochemical reduction of carbon
563 monoxide to acetate. *Nat. Catal.* *2*, 423–430.
- 564 48. Resasco, J., Lum, Y., Clark, E., Zeledon, J.Z., and Bell, A.T. (2018). Effects of anion identity
565 and concentration on electrochemical reduction of CO₂. *ChemElectroChem* *5*, 1064–1072.
- 566 49. Weng, L.-C., Bell, A.T., and Weber, A.Z. (2018). Modeling gas-diffusion electrodes for CO₂
567 reduction. *Phys. Chem. Chem. Phys.* *20*, 16973–16984.
- 568 50. Wu, K., Birgersson, E., Kim, B., Kenis, P.J.A., and Karimi, I.A. (2014). Modeling and
569 Experimental Validation of Electrochemical Reduction of CO₂ to CO in a Microfluidic Cell.
570 *J. Electrochem. Soc.* *162*, F23.
- 571 51. Delacourt, C., and Newman, J. (2010). Mathematical Modeling of CO₂ Reduction to CO in
572 Aqueous Electrolytes: II. Study of an Electrolysis Cell Making Syngas from and Reduction at
573 Room Temperature. *J. Electrochem. Soc.* *157*, B1911.
- 574 52. Kas, R., Yang, K., Bohra, D., Kortlever, R., Burdyny, T., and Smith, W.A. (2020).
575 Electrochemical CO₂ reduction on nanostructured metal electrodes: fact or defect? *Chem. Sci.*
576 *11*, 1738–1749.

- 577 53. Zong, Y., Chakthranont, P., and Suntivich, J. (2020). Temperature Effect of CO₂ Reduction
578 Electrocatalysis on Copper: Potential Dependency of Activation Energy. *J. Electrochem.*
579 *Energy Convers. Storage* *17*.
- 580 54. Lobaccaro, P., Singh, M.R., Clark, E.L., Kwon, Y., Bell, A.T., and Ager, J.W. (2016). Effects
581 of temperature and gas–liquid mass transfer on the operation of small electrochemical cells for
582 the quantitative evaluation of CO₂ reduction electrocatalysts. *Phys. Chem. Chem. Phys.* *18*,
583 26777–26785.
- 584 55. Weng, L.-C., Bell, A.T., and Weber, A.Z. (2019). Towards membrane-electrode assembly
585 systems for CO₂ reduction: a modeling study. *Energy Environ. Sci.*
- 586 56. Leonard, M.E., Orella, M., Aiello, N., Roman-Leshkov, Y., Forner-Cuenca, A., and Brushett,
587 F. (2020). Editors' Choice—Flooded by Success: On the Role of Electrode Wettability in CO₂
588 Electrolyzers that Generate Liquid Products. *J. Electrochem. Soc.*
- 589 57. Guancheng, J. (2018). Chapter 1 - Concept of Gas Wettability and Research Status. In *Gas*
590 *Wettability of Reservoir Rock Surfaces with Porous Media*, J. Guancheng, ed. (Gulf
591 Professional Publishing), pp. 1–27.
- 592 58. Zhuang, T.-T., Liang, Z.-Q., Seifitokaldani, A., Li, Y., De Luna, P., Burdyny, T., Che, F.,
593 Meng, F., Min, Y., Quintero-Bermudez, R., et al. (2018). Steering post-C–C coupling
594 selectivity enables high efficiency electroreduction of carbon dioxide to multi-carbon alcohols.
595 *Nat. Catal.* *1*, 421–428.
- 596 59. Hoang, T.T., Ma, S., Gold, J.I., Kenis, P.J., and Gewirth, A.A. (2017). Nanoporous copper
597 films by additive-controlled electrodeposition: CO₂ reduction catalysis. *ACS Catal.* *7*, 3313–
598 3321.
- 599 60. Wang, Y., Wang, Z., Dinh, C.-T., Li, J., Ozden, A., Kibria, M.G., Seifitokaldani, A., Tan, C.-
600 S., Gabardo, C.M., Luo, M., et al. (2020). Catalyst synthesis under CO₂ electroreduction
601 favours faceting and promotes renewable fuels electrosynthesis. *Nat. Catal.* *3*, 98–106.
- 602 61. Li, Y.C., Wang, Z., Yuan, T., Nam, D.-H., Luo, M., Wicks, J., Chen, B., Li, J., Li, F., Garcia
603 de Arquer, F.P., et al. (2019). Binding site diversity promotes CO₂ electroreduction to ethanol.
604 *J. Am. Chem. Soc.*
- 605 62. Yang, P.-P., Zhang, X.-L., Gao, F.-Y., Zheng, Y.-R., Niu, Z.-Z., Yu, X., Liu, R., Wu, Z.-Z.,
606 Qin, S., Chi, L.-P., et al. (2020). Protecting Copper Oxidation State via Intermediate
607 Confinement for Selective CO₂ Electroreduction to C₂₊ Fuels. *J. Am. Chem. Soc.* *142*, 6400–
608 6408.
- 609 63. Li, F., Thevenon, A., Rosas-Hernández, A., Wang, Z., Li, Y., Gabardo, C.M., Ozden, A., Dinh,
610 C.T., Li, J., Wang, Y., et al. (2020). Molecular tuning of CO₂-to-ethylene conversion. *Nature*
611 *577*, 509–513.
- 612 64. Gabardo, C.M., O'Brien, C.P., Edwards, J.P., McCallum, C., Xu, Y., Dinh, C.-T., Li, J.,
613 Sargent, E.H., and Sinton, D. (2019). Continuous carbon dioxide electroreduction to

614 concentrated multi-carbon products using a membrane electrode assembly. *Joule* 3, 2777–
615 2791.

616