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Understanding surface charge regulation in silica nanopores

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Abstract: Nanoporous silica is used in a wide variety of applications, ranging from bioanalytical tools and materials for energy storage and conversion as well as separation devices. The surface charge density of nanopores is not easily measured by experiment yet plays a vital role in the performance and functioning of silica nanopores. Herein, we report a theoretical model to describe charge regulation in silica nanopores by combining the surface-reaction model and the classical density functional theory (CDFT). The theoretical predictions provide quantitative insights into the effects of pH, electrolyte concentration, and pore size on the surface charge density and electric double layer structure. With a fixed pore size, the surface charge density increases with both pH and the bulk salt concentration similar to that for an open surface. At fixed pH and salt concentration, the surface charge density rises with the pore size until it reaches the bulk asymptotic value when the surface interactions become negligible. At high pH, the surface charge density is mainly determined by the ratio of the Debye screening

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22 length to the pore size (λ_D/D).

23 **Keywords:** nanoporous silica, surface charge regulation, CDFT, electric double layer

24

25 **1 Introduction**

26 Recent progress in nanotechnology enables various applications of nanoporous silica such
27 as bioanalytical tools^{1,2,3}, separation devices⁴, or energy conversion/storage.^{5,6} Its broad
28 applicability is owing to high surface area, tunable pore size, biocompatibility, and chemical
29 stability. The surface charge of nanoporous silica is mostly negative due to
30 protonation/deprotonation of the dissociable functional groups at the solid interface.^{7,8} It has
31 been generally recognized that the electrophoretic behavior of nanoporous silica depend on
32 their surface charge densities, which vary with the solution conditions⁹ and the pore sizes.^{9,10}

33 Several investigations have been attempted to characterize the surface charge density of
34 silica nanopores through experiment. Electrokinetic techniques including streaming potential
35 or electrophoresis analysis mostly focused on the electrical surface potential (zeta potential).⁹⁻
36 ¹³ However, the interpretation of such experiments is problematic when the electrical double
37 layers within the nanopore are highly overlapped. The problem arises mainly due to the data
38 analysis based on the Helmholtz-Smoluchowski and the Gouy-Chapman equations that are
39 unreliable for such systems.^{14,15} Another method to estimate the surface charge is through
40 potentiometric titration by monitoring the number of ions and protons absorbed at the interface
41 of nanoporous silica.¹⁶⁻²⁰ Even though the charge density and the local ionic composition
42 remain elusive, it provides a quantitative connection between ion adsorption and the regulation
43 of the surface charge.

44 Theoretical modeling provides a valuable alternative to experimental investigation of the
45 surface charge regulation mechanism for nanoporous silica.²¹ Conventional methods based on
46 Poisson-Boltzmann (PB) or the Poisson-Nernst-Planck (PNP) equations have been widely used
47 due to their high efficiency in describing the electrical double layer and electroosmotic flows.²²⁻
48 ²⁷ Recent studies provided the pore level analysis on the mesoporous internal surface charge at
49 various porosities and ionic conditions.^{28,29} However, the conventional methods ignore the
50 ionic volume exclusion effects and electrostatic correlations important for concentrated
51 electrolytes in nanoscale pores.^{30,31} Such effects should be considered to understand the charge
52 regulation of silica nanopore under nanoconfinement.

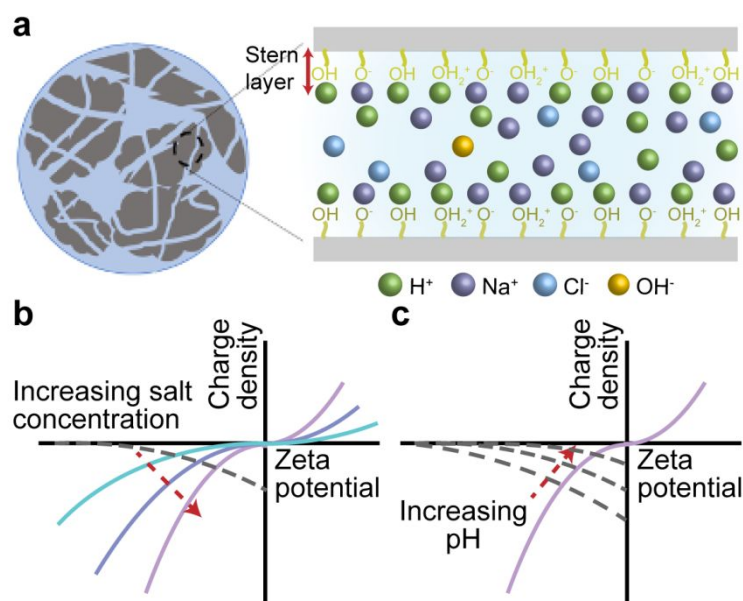
53 Increasing studies suggest that classical density functional theory (CDFT) is able to
54 describe various equilibrium and transport properties of confined ionic systems including
55 electrolytes in nanoscale porous materials.³²⁻³⁷ In this work, we combine CDFT with a surface
56 reaction model to investigate the effects of pH, electrolyte concentration, and pore size (as
57 small as 1 nm) on the surface charge density of porous silica. Whereas disordered porous silica
58 has a complex pore size distribution and diverse pore shapes, theoretical investigations in this
59 work are focused on a slit pore model that intends to capture the essential features of realistic
60 porous silica submerged in electrolytes. The theoretical results show the significant role of the
61 pore size in determining the surface charge density, in particular when it is in the range of the
62 Debye screening length. The remaining of this paper is structured as follows. First, we
63 recapitulate the thermodynamic and kinetic models and the computational scheme
64 underpinning the CDFT and the surface-equilibrium calculations. The technical details can be
65 found in our previous publications.^{38,39} Next, we discuss the theoretical predictions on the pH,

66 salt concentration, and pore size effects on the surface charge density of various silica
67 nanopores. Finally, we summarize the results. The single silt pore modeling methods could be
68 further integrated with other theories such as effective medium approximation (EMA)³⁰ for
69 understanding the charge regulation of more realistic porous silica in the future, which would
70 provide implications for possible applications.

71 **2 Model and methods**

72 ***2.1 Theoretical model***

73 We consider a generic model for a silica nanopore as shown schematically in Figure 1a.
74 The nanopore is submerged in an aqueous electrolyte solution with various ionic species
75 (throughout this work, subscript $i = 1$ stands for H^+ , $i = 2$ for Na^+ , $i = 3$ for Cl^- , and $i = 4$
76 for OH^-). As explained in the next subsection, we use the restricted primitive model to
77 investigate the effects of the pore size, electrolyte concentration, and pH on the surface
78 properties of nanoscale silica pores. Figure 1b and 1c show schematically how the surface
79 charge density and the surface electrical potential (zeta potential) are determined by a
80 combination of the electric double layer (EDL) and the surface-reaction models.



81

82 **Figure 1.** (a) Schematic of the functional groups and the electrical double layer (EDL) in a silt
 83 pore of porous silica submerged in an electrolyte solution. Panels b and c illustrate how, at a
 84 fixed salt concentration and a fixed pH, respectively, the surface charge density and the zeta
 85 (surface) potential are determined by the interception of the corresponding curves predicted by
 86 the EDL (solid lines) and the surface reaction models (dashed lines).

87 **2.2 CDFT for electric double layers**

88 Ion distributions in silica nanopores are described with the restricted primitive model
 89 (RPM) of electrolyte solutions. Both cations and anions are represented by charged hard
 90 spheres of the same size since the hydrated radius of sodium and chloride are similar to each
 91 other.⁴⁰ While atomic details are neglected, the primitive model is able to account for
 92 electrostatic correlations and mean ionic excluded volume effects important for understanding
 93 charge regulation. According to this model, the pair ionic potential is given by

$$94 \mu_i = \begin{cases} \infty, & r < (\sigma_i + \sigma_j)/2 \\ Z_i Z_j e^2 / (4\pi\epsilon_0\epsilon_r r), & r \geq (\sigma_i + \sigma_j)/2 \end{cases} \quad (1)$$

95 where r is the center-to-center distance between ions, e is the elementary charge, ϵ_r is the
 96 solvent dielectric constant, ϵ_0 is the permittivity of the free space, σ_i and Z_i are the hard-

97 sphere diameter and the valence of ionic species i , respectively. In our CDFT calculations,
 98 both anions and cations are assumed to be monovalent, the diameter of all particles (solvated
 99 ions) is fixed at $\sigma = 0.3 \text{ nm}$, and the dielectric constant for liquid water is $\epsilon_r = 78$.⁴¹ In
 100 consistent with the primitive model, the silica nanopore is represented by the slit model with
 101 pore width D and uniform surface charge density Q . The non-electrical component of
 102 interaction between the silica nanopore and ion species i is modeled as a hard-wall potential
 103 $V_i(z)$

$$104 \quad V_i(z) = \begin{cases} \infty, & z \leq \frac{\sigma_i}{2} \text{ or } z \geq D - \frac{\sigma_i}{2} \\ 0, & \text{otherwise} \end{cases} \quad (2)$$

105 where the z is the perpendicular distance from the pore surface.

106 At given temperature T and bulk ion concentrations $\rho_i^0(z)$, CDFT predicts that the ionic
 107 density distributions inside the pore are given by

$$108 \quad \rho_i(z) = \rho_i^0 \exp[-\beta V_i(z) - \beta Z_i e \psi(z) - \beta \Delta\mu_i^{ex}(z)] \quad (3)$$

109 where $\beta = 1/(k_B T)$, $T = 300 \text{ K}$, k_B is the Boltzmann constant, and $\Delta\mu_i^{ex}(z)$ accounts for
 110 electrostatic correlations and ionic excluded volume effects. The detailed equation for the local
 111 chemical potential can be found in our previous publications.³⁸³⁹ The mean electric potential,
 112 $\psi(z)$, is related to the local charge density by the Poisson equation

$$113 \quad \nabla^2 \psi(z) = -\frac{e}{\epsilon_0 \epsilon_r} \sum Z_i \rho_i(z) \quad (4)$$

114 with the following boundary conditions

$$115 \quad \psi(0) = \psi(D) = \psi_0. \quad (5)$$

116 As discussed below, the surface potential, ψ_0 , depends on the protonation/deprotonation
 117 reactions at the surface of porous silica.

118 To solve Eqs.(3-5) numerically, we start with an initial guess of the ionic density profiles,

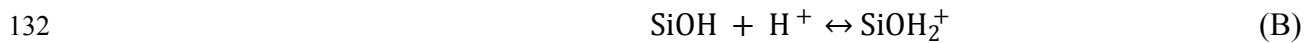
119 $\rho_i(z) = \rho_i^0$. The local excess chemical potential for each ionic species and the local electrical
 120 potential, $\Delta\mu_{ex}^i(z)$ and $\psi(z)$, are then calculated from the initial density profiles and the
 121 Poisson equation with known electric potentials at the boundaries. Next, a new set of ionic
 122 density profiles are obtained from Eq. (3), and the numerical procedure repeats until
 123 convergence ($|\frac{\Delta\rho_i}{\rho_o}| < 10^{-3}$ at all positions). From the ion distributions and the electrical potential,
 124 we calculate the surface charge density according to the overall charge neutrality condition

$$125 \quad Q = - \sum_i Z_i e \int_0^{H/2} dz \rho_i(z). \quad (6)$$

126 In our CDFT calculations, we assume a layer of surface charge at the surface of the silica
 127 nanopore (*viz.*, the center of silanol groups) where $z = 0$.

128 **2.3 Surface reaction model**

129 Charge regulation takes place at the silica surface by protonation/deprotonation reactions
 130 of the dissociable functional groups ⁷⁸:



133 Accordingly, the equilibrium constants are

$$134 \quad K_A = \frac{N_{\text{SiO}^-} [\text{H}^+]_s}{N_{\text{SiOH}}} \quad \text{and} \quad K_B = \frac{N_{\text{SiOH}_2^+}}{N_{\text{SiOH}} [\text{H}^+]_s} \quad (7)$$

135 where N_{SiOH} , N_{SiO^-} , and $N_{\text{SiOH}_2^+}$ are the number densities of SiOH, SiO⁻, and SiOH₂⁺
 136 groups at the silica surface, respectively. In Eq.(7), $[\text{H}^+]_s$ stands for the proton concentration
 137 at the silica interface. Neglecting proton interaction with other ionic species in the system, we
 138 can predict the surface density of protons from the Boltzmann equation:

$$139 \quad [\text{H}^+]_s = C_{\text{H}^+}(z)|_{z=s} = C_1 \exp\left(-\frac{z_i F \psi_0}{RT}\right) \quad (8)$$

140 where ψ_0 is the surface electrical potential, C_1 is the bulk concentration of $[\text{H}^+]$ ions,

141 which is related to the solution pH by $\text{pH} = -\log(C_{\text{H}^+})$.

142 The total number density of silanol functional groups on the solid/liquid interface consists
143 of three contributions:

$$144 \quad N_{\text{total}} = N_{\text{SiOH}} + N_{\text{SiO}^-} + N_{\text{SiOH}_2^+} \quad (9)$$

145 From Eqs. (7) - (9), we can obtain the surface charge density of the silica nanopore

$$146 \quad Q = F(N_{\text{SiOH}_2^+} - N_{\text{SiO}^-}) = -FN_{\text{total}} \frac{K_A - K_B[\text{H}^+]_s^2}{K_A + [\text{H}^+]_s + K_B[\text{H}^+]_s^2} \quad (10)$$

147 where $F = 96485 \text{ C/mol}$ is the Faraday constant. The parameters for the silica surface can
148 be found from the literature: $N_{\text{total}} = 2 \times 10^{-6} \text{ mol/m}^2$, $\text{p}K_A = -\log(K_A) = 6.8$ and p
149 $K_B = -\log(K_B) = 1.7$.^{42,43}

150 The above procedure can be calibrated with experimental data for charge regulation of
151 open silica surfaces. As shown in Fig. S1, a combination of CDFT and the surface reaction
152 model is able to reproduce the experimental results at different pH and salt concentrations. For
153 an open silica surface, similar results can be accomplished by replacing CDFT with the PB
154 equation. It suggests that our silt pore model can successfully capture the surface charge
155 properties of realistic porous silica. Whereas CDFT and PB yield similar results for this
156 particular set of data, the ion distributions obtained from CDFT calculations can be
157 significantly different from those solved from the PB equation when the pore size is
158 comparable with or smaller than the Debye screening length. For example, Figure S2 shows
159 density profiles for ion distributions in a nanopore. While the PB equation predicts an
160 unrealistic contact density at the surface, CDFT predicts a more reasonable counterion density
161 profile, with the peak position being 0.15 nm to the surface due to the ionic excluded volume
162 effects.

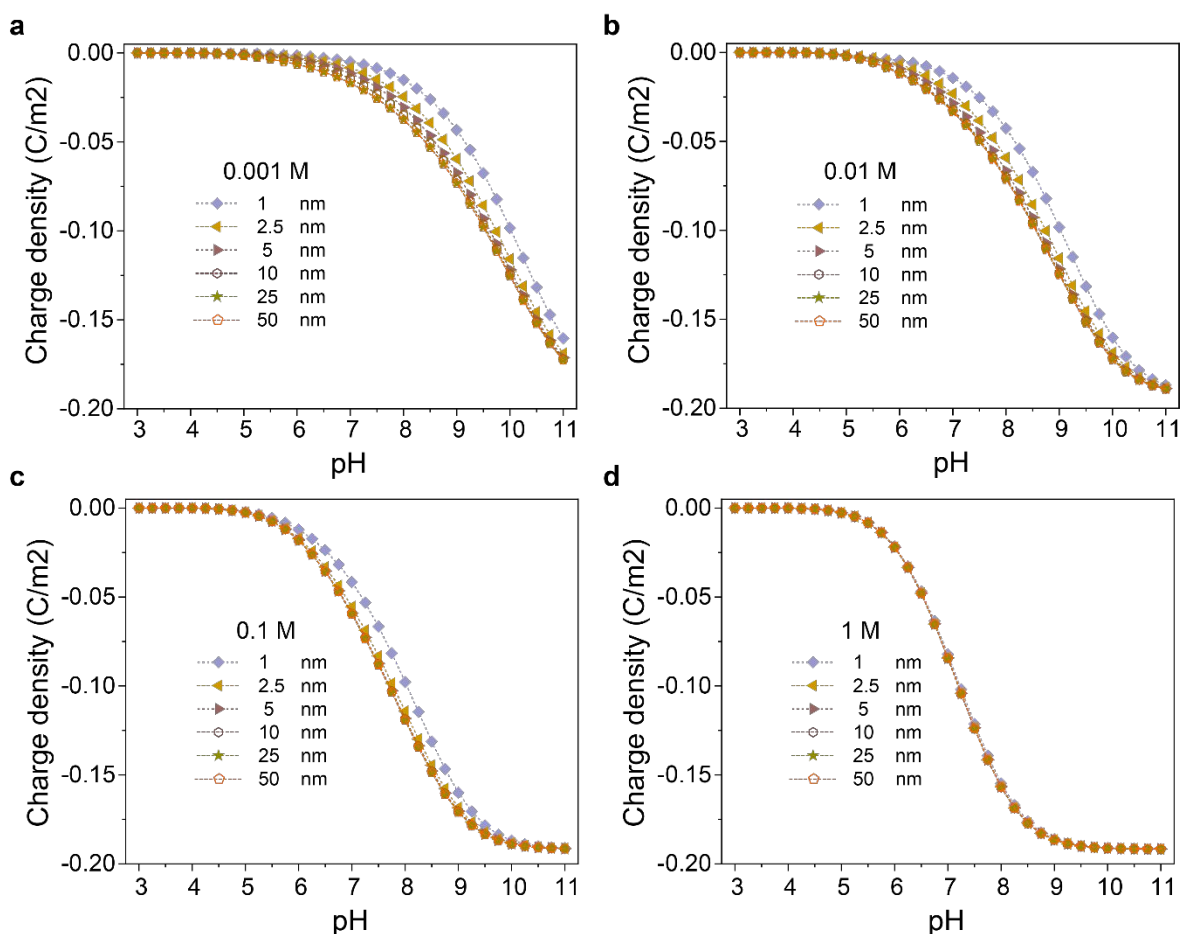
163 **3 Results and discussion**

164 The surface charge of a silica nanopore depends on solution pH as well as ionic
165 distributions. Because neither the surface charge nor the surface electrical potential is known
166 from experiment, an iterative procedure is required to calculate the ionic distributions inside
167 the pore and the surface composition self-consistently (Figure. 1). For all results discussed in
168 the following, we obtain these quantities by solving the EDL and surface-reaction models
169 simultaneously. First, we construct a series of curves for the surface charge density Q versus
170 the surface potential ψ_0 by CDFT calculations at different bulk salt concentrations, as shown
171 schematically by the violet, blue and cyan lines in Figures 1b and 1c. Next, we establish the
172 relations between Q and ψ_0 at different pH values by using the surface-reaction model. These
173 curves are shown as the grey dashed lines in Figures 1b and 1c. Finally, we obtain the surface
174 charge density and the surface electrical potential (viz., zeta potential) at the given pH and the
175 electrolyte concentration.

176 ***3.1 pH effects on the surface charge densities of silica nanopores***

177 Because of deprotonation, the surface charge density of a silica surface becomes more
178 negative as the pH increases. Such an effect is expected to be sensitive to both the pore size
179 and salt concentration because the confinement alters the surface electrical potential and the
180 local proton concentration. As shown in Figure 2, the surface charge density becomes more
181 negative as pH increases, which agrees quantitatively with the experimental result for an open
182 silica surface.⁴⁴ The absolute value of the surface charge density declines with the pore size
183 because of the surface-surface interactions. Similar results can be found in previous studies of
184 two planar electrodes, the surface charge will increase upon on the increase of the plate

185 separation.⁴⁵ As expected, the pore size effect is most significant at small salt concentrations
 186 because of the long-range electrostatic interactions.



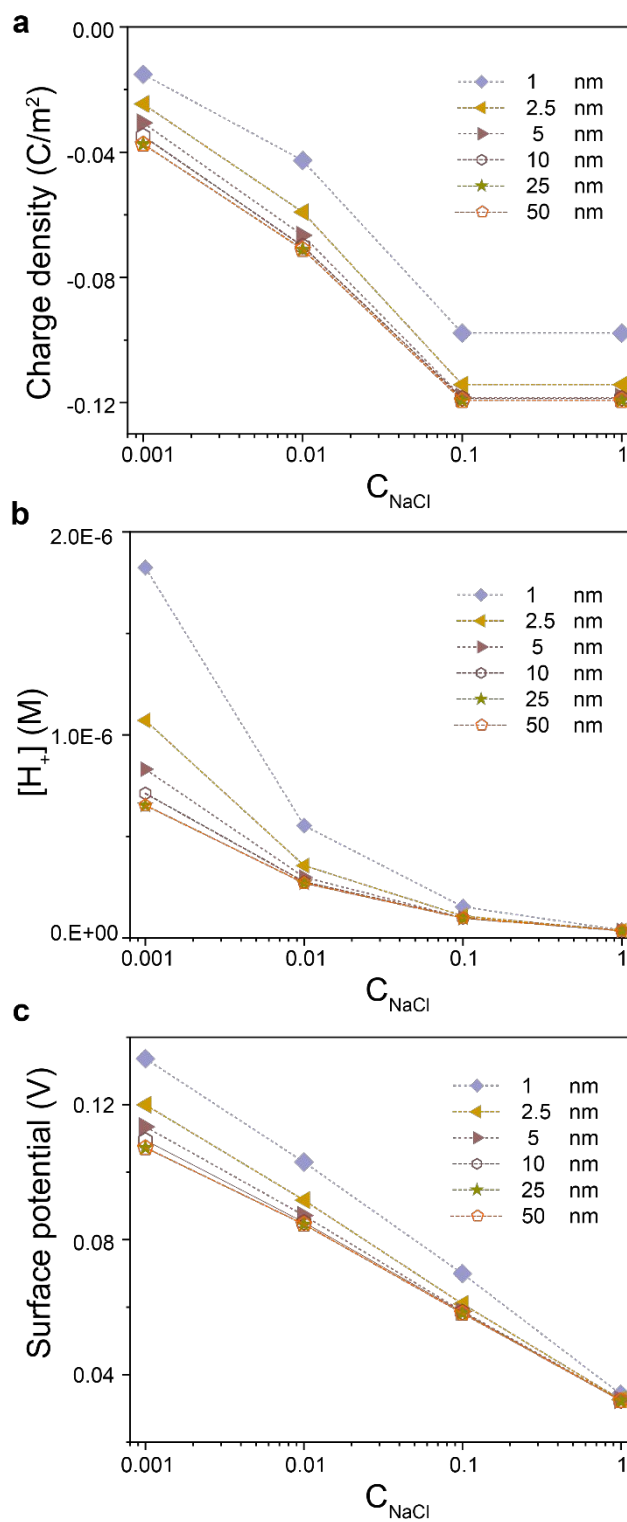
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 188 **Figure 2.** Variation of the surface charge density of silica nanopores of different pore width
 189 (legends, 1 nm -50 nm) with pH at different NaCl concentrations, 0.001M (a), 0.01M (b), 0.1M
 190 (c), and 1 M (d).

191 3.2 Salt concentration effects in different nanopores

192 Figure 3a shows how the salt concentration affects the surface charge densities of
 193 different silica nanopores. At the same pH and pore size, the absolute value of the surface
 194 charge density increases with the salt concentration because of the electrostatic screening effect.
 195 The trend is consistent with the experimental data and other theoretical results for single silica
 196 surfaces.^{46,47} Take pH = 8 and D = 50 nm, as an example, the surface charge density is about -

197 0.04 C/m^2 for $C_{NaCl} = 0.001 \text{ M}$, while it is increased to -0.16 C/m^2 for $C_{NaCl} = 1 \text{ M}$.

198 The salt concentration affects the surface charge density of silica nanopore by altering the
199 proton concentration at the surface. As shown in Figure 3b, the proton concentration at the
200 surface falls as C_{NaCl} increases, suggesting that H^+ ions are excluded from the surface because
201 of the reduction of the surface electrical potential (Figure 3c). In other words, a lower surface
202 H^+ concentration leads to more deprotonation of the silica surface thus more negative charge.
203 When the pore size of a silica nanopore increases from 1 nm to 50 nm in 0.001 M NaCl, the
204 proton concentration is reduced accordingly from $1.76 \times 10^{-6} \text{ M}$ to $6.35 \times 10^{-7} \text{ M}$. It is worth
205 noting that the surface concentration of H^+ ions varies more significantly with the pore size at
206 low salt concentrations (Figure 3b). Again, the pore size effect is most significant at low salt
207 concentrations.



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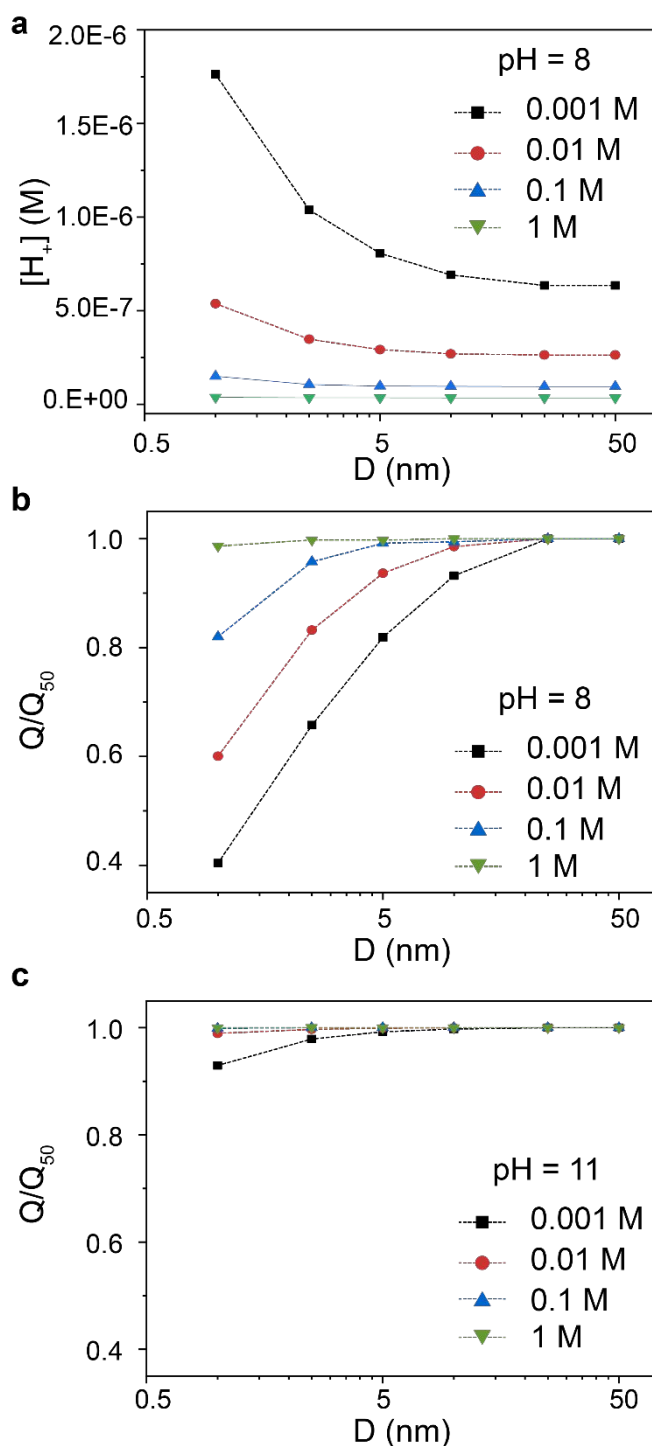
209 **Figure 3.** Variation of the surface charge density (a), the surface proton concentration (b), and
 210 the surface potential (c) with C_{NaCl} at different pore sizes. In all cases, pH = 8 and the different
 211 pore sizes are shown with different markers.

212 **3.2 The pore size effects on ionic screening**

213 From the above discussions, we see that the pore size plays a vital role in determining the
214 surface charge density of a silica nanopore at high pH and low salt concentrations. At all pH
215 values and different salt concentrations, the absolute value of the surface charge density
216 decreases with the pore width (Figure 2) because of the increased proton concentration at the
217 surface (Figure 3b). The surface charge density approaches an asymptotic limit when the pore
218 size is sufficiently large, i.e., when the EDLs from the opposite surfaces of the slit pore are
219 independent. In other words, the pore size effect becomes less important when it is larger than
220 a critical value that depends on both the pH and salt concentration. For instance, at pH = 8, the
221 critical pore size is about 25 nm at $C_{NaCl} = 0.001$ M, while it is about 1 nm at $C_{NaCl} = 1$ M.
222 The lower of the salt concentration, the larger the critical pore value.

223 As well known, the Debye screening length, defined as $\lambda_D = \sqrt{\varepsilon_0 \varepsilon_r RT / \sum_i^2 F^2 Z_i^2 C_{i0}}$,
224 provides a measure of the EDL thickness. In nanopores, both the ionic concentrations and the
225 EDL thickness will be influenced by confinement due to overlapped EDL, which has been
226 demonstrated in previous work.⁴⁸ To characterize the pore-size effects on ionic screening, we
227 present in Figure 4 the proton concentration and the surface charge density at different pH and
228 salt concentrations. Here the surface charge density is normalized by the asymptotic value
229 approximated by that for a slit pore at $D = 50$ nm, Q/Q_{50} . For the four electrolyte concentrations
230 considered in this work, the normalized surface charge density increases with the pore size at
231 all pH values. When the pH is fixed (e.g., pH=8), the normalized surface charge densities are
232 close to 1 for $D > 25$ nm. In this case, the Debye screening length is much smaller than the pore
233 size, explaining a weak influence of the pore size on the surface charge density. Interestingly,
234 the normalized surface charge density reaches the asymptotic limit at a smaller pore size ($D <$

235 5 nm) when the pH is increased to 11. For example, the normalized surface charge density is
 236 0.40 for $D = 1$ nm, $C_{NaCl} = 0.001$ M at $\text{pH} = 8$ while it is 0.93 at $\text{pH} = 11$. Similar to the
 237 explanation above, the extent of EDL overlap is relatively insignificant when $\text{pH} = 11$
 238 compared with $\text{pH} = 8$ so that the effects of pore size on surface charge is less evident. \

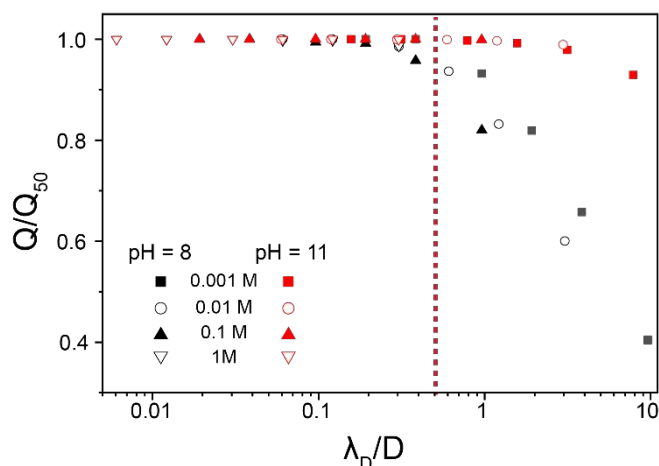


239

240 **Figure 4.** Surface proton concentration versus pore size at $\text{pH} = 8$ (a) and the normalized

241 surface charge density of a silica nanopore as a function of the pore size at pH = 8 (b) and 11
242 (c). The legends denote different NaCl concentrations.

243 To investigate the combined effects of the pore size (D) and the electrolyte concentration
244 on the surface charge density of silica nanopores, we further consider the surface charge density
245 as a function of the reduced Debye length, λ_D/D . This parameter reflecting the EDL overlap
246 has also been used in previous theoretical work (e.g. PNP).^{15,28,29} Results show that the large
247 value of λ_D/D significantly causes the surface charge density to deviate from that of an open
248 surface, which aligns to our predictions. As shown in Figure 5, the normalized surface charge
249 density is virtually invariant when the pore size is larger than the Debye length but it declines
250 sharply as the reduced Debye length, λ_D/D , is beyond a critical value that depends on both pH
251 and salt concentration. Interestingly, the reduced surface charge density is close to unity when
252 λ_D/D is smaller than 0.5 regardless of pH and salt concentration. Overall, the surface charge
253 density is weakly correlated with the silica nanopore size when $D > 2\lambda_D$, while it becomes
254 more sensitive to the pore size when $D < 2\lambda_D$ due to the EDL overlap. Notably, though
255 conventional theoretical works (e.g. PB or PNP) have reported the significant pore size effects
256 on the surface charge density when the pore size is comparable to that of the EDL dimension.
257 In this work, we identified the surface charge regulation of silica nanopore using CDFT when
258 the pore size is similar to the ion diameter as small as 1 nm, where the ion size effect becomes
259 more important and could not be ignored, as shown in Figure S1. We hope that our model would
260 provide more accurate microscopic insights into both the EDL structures inside silica nanopore
261 and the surface properties of the porous silica.



262

263 **Figure 5.** Normalized surface charge density versus the ratio of the Debye length λ_D to the264 pore size (D). Different symbols represent different electrolyte concentrations, while different

265 colors show different pH values (black for pH = 8; red for pH = 11).

266 **4 Conclusion**

267 In summary, the classical density functional theory (CDFT) is combined with a surface

268 reaction model to predict the pore size effects on the surface charge density of silica nanopores

269 at different pH and NaCl concentrations. At the asymptotic limit of large pores, the theoretical

270 predictions agree well with the experimental data for the surface charge densities

271 corresponding to planar silica surfaces or silica particles. As expected, the surface charge

272 density becomes more negative as the pH increases due to the deprotonation of the SiOH

273 functional groups. The coarse-grained model is able to capture a decrease of H^+ concentration

274 at the surface as the electrolyte concentration increases. Furthermore, the surface charge

275 density increases with the pore size and approaches an asymptotic limit as the EDL overlapping

276 becomes negligible. The pore size effect is less significant at high electrolyte concentration and

277 a large pH value.

278 Charge regulation for silica nanopores reflects a combined effect of the pore size and

279 electrolyte concentration as represented by the ratio of the Debye screening length (λ_D) to the
280 nanopore size (D). When the pore size is much larger than the Debye length, the pore size effect
281 on the surface charge density is relatively insignificant regardless of the pH and electrolyte
282 concentration. By contrast, the pore size has a strong influence on the surface charge density
283 when the pore size is close to the Debye screening length. In all cases, the absolute value of the
284 surface charge density decreases as the pore size is reduced. This work demonstrates that a
285 combination of the CDFT and a surface reaction model is able to capture the pore size effects
286 under nanoconfinement. The molecular theory provides insights into the surface properties of
287 silica nanopores useful for their broad practical applications.

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295 **REFERENCES**

- 296 1 Y. Wang and H. Gu, Core–Shell-Type Magnetic Mesoporous Silica Nanocomposites for
297 Bioimaging and Therapeutic Agent Delivery, *Adv. Mater.*, 2015, **27**, 576–585.
- 298 2 Z. Wang, Y. Liu, L. Yu, Y. Li, G. Qian and S. Chang, Nanopipettes: a potential tool for
299 DNA detection, *Analyst*, 2019, **144**, 5037–5047.
- 300 3 X. He, K. Zhang, Y. Liu, F. Wu, P. Yu and L. Mao, Chaotropic Monovalent Anion-Induced

- 301 Rectification Inversion at Nanopipettes Modified by Polyimidazolium Brushes, *Angew. Chem.*,
302 2018, **130**, 4680–4683.
- 303 4 S. J. Kim, G. Chase and S. C. Jana, The role of mesopores in achieving high efficiency
304 airborne nanoparticle filtration using aerogel monoliths, *Sep. Pur. Technol.*, 2016, **166**, 48–54.
- 305 5 W. Li, J. Liu and D. Zhao, Mesoporous materials for energy conversion and storage
306 devices, *Nat. Rev. Mat.*, 2016, **1**, 16023.
- 307 6 J. Liang, Z. Liang, R. Zou and Y. Zhao, Heterogeneous Catalysis in Zeolites, Mesoporous
308 Silica, and Metal–Organic Frameworks, *Adv. Mater.*, 2017, **29**, 1701139.
- 309 7 D. A. Sverjensky and N. Sahai, Theoretical prediction of single-site surface-protonation
310 equilibrium constants for oxides and silicates in water, *Geochim. Cosmochim. Acta*, 1996, **60**,
311 3773–3797.
- 312 8 S. H. Behrens and D. G. Grier, The charge of glass and silica surfaces, *J. Chem. Phys.*,
313 2001, **115**, 6716–6721.
- 314 9 M. Nyström, A. Pihlajamäki and N. Ehsani, Characterization of ultrafiltration membranes
315 by simultaneous streaming potential and flux measurements, *J. Membr. Sci.*, 1994, **87**, 245–
316 256.
- 317 10 T. Jimbo, M. Higa, N. Minoura and A. Tanioka, Surface Characterization of
318 Poly(acrylonitrile) Membranes Graft-Polymerized with Ionic Monomers As Revealed by ζ
319 Potential Measurement, *Macromolecules*, 1998, **31**, 1277–1284.
- 320 11 T. Jimbo, A. Tanioka and N. Minoura, Characterization of an Amphoteric-Charged Layer
321 Grafted to the Pore Surface of a Porous Membrane, *Langmuir*, 1998, **14**, 7112–7118.
- 322 12 C. Causserand, M. Nyström and P. Aimar, Study of streaming potentials of clean and

- 323 fouled ultrafiltration membranes, *Journal of Membrane Science*, 1994, **88**, 211–222.
- 324 13 M. R. Teixeira, M. J. Rosa and M. Nyström, The role of membrane charge on nanofiltration
325 performance, *J. Membr. Sci.*, 2005, **265**, 160–166.
- 326 14 M. Jia and T. Kim, Multiphysics Simulation of Ion Concentration Polarization Induced by
327 Nanoporous Membranes in Dual Channel Devices, *Anal. Chem.*, 2014, **86**, 7360–7367.
- 328 15 C.-C. Chang, R.-J. Yang, M. Wang, J.-J. Miao and V. Lebiga, Liquid flow retardation in
329 nanospaces due to electroviscosity: Electrical double layer overlap, hydrodynamic slippage,
330 and ambient atmospheric CO₂ dissolution, *Phys. Fluids*, 2012, **24**, 072001.
- 331 16 R. P. Abendroth, Surface charge development of porous silica in aqueous solution, *J. Phys.*
332 *Chem*, 1972, **76**, 2547–2549.
- 333 17 M. Karlsson, C. Craven, P. M. Dove and W. H. Casey, Surface Charge Concentrations on
334 Silica in Different 1.0 M Metal-Chloride Background Electrolytes and Implications for
335 Dissolution Rates, *Aquat. Geochem.*, 2001, **7**, 13–32.
- 336 18 A. Salis, D. F. Parsons, M. Boström, L. Medda, B. Barse, B. W. Ninham and M. Monduzzi,
337 Ion Specific Surface Charge Density of SBA-15 Mesoporous Silica, *Langmuir*, 2010, **26**,
338 2484–2490.
- 339 19 D. C. Martínez Casillas, M. P. Longinotti, M. M. Bruno, F. Vaca Chávez, R. H. Acosta
340 and H. R. Corti, Diffusion of Water and Electrolytes in Mesoporous Silica with a Wide Range
341 of Pore Sizes, *J. Phys. Chem.*, 2018, **122**, 3638–3647.
- 342 20 H. Rho, K. Chon and J. Cho, Surface charge characterization of nanofiltration membranes
343 by potentiometric titrations and electrophoresis: Functionality vs. zeta potential, *Desalination*,
344 2018, **427**, 19–26.

- 345 21 C. Lian, X. Kong, H. Liu and J. Wu, Flow effects on silicate dissolution and ion transport
346 at an aqueous interface, *Phys Chem Chem Phys*, 2019, **21**, 6970–6975.
- 347 22 H. Daiguji, P. Yang and A. Majumdar, Ion Transport in Nanofluidic Channels, *Nano Lett.*,
348 2004, **4**, 137–142.
- 349 23 H. Daiguji, P. Yang, A. J. Szeri and A. Majumdar, Electrochemomechanical Energy
350 Conversion in Nanofluidic Channels, *Nano Lett.*, 2004, **4**, 2315–2321.
- 351 24 A. P. Thompson, Nonequilibrium molecular dynamics simulation of electro-osmotic flow
352 in a charged nanopore, *J. Chem. Phys.*, 2003, **119**, 7503–7511.
- 353 25 R. Qiao and N. R. Aluru, Charge inversion and flow reversal in a nanochannel electro-
354 osmotic flow, *Phys. Rev. Lett.*, 2004, **92**, 198301.
- 355 26 R. Qiao and N. R. Aluru, Surface-charge-induced asymmetric electrokinetic transport in
356 confined silicon nanochannels, *Appl. Phys. Lett.*, 2005, **86**, 143105.
- 357 27 M. Wang and S. Chen, Electroosmosis in homogeneously charged micro- and nanoscale
358 random porous media, *J Colloid Interface Sci.*, 2007, **314**, 264–73.
- 359 28 S. Tumcan and B. Murat, Size Dependent Surface Charge Properties of Silica Nano-
360 Channels: Double Layer Overlap and Inlet/Outlet Effects, *Phys. Chem. Chem. Phys.*, 2016, **18**,
361 21–46.
- 362 29 T. Sen and M. Barisik, Internal surface electric charge characterization of mesoporous
363 silica, *Sci. Rep.*, 2019, **9**, 137.
- 364 30 C. Lian, H. Su, C. Li, H. Liu and J. Wu, Non-Negligible Roles of Pore Size Distribution
365 on Electroosmotic Flow in Nanoporous Materials, *ACS Nano*, 2019, **13**, 8185–8192.
- 366 31 R. B. Schoch, J. Han and P. Renaud, Transport phenomena in nanofluidics, *Rev. Mod.*

- 367 *Phys.*, 2008, **80**, 839–883.
- 368 32 D. E. Jiang, Z. Jin and J. Wu, Oscillation of capacitance inside nanopores, *Nano Lett.*, 2011,
369 **11**, 5373–7.
- 370 33 C. Lian, D. Jiang, H. Liu and J. Wu, A Generic Model for Electric Double Layers in Porous
371 Electrodes, *J. Phys. Chem. C*, 2016, **120**, 8704–8710.
- 372 34 D. E. Jiang and J. Wu, Unusual effects of solvent polarity on capacitance for organic
373 electrolytes in a nanoporous electrode, *Nanoscale*, 2014, **6**, 5545–50.
- 374 35 C. Lian, K. Liu, K. L. Van Aken, Y. Gogotsi, D. J. Wesolowski, H. L. Liu, D. E. Jiang and
375 J. Z. Wu, Enhancing the Capacitive Performance of Electric Double-Layer Capacitors with
376 Ionic Liquid Mixtures, *ACS Energy Lett.*, 2016, **1**, 21–26.
- 377 36 C. Lian, H. Liu, C. Li and J. Wu, Hunting ionic liquids with large electrochemical potential
378 windows, *AIChE J.*, 2019, **65**, 804–810.
- 379 37 D. Gillespie, D. N. Petsev and F. van Swol, Electric Double Layers with Surface Charge
380 Regulation Using Density Functional Theory, *Entropy*, 2020, **22**, 132.
- 381 38 Y. X. Yu, J. Wu and G. H. Gao, Density-functional theory of spherical electric double
382 layers and zeta potentials of colloidal particles in restricted-primitive-model electrolyte
383 solutions, *J Chem. Phys.*, 2004, **120**, 7223–33.
- 384 39 Y.-X. Yu and J. Wu, Structures of hard-sphere fluids from a modified fundamental-
385 measure theory, *J. Chem. Phys.*, 2002, **117**, 10156–10164.
- 386 40 B. Tansel, J. Sager, T. Rector, J. Garland, R. F. Strayer, L. Levine, M. Roberts, M.
387 Hummerick and J. Bauer, Significance of hydrated radius and hydration shells on ionic
388 permeability during nanofiltration in dead end and cross flow modes, *Sep. Pur. Technol.*, 2006,

- 389 **51**, 40–47.
- 390 41 M. Addicoat, R. Atkin, J. N. Canongia Lopes, M. Costa Gomes, M. Firestone, R. Gardas,
391 S. Halstead, C. Hardacre, L. J. Hardwick, J. Holbrey, P. Hunt, V. Ivanistsev, J. Jacquemin, R.
392 Jones, B. Kirchner, R. Lynden-Bell, D. MacFarlane, G. Marlair, H. Medhi, M. Mezger, A.
393 Padua, I. Pantenburg, S. Perkin, J. Reid, M. Rutland, S. Saha, K. Shimizu, J. M. Slattery, M.
394 Swadzba-Kwasny, S. Tiwari, S. Tsuzuki, B. Uralcan, A. van den Bruinhorst, M. Watanabe and
395 J. Wishart, Structure and dynamics of ionic liquids: general discussion, *Faraday Discuss.*, 2018,
396 **206**, 291–337.
- 397 42 M. B. Andersen, J. Frey, S. Pennathur and H. Bruus, Surface-dependent chemical
398 equilibrium constants and capacitances for bare and 3-cyanopropyldimethylchlorosilane coated
399 silica nanochannels, *J. Colloid Interface Sci.*, 2011, **353**, 301–310.
- 400 43 M. Barisik, S. Atalay, A. Beskok and S. Qian, Size Dependent Surface Charge Properties
401 of Silica Nanoparticles, *J. Phys. Chem. C*, 2014, **118**, 1836–1842.
- 402 44 J. Sonnefeld, M. Löbbus and W. Vogelsberger, Determination of electric double layer
403 parameters for spherical silica particles under application of the triple layer model using surface
404 charge density data and results of electrokinetic sonic amplitude measurements, *Colloids Surf.,*
405 *A*, 2001, **195**, 215–225.
- 406 45 M. Polat and H. Polat, Analytical solution of Poisson-Boltzmann equation for interacting
407 plates of arbitrary potentials and same sign, *J. Colloid Interface Sci.*, 2010, **341**, 178–185.
- 408 46 Z. Ovanesyan, A. Aljzmi, M. Almusaynid, A. Khan, E. Valderrama, K. L. Nash and M.
409 Marucho, Ion-ion correlation, solvent excluded volume and pH effects on physicochemical
410 properties of spherical oxide nanoparticles, *J. Colloid Interface Sci.*, 2016, **462**, 325–33.

- 411 47 S. Atalay, M. Barisik, A. Beskok and S. Qian, Surface Charge of a Nanoparticle Interacting
412 with a Flat Substrate, *J. Phys. Chem. C*, 2014, **118**, 10927–10935.
- 413 48 W. Qu and D. Li, A Model for Overlapped EDL Fields, *Journal of Colloid and Interface*
414 *Sci*, 2000, **224**, 397–407.
- 415