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Predicted CsSi Compound: A Promising Material for Photovoltaic Applications

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Exploration of photovoltaics materials has received enormous interest in a wide range of both fundamental and applied research. Therefore, in this work, we identify a CsSi compound with a Zintl phase for a promising candidate of photovoltaic materials by using global structure prediction method. Electronic structure calculations indicate that this phase possesses a quasi-direct band gap of 1.45 eV, suggesting that its optical properties could be superior to diamond-Si for capturing sunlight from the visible to the ultraviolet range. In addition, a novel silicon allotrope is obtained by removing Cs atoms from this CsSi compound. The superconducting critical temperature of this phase was estimated as a T_c of 9 K in terms of a substantial density of states at the Fermi level. Our findings represent a new promising CsSi material for photovoltaic applications, as well as a potential precursor of a superconducting silicon allotrope.

Introduction

The search for a candidate material for solar energy has attracted great attention, since it is a promising and available renewable resource for its abundant, clean, and sustainable merits.¹⁻⁵ Photovoltaics (PV), as important one for solar energy, could directly convert sunlight into electricity, where silicon-based materials are commonly accepted to be the most common PV materials.^{6,7} Compared to non-siliconbased PV materials, such as GaAs, CdS, CuInSe₂, silicon-based materials have their own advantages, including stability, nontoxicity and low cost. However, due to silicon's indirect fundamental band gap (1.1 eV) and much wider direct band gap (3.4 eV)⁸, it is still far from the optimal value (around 1.4 eV) for solar cell applications.⁹ Silicon can react with other elements to form a variety of stable binary Zintl phases, which were named after the pioneering chemist Eduard Zintl.¹⁰ Zintl phases are intermetallic compounds that contain a strongly electropositive metal, such as an alkali metal, alkaline earth metal, or lanthanoid, and a somewhat less electropositive metal (typically from the late d block or the

early p block).¹¹⁻¹⁴ In Zintl silicides, electron charge could transfer to the silicon atoms from other elements, forming a variety of sub-network structures with Si–Si covalent bonds.¹⁴⁻¹⁶ In past decades, Zintl silicides have drawn much attention owing to their excellent physical and chemical properties.¹⁷⁻³²

For Cs-Si Zintl silicides, CsSi, $Cs_{8-x}Si_{46}$ and Cs_7Si_{136} have been studied in experiments. The ambient pressure phase of CsSi is thought to adopt the KGe structure type (space group P-43n, Z=32),^{33,34} while the high-pressure phase adopts the NaPb structure type (space group /41/acd, Z=32).³⁵ Meanwhile, it is found that the Clathrate-I phase of Cs_{8-x}Si₄₆ and Clathrate-II phase of $\mathsf{Cs}_7\mathsf{Si}_{136}$ can be obtained using CsSi as a starting under high pressure or thermal material via decomposition.^{36,37} Recently, a semiconducting Im-3m symmetry CsSi₆ phase is predicted.³⁸ Moreover, alkali metal silicides are found to be important precursors in the preparation of new silicon allotropes, which can be synthesized by oxidation or thermal decomposition to remove the metal atoms.³⁹ For example, a new Cmcm-Si₆ phase with a quasi-direct band gap near 1.3eV was obtained by removing Na from Na₄Si₂₄.⁴⁰ Furthermore, an Im-3m symmetry Si allotrope with a band gap of 1.17eV is predicted to be stable after removing Cs from Im-3m CsSi₆, ³⁸ and also, ametallic P6/m-Si₆ allotrope with an interesting T_c of 12.2K at ambient pressure was proposed via a P6/m-NaSi₆ compound as a precursor.41

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As mentioned above, it is still required to search for new PV materials for science interests and potential application. In this work, we predict a new CsSi compound with a Zintl phase at ambient pressure. Electronic structure calculations reveal that it is a semiconductor with a quasi-direct band gap of 1.45 eV. Moreover, the optical properties of this phase exceeded those of diamond-Si (*dia*-Si) as indicated from the simulations. CsSi, based on our simulations, could be a precursor material to synthesize a *Immm* symmetry silicon allotrope by removing the Cs atoms. *Immm*-Si₄ is dynamically and thermally stable, and also exhibits remarkable superconductivity feature at atmospheric pressure.

Computational details

CsSi compounds are predicted at ambient pressure via the CALYPSO (Crystal structure Analysis by Particle Swarm Optimization) methodology,^{42,43} combined with firstprinciples calculations. This method has been successfully used for the prediction of new stable and mestastable materials.44-47 The *ab initio* structural relaxations and electronic properties calculations were performed within the framework of density functional theory (DFT) as implemented in the Vienna Ab-initio simulation package (VASP).48 The generalized gradient approximation (GGA) with the Perdew-Burke-Ernzerhof (PBE)⁴⁹ functional was used to approximate the electronic exchange and correlation interaction. We used projector augmented wave (PAW)⁵⁰ potentials with 5s²5p⁶6s¹ and 3s²3p² electrons treated as valence for Cs and Si atoms, respectively. An energy cutoff of 700 eV and dense Monkhorst-Pack k-meshes were used to ensure the total energies were converged to within around 1 meV/atom in the structural optimizations. Phonon calculations were performed to determine the dynamical stability of the predicted structures using a supercell approach as implemented in the PHONOPY package.⁵¹ Since the semi-local DFT PBE functional is known to underestimate the band gap of predicted structures, the hybrid Heyd-Scuseria-Ernzerhof functional (HSE06)⁵² was used to calculate the electronic and optical properties. The electron phonon coupling constant was obtained from first-principles density-functional perturbation theory as implemented in the QUANTUM ESPRESSO package.53 Ultrasoft pseudopotentials and an energy cutoff of 80 Ry were employed, with a $3 \times 5 \times 5 q$ -grid and $12 \times 20 \times 20$ k-grids. The broadening for which was converged to within 0.001 was 0.04Ry.

Results and discussion

Herein, the computational prediction of novel CsSi structures was carried out using the particle swarm method with simulation cells of up to four formula units at ambient pressure. Our structure search identified a previously unknown orthorhombic CsSi Zintl phase (space group *I*-4, 16 atoms/cell), as shown in Fig. 1. In our predicted *I*-4-CsSi structure, silicon atoms form isolated tetrahedra,



Fig. 1 Predicted CsSi structure. (a) A polyhedral view. (b) A view along the c

corresponding to the typical characteristics of Zintl monosilicides with an alkali metal. At ambient pressure, the optimized lattice parameters are a = 10.107Å and c = 7.022Å, with Cs atoms occupying the 8g (0.112, 0.259, 0.909) and Si atoms occupying the 8g (0.115, 0.96, 0.619) Wyckoff sites. A Bader charge analysis⁵⁴ revealed that 0.68e are transferred from the Cs atoms to the Si atoms in *I*-4-CsSi.

The phase stability of I-4-CsSi was investigated by calculating the enthalpy of formation, which is given via ΔH = [H(CsSi) - H(Cs) - H(Si)]/2, where H(CsSi) is the enthalpy of I-4-CsSi, H(Cs) is the enthalpy of elemental bcc Cs, and H(Si) is the enthalpy of elemental Si in the diamond structure. The formation enthalpies of the known P-43n and I41/acd-CsSi phases were also considered in our simulations. The enthalpy of I41/acd-CsSi is lower than that of the P-3n structure above 3.3 GPa, indicating that the /41/acd phase becomes stable above this pressure, which is consistent with the experimental preparation of /41/acd-CsSi at 4 GPa.³⁵ Our results revealed that ΔH of I-4-CsSi is -0.032 eV/ atom at ambient pressure, which is 27 and 50 meV/atom higher than that of I41/acd and P-43n-CsSi, respectively. To investigate the dynamic stability of I-4-CsSi, we computed the phonon dispersion curves at ambient pressure, as shown in Fig.2a. There is no imaginary frequency in the whole Brillouin Zone, demonstrating the dynamically stable nature of our predicted I-4-CsSi. The I-4-CsSi compound is thus a low-lying metastable phase. With increasing the pressure to 49.2 GPa, I-4-CsSi becomes enthalpically more favorable than P-43n-CsSi, although *I*41/*acd* still remains the most stable phase.



Fig. 2 Calculated (a) phonon dispersion, and (b) band structure of *I*-4-CsSi using the PBE (black lines) and HSE06 (red lines) functionals.

For materials used in PV devices, the band gap is a critical parameter because the conversion efficiency from sunlight into electrical power is a function of the band gap. The electronic band structure of I-4-CsSi was calculated, as shown in Fig. 2b. It is well known that the PBE functional normally underestimates the band gap, while the screened hybrid functional of Heyd, Scuseria, and Ernzerhof (HSE) usually gives more accurate electronic band gaps. I-4-CsSi was calculated to have a band gap of 1.42 eV with the HSE06 functional, and 0.92 eV with the PBE functional, from the minimum of the conduction band at the M-point to the maximum of the valence band close to the Γ -point along the Γ-Z direction. The direct band gap of I-4-CsSi, which is located at the M-point, was 1.45 eV with the HSE06 functional (0.93 eV with PBE), indicating that I-4-CsSi is a semiconductor with a quasi-direct band gap.

To further explore the potential applications for photovoltaic devices, we simulated the optical absorption spectrum of I-4-CsSi using the HSE06 functional, as shown in Fig. 3. The calculated absorption spectrum of the dia-Si phase is also simulated for comparison with the reference air mass (AM) 1.5 solar spectral irradiance.⁵⁵ For *I*-4-CsSi, the absorption of low-energy photons starts from ~1.5 eV, which is close to its band gap. The I-4-CsSi phase exhibits considerably stronger absorption than dia-Si in the visible light range of 1.7-3.2eV, and the absorption spectrum overlaps significantly with the solar spectrum. The decrease of the direct optical band gap as compared to that of dia-Si is beneficial for improving the absorption, thus, the photo current of the solar cell. This suggests that the I-4-CsSi phase could potentially be employed in highly efficient Si-based solar cells.

It is well known that Zintl phases play a critical role as precursors for the synthesis of Si and Ge allotropes or other



metastable compounds using chemical oxidation and thermal decomposition. Thus, the Cs atoms in the *I*-4-CsSi structure were removed to obtain the pure silicon allotrope. After fully structural optimization, the phase evolved into a new orthorhombic silicon allotrope with the *Immm* space group (Fig. 4a). This *Immm*-Si₄ phase contains open channels formed from six-membered Si rings along the crystallographic *c* axis, and differs markedly from the tetrahedral motifs present in the *I*-4-CsSi structure.

The silicon sublattice of the product differs from that of the precursor has been found experimentally in the formation of Ge allotropes by oxidation of Na₁₂Ge₁₇ or Na₄Ge₉.^{56,57} The Na atoms template the formation of the Ge₁₃₆ framework as the $[Ge_4]^{4-}$ and $[Ge_9]^{4-}$ polyanions are oxidized and are then emptied from the clathrate framework cages as the reaction progresses.³⁹ The transformation of polyanionic clusters into the Ge₁₃₆ framework requires not only new Ge-Ge bonds to be formed but also existing Ge–Ge bonds to be broken. ³⁹ At ambient pressure, the optimized lattice parameters of *Immm*-Si₄ are a = 6.195, b = 3.248, and c = 3.496 Å, with Si atoms occupying the 4e (0.191, 0.5, 0.5) Wyckoff sites. The bond lengths of Si₁-Si₂ and Si₁-Si₃ are 2.495 and 2.368 Å, respectively (Fig. 4a).

The total energies of the experimentally known *dia*-Si, R8,^{58,59} *Cmcm*-Si₆,⁴⁰ and *Fd*-3*m*-Si₁₃₆⁶⁰ phases, and the theoretically predicted *Immm*-Si₄, and *P6/m*-Si₆⁴¹ phases of silicon were calculated. Fig.4b shows that *dia*-Si is the most thermodynamically stable phase, as expected. The total energies of *Cmcm*-Si₆ and *P6/m*-Si₆ are 0.09eV/atom and 0.35eV/atom higher than that of *dia*-Si, respectively, which is in good agreement with previous computations.⁴¹ Moreover, the energy of *Immm*-Si₄ is lower than that of *Fd*-3*m*-Si₁₃₆, a phase that has been synthesized using thermal decomposition.⁶⁰ Fig. 4c shows no imaginary frequencies are observed in *Immm*-Si₄ throughout the whole Brillouin zone, indicating that it is dynamically stable at ambient pressure.





Fig. 4 (a) The crystal structure of Immm-Si₄. (b) Energy vs volume curves of various silicon allotropes. (c) Phonon dispersion, Eliashberg spectral function, and electron phonon coupling integral, as well as (d) The band structure and total densities of states of the Immm-Si₄ phase at ambient pressure.

To further understand the electronic properties, we calculated the band structure and density of states of Immm-Si₄, as shown in Fig. 4d. The results indicate that Immm-Si₄ exhibits metallic behavior. The previously reported pure Si allotropes are almost semiconductors except the recently proposed metallic P6-m Si phase with T_c of 12.2 K at ambient pressure.⁴¹ Therefore, we calculated the electron-phonon coupling constant, λ , from first-principles density-functional perturbation theory, and the superconducting critical temperature, T_c , was estimated using the Allen-Dynes-McMillan formula,61,62 with a Coulomb modified pseudopotential μ^* of 0.1. The logarithmic average of phonon frequencies, $\omega_{\text{log}},$ and λ obtained from the Eliashberg spectral function, $\alpha^2 F(\omega)$, were 259 K and 0.70, respectively. Our results predict that the Immm-S_{i4} allotrope is a superconducting phase with an estimated T_c of 9 K at ambient pressure.

Conclusions

In summary, we predicted an *I*-4-CsSi compound via crystal structure search carried out at ambient pressure. Compared to the known CsSi phases, it is thermally metastable and dynamically stable. Electronic structure calculations indicate that *I*-4-CsSi has a quasi-direct band gap of 1.45 eV, which perfectly matches the optimal value of ~1.4 eV for solar cell applications. The calculated optical absorption of *I*-4-CsSi reveals that its optical properties are superior to those of *dia*-Si. Moreover, a novel silicon allotrope of *Immm*-Si₄ was obtained by removing Cs atoms from the *I*-4-CsSi compound

followed by structural relaxation. *Immm*-Si₄ is dynamically stable at ambient pressure with an estimated T_c value of 9 K. *Immm*-Si₄ is 0.05eV/atom more stable than the previously synthesized *Fd*-3*m*-Si₁₃₆ phase, indicating that it is likely to be synthesized in the lab. Our current results represent that *I*-4-CsSi is not only a new promising material for photovoltaic applications, but also a potential precursor of a novel superconducting *Immm*-Si₄ allotrope.

Conflicts of interest

There are no conflicts of interest to declare.

Acknowledgements

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