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Abstract: ABO₃ perovskites are increasingly being explored as catalysts, but it is unclear how they behave as supports for single atoms and how the subsequent single-atom catalysts can be employed for important reactions such as methane activation. Here we examine the stability of Pt single atom (Pt₁) on the commonly exposed (100) surfaces of SrBO₃ perovskites (B = 3d transition metals) and its methane-adsorption property by first principles density functional theory. We find that the stability and charge state of Pt_1 on the $SrBO_3(100)$ surfaces are terminationsensitive. Due to polar compensation, Pt_1 is negatively charged on the A termination but positively charged on the B termination. This charge state greatly impacts methane adsorption: negatively charged Pt₁ on the A-termination chemisorbs methane (in some cases, dissociatively), but positively charged Pt_1 on the B-termination adsorbs methane physically. Analysis of density of states of the negatively charged Pt_1 reveals that its sp states are key to methane chemisorption and C-H activation. Our work shows that polar compensation on the perovskite surfaces can be used to tune the charge state of a single atom for methane chemisorption and C-H activation.

1. Introduction

Single-atom catalysts (SACs) are a prominent class of heterogeneous catalysts due to their unique activity and atomic efficiency.¹⁻³ In most cases, noble-metal atoms are atomically dispersed on an oxide support, such as Pt_1 on CeO_2 .⁴⁻⁶ SACs also include surface-dispersed mononuclear organometallic catalysts,⁷ micropore-confined single atom catalyst, 8 and atomically dispersed metal atoms on a metal surface⁹ or a nanoparticle.¹⁰ Moreover, SACs have also been increasingly used for electrocatalysis¹¹ and organic chemistry.¹²

The most studied oxide supports for SACS are binary oxides including FeO_x , $Al₂O₃$, TiO₂ and $CeO₂.^{4, 13-15}$ A great deal of efforts have been devoted to understanding the interaction between the single atoms and the oxide supports, to avoid sintering of the single atoms into larger particles.¹⁶⁻ ¹⁹ On the other hand, more complex oxides have been explored as supports to tune the interactions.20-22

Perovskites (ABO₃) are the most common type of ternary oxides.^{23, 24} The great variability in choices of A and B serves as an ideal test ground for probing their interactions with a single atom.25, ²⁶ Recently, there has been great interest in using perovskites for catalysis and understanding the structure-activity-selectivity relationships for various facets and terminations of perovskites.²⁷⁻²⁹

 $CH₄$ activation and conversion on SACs have attracted interest lately.³⁰⁻³² For example, it was reported that Pt/Cu(111) single-atom alloys can activate C-H bond while efficiently preventing coke formation.³³ And density functional theory (DFT) predicted that single atoms such as $Ir₁$ and $Pt₁$ substitutionally doped on the (110) surface of rutile oxides can chemisorb methane, leading to a facile C-H activation.34, 35

From the perspective of CH_4 adsorption and activation on SACs, we think that a great opportunity awaits in exploring how the single atoms interact with perovskites and how the resulting SACs adsorb and activate methane. Hence, herein we use $Pt₁$ as a prototypical single atom and Sr-based perovskite $(SrBO₃)$ as the support, to examine from first principles the effect of varying B ions on the interaction between Pt_1 and $SrBO_3$ and the subsequent impact on CH_4 adsorption. Below we first describe the DFT method employed.

2. Computational method

The Vienna Ab Initio Simulation Package (VASP) was used to perform spin-polarized density functional theory calculations in the generalized-gradient approximation (GGA) with PerdewBurke-Ernzerhof (PBE) functional.³⁶⁻³⁸ (Here we did not employ any dispersion correction, so that we could clearly distinguish chemisorption and physisorption of methane, a strategy employed previously;³⁴ including the dispersion correction did not change our conclusions.) Electronic wave functions were expanded in plane waves with a cutoff energy of 450 eV and the core electrons with the nucleus were approximated by the projector augmented-wave (PAW) method.³⁹ The convergence criteria were 10^{-4} eV and 0.01 eV/Å for energy and force, respectively.

As in previously pursued strategies, $25, 26, 29$ we focus on the two main terminations of the SrBO₃ (100) surface, one of the most stable and commonly exposed surfaces. The (100) facet of SrBO₃ in either A- or B-termination was modelled as a 2×2 supercell with a vacuum space of 15 Å along *z* direction. The surface slabs have six layers, with the bottom four layers fixed in their bulk positions, and were sampled by a $3\times3\times1$ k-point mesh in the Monkhorst-Pack Scheme.⁴⁰ The interaction of Pt_1 with the surface was determined by the Pt_1 adsorption energy

$$
E_{ads-Pt1} = E_{Pt1 - ABO_3} - E_{Pt1} - E_{ABO_3},
$$

where $E_{Pt1 - ABO_3}$, E_{Pt1} and E_{ABO_3} represent energies of ABO₃-supported Pt₁ single-atom catalyst, a Pt atom in the gas phase, and the ABO₃ surface, respectively. The same approach was used to compute methane adsorption energy on ABO_3 -supported Pt_1 single-atom catalysts. O vacancy formation energy (E_{O-VFE}) was calculated with the following equation:

$$
E_{O-VFE} = E_{defect - ABO_3} - \frac{1}{2}E_{O_2} - E_{ABO_3},
$$

where $E_{defect-AlBO_3}$ and E_{O_2} represent the calculated energies of ABO₃ surface with an O vacancy and a gas phase O_2 molecule, respectively.

The charge-density difference was calculated with the following equation:

$$
\Delta \rho = \rho_{Pt_1 - SrTiO_3} - \rho_{SrTiO_3} - \rho_{Pt_1},
$$

where $\rho_{Pt_1-SrTiO_3}$, ρ_{SrTiO_3} and ρ_{Pt_1} represent the calculated electron density of Pt₁-SrTiO₃, SrTiO₃, and Pt_1 , respectively.

3. Results and discussion

Sr-based perovskites have been widely studied both from experiment and theory,²⁵ so we chose Sr^{2+} as the A cation in this work. We vary the B cation in $SrBO_3$ in 3d transition metals (B = Ti, V, Cr, Mn, Fe, Co, Ni, Cu). (100) facets are usually the most stable facets of perovskite (ABO₃).^{41,} 42 In this SrBO₃ (100) system, we first study the adsorption of Pt single atom on SrBO₃ (100) surface to evaluate the interaction between $Pt₁$ and perovskite oxide.

3.1. Adsorption of Pt single atom on SrBO3 (100)

The $\rm SrBO_3(100)$ facet can have either the A- or B-termination. The A-termination exposes the SrO layer (Fig. 1a) while the B-termination exposes the $BO₂$ layer (Fig. 1b). On A-termination, we found that Pt₁ prefers to locate at the top of a surface oxygen atom (Fig. 1c), while on B-termination, $Pt₁$ is most stable at the hollow site and coordinated by four O (Fig. 1d).

Fig. 1. Typical structures of $SrBO_3(100)$ and $Pt_1-SrBO_3(100)$: (a) $SrBO_3(100)$ A-termination; (b) SrBO₃(100) B-termination; (c) Pt₁ on SrBO₃(100) A-termination; (d) Pt₁ on SrBO₃(100) B-termination. Color mode: O, red; Sr, green; B, gray; Pt, blue.

Fig. 2. Adsorption energies of Pt single atom on $SrBO_3(100)$ surfaces, for B = Ti, V, Cr, Mn, Fe, Co, Ni, Cu.

The Pt₁ adsorption energies (relative to a gas-phase Pt atom) on the two $SrBO₃(100)$ terminations with varying B are plotted in Fig. 2. Interestingly, one can see that on A-termination, $Pt₁$ adsorption energy is relatively constant, fluctuating slightly around -3.0 eV, with varying B cations. In contrast, the Pt_1 adsorption energy on the B-termination increases in magnitude from -3.09 eV on SrTiO₃(100) to -9.22 eV on SrCuO₃(100). Apparently, the changing B ion has a much greater impact for Pt_1 interaction on the B-termination than on the A-termination. This is consistent with the difference in the Pt₁ adsorption geometry on the two terminations (Fig. 1c, d): the formation of 4 Pt-O bonds in the B-termination suggests that the Pt adsorption on the B termination will be much more impacted by the B ion. Below we further examine this difference by correlating with descriptors that vary with the B ion.

3.2. Correlation with the vacancy formation energy

Commonly used descriptors for oxides and perovskites in terms of their surface reactivity include O-vacancy formation energy (O-VFE), H adsorption energy, and surface-oxygen Bader charges.^{19, 43, 44} To understand the trend of Pt adsorption on the B-termination (Fig. 2), we calculated surface O-VFE of the $SrBO₃(100)$ B terminations. As one can see from Fig. 3, the O-VFE correlates well with Pt adsorption energy. In other words, as B changes from Ti to Cu, the surface becomes more reducible and the surface O becomes more reactive, leading to stronger interaction of Pt on the surface.

Fig. 3. Correlation between Pt single atom adsorption energy and surface O vacancy formation energy on $SrBO₃(100)$ B termination. The red line represents the best linear fit.

3.3 Charge distribution on Pt1 atom supported by SrBO3 (100)

Since charge state is an important indicator of the reactivity of a single atom site, we have examined the Bader charge of the Pt single atom on $S₁BO₃$ (100) in preparation for examining their methane-adsorbing ability.⁴⁵ Interestingly, Fig. 4 shows that Pt atoms are negatively charged on the A-termination but positively charged on the B-termination (Bader charges of other surface atoms are shown in Table 1 after/before Pt adsorption). When B cation changes from Ti to Cu, the Pt charge increases: becoming more positive on the B termination and less negative on the A termination.

Fig. 4. Partial atomic charge of Pt single atom on $SrBO₃(100)$.

Table 1. Bader charges (in $|e|$) of Pt₁, O atoms bound to Pt₁, and nearest A and B ions to Pt₁ on the surface after/before Pt adsorption on the A- and B-terminations of $\rm SrBO_3(100)$.

B in SrBO ₃	A-termination			B -termination		
	Pt ₁	O	Sr	Pt ₁	O	B
Ti	$-0.39/0.00$	$-1.04/-1.27$	1.57/1.58	0.48/0.00	$-1.01/-1.11$	1.85/1.91
$\mathbf v$	$-0.60/0.00$	$-1.02/-1.25$	1.57/1.57	0.52/0.00	$-1.00/-1.05$	1.73/1.77
Cr	$-0.39/0.00$	$-1.02/-1.21$	1.56/1.57	0.60/0.00	$-0.96/-0.99$	1.66/1.71
Mn	$-0.28/0.00$	$-0.95/-1.18$	1.56/1.57	0.70/0.00	$-0.91/-0.95$	1.60/1.65
Fe	$-0.35/0.00$	$-0.97/-1.30$	1.57/1.62	0.70/0.00	$-0.92/-0.94$	1.50/1.53
Co	$-0.28/0.00$	$-1.33/-1.16$	1.37/1.56	0.84/0.00	$-0.86/-0.83$	1.32/1.37
Ni	$-0.10/0.00$	$-0.81/-1.10$	1.57/1.57	0.98/0.00	$-0.79/-0.78$	1.19/1.24
Cu	$-0.06/0.00$	$-0.78/-1.10$	1.57/1.57	1.00/0.00	$-0.78/-0.77$	1.15/1.17

To pinpoint how the electron transfer between Pt single atom and the surface, we use $SrTiO₃$ (100) as an example and analyzed the charge-density difference after supporting the Pt single atom. As one can see from Fig. 5, Pt gains electron from the O atom underneath and becomes negatively charged on the A-termination (Fig. 5a), but on the B-termination, the charge transfer is more complex (Fig. 5b) and overall Pt_1 loses some electron density while Ti gains.

Fig. 5. Charge-density difference plots of Pt_1 -SrTiO₃ (100): (a) A-termination; (b) B-termination. Blue denotes depletion of electron density while yellow represents accumulation. Color code: O, red; Sr, green; Ti, gray. Pt₁ is at the center of the charge transfer. Isosurface value equal to 0.0038 e/ \AA ³.

The different charges of Pt₁ on A- and B-terminations of $SrBO₃(100)$ can be understood by polar compensation.⁴⁶⁻⁵¹ Using SrTiO₃(100) as an example, Fig. 6 shows that the A-termination is slightly positively charged, while the B-termination is negatively charged. This is in agreement with previous theoretical finding that the $SrTiO₃(100)$ surfaces are slightly polar.⁵² To compensate the surface charges, Pt_1 should have negative charge on the A-termination and positive charge on the B-termination. That is exactly what we have found from Fig. 4.

Fig. 6. Charge of surface layers on $SrTiO₃(100)$ for A-termination and B-termination. The surface-layer charge is determined by the sum of the Bader charges of all the atoms in the top layer of the surface before Pt₁ adsorption (Table 1).

Fig. 7. CH₄ adsorption energy on Pt_1 -SrBO₃ (100).

3.4. CH4 adsorption on Pt1-SrBO3(100)

We next examine CH₄ adsorption on Pt_1 -SrBO₃ (100). Fig. 7 shows that on the B-termination, adsorption of CH_4 on Pt_1 -SrBO₃ (100) is physical in nature and weak, but on the A-termination, $CH₄$ chemisorbs strongly with adsorption energies in the range of -0.8 to -1.1 eV. To understand this strong adsorption, we further inspected the optimized geometry of the adsorbed CH4. As shown in Fig. 8, for B being the early transition metals, such as in $SrTiO₃$, $SrVO₃$ and $SrCrO₃$, dissociative adsorption happens, leading to co-adsorbed H and CH_3 on Pt₁. For B = Mn to Cu, one or two C-H bonds of CH_4 are significantly activated and elongated, though not broken, in the chemisorbed state (Fig. 8).

The spontaneous CH₄ dissociation that we found on systems such as Pt_1 -SrTiO₃-(100)-Atermination is consistent with several recent studies of CH₄ activation by gas phase atom or clusters. For example, Perera *et al.*⁵³ found that the insertion of a gas-phase Pt atom into the C-H bond of $CH₄$ is barrierless to form $CH₃$ -Pt-H, based on a joint vacuum ultraviolet photoionization and DFT investigation. More relevantly, Zhao *et al.* found from mass spectrometry and DFT calculations that small gas-phase anionic clusters such as $PtAl_2O_4^-$ and $AuTi_3O_7^-$ can activate CH₄.^{54, 55}

Fig. 8. Optimized configuration of CH₄ adsorption on Pt₁-SrBO₃ (100) A-termination. Color code: O, red; Sr atom, green; Pt atom, blue; C atom, black; H atom, white; B, varies. Key bond distances are also given for the C-H bond being activated: Pt-C (left), Pt-H (lower right), and C-H (upper right).

To understand why CH₄ adsorbs strongly on Pt_1 -SrBO₃ (100) A-termination, we plotted the local density of states (DOS) of Pt_1 on $SrTiO_3(100)$ (similar results have been found for other strong-interaction cases such as $B = V$ and Cr). As one can see, there are empty 6s and 6p states (especially p_z) of Pt₁ right above the Fermi level (Fig. 9a). These states serve as acceptor for σ donation from the C-H bond of CH₄ and have much greater change after adsorption. As revealed previously,^{34, 35} the key to methane chemisorption from the d states is the empty d_z^2 orbital, but it is occupied in the case of Pt_1 -SrTiO₃-(100)-A (Fig. 9b). Hence, the empty sp states on Pt_1 -SrBO₃-(100)-A play the dominant role in methane chemisorption. In contrast, the weak methane chemisorption on Pt₁-SrTiO₃-(100)-B (Fig. 7) can be attributed to the lack of either empty sp states (Fig. 10a) or empty d_z^2 states near the Fermi level (Fig. 10b).

Fig. 9. Projected density of state plots of Pt single atom supported on SrTiO₃ (100) A-termination before $CH₄$ adsorption: (a) 6s states; (b) 6p states; (b) 5d states. The Fermi level is set to 0 eV.

Fig. 10. Projected density of state plots of Pt single atom supported on SrTiO₃ (100) B-termination before $CH₄$ adsorption. (a) s, p states; (b) d states. The Fermi level is set to 0 eV.

3.5. Implications for methane activation

Considering both the stability (Figure 2) and methane adsorption (Figure 7), one can see that the A terminations of $SrTiO₃(100)$ and $SrVO₃(100)$ are the most promising to realize methane chemisorption and facile activation at the Pt_1 site, because Pt_1 has about equal stability on the A and B terminations of those two surfaces. For the (100) surfaces of the other perovskites explored, $Pt₁$ strongly prefers the B termination where methane only physisorbs at the $Pt₁$ site; but as shown previously,²⁴ the perovskite surfaces can reconstruct at various pretreatment methods and at certain conditions the A termination can be exposed. In this case, Pt_1 can be anchored on the A termination to test methane activation. Fig. 2 shows that the adsorption energy of $Pt₁$ on the A terminations is about -3.0 eV relative to a gas-phase Pt atom, while bulk Pt has a cohesive energy of 5.5 eV/atom from our DFT calculation. This means that Pt_1 on the A terminations is much less stable than bulk Pt and may sinter at high temperatures, which brings challenges in synthesis. On the other hand, $Pt₁$ may be kinetically stabilized on the A termination. We have investigated minimum-energy diffusion pathways of Pt₁ on the A- and B-terminations of $SrTiO₃(100)$, as an example. We found that the diffusion barrier is actually higher on the A-termination ($E_a = 1.37$ eV) that on the Btermination ($E_a = 1.18$ eV).

4. Conclusions

In summary, we have studied methane adsorption on the Pt single atom stabilized on the (100) surfaces of SrBO₃ perovskites (B = 3d transition metals) by first principles density functional theory. We found that as B varies from Ti to Cu, the binding energy of Pt_1 with the A (or SrO) termination is about the same, but the binding of Pt_1 with the B (or BO_2) termination becomes stronger which correlates with the increasing reducibility as measured by the oxygen-vacancy formation energy. More interestingly, Pt_1 is negatively charged on the A termination but positively charged on the B termination, due to polar compensation. This charge state has a profound impact on methane adsorption: methane adsorbs strongly, chemically, and, in some cases, dissociatively on Pt₁-SrBO₃(100)-A-termination, but weakly and physically on Pt₁-SrBO₃(100)-B-termination. Analysis of local density of states revealed that the sp states of the negatively charged $Pt₁$ is key to methane chemisorption and C-H activation. Our work shows that negatively charged $Pt₁$ single

atom on the A termination of the $SrBO₃(100)$ has the potential to achieve facile C-H activation of methane.

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TOC graphic

Due to polar compensation, Pt_1 is negatively charged on the A termination of $SrBO₃(100)$ surfaces but positively charged on the B termination, which greatly impacts methane adsorption