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Journal:	Soft Matter
Manuscript ID	SM-ART-12-2019-002414.R2
Article Type:	Paper
Date Submitted by the Author:	25-Mar-2020
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# How clay particulates affect flow cessation and the coiling stability of yield stress-matched cementing suspensions

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### 7 ABSTRACT

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9 The remarkable increase in the flow resistance of dense suspensions can hinder 3D-printing processes on account of flow cessation in the extruder, and filament fragility/rupture following 10 11 deposition. Understanding the nature of rheological changes that occur is critical to manipulate 12 flow conditions or to dose flow modifiers for 3D-printing. Therefore, this paper elucidates the influences of clay particulates on controlling flow cessation and the shape stability of dense 13 14 cementing suspensions that typically feature poor printability. A rope coiling method implemented with varying stand-off distances was used to probe the buckling stability and 15 tendency to fracture of dense suspensions that undergo stretching and bending during 16 17 deposition. The contributions of flocculation and short-term percolation due to the kinetics of 18 structure formation to deformation rate were deconvoluted using a stepped isostress method. 19 It is shown that the shear stress shows a divergence with a power-law scaling when the particle volume fraction approaches the jamming limit;  $\phi 
ightarrow \phi_{J} 
ightarrow \phi_{max}$ . Such a power-law divergence of 20 21 the shear stress decreases by a factor of 10 with increasing clay dosage. Such behavior in clay-22 containing suspensions arises from a decrease in the relative packing fraction ( $\phi/\phi_{max}$ ) and the formation of fractally-architected aggregates with stronger interparticle interactions, whose 23 uniform arrangement controls flow cessation in the extruder and suspension homogeneity, 24 25 thereby imparting greater buckling stability. The outcomes offer new insights for 26 assessing/improving the extrudability and printability behavior during slurry-based 3D-printing 27 process. 28 29 Keywords: rope coiling, rheology, buckling stability, flow cessation, dense suspension, fractal 30 structuring. 31 32 33 **1. INTRODUCTION AND BACKGROUND** 34 To create materials that are amenable to 3D-printing, it is advantageous to combine powder 35 feedstocks with a carrier liquid (often water) to form a concentrated suspension ("slurry").<sup>1,2</sup> The suspension can then be printed layer-by-layer (i.e., in the form of a filament) by extrusion 36 through a printer nozzle assisted by air-pressure, a ram or screw auger.<sup>2-4</sup> It is broadly accepted 37 38 that suspensions amenable to layer-wise extrusion/deposition 3D-printing can be composed by adjusting the rheological properties of the suspension.<sup>5–9</sup> Suspensions with suitable rheological 39 properties feature the ability: (i) to be extruded through a print nozzle without experiencing 40 flow cessation,<sup>10–12</sup> (ii) to support the weight of the printed layer itself, and of overlying 41 layers,<sup>9,13</sup> and (iii) to be shape stable while limiting the potential for filament rupture during 42 extrusion/deposition.<sup>12,14</sup> In 3D-printing of suspensions, two competing failure modes are 43 44 typically noted: (i) material failure by yielding, and (ii) elastic buckling failure through local or global instability, which in turn may be accompanied by material failure.<sup>15,16</sup> The failure mode 45 depends on the printed object's geometry, the kinetics of (the suspension's) structure 46

- 47 formation, i.e., post-extrusion/-deposition, the shear history of the suspension prior to
- 48 extrusion,<sup>17–19</sup> loading rate, boundary conditions, and process parameters (e.g., print speed,
- 49 nozzle distance).<sup>12,20</sup>

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#### 50

51 To ensure shape stability during the 3D-printing process, particulate suspensions are often composed at a high solid volume fraction ( $\phi \ge 0.40$ ),<sup>9,21,22</sup>. Such dense particulate suspensions 52 generally display non-Newtonian flow behaviors, including shear thinning, shear thickening, and 53 54 shear-induced jamming.<sup>23–26</sup> Shear jamming, where flow is arrested and the suspension exhibits 55 solid-like behavior, is understood to be strongly dominated by particle volume fraction and interparticle interactions.<sup>27–31</sup> To approach the shear jamming regime, both a minimum stress 56 level (or, equivalently, a minimum shear rate) and a sufficiently high packing fraction ( $\phi \rightarrow \phi_i$ ) 57 are required.<sup>30</sup> Dense suspensions exhibit high shear rate dependence: at low applied shear, 58 most particles in the suspension are lubricated by solvent layers, thus shear thinning is 59 observed, whereas at high shear, a large fraction of particles are forced into close proximity, 60 such that the lubrication layer can rupture or is reduced to molecular length-scales, resulting in 61 shear thickening behavior.<sup>25,26,32–34</sup> As a result, particles make direct (Hertzian) contacts and 62 experience friction. This effect becomes more critical as  $\phi$  approaches  $\phi_{max}$  where  $\phi_{max}$  ranges 63 between 0.50 to 0.60, which can lead to flow cessation and shear-induced jamming.<sup>33,35–37</sup> This 64 behavior can cause heterogeneity in the arrangement of neighboring particles and fragility<sup>\*</sup> in 65 dense suspensions when subjected to shear.<sup>36,38,39</sup> For instance, the consolidation of dense 66 67 suspensions during 3D-printing has been observed to result in differing (spatial) solid 68 concentrations as a function of position due to liquid phase migration.<sup>10,11,39</sup> Such increases in 69 solid concentration due to liquid phase migration can cause the suspension's transition from 70 flowing to solid-like state, thereby resulting in flow cessation in the extruder. Although the 71 complexity of the rheological behavior of dense suspensions has been extensively studied, it 72 remains unclear how such attributes impact printability and filament homogeneity during the 73 3D-printing processes. 74

75 The kinetics of structure formation in a dense cementitious suspension are critical for 3Dprinting. This requires, in general, that the time-dependent evolution of the yield stress of 76 suspension should be sufficient to prevent the flow of the printed layer itself, and the ability to 77 78 resist force amplification associated with the deposition of overlying layers (see Figure 1a). 79 Furthermore, the rate and extent of structure formation also control the rate and extent of 80 deformation of the printed layer, i.e., both at short and later times (see Figure 1b). For example, while thixotropic rebuilding associated with flocculation is relevant at short time 81 scales, rigid structure formation associated with the progress of hydration reactions dominates 82 at later times.<sup>13,40</sup> While structure formation is necessary and important, it needs to be 83 84 controlled, since it can induce issues such as: (i) flow cessation during extrusion, (ii) filament rupture during deposition, and (iii) poor interface bonding between layers after deposition. To 85 affect such structure formation rates, colloidal particles such as clays have been added to 86 87 cementitious suspensions. Clay particulates feature a platelike morphology consisting of negatively charged faces and positively charged edges.<sup>41–43</sup> This anisotropic surface charge 88 results in the formation of the *house of cards-like* structures within the suspension.<sup>44–46</sup> 89 90 Therefore, clay suspensions feature high thixotropic rebuilding rates (i.e., recovery rate)<sup>47,48</sup> and

<sup>\*</sup> Suspensions in the vicinity of jamming can feature fragility (i.e., particle-particle contact breakage) when the material experiences a rapid change in the direction of applied stress.<sup>36,38</sup>

- 91 stretchability, <sup>49,50</sup> which leads to superior shape stability while minimizing the risk of filament
- 92 rupture during printing. This enables accelerated printing and allows flexibility in the stand-off
- distance (i.e., distance from the print nozzle tip to substrate) and print path curvature.
- 94



**Figure 1:** Representative schematic that illustrates: **(a)** time-dependent yield stress evolution of a printed layer that is required to prevent the flow of the layer itself after extrusion as well as its ability to support subsequent overlying layers, and **(b)** reduced deformation of the deposited layer due to its structure build-up following deposition of subsequent overlying layers. Here,  $\alpha$  and  $\delta$  refer to the rate and extent of deformation of the printed layer, respectively, following the deposition of subsequent layers. In **(a)**, the interface bonding between layers is controlled by the yield stress of the layer immediately after deposition. If the yield stress of a printed layer is much greater than the minimum required to support itself, it can result in poorly-bonded interfaces between successive layers due to the lack of local remixing/homogenization and formation of high local porosity.<sup>51</sup>

95

96 Although the role of clay additions on improving suspension printability is well-known, the

97 means by which clay particulates mitigate flow cessation and enhance the homogeneity of

98 suspension during extrusion is less well-understood. To better address these questions, this

99 paper seeks to develop new insights into how clay incorporation affects flow cessation in

100 extruder and suspension homogeneity during extrusion and its resulting shape stability

101 following accelerated 3D-printing. Towards this end, a viscoelastic rope coiling approach (i.e.,

102 the examination of the coiling and falling of a viscoelastic rope onto a surface)<sup>52,53</sup> is used to

assess the buckling stability of a suspension that experiences stretching and bending during

deposition. Taken together, the outcomes of this work help in the development of new

105 guidelines for improving the printability and filament homogeneity of dense suspensions.

106

### 107 2. MATERIALS AND METHODS

### 108 **2.1.** Materials and sample preparation

109 Dense suspensions were composed using an ASTM C150-compliant<sup>54</sup> ordinary portland cement

110 (OPC) and kaolin clay (ACROS Organics)\* at varying kaolin-to-solid (i.e., OPC + clay) volume

<sup>\*</sup> Certain commercial equipment, software and/or materials are identified in this paper in order to adequately specify the experimental procedure. In no case does such identification imply recommendation or endorsement by

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- ratios between 0 % and 31 %. The upper bound of clay dosage was determined based on: (i)
- establishing the total solid volume fraction of suspension, that is required to ensure printability,
- i.e., typically in excess of 0.40, and (ii) the stiffening that results from OPC hydration, which is
- retarded/suppressed at high clay replacement levels. The median particle diameters ( $d_{50}$ ) of the
- 115 OPC and kaolin were 17.2  $\mu$ m and 1.4  $\mu$ m, respectively, as determined using static light
- scattering (SLS; LS13-320, Beckman Coulter). The densities of OPC and kaolin were measured as
- 117 3140 kg/m<sup>3</sup> and 2650 kg/m<sup>3</sup>, respectively, using helium pycnometry (Accupyc II 1340,
- 118 Micromeritics). The suspensions were formed by mixing the powders and water in a 500 mL
- beaker using a high shear, four-blade impeller-type mixer (RW 20 Digital, IKA) for 180 s at 500
- 120 rpm, followed by an additional 180 s of mixing at 600 rpm.
- 121

## 122 **2.2.** Printing and fabrication

- 123 A layer-wise extrusion system (LUTUM dual v2.0, VormVrij)<sup>55</sup> fitted with a 5 mm diameter
- 124 nozzle was used to print the suspensions under controlled air-pressures and flow rates (see
- 125 Figure 2a). Following mechanical mixing, the suspension was immediately loaded in a cylindrical
- barrel ( $\approx 600 \text{ cm}^3 \text{ volume}$ ) above the print head and the pressure generated by compressed air
- 127 was used to force the suspension into a single-screw auger. Different air pressure values
- ranging between 0.15 MPa and 0.41 MPa were applied to the print cartridge to vary the
- 129 extrusion velocity of material and the resulting gravitational stress rate that is induced in the
- print layers/coils (see Supporting Information, Figure S1). The non-linearity between coiling
- 131 speed and air pressure in Figure S1 is due to the lubrication layer formation with increasing air
- 132 pressure that can facilitate material movement in the extruder.
- 133

To evaluate the suspension's bucking stability, a rope coiling approach with different coil 134 135 angular velocities was implemented (see Figure 2b). A coiling viscous rope features a quasi-136 vertical tail that ruptures mainly by gravitational-induced stretching and a helical coil that 137 ruptures primarily by bending. The stand-off distance (i.e., the distance from the nozzle tip to the substrate) was maintained at 50 mm for all suspensions during printing, which enabled the 138 formation of both the tail and coil during printing. During 3D-printing, layers were continuously 139 deposited until failure occurred and the critical height,  $H_{cr}$ , corresponding to the buckling 140 instability of the printed coil at failure, was determined. The buckling stability of the coils is 141 142 primarily controlled by the resistance of the coil to compressive stresses induced by the deposited coils as well as bending resistance of the tail following coiling.<sup>52,53</sup> The total print time 143 was limited to 100 s. It should be noted that the total print time was substantially lower than 144 145 the time required to form a rigid structure due to OPC hydration (see Supporting Information, 146 Figure S2a). The kinetics of structure formation of cementing suspensions can be decomposed 147 in three successive phases: (i) colloidal interaction, (ii) percolated network formation due to the formation of flocs and hydrate nucleation between flocculated particles, followed by (iii) 148 stiffening due to the formation of bonds amongst particles by OPC hydrates.<sup>40,56–58</sup>. 149 150

the National Institute of Standards and Technology, nor does it imply that the equipment and/or materials used are necessarily the best available for the purpose.



**Figure 2: (a)** A schematic of the configuration of the 3D-printer consisting of the extruder and cylindrical barrel. All dimensions are expressed in mm. **(b)** A schematic that illustrates rope coiling of the suspension during extrusion/deposition. The filament with diameter *D* travels a stand-off distance *H* from the nozzle tip to the substrate on which it forms a helical coil of radius  $R_c$  that rotates with angular frequency  $\omega_c$  with respect to a vertical axis.

151

### 152 **2.3. Experimental methods**

### 153 **2.3.1.** Rheological characterization

154 The rheology of the suspension was characterized using a combined motor-transducer (CMT) 155 rheometer (Discovery Hybrid Rheometer, DHR-2, TA Instruments) using: (i) vane-in-cup

- 156 geometry for shear flow/oscillatory rheology, and (ii) parallel plate configuration for extensional
- 157 rheology. For all measurements, suspensions were conditioned to a temperature of 25 °C  $\pm$
- 158 0.1 °C and remained in the dormant hydration period (i.e., when the mixture is still plastic prior
- to cement hardening)<sup>59</sup> through the entire time of testing. Four different rheometry protocols
   were carried out as follows:
- The yield stress  $\sigma_y$  and plastic viscosity  $\eta$  were determined via a shear rate sweep. Before
- 162the sweep, a 60 s pre-shear at  $\dot{\gamma} = 100 \text{ s}^{-1}$  was carried out to remove shear history effects.16362 A reversible (ascending-descending) sweep procedure was applied in logarithmically-
- spaced steps (5 points per decade) from  $\dot{\gamma} = 0.001 \text{ s}^{-1}$  to 200 s<sup>-1</sup> with a 10 s data-averaging period. The *apparent* yield stress and plastic viscosity were identified as the peak shear
- 165 period. The *apparent* yield stress and plastic viscosity were identified as the peak shear 166 stress prior to the suspension's transition to plastic flow during ascending shear rate sweep
- 167  $^{61,63-65}$  and the slope of the rising portion of shear stress-shear rate curves ( $\dot{\gamma} > 1 \text{ s}^{-1}$ ), 168 respectively. The ascending sweep (up sweep) was followed by a descending sweep (down 169 sweep) over the same shear rate range. It should be noted that this rheological protocol to
- characterize stress response of suspension does not reveal the *static yield stress* when no rest time was allowed between pre-shearing and shear rate sweep (see Supporting
- 172 Information, Figure S2b).<sup>66</sup> The rheological protocols used herein are adopted to be relevant
- for a 3D-printing process wherein no rest time is typically permitted.
- The kinetics of structure formation in the suspensions was characterized via stepwise
   changes in the shear stress.<sup>67,68</sup> In this method, the shear stress was stepped up between 0
- and 200 Pa at 20 Pa increments and held constant for a 30 s period prior to the next step.
- 177 The shear stress upper bound was selected such that it did not exceed the yield stress of the
- suspensions ( $\sigma_i \leq \sigma_y$ ). The pre-shearing regime used was similar to the one used for the

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- shear rate sweep. To mimic the 3D-printing process, no rest time was permitted as shownschematically illustrated in Figure 1(b).
- The adhesive properties that determine the stretchability of the suspension were 181 characterized via extensional rheology (i.e., tack test).<sup>69</sup> The sample was placed on the 182 183 bottom plate and a 1 mm gap was set between the top and bottom plates. A pre-shearing 184 regime similar to that used for the shear rate sweep was used, and no rest time was 185 allowed to suspension. The top plate was raised vertically upward at a controlled velocity of 10  $\mu$ m/s until the sample became separated from the top plate. The normal force 186 experienced by the top plate, corresponding to the stretching force exerted by the 187 188 suspension, was recorded as a function of the plate's separation distance. The adhesive
- energy of suspension was then calculated as the area under the force-displacement curve.
  The viscoelastic behavior of the suspension over time was characterized via small amplitude
- 191 oscillatory shear (SAOS) rheometry. A time sweep (up to 1000 s) was performed at a fixed 192 frequency of 1 Hz and strain amplitude  $\gamma = 0.05$  %. This strain amplitude was selected such
- that all the suspensions remained in the linear viscoelastic region (LVR).
- 194

### 195 **2.3.2.** Structure characterization

- To assess the floc size of suspensions for varying clay dosages, dynamic light scattering (DLS)
   analysis (Malvern, Zetasizer Nano) was carried on dilute suspensions (0.05 g<sub>solid</sub>/L<sub>solution</sub>) of OPC-
- clay solids in aqueous medium of cement pore solution. The Z-average size (intensity-based
- 199 overall average size) of aggregate was determined by cumulants analysis (Malvern, Zetasizer
- 200 Software). Optical microscopy (Leica DM750P) was carried out with image capture using a
- digital camera on similar dilute suspensions as used to assess fractal structuring. Samples used
   for the optical microscopy were prepared by dripping dilute suspension on the surface of glass
- slide. To minimize dynamic flow and its subsequent structural changes, the sample was
- immediately smoothened and squeezed using the edge of another glass slide. Before
- application, the surfaces of the glass slides were cleaned using isopropanol.
- 206

### 207 2.3.3. Printability and filament homogeneity characterization

- 208 To quantify the printability of a suspension following the rope coiling method, the self-buckling
- 209 instability was determined when the stack of coils buckled under its own weight upon
- exceeding a critical height  $H_{cr}$ . The gravity-induced stretching of the filament was assessed by
- determining the maximum stand-off distance (MSOD) at which the extruded filament ruptured
- under its own weight. To evaluate the potential for liquid phase migration and heterogeneity of
- 213 materials, the local solid volume fractions were quantified by determining the solid mass of the
- extrudate by oven drying at 110 °C  $\pm$  5 °C for 24 h. The measurements were performed on
- extruded filaments at: (i) different positions of material across the height of the cylindrical
- barrel, and (ii) different positions across the diameter of the filaments. In (i), at different
- material displacement in the cylindrical barrel, the air pressure was set at zero to stop
   extrusion, and then the material remaining in the extruder (around 50 g) was removed and
- 219 used for the analysis of its solid volume fraction by drying material and determining water
- content. This procedure was repeated at different positions of material in the cylindrical barrel
- (L\* = x/L, see Figure 2a). In (ii), extrusion was stopped at L\* = x/L = 0.5, and the material
- remaining in the extruder was extruded to form a filament. The filament (5 mm × 100 mm; D ×

- 223 H) was then sectioned across its diameter to form 3 strips consisting of sections of the following
- dimensions 1.5 mm × 100 mm, 2 mm × 100 mm, and 1.5 mm × 100 mm. The solid volume
- fraction of each strip was then determined to evaluate the homogeneity of filament across its diameter.
- 227

#### 228 3. RESULTS AND DISCUSSION

#### **3.1.** *Rheology and fractal structuring of OPC-clay suspensions*

The influence of clay dosages on rheology was examined by analysis of the scaling exponents of 230 231 the measured (apparent) yield stress (see Figure 3a). Herein, the yield stress-solid volume fraction curves for varying clay dosages were fitted by a power-law function of the form  $\sigma_v = a$ 232  $(\phi)^{b}$ [Eq. 1], where a and b are fitting variables. Increasing the clay dosage required a lower 233 total solid volume fraction to achieve equivalent yield stress. This reduction arises from strong 234 interparticle interactions and high aspect ratio of the clay particulates, which results in the 235 formation of a house-of-cards structure and a reduction in the interparticle spacing for a given 236 237 solid content.<sup>2,70,71</sup> The results of dynamic light scattering (DLS) analysis confirmed that the 238 addition of clay resulted in an increase in floc size (see Supporting Information, Figure S3) while 239 producing a structural transition from densely packed flocs in the neat OPC suspension to highly-branched flocs (open structure) in OPC-clay suspension, as evidenced by the optical 240 microscope images (see Figure 4e-f). The power-law scaling exponent, b, of the yield stress 241 decreased with clay loading (see Supporting Information, Figure S4), meaning that the 242 dependence of yield stress on solid volume fraction reduced with clay dosage over a wide range 243 of yield stress-matched suspensions (100 Pa  $\leq \sigma_v \leq$  1000 Pa). This results from the formation 244 of fractally-architected aggregates that lead to more uniform arrangements of particles in the 245 clay-containing suspensions, as indicated by the optical microscopy imaging (see Figures 4e-f). 246 This behavior is desirable for 3D-printing, since dense suspension can experience shear- and /or 247 pressure-induced liquid phase migration during extrusion. Such changes in solid volume fraction 248 249 - induced due to liquid phase migration (i.e., dewatering of the slurry, locally) can result in the 250 onset of jamming  $(\phi \rightarrow \phi_i)$ , which causes the cessation of flow in the extruder. At an equivalent total solid volume fraction, the yield stress also exhibited a power-law scaling with clay loading 251 C/S (see Figure 3b). With increasing total solid volume fraction,  $\phi$ , the yield stress diverges at a 252 253 lower C/S, indicating that the maximum possible clay loading, C/S<sub>max</sub>, is dominated by the volume fraction of OPC contained in the suspension. This is on account of the formation of 254 255 stronger space-spanning networks between clay and OPC particles and reduced interparticle 256 spacing that feature a reduced lubricating liquid film thickness. The electrokinetic interactions in the suspensions were determined using zeta potential data, and calculating the interparticle 257 interaction potential (i.e., the contributions of electrostatic repulsion and van der Waals 258 259 attraction) between clay particles in DI water and OPC pore solutions (see Supporting 260 Information, Figure S5). Cement dissolution produces high ionic strengths (i.e., 0.52 M)<sup>72</sup> that results in a strong screening of electrostatic repulsions between particles. First, the high ionic 261 262 strength of OPC pore solutions compresses the electric double layer (EDL) around OPC particles, 263 thereby screening electrostatic repulsions, while increasing the tendency to form aggregates between OPC-OPC particles. Second, only attractive van der Walls forces operate between clay-264 265 clay particles in OPC pore solution thus enhancing the potential of their aggregation between 266 clay-clay particles (see Supporting Information, Figure S5). Third, the attractive electrostatic

- 267 interactions between positively charged OPC particles (zeta potential  $\zeta = +3.5$  mV) and
- 268 negatively charged clay particles (zeta potential  $\zeta = -0.5$  mV), suggest electrostatically driven 269 heteroaggregate formation.<sup>73</sup>
- 270



Figure 3: (a) The yield stress-particle volume fraction curves of dense suspensions for varying clay dosages. The data was fitted by a power-law function of the form  $\sigma_y = a(\phi)^b$  to determine the scaling behavior of the yield stress (see Supporting Information, Figure S4). (b) The variations in yield stress of suspensions for varying clay dosages as a function of total solid volume fraction. The data was fitted by a power-law function of the form  $\sigma_y = (C/S_{max} - C/S)^{-m}$  to determine maximum clay dosage  $C/S_{max}$  that is achievable for a given total solid volume fraction. (c) The dependence of intrinsic viscosity [ $\eta$ ] and maximum solid volume fraction  $\phi_{max}$  on clay dosage. The data was fitted by the Krieger–Dougherty equation (see Supporting Information, Figure S4). Here, based on three replicate measurements, a highest uncertainty of 3 % in the yield stress and viscosity was observed.

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In an effort to determine the maximum possible solid fraction or to predict the apparent 272 viscosities of suspensions as a function of particle loading several semi-empirical equations 273 have been proposed.<sup>63</sup> A convenient way to describe viscosity-concentration data is to use the 274 Krieger–Dougherty (K–D) equation.<sup>74</sup> The K–D equation is written as:  $\eta_r =$ 275  $(1 - \phi/\phi_{max})^{-[\eta]\phi_{max}}$  [Eq. 2],<sup>74</sup> where  $\eta_r$  is the relative viscosity (i.e., ratio of viscosity of the 276 277 suspension to continuous phase (water)),  $\phi_{max}$  is the maximum solid volume fraction (i.e., 278 cement + clay), and  $[\eta]$  is intrinsic viscosity of particles, was fitted to the viscosity-particle volume fraction curves. This equation is not exact, except perhaps in the dilute limit, where it 279 280 gives the relative viscosity equals 1 plus the intrinsic viscosity times the volume fraction. Therefore, the results obtained for  $[\eta]$  and  $\phi_{max}$  from fitting the K-D equation to the 281 measurements must be seen as displaying trends, and not precisely calculated values. With this 282 proviso, both the values of  $[\eta]$  and  $\phi_{max}$  were noted to linearly vary, such that increasing the 283 clay dosage resulted in an increase in [ $\eta$ ] and conversely, a reduction in  $\phi_{max}$  (see Figure 3c). 284 The greatest value of  $\phi_{max}$  = 0.60 was obtained for the neat polydisperse OPC suspension (see 285 Figure 3c), which is surprisingly close to the value of 0.638 noted for random close packing 286 (RCP) of monodisperse hard-spheres.<sup>28</sup> This can be used as an indication that the K-D equation 287 288 is approximate and can only display trends, since even a polydisperse perfect sphere 289 suspension is expected to feature  $\phi_{max}$  value much closer to 1 (e.g.,  $\phi_{max}$  = 0.87 for bimodal

system<sup>28</sup>). The non-sphericity of the cement particles<sup>75–77</sup> would tend to reduce this value, but it 290 still should be well above the monodisperse sphere results. It should be noted that [n] = 2.5 for 291 292 monodisperse hard-spheres,<sup>25,59</sup> and any deviations from the sphere value indicate particle aspect ratio, crowding, and flocculation.<sup>28,78</sup> For instance, it has been observed that  $[\eta]$  varies 293 with particle dispersion state ranging from 5.1 for dispersed suspension and 6.3 for flocculated 294 suspension.<sup>79</sup> Since [ $\eta$ ] is inversely proportional to the skeletal density of clusters,<sup>28</sup> a higher 295 value of  $[\eta]$  implies a lower skeletal density of particle agglomerates, suggesting a more open 296 structure as evidenced by increasing floc size with clay addition (see Supporting Information, 297

- Figure S3). This arises from the dominant interactions between clay-clay particles and their
- 299 higher particle aspect ratio than OPC particles.
- 300

301 To enable comparison of the structure formation kinetics and the printability of suspensions composed with different clay dosages, their solid volume fractions were adjusted to ensure 302 equivalent yield stress;  $\sigma_v = 300 \pm 25$  Pa. On account of increasing interparticle interactions, 303 304 the solid volume fraction required to match yield stress scaled linearly with clay dosage (see Figure 4a). Increasing the C/S from 0 % to 31 % reduced the total solid (i.e., OPC + clay) volume 305 306 fraction from about 60 % for the neat OPC suspension to about 42 % for C/S = 31 %. This is on 307 account of the fine nature of the clay, and its tendency to adhere water to its surfaces, which 308 demands dilution to achieve an apparent yield stress equivalent to the clay-free, neat OPC 309 suspension. A closer look at the reversible (ascending-descending) shear rheology curves in Figure 4b reveals that suspensions composed at equivalent yield stress:  $\sigma_v$  (300  $\pm$  25 Pa) exhibit 310 311 a shear stress divergence during both shear-up and shear-down sweeps. This divergence (i.e., non-monotonic trend) was much more significant for the neat OPC suspension as compared to 312 the clay-enriched suspensions. The shear stress divergence with decreasing shear rate followed 313 a power-law relation of the form  $\sigma_r = a(\gamma')^n$  [Eq. 3] where the exponent *n* decreased from 0.33 314 for the neat OPC suspension by one order of magnitude with increasing clay dosage (see Figure 315 4c). It should be noted that this observation in congruent with rheological hysteresis loop data 316

- 317 (see Supporting Information, Figure S6), in which the plain OPC suspension exhibited
- divergence in shear stress during a downward shear rate sweep, while this behavior was substantially suppressed for the OPC-clay suspension. Furthermore, on account of higher
- 320 thixotropic structure rebuilding, the OPC-clay suspension featured a greater hysteresis loop
- area than that of the plain OPC suspension.<sup>48,80</sup>
- 322





**Figure 4: (a)** The variations in the total solid volume fraction (OPC + clay) of the yield stressmatched suspensions in relation to the clay dosage needed to achieve equivalent yield stress;  $\sigma_y = 300 \pm 25$  Pa. Here, based on three replicate measurements, the highest uncertainty of 5 % in the solid volume fraction was observed. **(b)** The dependence of shear stress on clay dosage over ascending and descending shear rate sweeps for suspensions with  $\sigma_y = 300 \pm 25$ Pa. The scaling of shear stress divergence during the descending shear rate sweep was determined by fitting by a power-law function of the form  $\sigma_y = a(\gamma')^n$ . **(c)** The dependence of power-law scaling *n* of shear stress divergence across varying clay dosages. **(d)** The dependence of the fractal dimension of the solids in the suspensions on clay dosage. The diamond symbols in **(d)** are marked to the secondary y-axis (i.e., relative packing fraction  $\phi/\phi_{max}$ ) as a function of the clay dosage. Representative optical microscope images that illustrate fractal structuring in **(e)** the neat OPC suspension and in **(f)** the clay-enriched suspension (*C/S* = 31 %).

323

The divergence of the shear stress curves observed in Figure 4(b) for the neat OPC suspension 324 can be attributed to shear-induced structural inhomogeneities that produce mechanically 325 326 unstable behavior. When particles are forced into close proximity at high shear rates, the 327 lubricating water film thickness is reduced, leading to the formation of frictional contacts between particles. This can result in solid-like behavior of suspensions when the shear rate 328 329 increases, indicating discontinuous shear thickening, as identified by the jump in viscosity-shear rate trend (see Supporting Information, Figure S7). When the applied shear stress was reversed 330 (downward shear sweep), the neat OPC suspension experienced non-monotonic behavior. 331 However, this non-monotonic response was found to diminish with increasing clay dosage (see 332 333 Figure 4c). More stable behavior in OPC-clay suspensions can be attributed to the (i) stronger 334 interparticle interactions between clay-clay and clay-OPC particles, (ii) the lower total solid 335 volume fraction in suspension (lower relative volume fraction  $\phi/\phi_{max}$ ) that is required to achieve an equivalent yield stress to that of the neat OPC suspension, and (iii) fractal 336 structuring transition from densely packed flocs to highly branched flocs (i.e., space-spanning 337 338 network) in the OPC-clay suspensions that can hinder local liquid phase migration amongst particles. To assess the structure of aggregates within the OPC-clay suspensions, their fractal 339 dimensions were determined assuming strong gels, in which the interfloc links are as strong as 340 the intrafloc linkages:  $b = (d' + X)/(d' - D_f)$  [Eq. 4],<sup>81</sup> where b is the power-law scaling from 341 the yield stress-particle volume fraction curves (see Figure 3a and Supporting Information, 342

- Figure S4),  $D_f$  is the fractal dimension of the aggregate cluster, d' is the Euclidean dimension (d'
- 344 = 3), and X is the bond dimension that describes the fractal geometry of the cluster backbone (X
- 345  $\approx 1.0^{81,82}$ ). A value of  $D_f \approx 3$  was determined for the neat OPC suspension with  $\phi/\phi_{max} \approx 1$
- indicating no fractal behavior wherein particles aggregate into closely-packed assemblages.
- 347 Conversely, the fractal dimension decreased with clay addition, wherein the aggregates tend to
- 348 form fractally-architected regions that feature a more open structure, thereby resulting in more
- uniform arrangements of particles, as evidenced by optical microscopy (see Figures 4e-f).
- 350

## **351 3.2.** *Structure formation kinetics of OPC-clay suspensions*

The kinetics of structure formation and the deformation resistance of the suspensions were thereafter probed using the stepped-isostress method (see Figure 5a). Within each step, the suspension experienced an instantaneous strain that was followed by transient strain, which reached a plateau. In general, the extent of instantaneous strain and the rate of transient strain were strongly reduced by clay dosage. To determine the time required to reach a plateau for

- each cycle, i.e., the relaxation time, a Kelvin–Voigt model of the form  $\gamma_t =$
- $\sum_{i=1}^{N=10} A_i (1-e^{-t/t_{r,i}})$  [Eq. 5],<sup>83</sup> was fitted to the data, where  $t_{r,i}$  is the retardation time (i.e., 358 relaxation time) and  $A_i$  is a fitting parameter. This approach is analogous to creep flow 359 360 characterization using the Boltzmann superposition principle (i.e., time-aging-stress) of soft glassy materials that show strong time and stress dependency.<sup>84,85</sup> Semi-log plots showing  $y-t_r$ 361 collapses all cycles on a single master curve for each suspension (see Figure 5b). It should be 362 noted that although the retardation times were extracted by fitting Eq. 5 to each shear stress 363 cycle of 30 s, in effect, some strain trends did not truly reach a plateau, specially at higher shear 364 stress cycles, as indicated in the zoomed view (see Supporting Information, Figure S8). Herein, it 365 is seen that structural rebuilding that is induced by the clay particles results in a lower 366 deformation rate as indicated by a greater power-law scaling b for clay-enriched suspensions. 367 The rate of strain development is controlled by: (i) thixotropic rebuilding that is dominated by 368 flocculation, and (ii) short-term stiffening that is induced by the formation of a percolated 369 network at pseudo-contact points<sup>\*</sup>.<sup>40,86</sup> The formation of a percolated network is suggested to 370 be caused by both physical and chemical changes.<sup>58,86</sup> The physical effect is linked to the time-371 372 dependent colloidal flocculation, which tends to bring particles into contacts while the chemical effect results from formation of percolated network between particles via nucleation and 373 growth of hydrates at pseudo-contact points between flocculated particles.<sup>40</sup> 374
- 375

<sup>\*</sup> Pseudo-contact points refer to connectivity between particles in cementing suspensions that are connected by colloidal interactions rather than by cement hydration products, which, all together, results in formation of percolated networks between particles.





Figure 5: (a) The evolution of strain with stepwise changes in shear stress for varying clay dosages. The shear stress was stepped up between 0 and 200 Pa at 20 Pa increments over 10 steps and held constant for a 30 s period prior to next step over a cumulative period of 300 s.
(b) The master curves of strain responses using data shown in (a) that are obtained by fitting to the Kelvin–Voigt model. (c) The dependence of strain rate on clay dosage. The deviation from the initial slope captures the transition from flocculation to percolation (i.e., short-term stiffening) in the suspension.

376

- Closer analysis of the kinetics of structure formation of OPC-clay suspensions indicates that the suspensions demonstrate transition points with initial slope  $m_1$  corresponding to flocculation with a progressive switchover, with time, to percolation slope  $m_2$  (see Figure 5c). The
- 380 contribution of flocculation to the strain rate remains constant as indicated by a constant slope
- $m_1$  in Figure 5c. However, due to the progressive formation of percolated networks between
- 382 particles, the contribution of short-term kinetics to reducing the strain rate increases
- progressively (see Figure 5c). As such, the contributions of flocculation and percolation due to the short-term kinetics of suspension structure formation to deformation associated with
- deposition of subsequent overlying layers can be estimated by an equation of the form  $\frac{d\gamma}{dt} =$
- 386  $m_{1, for t < t_c} + \exp(m_2 t)_{for t > t_c}$  [Eq. 6], where  $m_1$  is the flocculation rate, which is constant up
- to the onset of percolated network formation, and  $m_2(t_c,t)$  is the percolation rate, which
- initiates at time  $t_c$  (i.e., transition from flocculation to percolation) and varies exponentially with
- time. Notably, as a result of OPC dilution, clay dosages delay stiffening. For example, the 31 %
- clay suspension exhibited solely the early-age slope  $m_1$ , suggesting that the contribution of
- 391 short-term stiffening is minimal. These observations are in agreement with the non-linear
- 392 structural build-up of storage modulus (see Supporting Information, Figure S9b). The initial
- 393 storage modulus is related to the degree of flocculation while the evolution rate of storage
- 394 modulus is controlled by percolated network formation.
- 395

### **396 3.3.** *Buckling stability and filament homogeneity of OPC-clay suspensions*

- 397 The resistance to buckling, i.e., the buckling stability of the filament is controlled by both the
- 398 structure formation kinetics and filament homogeneity. The former is important to sustain
- 399 stability during layer build-up while the latter ensures consistent properties of the filament
- 400 during deposition to minimize imperfections during printing. The critical buckling height, *H*<sub>cr</sub>, of

401 the suspensions was noted to enhance significantly with clay dosage across coiling speeds (see

402 Figure 6a). This enhancement is attributed to both the (i) improved structural recovery

following deposition, which reduces deformations of the deposited filament as indicated by its

faster flocculation (see Figure 5), and (ii) greater stretchability, which limits the potential for

- filament rupture when subjected to stretching (tail portion) and bending (coil portion).
- 406



Figure 6: (a) The dependence of critical buckling height on clay dosage across different coiling speeds. The red zone represents coiling speeds beyond which the flow of suspensions was arrested in the extruder. Based on three replicate measurements, the highest uncertainty of 9 % in the critical buckling height was observed. (b) Representative normal force-plate displacement responses for suspensions subjected to extensional rheology across clay dosages. The adhesive energy of suspension was calculated as the area under the force-plate displacement curve. (c) The critical buckling heights corresponding to the onset of buckling instability. Unlike clay-enriched suspensions, the neat OPC suspension featured rapid filament rupture during deposition.

407

408 The buckling stability of coiling of an elastic rope has been found to scale with the gravitationalbending length  $L_{gb}$ , which is driven the balance between bending and gravitational energies: 409  $L_{ab} = (Er_0/8\rho g)^{1/3}$ [Eq. 7], where E is the material Young's modulus,  $r_0$  is the radius of the 410 filament, and  $\rho$  is the material density.<sup>87</sup> As such, materials with a higher zero-shear rate 411 storage modulus G' are expected to have greater buckling stability. In addition to the elastic 412 property, stretchability is another important factor controlling the stability of suspension 413 following coiling. To quantitatively assess the stretchability, the adhesive properties (i.e., 414 415 cohesion and adhesion)<sup>88</sup> of the suspensions were determined using extensional rheology (see Figure 6b). The greater force peak and slower force decay indicate greater cohesion of the clay-416 417 enriched suspensions. This also results in greater adhesive energy in clay-enriched suspensions 418 as evidenced by the larger area under the load-displacement curve obtained from extensional 419 rheology. Unlike OPC-clay suspensions, the neat OPC suspension exhibited a strain-hardening 420 response. It should be noted that both plain OPC and OPC-clay suspensions were drawn 421 towards the center of the plate and no fingering instability was observed for the OPC-clay 422 suspensions during extensional rheology. The strain hardening behavior for the plain OPC suspension is expected to be due to its high solid volume fraction, which can result in a higher 423 424 interparticle friction and interlocking when suspension moves towards the center of the plate.<sup>88</sup>

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- 425 Increasing frictional contacts between particles induce an enhanced tendency to rupture as
- 426 evidenced by the sharp force decay following the peak force.
- 427
- 428 Coming back to buckling, it is notable that the 31 % clay suspension showed essentially an
- 429 unchanged critical buckling height across coiling speeds. This suggests that the structural
- 430 recovery of these suspensions was fast enough to sustain the increased loading rate. On the
- 431 other hand, the buckling stability of the neat OPC suspension (and suspensions composed of
- lower clay dosages) diminished significantly with coiling speed, suggesting that the rate of
- 433 structural recovery was inferior to the coiling speed and the loading rate imposed therein. In
- addition to retarded structural recovery, the neat OPC suspension featured filament fragility
- 435 (i.e., rupture and discontinuity) following extrusion and deposition due to its low stretching and
- 436 bending capacities (see Figure 6c). Such filament rupture creates imperfections that accelerate
- 437 buckling instability and yielding. As a result, the printed coils consisting of the neat OPC
- 438 suspension dominantly fail by material yielding. However, the clay-containing suspensions fail
- dominantly due to a buckling instability since the accelerated structural recovery and enhanced
- stretchability offered by clay particulates enhances both the normal and bending capacities of
- 441 the filament, thus enhancing its resistance to yielding.
- 442



Figure 7: (a) The correlation between adhesive energy of the suspension and maximum stand-off distance of its filament. The maximum stand-off distance (MSOD) was determined as the allowable filament length that was able to support its weight when subjected to gravity-induced stretching. The insets compare the rupture modes of the OPC and clay-enriched suspensions subjected to gravity-induced stretching. (b) The variation in the relative solid volume fraction of the filament as a function of the relative length of material in the cylindrical barrel during extrusion. (c) The variation in the relative solid volume fraction of its relative diameter during extrusion. The inset illustrates the defined relative position across filament diameter. Based on three replicate measurements, a maximum 3 % deviation in solid volume fraction was observed.

- 443
- To better assess the stretchability of the filament, the maximum stand-off distance (MSOD) at
- which a filament can withstand its own weight was measured (see Figure 7a). It was noted that
- the MSOD scaled linearly with the adhesive energy of the suspension, indicating that –
- 447 expectedly a more "sticky" suspension features better filament continuity. Importantly, a

sequence of images obtained from video recording of crack propagation and rupture mode 448 revealed that the neat OPC suspension exhibited elastic rupture wherein a crack initiated at the 449 450 edge and propagated inward (i.e., edge fracture) while the clay-enriched suspension featured 451 necking failure (i.e., ductile failure) – more similar to plasticity-driven failure. This suggests that in the neat OPC suspension, strain localization, caused by local dewatering along the filament's 452 length/diameter, formed fracture critical regions during extrusion and deposition. Indeed, more 453 454 detailed analyses of local particle density variations of the filament as a function of both relative position of material in the cylindrical barrel and along the filament diameter indicated 455 456 that the neat OPC suspension underwent significant changes in particle volume fraction during 3D-printing/extrusion; unlike the clay-enriched suspensions (see Figures 7b-c). This was 457 gualitatively suggested by visual evidence of water leakage from the extruder indicative of the 458 dewatering of the OPC suspension. It has been observed that the imposition of extrusion 459 pressure and shear can lead to the formation of different zones, including plug flow, shearing, 460 and dead regions, within the material in the extruder.<sup>10,11</sup> This can induce liquid phase 461 462 migration, which may result either in flow cessation due to a reduced lubricating water film thickness or the formation of solid-poor regions in the filament.<sup>6,10</sup> This results in 463 heterogeneous gradients of particle volume fraction in the filament (see Figure 7c). This 464 465 behavior is critical for suspensions whose particle volume fraction is in the vicinity of the jamming volume fraction  $\phi_i$ , where a small local increase in solid concentration associated with 466 liquid phase migration can cause cessation of flow and bring the suspension to the jammed 467 state,  $\phi \rightarrow \phi_i$ , in the extruder.<sup>89</sup> This is indicated by the sharp power-law scaling of yield stress 468 with  $\phi$  for the neat OPC suspension (see Supporting Information, Figure S4) wherein the yield 469 stress rapidly increases with a small increase in solid concentration. On the other hand, minimal 470 471 changes in particle volume fraction induced by shear are noted in the clay-enriched suspension, since their lower relative volume fraction  $\phi/\phi_{max}$  and the fractally-architected arrangement of 472 473 aggregates arrests water within flocs and consequently reduces liquid phase migration (see 474 Figures 7b-c). Thus, reducing the dewatering of the suspension is a critical need to ensure ease 475 and consistency of extrusion during the 3D-printing process.

476

### 477 4. SUMMARY AND CONCLUSIONS

This paper has elucidated the influences of clay particulates on controlling fractal structuring 478 479 and flow cessation in extruder and homogeneity of cementing suspensions in the context of the 480 3D-printing process. Special focus was placed on understanding how the rope-coiling method can be used to assess the buckling stability and fragility of the extruded filament – during 3D-481 482 printing – for suspensions loaded with clay additives. In general, it is noted that neat OPC 483 suspensions feature a steep stress-shear rate relationship, resulting from their closely-packed structure with  $\phi \approx \phi_{max}$  and fractal dimension  $D_f \approx$  3. The dosage of clay particulates is found 484 to decrease the power-law divergence (i.e., non-monotonic behavior) of shear stress and 485 mitigate flow cessation in the extruder. This is linked to the (i) stronger interparticle 486 interactions between clay/OPC particles, (ii) decreased relative packing fraction ( $\phi/\phi_{max}$ ), and 487 488 (iii) the formation of fractally-architected aggregates in the OPC-clay suspensions which can suppress the mechanically unstable behavior in shear stress response. Alterations in fractal 489 490 structuring of aggregates from densely packed flocs in the neat OPC suspension to highlybranched flocs (i.e., space-spanning network) in the OPC-clay suspension results in more 491

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- 492 uniform particle arrangements that can increase the resistance to liquid-phase migration (i.e.,
- 493 slurry dewatering) when shear stress is applied. The analysis of interparticle interactions
- 494 indicates that high ionic strengths resulting from cement dissolution disrupt electrostatic
- 495 repulsion between clay-clay particles by screening charges, thus resulting in a significant
- 496 interparticle interactions between all particles (clay-clay, OPC-clay, and OPC-OPC particles) in
- the OPC-clay suspensions. Importantly, the clay-enriched suspensions due to their greater
- 498 structural recovery rate and stretchability feature higher critical buckling heights across varying
- 499 coiling speeds than the neat OPC suspension. The ability of clay additions to prevent
   500 dewatering is critical for dense suspensions whose particle volume fraction is in the vicinity of
- the jamming volume fraction  $\phi_i$ . Since dewatering causes the solid volume fraction to approach
- 502  $\phi \rightarrow \phi_i$ , which results in flow cessation in the extruder, maintaining  $\phi < \phi_j \approx \phi_{max}$  is a key factor
- for ensuring the printability of dense suspensions. The understanding gained herein offers new
- means to design, evaluate, and rank dense suspensions to ensure superior filament
- 505 homogeneity (i.e., hindering dewatering) and stretchability. These attributes are of relevance to
- slurry-based 3D-printing processes wherein the filament undergoes substantial stretching and
- 507 bending actions due to changes in the stand-off distance or print path curvature.
- 508

### 509 **5.0. ACKNOWLEDGEMENTS**

- 510 The authors acknowledge financial support for this research from: The Anthony and Jeanne
- 511 Pritzker Family Foundation, TRANSCEND: a joint UCLA-NIST consortium that is supported by its
- industry and agency partners, and the National Science Foundation (DMREF: 1922167). This
- research was conducted in the Laboratory for the Chemistry of Construction Materials (LC<sup>2</sup>). As
- such, the authors gratefully acknowledge the support that has made these laboratories and
- their operations possible. The contents of this paper reflect the views and opinions of the
- authors, who are responsible for the accuracy of the datasets presented herein, and do not
- reflect the views and/or policies of the funding agencies, nor do the contents constitute a
- 518 specification, standard or regulation.
- 519

### 520 SUPPORTING INFORMATION

- 521 Coiling speed with air pressure, kinetics of structure formation at varying rest times, dynamic
- 522 light scattering of flocs, rheological properties with particle volume fraction, interparticle
- 523 interaction potential between clay particles, rheological hysteresis, shear thickening/thinning
- 524 behavior, strain-retardation time, and viscoelastic properties.
- 525

### 526 CONFLICTS OF INTEREST

- 527 There are no conflicts to declare.
- 528

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**Synopsis:** Transition from closely-packed to fractally-architected structures with clay addition improves homogeneity and prevents local dewatering, thus enhancing coiling stability of layer-wise extruded cementing suspensions during 3D-printing.