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Graphene-Covered Transition Metal Halide as Efficient and Durable Electrocatalysts for Oxygen Reduction and Evolution Reactions

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6 **Graphene-Covered Transition Metal Halide Molecules as Efficient** 7 **and Durable Electrocatalysts for Oxygen Reduction and Evolution** 8 **Reactions**

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10 Proton exchange fuel cells (PEFC) are one of the most popular and promising energy conversion devices because of their 11 highly stable and efficient membranes in acidic media, but there is lack of durable non-noble metal electrocatalysts suitable 12 for the acidic environments. Herein, we designed a new type of electrocatalysts consisting of transition metal halide 13 molecules covered by graphene sheets, which is supported by the experiments. To rapidly screen the best catalysts from 14 numerous candidate materials, the electronic structures, reaction free energies and overpotentials of those graphene-15 covered halide catalysts were studied by the first-principles calculations to predict the catalytic activities for oxygen 16 reduction reaction (ORR) and oxygen evolution reaction (OER). An intrinsic descriptor, the electrostatic force induced by the 17 metallic ions, was found to well describe the catalytic activities and provides a better understanding of the local electrical 18 field effects on catalytic activities. The spin-down d-band center was also introduced to describe catalytic activities of the 19 catalysts. The results demonstrate that the graphene-covered CrBr₂ shows the best bifunctional catalytic activities for fuel 20 cells while graphene-covered CoF₂ could well facilitate H₂O₂ production. These catalysts are better than the best commercial 21 noble metal catalysts (e.g., Pt and RuO₂) in terms of overpotentials and activities. This work provides a theoretical base for 22 rationally designing durable electrocatalysts with excellent catalytic activities.

23 **1. Introduction**

24 Clean energy conversion such as hydrogen fuel to electricity by fuel cells and energy storage through metal-air batteries are considered as 25 critical emerging technologies to replace traditional fossil fuels. In these energy technologies, oxygen reduction reaction (ORR) and oxygen 26 evolution reaction (OER) are two key reactions that determine conversion efficiency, and electrocatalysts are usually needed to promote 27 those reactions¹. Among the traditional electrocatalysts, noble metal catalysts are often applied to catalyse the ORR and OER (e.g., Pt for 28 ORR², and RuO₂ for OER³). However, the high cost and limited resources of the noble metals hinder their large-scale commercial application 29 of the energy technologies ¹.

30 In the past decades, nonprecious metal and metal-free catalysts have been developed to preplace or reduce the use of noble metal catalysts. In particular, metal-free catalysts, such as nitrogen doped carbon nanotubes and graphene, performed much better ORR catalytic 32 activity than Pt⁴. Multitudinous carbon catalysts, including heteroatom (e.g., B, N, F, O, S, P and transition metals) doped graphene, carbon 33 dots, carbon nitride, graphite, and defective graphene were also developed for efficient electrocatalysts ⁵⁻¹⁰. However, those carbon-based electrocatalysts, like nonprecious metal catalysts, still suffer from poor durability, in particular in acid media although most of them works well in alkaline environments. This downgrade the proton exchange fuel cells (PEFC), the most popular and promising ones in commercial applications because of their highly stable and efficient membranes in acidic media. It is therefore necessary to develop electrocatalysts well suitable for acidic environments.

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 Apart from the doping in graphene, intermolecular charge-transfer via non-covalent functionalization has also been demonstrated to enhance ORR electrocatalytic activities of graphene. The non-covalent functionalization involves weak interactions, which cause no change 40 on the basal plane structure of graphene and its electronic properties being largely retained.¹¹ The enhanced electrocatalytic activities originate from the intermolecular charge transfer between the undoped graphene and physically adsorbing polarized organic molecules (e.g., 42 poly(diallyl dimethylammonium chloride) (PDDA), and Tetracyanoethylene (TCNE)) on the basal plane of pristine graphene sheets.¹¹⁻¹³ This "intermolecular charge-transfer" offers a simple, low-cost approach to develop carbon-based metal-free ORR catalysts. However, these adsorbed organic molecules still suffer from the acidic attack in the acidic environments, which reduces the stability of the catalysts.

45 In this paper, we designed novel electrocatalysts consisting of graphene sheets embedded with transition metal halide molecules. 46 Graphene is extremely flexible with large surface area, excellent mechanical and electrical properties, and highly stable in fuel cell 47 environments, while halide molecules are polarized and could be embedded into the graphene sheets ^{14,15}. While the halide molecules are 48 covered and separated from the reactive environment by the graphene, their high polarization would enhance the catalytic activity through 49 spin or charge transfer between the graphene and the halide molecules. Such graphene-covered halide catalysts would be durable while 50 having highly catalytic activity in acidic environments.

 The above new catalyst design involves the selection of catalysts from numerous candidate materials. However, experimentally searching for the best catalysts would be time-consuming and costly. In contrast, computational approaches could be used to rapidly screening of the best catalysts, and provide the guidelines to the experiment. Recently computational and theoretical methods were used to elucidate the 54 microscopic details underlying the mechanisms of operation behind the functioning of PEFC. 16-18 Activity descriptors were developed to 55 identify the best ORR/OER catalysts for heteroatom-doped carbon nanomaterials and covalent organic framework (COF). 19,20 In this work, DFT calculation method will be used to establish a design principle to guide the structural design for the family of graphene-halide catalysts.

57 **2. Simulation methods**

58 Transition metal (TM) halide (TMX₂) absorbed graphene (G) sheets were constructed to calculate their catalytic performances (TM = Ti, 59 V, Cr, Mn, Fe, Co, Ni, Cu, Zn, and X = F, Cl, Br, I). The TMX₂ molecules we studied are listed in Table S1. Such graphene structures embedded 60 with salts have been achieved experimentally.^{14,15,21} As illustrated in Figure 1, the unit cells of the structures used in this study are composed 61 of a single-layer graphene sheet adsorbed with a metal halide molecule, the size of which was set to 12.82 Å \times 9.87 Å \times 20Å. The Van der 62 Waals correction was also taken into consideration in the calculations. Gaussian smearing method was used to determine the electronic 63 partial occupancies. In this design, only graphene surface is exposed to the reactants while the halides are hided behand the graphene. The 64 G/TMX₂ was constructed as a three-dimensional periodic structure, where the vacuum layer was set to be around 20 Å in the z-directions to 65 avoid interaction between slabs.

66

67 **Figure1.** The unit cell model of TMX2 adsorbed graphene. The brown, yellow and blue atoms represent Carbon, Halogen, and TM, 68 respectively.

69 The free energies and overpotentials of OER and ORR on the structures were calculated with the density functional theory (DFT) method, 70 the Vienna Ab initio Simulation Package (VASP) ^{22,23}. The Perdew-Burke-Ernzenhof (PBE) exchange correlation functional and soft projector-71 augmented wave (PAW) pseudopotentials were implemented in VASP code via the DFT. 24,25. The PBE functional which in our previous studies

- 72 have been shown a great performance of the carbon based system.^{19,20} For 3d transition metals in this work, DFT+U methods ²⁶ were also
- 73 considered in the calculations, and the values of U-J are listed in Table S2. The cut off energy of plane wave basis was set 550 eV, and the 74 structures were sampled by 3×3×1 Monkhorst-Pack k-point mesh. Both the cutoff energy and k point meshes were converged within 1 meV
- 75 in per atom. The whole calculations in spin-polarized with the lowest energy magnetic configurations were identified for all the surfaces and
- 76 the geometries were relaxed until a maximum force of 0.01 eV /Å was converged.

77 **3. Results and Discussion**

78 **3.1 Electronic structures and reaction pathways**

83 **Figure 2.** Differential charge density distribution on the graphene absorbed with ZnBr₂ (no charge transfer between the graphene and ZnBr₂) 84 in (A) top view and (B) side view. The gray and red dotted cycles represent Br and Zn, respectively. Differential charge density distribution on 85 the graphene absorbed with CoBr₂ with a charge transfer between the graphene and CoBr₂ (0.78e) in (C) top view and (D) side view. The gray 86 and red dotted cycles represent Br and Co, respectively. The blue and yellow colors indicate the positive and negative values of electron 87 quantities, respectively. The isosurface level is 0.0005.

88 The electronic structures, the electron transfer between TMX₂ and graphene, and the effect of electronic field on ORR and OER were 89 calculated by the DFT. Both 2 and 4-electron pathways of OER and ORR (Supplementary information) and reaction free energy were analyzed 90 following the approaches described in references.^{3,27-33} Figure S1 shows the optimized configurations of the G/TMX₂ and Bader differential 91 charge density distribution on the graphene with the largest and smallest Bader charge transfer of every group. Figure 2A and B shows 92 optimized structures and Bader differential charge density distribution of G/ZnBr₂ with zero charge transfer while Figure 2 C and D shows 93 optimized structures and Bader differential charge density distribution of G/CoBr₂ with the largest charge transfer (0.78e) among G/TMX₂. 94 The metal ions are highly positively charged, while the charge on the graphene redistributes near the halide molecule. In addition, there 95 exists electron transfer between the TMX₂ molecules and graphene. The charge transfer could be positive, negative or zero, depending on 96 the type of the metal ions in halides (Table S1 and S3). Thus, the adsorption of TMX₂ induces charge redistribution on graphene as well as

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97 the charge transfer. The highly positively charged ions, charge transfer and charge redistribution may have a strong effect on the catalytic 98 activities.

99 The catalytic activities of G/TMX2 were evaluated by calculating the minimum overpotential of ORR and OER on the graphene surface. 100 The overpotential η of ORR/OER has been considered as an important measure of the catalytic activities of catalysts². For an ideal ORR/OER 101 catalyst, it should promote the OER/ORR just above the equilibrium potential, with zero overpotential². However, since the binding energies 102 of the intermediates are correlated, this perfect state cannot be achieved. Thus, the lower the overpotential, the better the catalysts.

103 To find out the lowest overpotentials of active sites on the G/TMBr₂, we calculated the free energy of the four elementary steps of 104 OER/ORR on all the possible sites that could occur on the graphene. Because the four elementary steps are correlated with each other, we 105 analyzed the adsorption energies of the HOO*, OH*, and O* intermediates adsorbed on graphene during ORR and OER (Table S4 and S5). 106 Figure 3A illustrates the catalytic cycle of ORR in 4-electron pathway (A-B-C-D-A) and 2-electron pathways (A-E-A). In 2-electron transfer 107 pathways, we only analyzed the HOO* (Table S6) for it is the only intermediate in this reaction.

113 We firstly discuss the 4-electron transfer ORR and OER. Figure 3B-C and Figure S2-S3 show the free energy diagrams of ORR and OER 114 on G/TMX₂ with the lowest overpotentials. For OER, nearly all the G/TMX₂ structures are uphill at an electrode potential of 0V except for 115 G/MnF₂ and G/MnBr₂ structures. For most of the G/TMX₂ structures the rate limiting step is the first electron transfer to absorb water 116 molecule to form HO*, except for G/TMX₂ (G/VF₂, G/VF₂, G/VBr₂, G/CrBr₂, G/CoBr₂, and G/CoI₂) with their rate limiting step at the third 117 electron transfer (the transformation of O* to HOO*). Our results show that the OER free energy change of G/VF₂, G/CrCl₂, G/CrBr₂ and G/VI₂

118 are very close to that of the ideal catalyst, indicating that these structures may have better catalytic performance than others in their own 119 groups. Among all the structures, G/CrBr₂ has the lowest overpotential and may exhibit the best OER catalytic activity among all G/TMX₂ 120 structures. Similarly, for ORR in acid solutions, the rate limiting step is the first electron transfer, $O_2^* + H^* + e^- \rightarrow HOO^*$, which are consistent 121 with the previous reports ^{34,35}. Among all G/TMX₂ structures, CrBr₂ absorbed graphene shows the lowest overpotential for ORR and thus is 122 the best ORR catalyst among them. Overall, CrBr₂ absorbed graphene performs the best catalytic activity in both OER and ORR process, 123 indicating that the structure of G/CrBr₂ may be a promising bifunctional OER and ORR catalyst.

124 For 2-electron pathways (Figure 3D and Figure S4), among them, both G/TiX₂ and G/CrX₂ could occur spontaneously, but G/MnX₂ and 125 G/CuX₂ cannot lead to spontaneous reaction both in 2e- and 4e-transfer pathways. In particular, the overpotentials of G/CoF₂, G/TiCl₂, 126 G/CrBr₂ and G/ CoI₂ are much closer to the ideal catalysts of their own group. These catalysts are better in catalyzing the production of H₂O₂ 127 compared with the other structures in their own groups. Among all the structures, $G/CoF₂$ is one of the best catalysts (Figure 3dD) for H₂O₂. 128 Compared with the best Pd catalysts with an overpotentials of 0.08V³⁶, G/CoF₂ is even lower than Pd catalysts, indicating that it could have 129 better catalytic activity than Pd.

130 **3.2 Scaling factor and descriptors for volcano relationship**

131To screen highly active catalysts from the candidate materials, it is necessary to find descriptors that well describe the catalytic activities of

132 the halide-adsorbed graphene. Figure 4A shows the plots of ΔG_{HOO*} vs ΔG_{HO*} (where ΔG is the free energy of adsorption) for the active sites

133 with the lowest overpotentials of ORR and OER on G/TMX₂ structures in 4-electron transfer pathways. From Figure 4a, a linear relationship

134 is expected between $\Delta G_{HO\sigma^*}$ and $\Delta G_{HO\sigma^*}$, which can be used to calculate the lower limit of ORR and OER overpotentials of G/TMX₂ structures

136 **Figure 4.** (A) Absorption energy of OOH* versus that of OH* on G/TMBr₂ for 4-electron transfer pathways. The lowest (B) ORR and (C) OER 137 overpotentials for G/TMBr₂ versus the absorption energy of OOH* in 4-electron transfer pathways.

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138 To find out the most active sites on the TMX₂ absorbed graphene, we took ΔG_{HOP} as a descriptor to describe the relationship between 139 ΔG_{HOO*} and the overpotentials. Figure 5B and C shows the overpotentials η^{OER} and η^{ORR} versus descriptor, ΔG_{HOO*}. The lowest OER and ORR 140 overpotentials for 4-electron transfer pathways are listed in Table S3. As shown in Figure 4B, C, there are volcano relationships and the 141 G/CrBr₂ sits on the top of the volcano (ORR ~ 0.35 V), and also shows the lowest OER overpotentials (OER ~ 0.36V) among the G/TMBr₂. 142 These predicted OER and ORR overpotentials are much lower than those of the noble metal catalysts (OER on RuO₂-rutile is 0.42 V² and ORR 143 on Pt is 0.45V³). Thus, G/CrBr₂ structure are identified as the most effective bifunctional catalyst for both OER and ORR among all the G/TMX₂ 144 structures.

145 **3.3 Intrinsic descriptors and role of electric field induced by buried salts**

146 Although the above descriptor can well predict the activities of the G/TMX₂, it is highly desirable to find an intrinsic descriptor that 147 correlate the G/TMX₂ structures with the catalytic activities. It has been shown that the TM ions carry different charge in different G/TMX₂ 148 structures due to the interactions with the haloid elements and graphene (Table S1 and S3). The TM ions with strongly positive charge could 149 induce an interfacial electric field³⁷, which significantly influences the adsorption of intermediates and thus ORR/OER occurring on the surface 150 of the graphene. We therefore consider the electrostatic force as an intrinsic descriptor to describe the catalytic activities. According to the 151 Coulomb's law, the electrostatic interaction force between TMX₂ and intermediates (F_a) may be related to the charge that the TM carries (Q) 152 and the distance between TM and graphene surface (R). Hence, we defined the new descriptor as

$$
\mathsf{F}_a = \frac{\varrho_e}{R^2} \tag{4}
$$

154 where e is the charge of an electron.

155 Figure 5. The electrostatic interaction force between TMX₂ and intermediates (F_a) of (A) Group G/TMF₂ and G/TMI₂ and (B) Group G/TMBr₂ 156 and G/TMCl₂ vs overpotentials of OER. (C) Sites at which ORR and OER could occur. The numbers are the values of overpotentials (n_{OEF}/n_{OBE}); 157 the red numbers and cycles represent the lowest overpotential of OER on this site, and the green numbers and cycles represent the lowest 158 overpotential of ORR on this site, respectively. (D) The overpotentials of OER on catalytic sites near the Cr ion of CrBr₂ as a function of the 159 electrostatic force Fa.

160 The overpotentials were plotted as a function of new descriptor in Figure 5A and B. To more clearly show the relationship for different 161 G/TMX₂ structures, we divide them into 2 groups according to the adsorption energy of TMX₂ on graphene (Table S7 and Figure S5). One 162 group includes G/TMF₂ and G/TMI₂ with the adsorption energy around 9eV, and the other consist of G/TMCI₂ and G/TMBr₂ with the 163 adsorption energy around -0.5eV. There is a volcano relationship between F_a and the overpotentials of OER. At the summit of the volcano 164 plot is G/CrBr₂, from which the best catalysts are predicted.

165 To get more insight into the role of electrical field in OER, we further calculated the distance between the TM ion and the active sites 166 (carbon atoms) in G/CrBr₂, the best bifunctional catalyst identified so far. For OER on G/CrBr₂, the relatively low overpotentials locate at C18, 167 C20 and C31 with overpotentials of 0.36V, 0.37V and 0.38V, respectively. All the sites for ORR and OER are marked in Figure 5C. We measured 168 the distance (R) from the carbon atoms to TM ion (Cr) of the G/CrBr₂ structure (Table S8). When the distance between Cr and C is less than 169 4Å, the catalytic activity shows a volcano relationship with Qe/R² (Figure 5D), indicating that the electric field induced by the adsorbed 170 molecule dominates the OER on the graphene surface. However, when the distance between Cr and C is larger than 4Å, the effect of electrical 171 field becomes very weak or even disappear. A larger model (G/CrBr₂) with 80 atoms shows the similar trend (Figure S6).

172 Unlike the OER active sites that usually locate near TMX₂, those mostly active sites for ORR are far from the metal ion and therefore do not 173 form the distinctive volcano relationship with the electrostatic force Fa (Figure S7). To understand this phenomenon, we calculated the 174 charge density on the graphene structures and found that all the active sites are located in the part with positive charge distribution (Figure 175 S8), which is consistent with the previous results²¹.

176

177 **3.4 Spin-down d band center for determining the catalytic activities**

178 For metallic catalysts, d-band center^{38,39} (denoted as ε) has been successfully used to describe the catalytic activity of transition metal 179 catalysts. According to the d-band center model, the higher the d-band center, the stronger the adsorption. However, the adsorption 180 behavior on some material surfaces cannot be well explained from the d-band center.⁴⁰⁻⁴² Since the d-orbitals' spin states can be divided into 181 spin-up and spin-down states for 3d transition metals, we take spin-down d-orbitals into consideration. Similarly, we define the spin-down 182 d-band center (denoted as ϵ_{down}) as a descriptor for the G/TMBr₂. The ϵ_{down} was calculated by

$$
z_{down} = \frac{\int_{-\infty}^{+\infty} x \rho(x) dx}{\int_{-\infty}^{+\infty} x dx}
$$
 (5)

184 where x is the energy of spin-down d-electrons and ρ is the density of state of spin-down d-electrons. The equation of $\varepsilon_{\rm down}$ is similar to the 185 calculation method for ε, but the difference between them is that the DOS is the spin-down d-DOS. The calculation results show that there 186 is a nearly proportional relationship between ε-down and ε, as shown in Figure S9. Since the metal ions in halide-adsorbed graphene play an 187 important role in the catalytic activity, we use $\varepsilon_{\text{down}}$ as a descriptor to describe the relationship between overpotentials and $\varepsilon_{\text{down}}$. As shown 188 in Figure 6, the overpotential- ε_{down} plot can be divided into two regions: for ε_{down} < 0, there are very large overpotentials for both OER and 189 ORR, indicating that the catalytic activities are very weak. When $\varepsilon_{\text{down}} > 0$, an excellent linearity can be observed between the overpotential 190 and ε_{down} for both OER and ORR. The overpotentials nearly linearly decrease with increasing ε_{down} for $\varepsilon_{down} > 0$, suggesting that the best 191 catalysts could be found by searching for the materials with the lowest spin-down d-orbitals of TM (Figure S10). A larger $\varepsilon_{\text{down}}$ means that 192 the most spin down d orbitals are with high level energy, and the materials with larger ε_{down} needs less energy to absorb the intermediates 193 in the catalytic process. Therefore, increasing $\varepsilon_{\text{down}}$ of G/TMX₂ could enhance the bifunctional catalytic activities for both OER and ORR.

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195 **4. Conclusion**

196 We have calculated the overpotentials of OER and ORR on TMX₂ (TM = Ti, V, Cr, Mn, Fe, Co, Ni, Cu, and Zn; X = F, Cl, Br, I) absorbed graphene 197 and introduced intrinsic descriptors to describe the catalytic activities of TMX₂ absorbed graphene. A design principle has been established 198 to describe the relationship between the catalytic activates and the descriptor. The catalytic activity of the G/TMX₂ are also associated with 199 the spin-down d-band center. Among the TMX₂ absorbed graphene structures, G/CrBr₂ is identified to be the best bifunctional 200 electrocatalysts for both OER and ORR with overpotentials of 0.36V and 0.35V in 4-electron pathways, respectively. G/CoF₂ is considered as 201 the best electrocatalysts to facilitate H₂O₂ in 2-electron pathways with overpotentials of 0.03V. The high catalytic activities of the CrBr₂ 202 absorbed graphene are originated from the synergistic effects of 3d transition metals, halogens adsorbed TMX₂ that leads to the charge 203 redistribution and charge transfer between CrBr₂ and graphene. Our results would provide an approach for rational design of novel hybrid 204 catalysts, which would work effectively and durably in acidic environments.

205 **Conflicts of interest**

206 There are no conflicts to declare.

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212 **References**

- 213 1 C. Sealy, Materials Today, 2008, 11, 65-68.
- 214 2 J. K. Nørskov, J. Rossmeisl, A. Logadottir, L. Lindqvist, J. R. Kitchin, T. Bligaard and H. Jonsson, *The Journal of Physical Chemistry B*, 2004, 215 108, 17886-17892.
- 216 3 I. C. Man, H. Y. Su, F. Calle-Vallejo, H. A. Hansen, J. I. Martínez, N. G. Inoglu, J. Kitchin, T. F. Jaramillo, J. K. Nørskov and J. Rossmeisl*,* 217 *ChemCatChem*, 2011, 3, 1159-1165.
- 218 4 K. Gong, F. Du, Z. Xia, M. Durstock and L. Dai, *Science*, 2009, 323, 760-764.
- 219 5 X.-K. Kong, C.-L. Chen and Q.-W. Chen, *Chemical Society Reviews*, 2014, 43, 2841-2857.
- 220 6 L. Qu, Y. Liu, J.-B. Baek and L. Dai, ACS nano, 2010, 4, 1321-1326.
- 221 7 X. Liu and L. Dai, *Nature Reviews Materials*, 2016, 1, 16064.
- 222 8 Y. Jiao, Y. Zheng, M. Jaroniec and S. Z. Qiao, *Chemical Society Reviews*, 2015, 44, 2060-2086.
- 223 9 L. Dai, Y. Xue, L. Qu, H.-J. Choi and J.-B. Baek, *Chemical reviews*, 2015, 115, 4823-4892.
- 224 10 L. Zhang, J. Niu, L. Dai and Z. Xia, *Langmuir*, 2012, 28, 7542-7550.
- 225 11 S. Wang, D. Yu, L. Dai, D. W. Chang and J.-B. Baek, *ACS Nano*, 2011, 5, 6202-6209.

- 12 A. Shen, W. Xia, L. Zhang, S. Dou, Z. Xia and S. Wang, *Nanotechnology*, 2016, 27, 185402.
- 13 Z. Zhao, L. Zhang and Z. Xia, *The Journal of Physical Chemistry C*, 2016, 120, 2166-2175.
- 14 G. Colherinhas, E. E. Fileti and V. V. Chaban, *Physical Chemistry Chemical Physics*, 2015, 17, 17413-17420.
- 15 C. Yang, J. Chen, X. Ji, T. P. Pollard, X. Lü, C.-J. Sun, S. Hou, Q. Liu, C. Liu and T. Qing, *Nature*, 2019, 569, 245.
- 16 G. Cassone, P. V. Giaquinta, F. Saija, and A. M. Saitta, The Journal of Physical Chemistry B, 2014, 118, 12717-12724.
- 17 G. Cassone, P. V. Giaquinta, F. Saija and A. M. Saitta, *The Journal of chemical physics*, 2015, 142, 054502.
- 18 F. Creazzo, D. R. Galimberti, S. Pezzotti and M. P. Gaigeot, *The Journal of chemical physics*, 2019, 150, 041721.
- 19 Z. Zhao, M. Li, L. Zhang, L. Dai and Z. Xia, *Advanced Materials*, 2015, 27, 6834-6840.
- 20 C. Y. Lin, L. Zhang, Z. Zhao and Z. Xia, Advanced Materials, 2017, 29, 1606635.
- 21 C. Zhu, J. Yun, Q. Wang and G. Yang, *Applied Surface Science*, 2018, 435, 329-337.
- 22 G. Kresse and J. Furthmüller, *Computational materials science*, 1996, 6, 15-50.
- 23 G. Kresse and J. Furthmüller, *Physical Review B*, 1996, 54, 11169.
- 24 P. E. Blöchl, *Physical review B*, 1994, 50, 17953.
- 25 J. P. Perdew, K. Burke and M. Ernzerhof, *Physical Review Letters*, 1996, 77, 3865.
- 26 I. Solovyev, P. Dederichs and V. Anisimov, *Physical Review B*, 1994, 50, 16861.
- 241 27 K. Kinoshita, Electrochemical oxygen technology, John Wiley & Sons, 1992.
- 242 28 B. C. Steele and A. Heinzel, in Materials For Sustainable Energy: A Collection of Peer-Reviewed Research and Review Articles from Nature Publishing Group, World Scientific, 2011, pp. 224-231.
- 29 Z. Zhao, L. Zhang, C. Y. Lin and Z. Xia, Carbon-Based Metal-Free Catalysts: Design and Applications, 2018, 1, 1-33.
- 30 J. K. Nørskov, J. Rossmeisl, A. Logadottir, L. Lindqvist, J. R. Kitchin, T. Bligaard and H. Jonsson, *The Journal of Physical Chemistry B*, 2004, 108, 17886-17892.
- 31 M. Li, L. Zhang, Q. Xu, J. Niu and Z. Xia*, Journal of Catalysis*, 2014, 314, 66-72.
- 32 F. Calle-Vallejo, J. I. Martinez and J. Rossmeisl, *Physical Chemistry Chemical Physics*, 2011, 13, 15639-15643.
- 249 33 J. Rumble, CRC handbook of chemistry and physics, CRC press, 2017.
- 34 R. A. Sidik, A. B. Anderson, N. P. Subramanian, S. P. Kumaraguru and B. N. Popov, *The Journal of Physical Chemistry B*, 2006, 110, 1787- 1793.
- 35 Y. Okamoto, Applied Surface Science, 2009, 256, 335-341.
- 36 A. Verdaguer-Casadevall, D. Deiana, M. Karamad, S. Siahrostami, P. Malacrida, T. W. Hansen, J. Rossmeisl, I. Chorkendorff and I. E. Stephens, *Nano Letters*, 2014, 14, 1603-1608.
- 37 T. Meng, Y.-N. Hao, J. Qin and M. Cao, *ACS Sustainable Chemistry & Engineering*, 2019, 7, 4657-4665.
- 38 B. Hammer and J. Norskov, *Nature*, 1995, 376, 238.
- 39 A. Ruban, B. Hammer, P. Stoltze, H. L. Skriver and J. K. Nørskov, *Journal of Molecular Catalysis A: Chemical*, 1997, 115, 421-429.
- 40 M. Qiu, Z. Fang, Y. Li, J. Zhu, X. Huang, K. Ding, W. Chen and Y. Zhang, *Applied Surface Science*, 2015, 353, 902-912.
- 41 F. Calle-Vallejo, J. Martínez, J. M. García-Lastra, J. Rossmeisl and M. Koper, *Physical review letters*, 2012, 108, 116103.
- 42 H. Xin and S. Linic, *The Journal of Chemical Physics*, 2010, 132, 221101.