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Complete List of Authors:	Dimitrievska, Mirjana; National Renewable Energy Laboratory, ; National Institute of Standards and Technology, NCNR Chong, Marina; Pacific Northwest National Laboratory, Bowden, Mark; Pacific Northwest National laboratory, Energy Processes & Materials Division Wu, Hui; University of Maryland, Department of Materials Science and Engineering Zhou, Wei; National Institute of Standards and Technology, NCNR Nayyar, Iffat; Pacific Northwest National Laboratory Ginovska, Bojana; Pacific Northwest National Laboratory Gennett, Thomas; NREL, Autrey, Tom; Pacific Northwest National Laboratory, FCS Jensen, Craig; Department of Chemistry, University of Hawaii at Manoa Udovic, Terrence; National Institute of Standards and Technology,

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# Structural and Reorientational Dynamics of Tetrahydroborate (BH<sub>4</sub>-) and Tetrahydrofuran (THF) in Mg(BH<sub>4</sub>)<sub>2</sub>·3THF Adduct: Neutron-Scattering Characterization

Mirjana Dimitrievska,\*<sup>1,2</sup> Marina Chong,<sup>3</sup> Mark E. Bowden,<sup>3</sup> Hui Wu,<sup>2</sup> Wei Zhou,<sup>2</sup> Iffat Nayyar,<sup>3</sup> Bojana Ginovska,<sup>3</sup> Thomas Gennett,<sup>1</sup> Tom Autrey,<sup>3</sup> Craig M. Jensen,<sup>4</sup> and Terrence J. Udovic<sup>2</sup>

<sup>1</sup>National Renewable Energy Laboratory (NREL), 5013 Denver W Pkwy, Golden, CO 80401, United States

<sup>2</sup>NIST Center for Neutron Research, National Institute of Standards and Technology, Gaithersburg, MD 20899-6102, United States

<sup>3</sup>Pacific Northwest National Laboratory (PNNL), Richland, WA 99352, United States

<sup>4</sup>University of Hawaii Manoa, Department of Chemistry, Honolulu, HI 96822-2275, United States

\*Author to whom correspondence should be addressed. E-mail: <u>mirjana.dimitrievska@nist.gov;</u> <u>mirjana.dimitrievska@nrel.gov;</u>

#### Abstract

Metal borohydrides are considered promising materials for hydrogen storage applications due to a high volumetric and gravimetric hydrogen density. Recently, different Lewis bases have been complexed with  $Mg(BH_4)_2$  in efforts to improve hydrogenation/dehydrogenation properties. Notably,  $Mg(BH_4)_2 \cdot xTHF$ adducts involving tetrahydrofuran (THF; C<sub>4</sub>H<sub>8</sub>O) have proven to be especially interesting. This work focuses on exploring physicochemical properties of the THF-rich Mg(BH<sub>4</sub>)<sub>2</sub>·3THF adduct using neutronscattering methods and molecular DFT calculations. Structural analysis, based on neutron diffraction measurements of  $Mg(^{11}BH_4)$ , 3TDF (D - deuterium), has confirmed a lowering of the symmetry upon cooling, from monoclinic C2/c to P-1 via a triclinic distortion. Vibrational properties are strongly influenced by the THF environment; showing a splitting in spectral features as a result of changes in the bond lengths, force constants, and lowering of the overall symmetry. Interestingly, the orientational mobilities of the BH<sub>4</sub><sup>-</sup> anions obtained from quasielastic neutron scattering (QENS) are not particularly sensitive to the presence of THF and compare well with the mobilities of BH4- anions in unsolvated Mg(BH<sub>4</sub>)<sub>2</sub>. The QENS data point to uniaxial 180° jump reorientations of the BH<sub>4</sub><sup>-</sup> anions around a preferred  $C_2$  anion symmetry axis. The THF rings are also found to be orientationally mobile, undergoing 180° reorientational jumps around their  $C_2$  molecular symmetry axis with jump frequencies about an order of magnitude lower than those for the BH<sub>4</sub><sup>-</sup> anions. In contrast, no dynamical behavior of the THF rings is observed with QENS for a more THF-deficient  $2Mg(BH_4)_2$  THF adduct. This lack of comparable THF mobility may reflect a stronger Mg<sup>2+</sup>-THF bonding interaction for lower THF/Mg(BH<sub>4</sub>)<sub>2</sub> stoichiometric ratios, which is consistent with DFT calculations showing a decrease in the binding energy with each additional THF ring in the adduct. Based on the combined experimental and computational results, we propose that using fractions of THF to Mg(BH<sub>4</sub>)<sub>2</sub> is beneficial in (i) preventing weakly bound THF from coming free from the Mg<sup>2+</sup> cation and reducing the concentration of an unwanted impurity in the hydrogen and (ii) disrupting the stability of crystalline phase leading to a lower melting point and enhanced kinetics for any potential hydrogen storage applications.

## Introduction

Using hydrogen as an energy carrier provides an opportunity to achieve zero-emission propulsion, which will help to reduce the consumption of fossil fuels and limit the extent of anthropogenically contributions to climate change.<sup>1</sup> Storing hydrogen in chemical bonds, i.e., hydrogen storage materials, is one approach to storing and transporting hydrogen that provides for greater volumetric and gravimetric hydrogen densities compared to compressed hydrogen gas. Hydrogen storage materials must fulfill a wide range of requirements to function as hydrogen storage materials. Among those, high hydrogen content, fast kinetics of hydrogen desorption and absorption at low operating temperature and pressure, as well as high purity of the released hydrogen are the most important. Promising alternatives for hydrogen storage include carbon nanostructures<sup>2,3</sup>, liquid organic hydrogen carriers<sup>4</sup>, metal organic frameworks<sup>5</sup> and complex borohydrides.<sup>6</sup>

Metal borohydrides  $[M(BH_4)_n]$  are considered as attractive materials for hydrogen storage applications due to their high volumetric and gravimetric hydrogen densities.  $M(BH_4)_n$  are largely ionic compounds composed of tetrahedral hydrogen-rich molecular  $BH_4^-$  anions counterbalanced by metal cations  $M^{n+}$ . Among the alkali- and alkaline-earth-based borohydrides, magnesium borohydride  $[Mg(BH_4)_2]$  displays very interesting properties both from fundamental and applicative points of view.  $Mg(BH_4)_2$  has a more favorable thermodynamic stability in comparison to, e.g., LiBH<sub>4</sub>, while still maintaining attractive hydrogen storage capacities of 14.9 mass %.<sup>7,8</sup> However, the reaction kinetics of  $Mg(BH_4)_2$ , which is still not fully understood, along with its decomposition pathways occurring in a multi-step process only at 600 K,<sup>9,10</sup> and its limited reversibility,<sup>11</sup> make  $Mg(BH_4)_2$  still unsuitable for practical applications.

Recently, different additives have been used in order to enhance the kinetics of hydrogen release from  $Mg(BH_4)_2$ .<sup>12–15</sup> These studies have found that the onset temperature of hydrogen release could be reduced upon addition of metal fluorides<sup>13–15</sup> or by mechanically milling  $Mg(BH_4)_2$  with TiO<sub>2</sub>.<sup>15</sup> Additionally, accelerated thermal dehydrogenation of  $Mg(BH_4)_2$  was observed in mixtures with LiBH<sub>4</sub>.<sup>16–18</sup>

Several studies have also suggested that adducts of  $Mg(BH_4)_2$  complexed with solvent molecules might lead to beneficial enhancements in hydrogen kinetics.<sup>19–23</sup> Among these, adducts with tetrahydrofuran (THF = C<sub>4</sub>H<sub>8</sub>O) have proven to be especially interesting, as it was found that THF binds to the Mg<sup>2+</sup> and appears to modify its chemistry, increasing both selectivity and kinetics for H<sub>2</sub> release (and reversible uptake).<sup>23</sup> However, the specific role of THF on the hydrogenation and dehydrogenation process is still ambiguous, and as such, in-depth foundational studies of the THF interactions with the BH<sub>4</sub><sup>-</sup> and Mg<sup>2+</sup> may provide further insight.

Investigating physicochemical properties of  $Mg(BH_4)_2 \cdot xTHF$  adducts constitutes a challenge from an experimental point of view, as selectivity to the different components, e.g.,  $BH_4^-$  and THF, is required. Neutron scattering techniques are invaluable in this case, as they offer selectivity to a given component through isotopic (H/D) labeling. In the case of incoherent neutron scattering techniques (neutron vibrational spectroscopy (NVS) and quasielastic neutron scattering (QENS)) related to hydrogen selfmotions, deuteration can significantly reduce the scattered intensity, thereby masking the deuterated component to neutrons. Additionally, QENS also provides space/time resolution at a molecular level through the momentum transfer dependence of the magnitudes measured. On the other hand, structural information, which is crucial for the understanding of vibrational and dynamical results, can be provided from neutron powder diffraction measurements (NPD).

This work focuses on exploring structural, vibrational, and dynamical properties of the THF-rich  $Mg(BH_4)_2$ ·3THF adduct through neutron-scattering methods and molecular DFT calculations. Some additional limited experimental and theoretical comparisons with  $Mg(BH_4)_2$  and the relatively THF-

poorer Mg(BH<sub>4</sub>)<sub>2</sub>·(2/3)THF adduct provide further insights concerning the dependences of BH<sub>4</sub><sup>-</sup> and THF dynamics on adduct stoichiometry. In some experiments, selective masking of the THF molecules within the Mg(BH<sub>4</sub>)<sub>2</sub>·*x*THF adducts via their deuteration, allowed us to isolate and characterize more effectively the individual behaviors of the BH<sub>4</sub><sup>-</sup> anions and THF molecules.

## **Experimental Details**

## Materials

To facilitate neutron scattering measurements, all samples were prepared using <sup>11</sup>B-enriched Mg(<sup>11</sup>BH<sub>4</sub>)<sub>2</sub> from Katchem<sup>24</sup> as a starting material; this leads to a significant reduction of the neutron absorption otherwise due to the <sup>10</sup>B isotope in natural boron. Moreover, for some neutron scattering measurements where appropriate, synthesized adducts with fully deuterated THF (i.e., TDF) were used.

All sample preparation and storage was conducted either in a nitrogen glovebox or on a Schlenk line. Solvents were dried over molecular sieves and verified by NMR for purity before use.

Two Mg(BH<sub>4</sub>)<sub>2</sub>·*x*THF adduct stoichiometries were prepared with THF/Mg(BH<sub>4</sub>)<sub>2</sub> stoichiometric ratios x = 3 and 0.5, as previously described in ref. [23]. Briefly, diethyl ether was added to a round-bottom flask containing 400 mg  $\alpha$ -phase Mg(BH<sub>4</sub>)<sub>2</sub> until the borohydride was dissolved. In a subsequent step, a stoichiometric quantity, 0.5 or 3 equivalents of THF or TDF per Mg(BH<sub>4</sub>)<sub>2</sub> was added by micropipette to the ether solution at room temperature and stirred for 30 min. Excess ether was then removed in vacuo at room temperature for 12 hours to yield the THF adducts of Mg(BH<sub>4</sub>)<sub>2</sub>. This vacuum treatment did not remove THF, since prolonged exposure of Mg(BH<sub>4</sub>)<sub>2</sub>·3THF under vacuum did not change its stoichiometry or crystallinity.

Room-temperature XRD data for  $Mg(BH_4)_2 \cdot 3THF$  confirmed formation of the known monoclinic structure. The indexing of comparable XRD data for the more THF-deficient  $2Mg(BH_4)_2 \cdot THF$  (i.e., x = 0.5) also suggested primarily a single phase with monoclinic symmetry but of as-of-yet unknown structure and exact stoichiometry.

## Methods

All neutron scattering measurements were performed at the National Institute of Standards and Technology Center for Neutron Research using 0.5 g of each powdered sample mounted in a thin-walled annular geometry (5 cm height  $\times$  1.2 cm annulus diameter) inside a He-filled, indium o-ring-sealed, cylindrical Al sample cell.

Neutron powder diffraction (NPD) measurements were performed between 5 K and 300 K on the BT-1 High-Resolution Neutron Powder Diffractometer<sup>25</sup> with the Cu(311) monochromator at  $\lambda = 1.5397(2)$  Å and an in-pile collimation of 60 min of arc. Neutron vibrational spectroscopy (NVS) measurements were performed at 4 K on the Filter-Analyzer Neutron Spectrometer (FANS)<sup>26</sup> using the Cu(220) monochromator with pre- and post-collimations of 20' of arc, yielding a full-width-at-half-maximum (fwhm) energy resolution of about 3 % of the neutron energy transfer. QENS measurements were performed up to 315 K on two complementary instruments: the Disc Chopper Spectrometer (DCS)<sup>27</sup> between 4 K and 315 K using incident neutron wavelengths ( $\lambda$ ) of 4.5 Å, 10 Å and 12 Å with respective fwhm resolutions of 67 µeV, 17 µeV and 10.9 µeV and respective maximum attainable elastic scattering vector (*Q*) values of around 2.62 Å<sup>-1</sup>, 1.18 Å<sup>-1</sup> and 0.98 Å<sup>-1</sup>; and the High Flux Backscattering fixed-window scan (FWS)] using an incident neutron wavelength of 6.27 Å (2.08 meV) with a fwhm resolution of 0.8 µeV and a maximum attainable *Q* value of 1.75 Å<sup>-1</sup>. Instrumental resolution functions were

determined from QENS spectra at lower temperatures free of quasielastic scattering, i.e., at 4 K. QENS data were analyzed using the DAVE software package.<sup>29</sup>

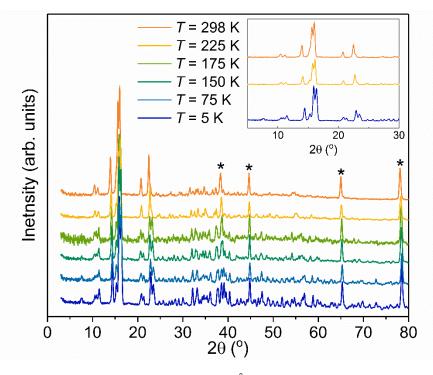
Solid-state density-functional theory (DFT) calculations were performed using the Quantum-Espresso package.<sup>30</sup> A cutoff energy of 544 eV and a  $2\times2\times1$  k-point mesh (generated using the Monkhorst-Pack scheme) were used and found to be enough for the total energy to converge within 0.01 meV per atom. Crystal structures were first fully optimized, based on the experimental structures, and are presented in the Supporting Information. For comparison with the NVS measurements, simulated phonon densities of states (PDOSs) were generated from the DFT-optimized structures using the supercell method ( $2\times2\times1$  cell size) with finite displacements (0.02 Å) and were appropriately weighted to take into account the total neutron scattering cross sections of the different elements. Gas phase DFT calculations were done using the B3LYP functional<sup>31–34</sup> and the gaussian 6-31+G\* basis set<sup>35,36</sup> to calculate the binding energies of 1<sup>st</sup>, 2<sup>nd</sup> and a 3rd THF molecule to Mg(BH<sub>4</sub>)<sub>2</sub>. The optimized geometries were verified by the absence of imaginary frequencies from the frequency analysis in the harmonic oscillator/rigid-rotor approximation. The frequencies were also used to calculate the enthalpic and entropic contributions to the free energy for binding. All calculations were done using the NWChem software.<sup>37</sup>

Structural depictions were made using the VESTA (Visualization for Electronic and Structural Analysis) software.<sup>38</sup> For all figures, standard uncertainties are commensurate with the observed scatter in the data, if not explicitly designated by vertical error bars.

#### **Results and discussion**

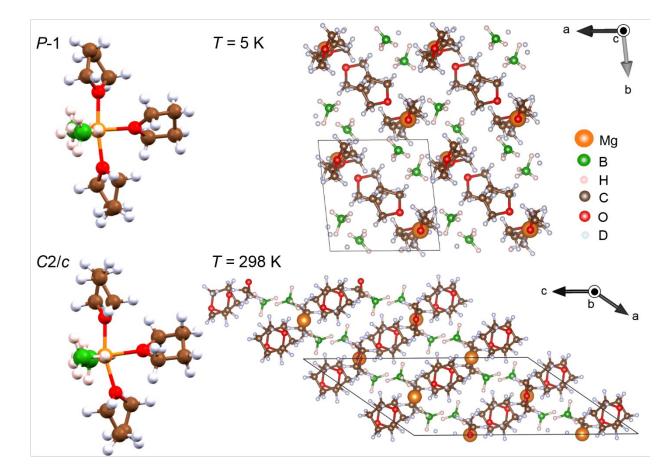
#### Structural behavior and vibrational dynamics of Mg(BH<sub>4</sub>)<sub>2</sub>·3THF

Neutron powder diffraction measurements of the partially deuterated sample,  $Mg(BH_4)_2$  3TDF, upon heating from 5 K to 298 K indicate a phase transition in the temperature range between 175 and 225 K, as shown in Figure 1. Prior low-temperature (100 K) and room-temperature (298 K) XRD measurements of Mg(BH<sub>4</sub>)<sub>2</sub>·3THF suggested two crystalline forms of this material with triclinic (space group P-1)<sup>39</sup> and monoclinic (space group C2/c)<sup>40</sup> structures, respectively. To confirm these structures, Rietveld refinements of the NPD patterns (at 298 K and 225 K) above and (at 5 K) below this phase transition were performed. The analysis corroborated a lowering of the symmetry upon decreasing the temperature, from monoclinic C2/c to P-1 via a triclinic distortion from respective  $\alpha$ ,  $\beta$ , and  $\gamma$  values of 90°, 144.992(4)°, and 90° at 298 K to 102.596(3)°, 103.984(4)°, and 92.239(3)° at 5 K. Figure S1 in the Supporting Information (SI) presents the model refinements of the NPD data at 298 K, 225 K, and 5 K. Table S1 and supplemental CIF files, also in the Supporting Information, summarize the structural analysis details. It should be noted that the lattice parameters from ref. [40] were associated with a different unit cell convention than used in this paper. Converting these parameters to the present convention yields values of a = 12.120 Å, b = 12.646 Å, c = 20.216 Å, and  $\beta = 144.96^{\circ}$ , which are in good agreement with our values. We note that the slightly smaller lattice constants associated with the present results are consistent with the incorporation of TDF, instead of THF, in our NPD samples. Molecular vibrational amplitudes associated with deuterated molecules are somewhat smaller than hydrogenated analogues, leading to slightly less space required to accommodate the former molecules compared to that for the latter.



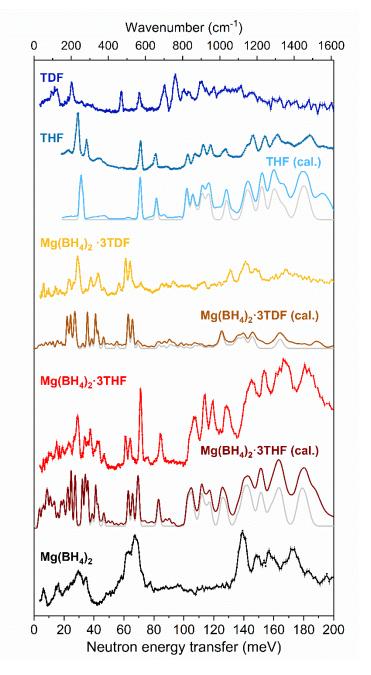
**Figure 1.** Progression of NPD patterns  $[\lambda = 1.5397 \text{ Å}]$  of Mg(BH<sub>4</sub>)<sub>2</sub>·3TDF with temperature, upon heating from 5 K to 298 K, showing the structural transformation from triclinic *P*-1 symmetry to monoclinic *C*2/*c* symmetry. Inset shows patterns at 5 K, 225 K, and 298 K for clarity to distinguish from the intervening patterns measured at other temperatures. Asterisks mark the reflections obtained from the Al container.

Figure 2 shows the molecular and crystal structures of the low- and room-temperature Mg(BH<sub>4</sub>)<sub>2</sub>·3TDF phases. It is apparent that the packing of molecules in both crystalline forms takes place in a layered lattice, with interchangeable layers of  $BH_4^-$  anions and TDF rings. The three TDF molecules and two  $BH_4^$ anions are located around the central Mg<sup>2+</sup> cation in a trigonal bipyramidal motif. The BH<sub>4</sub><sup>-</sup> anions are bonded to the Mg<sup>2+</sup> cation via double hydrogen bridges. The coordination polyhedron of Mg<sup>2+</sup> can be described by a distorted pentagonal-bipyramidal geometry, where the axial vertices are occupied by the O atoms of the two apical TDF rings, while the four bridging H atoms from the  $BH_4^-$  anions and the O atom from the third TDF ring lie mostly in the equatorial plane. The detachment of the apical TDF rings from the BH<sub>4</sub><sup>-</sup> anions is observed from the O-Mg-O angles of 172.1° and 163.6° in the respective P-1 and C2/cstructures, as well as from the slightly longer Mg-O distances to the apical TDF molecules. However, the arrangement of the Mg, O, and two B atoms in the equatorial plane is perfectly coplanar, with low deviations from the ideal angle of 120° in a trigonal bipyramidal coordination geometry. Higher values of the B-Mg-B angle in comparison to both B-Mg-O angles are observed due to a more repulsive force between the negatively charged BH<sub>4</sub><sup>-</sup> anions than between the coordinated TDF ring and BH<sub>4</sub><sup>-</sup> anion. No significant distortion of the TDF rings or  $BH_4^-$  anions have been observed in either crystalline form, with bond lengths and bond angles not being different from the expected values.<sup>41</sup> However, the interatomic Mg and B separations of 2.49 Å in P-1 and 2.46 Å in  $C^{2/c}$  are slightly greater than 2.41 Å reported for pristine  $\alpha$ -Mg(BH<sub>4</sub>)<sub>2</sub>.<sup>41</sup>



**Figure 2.** Molecular and crystal structures of the low-temperature (*P*-1 symmetry) and room-temperature (C2/c symmetry) Mg(BH<sub>4</sub>)<sub>2</sub>·3TDF phases.

The neutron vibrational spectra for  $Mg(BH_4)_2 \cdot 3THF$  and  $Mg(BH_4)_2 \cdot 3TDF$  at 4 K based on FANS measurements are shown in Figure 3 and compared with the simulated phonon density of states (PDOS) from the DFT-optimized triclinic structure, indicating overall good agreement between experiment and theory. Further information about the character, symmetry, and energies of the different phonon modes contributing to the simulated PDOSs can be found in the Supporting Information.



**Figure 3.** Neutron vibrational spectra for Mg(BH<sub>4</sub>)<sub>2</sub>·3THF (red) and Mg(BH<sub>4</sub>)<sub>2</sub>·3TDF (orange) at 4 K compared with the simulated 0 K 1-phonon- (light gray) and 1+2-phonon-density-of-states spectra (dark red and brown) after DFT energy optimization of the triclinic *P*-1-symmetric structure derived from the 5 K NPD data. Neutron vibrational spectra of pristine  $\alpha$ -Mg(BH<sub>4</sub>)<sub>2</sub> (black), THF (blue), and TDF (dark blue), as well as the simulated 0 K 1-phonon- (light gray) and 1+2-phonon-density-of-states spectrum (light blue) of THF are shown for reference and comparison. The Cu(220) and pyrolytic graphite (002) monochromators were used above and below 40 meV, respectively. Vertical error bars denote ±1  $\sigma$ .

Upon comparison of the Mg(BH<sub>4</sub>)<sub>2</sub>·3THF NVS spectrum with that of pristine  $\alpha$ -Mg(BH<sub>4</sub>)<sub>2</sub>, obvious peak splitting in the lower frequency region (< 80 meV) is observed. This is mostly due to the coordination of THF rings, which affect the bonding between the Mg atoms and BH<sub>4</sub>- anions. Based on the structural data,

the presence of THF rings is causing extension of the interatomic distance between the Mg and B atoms and reduction in the B-Mg-B angle in comparison to the pristine  $\alpha$ -Mg(BH<sub>4</sub>)<sub>2</sub>.

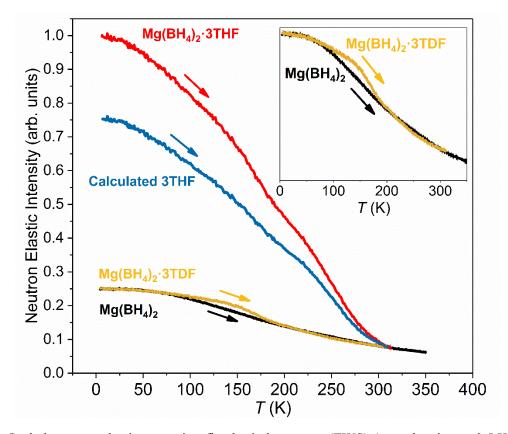
Direct comparison between NVS spectra of  $Mg(BH_4)_2 \cdot 3THF$  and  $Mg(BH_4)_2 \cdot 3TDF$ , as well as the mode identification obtained from the DFT PDOS calculations, can help distinguish the modes dominated by the vibrational displacements of the  $BH_4^-$  anions and those of the THF molecules. This is due to the fact that, while the TDF spectrum by itself is measurable as can be seen in Figure 3, its overall contribution to the  $Mg(BH_4)_2 \cdot 3TDF$  spectrum is very small, thus accentuating the  $BH_4^-$  contributions. Additionally, it should be mentioned that the noticeable shift in peak positions between the crystalline THF and TDF compounds is due to the mass difference between H and D atoms, where heavier D atoms cause peak-shifting to lower energy transfers.

Analyzing the spectra in this way, most peaks that are observed in the low-energy region from 20 to 65 meV (1 meV  $\approx 8.066$  cm<sup>-1</sup>) for Mg(BH<sub>4</sub>)<sub>2</sub>·3THF in Figure 3 correspond to the BH<sub>4</sub>- librational (torsional) modes.<sup>42</sup> Moving to higher energy-transfer values, two sharp features are observed at around 70 and 84 meV, and are attributed to vibrations related to THF ring deformations and CH<sub>2</sub> wagging.<sup>43</sup> Furthermore, a number of sharp peaks are noticeable in the 100 - 190 meV spectral region of Mg(BH<sub>4</sub>)<sub>2</sub>·3THF, of which eight peaks centered at around 107, 114, 120, 128, 144, 154, 167 and 182 meV are particularly intense. According to group-theory calculations,<sup>44,45</sup> BH<sub>4</sub>- anions have a high tetrahedral ( $T_d$ ) symmetry for which the irreducible representation in the Mulliken notation can be written as  $\Gamma = A_1 \oplus E \oplus 2F_2$ , while THF rings have two-fold rotational ( $C_2$ ) symmetry with the irreducible representation given as  $\Gamma$  =  $17A \oplus 16B$ . Here the A and B modes are non-degenerate, while the E modes are doubly degenerate, and the F modes are triply degenerate. Having in mind this notation and comparing the results with the DFT simulations and the experimental spectra of pristine  $\alpha$ -Mg(BH<sub>4</sub>)<sub>2</sub>, THF, and TDF, the very intense peaks observed in the 100 - 190 meV region could be resolved as follows: the peaks observed at 144 and 154 meV mostly correspond to the asymmetric bending vibrations of  $BH_4^-$  (F<sub>2</sub> symmetry mode), while the peaks at 167 and 183 meV are attributed to the symmetric bending vibrations of BH<sub>4</sub><sup>-</sup> (E symmetry mode). The lower energy peaks at 107, 114, 120, 128, 144 meV are related to the A and B modes of THF, corresponding to mostly stretching vibrations.

#### **Reorientational Dynamics of Mg(BH<sub>4</sub>)<sub>2</sub>·3THF**

The neutron-elastic-scattering fixed-window scan (FWS) presents a plot of the temperature dependence of the neutron scattering intensity at zero energy transfer. As such, a direct comparison of the FWS scans measured from the different compounds provided some insights into the reorientational dynamics of the THF and BH<sub>4</sub><sup>-</sup> groups in these materials. Figure 4 shows the FWS for Mg(BH<sub>4</sub>)<sub>2</sub>·3THF and Mg(BH<sub>4</sub>)<sub>2</sub>·3TDF measured on HFBS at Q = 0.87 Å<sup>-1</sup> between 4 K and 315 K [limited by the decomposition temperature of Mg(BH<sub>4</sub>)<sub>2</sub>·3THF]. The FWS for pristine  $\alpha$ -Mg(BH<sub>4</sub>)<sub>2</sub> was measured as a reference in a similar temperature region, from 4 K to 350 K. Again, due to the much lower incoherent neutron scattering cross section of D in comparison to H, the contribution of TDF molecules to the FWS could be neglected (to first order), allowing for all FWS to be scaled according to their number of H atoms. The FWS plots of Mg(BH<sub>4</sub>)<sub>2</sub>·3THF at the lowest measured temperature of 4 K. Assuming that the FWS for Mg(BH<sub>4</sub>)<sub>2</sub>·3TDF reflects only the dynamical contributions of the BH<sub>4</sub><sup>-</sup> anions allows the estimation of the dynamical contributions from solely the three THF molecules in Mg(BH<sub>4</sub>)<sub>2</sub>·3THF

(labeled in Figure 4 as "Calculated 3THF") by subtracting the FWS of  $Mg(BH_4)_2$ ·3TDF from that of  $Mg(BH_4)_2$ ·3THF.

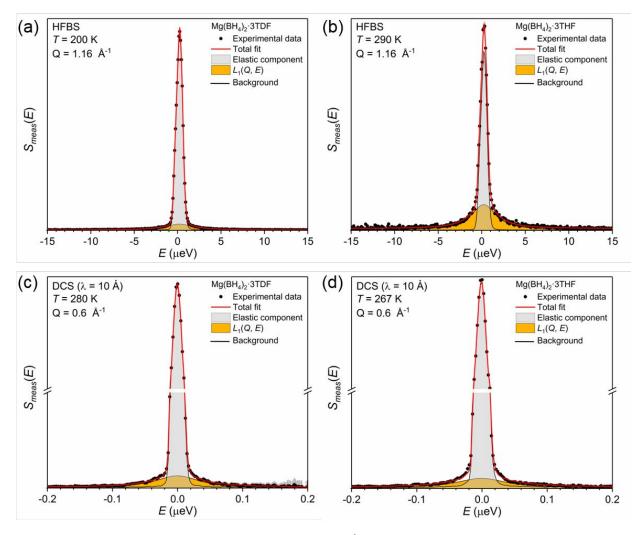


**Figure 4.** Scaled neutron-elastic-scattering fixed-window scans (FWS) (upon heating at 0.5 K min<sup>-1</sup>) for Mg(BH<sub>4</sub>)<sub>2</sub>·3THF and Mg(BH<sub>4</sub>)<sub>2</sub>·3TDF measured between 4 K and 315 K, at Q = 0.87 Å<sup>-1</sup> with 0.8 µeV fwhm instrumental resolution on HFBS. FWS of pristine  $\beta$ -Mg(BH<sub>4</sub>)<sub>2</sub> was measured as a reference in the similar temperature region, from 4 K to 350 K under the same conditions. The scan labeled "Calculated 3THF" was obtained by subtracting the FWS of Mg(BH<sub>4</sub>)<sub>2</sub>·3TDF from that of Mg(BH<sub>4</sub>)<sub>2</sub>·3THF using the procedure explained in the text. Inset shows zoomed comparison of the FWS scans from Mg(BH<sub>4</sub>)<sub>2</sub>·3TDF and  $\alpha$ -Mg(BH<sub>4</sub>)<sub>2</sub>.

Upon heating from 4 K, a fall-off in intensity with increasing temperature occurs for all scans depicted in Figure 4, reflecting the composite Debye-Waller-type hydrogen vibrational displacements associated with the THF and  $BH_4^-$  groups. Above certain threshold temperatures (at around 230 K for THF and 160 K for  $BH_4^-$  in the adducts) the fall-offs accelerate, signaling that THF and  $BH_4^-$  reorientational jumps are coming into the HFBS resolution window with frequencies beginning to exceed  $10^7$ - $10^8$  s<sup>-1</sup>. As temperatures further increase, the fall-off in intensity begins to slow again, signaling that the reorientational jump frequencies are now approaching the order of  $10^{10}$  s<sup>-1</sup> and moving out of the HFBS dynamic range window. For Mg(BH<sub>4</sub>)<sub>2</sub>·3THF, this seems to occur above around 300 K for THF and 190 K for BH<sub>4</sub><sup>-</sup>.

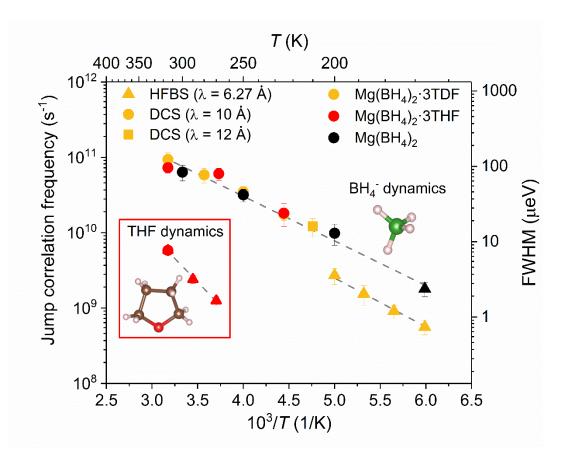
Comparison of the FWS scans from Mg(BH<sub>4</sub>)<sub>2</sub>·3TDF and pristine  $\alpha$ -Mg(BH<sub>4</sub>)<sub>2</sub>, presented in the inset in Figure 4, shows differences in the slopes of the two curves in the lower-temperature region, up to 200 K, leading to the conclusion that the BH<sub>4</sub><sup>-</sup> anions in Mg(BH<sub>4</sub>)<sub>2</sub>·3TDF have lower reorientational mobilities compared to the BH<sub>4</sub><sup>-</sup> anions in Mg(BH<sub>4</sub>)<sub>2</sub> at the same temperature in this region. However, above 200 K, BH<sub>4</sub><sup>-</sup> anions in both compounds behave the same, based on the overlap of the FWS plots. The drastic change of the reorientational mobilities of the BH<sub>4</sub><sup>-</sup> anions around the 200 K temperature region coincides with the triclinic to monoclinic structural phase transition for Mg(BH<sub>4</sub>)<sub>2</sub>·3TDF, as explained previously by the NPD experiments. Additionally, the noticeable change in slope above about 175 K for the FWS component intensity for THF in Mg(BH<sub>4</sub>)<sub>2</sub>·3THF likely marks a change in the Debye-Waller factor for THF due to this same phase transition.

Complementary QENS measurements on the DCS and HFBS instruments were undertaken to gain further insights concerning the reorientational behavior of both THF and BH<sub>4</sub><sup>-</sup> groups. Figure 5 presents two exemplary QENS spectra for Mg(BH<sub>4</sub>)<sub>2</sub>·3TDF measured at (a) 280 K and (c) 200 K, and two for Mg(BH<sub>4</sub>)<sub>2</sub>·3THF measured at (b) 267 K and (d) 290 K, on two instruments at different resolutions [(a) and (b) on HFBS with 0.8 µeV resolution; and (c) and (d) on DCS with 16.9 µeV resolution]. All QENS spectra were adequately fit with a delta function and a single Lorentzian component, both convoluted with the resolution function, above a flat background. For all temperatures, the width of the Lorentzian function,  $\Gamma$ , associated with each component group was found to be *Q*-independent over the measured *Q*-range (Figure S2 in the Supporting Information), consistent with simple localized hydrogen jump motions<sup>46</sup> for both THF and BH<sub>4</sub><sup>-</sup> groups, as opposed to long-range diffusive motions.



**Figure 5.** QENS spectra measured on HFBS at  $Q = 1.16 \text{ Å}^{-1}$ , 0.8 µeV resolution, for Mg(BH<sub>4</sub>)<sub>2</sub>·3TDF at (a) 200 K, and Mg(BH<sub>4</sub>)<sub>2</sub>·3THF at (b) 290 K. QENS spectra measured on DCS at  $Q = 0.6 \text{ Å}^{-1}$ , 16.9 µeV resolution, for Mg(BH<sub>4</sub>)<sub>2</sub>·3TDF at (c) 280 K, and Mg(BH<sub>4</sub>)<sub>2</sub>·3THF at (b) 267 K.

Figure 6 conveys the Arrhenius-like dependencies between 160 K and 320 K of the fundamental jump correlation frequencies  $\tau_1^{-1} = \Gamma/(2\hbar)$  vs.  $T^{-1}$  for Mg(BH\_4)<sub>2</sub>·3TDF, Mg(BH\_4)<sub>2</sub>·3THF, and pristine  $\alpha$ -Mg(BH\_4)<sub>2</sub>.



**Figure 6.** Jump correlation frequencies  $\tau_{i1}^{-1}$  vs. inverse temperature for Mg(BH<sub>4</sub>)<sub>2</sub>·3TDF, Mg(BH<sub>4</sub>)<sub>2</sub>·3THF, and pristine Mg(BH<sub>4</sub>)<sub>2</sub>. Different symbols denote the different instruments used. Different colors denote the different samples. Dashed lines represent linear fits to the QENS data.

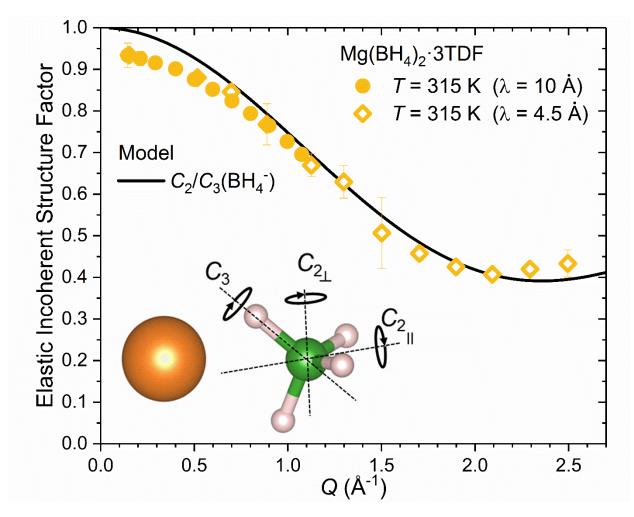
Similarities in the BH<sub>4</sub><sup>-</sup> reorientational jumps for Mg(BH<sub>4</sub>)<sub>2</sub>·3TDF, Mg(BH<sub>4</sub>)<sub>2</sub>·3THF and unsolvated Mg(BH<sub>4</sub>)<sub>2</sub> lead to the conclusion that the mobilities of the BH<sub>4</sub><sup>-</sup> anions are not noticeably affected by the presence of THF rings in the higher-temperature region from 320 K to 200 K, displaying jump correlation frequencies of the order of 10<sup>10</sup> to 10<sup>11</sup> s<sup>-1</sup>, in agreement with those previously reported in Mg(BH<sub>4</sub>)<sub>2</sub>.<sup>47</sup> Further lowering of the temperature, from 200 K to 160 K, leads to a sudden 75 % drop in the BH<sub>4</sub><sup>-</sup> jump frequencies, compared to the trending values above 200 K. This decrease is explained by the phase transition of Mg(BH<sub>4</sub>)<sub>2</sub>·3TDF from the monoclinic to the triclinic structure with a smaller unit cell size, presumably leading to enhanced steric effects and thus smaller jump frequencies. This is further portrayed by the calculated activation energies for reorientations,  $E_a$ , (derived from the slopes ( $-E_a/k_B$ ) of the linear fits to the ln( $\tau_{i1}$ <sup>-1</sup>) vs. T<sup>-1</sup>) where a higher barrier is observed in the triclinic structure, i.e.,  $E_a = 137(3)$  meV for  $T \le 200$  K, compared to the monoclinic structure with  $E_a = 113(3)$  meV for  $T \ge 200$  K.

The nature of the reorientational mechanism of the BH<sub>4</sub><sup>-</sup> anions in Mg(BH<sub>4</sub>)<sub>2</sub>·3TDF and Mg(BH<sub>4</sub>)<sub>2</sub>·3THF was determined by the calculated Elastic Incoherent Structure Factor (EISF) over the large *Q*-range accessible with DCS and using 4.5 Å incident wavelength neutrons. Figure 7 shows the experimentally derived EISF data at 315 K due solely to the BH<sub>4</sub><sup>-</sup> anions in Mg(BH<sub>4</sub>)<sub>2</sub>·3TDF. Based on the invariant Lorentzian widths with *Q*, two possible localized hydrogen motions are expected in this system, rotations around any one of the three two-fold (*C*<sub>2</sub>) or four three-fold (*C*<sub>3</sub>) molecular symmetry axes of the BH<sub>4</sub><sup>-</sup> units. For BH<sub>4</sub><sup>-</sup> rotations around a *C*<sub>2</sub>-axis, the tetrahedral anions have two equal equilibrium orientations, while for BH<sub>4</sub><sup>-</sup> rotations around a *C*<sub>3</sub>-axis, the tetrahedral anions have three equal equilibrium orientations,

while one hydrogen atom remains immobile. In both cases the final expression for the EISF is identical and given by:<sup>46</sup>

$$\text{EISF}_{C_2/C_3}(\text{BH}_4^-) = \frac{1}{2}(1 + j_0(Qd)) \tag{1}$$

where  $j_0(x)$  is a zeroth-order Bessel function equal to  $\sin(x)/(x)$ , and  $d \approx 2.0$  Å is the jump distance between two BH<sub>4</sub> hydrogen atom positions. Excellent agreement is seen between the plotted  $C_2/C_3$  model and the experimental data at 315 K. (It should be noted that the minor deviations seen at lower Q-values is a signature of minor multiple-scattering effects due to a sample that is somewhat thicker than desired.) As the EISF by itself cannot distinguish between a  $C_2$ - or  $C_3$ -type jump mechanism for the BH<sub>4</sub><sup>-</sup> anions, it is also necessary to consider the jump frequencies and the activation energies corresponding to this process. In principle, different jump frequencies and activation energies are expected for different types of reorientations of the same  $BH_4^-$  anion. The coordination of the  $BH_4^-$  anions in Mg( $BH_4$ )<sub>2</sub>·3TDF with respect to the Mg<sup>2+</sup> cations is similar to that in unsolvated Mg(BH<sub>4</sub>)<sub>2</sub> compounds, namely, the BH<sub>4</sub><sup>-</sup> anions are oriented toward the Mg<sup>2+</sup> cations roughly along a molecular two-fold symmetry axis so that the Mg<sup>2+</sup> cations are straddled by two of the four BH4- hydrogen atoms. Although the BH4-Mg2+ bond distances are also similar among the solvated and unsolvated compounds, the structures differ somewhat in that each BH<sub>4</sub><sup>-</sup> anion in unsolvated Mg(BH<sub>4</sub>)<sub>2</sub> is in nearly linear coordination with two near-neighbor Mg<sup>2+</sup> cations, whereas each BH<sub>4</sub><sup>-</sup> anion in Mg(BH<sub>4</sub>)<sub>2</sub>·3TDF coordinates to only one Mg<sup>2+</sup> cation (see Figure 7). This coordinative arrangement suggests that the rotation around the particular two-fold symmetry axis that is roughly parallel with that connecting B and Mg atoms (horizontal  $C_2$ -axis in Figure 7) does not break any Mg-H bonds. Thus, this type of motion is expected to have a lower activation energy than the rotation around the two other two-fold symmetry axes of the BH<sub>4</sub><sup>-</sup> anion, which breaks all Mg-H bonds, or, the rotation around any of the three-fold symmetry axes (one of which is shown as  $C_3$  in Figure 7), which breaks half of the Mg-H bonds. The activation energy and the jump frequencies measured for Mg(BH<sub>4</sub>)<sub>2</sub>·3TDF and Mg(BH<sub>4</sub>)<sub>2</sub> in this case compare well with the values obtained by the recent NMR measurements for Mg(BH<sub>4</sub>)<sub>2</sub> ( $E_a = 116(6)$  meV and  $\tau_1^{-1} = 1.5 \times 10^8$  s<sup>-1</sup> at 170 K),<sup>48</sup> which correspond to the rotation around the two-fold axis connecting B and Mg atoms. Therefore, the most likely mechanism in the present case for the BH<sub>4</sub>- anions in Mg(BH<sub>4</sub>)<sub>2</sub>·3TDF involves two-fold jumps around the symmetry axis nearly parallel to that connecting B and Mg atoms ( $C_2$  model). Additionally, based on the EISF dependence with temperature for  $Mg(BH_d)_2$ ·3TDF, no differences in the reorientational mechanisms are observed upon the phase transition from monoclinic to triclinic structure.



**Figure 7.** The EISF for the BH<sub>4</sub><sup>-</sup> anions in Mg(BH<sub>4</sub>)<sub>2</sub>·3TDF determined from 315 K DCS data (using 4.5 Å and 10 Å incident wavelength neutrons) compared with the model curve corresponding to uniaxial reorientations involving either two-fold or three-fold jumps of BH<sub>4</sub><sup>-</sup> anions ( $C_2/C_3$ (BH<sub>4</sub><sup>-</sup>) model).

Another type of dynamics could be identified from Figure 6, observed from the HFBS measurements of  $Mg(BH_4)_2$ ·3THF. Order-of-magnitude-slower jump correlation frequencies, in the range from 10<sup>9</sup> s<sup>-1</sup> to 10<sup>10</sup> s<sup>-1</sup>, when compared to BH<sub>4</sub><sup>-</sup> anions at the same temperature, could be attributed to the localized hydrogen motion associated with the THF rings. This is further confirmed by investigating possible reorientational mechanisms of the THF molecules in Mg(BH<sub>4</sub>)<sub>2</sub>·3THF and comparing with the experimentally obtained EISF.

Due to the  $C_2$  symmetry of the THF ring and the *Q*-independent Lorentzian widths obtained from the QENS fitting of Mg(BH<sub>4</sub>)<sub>2</sub>·3THF measured on the HFBS instrument, the only possible localized motion of H atoms involves two-fold jumps around the anchored  $C_2$ -axis which passes through the O atom and the middle of the opposite C-C bond within the THF ring [ $C_2$ (THF) model]. Such THF reorientations involve much larger moments of inertia than for BH<sub>4</sub><sup>-</sup> reorientations, since the former also require the jump displacements of the C atoms. In this case, the EISF is given as:

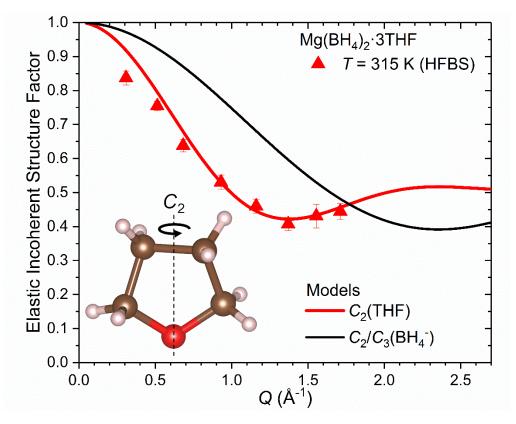
$$\text{EISF}_{C_2}(\text{THF}) = \frac{1}{2} + \frac{1}{8} \cdot \left( j_0(Q \cdot d_1) + j_0(Q \cdot d_2) + j_0(Q \cdot d_3) + j_0(Q \cdot d_4) \right)$$
(2)

where  $d_1 = 2.89$  Å,  $d_2 = 2.57$  Å,  $d_3 = 3.70$  Å and  $d_4 = 3.89$  Å are distances between the exchanging H atoms, as shown in Figure S3 in the Supporting Information.

Considering that the  $BH_4^-$  anions have an active dynamical behavior in the measured temperature range, the EISF associated with the  $BH_4^-$  anions should also be included in the calculation of the total EISF for  $Mg(BH_4)_2 \cdot 3THF$ . The total EISF is calculated by including the weighting factors based on the number of H atoms in front of the EISF of each chemical group. Considering that there are a total of 32 H atoms, with 24 present in the three THF rings, and 8 present in the two  $BH_4^-$  anions, the total EISF is given as:

$$\operatorname{EISF}(\operatorname{total}) = \frac{24}{32} \cdot \operatorname{EISF}_{C_2}(\operatorname{THF}) + \frac{8}{32} \cdot \operatorname{EISF}_{C_2/C_3}(\operatorname{BH}_4^-) \quad . \tag{3}$$

In the HFBS data measured, the elastic peak contribution is actually the sum of the elastic peak intensities due to both THF and  $BH_4^-$  reorientations, whereas the fitted quasielastic contribution is due only to the slower THF reorientations. The much faster  $BH_4^-$  reorientations yield quasielastic scattering that is an order of magnitude too broad to be observed by HFBS and resides as broad background scattering buried under the flat baseline. In order to extract the EISF solely due to the THF molecules, it is necessary to determine the amount of THF elastic scattering intensity by calculating and subtracting the  $BH_4^-$  elastic scattering intensity from the total elastic scattering measured. This can be done with the aid of Eq. 3 and the fact that the EISF behavior for the  $BH_4^-$  anions is known. The mathematical details are outlined in the Supporting Information. The resulting experimental EISF for THF reorientations in  $Mg(BH_4)_2$ .<sup>3</sup> THF at 315 K is shown in Figure 8 and is in good agreement with the theoretically predicted EISF from eq. (2) for two-fold THF jumps. The very different *Q*-dependent behavior of the comparative EISF plot for  $BH_4^$ reorientations in Figure 8 provides further corroboration that we are indeed observing two-fold THF reorientations in the HFBS data.



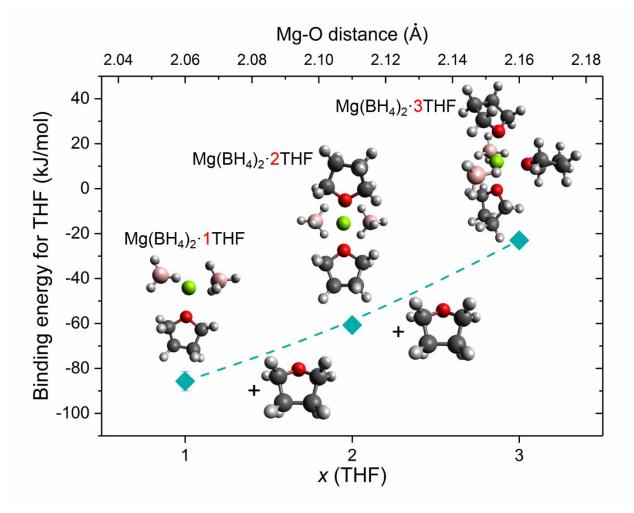
**Figure 8.** The derived EISF for the THF molecules in Mg(BH<sub>4</sub>)<sub>2</sub>·3THF determined from 315 K HFBS data matched with the model curve (blue) corresponding to uniaxial two-fold jumps [ $C_2$ (THF)]. The EISF (black) corresponding to uniaxial two- or three-fold jumps of BH<sub>4</sub><sup>-</sup> anions [ $C_2/C_3$ (BH<sub>4</sub><sup>-</sup>)] is shown for comparison.

## Effect of the THF concentration on the Mg(BH<sub>4</sub>)<sub>2</sub>·xTHF properties

The structural results show that the THF is bound to the  $Mg^{2+}$  cation and there is little direct interaction between THF and the  $BH_4^-$  anion at the investigated temperatures. With three THF molecules bound to each  $Mg^{2+}$  cation, it is tempting to predict that the interactions between the  $Mg^{2+}$  cation and the  $BH_4^$ anion will be weaker in THF-containing compounds. It is therefore surprising that no measurable change was observed in the rotational dynamics of the  $BH_4^-$  anion between the unsolvated  $\alpha$ -Mg( $BH_4$ )<sub>2</sub> and the solvated Mg( $BH_4$ )<sub>2</sub>·3THF adduct.

One possible explanation for this is the asymmetry in the Mg<sup>2+</sup>-BH<sub>4</sub><sup>-</sup> alignment that arises in the THF adducts. In unsolvated  $\alpha$ -Mg(BH<sub>4</sub>)<sub>2</sub>, the BH<sub>4</sub><sup>-</sup> anions sit approximately equidistant between two Mg<sup>2+</sup> cations, with distances of around 2.41 Å and are oriented such that one of their  $C_2$  anion symmetry axes is effectively aligned with the Mg-B-Mg bond direction. In Mg(BH<sub>4</sub>)<sub>2</sub>·3THF, the Mg-B distance is slightly larger, 2.46 - 2.49 Å, but each BH<sub>4</sub><sup>-</sup> anion only interacts with one Mg<sup>2+</sup> cation. The next-nearest Mg<sup>2+</sup> is more than 6 Å away. One would expect that this decreased Mg<sup>2+</sup>-BH<sub>4</sub> interaction, arising from the competing Mg<sup>+</sup>-THF coordination and evidenced by a larger Mg-B distance would lead to an enhanced BH<sub>4</sub><sup>-</sup> orientational mobility compared to that in MgBH<sub>4</sub>. Yet, in the Mg(BH<sub>4</sub>)<sub>2</sub>·3THF crystal structure, the Mg-B bond direction is not as well aligned with the corresponding  $BH_4^- C_2$  symmetry axis (see Figure 7) as in the unsolvated compound. Because of this misalignment, one can expect more steric interference with the adjacent Mg<sup>2+</sup> cation while performing the BH<sub>4</sub><sup>-</sup> two-fold reorientation. The counterpositioning of these two opposing effects on BH<sub>4</sub><sup>-</sup> mobility may indeed result in similar BH<sub>4</sub><sup>-</sup> reorientational dynamics for Mg(BH<sub>4</sub>)<sub>2</sub> and Mg(BH<sub>4</sub>)<sub>2</sub>·3THF as observed. This is also corroborated by the fact that the BH<sub>4</sub>librational energy around this easy  $C_2$ -axis in Mg(BH<sub>4</sub>)<sub>2</sub>·3THF is located between 25 meV and 28 meV by the DFT PDOS calculations and NVS experiments, which is very similar to the calculated and observed BH<sub>4</sub><sup>-</sup> librational energy around the Mg-B-Mg axes in unsolvated Mg(BH<sub>4</sub>)<sub>2</sub> phases.<sup>42</sup> Hence, it is not unreasonable to expect the corresponding rotational potential wells and dynamics to also be similar.

In addition to the observed BH<sub>4</sub>- mobility, one can speculate that a relatively weak Mg<sup>2+</sup>-THF interaction in  $Mg(BH_4)_2$ ·3THF is one key factor for enabling the THF rings to be orientationally mobile, undergoing 180° reorientational jumps around their  $C_2$  molecular symmetry axis. In order to explore this proposition further, molecular DFT calculations and additional OENS and NVS experiments were conducted for samples with a lower amount of THF present in the adduct. DFT calculations for Mg(BH<sub>4</sub>)<sub>2</sub>·THF, Mg(BH<sub>4</sub>)<sub>2</sub>·2THF and Mg(BH<sub>4</sub>)<sub>2</sub>·3THF isolated clusters showed a slight lengthening of the Mg-O distance with each additional THF molecule that coordinates to the Mg<sup>2+</sup> cation. The calculated Mg-O distance for Mg(BH<sub>4</sub>)<sub>2</sub>·THF adduct is 2.06 Å, whereas addition of a second THF results in a symmetric structure with two Mg-O distances of 2.11 Å. Adding the third THF molecule results in further distancing of the THF molecules from Mg<sup>2+</sup> and bond distances of 2.15-2.17 Å. Figure 9 presents the zero-point corrected electronic energies calculated based on the gas-phase reference models for all compounds, which correspond more closely to low-temperature conditions. The calculated energies show that the first THF molecule binding most strongly with -85.7 kJ/mol, the second one binding with -60.7 kJ/mol, and the third one with -23.0 kJ/mol (Table S2 in the Supporting Information). Absolute energies and geometries of the structures are provided in Tables S3 and S4 in the Supporting Information. We expect that the simplicity of the molecular model and its treatment in the gas phase introduce entropic inaccuracies, and as such, result in positive binding free energies. However, because the DFT treatment is consistent for each addition reaction, we can reliably estimate the relative weakening in the binding energy for each subsequent THF molecule. This observation suggests that the reorientational dynamics of the THF rings may be more hindered in a material with a lower amount of solvent present. Indeed, comparative QENS measurements on the samples with the substoichiometric compositional ratio  $2Mg(BH_4)_2$ ·THF and  $2Mg(BH_4)_2$ ·TDF have shown no observable reorientational dynamics of the THF rings within the resolution window of the spectrometer (Figure S4 in the Supporting Information), which is in line with a stronger binding energy of the THF rings to the Mg<sup>2+</sup> cations. Additionally, no noticeable changes in the H dynamics of the BH<sub>4</sub><sup>-</sup> anions is observed in this case either (Figure S5 in the Supporting Information), leading to the conclusion that the THF solvent does not seem to appreciably affect the BH<sub>4</sub><sup>-</sup> dynamical behavior in Mg(BH<sub>4</sub>)<sub>2</sub>·xTHF adducts. (Comparative NVS spectra of these substoichiometric compounds are provided in Figure S6 in the Supporting Information.)



**Figure 9.** THF binding energies for  $Mg(BH_4)_2$ ·THF,  $Mg(BH_4)_2$ ·2THF and  $Mg(BH_4)_2$ ·3THF obtained from the molecular DFT calculations. Red, green, pink, dark gray, and light gray spheres denote the respective O, Mg, B, C and H atoms.

The presence of a much stronger interaction between the THF and the  $Mg^{2+}$  cation for lower THF/Mg(BH<sub>4</sub>)<sub>2</sub> stoichiometric ratios implies that using a more THF-deficient Mg(BH<sub>4</sub>)<sub>2</sub>·*x*THF adduct will reduce the amount of THF that might be released from the adduct during the concomitant thermal release of hydrogen. Indeed, thermal desorption measurements showed that such adducts with lower THF/Mg(BH<sub>4</sub>)<sub>2</sub> ratios required higher temperatures to liberate THF. Hence, with respect to potential

hydrogen-storage applications, the beneficial properties of the THF solvent can be retained over a greater temperature range while reducing the concentration of an unwanted impurity in the released hydrogen.

#### Conclusions

Various neutron scattering experiments, including NPD, NVS, FWS, and QENS, as well as molecular DFT calculations were employed to gain deeper insights into the structural, vibrational, and dynamical properties of  $Mg(BH_4)_2$ ·3THF. Structural analysis has confirmed a lowering of the symmetry upon cooling, from monoclinic C2/c to P-1 via a triclinic distortion, with the phase transition occurring around 200 K. Vibrational properties are strongly influenced by the THF environment. In particular, splitting in spectral features is observed and explained by the lowering of the overall symmetry. Interestingly, for the monoclinic structure, the orientational mobilities of the BH<sub>4</sub>- anions obtained from QENS are not particularly sensitive to the presence of THF and compare well with the mobilities of  $BH_{4}$ - anions in unsolvated Mg(BH<sub>4</sub>)<sub>2</sub>. In contrast, a decrease in the BH<sub>4</sub><sup>-</sup> jump frequencies is observed in the triclinic low-temperature structure and correlated with the enhanced steric effects presumably due to the smaller unit cell volume. Uniaxial jump reorientations of the  $BH_4^-$  anions around a two-fold ( $C_2$ , 180° jumps) axis is observed over the whole temperature range. The THF rings are also shown to be dynamically active, with orientational mobilities about an order of magnitude slower than for the  $BH_4^-$  anions, and with a reorientational mechanism also corresponding to uniaxial two-fold 180° jumps around the THF  $C_2$ symmetry axis. In contrast, no noticeable reorientations of the THF rings are observed in the case of a more THF-deficient 2Mg(BH<sub>4</sub>)<sub>2</sub>. THF adduct. This may be, at least in part, a consequence of a relatively stronger Mg<sup>2+</sup>-THF bonding interaction, as the Mg<sup>2+</sup> cation likely coordinates with only one THF molecule in this adduct. This is unlike THF-rich Mg(BH<sub>4</sub>)<sub>2</sub>·3THF, where each Mg<sup>2+</sup> cation coordinates with three THF molecules. Indeed, DFT calculations show a decrease in the binding energy with each additional THF coordinated to the Mg<sup>2+</sup> cation.

Although the physicochemical behavior of this THF-solvated  $Mg(BH_4)_2$  adduct determined from this combined experimental and computational study is of general fundamental interest, the observed and calculated interplay between  $Mg^{2+}$  cations,  $BH_4^-$  anions, and THF adduct molecules also has likely technological relevance toward understanding better the role of THF for improving hydrogen desorption/absorption cycling properties of  $Mg(BH_4)_2$ . The dynamical results for different THF/Mg( $BH_4$ )<sub>2</sub> stoichiometric ratios suggest little direct interaction, in general, between the THF molecules and the  $BH_4^-$  anions, whereas the THF-Mg<sup>2+</sup> interaction is strongly ratio-dependent, with stronger interactions occurring for the lower THF/Mg( $BH_4$ )<sub>2</sub> ratios. Hence, we would propose that using lower THF/Mg( $BH_4$ )<sub>2</sub> ratios is beneficial in (i) preventing weakly bound THF from coming free from the Mg<sup>2+</sup> cation and (ii) disrupting the stability of the crystalline phase leading to a lower melting point and enhanced kinetics.

#### **Conflicts of interest**

The authors declare no competing financial interest.

#### Authors' contributions

M.D., T.A. and T.J.U. conceived and planned the experiments. M.D. and T.J.U. carried out the experiments. W.Z., I.N. and B.G. planned and carried out the DFT calculations. M.C. contributed to sample preparation. M.D., M.B., H.W., T.A., B.G., T.G., C.J. and T.J.U contributed to the interpretation

of the results. M.D. took the lead in writing the manuscript. All authors provided critical feedback and helped shape the research, analysis and manuscript.

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