Soft Matter

Nondestructive Characterization of Soft Materials and Biofilms by Measurement of Guided Elastic Wave Propagation Using Optical Coherence Elastography

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1   **Nondestructive Characterization of Soft Materials and Biofilms by**

2   **Measurement of Guided Elastic Wave Propagation Using Optical Coherence**

3   **Elastography**

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21 **ABSTRACT**

22

23 Biofilms are soft multicomponent biological materials composed of microbial communities attached to

24 surfaces. Despite the crucial relevance of biofilms to diverse industrial, medical, and environmental

25 applications, biofilm mechanical properties are understudied. Moreover, most available techniques for

26 characterization of biofilm mechanical properties are destructive. Here, we detail a model-based approach

27 developed to characterize the viscoelastic properties of soft materials and bacterial biofilms based on

28 experimental data obtained with the nondestructive dynamic optical coherence elastography (OCE)

29 technique. The model predicted the frequency- and geometry-dependent propagation velocities of elastic

30 waves in a soft viscoelastic plate supported by a rigid substratum. Our numerical calculations suggest that

31 the dispersion curves of elastic waves recorded in thin soft plates by the dynamic OCE technique was

32 dominated by guided waves, whose phase velocities strongly depended on the viscoelastic properties and

33 the plate thicknesses. The numerical model was validated against experimental measurements in agarose

34 phantom samples with different thicknesses and concentrations. The model was then used to interpret

35 guided wave dispersion curves obtained by OCE technique in bacterial biofilms developed in a rotating

36 annular reactor, which allowed for a quantitative characterization of biofilm shear modulus and viscosity.

37 This study is the first to employ measurements of elastic wave propagation to characterize biofilms, and

38 provides a novel framework combining a theoretical model and experimental approach for studying the

39 relationship between biofilm internal physical structure and mechanical properties.

40

41 **Keywords:** nondestructive optoacoustic imaging, optical coherence elastography, viscoelastic properties,

42 guided elastic wave propagation, biofilms

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1. Introduction

Biofilms are multicomponent biological materials composed of communities of microorganisms attached to a surface and encased in a self-produced matrix of extracellular polymeric substances (EPS)[1, 2]. Biofilms are the dominant mode of microbial life in aquatic systems, soil and sediment, and play a critical role in biogeochemical cycling, food webs, and symbioses[3-6]. The mechanical properties of biofilms have recently attracted substantial attention from researchers but remain understudied[7-9]. It has been shown that biofilms display properties of both elastic solids and viscous liquids in response to stress, and can thus be viewed as viscoelastic biomaterials analogous to soft biological tissues[8, 10-12]. Biofilm mechanical properties depend on morphology and composition, and are thought to influence important processes like detachment, attachment, and mass transfer characteristics that are crucial to biofilm functions[13, 14]. Recent macro-scale quasi-static experiments on biofilm mechanical properties suggest that (1) biofilm mechanical properties are heterogeneous[8, 15]; (2) viscoelastic behavior occurs at small deformations[16, 17]; (3) viscoplastic behavior occurs at large deformations when the internal stresses are relieved[10, 15, 18, 19]; and (4) biofilms are stiffer near the attachment surface and more flexible in their canopy[20]. The latter suggests biofilm mechanical properties can vary with distance from the attachment surface.

Most work to date on biofilm mechanical properties has employed macro- and micro-rheological techniques to measure mechanical properties. These techniques have some limitations: macro-rheological techniques only measure bulk average of the properties and do not reveal spatial variability and complexity of living biofilms[21-23], whereas micro-rheological techniques yield only highly localized measurements, often on biofilms *ex situ*, and are not capable of spatial mapping at the mm to cm scale[24-28]. Additionally, these techniques require sample disruption or structural changes while testing. Recently, elastography techniques have enabled biomechanical characterization of soft structures, particularly in the biomedical engineering community, by combining diagnostic imaging tools with specimen deformation approaches. In these techniques, the spatial deformation of a biological specimen is mapped under an applied external force, allowing for identification of mechanical contrast regions and stiff tissues associated with different

71 disease states. Among existing elastography techniques including magnetic resonance elastography

72 (MRE)[29], ultrasound elastography (USE)[30, 31], and optical coherence elastography (OCE)[32-34], the OCE

73 technique provides superior characteristics such as (1) micron-scale spatial resolution, (2) high sample

74 displacement sensitivity on the nanometer scale, and (3) high temporal resolution and fast image acquisition

75 time. These features facilitate the detection of small sample deformations and provide the potential to track

76 dynamic mechanical deformations in real time. OCE techniques can be classified as static or dynamic

77 methods, depending on the time scale of the specimen deformation. Static OCE methods have been widely

78 applied in biomechanical characterization experiments in the biomedical research community, where the

79 heterogeneous strain map of tissue specimens produced in response to a uniform stress field is luied to

80 predict the local Young's modulus. This modulus is based on the ratio of the stress and strain, as obtained

81 in a linear elastic solid. Measurements of viscoelastic properties, on the other hand, rely on tracking

82 temporal dynamics of the specimen deformation under the applied stress field. This is achieved using

83 dynamic OCE methods, in which creep relaxation dynamics[35], elastic stress wave propagation[36-41], or

84 underdamped acoustic vibrations[42, 43] are recorded using various motion tracking methods. These methods

85 can be classified into speckle tracking methods and phase sensitive optical coherence tomography (OCT).

86 The latter is of great interest because it provides a larger measurement dynamic range and inexpensive

87 options for data acquisition[38].

88   This paper provides the first-of-its-kind OCE characterization of viscoelastic properties in bacterial

89 biofilms based on elastic stress wave propagation measurements. Elastic wave based dynamic OCE

90 methods have been explored exclusively in biomedical applications for characterization of viscoelastic

91 properties in soft tissues[36, 37, 44]; however, biofilms have more complex geometrical and compositional

92 features. These features, including heterogeneous composition, surface roughness, non-uniform porosity

93 distribution, and bacterial hierarchical stratification[11, 19, 45-47], make modeling of elastic waves in these

94 materials challenging and invaluable for interpreting the experimental data. In this paper, we report a

95 layered theoretical model that predicted the velocities of guided elastic waves at different frequencies in a

96 soft viscoelastic plate with various thicknesses, shear moduli, and complex shear viscosities. The layered

97   model simulated a soft plate in contact with a semi-infinite water/vacuum medium and a rigid substratum,

98   which approximates the sample geometries tested by the OCE technique. The theoretical model was

99   validated against experimental measurements in agarose gel phantoms of different thicknesses and

100   concentrations. Then, the model was applied to estimate the viscoelastic properties of a mixed-culture

101   bacterial biofilm from OCE measurements of the dispersion curves at frequencies up to 1 kHz. This work

102   provides a promising novel experimental framework for nondestructive quantification of biofilm

103   viscoelastic properties based on elastic wave propagation measured by OCE technique. Furthermore, the

104   potential to obtain co-registered 2D and 3D images of biofilm morphology and viscoelastic properties using

105   the OCE technique can facilitate our understanding of the roles of composition, internal structure, and

106   mechanical properties on the functional performance of bacterial biofilms in a range of applications of high

107   societal relevance. Potential applications of the technique include characterizing fundamental biofilm

108   properties in order to develop strategies to mitigate detrimental biofilms (biofouling) that lead to billions of

109   dollars of cost per year in diverse water/wastewater, food, beverage, petrochemical, industrial equipment

110   and piping, and medical settings[48, 49], and conversely to manage (retain) beneficial biofilms that are

111   increasingly used to clean water and remediate groundwater and soil[50]. In addition, the application of OCE

112   offers the opportunity to enhance understanding of material properties of biofilms in biofilm-linked

113   infections that affect 17 million Americans annually, cause at least 550,000 deaths, and place an enormous

114   economic burden on the US health care system[49].

115

116   **2. Experimental section**

117

118   **2.1 Sample preparation**

119   Soft agarose (Fisher Bioreagents, BP1423-500, PA, USA) gel phantoms with 1.0% and 2.0% weight-to-

120   volume (w/v) concentrations were prepared by mixing one and two grams of agarose powder, respectively,

121   with a 100 mL solution made from 94 mL of nano-purified water and 6 mL of 5.0% w/v skim milk (Becton,

122   Dickinson and Company, 232100, MD, USA). Milk was used to enhance optical scattering in the agarose

5

123   samples and to improve the OCT image contrast of the sample morphology. A sample preparation protocol

124   (Section S1 of the Supplemental Information) was followed to obtain samples with consistent mechanical

125   properties and boundary flatness. A series of heterotrophic biofilm samples were also developed for method

126   proof-of-concept using a rotating annular reactor (RAR Model 1320, Biosurface Technologies, Bozeman,

127   MT, USA). The RAR was operated in batch mode for 24 hours after inoculation to allow for attachment of

128   biomass to the coupons. After 24 hours, synthetic wastewater was fed into the RAR at a dilution rate of 5

129   d^{-1}. The system was inoculated with 25 mL of activated sludge from a local water reclamation plant

130   (Hanover Park, IL, USA) and operated at 30 rpm and at a room temperature of 20-23°C. The reactor was

131   constantly aerated and fed synthetic wastewater for 30 days in order to develop a thick mixed-culture

132   bacterial biofilm, analogous to environmental biofilms commonly employed for contaminant removal in

133   wastewater treatment biofilm reactors. The biofilm was predominantly composed of aerobic heterotrophs

134   and growth was achieved on rectangular polycarbonate coupons designed with a special angled edge to

135   match the slot inside the reactor and allowing them to stay in place during long duration experiments (width

136   12.7 mm, length 150 mm; Biosurface Technologies, Bozeman, MT, USA). Additional details of biofilm

137   growth and reactor operation and monitoring are available in a recent publication from our research group[51].

138   After 30 days, and reaching a thickness of 2.5 mm, the coupon with intact biofilm was removed from the

139   RAR and placed in the OCE setup to carry out measurements. Further details regarding OCE measurements

140   of both agarose gel phantoms and biofilms are discussed in the next section.

141

142   **2.2 Phase-sensitive optical coherence elastography**

143   A schematic of the phase-sensitive OCE is shown in Fig. 1. The setup was used for local excitation and

144   detection of elastic waves in the agarose gel phantoms and mixed-culture bacterial biofilm samples. In the

145   setup, a paddle actuator, composed of a 10 mm wide razor blade glued to the end of an 18-gauge syringe

146   needle, was used to excite elastic waves. The other end of the needle was attached to a piezoelectric

147   transducer (Thorlabs PZS001) that was driven by a sinusoidal voltage from a radio frequency function

148   generator (Agilent 33120A, CA, USA). The blade was guided towards the sample and made light contact

149　　with the sample surface using a two-axis translation stage. When the piezoelectric transducer was excited,

150　　the blade indented the sample periodically and generated harmonic elastic waves including compressional

151　　waves, shear waves, and surface waves. Compressional and shear waves are bulk waves that travel into the

152　　sample, whereas surface waves travel near the sample boundary. A maximum 10 V of the peak-to-peak

153　　voltage applied to the transducer led to a peak-to-peak axial displacement of 5.8 µm of the needle. The local

154　　sample displacement induced by the elastic waves was then recorded with a phase-sensitive spectral-domain

155　　OCT system (operated with a near-infrared light source: center wavelength 930 nm and bandwidth 100 nm)

156　　that is capable of recording the sample morphology and the local dynamic response. The gray-scale sample

157　　morphology image was obtained by collecting a series of adjacent A-scans, which correspond to the one-

158　　dimensional scattering intensity along the vertical (z) direction through the depth of the sample, and

159　　assembling the A-scans to a two-dimensional B-scan image in the x-z plane. The intensity distribution in

160　　the B-scan image represents the spatial variation of the local refractive index in the sample, which is

161　　correlated with the sample's internal structure. In addition, the OCT acquired the local dynamic response

162　　in the sample by calculating the optical phase difference $\Delta\varphi$ between two adjacent A-scans recorded with

163　　a time delay dt, and relating $\Delta\varphi$ to the vertical component of the local sample displacement $u_z(x,z,t)$ by

164　　the relationship $\Delta\varphi(x,z,t) = 4\pi n(x,z)\Delta u_z(x,z,t)/\lambda_0$, where n is the local refractive index of the sample and

165　　λ_0 is the center wavelength of the OCT light source. The motion of the scanning optics in the OCT system

166　　and the acquisition of the A-scans were synchronized with the sinusoidal driving function of the

167　　piezoelectric transducer using a custom-built microcontroller trigger circuit, so that the local phase

168　　difference $\Delta\varphi(x,z)$ along the x-direction could be recorded with respect to a fixed trigger reference and

169　　assembled together to obtain a 2D B-scan image. This image profiles the spatial distribution of the

170　　displacements induced by the waves. Additional details about the measurement approach of the dynamic

171　　response, especially the effect of the delay time dt on the measured data, are discussed in section S2 of the

172　　Supplemental Information.

173   Fig. 2 shows representative OCT and OCE B-scan images obtained in the 2.0% agarose gel

174   phantom with 10 mm thickness. The sample was supported by a 1 mm thick glass substratum and loaded

175   with a water layer over the top surface. The excitation frequency for this experiment was 1.4 kHz. The OCT

176   and OCE images were acquired over a lateral distance of 9 mm. The sample was tilted by 10 degrees relative

177   to the vertical optical axis of the microscope objective to eliminate strong direct reflection of the probe light

178   from the air-water and the water-agarose interfaces that would create artifacts in the images due to multiple

179   interferences. The bright band in the OCT image (Fig. 2a) is due to a strong contrast of the refractive index

180   at the interface between the air and the water. In addition, the OCT image shows limited contrast in the

181   agarose gel layer, suggesting the sample is homogeneous without apparent structural features such as voids

182   or cracks. On the other hand, the OCE image (Fig. 2b) shows a periodic distribution of the phase difference

183    $\Delta\varphi$ alternating between the maximum π and minimum $-\pi$ radians along the lateral direction which is

184   associated with the periodic displacement of the elastic wave. The phase values were plotted within a

185   smaller span $[-\pi/2, \pi/2]$ to enhance the color contrast. The spatial frequency ($\nu = 1/\lambda$, where λ is the

186   wavelength) of the elastic wave was obtained by implementing spatial fast Fourier transform from the data

187   along the white dotted line, and the phase velocity c of the elastic wave was determined based on the

188   relationship $c = f\lambda$ where f is the excitation frequency. The measurement was repeated at different

189   excitation frequencies to collect the frequency-dependent phase velocity, the dispersion curve, for the

190   excited elastic waves in the sample. The dispersion curve of an agarose gel plate is a function of the plate

191   thickness and material properties, which was used to determine the shear modulus and the shear viscosity

192   through the inverse analysis based on the model presented in the following section.

193

194   **2.3 Theoretical model for elastic wave propagation in a multi-layered structure**

195   The choice of the elastic wave type used in measurements affects the achievable spatial resolution. In soft

196   samples, the wavelength of compressional waves in the kHz range is typically in the range of meters, while

197   the wavelength of shear waves in the same frequency range is three orders of magnitude smaller. As such,

198   shear waves in the kHz range are favored for acoustic mapping of elastic property variations in soft

199 samples[52]. Operating with shear waves at MHz frequencies can lead to spatial resolution in the micron and

200 sub-micron range; however, this is prohibited by attenuation of elastic waves resulting from the viscoelastic

201 behavior of the materials. Furthermore, in thin samples where the elastic wavelength is comparable to the

202 sample thickness, shear waves reflect from the sample boundaries and overlap through the sample thickness

203 to produce standing wave interference patterns and propagate as guided waves in the lateral direction[53-55].

204 These guided waves can propagate as Lamb, Love, or surface acoustic waves, having dispersive phase

205 velocities that depend on frequency, sample geometry, and sample material properties. Therefore, a model

206 capable of predicting the dispersion curves of the guided waves is necessary for the inverse analysis which

207 estimates the viscoelastic properties in the samples from the experimental measurements of wave velocity.

208 In this section, we present a model for guided elastic wave propagation in a multi-layer structure

209 composed of an isotropic, viscoelastic, and homogenous gel plate loaded by a water half-space on the top

210 surface and attached to a stiff half-space at the bottom surface. A schematic diagram of the layered model

211 system is shown in Fig. 3. The stiff (glass) substrate was assumed to be rigid since its Young's and shear

212 moduli are orders of magnitude larger than those of the agarose gel layer. The water layer was assumed to

213 be an ideal liquid which is homogenous, isotropic, inviscid, and does not support shear stresses. The model

214 predicts the dispersion relation for the water loaded viscoelastic layer based on the solution to the

215 elastodynamic wave equation for an isotropic and homogeneous material in the frequency domain, given

216 by[56]

$$(\lambda^* + \mu^*)\nabla(\nabla \cdot \vec{u}) + \mu^*\nabla^2\vec{u} + \rho\omega^2\vec{u} = 0 \tag{1}$$

218 where $\vec{u} = u_x\hat{e}_x + u_y\hat{e}_y + u_z\hat{e}_z$ is the displacement vector which comprises its components u_x, u_y, and u_z

219 along x-, y-, and z- Cartesian axes with unit vectors $\hat{e}_x$, $\hat{e}_y$, and $\hat{e}_z$. ∇ is the differential operator in the three-

220 dimensional space, ω is the angular frequency, ρ is the material density, and λ^* and μ^* are the complex

221 frequency-dependent relaxation functions of the Lamé material properties defined by

$$\lambda^*(\omega) = \lambda + i\eta_\lambda\omega \tag{2}$$

$$\mu^*(\omega) = \mu + i\eta_\mu\omega \tag{3}$$

224   where λ and μ are the asymptotic values of the relaxation functions, and η_λ and η_μ are complex viscosity

225   terms. μ and η_μ represent the shear modulus and shear viscosity[57, 58], respectively, which are of particular

226   interest in this work. It can be shown that the solution to eqn (1) under plane strain conditions (u_y

227    $= 0, \partial/\partial y = 0$) in a plate is a linear superposition of compressional and shear waves, which can be

228   expressed in terms of a scalar potential ϕ and vector potential component ψ given by

229
$$\phi = A_{L+}\, e^{i(k_x x + k_z^L z - \omega t)} + A_{L-}\, e^{i(k_x x - k_z^L z - \omega t)} \tag{4}$$

230
$$\psi = A_{S+}\, e^{i(k_x x + k_z^S z - \omega t)} + A_{S-}\, e^{i(k_x x - k_z^S z - \omega t)} \tag{5}$$

231   where $A_{L\pm}$ and $A_{S\pm}$ are complex amplitude constants with the letters L and S distinguishing partial

232   longitudinal and shear waves and the $\pm$ sign distinguishing downward (positive) and upward (negative)

233   traveling directions of the partial bulk waves as illustrated in Fig. 3. $k_z^{L,S}$ are the complex wave numbers in

234   the vertical (z) direction which satisfy the following relationship:

235
$$\pm\, k_z^L = \pm\, \sqrt{\frac{\omega^2}{a_L^2} - k_x^2} \tag{6}$$

236
$$\pm\, k_z^S = \pm\, \sqrt{\frac{\omega^2}{a_S^2} - k_x^2} \tag{7}$$

237   where a_L and a_S are complex compressional and shear wave speeds, which are related to the material

238   properties by the following relationships:

239
$$a_L^2 = \frac{\lambda^* + 2\mu^*}{\rho} \tag{8}$$

240
$$a_S^2 = \frac{\mu^*}{\rho} \tag{9}$$

241   The real parts of a_L and a_S yield the bulk compressional and shear wave speeds, c_L and c_S, and their

242   imaginary values are the attenuation coefficients of the bulk waves. The local displacement vector

243   associated with the harmonic plane wave potential solutions is obtained using the vector relationship:

244
$$\vec{u} = \nabla\phi + \nabla \times \vec{\psi} \tag{10}$$

245   where $\vec{u} = (u_x, 0, u_z)$ and $\vec{\psi} = (0, \psi, 0)$ under plane strain conditions. The local stress tensor is given by the

246   Lamé constitutive relations

$$\sigma_{zz} = \lambda^* (\varepsilon_{xx} + \varepsilon_{zz}) + 2\mu^* \varepsilon_{zz} \tag{11}$$

$$\sigma_{xz} = 2\mu^* \varepsilon_{xz} \tag{12}$$

249   where $\varepsilon_{xx} = \frac{\partial u_x}{\partial x}$, $\varepsilon_{zz} = \frac{\partial u_z}{\partial z}$, and $\varepsilon_{xz} = \frac{1}{2}\left(\frac{\partial u_x}{\partial z} + \frac{\partial u_z}{\partial x}\right)$ [57, 59-61].

250   From eqn (4) to (12), the components of the local displacement and stress tensor are expressed in

251   the matrix form:

$$\begin{bmatrix} u_x \\ u_z \\ \sigma_{zz} \\ \sigma_{xz} \end{bmatrix} = i\boldsymbol{D} \begin{bmatrix} A_{L+} \\ A_{L-} \\ A_{S+} \\ A_{S-} \end{bmatrix} \exp\left[i\left(k_x x - \omega t\right)\right]$$

253   where

$$\boldsymbol{D} = \begin{bmatrix} k_x e^{ik_z^L z} & k_x e^{-ik_z^L z} & -k_z^S e^{ik_z^S z} & k_z^S e^{-ik_z^S z} \\ k_z^L e^{ik_z^L z} & -k_z^L e^{-ik_z^L z} & k_x e^{ik_z^S z} & k_x e^{-ik_z^S z} \\ i\rho(\omega^2 - 2\alpha_S^2 k_x^2) e^{ik_z^L z} & i\rho(\omega^2 - 2\alpha_S^2 k_x^2) e^{-ik_z^L z} & 2i\rho k_x k_z^S \alpha_S^2 e^{ik_z^S z} & -2i\rho k_x k_z^S \alpha_S^2 e^{-ik_z^S z} \\ 2i\rho k_x k_z^L \alpha_S^2 e^{ik_z^L z} & -2i\rho k_x k_z^L \alpha_S^2 e^{-ik_z^L z} & -i\rho(\omega^2 - 2\alpha_S^2 k_x^2) e^{ik_z^S z} & -i\rho(\omega^2 - 2\alpha_S^2 k_x^2) e^{-ik_z^S z} \end{bmatrix}$$

255   (13)

256   The local displacement vector in the water layer can be expressed in terms of a potential function

257   ϕ^w by the relationship:

$$\overrightarrow{u^w} = \nabla \phi^w, \tag{14}$$

259   which is a special case of eqn (10) where the vector potential ψ, related to the shear partial waves, vanishes.

260   The scalar potential in eqn (14) must satisfy the wave equation

$$\nabla^2 \phi^w + (k^w)^2 \phi^w = 0 \tag{15}$$

11

262 where $(k^w)^2 = k_x^2 + (k_z^w)^2$ is the wavenumber of the compressional wave in the water layer that comprises

263 the components k_x and k_z^w in x- and z-directions and has the relation with the compressional wave speed k^w

264 $= \omega/c^w$ as $c^w =$ 1481 m/s in water. The water layer is treated as a half-space without wave sources. As

265 such, only partial waves travelling in the negative z-direction exist, as illustrated in Fig. 3. In addition, we

266 seek guided elastic wave solutions in the soft plate that travel at the water-plate interface as interface

267 waves[62-64], so the scalar potential can be expressed with the amplitude A_{L-}^w and the wavenumber k_x by

$$\phi^w = \left(A_{L-}^w e^{-ik_z^w z}\right)\exp\left[i\left(k_x x - \omega t\right)\right] \tag{16}$$

269   From eqn (11), (12), (14) and (16), the displacement and the pressure (equivalent to the normal

270 stresses in solids) in the vertical direction in the water layer, u_z^w and p, can be derived

$$u_z^w = -i\left(k_z^w e^{-ik_z^w z}\right)A_{L-}^w \exp\left[i(k_x x - \omega t)\right] \tag{17}$$

$$p = i(i\rho^w \omega^2 e^{-ik_z^w z})A_{L-}^w \exp\left[i(k_x x - \omega t)\right] \tag{18}$$

273   Since the water layer was assumed to be an inviscid liquid that does not support the propagation of

274 shear waves, the shear stress in the water layer is zero, $\sigma_{xz}^w = 0$.

275   Five boundary conditions are needed to solve for the unknown coefficients for the potential

276 functions in the soft plate and water layers. The boundary conditions include zero displacement at the

277 bottom surface of the soft plate, $u_x|_{z=h} = u_z|_{z=h} = 0$, due to the rigid glass substratum, continuity of the

278 vertical displacements between the soft plate and the water layer, $u_z|_{z=0} = u_z^w|_{z=0}$, continuity of the

279 normal traction in the soft plate and the pressure in the water, $\sigma_{zz}|_{z=0} = p|_{z=0}$, and zero shear traction at

280 the interface between the soft plate and the water layer, $\sigma_{xz}|_{z=0} = 0$. Applying these conditions leads to

281 five equations for the potential function coefficients, which are expressed below in the matrix form

282
$$\begin{bmatrix} k_x e^{ik_z^L h} & k_x e^{-ik_z^L h} & -k_z^S e^{ik_z^S h} & k_z^S e^{-ik_z^S h} & 0 \\ k_z^L e^{ik_z^L h} & -k_z^L e^{-ik_z^L h} & k_x e^{ik_z^S h} & k_x e^{-ik_z^S h} & 0 \\ i\rho(\omega^2 - 2\alpha_S^2 k_x^2) & i\rho(\omega^2 - 2\alpha_S^2 k_x^2) & 2i\rho k_x k_z^S \alpha_S^2 & -2i\rho k_x k_z^S \alpha_S^2 & -i\rho^w \omega^2 \\ 2i\rho k_x k_z^L \alpha_S^2 & -2i\rho k_x k_z^L \alpha_S^2 & -i\rho(\omega^2 - 2\alpha_S^2 k_x^2) & -i\rho(\omega^2 - 2\alpha_S^2 k_x^2) & 0 \\ k_z^L & -k_z^L & k_x & k_x & k_z^w \end{bmatrix} \begin{bmatrix} A_{L+} \\ A_{L-} \\ A_{S+} \\ A_{S-} \\ A_{L-}^w \end{bmatrix} = \mathbf{0}$$

283   or

284
$$\boldsymbol{SA} = \mathbf{0} \tag{19}$$

285   To obtain non-trivial solutions for $\boldsymbol{A}$ from eqn (19), the determinant of the coefficient matrix $\boldsymbol{S}$ must

286   equal to zero:

287
$$\det(\boldsymbol{S}) = 0 \tag{20}$$

288   which is the characteristic equation of the layered model. The solutions (ω, k_x) of eqn (20) yield the

289   dispersion relation of the guided elastic waves in the water-loaded viscoelastic layer as the speed of the

290   elastic wave propagating along the x-direction, $c = \omega/k_x$.

291

292   **3. Results and discussion**

293

294   **3.1. Numerical simulation**

295   In this section, we present a set of dispersion curves predicted by the model for the guided elastic waves in

296   the layered structure under different boundary conditions and mechanical properties as shown in Fig. 4.

297   Each dispersion curve represents a unique elastic wave mode associated with a group of the solutions (ω,

298   k_x) of eqn (20). Although there are an infinite number of dispersion curves for the model since the frequency

299   can go to infinity, we only plotted the curves within the frequency range 0-3000 Hz. First, we calculated

300   the dispersion curves for a purely elastic plate with the mechanical properties and geometry such as

301   compressional wave speed $c_L = 1481$ m/s, shear wave speed $c_S = 3$ m/s, zero complex viscosities $\eta_\lambda = \eta_\mu$

302    $= 0$, and layer thickness $h = 10$ mm. The compressional wave speed in the gel was assumed to be the same

303   as found in water because of the low concentration of agar in the gel phantoms. The large ratio of the

304   compressional to the shear wave speeds corresponds to a Poisson's ratio of 0.5.

305   Fig. 4a shows the results from the stress-free and clamped boundary conditions imposed on the top

306   and the bottom surfaces of the agarose layer, in which the water layer in Fig. 3 was replaced by vacuum.

307   Except for the non-dispersive mode with a frequency-independent phase velocity c_S, corresponding to the

308   bulk shear wave propagation, each dispersion curve has a cut-off frequency (f_c) below which the associated

309   elastic wave mode does not propagate. The cut-off frequencies occur at values of $f_c = \frac{c_S}{2h}, \frac{c_S}{h}, \frac{3c_S}{2h}, \frac{2c_S}{h}$ [58], etc.

310   where the phase velocity tends to approach infinity. An implication of the infinite phase velocity at the cut-

311   off frequency is that all points on the free surface of the layer vibrates in phase, leading to a shear-thickness

312   resonance. We note that the phase velocities of all dispersive modes have decreasing trends with frequency

313   and asymptotically approach the bulk shear wave speed c_S, except for the first mode. The asymptotic value

314   of the first mode is the Rayleigh wave velocity of the layer, $c_R = 2.87$ m/s. The ratio $c_R/c_S = 0.96$ agrees

315   with the theoretical prediction for a material that has a Poisson's ratio of 0.5[62, 63]. The penetration depth of

316   the Rayleigh wave is approximately one wavelength, over which the energy of the wave attenuates to $1/e$

317   of its maximum value at the layer surface. As such, the elastic mode reaches the asymptotic value and

318   propagates as a pure surface wave without the interference by the energy reflected from the bottom surface.

319   As an example, consider an agarose gel phantom with thickness 10 mm. When the frequency is larger than

320   500 Hz, the wavelength is 5.7 mm, which is smaller than the thickness of the agarose gel layer.

321   When the agarose gel layer is loaded with water on the top surface, the water loading decreases the

322   Rayleigh wave velocity to the Scholte wave velocity $c_{Sch} = 2.52$ m/s as shown in Fig. 4b. The Scholte

323   wave propagates at the interface between the gel and the water layers whose elastic energy is attenuated

324   along the transverse direction as part of the energy couples into the water layer. The ratio $c_{Sch}/c_S = 0.84$

325   from our numerical calculation agrees well with analytical predictions[62, 63].

326   Fig. 4c shows the dispersion curves predicted by the model for a water-loaded viscoelastic layer of

327   agarose gel. In this calculation, the material properties and the plate geometry remained the same as the

14

328   ones used to obtain the results shown in Fig. 4a and 4b, with the sole exception that the complex shear

329   viscosity was changed to $\eta_\mu = 0.15$ Pa-s. One significant difference between the dispersion curves in Fig.

330   4b and 4c is the increasing bulk shear and Scholte wave velocities in the high-frequency region due to the

331   effect of viscosity, which is of particularly interest in this study. Measurement of the dispersive phase

332   velocity over this region can support characterization of the shear modulus and complex shear viscosity of

333   the agarose gel sample. The dispersion curves also depend on the layer thickness as shown in Fig. 5, where

334   the layer thickness is changed to 1 mm with the same material properties as in Fig. 4c. Comparing Fig. 5

335   with Fig. 4c, the most significant difference is that only three elastic modes are supported in the 1 mm thick

336   layer, which correspond to the lowest interface wave, the bulk shear wave, and the second guided wave

337   modes. The lowest mode in Fig. 5 has a higher cut-off frequency since this frequency is inversely

338   proportional to the layer thickness.

339

340   **3.2. Experimental results**

341   **3.2.1. Agarose gel phantoms**

342   Fig. 6 shows experimental dispersion curves obtained for the 1.0% and 2.0% agarose gel phantoms of two

343   thicknesses, 10 mm and 1 mm. The bandwidth of the measured dispersion curves was limited to 5 kHz due

344   to the low signal-to-noise ratio of the OCE B-scans, which stemmed from attenuation of the excited waves

345   above this frequency. Each dispersion curve in Fig. 6 represents the average of nine measurements of the

346   phase velocity versus frequency obtained at random positions in the sample. The error bars represent the

347   standard deviation of these nine measurements. For all the experimental dispersion curves, the coefficient

348   of variation (COV), defined as the ratio of standard deviation to the average value of the phase velocity, is

349   less than 2.5%, suggesting homogeneity of the sample elastic properties. The dispersion curves for the 1

350   mm thick samples have a relatively constant wave speed within the high frequency range. The wavelength

351   of the excited waves in the 1.0% agarose sample, for example, decreased from 1.29 mm to 0.38 mm between

352   the frequency range of 1.2 to 4 kHz, which suggests that the excited wave changed from a guided wave at

15

353   lower frequencies to a Scholte wave at higher frequencies. On the other hand, for the 10 mm thick samples,

354   the Scholte wave mode was supported over the frequency bandwidth of the measurement since the

355   wavelength was smaller than the sample thickness. As observed in the numerically-calculated dispersion

356   curves, the phase velocity of the wave modes obtained in the 10 mm samples increase markedly with

357   frequency due to the complex shear viscosity of the samples. In addition, the average phase velocity

358   increased with the agarose concentration in the gel, as expected.

359   The dispersion curves in Fig. 6b have a decreasing trend in the low-frequency range, indicating that

360   the measured mode belongs to the lowest elastic mode illustrated in Fig. 4. This is a reasonable observation

361   since the Scholte wave was the predominant propagation mode near the interface[63, 65]. Fig. 7 compares

362   numerical calculations of the dispersion curve for the first mode in the layered model to the experimental

363   data. The shear modulus and complex shear viscosity were used as free fitting parameters in the numerical

364   model for samples of 10 mm thickness, while shear modulus was the only free fitting parameter for samples

365   of 1 mm thickness owing to the limited dispersion observed in these experiments. Good agreement was

366   found between the experimental and numerical results, except for the larger errors observed within the low

367   frequency range in Fig. 7c and 7d, which were due to lack of sufficient periodic cycles of the waves within

368   the OCE field of view to estimate the wavelength through fast Fourier transform. The shortest wavelength

369   below 2 kHz in Fig. 7c and 7d was 1.65 mm; this is equivalent to less than six cycles within the total

370   sampling distance in the OCE B-scan, which limited the accuracy of the wavelength estimation. The best-

371   fit material properties for the samples are listed in Table 1. The shear modulus and viscosity increase, as

372   expected, with the concentration of agarose in the gel. The shear moduli measured from 1 and 10 mm

373   samples were similar at 1.0% and indistinguishable at 2.0%, and the values agree well with those reported

374   in the literature[38, 52]. This finding validates the use of the layered model to determine the mechanical

375   properties of viscoelastic materials.

376

377   **3.2.2. Mixed-culture bacterial biofilm**

16

378   After 30 days of growth, the mixed-culture bacterial biofilm reached a ~2.5 mm thickness with a ~ 250 μm

379   variation over a 4 mm lateral extent due to surface roughness as shown in the OCT B-scan (Fig. 8a) that

380   illustrates the sample morphology. The OCE B-scan obtained at an excitation frequency of 660 Hz and the

381   dispersion curves from the numerical simulation and experimental measurements are shown in Fig. 8b and

382   8c, respectively. In order to more precisely calculate the speed of the elastic wave travelling along the

383   curved surface, a cubic function was fitted to the curved region of the biofilm surface to approximate the

384   propagation path of the elastic waves. The amplitudes at discrete intervals along the path were extracted,

385   and the fast Fourier transform was applied to the data to obtain the spatial frequency (or inverse wavelength)

386   of the elastic waves (See Section S4 in the Supplemental Information). The maximal excitation frequency

387   in the measurement was limited to 1 kHz due to attenuation of elastic waves in the sample. Unlike the

388   agarose gel sample, the OCT image of the biofilm shows internal structural variations due to the presence

389   of pores (dark bands), which also results in the low phase amplitude of the elastic waves in the OCE image.

390   This finding aligns with other observations that pores and structural heterogeneity are common in biofilms[13,

391   66].

392   The bright white bands in the OCT image (Fig. 8a) indicate the boundary of the biofilm with air,

393   and the propagation mode illustrated in the OCE image (Fig. 8b) is a Rayleigh surface wave. The use of the

394   Rayleigh wave measurement is particularly advantageous in this case since the penetration depths of the

395   wave, 1.2 mm at 550 Hz and 0.76 mm at 1 kHz, were less than the sample thickness and thus less sensitive

396   to sample thickness variations. The dispersion curve for the measured data in Fig. 8c shows a 15% increase

397   in phase velocity within the frequency bandwidth of the measurement. The measured dispersion curve was

398   fitted by the numerical model using the biofilm shear modulus and viscosity as free fitting parameters.

399   Similar to Fig. 7c and 7d, large disagreement between the experimental data and the numerical model is

400   observed within the low frequency range (≤ 400 Hz) due to the limited number of wavelengths within the

401   total sampling distance. The best-fit shear modulus and complex shear viscosity are 429 Pa and 0.06 Pa-s.

402   These estimated properties represent the average bulk viscoelastic properties of the biofilm, and are within

403   the broad range of reported values for the viscoelastic properties of biofilms (shear modulus = 10^{-1} - 10^5 Pa

404 and viscosity = 10^{-1} - 10^{10} Pa-s)[18]. The broad range reflects both the diversity of different type of biofilms

405 used in other studies and additional differences resulting from the inconsistencies and disadvantages of the

406 characterization methods used. We highlight that most of the rheometry techniques employed for property

407 characterization are destructive or involve large disturbances in sample geometry when testing, which

408 inevitably causes changes of the sample morphology and properties. On contrary, our novel technique has

409 a nondestructive nature that prevents any structural change and allows for the estimation of viscoelastic

410 properties in the intact forms of the samples, which makes this more advantageous compared to other

411 techniques.

412 We remark that in our inverse modeling analysis, we assumed the bulk modulus of the biofilm to

413 be equal to the bulk modulus of water due to high water content (> 90%) of the sample. Overall, this novel

414 approach provides a nondestructive, direct, local and *in situ* option to interrogate mechanical properties in

415 these type of systems. Further work that refines the layered model is needed to address the heterogeneous

416 spatial distribution of the shear modulus in the sample as suggested in Fig. 8b.

417

418 **4. Conclusion**

419 We developed methods involving a combination of OCE measurements and inverse modeling to

420 characterize the mechanical properties of soft viscoelastic materials and bacterial biofilms. OCE was

421 employed to obtain the dispersion curve—the frequency-dependent phase velocity—of the surface acoustic

422 wave travelling in a biofilm supported by a rigid substrate. This is the first work to present wave propagation

423 in biofilms, discover the frequency-dependent wave velocity, and interpret the dispersive wave velocity by

424 a theoretical model to estimate the mechanical properties. The theoretical model obtained the dispersion

425 curves of guided elastic wave modes by solving the elastodynamic wave equation for a layered viscoelastic

426 plate attached to a rigid substratum and a semi-infinite water/vacuum layer on both sides. Dispersion in

427 these materials depends on the mechanical properties, the geometry of the plate, and the presence or absence

428 of water on the surface of the viscoelastic material. The model was validated by estimating the shear moduli

429   and complex shear viscosities from the OCE measurements of phase velocities in 10 and 1 mm thick agarose

430   gel plates with 1.0% and 2.0% agarose concentrations. The estimation of the agarose gel properties agrees

431   well with the ones in literature. These results suggest that the wave propagation observed in the OCE

432   measurements of agarose gel plates belong to the lowest elastic mode travelling near the top boundary of

433   the plate. The influence of the plate geometry is crucial since the guided waves interact with the bottom

434   boundary when the acoustic wavelength is larger than the plate thickness. We then used this approach to

435   measure the shear modulus and complex shear viscosity in a bacterial biofilm, and obtained reasonable

436   results that are within the range reported in literature. Since there is no "gold standard" measurement for

437   mechanical properties in soft materials and biofilms, our nondestructive technique provides a novel

438   approach for characterizing these properties without affecting the original status of the samples.

439   Furthermore, this framework enhances our understanding of elastic wave propagation in soft viscoelastic

440   materials and provides a first proof-of-concept of OCE application to quantify mechanical properties of

441   biofilms that are critically important in diverse environments and applications. The OCE technique can be

442   further employed to study relationships between biofilm morphology, growth conditions, and elastic

443   properties. Future work should focus on refining the layered model to address variations of the geometry

444   and heterogeneity of material properties in soft materials as well as on utilizing the technique for rapid,

445   nondestructive, spatially-resolved characterization of biofilm mechanical properties across a range of

446   microbial systems and applications.

447

448   **Conflicts of interest**

449   The authors declare no conflicts of interest.

450

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457

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458   **REFERENCES**

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566

Figures

Nondestructive Characterization of Soft Materials and Biofilms by Measurement of Guided Elastic Wave Propagation Using Optical Coherence Elastography

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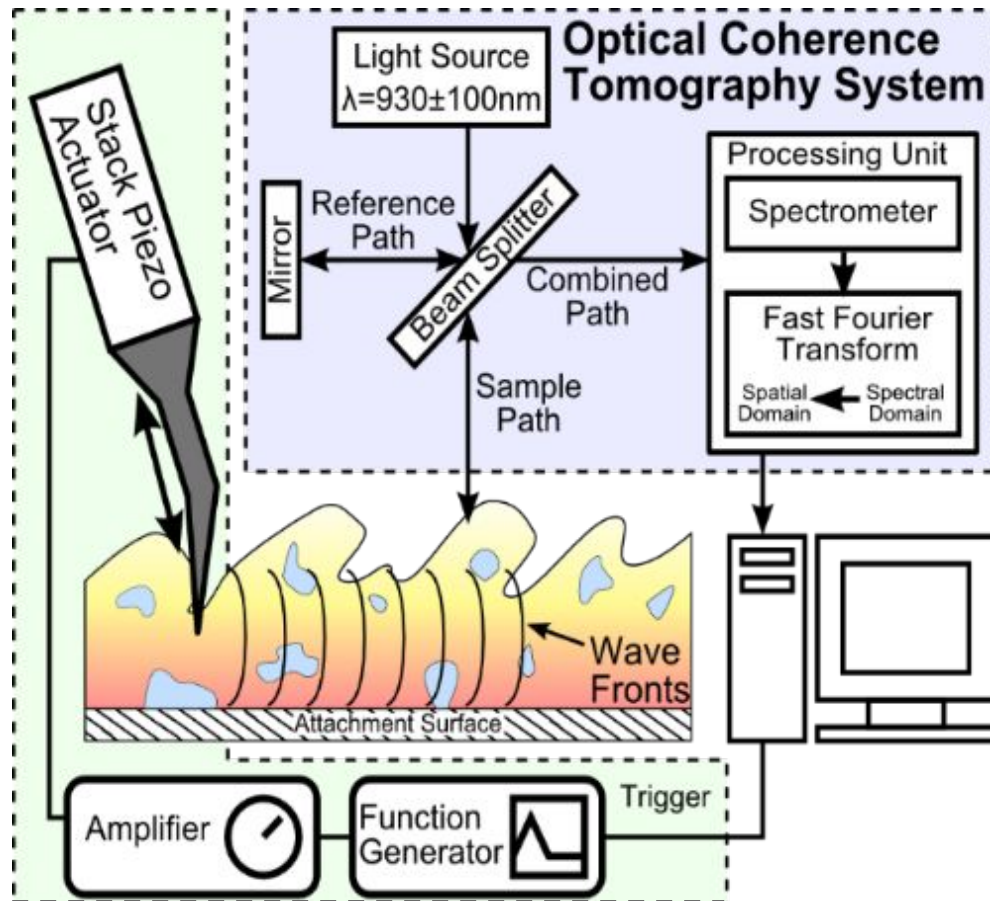
Fig. 1. Phase-sensitive optical coherence elastography setup
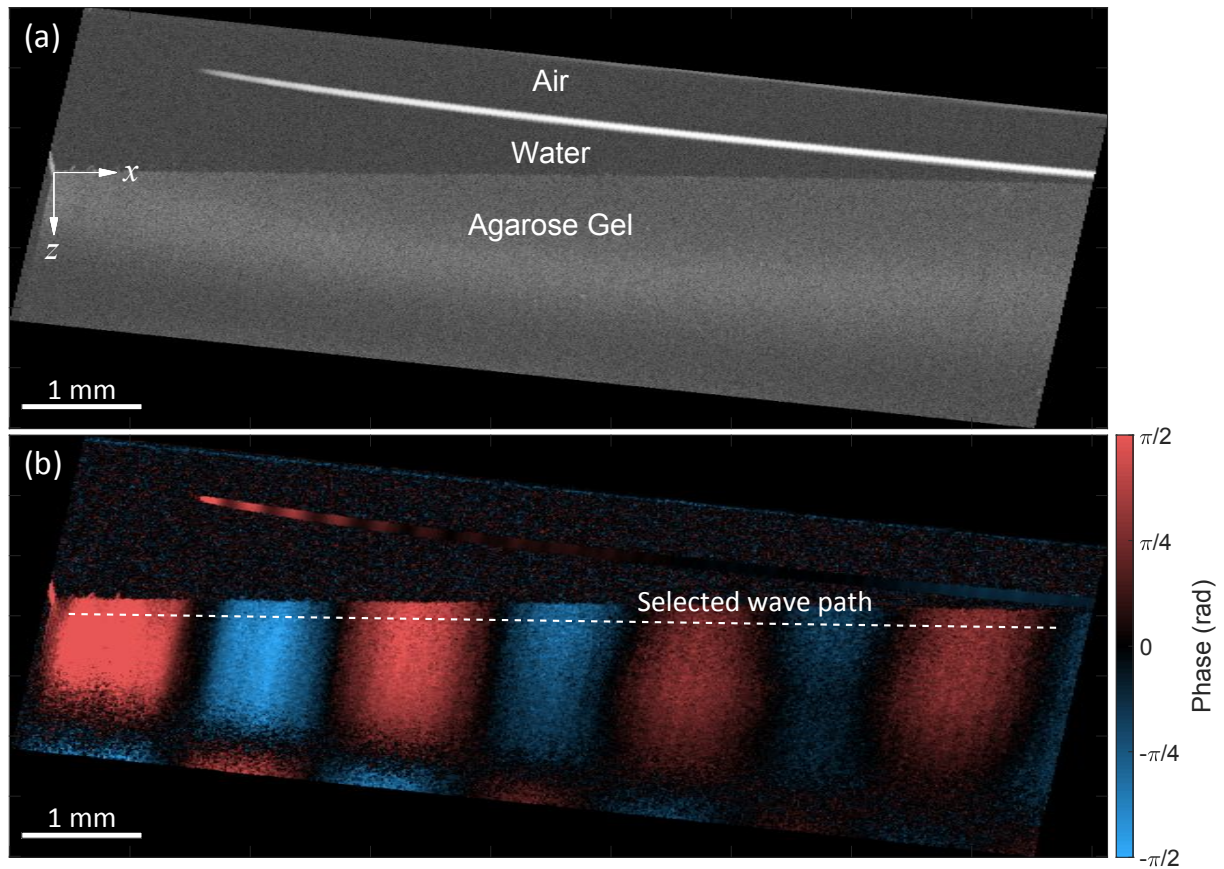
Fig. 2. (a) Optical coherence tomography image of a 2.0% agarose gel phantom. The thickness of the water layer was reduced in this figure for visualization purposes. The experiments were conducted with a water layer of >2 mm thickness. (b) Optical coherence elastography image showing phase distribution of 1.4 kHz elastic waves in the sample. The pixel sizes along the x- and z-directions are 4 and 2 μm.
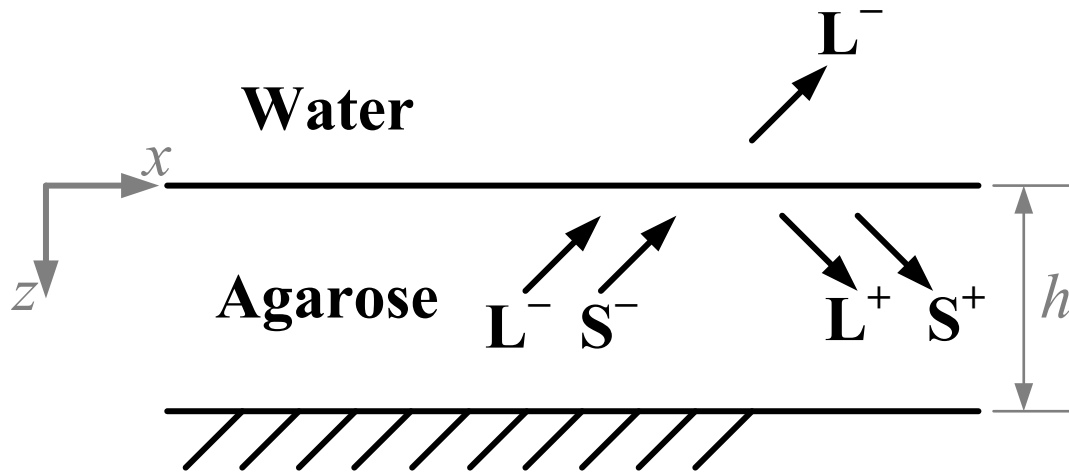
Fig. 3. Geometry of the layered numerical model. Symbols L and S represent compressional and shear waves. Positive and negative superscripts are used to represent forward and backward propagating partial waves.

(a)

First Mode c_R = 2.87 m/s

(b)

First Mode c_{Sch} = 2.52 m/s
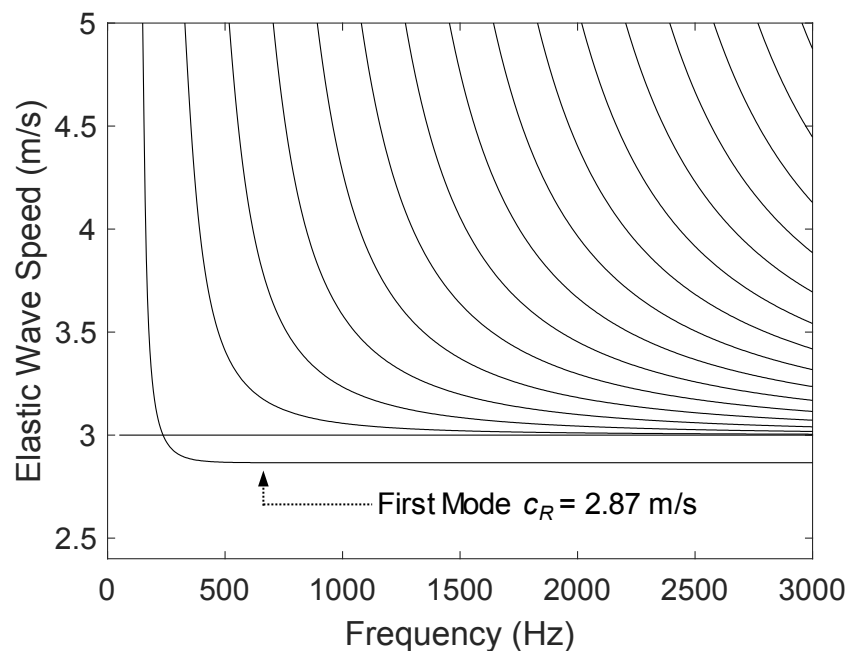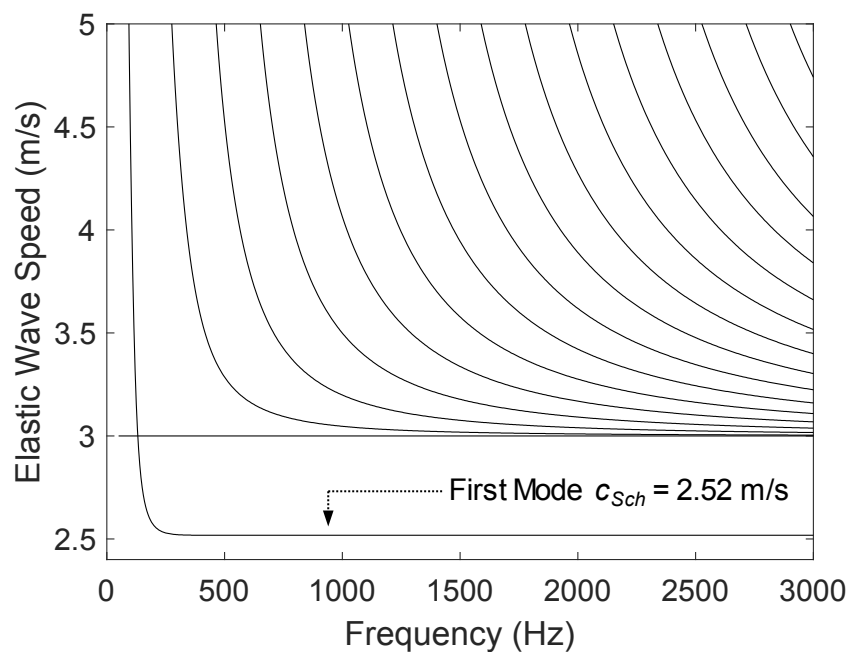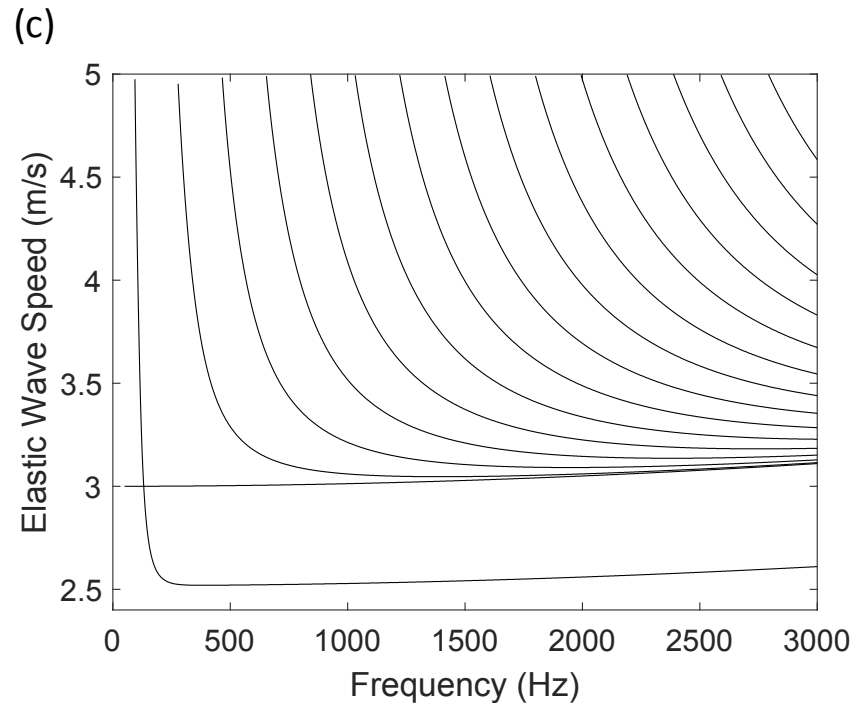
Fig. 4. Dispersion curves for (a) a pure-elastic layer with free-clamped boundary condition, (b) a pure-elastic plate with liquid loading on the top surface and clamped boundary condition at bottom surface, and (c) a viscoelastic layer with liquid loading and clamped boundary conditions. The layer thickness $h = 10$ mm.
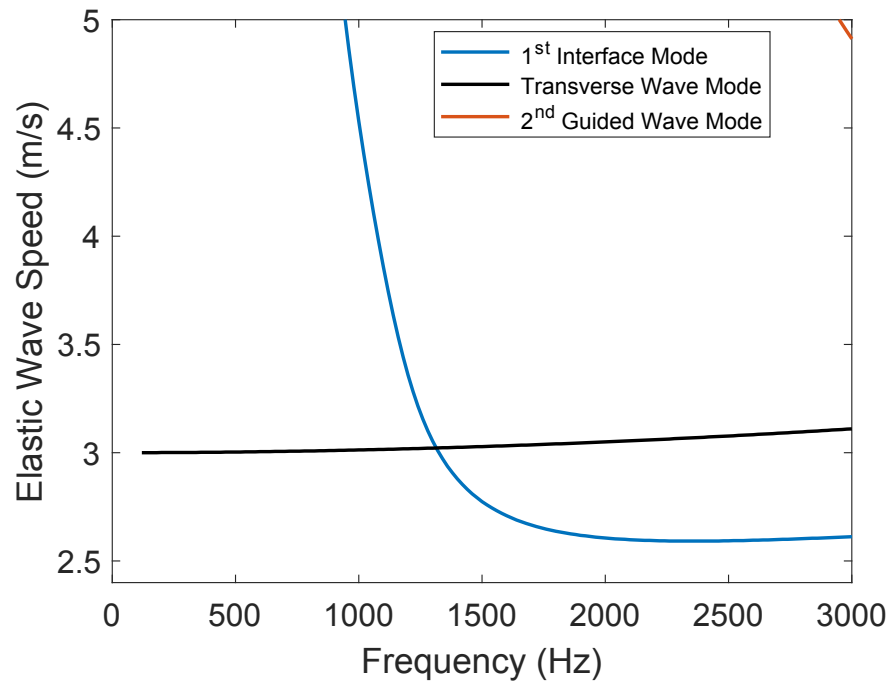
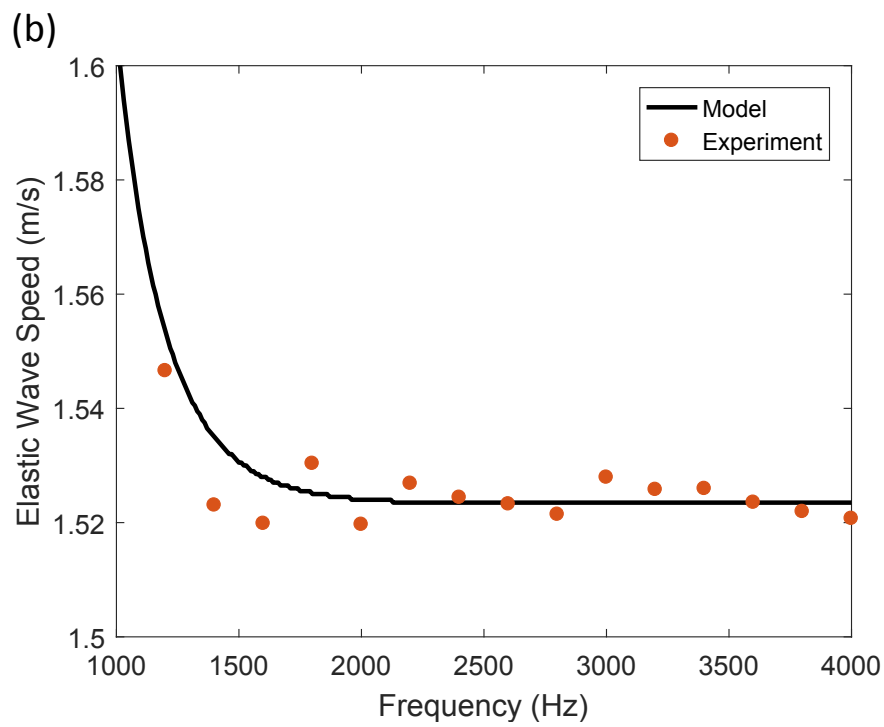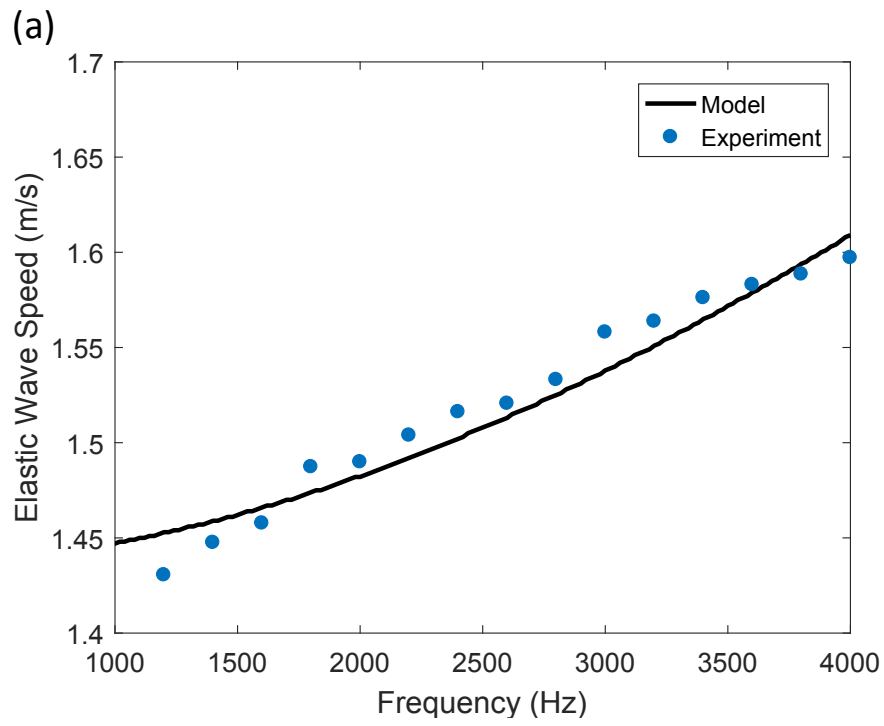**Fig. 5.** Dispersion curves for a 1 mm thick viscoelastic layer with liquid loading and clamped boundary conditions.
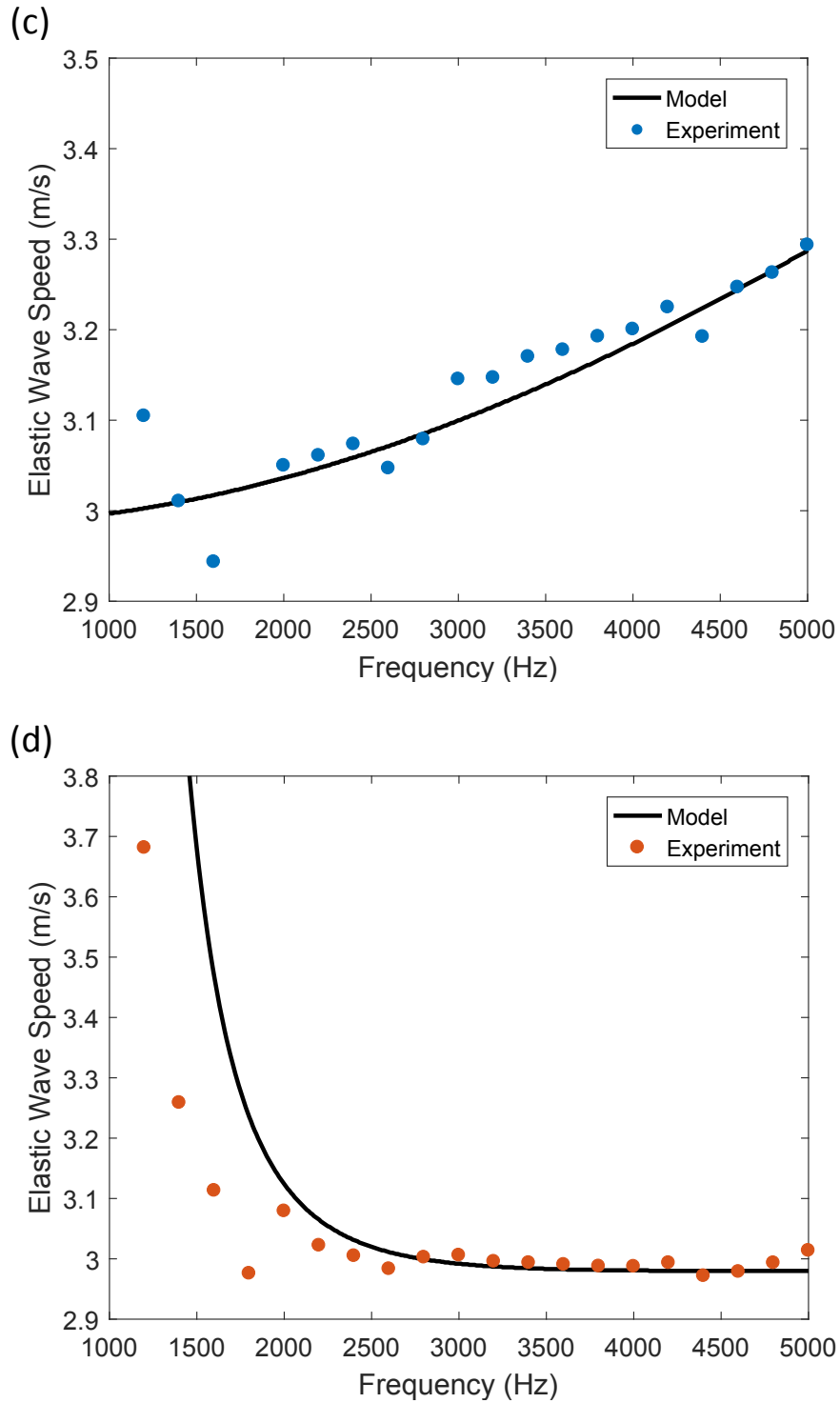
(a)
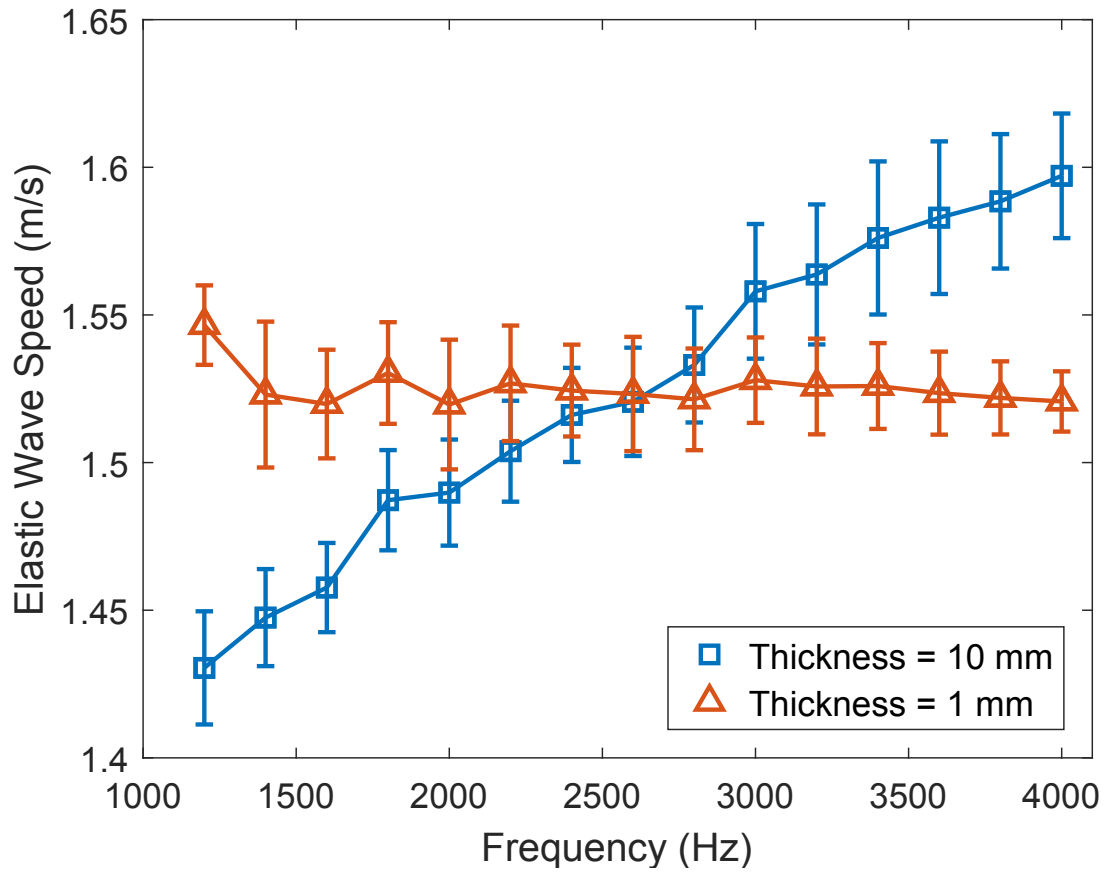
(b)
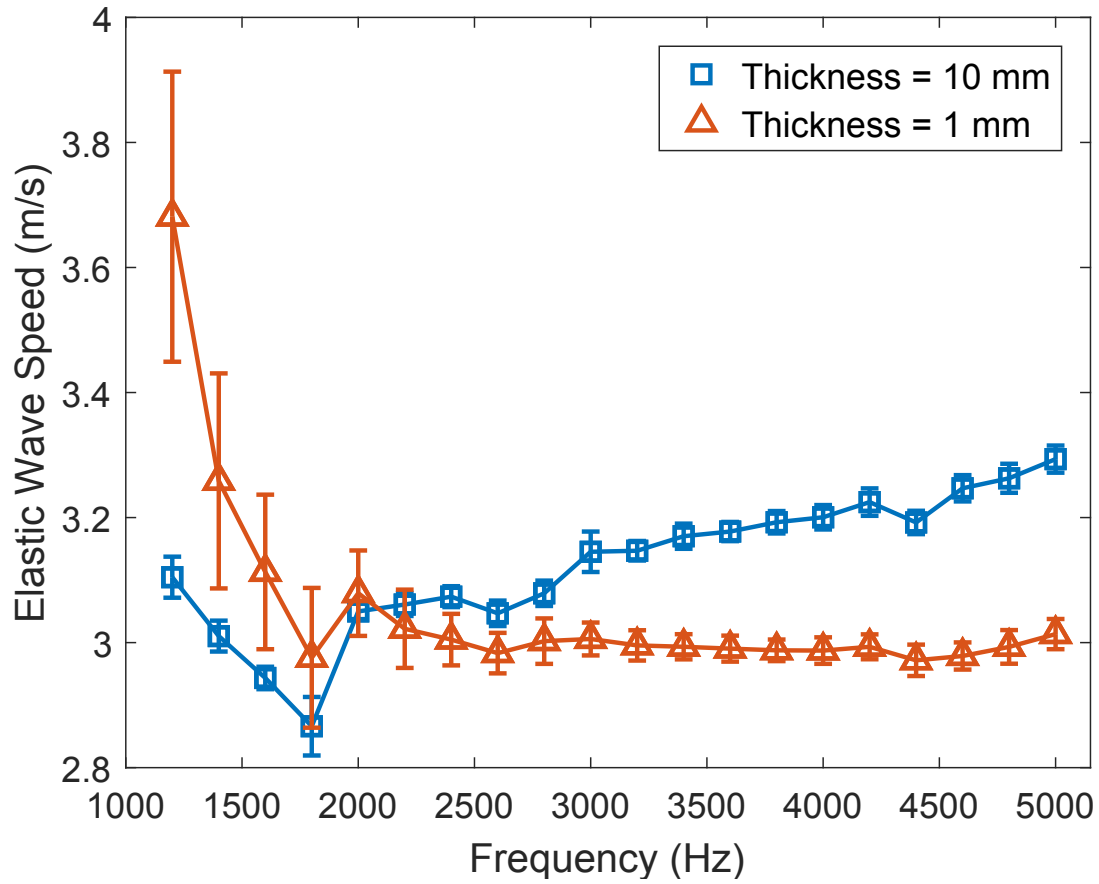
Fig. 6. Measured guided elastic wave dispersion curves in agarose gel phantoms with (a) 1.0% and (b) 2.0% agarose concentrations, for different sample thicknesses.

(a)

(b)

Fig. 7. Comparison of measured and numerical guided elastic wave dispersion curves in gel samples with 1.0% (a and b), and 2.0% (c and d) agarose concentrations, for different sample thicknesses. Sample thickness = 10 mm in (a) and (c), and 1 mm in (b) and (d). Model parameters are listed in Table 1.

Sample Properties	(a)	(b)	(c)	(d)
Agarose gel concentration	1.0%		2.0%	
Thickness (mm)	10	1	10	1
Shear Wave Speed (m/s)	1.71	1.81	3.55	3.55
Complex Shear Viscosity (Pa-s)	0.07		0.22	
Shear Modulus (kPa)	2.9	3.2	12.6	12.6

Table 1: Estimated mechanical properties for agarose gel phantoms by model
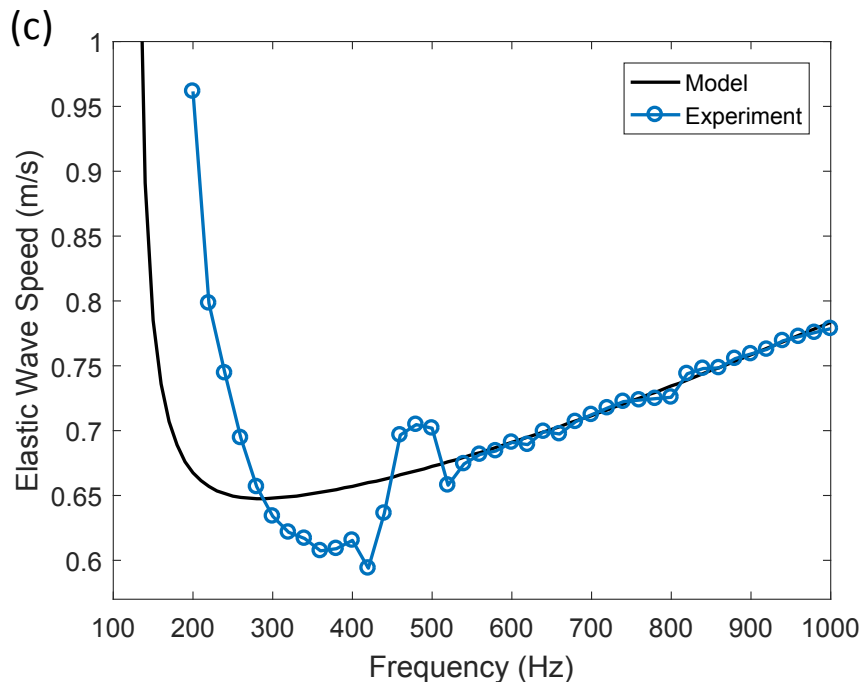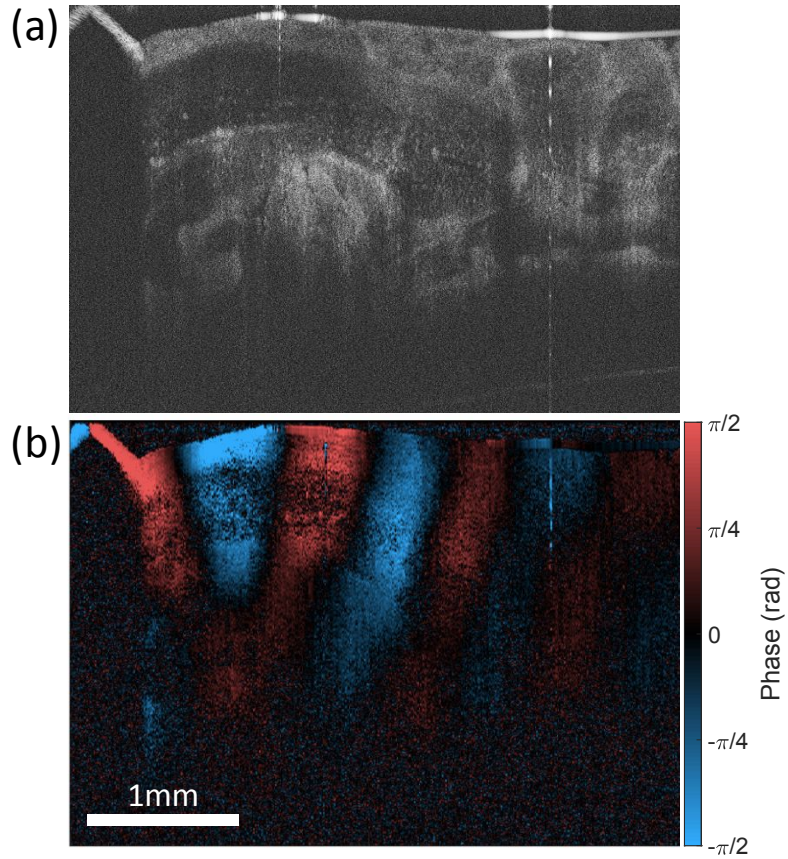
Fig. 8. (a) Optical coherence tomography image of a mixed-culture bacterial biofilm, (b) optical coherence elastography image showing phase distribution of 660 Hz elastic waves in the sample, and (c) dispersion curves for the model (shear modulus 429 Pa and complex shear viscosity 0.06 Pa-s) and excited elastic waves in the sample.