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ARTICLE

Porous Silicon Carbide derived from apple fruit with high electromagnetic absorption performance

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Porous silicon carbide (SiC) material was synthesized by a novel technique, which incorporates the natural porous structure of a fruit, in particular apple. Its characteristic morphological structure has been difficult to synthesize with traditional materials. The novel porous SiC structure was characterized by scanning electron microscopy, X-ray diffraction, and vector network analyses. Microstructural analysis indicated that the microstructure of the porous SiC can be adjusted simply by varying the mass ratio of the Si: C precursors. When stacking faults were introduced into the porous SiC, the EM absorption was greatly enhanced. The lowest reflection loss value reduced from -30 dB to -60 dB. This enhancement was attributed to the introduction of stacking faults in the SiC which result in charge redistribution and generate significant dipole polarization within the porous SiC. This novel porous structure of SiC also facilitated the multi-reflection of EM waves within the absorber which lead to enhanced absorption performance.

Introduction

Electromagnetic (EM) wave absorbing materials are now widely used in civil, commercial and military applications. Extensive studies to enhance their EM absorption performance have been conducted due partly to the ever-increasing EM interference problems. In addition, EM waves have been associated with weakened biological immune systems, breaking of DNA strands, promoting cancers and presenting a threat to human health.¹⁻³ The EM absorbers which are cost effective, relatively light-weight, highly thermal stable, and with wide absorption frequency are in high demand. As a result, there is a need to fabricate new EM absorbing materials with excellent EM absorption performance to meet this high demand.

Porous ceramics have been engineered to combine several advantages (e.g. light-weight, open porosity) inherent to their architecture, making them useful as electrode materials for batteries and supercapacitors¹, super adsorbents⁴, catalyst supports⁵, and chemical and biological sensors.^{6,7} However, porous materials with low densities can be exploited as potential light-weight EM attenuation materials. The attenuation is attributed to internal multiple reflection of EM waves, but this property is rarely reported. Porosity can

dissipate the incident microwaves multiple times to enhance the EM wave absorption property.^{8,9}

Other than the morphological structure, the chemical nature of the material also plays an important role in determining its EM absorption property. For practical application of EM wave absorbing materials, the dielectric property and weight are the two most important elements.¹⁰⁻¹⁴ Among many EM wave absorbing materials, such as MWCNTs, SiC, graphene oxides, and organic polymeric materials,¹⁵⁻²⁷ SiC has many advantages. It is a bio-inert, large band gap semiconductor²⁸, good dielectric property²⁹, and has excellent thermal chemical stability at high temperatures or harsh working environment.^{30,31}

In this study, we report a facile, economic, and environmentally friendly method of fabricating porous SiC with high electromagnetic absorption performance. Currently there is interest in producing carbon based platforms from biomass materials^{32,33}, as they are extremely cheap, easy to obtain, renewable, and nontoxic.³⁴ We took advantage of the natural inter-connected porous structure of a common fruit, apple, as the starting material. Carbonized apple was obtained by pyrolyzing dried apple and found that the porous fibrous structure could be maintained. Then the porous structure was converted to SiC. The conversion was achieved by infiltrating reactive Si vapour in an argon atmosphere without use of any catalyst in the temperature of 1600 °C. By using a natural raw material, we allowed nature to guide the design of the absorber.³⁵ The EM wave absorption performance of porous SiC was evaluated in the frequency range of 2-18 GHz.

Experimental

Materials

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Apples (golden delicious) were obtained from the local market. Morphological analysis of dried apples of different cultivars (such as gala apples, red delicious apples, mcIntosh apples and granny smith apples) indicated that the internal porous structure was essentially the same. Silicon powder was supplied by Sinopharm Chemical Reagent Co., Ltd., China.

Synthesis of porous SiC

Firstly, peeled apple was cut into a particular shape, and then freeze-dried for 48h to produce dried apple. The dried apple was pyrolyzed at 1300 °C under argon atmosphere to generate black carbonized apple. Silicon powder was put in an alumina boat, and then covered by the carbonized apple. The mass ratios of Si to carbonized apple that were investigated were 2:1, 4:1, 6:1, and 8:1, respectively. The alumina boat was placed into the center of a furnace. The temperature of the furnace was ramped to 1600 °C at a rate of 5 °C min⁻¹ and held for 2 h under argon atmosphere at a flow rate of 20 ml min⁻¹ to synthesize porous SiC.

Characterizations

X-ray diffraction (XRD) analysis of the products was conducted with a Rigaku D/max-gB diffractometer. The morphologies of the products were evaluated with a scanning electron microscope (SEM, JSM-6480). An energy-dispersive X-ray spectrometer (EDS) attached to the SEM provided *in situ* determination of composition of the samples. Electromagnetic parameters were analyzed using a vector network analyzer (Agilent, N5230A) in transmission-reflection mode in the 2–18 GHz band. Using paraffin as the binder, the crushed sample powders were pressed to form a cylinder with an inner diameter of 3.0 mm, outer diameter of 7.0 mm and thickness of about 5 mm for electromagnetic measurement.

Results and discussion

Fabrication of porous SiC from apples

Apples were freeze-dried and pyrolyzed to produce the carbon frame of the porous structure. SiC was then *in situ* synthesized by reacting carbon with Si vapor at the high temperature of 1600 °C. The main reaction that took place can be represented by Equation 1. The negative value of the change of Gibbs free energy for the reaction at 1600 °C ($\Delta G^\circ = -51 \text{ kJ/mol}$)³⁶ indicates that the formation of SiC is favorable. Thermodynamic calculations using a chemical thermodynamic software FactSage showed that the Si vapor pressure is in the order of 1×10^{-5} units' atmospheres pressure under the experimental conditions. This relatively high vapor pressure of Si at 1600 °C allowed its quick reaction with carbon to form solid SiC.

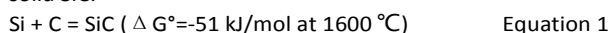


Fig. 1 illustrates the processing steps for making the porous SiC from apple. Apple was firstly freeze-dried in vacuum to maintain the internal structure to the greatest extent. The dried apple was then pyrolyzed at a high temperature of 1300 °C to produce carbonized apple, which resulted in

approximately 20% volume reduction (Supporting Information Figure S1). The final step involved reacting carbonized apple with Si vapor in argon atmosphere at 1600 °C. Si vapor was produced by spreading a layer of Si powder beneath the apple sample in an alumina boat, which vaporized and diffused into the porous apple sample.

The SEM image of the apple as shown in Fig. 2a reveals the cells of the apple. Fig. 2b, c and d illustrate that the freeze-drying process does not destroy the dried apple's cellular, interconnected and well-organized 3D network porous structure. The typical diameter of the pores is 80-100 μm. A SEM image of the carbonized apple in Fig. 2d shows that the porous structure of the apple was well maintained after pyrolysis treatment. The diameter of the pores decreased to 60-80 μm, which correlates well with the total volume shrinkage of about 20% of its macroscopic size (Fig. S1). A SEM image of prepared porous SiC as shown in Fig. 2e confirms that the material had relatively uniform pore sizes which resembled the interconnected network structure of the carbonized apple. Little difference in their morphologies or porous structure could be seen between the carbonized apple and the Si treated carbonized apple. By using this experimental process, the morphology of the final product maintained the natural structure of the starting material (i.e. the apple). This morphology has been proven to be difficult to replicate synthetically but has great potential in the field of EM wave absorption.^{33, 34}

Fig. 3 shows the XRD patterns of the samples. In the XRD pattern of carbonized apple, the peak at 22.4° corresponds to the (002) plane of graphite. While for the porous SiC material, crystalline zinc-blend (cubic) form of 3C-SiC (β -SiC, JCPDS Card No. 29-1129) was found. The diffraction peaks located at 35.60°, 41.38°, 59.98°, 71.78° and 75.49° are associated with the (111), (200), (220), (311) and (222) planes of β -SiC, respectively. A small shoulder which was labelled as SF at $2\theta = 34.7^\circ$ is attributed to stacking faults (SFs) of the β -SiC. The SF peak is found when mass ratio of Si: C is increased to above six. The changes in the diffractogram which were obtained by varying the mass ratio of the precursors, reveals that the porous SiC consisted of different polytypes. The diffraction peaks at 28.5°, 47.3°, and 56.2° correspond to Si which can be detected when the mass ratio of Si: C is below four or above six, which means that the reaction has not gone to completion. In macroscopic view, the treated samples show elemental carbon (carbon residue) and a coating of Si on the carbon surface (excess Si powder) for the Si: C mass ratios of two or eight, respectively. Some morphological features are shown in Fig. 4.

The SEM images of the porous SiC (Si: C mass ratio of 4:1) in Figs. 4a-c illustrate its characteristic porous foam structure. The porous structure has a rough surface because there are large quantities of micropores covering the surface (as shown in Fig 4c). The EDS result in Fig. 4j shows that the pore consists of Si, C and little K, meaning that the porous framework is constituted of SiC matrix with excess carbon. Figs. 4d-f show the porous SiC (Si: C mass ratio of 6:1) at various magnifications. Compared with the porous SiC at the starting

Si: C ratio of 4:1, this porous SiC appears denser. No micro-pores could be observed, and the wall of the pores are crystalline SiC. The EDS analysis which is shown in Fig. 4k reveals that the pore consisted of Si and C with the atomic ratio of approximately 1:1, since more Si participated in the reaction. From the images with higher magnifications (Fig. 4g-i), SiC fibres were found to have grown *in situ* on the porous framework and the diameters of the micro-fibres are in the range of 0.1-1 μm.

The possible mechanism for the growth of the SiC wires are explained as follows. It was mentioned earlier that the partial pressure of Si vapor was very high under the current experimental conditions. It was even higher at a higher ratio of Si: C (e.g. 6:1). FactSage³⁷ simulation revealed that the partial pressure of carbon vapor was in the order of 1×10^{-14} units' atmospheres pressure under the current experimental conditions, which was 9 orders of magnitude lower than that of Si vapor. As a result, the carbon vapor contributed little to the formation of SiC. Si vapor firstly deposited on the sample surface, followed by the diffusion of carbon from inside the sample to the newly-deposited Si layer to allow their reaction forming SiC. Si vapor concentration gradient would be developed due to its continuous consumption on the sample surface. The presence of the concentration gradient resulted in preferential deposition of Si on the tip of SiC wires, because of the presence of highest concentrations of Si vapor around the tips of SiC wires. In order to allow the continuous growth of the SiC wires, carbon needs to diffuse from the bulk of sample through the SiC wires all the way to the tips. The self-diffusion rate of C in β-SiC can be represented by Equation (2)³⁸, where D_c is the lattice self-diffusion coefficient of carbon in β-SiC, k is the Boltzmann constant, and T is temperature. Assuming the diffusion of carbon is the rate-limiting step in the formation of SiC wires, it was calculated that approximately 5 μm long SiC wires would form after 2 hours of reaction at 1600 °C, which is in good agreement with the observations under SEM (Figure 4).

$$D_c = (1.97 \pm 1.50) \times 10^8 \exp\left(\frac{-8.67 \pm 15eV/atom}{kT}\right) cm^2 sec^{-1} \quad \text{Equation (2)}$$

Fig. 4i illustrates the highly populated stacking faults which is characteristic of the micro-fibres. Plenty of step-like streaked lines can be seen as strips in the domains, corresponding to the SF peak in XRD pattern (Fig. 3b). The stacking faults could enhance the dipole polarization and magnify energy dissipation under a varying electromagnetic field.³⁹ Therefore, such porous SiC materials are expected to exhibit good performance in the electromagnetic absorption. It is worth noticing that the two types (crystalline and fibre) of the porous SiC show different porous micro-structures, which were developed by simply changing the Si: C ratios.

EM absorption performance of the porous SiC

Generally, the microwave absorption property is determined by the complex permittivity ($\epsilon_r = \epsilon' - j\epsilon''$) and permeability ($\mu_r = \mu' - j\mu''$). The real parts (ϵ' and μ') of the complex permittivity and permeability represent the storage capability of the electric

and magnetic energy. The imaginary parts (ϵ'' and μ'') represent the loss capability of electric and magnetic energy. Figure 5 shows the frequency dependence of the real part (ϵ') and imaginary part (ϵ'') of the permittivity of the samples in the range of 2-18 GHz. The real part (μ') and imaginary part (μ'') of permeability are 1.0 and 0.0 (not shown). It can be seen that both ϵ' and ϵ'' increase with the filler content (mass ratio of Si: C being 6:1). When the loading of SiC is 60 wt%, ϵ' is around 10, which is twice that when the loading is 40 wt% ($\epsilon' \approx 5$). ϵ' increases from 7.5 to 10 when the precursor Si: C mass ratio is increased from 4:1 to 6:1. Compared with the significant changes of ϵ' , there is little change for ϵ'' . β-SiC exhibit dielectric resonance features in the permittivity spectra. There is a resonance peak located at approximately 9 GHz with the loading of 60 wt% (mass ratio of Si: C being 6:1). No obvious resonance peak can be found when the Si: C mass ratio was 4:1. We propose that the dielectric resonance of SiC is enhanced by the presence of the stacking faults. The SF phenomenon correlates with specific processing conditions (Fig. 3). The stacking faults alter the porous crystal structure of SiC, change the electric charge distribution balance and introduces an orientational polarization. These enhance the permittivity.

EM wave absorption performance is basically estimated by reflection loss (R_L), which can be calculated from the complex permeability and permittivity at a given frequency (2-18 GHz) and the absorber thickness by the transmission line theory as the following equations:

$$Z_{in} = Z_0 (\mu_r / \epsilon_r)^{1/2} \tan h [j(2\pi f d / c)(\mu_r \epsilon_r)^{1/2}] \quad \text{Equation (3)}$$

$$R_L (dB) = 20 \log \left| \frac{Z_{in} - Z_0}{Z_{in} + Z_0} \right| \quad \text{Equation (4)}$$

where f is the EM wave frequency, d is the thickness of the absorber, c is the velocity of light in free space, Z_{in} is the input impedance of the absorber, and Z_0 is the impedance of air.⁴⁰ Figs. 6a and b show the EM wave absorption characteristics of the porous SiC starting with a Si: C mass ratio of 6:1. It can be seen that the minimum R_L s are -1.1 dB at 12-14.8 GHz and -11 dB at 17.6-18 GHz with loadings of 40 wt% and 50 wt%, respectively. These composites exhibit little microwave absorption. However, a significant enhancement of the absorption ability was noted when the loading increased to 60 wt%, where the minimum R_L value reaches as low as -60 dB at 17-18 GHz. This substantial enhancement of the permittivity was achieved while there was little impact on the permeability (shown in Fig. 5). Fig. 6e demonstrates that the absorber exhibits an excellent absorption performance ($R_L < -10$ dB) at a high frequency range from 14 to 18 GHz with the sample thickness of 4.0-5.0 mm. For comparison, the lowest reflection loss for SiC porous materials (Si: C mass ratio being 4:1, loading 60 wt%) is -30 dB at 17.8 GHz (Fig. 6d), which demonstrates a good absorption performance. This supports the claim that the enhancement of the EM absorption (compared with that of the SiC with Si: C of 6:1, loading 60 wt%) could be attributed to the presence of stacking faults (SFs) in the β-SiC. This

dependency of the EM absorption performance on the presence of stacking faults has been reported in our previous work³⁹. Table 1 lists the EM absorption performance of some porous or SiC materials obtained from literature. Compared to these reported materials, porous SiC has the best performance on high frequency range. Along with its high strength, thermal stability, oxidation and corrosion resistance, the porous SiC has the potential to become one of the best EM absorbers.

Table 1 EM absorption of some representative porous structure or SiC absorber materials.

Samples	Minimum R_L (dB)	Frequency range (GHz) ($R_L < -10$ dB)	Frequency coverage (GHz) ($R_L < -10$ dB)	Ref.
Porous Co/CoO	-50	13.8-18	4.2	41
Porous C fiber	-25.65	8-18	10	42
SiC NWs	-31.7	7.1-9.6	2.5	43
SiC NPs	-18	9.5-11.8	2.3	44
Porous SiC	-60	14-18	4	Current study

EM absorption mechanism of the porous SiC

The enhanced EM absorption performance can be essentially attributed to two factors. The first factor is the presence of its unique porous structure.⁴⁵⁻⁴⁷ The porous SiC with appropriate filler content has a well-connected architecture. The architecture was inherited from the natural honeycomb porous structure from apple which well integrates macropores, mesopores, and micropores. The macropores guaranteed and enabled the accessibility of the mesopores and micropores. This may be critical to the remarkable increase of permittivity which improves the impedance matching between porous structure and free space, which results in the high EM absorption performance. Furthermore, the porous structures inherited from nature which have a broad pore size distribution are robust and interconnected. The EM waves could easily irradiate into the porous structure. The incident waves have higher chances of being attenuated through reflection and scattering for numerous times. The absorbed electromagnetic energy must be readily converted to thermal energy. Absorption performance is enhanced significantly because the EM waves cannot escape from the porous structure.

Secondly, as a typical dielectric absorber, SiC accomplishes EM absorption by intrinsic electric dipole polarization.⁴⁸ The produced porous SiC with a highest stacking fault concentration exhibits the best EM absorption ability. Stacking fault enhancement of EM absorption has been previously reported by our group³⁹. The stacking faults in the SiC material changes its crystal structure which produces a large number of vacancies, antisites and layer interruptions, and thus destroy the balance of the electric charge distribution. The SFs cause dipole fields to be introduced with dipole moments which cause multiple dipole polarizations and orientational

polarizations all of which enhance energy dissipation and, as a consequence, enhance EM absorption.

Conclusions

Porous SiC was produced with a unique morphological structure. The fabrication took advantage of the natural microstructure of the apple fruit. This porous SiC material showed high performance for EM absorption. The porous framework was composed of different SiC crystal structures, and the SiC polytype composition was able to be adjusted simply by varying the mass ratio of the C and Si precursors. The stacking faults in the SiC was proposed to induce enhanced electric charge redistribution and generate extensive polarizations and energy dissipation, thus attenuating the incident EM waves. The cellular, interconnected, well-organized 3D network porous structure likely reflects and scatters the EM waves numerous times to further enhance the EM absorption performance. When the porous SiC was produced with the appropriate filler content, the reflection loss of SiC porous materials reached as low as -60 dB. The porous SiC material is promising as a cost effective, lightweight, corrosion-resistant and highly thermostable multifunctional EM absorber.

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Notes and references

- 1 D. R. Rolison, J. W. Long, J. C. Lytle, A. E. Fischer, C. P. Rhodes, T. M. McEvoy, M. E. Bourga and A. M. Lubers, *Chem. Soc. Rev.*, 2009, **38**, 226-252.
- 2 R. Girgert, C. Grundker, G. Emons and V. Hanf, *Bioelectromagnetics*, 2008, **29**, 169-176.
- 3 J. R. Jauchem, K. L. Ryan and M. R. Frei, *Bioelectromagnetics*, 2000, **21**, 159-166.
- 4 X. Gui, J. Wei, K. Wang, A. Cao, H. Zhu, Y. Jia, Q. Shu and D. Wu, *Adv. Mater.*, 2010, **22**, 617.
- 5 C. Moreno-Castilla, and F. J. Maldonado-Hodar, *Carbon*, 2005, **43**, 455-465.
- 6 A. Richter, G. Paschew, S. Klatt, J. Lienig, K. F. Arndt and H. J. Adler, *Sensors*, 2008, **8**, 561.
- 7 J. H. Zou, J. H. Liu, A. S. Karakoti, A. Kumar, D. Joung, Q. A. Li, S. I. Khondaker, S. Seal and L. Zhai, *ACS Nano*, 2010, **4**, 7293.
- 8 J. Wang, C. Xiang, Q. Liu, Y. Pan and J. Guo, *Adv. Funct. Mater.*, 2008, **18**, 2995-3002.
- 9 J. Zhou, J. He, G. Li, T. Wang, D. Sun, X. Ding, J. Zhao and S. Wu, *J. Phys. Chem. C*, 2010, **114**, 7611-7617.
- 10 H. Yang, M. Cao, Y. Li, H. Shi, Z. Hou, X. Fang, H. Jin, W. Wang and J. Yuan, *Adv. Opt. Mater.*, 2014, **2**, 214-219.
- 11 Z. M. Dang, T. Zhou, S. H. Yao, J. K. Yuan, J. W. Zha, H. T. Song, J. Y. Li, Q. Chen, W. T. Yang and J. Bai, *Adv. Mater.*, 2009, **21**, 2077.

- 12 P. C. P. Watts, W. K. Hsu, A. Barnes and B. Chambers, *Adv. Mater.*, 2003, **15**, 600-603.
- 13 Y. Duan, Z. Liu, H. Jing, Y. Zhang and S. Li, *J. Mater. C* 2012, **22**, 18291-18299.
- 14 Z. Chen, C. Xu, C. Ma, W. Ren and H. M. Cheng, *Adv. Mater.*, 2013, **25**, 1296-1300.
- 15 L. Kong, X. W. Yin, X. Y. Yuan, Y. J. Zhang, X. M. Liu, L. F. Cheng and L. T. Zhang, *Carbon*, 2014, **73**, 185-193.
- 16 M. K. Han, X. W. Yin, L. Kong, M. Li, W.Y. Duan, L.T. Zhang and L. F. Cheng, *J. Mater. Chem. A*, 2014, **2**, 16403-16409.
- 17 L. Kong, X. W. Yin, Y. J. Zhang, X. Y. Yuan, Q. Li, F. Ye, L. F. Cheng, and L. T. Zhang, *J. Phys. Chem. C*, 2013, **117**, 19701-19711.
- 18 S. L. Shi, and J. Liang, *Nanotechnology*, 2008, **19**, 255707.
- 19 Y. X. Gong, L. Zhen, J. T. Jiang, C. Y. Xu, W. S. Wang and W. Z. Shao, *Crystengcomm*, 2011, **13**, 6839-6844.
- 20 B. Wen, M. Cao, M. Lu, W. Cao, H. Shi, J. Liu, X. Wang, H. Jin, X. Fang, W. Wang and J. Yuan, *Adv. Mater.*, 2014, **26**, 3484-3489.
- 21 W. L. Song, M. S. Cao, M. M. Lu, J. Liu, J. Yuan and L. Z. Fan, *J. Mater. Chem. C*, 2013, **1**, 1846-1854.
- 22 Y. Sun, F. Xiao, X. Liu, C. Feng and C. Jin, *Rsc Advances*, 2013, **3**, 22554-22559.
- 23 M. S. Cao, J. Yang, W. L. Song, D. Q. Zhang, B. Wen, H. B. Jin, Z. L. Hou and J. Yuan, *Acs Appl. Mater. Inter.*, 2012, **4**, 6948-6955.
- 24 C. Y. Liu, Y. J. Xu, L. N. Wu, Z. H. Jiang, B. Z. Shen and Z. J. Wang, *J. Mater. Chem. A*, 2015, **3**, 10566-10572.
- 25 G. S. Wang, Y. Wu, Y. Z. Wei, X. J. Zhang, Y. Li, L.D. Li, B. Wen, P. G. Yin, L. Guo and M. S. Cao, *Chempluschem*, 2014, **79**, 375-381.
- 26 X. Luo, G. S. Wang, H. Y. Guo, X. J. Zhang, W. Q. Cao, Y. Z. Wei, L. Guo and M. S. Cao, *Chempluschem*, 2014, **79**, 1089-1095.
- 27 X. J. Zhang, G. S. Wang, W. Q. Cao, Y. Z. Wei, J. F. Liang, L. Guo and M. S. Cao, *Acs Appl. Mater. Inter.*, 2014, **6**, 7471-7478.
- 28 N. Leventis, A. Sadekar, N. Chandrasekaran and C. Sotiriou-Leventis, *Chem. Mater.*, 2010, **22**, 2790-2803.
- 29 L. S. Sigl, *J. Eur. Ceram. Soc.*, 2003, **23**, 1115-1122.
- 30 R. Naslain, *Compos. Sci. Technol.*, 2004, **64**, 155-170.
- 31 N. G. Wright and A. B. Horsfall, *J. Appl. Phys.*, 2007, **40**, 6345-6354.
- 32 L. J. Bonderer, A. R. Studart and L. J. Gauckler, *Science*, 2008, **319**, 1069-1073.
- 33 M.E. Randall, L. Rafael, R. Nuria and R. S. Andre, *Science*, 2012, **13**, 199-204.
- 34 H. W. Liang, Q. F. Guan, L. F. Chen, Z. Zhu, W. J. Zhang and S. H. Yu, *Angew. Chem. Int. Edit.*, 2012, **51**, 5101-5105.
- 35 S. Deville, E. Saiz, R. K. Nalla and A. P. Tomsia, *Science*, 2006, **311**, 515-518.
- 36 A. Roine: HSC Chemistry, vesion 6.12 for Windows, Outokumpu Research Oy, Pori, Finland, 2007.
- 37 C. W. Bale, E. Bélisle, P. Chartrand, S. A. Decterov, G. Eriksson, K. Hack, I. H. Jung, Y. B. Kang, J. Melançon, A. D. Pelton, C. Robelin and S. Petersen, *Calphad*, 2009, **33**, 295-311.
- 38 M. H. Hon and R. F. Davis, *J. Mater. Sci.*, 1979, **14**, 2411-2421.
- 39 H. Zhang, Y. Xu, J. Zhou, J. Jiao, Y. Chen, H. Wang, C. Liu, Z. Jiang and Z. Wang, *J. Mater. Chem. C*, 2015, **3**, 4416-4423.
- 40 P. Xu, X. Han, B. Zhang, Y. Du and H.-L. Wang, *Chem. Soc. Rev.*, 2014, **43**, 1349-1360.
- 41 Q. L. Liu, D. Zhang and T. X. Fan, *Appl. Phys. Lett.*, 2008, **93**, 013110.
- 42 W. Xie, H. F. Cheng, Z. Y. Chu, Y. J. Zhou, H. T. Liu and Z. H. Chen, *Mater. Design*, 2009, **30**, 1201-1204.
- 43 S.C. Chiu, H.C. Yu and Y.Y. Li, *J. Phys. Chem. C*, 2010, **114**, 1947.
- 44 Y. K. Dou, J. B. Li, X. Y. Fang, H. B. Jin and M. S. Cao, *Appl. Phys. Lett.*, 2014, **104**, 052102.
- 45 B. Zhao, G. Shao, B. B. Fan, W. Y. Zhao, Y. J. Xie and R. Zhang, *J. Mater. Chem. A*, 2015, **3**, 10345-10352.
- 46 B. Zhao, W. Y. Zhao, G. Shao, B. B. Fan and R. Zhang, *Dalton Trans.*, 2015, **44**, 15984-15993.
- 47 B. Zhao, B. B. Fan, Y. W. Xu, G. Shao, X. D. Wang, W. Y. Zhao and R. Zhang, *Acs Appl. Mater. Inter.*, 2015, **7**, 26217-26225.
- 48 R. B. Wu, K. Zhou, Z. H. Yang, X. K. Qian, J. Wei, L. Liu, Y. Z. Huang, L. B. Kong and L.Y. Wang, *Crystengcomm*, 2013, **15**, 570-576.

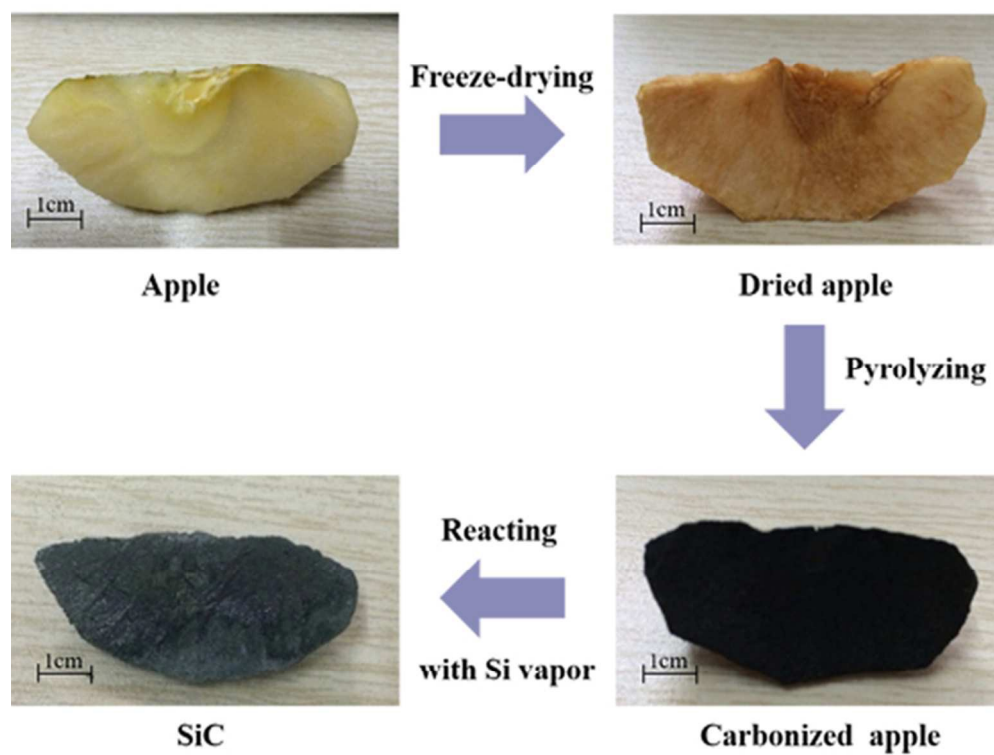


Fig. 1 Photographs illustrating fabrication process of porous SiC.
67x54mm (300 x 300 DPI)

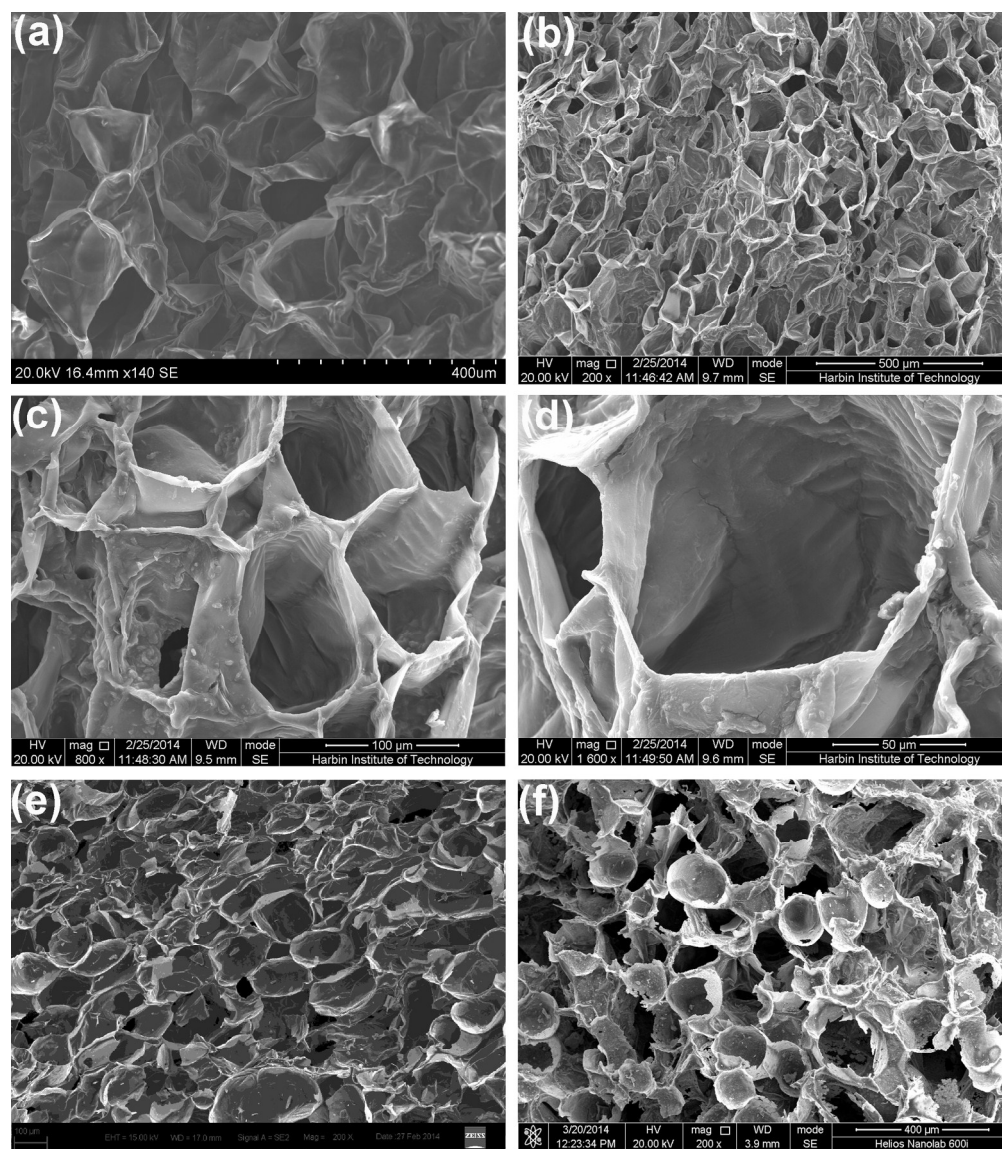


Fig. 2 SEM images of (a) apple; (b), (c), (d) dried apple at three various magnifications; (e) carbonized apple; (f) porous SiC.
94x107mm (600 x 600 DPI)

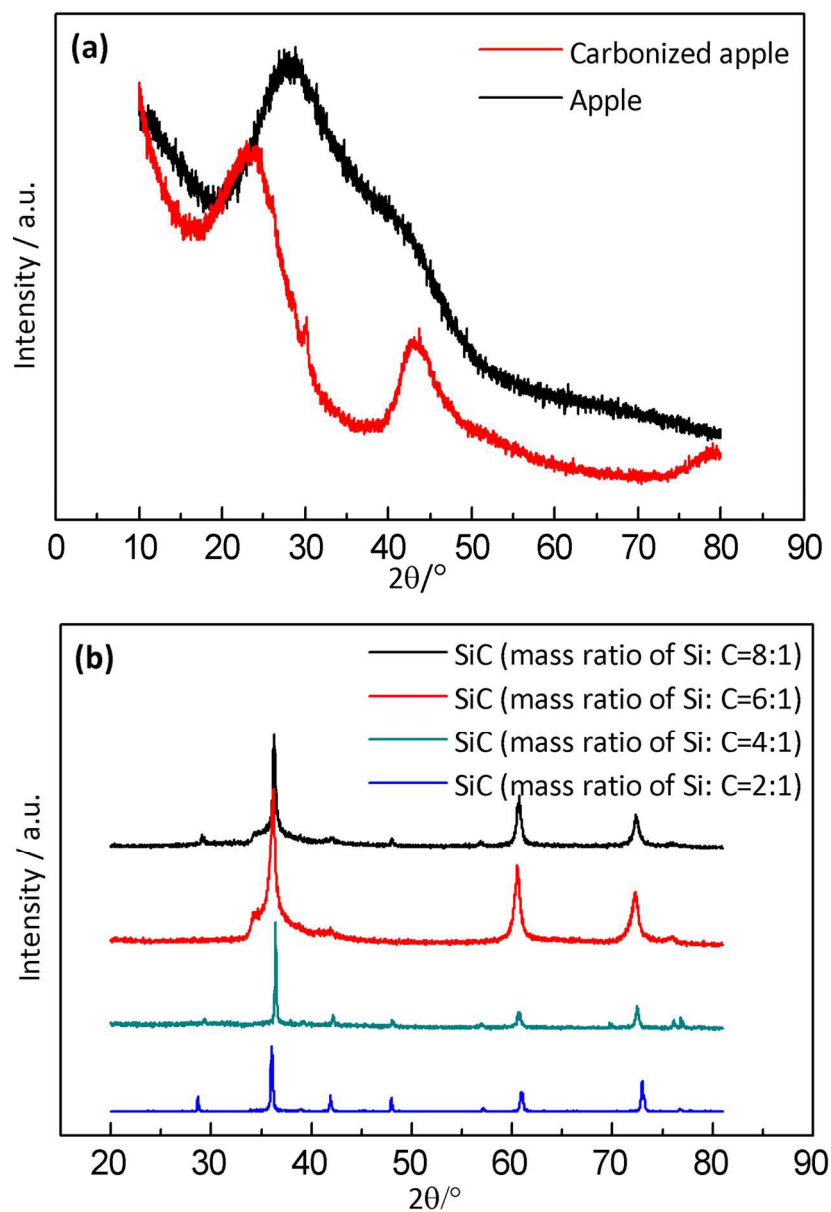


Fig. 3 XRD patterns of (a) the apple and carbonized apple; (b) porous SiC with the Si: C mass ratio of 2:1, 4:1, 6:1, and 8:1.
121x178mm (300 x 300 DPI)

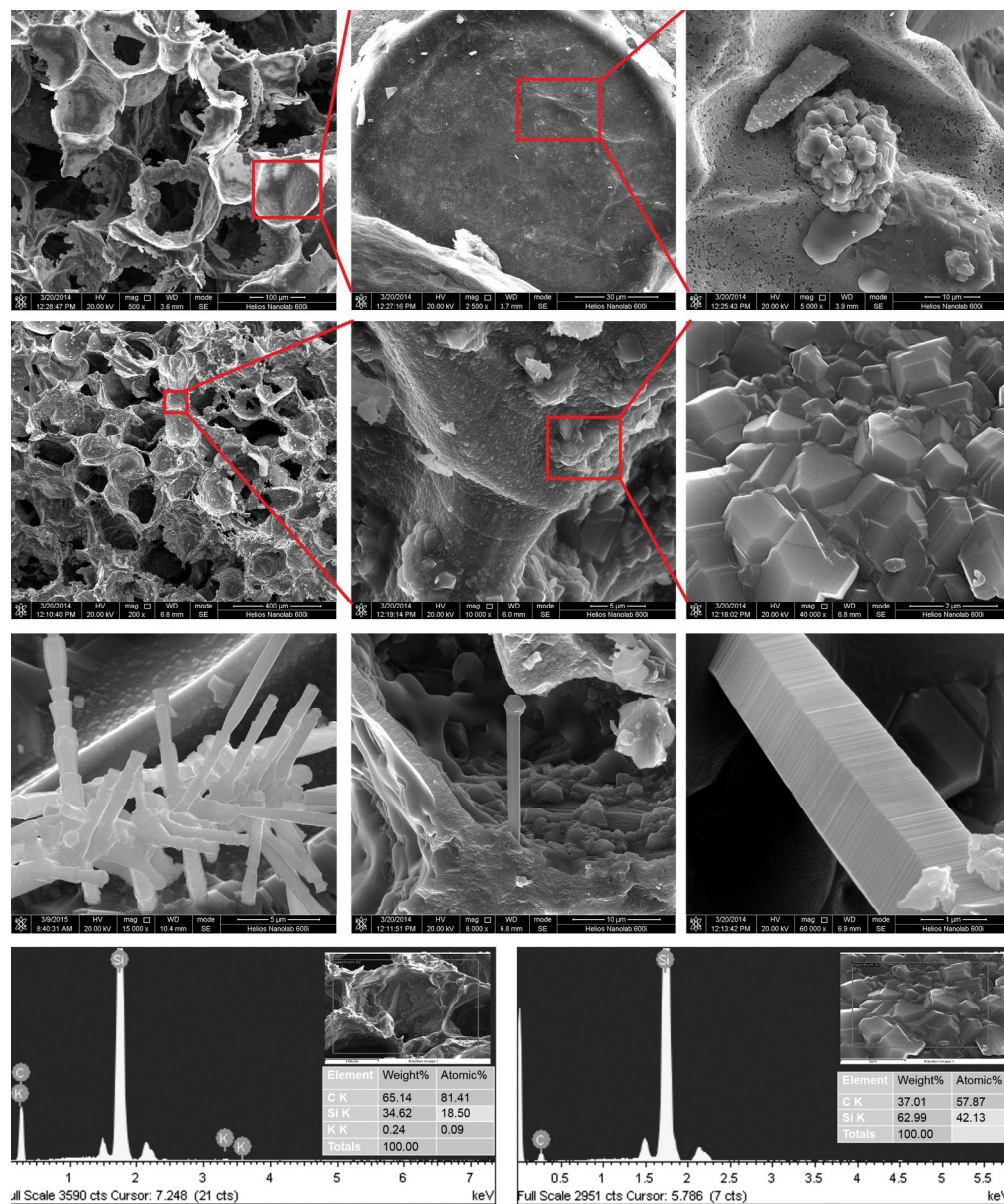


Fig. 4 SEM images of porous SiC at different magnifications. Mass ratio of Si: C being (a-c) 4:1; (d-i) 6:1 and corresponding EDS results of mass ratio of Si: C being (j) 4:1; (k) 6:1. 98x118mm (600 x 600 DPI)

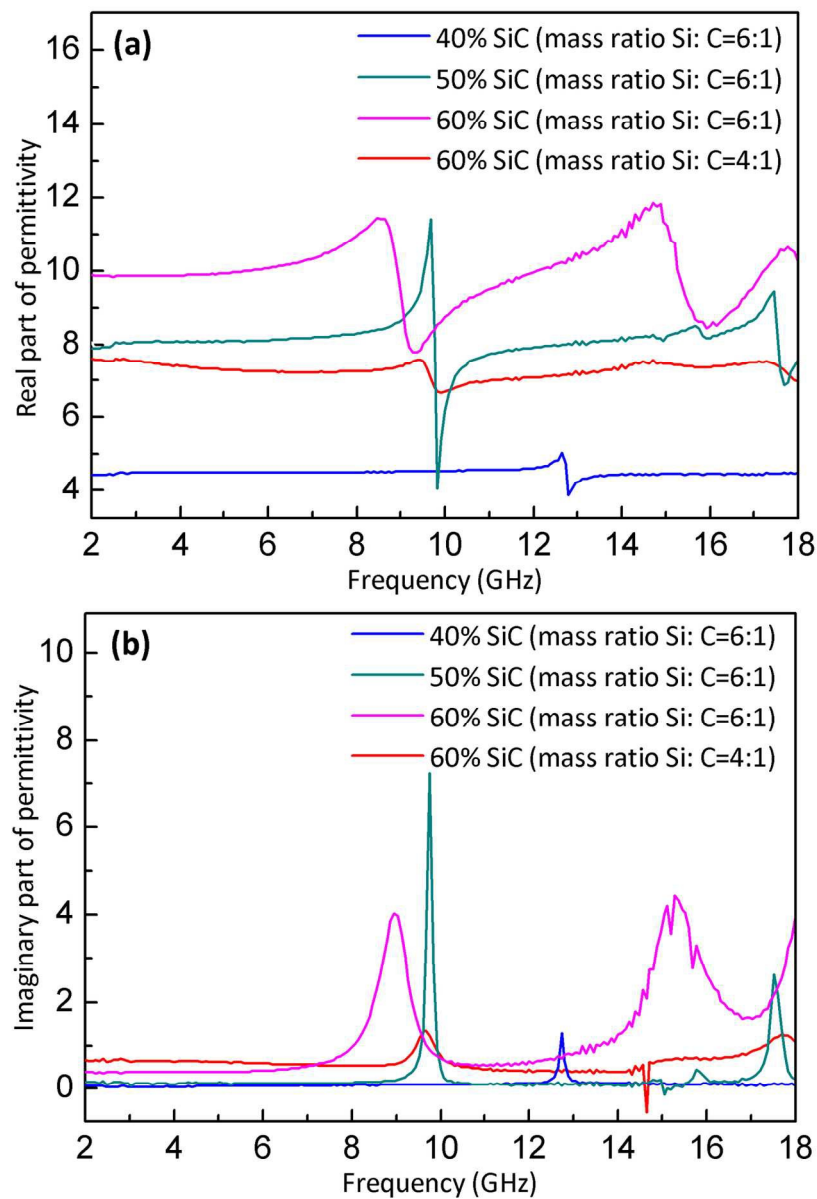


Fig. 5 Real part (a) and imaginary part (b) of the complex permittivity spectra of the samples in the 2–18 GHz range.

119x172mm (300 x 300 DPI)

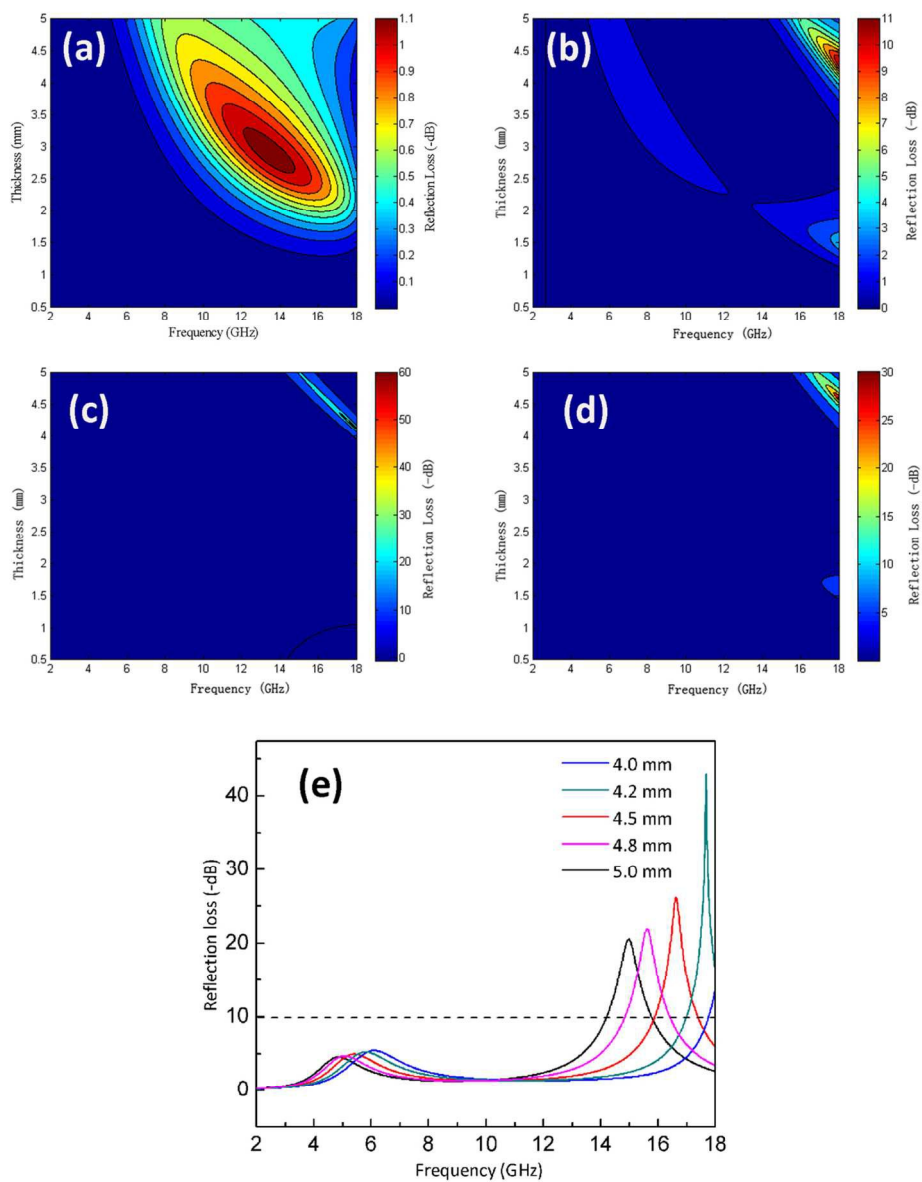


Fig. 6 EM wave absorption characteristics of the porous SiC (Si: C mass ratio of 6:1) with loadings of (a) 40%, (b) 50% and (c) 60%; (d) porous SiC (loading 60 wt%) as a function of mass ratio of Si: C being 4:1; (e) optimal reflection loss spectra plotted against is the frequency for porous SiC with different thicknesses. 104x131mm (300 x 300 DPI)

Table of contents entry:

Porous Silicon Carbide derived from apple fruit with high electromagnetic absorption performance

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A method for synthesizing porous SiC with excellent electromagnetic absorption performance which incorporates natural porous structure of apple was reported.

