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ARTICLE

## Solubility Product Difference-Guided Synthesis of $\text{Co}_3\text{O}_4$ - $\text{CeO}_2$ Core-Shell Catalysts for CO Oxidation

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It is still an important issue of developing a facile, environmental way to synthesize bimetal oxide materials. In this paper,  $\text{Co}_3\text{O}_4$ - $\text{CeO}_2$  core-shell catalysts were prepared by an interfacial reaction, where  $\text{Co}(\text{CO}_3)_{0.35}\text{Cl}_{0.2}(\text{OH})_{1.1}$  nanorods were dispersed in  $\text{Ce}^{3+}$  aqueous solution for 2 days, followed by a calcination step. The samples were characterized by X-ray diffraction (XRD), transmission electron microscopy (TEM), and X-ray photoelectron spectroscopy (XPS). Based on the characterization and comparative experimental results, we proposed that the  $\text{OH}^-$  ions slowly dissociated from  $\text{Co}(\text{CO}_3)_{0.35}\text{Cl}_{0.2}(\text{OH})_{1.1}$  precursor combine with  $\text{Ce}^{3+}$  to develop into  $\text{Ce}(\text{OH})_3$  nanoparticles because of its smaller solubility product constant than that of Co precursor or  $\text{Co}(\text{OH})_2$ . Neither additional precipitation agent nor stabilizing molecules were employed during the whole preparation. Raman spectroscopy and  $\text{H}_2$ -Temperature program reduction ( $\text{H}_2$ -TPR) analysis revealed that there is a synergistic effect between  $\text{Co}_3\text{O}_4$  and  $\text{CeO}_2$  in the as-prepared  $\text{Co}_3\text{O}_4$ - $\text{CeO}_2$  core-shell catalysts, which is responsible for their enhanced catalytic activity toward CO oxidation in comparison to pure  $\text{Co}_3\text{O}_4$  and  $\text{CeO}_2$ .

### 1. Introduction

It is well known that ceria ( $\text{CeO}_2$ ) is a key component in the formulation of catalysts owing to its excellent redox property and high oxygen storage capacity, thus, has attracted tremendous interest in fundamental studies and practical applications.<sup>1-14</sup> Over the last decade, the synthesis of  $\text{CeO}_2$ -based bimetal oxide catalysts and the optimization of their size, morphology, and composition were extensively studied because some bimetal oxides exhibit remarkable properties distinctly different from their monometallic counterparts, known as the synergistic effect in catalysis due to the interaction between two oxides.<sup>15-26</sup> During the catalytic reactions, the interfacial contact between  $\text{CeO}_2$  and secondary metal oxide is proposed to be active site, which is of great importance in determining their catalytic performance. Therefore, from a structural viewpoint, a considerable contact within components is highly expected in a good catalyst, which can enable an efficient diffusion of reactants into the active sites. Until now, co-precipitation,<sup>27</sup> impregnation<sup>28</sup> and sol-gel<sup>29</sup> approaches are widely used to prepare  $\text{CeO}_2$ -based bimetal oxides. Although the components mix well even at the atomic level, the exposure of the active sites to reactants is suppressed for the catalysts prepared using the former two routes, except in the case of the utilization of specific templates (e.g. mesoporous templates).<sup>28</sup> Since the hydrolysis and condensation rates of the precursors differ considerably, phase separation is ineluctable during the sol-gel process, thus leading to the decrease in contact within components.<sup>29</sup>

To address the issues above, the fabrication of bimetal oxides with

core-shell structures by depositing the secondary shell onto the pre-synthesized core surface seems feasible, because the sufficient contact between core and shell is guaranteed in this kind of structure.<sup>20,21,30,31</sup> However, the conventional layer-by-layer technique is a multistep process that requires precise control in surface modification and shell deposition. Specifically, to avoid homogeneous nucleation of the shell component, it is necessary to perform a surface modification to the core with the assistance of organic species (e.g. polymers or surfactants). Because the heterogeneously catalytic reaction occurs on the catalyst surface, these organic species need to be removed so as to fully expose the surface active sites. Unfortunately, a complete removal of these organic species is still challenging, and hence resulting in the deactivation of the catalyst. Moreover, the organic compounds and precipitation agents for the shell deposition always need to be carefully chosen. Obviously, this method is not economically feasible for a mass production of catalysts and is not preferred in industrial fields. Therefore, it is of highly interest to develop a surface modification-free, and easily scale-up approach to synthesize bimetal oxide catalysts with core-shell structure.

Solid-liquid interfacial reaction, proceeded by a sacrificial solid template reacting with one designated solution, has been identified as an effective method to synthesize core-shell structured materials. In this method, the surface modification-free manipulation can be realized as the formation of new shells is merely driven by the chemical reactions between templates and ions in solution. For example, a simple redox reaction in  $\text{Ce}^{4+}$  ion solution with the presence of  $\text{Cu}_2\text{O}$  as the sacrificial template leads to the formation of  $\text{Cu}_2\text{O}$ - $\text{CeO}_2$  core-shell bimetal oxides.<sup>20</sup> Recently, we also prepared  $\text{CeO}_2$ - $\text{MnO}_2$  bimetal oxides by treating  $\text{Ce}(\text{OH})\text{CO}_3$  templates with  $\text{KMnO}_4$  aqueous solution, where  $\text{MnO}_4^-$  is reduced to  $\text{MnO}_2$  and the  $\text{Ce}^{3+}$  in  $\text{Ce}(\text{OH})\text{CO}_3$  is simultaneously oxidized to form  $\text{CeO}_2$ .<sup>32</sup> In addition to the redox interfacial reaction, ion exchange is a well-

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known way to synthesize core-shell nanomaterials. For example,  $\text{Cu}_2\text{O-Cu}_2\text{S}$  core-shell structure can be achieved upon the addition of  $\text{Cu}_2\text{O}$  templates into  $\text{Na}_2\text{S}$  solution due to the small solubility product constant of  $\text{Cu}_2\text{S}$  ( $K_{\text{sp}}=10^{-48}$ ).<sup>33</sup> So far, the ion exchange procedure is restricted to chalcogenide based core-shell nanomaterials,<sup>34</sup> in contrast, the synthesis of bimetal oxide with core-shell structures via an ion exchange method has not been extensively studied yet.

In this paper, we choose  $\text{Co}_3\text{O}_4\text{-CeO}_2$  nanostructure as an example to demonstrate the interfacial process capable of generating core-shell bimetal oxides.  $\text{Co}_3\text{O}_4\text{-CeO}_2$  bimetal oxide is an interesting catalyst that was studied in CO, hydrocarbon and diesel oxidation reactions.<sup>24,25,29,30</sup> Herein,  $\text{Co}_3\text{O}_4\text{-CeO}_2$  core-shell structured nanorods were successfully fabricated by easily reacting the  $\text{Co}(\text{CO}_3)_{0.35}\text{Cl}_{0.2}(\text{OH})_{1.1}$  nanorods with  $\text{Ce}^{3+}$  aqueous solution, followed by calcination. Compared to the traditional synthetic process, neither additional precipitation agent nor stabilizing molecules are involved during the whole preparation. The solubility product difference between  $\text{Ce}(\text{OH})_3$  and the employed Co precursor plays the key role for the formation of the  $\text{CeO}_2$  shell.

## 2. Experimental

### 2.1 Materials

Cobalt chloride hexahydrate ( $\text{CoCl}_2 \cdot 6\text{H}_2\text{O}$ ) were purchased from Aladdin Industrial Corporation, urea ( $\text{H}_2\text{NCONH}_2$ ), cerium(III) nitrate hexahydrate ( $\text{Ce}(\text{NO}_3)_3 \cdot 6\text{H}_2\text{O}$ ), ammonium cerium(IV) nitrate ( $(\text{NH}_4)_2\text{Ce}(\text{NO}_3)_6$ ) and ethanol were purchased from Sinopharm Chemical Reagent Co. Ltd. They were of analytical grade and were used without further purification.

### 2.2 Synthesis of the $\text{Co}(\text{CO}_3)_{0.35}\text{Cl}_{0.2}(\text{OH})_{1.1}$ precursor

$\text{Co}(\text{CO}_3)_{0.35}\text{Cl}_{0.2}(\text{OH})_{1.1}$  can be easily prepared by mixing  $\text{CoCl}_2$  and urea.<sup>24</sup> In a typical synthesis, 1.1897 g (5 mmol) of  $\text{CoCl}_2 \cdot 6\text{H}_2\text{O}$  was first dissolved in 40 mL of deionized water under vigorous magnetic stirring and 300.3 mg (5 mmol) of urea was then added into the water solution. The obtained claret-red solution was transferred into a Teflon-lined steel autoclave and heated at 100 °C for 12 h in an electric oven. After the autoclave was cooled to room temperature, pink products were collected and washed with deionized water and ethanol three times and dried overnight at 80 °C.

### 2.3 Synthesis of the $\text{Co}_3\text{O}_4\text{-CeO}_2$ , pure $\text{Co}_3\text{O}_4$ and $\text{CeO}_2$

Different amounts of  $\text{Ce}(\text{NO}_3)_3$  were dissolved in 30 mL of deionized water under stirring. Then the as-prepared  $\text{Co}(\text{CO}_3)_{0.35}\text{Cl}_{0.2}(\text{OH})_{1.1}$  was added into the above clear solutions and kept still for 2 days. Finally, the samples were thoroughly washed with ultrapure water, dried, followed by a calcination at 550 °C for 4 h.

Pure  $\text{Co}_3\text{O}_4$  was synthesized from direct thermal decomposition of  $\text{Co}(\text{CO}_3)_{0.35}\text{Cl}_{0.2}(\text{OH})_{1.1}$  at 550 °C for 4 h. In the case of pure  $\text{CeO}_2$ , it was prepared by mixing  $\text{Ce}(\text{NO}_3)_3$  and NaOH solution. Then the product was washed with deionized water, dried and calcined at 550 °C for 4 h.

### 2.4 Characterization

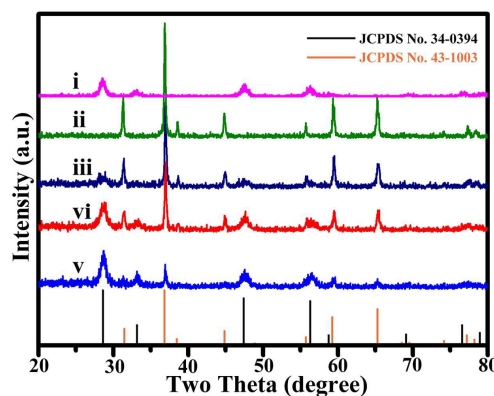
The phases and purity of the prepared samples were examined by X-ray powder diffraction (XRD) performed on a Rigaku D/Max- $\gamma$ A rotating anode X-ray diffractometer with Cu K $\alpha$  radiation ( $\lambda = 1.54178 \text{ \AA}$ ).  $\text{N}_2$  adsorption-desorption isotherms were measured at

77 K on a Micromeritics TriStar II 3020 surface area & pore size analyzer. Before measurement, the samples were outgassed in a vacuum at 300 °C for 4 h. The Brunauer-Emmett-Teller (BET) method was used to calculate the surface areas of the samples. The morphology and structure of samples were observed by transmission electron microscopy (TEM, JEM 2100) equipped with an energy dispersive X-ray spectrometer (EDS). X-ray photoelectron spectrometer (XPS) spectra were recorded by a PHI 5300 X-ray photoelectron spectrometer with Al K $\alpha$  radiation. Micro-Raman measurement was conducted with an LabRAM HR Evolution High Resolution Raman Spectrometer.  $\text{H}_2$ -Temperature-programmed reduction ( $\text{H}_2\text{-TPR}$ ) experiment was performed with a thermal conductivity detector on 50 mg sample using a gas mixture composed of 95% (molar) argon and 5% (molar) hydrogen at a flow rate of 30 mL  $\text{min}^{-1}$ . The temperature ramping rate was set to be 10 K  $\text{min}^{-1}$ . The  $\text{CeO}_2$  contents in  $\text{Co}_3\text{O}_4\text{-CeO}_2$  samples were determined using inductively coupled plasma mass spectrometer (ICP-MS, Thermo Scientific XSeries-2).

### 2.5 Catalyst Test

Catalytic activity was tested using a continuous flow fixed-bed micro-reactor at atmospheric pressure. In a typical experiment, the system was first purged with high purity  $\text{N}_2$  gas and then a gas mixture (1% CO, 10%  $\text{O}_2$ , 89%  $\text{N}_2$ ) passed through the reactor at a flow rate of 50 mL/min, corresponding to a space velocity of 60 000  $\text{mL} \cdot \text{h}^{-1} \cdot \text{g}^{-1}$  of catalyst. Composition of the gas exiting from the reactor was analyzed with an online infrared gas analyzer (Gasboard-3100, China Wuhan Cubic Co.), which can simultaneously detect CO,  $\text{CO}_2$  and  $\text{O}_2$ .

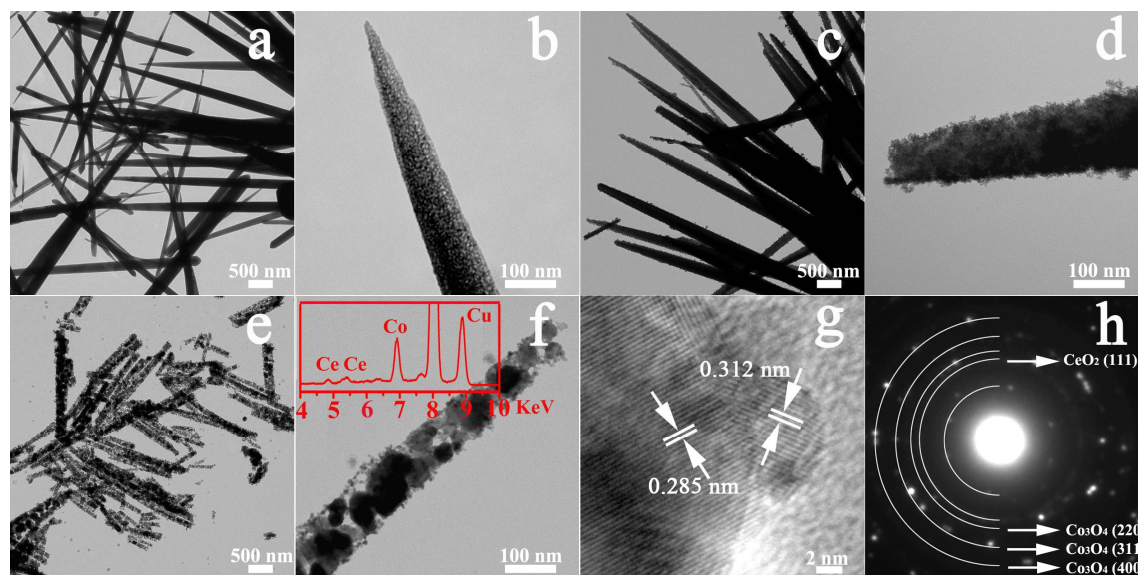
## 3. Results and discussion



**Fig. 1** XRD patterns of pure  $\text{CeO}_2$  (i),  $\text{Co}_3\text{O}_4$  (ii), and  $\text{Co}_3\text{O}_4\text{-CeO}_2$  bimetal oxide catalysts with different  $\text{CeO}_2$  compositions: 18.5 wt% (iii), 41.7 wt% (iv) and 63.4 wt% (v).

After interfacial reaction and heat-treatment at 550 °C, there are two sets of diffraction peaks, one set is corresponding to fluorite-phase  $\text{CeO}_2$  (JCPDS no.34-0394), and the other one is indexed to spinel-phase  $\text{Co}_3\text{O}_4$  (JCPDS no.43-1003) (Fig. 1), suggesting that  $\text{Co}_3\text{O}_4\text{-CeO}_2$  composite can be successfully synthesized through an interfacial reaction of  $\text{Co}(\text{CO}_3)_{0.35}\text{Cl}_{0.2}(\text{OH})_{1.1}$  precursor and  $\text{Ce}^{3+}$  followed by a thermal-treatment. Moreover, the relative intensity of  $\text{CeO}_2$  diffractions is gradually enhanced as the increase of the Ce/Co

mole ratio, suggesting that the  $\text{CeO}_2$  content is tunable by simply varying the  $\text{Ce}^{3+}$  concentration during the interfacial reaction.

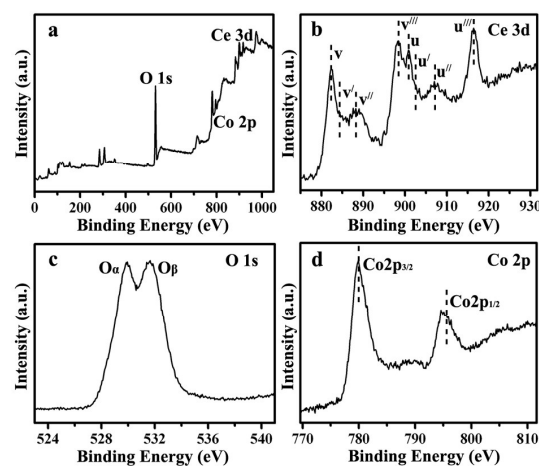


**Fig. 2** TEM images of the  $\text{Co}(\text{CO}_3)_{0.35}\text{Cl}_{0.2}(\text{OH})_{1.1}$  precursor (a, b), the sample obtained by reacting Co precursor with  $\text{Ce}^{3+}$  ions before (c, d) and after calcination (e-g), and SAED pattern (h) of the  $\text{Co}_3\text{O}_4\text{-CeO}_2\text{-18.5 wt\%}$  bimetal oxide catalyst. Inset in (f) is the EDS result detected from the rod in (f).

The morphologies of the  $\text{Co}(\text{CO}_3)_{0.35}\text{Cl}_{0.2}(\text{OH})_{1.1}$  precursor and the  $\text{Co}_3\text{O}_4\text{-CeO}_2$  bimetal oxide are characterized by TEM. As shown in Fig. 2a, the Co precursor is characteristic of one-dimensional (1D) rod (or needle) structure. The rod with a smooth surface is of several micrometers in length and average 80 nm in width (Fig. 2b). After dispersing Co precursor into the  $\text{Ce}(\text{NO}_3)_3$  solution, there are some dispersed nanoparticles (NPs) depositing on the Co precursor surface, as shown in Fig. 2c,d. Upon calcination, the one-dimensional structure is still maintained although the length of the rods becomes short. Higher magnification TEM image of the rod demonstrates that unlike the structure prior to the interfacial reaction, at this stage the rod turns less compact and is constructed by interconnected NPs surrounded by a NP-string-like thin shell (Fig. 2e,f). The co-existence of Ce and Co in the as-prepared sample is also supported by Energy Dispersive Spectroscopy (EDS) measurement (inset of Fig. 2f). Fig. 2g shows the HR-TEM image of the interface of the core-shell structure, constituted with interfused NPs. The lattice spacings of 0.312 nm and 0.285 nm correspond well to the (111) crystal plane of the fluorite-structured  $\text{CeO}_2$  and the (220) plane of the spinel-phase of  $\text{Co}_3\text{O}_4$ , respectively. These results regarding crystalline structure are consistent with that obtained from selected area electron diffraction (SAED) pattern shown in Fig. 2h, where two sets of diffraction rings belonging to the fluorite-phase  $\text{CeO}_2$ , and spinel-phase  $\text{Co}_3\text{O}_4$  can be recognized.

The chemical bonding states of the core-shell structure are further investigated by XPS analysis. Fig. 3 shows the XPS spectra of the typical  $\text{Co}_3\text{O}_4\text{-CeO}_2$  sample with  $\text{CeO}_2$  content of  $\sim 18.5\%$ . The coexistence of Co and Ce can be clearly observed from the wide spectrum (Fig. 3a). Fig. 3b presents the high resolution XPS spectrum of Ce 3d, where the peaks can be assigned to four pairs of spin-orbit doublets: the doublets ( $v \sim 882.05$  eV,  $u \sim 900.45$  eV), ( $u' \sim 906.97$  eV,  $v' \sim 888.26$  eV), and ( $u'' \sim 916.34$  eV,  $v'' \sim 897.94$  eV) are characteristic of  $\text{Ce}^{4+}$  state, whereas the doublet ( $u' \sim 902.19$  eV and  $v' \sim 884.34$  eV) is indicative of  $\text{Ce}^{3+}$  state, indicating the existence of  $\text{Ce}^{3+}$  species in the as-prepared  $\text{Co}_3\text{O}_4\text{-CeO}_2$  sample. In the case of O 1s XPS spectrum (Fig. 3c), two identical peaks are clearly displayed, indicating the presence of multi oxygen species. The peak at the lower binding energy side ( $\text{O}_a$ : 529-530 eV) is ascribed to lattice oxygen and the shoulder peak at higher binding

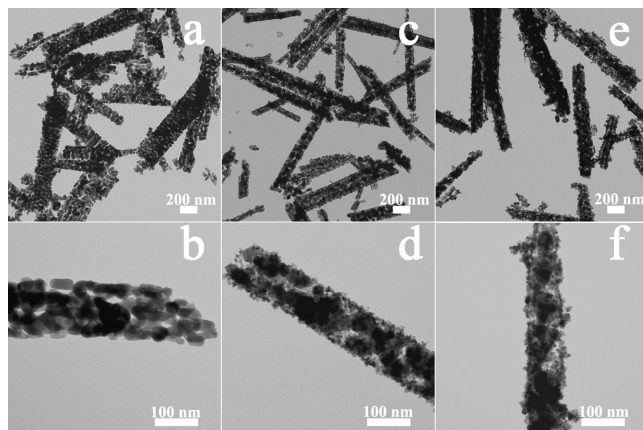
energy side ( $\text{O}_b$ : 531-532.8 eV) can be assigned to defective or adsorptive oxygen species. The Co 2p XPS spectrum in Fig. 3d



**Fig. 3** XPS wide spectrum (a), high resolution XPS spectra of Ce 3d (b), O 1s (c) and Co 2p(d) spectra of the  $\text{Co}_3\text{O}_4\text{-CeO}_2\text{-18.5 wt\%}$  bimetal oxide catalyst.

energy side ( $\text{O}_b$ : 531-532.8 eV) can be assigned to defective or adsorptive oxygen species. The Co 2p XPS spectrum in Fig. 3d

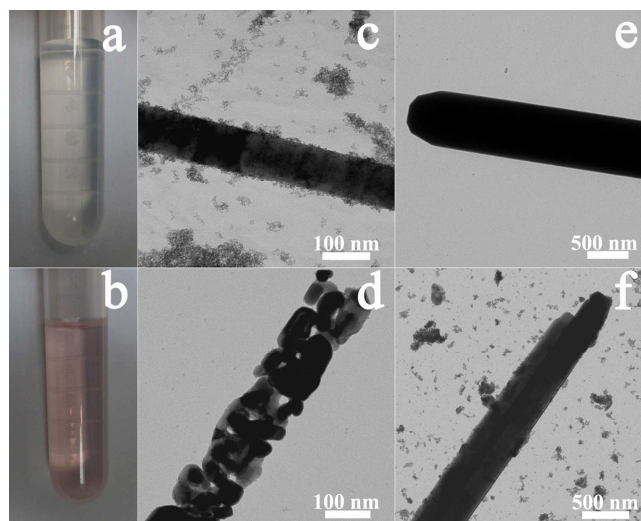
shows two major peaks at 795.5 and 780.4 eV which are corresponding to Co 2p<sub>1/2</sub> and Co 2p<sub>3/2</sub> spin-orbit signals, respectively. These results demonstrate that the surface of the Co<sub>3</sub>O<sub>4</sub>-CeO<sub>2</sub> core-shell sample is not fully covered with CeO<sub>2</sub> within the detection range of XPS.



**Fig. 4** TEM images of the pure Co<sub>3</sub>O<sub>4</sub> (a, b) derived from the thermal decomposition of Co(CO<sub>3</sub>)<sub>0.35</sub>Cl<sub>0.2</sub>(OH)<sub>1.1</sub> precursor, Co<sub>3</sub>O<sub>4</sub>-CeO<sub>2</sub>-41.7 wt%(c, d) and Co<sub>3</sub>O<sub>4</sub>-CeO<sub>2</sub>-63.4 wt% (e, f).

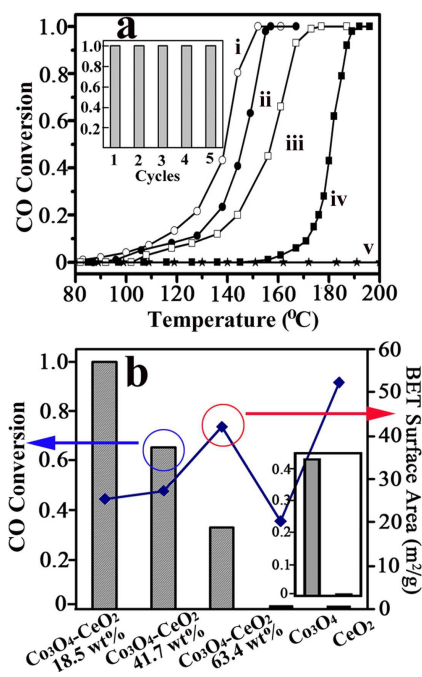
By simply changing the Ce<sup>3+</sup> concentration, a series of Co<sub>3</sub>O<sub>4</sub>-CeO<sub>2</sub> bimetal oxides with different surface coverage of CeO<sub>2</sub> are obtained. Fig. 4a, b shows the pure Co<sub>3</sub>O<sub>4</sub> nanorods derived from the thermal decomposition of Co(CO<sub>3</sub>)<sub>0.35</sub>Cl<sub>0.2</sub>(OH)<sub>1.1</sub>, which leads to the volume shrinkage and generates subsequent porosity structure throughout the rod (Fig. 4b). After interfacial reaction and heat-treatment, compared with that shown in Fig. 2f, more dense CeO<sub>2</sub> NPs are formed around the rod surface as the concentration of Ce(NO<sub>3</sub>)<sub>3</sub> increases (Fig. 4c-f). Specifically, a considerable surface coverage is reached for the sample with CeO<sub>2</sub> content of ~ 63.4 wt % (Co<sub>3</sub>O<sub>4</sub>-CeO<sub>2</sub>-63.4 wt %) as shown in Fig. 4e and f. In the Co<sub>3</sub>O<sub>4</sub>-CeO<sub>2</sub>-18.5 wt% and 41.7 wt% samples, the surface coverage by CeO<sub>2</sub> is less pronounced, where the shell is constructed by discontinuous CeO<sub>2</sub> NPs. In principle, this kind of structure enables more interfaces between Co<sub>3</sub>O<sub>4</sub> and CeO<sub>2</sub> to expose.

When the Co(CO<sub>3</sub>)<sub>0.35</sub>Cl<sub>0.2</sub>(OH)<sub>1.1</sub> precursor is added into Ce(NO<sub>3</sub>)<sub>3</sub> aqueous solution, Co<sup>2+</sup>, CO<sub>3</sub><sup>2-</sup>, Cl<sup>-</sup> and OH<sup>-</sup> ions are slowly dissociated simultaneously, and the released OH<sup>-</sup> ions prefer to combine with Ce<sup>3+</sup> to form Ce(OH)<sub>3</sub> due to the smaller solubility product constant of Ce(OH)<sub>3</sub> ( $K_{sp}=1.6\times 10^{-20}$ ). The newly produced Ce(OH)<sub>3</sub> nucleates *in situ* and grows on the surface of the Co(CO<sub>3</sub>)<sub>0.35</sub>Cl<sub>0.2</sub>(OH)<sub>1.1</sub> rods. Upon calcination, Ce(OH)<sub>3</sub> easily transferred into CeO<sub>2</sub>. Meanwhile, the residual Co(CO<sub>3</sub>)<sub>0.35</sub>Cl<sub>0.2</sub>(OH)<sub>1.1</sub> precursor is converted into Co<sub>3</sub>O<sub>4</sub> eventually to form Co<sub>3</sub>O<sub>4</sub>-CeO<sub>2</sub> bimetal oxide.



**Fig. 5** Photographs of the supernatant obtained from Co<sub>3</sub>O<sub>4</sub> rods (a) and Co precursor (b) dispersing into Ce<sup>3+</sup> aqueous solution; TEM images of Co precursor (c) and pure Co<sub>3</sub>O<sub>4</sub> (d) dispersed in Ce<sup>4+</sup> aqueous solution, and TEM images of Ce(OH)CO<sub>3</sub> precursor dispersed in Ce<sup>3+</sup> (e) and Ce<sup>4+</sup> (f) aqueous solution.

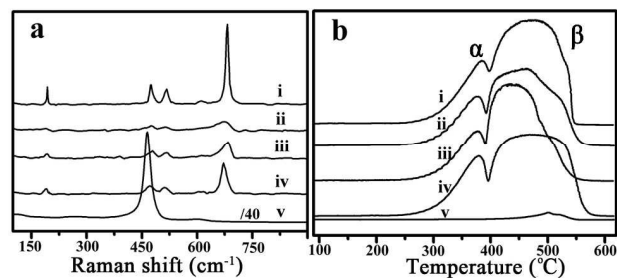
To further understand the transformation mechanism during the interfacial reaction, Co<sub>3</sub>O<sub>4</sub> rods derived from Co(CO<sub>3</sub>)<sub>0.35</sub>Cl<sub>0.2</sub>(OH)<sub>1.1</sub> are used as a Co precursor. Obviously, the supernatant still keeps colorless when Co<sub>3</sub>O<sub>4</sub> rods are dispersed in Ce<sup>3+</sup> aqueous solution even for more than 2 days (Fig. 5a). By stark contrast, light pink, characteristic color of Co<sup>2+</sup> ions, appears when Co(CO<sub>3</sub>)<sub>0.35</sub>Cl<sub>0.2</sub>(OH)<sub>1.1</sub> is dispersed in Ce<sup>3+</sup> aqueous solution for 2 days, as illustrated in Fig. 5b. In another case, (NH<sub>4</sub>)<sub>2</sub>Ce(NO<sub>3</sub>)<sub>6</sub> instead of Ce(NO<sub>3</sub>)<sub>3</sub> aqueous solution is mixed with Co(CO<sub>3</sub>)<sub>0.35</sub>Cl<sub>0.2</sub>(OH)<sub>1.1</sub> precursor, TEM result verifies that in addition to NPs depositing on the rod surface, a lot of “free” particles are individually dispersed in solution (Fig. 5c), possibly due to the smaller solubility product constant of Ce(OH)<sub>4</sub> ( $K_{sp}=2\times 10^{-48}$ ) than that of Ce(OH)<sub>3</sub> ( $K_{sp}=1.6\times 10^{-20}$ ). A crossover experiment was performed by dispersing Co<sub>3</sub>O<sub>4</sub> rods in (NH<sub>4</sub>)<sub>2</sub>Ce(NO<sub>3</sub>)<sub>6</sub> aqueous solution for 2 days, TEM analysis confirms that no small particles formed on the Co<sub>3</sub>O<sub>4</sub> rod surface (Fig. 5d). According to above phenomena, one can rule out the possibility that the formation of CeO<sub>2</sub> is due to the hydrolysis of Ce<sup>4+</sup>. Similarly, when Ce(OH)CO<sub>3</sub> nanorods instead of Co(CO<sub>3</sub>)<sub>0.35</sub>Cl<sub>0.2</sub>(OH)<sub>1.1</sub> is mixed with Ce<sup>3+</sup> and Ce<sup>4+</sup> aqueous solution, respectively, the Ce(OH)CO<sub>3</sub> nanorod surface is clean in the former case (Fig. 5e) while plenty of “free” NPs are observed in the latter case (Fig. 5f), because the solubility product constant of Ce(OH)<sub>4</sub> is far smaller than that of Ce(OH)<sub>3</sub>.



**Fig. 6** (a) CO conversion as a function of temperature for  $\text{Co}_3\text{O}_4$ - $\text{CeO}_2$  bimetal oxide catalysts with different  $\text{CeO}_2$  compositions: 18.5 wt% (i) 41.7 wt% (ii) 63.4 wt% (iii), and pure  $\text{Co}_3\text{O}_4$  (iv),  $\text{CeO}_2$  (v). (b) The comparisons in catalytic activity and BET surface area for the tested samples in (a). Inset in (a) shows the stability of  $\text{Co}_3\text{O}_4$ - $\text{CeO}_2$ -18.5 wt% sample for catalytic oxidation under the conditions of (a).

Herein, CO catalytic oxidation, being a model reaction, was carried out to evaluate the catalytic performance of the obtained  $\text{Co}_3\text{O}_4$ - $\text{CeO}_2$  core-shell materials. As for comparison, the catalytic activities of pure  $\text{Co}_3\text{O}_4$  rods derived from the thermal decomposition of  $\text{Co}(\text{CO}_3)_{0.35}\text{Cl}_{0.2}(\text{OH})_{1.1}$  precursor, and pure  $\text{CeO}_2$  prepared by precipitation method were also examined. Their light-off curves (CO conversion rate vs. temperature) for CO oxidation are presented. The temperature for 50% of CO conversion ( $T_{50}$ ) is an important parameter to assess the activity of the catalyst. As shown in Fig. 6a, pure  $\text{Co}_3\text{O}_4$  shows the  $T_{50}$  at  $\sim 180$  °C, while for the  $\text{Co}_3\text{O}_4$ - $\text{CeO}_2$  core-shell samples with different Co/Ce ratios, they exhibit higher catalytic activity than that of either pure  $\text{Co}_3\text{O}_4$  or pure  $\text{CeO}_2$ . Specifically, the one with lowest content of  $\text{CeO}_2$ ,  $\text{Co}_3\text{O}_4$ - $\text{CeO}_2$ -18.5 wt %, exhibits the superior catalytic activity with the  $T_{50}$  at  $\sim 139$  °C among all samples compared. Instead, the enrichment of  $\text{CeO}_2$  induces the decrease of the catalytic activity, reflected as the increase of  $T_{50}$  for samples of  $\text{Co}_3\text{O}_4$ - $\text{CeO}_2$ -41.7 wt% ( $T_{50} = 146$  °C) and  $\text{Co}_3\text{O}_4$ - $\text{CeO}_2$ -63.4 wt% ( $T_{50} = 157$  °C). The pure  $\text{CeO}_2$  sample does not show observable activity even at 200 °C. To evaluate the durability of the as-prepared  $\text{Co}_3\text{O}_4$ - $\text{CeO}_2$  catalyst, the  $\text{Co}_3\text{O}_4$ - $\text{CeO}_2$ -18.5 wt% sample was selected as a typical example for successive CO oxidation test under the same experimental conditions. As shown in the inset of Fig. 6a, after five cycles up to 150 °C, the as-prepared  $\text{Co}_3\text{O}_4$ - $\text{CeO}_2$  catalyst still maintained almost 100% conversion rate of CO. Generally, the catalytic performance is closely related to the surface area of the catalysts. However, BET

surface area analysis testified that no clear correlation between surface area and catalytic activity in our case, as shown in Fig. 6b. The pure  $\text{CeO}_2$  possesses the largest surface area while displays the lowest catalytic activity. Even for the  $\text{Co}_3\text{O}_4$ - $\text{CeO}_2$  samples prepared from same procedure, the catalytic activity is independent on the surface area. These observations strongly suggest that the formation of the  $\text{Co}_3\text{O}_4$ - $\text{CeO}_2$  bimetal oxide structure catalyst introduces distinctly different and significant synergistic effect in the catalytic CO oxidation.



**Fig. 7** Raman spectra (a) and  $\text{H}_2$ -TPR profiles (b) of pure  $\text{Co}_3\text{O}_4$  (i),  $\text{Co}_3\text{O}_4$ - $\text{CeO}_2$ -63.4 wt% (ii),  $\text{Co}_3\text{O}_4$ - $\text{CeO}_2$ -41.7 wt% (iii),  $\text{Co}_3\text{O}_4$ - $\text{CeO}_2$ -18.5 wt% (iv), and pure  $\text{CeO}_2$  (v).

It is widely recognized that the interaction between  $\text{Co}_3\text{O}_4$  and  $\text{CeO}_2$  is responsible for enhanced catalytic activity toward CO oxidation.<sup>24,30</sup> To investigate the surface chemistry feature of all samples, Raman spectroscopy characterization was performed. As illustrated in Fig. 7a of the Raman spectra, the band at  $\sim 466$   $\text{cm}^{-1}$  is related to the triply degenerated  $\text{F}_{2g}$  mode of fluorite  $\text{CeO}_2$ , and the bands at  $\sim 192$ ,  $\sim 475$ ,  $\sim 515$ ,  $\sim 614$  and  $\sim 682$   $\text{cm}^{-1}$  can be assigned to the vibrations of spinel  $\text{Co}_3\text{O}_4$ .<sup>35</sup> Interestingly, the peak intensity is strong at  $\sim 466$   $\text{cm}^{-1}$  for pure  $\text{CeO}_2$ , while nearly no obvious active Raman band can be detected at this position for all  $\text{Co}_3\text{O}_4$ - $\text{CeO}_2$  catalysts, indicating that the  $\text{Co}_3\text{O}_4$  highly likely enables the closely contacted  $\text{CeO}_2$  NPs to be structurally deformed.<sup>36,37</sup>  $\text{H}_2$ -TPR analysis was also conducted to study the interaction between  $\text{Co}_3\text{O}_4$  and  $\text{CeO}_2$ . As shown in Fig. 7b, pure  $\text{Co}_3\text{O}_4$  exhibits two reduction peaks (indexed as  $\alpha$ ,  $\beta$ ), corresponding to the reductions of  $\text{Co}_3\text{O}_4$  to  $\text{CoO}$  and  $\text{CoO}$  to metallic  $\text{Co}$ , respectively. In the case of  $\text{Co}_3\text{O}_4$ - $\text{CeO}_2$  catalysts, their reduction peaks ( $\alpha$ ) slightly shift to lower temperature in comparison to the pure  $\text{Co}_3\text{O}_4$ , further demonstrating a synergistic effect between  $\text{Co}_3\text{O}_4$  and  $\text{CeO}_2$ .<sup>30</sup>

During CO catalytic oxidation, it is consensus that the whole process mainly involves the adsorption and desorption of gas molecules on the surface/or at the interface of the catalyst. In the case of  $\text{Co}_3\text{O}_4$ - $\text{CeO}_2$  structures, the synergistic interaction between  $\text{Co}_3\text{O}_4$  and  $\text{CeO}_2$  could modify  $\text{Ce}^{3+}/\text{Ce}^{4+}$  and  $\text{Co}^{2+}/\text{Co}^{3+}$  redox cycles and increases oxygen mobility. In addition, the oxygen vacancies, widely existed in  $\text{CeO}_2$ , could increase the adsorption amount and capacity of  $\text{Co}_3\text{O}_4$ - $\text{CeO}_2$  for oxygen molecules, and promote the dissociation of  $\text{O}_2$  into  $\text{O}_{\text{ads}}$ , thus facilitating the CO oxidation. However, when the  $\text{CeO}_2$  species increases to certain extent, e.g.  $\text{Co}_3\text{O}_4$ - $\text{CeO}_2$ -63.4 wt% catalyst, it may significantly cover the  $\text{Co}_3\text{O}_4$  surface, thus blocking the pore channel and causing the decrease of catalytic activity.<sup>28,38</sup>

## 4. Conclusions

In summary,  $\text{Co}_3\text{O}_4\text{-CeO}_2$  core-shell catalysts were successfully synthesized by a facilely interfacial reaction between  $\text{Co}(\text{CO}_3)_{0.35}\text{Cl}_{0.2}(\text{OH})_{1.1}$  nanorods and  $\text{Ce}^{3+}$  aqueous solution, followed by a calcination step. The  $\text{OH}^-$  ions dissociated from  $\text{Co}(\text{CO}_3)_{0.35}\text{Cl}_{0.2}(\text{OH})_{1.1}$  combine with  $\text{Ce}^{3+}$  to form  $\text{Ce}(\text{OH})_3$ , which can be easily convert into  $\text{CeO}_2$  upon a calcination process. The small solubility product constant of  $\text{Ce}(\text{OH})_3$  is responsible for the heterogeneous nucleation of  $\text{CeO}_2$  NPs onto the Co precursor surface, even in the absence of any surfactants or polymers. Moreover, the coverage degree of  $\text{CeO}_2$  onto the  $\text{Co}_3\text{O}_4$  rod surface can be tuned by changing the concentration of  $\text{Ce}^{3+}$  aqueous solution. The as-prepared  $\text{Co}_3\text{O}_4\text{-CeO}_2$  core-shell catalysts display higher catalytic activity toward CO oxidation in comparison to either pure  $\text{CeO}_2$  or  $\text{Co}_3\text{O}_4$ , indicating a strong synergistic effect between two components. This work also demonstrates the feasibility of designing core-shell structure by virtue of the utilization of solubility product constant difference, and also expands the strategies of preparing binary oxide catalysts by means of interfacial reactions.

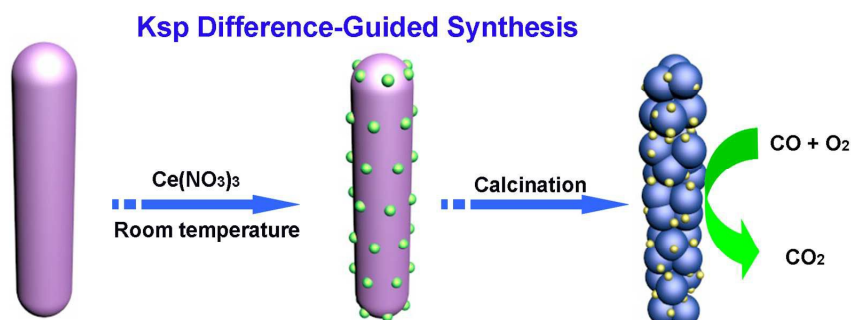
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## Graphical Abstract



Ion exchange procedure, one kind of solid-liquid interfacial reactions, is so far restricted to chalcogenide based core-shell nanomaterials, in contrast, the bimetal oxide with core-shell structures by means of ion exchange has not extensively studied yet. Herein, Co<sub>3</sub>O<sub>4</sub>-CeO<sub>2</sub> core-shell catalysts are successfully fabricated by an ion exchange procedure between Co(CO<sub>3</sub>)<sub>0.35</sub>Cl<sub>0.2</sub>(OH)<sub>1.1</sub> nanorods and Ce<sup>3+</sup> aqueous solution, followed by a calcination step.