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Graphical Abstract

The core-shell ternary $ZnS/Ni_3S_2@Ni$ composite exhibits superior electromagnetic wave absorption properties, which were attributed to synergistic effect between dielectric loss and magnetic loss, multiple interfacial polarization, good impedance match and unique urchin-like structure.



In-situ synthesis of novel urchin-like ZnS/Ni₃S₂@Ni composite with core-shell structure as an efficient electromagnetic absorption absorber

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ABSTRACT: The novel urchin-like ZnS/Ni₃S₂@Ni composite with core-shell structure was successfully synthesized by a facile twostage method. The structure and morphology are investigated by X-ray diffraction, scanning electron microscopy, transmission electron microscopy and energy dispersive spectrometry. The influence of reaction temperatures on the structure and morphology of ZnS/Ni₃S₂@Ni products were investigated by the aid of XRD and SEM techniques. A plausible formation mechanism of core-shell urchin-like architectures was proposed based on temperature dependent experiments. The electromagnetic absorption measurements show that the urchin-like ZnS/Ni₃S₂@Ni composite possesses the outstanding electromagnetic absorption properties compared with other ZnS/Ni₃S₂@Ni composites. The optimal reflection loss of -27.6 dB can be observed at 5.2 GHz and the effective absorption (below -10 dB, 90% electromagnetic absorption) bandwidth is 2.5 GHz (12.2-14.7 GHz) with the thin thickness of 1.0 mm. The location of minimal reflection loss can be tuned between 4.6 GHz and 18.0 GHz for the absorber with the thin thickness in 0.8-2.5 mm. The enhanced lectromagnetic absorption properties were attributed to synergistic effect between dielectric loss and magnetic loss, multiple interfacial polarization resulting from the heterogeneous structures of core-shell ternary ZnS/Ni₃S₂@Ni composite, good impedance match and unique urchin-like structure.

1. Introduction

The explosive development of information technology using electromagnetic waves in both civil and military fields leads to great concerns on electromagnetic interference (EMI) issues, which is expected to consider a new type of challenge on information security and physical health nowadays.^{1, 2} Therefore, ²⁵ considerable attention has been devoted to exploring effective electromagnetic absorbing materials with broad bandwidth and highly efficient absorption properties. Electromagnetic absorbers, which is distinct from the conventional reflection principles in electromagnetic shielding, provide an alternative to eliminate the ³⁰ unwanted electromagnetic (EM) waves by converting the EM energy into thermal energy or dissipating the EM waves through interference.^{3, 4}

In recent years, great efforts have been done to exploit the composite with core-shell structure, which is suit for the ³⁵ electromagnetic absorption absorber. ^{5, 6} Ohlan et al. prepared core-shell structured poly (3,4-Ethylenedioxy Thiophene)-barium ferrite nanocomposites and the minimal reflection loss is -22.5 dB at 15 GHz. ⁷ Zhou et al. synthesized core-shell Fe₃O₄-poly(3, 4- ethylenedioxythiophene) microspheres and the minimum ⁴⁰ reflection loss reached approximated -30 dB with the thickness of

4 mm. Moreover, the electromagnetic absorption properties of the composites are significantly influenced by the $(EDOT)/(Fe_3O_4)$ ratio.8 Core-shell composites, Fe₃O₄@C, with 500 nm Fe₃O₄ microspheres as cores were prepared by Du and co-workers. The 45 thickness of carbon shell, from 20 to 70 nm, can be well controlled. In comparison with pure Fe₃O₄, coating Fe₃O₄ microspheres with carbon shells endowed the composites with enhanced electromagnetic absorption properties due to multiple relaxation processes in these core-shell heterogeneous system.9 50 Core-shell structured Fe₃O₄-polyaniline nanoparticles were synthesized by Sun et al. Compared with pristine Fe₃O₄, the coreshell structured composites show enhanced electromagnetic absorption with the optimal reflection loss of -35.1 dB observed at 16.7 GHz. The excellent electromagnetic properties are 55 ascribed to the dielectric resonance arising from the special magnetic nanoparticle-conducting polymer core-shell structure.¹⁰ Yang et al ¹¹ designed Silica coated mesoporous Fe (Fe@SiO₂) microcubes with high performance electromagnetic wave attenuation in comparison with Fe cubes. In our previous 60 literatures, the Ni based core-shell composites, such as Ni@SnO₂,¹² Ni/CuO,¹³ and Ni@Al₂O₃ ¹⁴ exhibit the enhanced

electromagnetic wave absorption properties compared with raw Ni microspheres.

From above reported results, the electromagnetic wave absorption properties of core-shell composites are superior to ⁵ these of individual materials. In these composites, magnetic materials act as cores and dielectric materials regard as shells, which can provide cooperative effect between dielectric loss and magnetic loss, thus enhancing electromagnetic absorption capabilities. Moreover, these core-shell composites are mainly ¹⁰ bimaterials. There are scarce reports about electromagnetic wave absorption properties of ternary core-shell composites.^{15, 16}

Nanomaterial properties are dependent on the size, shape, and dimensionality. Metal sulfide nanomaterials have attracted significant attention because of their excellent properties and 15 promising applications in electronic and optoelectronic devices. Nickel subsulfide and ZnS are of importance in the metal sulfide family ^{17,18} because of the their variety applications such as in lithium ion batteries, supercapacitors, dye-sensitized ^{19, 20} and charge transfer, anion exchange, electricity generation,^{21,22} 20 respectively. Due to the semiconductor properties of nickel subsulfide and ZnS, it can induce dipole and space charge polarizations when placed under the alternated electromagnetic field.^{23, 24} Moreover, nickel subsulfide (Ni₃S₂) are highly metallic compared to insulating oxide compounds, which can cause ²⁵ conductive loss when irradiated by electromagnetic waves.²⁵ The core-shell Ni@ZnS composites with the improvement of electromagnetic properties were reported by our earlier literatures.^{26, 27} However, to the best of our knowledge, the

electromagnetic wave absorption core-shell structured ternary ³⁰ ZnS/Ni₃S₂@Ni composite was hardly found in the published papers. In this work, a novel ZnS/Ni₃S₂@Ni composite with coreshell structure was successfully synthesized. The urchin-like core-shell ZnS/Ni₃S₂@Ni composite obtained at 120 °C exhibits the excellent electromagnetic absorption and the absorption ³⁵ mechanism of such unique hierarchical microstructure was also discussed in detail.

2. Experimental section

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All the reagents are commercial available without any further purification. N₂H₄·H₂O (80%), ZnCl₂, and trisodium citrate were ⁴⁰ bought from Guangfu Chemical Co. Ltd. (Tianjin, China). Nickel chloride hexahydrate (NiCl₂·6H₂O), sodium hydroxide (NaOH), 1,2-isopropanol and sodium acetate were purchased from Xilong Chemical Reagent Co. Ltd. (Guangdong, China). Na₂S·9H₂O was provided by Kemiou Chemical Reagent Co. Ltd. (Tianjin, China) ⁴⁵ and Fengchuan Chemical Reagent Technologies Co. Ltd. (Tianjin, China), respectively. The preparation process of urchinlike ZnS/Ni₃S₂@Ni core-shell structured composites containing two steps: synthesis of Ni microspheres and subsequent growth of ZnS/Ni₃S₂ on the surfaces of Ni microspheres.

50 2.2 Preparation of monodispersed Ni microspheres

The monodipersed Ni microspheres were prepared according to our previous literatures.^{13, 14, 27} Simply, NiCl₂·6H₂O (1.2 g), trisodium citrate (0.3 g) and sodium acetate (3.0 g) were dissolved in the 60mL 1,2-isopropanol. Subsequently, N₂H₄·H₂O ⁵⁵ (6 mL) was added into above mixture before transferred into Teflon-lined autoclave. The sealed autoclave was heated at 140 °C for 15 h in an oven. The final products was washed with anhydrous ethanol and distilled water several times, and vacuumdried at 60 °C for 12 h.

60 2.3 Synthesis of urchin-like core-shell structured ZnS/Ni₃S₂@Ni composite

In brief, Ni (0.05 g) was dispersed in a mixture of aqueous solution of distilled water (30 mL) and ethanol (30 mL) containing 1.0 M NaOH. Then, 1 mmol ZnCl₂ and 2 mmol ⁶⁵ Na₂S·9H₂O were introduced into above mixture, respectively. The mixed solution was transferred to a Teflon-lined autoclave. The sealed autoclave was heated to 120 °C for 15 h. The products were washed in turn with distilled water and anhydrous ethanol, and then dried at 60 °C overnight in vacuum. To probe the ⁷⁰ possible formation mechanism of urchin-like core-shell structure and effects of temperatures on the morphologies of final products, the temperature-dependent experiments (60 °C, 80 °C , 100 °C and 120 °C) were also carried out in this work.

2.4 Characterization

The composition and phase purity of the as-received sample were analysed by an X-ray diffraction (XRD, XD-3, Beijing Purkinje General Instrument Co. Ltd.). The microscopic morphology and microstructure of the samples were observed ⁵ using transmission electron microscope (TEM, JEOL JSM-2100) and field-emission scanning electron microscope (FESEM, JSM-7001F) associated with an energy dispersive X-ray spectroscopy (EDS, Oxford Instruments). The electromagnetic parameters were measured on a vector network analyzer (Agilent, N5224A) in the ¹⁰ range of 1–18 GHz band at room temperature. The composite

samples for electromagnetic parameters measurements were prepared by evenly mixing the product with a paraffin wax in mass ratio of 1:1, and then being pressed into a ring with an outer diameter of 7.0 mm, inner diameter of 3.0 mm. Herein, the 15 paraffin serves as binder, which is transparent to electromagnetic

3. Results and discussion

wave.



Figure 1 XRD patterns of (a) pristine Ni microspheres and (b) as-20 prepared ZnS/Ni₃S₂@Ni.

The phase composition and structure of the as-prepared Ni microspheres and urchin-like ZnS/Ni₃S₂@Ni products were examined by XRD. Fig.1a shows the XRD pattern of pristine Ni microspheres, which can be well indexed to face-centred cubic ²⁵ structure of Ni (JCPDS No. 040850). As shown in Fig.1b, besides the diffraction peaks of Ni, the other diffraction peaks can be readily assigned to the zinc-blende ZnS (JCPDS Card no. 05-0566) and Ni₃S₂ (JCPDS Card no 76-1870). From these XRD patterns, it can be confirmed that the core-shell ZnS/Ni₃S₂@Ni ³⁰ composite consists of nickel, nickel sulfide and zinc sulfide. It can be inferred that nickel has functioned as the template for formation of Ni₃S₂ and deposition of ZnS.



Figure 2 FESEM images of (a) Ni microspheres, (b-c) urchin-³⁵ like core-shell structured ZnS/Ni₃S₂@Ni composite; (d) EDS profile of ZnS/Ni₃S₂@Ni composite.

To investigate the morphology of the products, FESEM images are taken for Ni microspheres and ZnS/Ni₃S₂@Ni composite and the corresponding results were shown in Fig.2. From the Fig.2a, it 40 can be seen that Ni products were composed of uniform distribution and smooth surface of microspheres with the average diameter of about 1 µm. Fig.2(b,c) show the different magnification FESEM images of core-shell ZnS/Ni₃S₂@Ni composite. Interestingly, from panoramic observation (Fig.2b), 45 urchin-like products were optionally grown on the surfaces of Ni microspheres and the outline of Ni spheres cannot be clearly observed due to the formation of ZnS/Ni₃S₂ on the surface of Ni particles. Notably, the diameter of Ni particles was also decreased, which indicates the consumption of Ni products. The 50 length and diameter of thorns were about 1 µm and 50 nm, respectively. Close observation from the high magnification FESEM image (Fig.2c), the presence of crinkled products

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encircle Ni particles and the crinkled products are expected to be links between Ni particles and thorns. Fig. 2d presents the EDS spectra of the ZnS/Ni₃S₂@Ni composite. The EDS spectrum reveals the presence of elements of S, Zn and Ni. Pt peaks are ⁵ also seen in the EDS curve because the SEM sample was prepared by sputtering of platinum onto the sample. Moreover, the high magnification FESEM images and corresponding elemental mapping were shown in Fig. S1. From the Fig.S1(a,b), it is noteworthy that the ZnS/Ni₃S₂@Ni composites possess the ¹⁰ features of urchin-like structure structure. The element mappings of Ni, Zn, and S are presented in Fig.S1(c–e), respectively. Interestingly, element Zn shows the larger area in the rim region than that in the center, which indicates that the ZnS/ Ni₃S₂ urchins were grown on Ni particles to form core-shell structure.



Figure 3 (a,b) TEM image of core-shell structured $ZnS/Ni_3S_2@Ni$ composite; (c) HRTEM image recorded on the B1 area in the Fig.3b, (d) HRTEM image recorded on the B2 area in the Fig.3b.

Detailed structural information about the ZnS/Ni₃S₂@Ni composite was further analysed by TEM and HRTEM. As exhibited in Fig. 3 (a,b), the TEM image of the ZnS/Ni₃S₂@Ni composite shows that Ni microspheres have a coarse outer layer and core-shell structure of composite can be clearly observed due ²⁵ to different contrast of cores and shells. Notably, the crinkled

products and thorn-like products are both grown on the surfaces of Ni microspheres, which is in accordance with the observation of above FESEM images (Fig.2b,c). Fig. 3 (c,d) shows a highresolution TEM (HRTEM) image recorded from the rim regions 30 of core-shell composite, respectively (as seen in Fig. 3b). From the Fig.3c, the HRTEM image derived from the B1 area in the Fig.3b indicates that the spacing of the lattice is 0.286 nm and 0.312 nm, which correspond to the spacing of (110) and (111) planes of Ni₃S₂ and ZnS, respectively. Based on these results, it $_{35}$ can be confirmed that the shell products are comprised of Ni₃S₂ and ZnS materials. Furthermore, the HRTEM image recorded from the B2 region in Fig.3b was shown in Fig.3d. The lattice fringe spacing of 0.287 nm and 0.311 nm were in accordance with Ni₃S₂ and ZnS, respectively. Notably, it can be found the ⁴⁰ Ni₃S₂ phase was located in the place which is close to the core of Ni spheres. Thus, the composite is ZnS/Ni₃S₂@Ni. The Ni₃S₂ stays away from the core of Ni spheres.

To study the influences of reaction temperature on the microstructure of final products, the temperature dependent ⁴⁵ experiments were carried out and the corresponding results were presented in Fig.(4,5). Fig.4 shows the XRD patterns of ZnS/Ni₃S₂@Ni composites prepared at various temperatures. All the diffraction peaks can be well attributed to the Ni, ZnS and Ni₃S₂ phases. No additional other peaks can be observed in the ⁵⁰ profiles, indicating the core-shell structured composites prepared at different temperatures were the mixture phases of ZnS, Ni₃S₂ and Ni. Interestingly, the relative intensity of Ni₃S₂ diffraction peaks was increased with increasing the reaction temperatures while the relative intensity of ZnS diffraction peaks clearly ⁵⁵ reduces, which indicates that the high crystallinity Ni₃S₂ products are supposed to generate at relatively high temperatures (120 °C)



Figure 4 XRD curves of $ZnS/Ni_3S_2@Ni$ composites prepared at different reaction temperatures: (a) 60 °C, (b) 80 °C, (c) 100 °C and 120 °C.



Figure 5 FESEM images of core-shell ZnS/Ni₃S₂@Ni composites prepared at various reaction temperatures: (a) 60 °C, (b) 80 °C, (c) 100 °C and (d) 120 °C.

It is assumed that the reaction temperature has an effect on ¹⁰ the morphology of core-shell heterstructure. At low temperature (60 °C), interestingly, there are plentiful waxberry-like products existed in Fig.5a. It is due to the fact that Ni microspheres were coated by wrinkle ZnS products. Meanwhile, combination of XRD patterns (Fig.4), Ni₃S₂ nanoparticles were in-situ formed by ¹⁵ consumption of Ni particles and then deposited on the surface of

remaining Ni products. With increasing the reaction temperatures

to high temperatures (80 °C), notably, some protuberant prickles were grown the surfaces of core-shell composites (Fig.5b). When the reaction temperature is 100 °C, more and strong protuberant ²⁰ prickles existed on the wrinkle surfaces of composite can be clearly observed (Fig.5c). With further increasing the temperature to 120 °C, the target urchin-like core-shell structured ZnS/Ni₃S₂@Ni composites were produced with a large number of thorns or rods grown on the rugged surfaces.



Figure 6 A plausible formation mechanism of urchin-like coreshell structured $ZnS/Ni_3S_2@Ni$ composites

Based on the temperature dependent experiments, a possible plausible formation mechanism of urchin-like core-shell 30 structured ZnS/Ni₃S₂@Ni composites was proposed (Fig.6). Firstly, when ZnCl₂ and Na₂S 9H₂O were introduced into this reaction process, it is not hard to understand that ZnS nanoparticles were yielded immediately generated from the reaction between ZnCl₂ and Na₂S due to the high solubility-³⁵ product constant (Ksp) of ZnS (Ksp =2.93×10⁻²⁵).^{21,22} According to our earlier reports,²⁶ the ZnS products are prone to form wrinkle and rough samples. Therefore, the Ni microspheres were coated by wrinkle ZnS at early stage. Secondly, because of high concentration of Na₂S (2 mmol), the unreacted Na₂S would be 40 reacted with Ni microspheres to produce Ni₃S₂ products. ¹⁷ Thirdly, the fresh Ni₃S₂ are expected to grow to thorns or rods.¹⁹ Thus, the urchin-like core-shell structured ZnS/Ni₃S₂@Ni composites were formed.



Figure 7 Frequency dependence of (a) real parts (ϵ') and (b) imaginary parts (ϵ'') of ZnS/Ni₃S₂@Ni composites prepared at various temperatures; (c) real parts (μ') and (b) imaginary parts (μ'') of ZnS/Ni₃S₂@Ni composites prepared at various temperatures.

Fig. 7 (a,b) exhibit the frequency dependencies of the real $_{5}(\varepsilon')$ and imaginary parts (ε'') of relative complex permittivity of the ZnS/Ni₃S₂@Ni composites prepared at different temperatures. It is well known that the ε' is vitally correlated with the amount of polarization happening in the material, and ε'' is associated to energy dissipation.^{8, 28} The performance of relative permittivity 10 was related with ionic, electronic, orientational, and space-charge polarization (interfacial polarization).^{14, 29} It can be observed that the real parts (ε') of ZnS/Ni₃S₂@Ni paraffin composites almost show the steady values with little fluctuation (Fig.7a). Notably, the ε' values of ZnS/Ni₃S₂@Ni composites prepared at 80 °C and 15 100 °C show the sharp peaks located at about 15-16 GHz. It may be due to the formation of unique fresh thorns, which can regard as links between Ni microspheres and wrinkle ZnS to form conductive gateway, leading to relatively high conductive. ^{30, 31} Fig.7b presents the imaginary parts (ε'') of four core-shell 20 ZnS/Ni₃S₂@Ni composites. Notably, it can be seen that urchin-

like $ZnS/Ni_3S_2@Ni$ composite shows the highest ϵ'' values among the four samples cover almost measured frequency, which indicates the highest dielectric loss for the urchin-like

 $ZnS/Ni_3S_2@Ni$ sample. Interestingly, the four $ZnS/Ni_3S_2@Ni$ 25 samples present the similar increase trend and multi-peaks on the ε'' values, meaning multiple resonances behavior. This resonance behavior is generally associated with highly conductive and electronic spin, skin effects,³² space charge polarizations, and polarized centers.²⁵ These phenomena are also clearly observed in 30 our earlier literatures. ^{13, 26, 27} The observed increase in ε'' with increasing frequency was assigned to the augment in interfacial polarization.³³ In a heterogeneous system, the accumulation of space charges at the interface of two materials will result in interfacial polarization and is known as Maxwell-Wagner 35 polarization. ^{34, 35} Herein, plentiful heterogeneous interfaces in ZnS/Ni₃S₂@Ni composite urchin-like core-shell make contribution to interfacial polarization. According to the free electron theory, ${}^{36} \varepsilon'' \approx 1/\pi \varepsilon_0 \rho f$, where ρ is the resistivity, high ε'' indicates high conductivity. As shown in Fig.8b, urchin-like 40 core-shell ZnS/Ni₃S₂@Ni composite exhibit high conductivity at the nearly whole frequency. Notably, ZnS/Ni₃S₂@Ni composites prepared at 80 °C and 100 °C show the relatively high conductivity at about 15-16 GHz, which is accordance with above

discussion of real parts (ε'). For the urchin-like core-shell ZnS/Ni₃S₂@Ni sample, the wrinkle ZnS and similar metal Ni₃S₂ thorns on the surfaces of Ni spheres can connect with each other and a continuous microcurrent gradually produces, which will ⁵ cause low electric resistivity and induce electric and conduction loss.

The real part (μ') and the imaginary part (μ'') of complex permeability of the ZnS/Ni₃S₂@Ni composites as a function of frequency from 1 to 18 GHz are shown in Fig. 7(c,d). The μ' ¹⁰ values show a decrease trend in the 1–18 GHz frequency range (Fig.7c). Moreover, from the Fig.7d, μ'' values present three broad peaks in the measured frequency. These multi-resonance peaks of μ'' can be attributed to the natural resonance at low frequency and 'exchange mode' resonance at high frequency, ³⁷,

^{15 38} which may be associated with the small size effect and shape anisotropy. In general, the contribution to magnetic loss contains domain-wall displacement, magnetic hysteresis, eddy current and natural resonance.²³ Herein, the magnetic loss was mainly attributed to natural resonance loss due to the weak applied ²⁰ electromagnetic field.³⁹ It is expected that the natural resonance will result in strong magnetic loss abilities in low frequency, while the interfacial polarization takes place in high frequency. This complementary effect is helpful for enhancement of electromagnetic absorption properties.



Figure 8 Attenuation constant of four core-shell ZnS/Ni₃S₂@Ni paraffin-composites versus frequency

Nevertheless, it is difficult to judge the electromagnetic absorption properties just relying on the electromagnetic ³⁰ parameters (impedance match), which let the electromagnetic wave go through the interior of absorber as much as possible. In fact, if we want to obtain the enhanced electromagnetic absorption properties, another important factor should also be considered, namely electromagnetic wave attenuation (α). The ³⁵ attenuation constant α which determines the attenuation properties of absorbing materials, which can be expressed by following equation ^{6,40}

$$\alpha = \frac{\sqrt{2\pi f}}{c} \times \sqrt{\left(\mu''\varepsilon'' - \mu'\varepsilon'\right) + \sqrt{\left(\mu''\varepsilon'' - \mu'\varepsilon'\right)^2 + \left(\mu'\varepsilon'' + \mu''\varepsilon'\right)^2}}$$
(1)

In which f and c are the frequency and velocity of light, ⁴⁰ respectively. Fig. 8 exhibits the dependence of the attenuation constant α of ZnS/Ni₃S₂@Ni paraffin-composites in the frequency range of 1–18 GHz. It can be noteworthy that the urchin-like ZnS/Ni₃S₂@Ni prepared at 120 °C shows the largest attenuation constant in the almost whole measured frequency ⁴⁵ range, which indicates its excellent electromagnetic absorption.



Figure 9 The calculated reflection loss of core-shell ZnS/Ni₃S₂@Ni composites prepared at various temperatures: (a) 60 ° C, (b) 80 ° C, (c) 100 ° C and (d) 120 ° C with different thickness in the frequency range of 1-18 GHz.

To reveal the electromagnetic wave absorption properties, ⁵ the representation reflection loss (RL, which can be used to judge the electromagnetic absorption abilities of absorber, the lower the value, the stronger the absorption) of the core-shell ZnS/Ni₃S₂@Ni paraffin-composites with different sample thicknesses are simulated based on the measured complex ¹⁰ permeability ($\mu_r = \mu' - j\mu''$) and complex permittivity ($\varepsilon_r = \varepsilon' - j\varepsilon''$) according to the transmit line theory, which can be described as the following equations: ^{11, 41, 42}

$$RL = 20\log_{10} |(Z_{in} - Z_0)/(Z_{in} + Z_0)|$$
⁽²⁾

$$Z_{in} = Z_0 \sqrt{\frac{\mu_r}{\varepsilon_r}} \tanh\left(j \frac{2\pi f d \sqrt{\mu_r \varepsilon_r}}{c}\right)$$
(3)

¹⁵ In which Z_{in} is the input characteristic impedance, Z_0 is the impedance of free space, f is the frequency, C is the velocity of light, and d is the absorber thickness. Fig.9 shows the reflection loss (RL) of the core-shell ZnS/Ni₃S₂@Ni composites prepared at different temperatures with thickness varies from 0.8

to 2.5 mm in the frequency range of 1-18 GHz. It can be found that urchin-like ZnS/Ni₃S₂@Ni composite exhibits outstanding electromagnetic absorption. The minimal RL is down to -27.6 dB at 5.2 GHz as the thickness is 2.5 mm. Notably, the reflection loss of urchin-like ZnS/Ni₃S₂@Ni can reach -21.6 dB at 13.3 GHz
when the absorber thickness is only 1.0 mm, and the effective bandwidth (RL less than -10 dB) is 2.5 GHz (12.2-14.7 GHz), which is superior to those of other reported dielectric/magnetic composites, such as Fe@SnO₂(-10.1dB),⁴³ and Co/CoO(-14 dB).⁴⁴ In fact, most of similar absorbing materials ^{13, 27, 43.49} are ³⁰ hardly obtain outstanding microwave absorption at the identical 1.0 mm, as listed in the Table 1. At the same time, the attenuation peaks would shift to lower frequencies and two RL peaks could be observed with the increasing thickness bigger than 2.5 mm. It can be attributed to the quarter-wavelength cancellation model. ⁵⁰

35	Table 1 The reflection loss data of some similar samples.					
	Sample	Minimum RL value	thickness (mm)	Frequency range	Ref.	
		(dB)		(RL<-10 dB)		

(GHz)

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Ni/SnO ₂	-18.6	7.0	13.8–15.3	[13]
PS@PPy@Ni	-20.06	2.0	9.16-13.75	[45]
Ni/Polypyrrole	-15.2	2.0	11–15.4	[46]
Ni/ZnS	-20.06	2.7	11.52–16.24	[27]
Fe/SnO ₂	>-10	1.5	None	[43]
Co/CoO	-14.5	2.5	12.24–15.68	[44]
Fe/graphene	-15	1.5	16.0-18.0	[47]
α-Fe ₂ O ₃ @CoFe ₂ O ₄	-15.2	1.5	14.8-18.0	[48]
MnO2@Fe- graphene	-17.5	1.5	14-18	[49]
ZnS/Ni ₃ S ₂ @Ni	-21.6	1.0	12.2-14.7	present



Figure 10 Schematic illustration of plausible electromagnetic wave absorption mechanism for the core-shell $ZnS/Ni_3S_2@Ni$ heterogeneous system.

- $_5$ Based on above analysis, a plausible electromagnetic wave absorption mechanism for the core-shell ZnS/Ni_3S_2@Ni heterogeneous system was proposed, as depicted in Fig.10. When the ZnS/Ni_3S_2@Ni composite is subjected to electromagnetic wave radiation, the Ni_3S_2 thorns grown on the surfaces of Ni
- ¹⁰ microspheres are expected as antenna receiver to allow electromagnetic waves penetrate into interior of absorber as much as possible, namely called good impedance match. ^{33, 51} Moreover, electromagnetic absorption is also understood from the viewpoint of Ohmic heating induced by an alternating magnetic
- ¹⁵ field, ⁵² in which ZnS/Ni₃S₂@Ni composite with relatively high electric conductivity from high ε" values. Thirdly, Due to the heterogeneous systems of core-shell ZnS/Ni₃S₂@Ni composites, the multi-interfaces between the ZnS, Ni₃S₂, and Ni are beneficial for enhancement of electromagnetic wave absorption because of ²⁰ the interaction of electromagnetic radiation with charged
- multipoles at the interfaces. ^{53, 54} The interfacial polarization ^{34, 55}

happening at the interfaces is attributed to the migration of charge carriers through different dielectric properties of the composite material, which gives rise to charge accumulation at the ²⁵ interfaces. During the activation of an electromagnetic wave, an additional interfacial relaxation is generated, which is helpful for the electromagnetic absorption.⁵⁶ Finally, the synergistic effect between dielectric loss at high frequency and magnetic loss at low frequency also makes contribution to the improvement of ³⁰ electromagnetic absorption.⁵⁷

4. Conclusion

In summary, novel and interesting urchin-like ZnS/Ni₃S₂@Ni composites were synthesized through a two-step process including solution reduction and subsequent template method. 35 The morphologies of core-shell ZnS/Ni₃S₂@Ni composites were crucially determined by the reaction temperatures. Based on serials of contrast experiments, a possible formation mechanism of urchin-like core-shell ZnS/Ni₃S₂@Ni composite was proposed. Different ZnS/Ni₃S₂@Ni composites prepared at different 40 temperatures show variable electromagnetc absorption responses, in which the urchin-like ZnS/Ni₃S₂@Ni obtained at 120 °C displays the best electromagnetic absorption performance due to its promising dielectric loss and magnetic loss, improved matched characteristic impedance, as well as unique urchin-like structure. 45 Multiple dielectric resonances originating from effective accumulation of various polarizations in the urchin-like structure are considered to be responsible for the enhancement of electromagnetic absorption. It is believed that the in-situ synthesis of core-shell ZnS/Ni₃S₂@Ni composites may open up a 50 new avenue for the design and preparation of novel microwave absorbers with promising application potential.

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