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**Copper Phosphide Modified Cadmium Sulfide Nanorods as a Novel p-n Heterojunction for Highly Efficient Visible-Light-Driven Hydrogen Production in Water** 

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Developing efficient photocatalysts made of earth-abundant elements for hydrogen (H<sub>2</sub>) production from water is considered to be a key pathway for future clean energy supply. Herein we give the first report that p-type copper phosphide (Cu<sub>3</sub>P) can be an efficient promoter to improve photocatalytic H<sub>2</sub> production from water when loaded on n-type cadmium sulphide nanorods (CdS NRs). The formation of a p-n junction in Cu<sub>3</sub>P/CdS NRs leads to fast charge transfer and enhanced photocatalytic activity under visible light irradiation. Under optimal conditions, the H<sub>2</sub> evolution rate was as high as ~200 µmol h<sup>-1</sup> mg<sup>-1</sup> ( $\lambda$  > 420 nm) and the apparent quantum yield at  $\lambda$  = 450 nm was ~25% in water.

Due to serious environmental issues and the increasing clean energy demand, developing new energy sources to substitute for fossil fuels is an urgent priority. Hydrogen is considered to be an ideal energy carrier for a future clean energy supply system because of its high energy capacity and environmental friendliness.<sup>1-2</sup> Since the discovery of hydrogen production through the photoelectrochemical water splitting on TiO<sub>2</sub> in early 1970s,<sup>3</sup> photocatalytic hydrogen production from water over semiconductors has attracted much attention during the past few decades.<sup>4-9</sup> The main challenge for its commercial application is to produce hydrogen efficiently using photocatalysts that have long-term stability and low cost.<sup>10-11</sup> To achieve hydrogen production under visible light, the photocatalysts should have a proper band gap for visible light absorption and enough redox potential for water splitting.<sup>12-13</sup>

Among all the visible-light-responsive photocatalysts, n-type semiconductor CdS appears to be one of the most promising materials for photocatalytic H<sub>2</sub> evolution due to its high activity under visible light (narrow band gap,  $E_g \sim 2.4 \text{ eV}$ ) and sufficiently negative flatband potential for proton reduction.<sup>14-17</sup> However, pure CdS is not very stable and exhibits low photocatalytic activity due to its fast electron-hole recombination and photocorrosion.<sup>18</sup> To solve these

problems, several approaches have been applied to functionalize CdS semiconductor, such as loading an appropriate cocatalyst. Cocatalysts, which can effectively enhance the transfer of photogenerated charge carriers, play a very important role in promoting the photocatalytic activity.<sup>19-21</sup> The most common and effective cocatalyst is Pt,<sup>15, 22</sup> and other noble-metal-based materials, such as Ru and Pd,<sup>23-24</sup> are also studied as cocatalysts owing to the physicochemical properties being similar to Pt. Unfortunately, the high price and scare supply of noble metals limit their large-scale application. An alternative that has been studied is cocatalysts based on non-noble metals, such as Mo,21, 25-27, Co,19, 28-31, and Ni.19, 32-33 Unhappily, the catalytic performances of these cocatalysts are generally worse than noble metals and therefore, it is necessary to develop more cost-effective and robust catalysts for H<sub>2</sub> production. Another approach to enhance the photocatalytic activity is to construct a p-n junction to retard the electron-hole recombination and improve the charge separation.

Cu<sub>3</sub>P, as a p-type semiconductor, was previously reported for application in lithium ion batteries.<sup>34-38</sup> However, the use of Cu<sub>3</sub>P for photocatalytic H<sub>2</sub> production has not received prior investigation. Herein we report the synthesis and evaluation of a novel and low-cost Cu<sub>3</sub>P/CdS NRs p-n heterojunction photocatalyst for highly efficient H<sub>2</sub> production from water. Under optimal conditions, the H<sub>2</sub> evolution rate using this Cu<sub>3</sub>P semiconductor was as high as ~200 µmol h<sup>-1</sup> mg<sup>-1</sup> ( $\lambda > 420$  nm) and the apparent quantum yield was ~25% in water with Na<sub>2</sub>S/Na<sub>2</sub>SO<sub>3</sub> as the electron donors ( $\lambda = 450$  nm). Furthermore, a photocatalytic mechanism based on the Cu<sub>3</sub>P/CdS NRs p-n junction is proposed and discussed.

Cu<sub>3</sub>P/CdS NRs were synthesized by a simple solvothermal method with the content ratio of Cu<sub>3</sub>P at 0 % - 2.90 wt %. The asprepared samples were labeled as CP1, CP2, CP3, and CP4, respectively. The copper ratio in each sample was measured by ICP-AES and the results are shown in Table S1. The Cu<sub>3</sub>P ratios were calculated based on the copper contents.

Figure 1a shows the powder XRD patterns of CdS, Cu<sub>3</sub>P/CdS, and Cu<sub>3</sub>P. The Cu<sub>3</sub>P/CdS and CdS samples show the peaks of  $2\theta$  at 24.8°, 26.5°, 28.2°, 36.7°, 43.7°, 47.9°, 51.9°, and 66.8°, which were

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indexed to the (100), (002), (101), (102), (110), (103), (112), and (203) lattice fringe diffractions of hexagonal CdS (PDF#77-2306), respectively. For comparison, the Cu<sub>3</sub>P shows peaks located at 36.0°, 39.0°, 41.5°, 45.1°, and 46.2°, which correspond respectively to the (112), (202), (211), (300), and (213) lattice diffractions of hexagonal Cu<sub>3</sub>P (PDF#71-2261). After loading Cu<sub>3</sub>P onto CdS, no obvious diffraction peaks belonging to Cu<sub>3</sub>P were observed and all reflections showed no significant difference from CdS, which might result from the low content of Cu<sub>3</sub>P in Cu<sub>3</sub>P/CdS samples and high dispersion of Cu<sub>3</sub>P nanomaterials deposited on the surface of CdS NRs.<sup>16, 39-40</sup> To identify the surface chemical composition and valence state of Cu<sub>3</sub>P/CdS NRs, the XPS spectra were measured (Figure 1b). The survey scan shows the existence of Cd, S, Cu, and P elements. No other elements were observed except C element as the reference and O element from the absorbed gaseous molecules.



Figure 1. (a) Powder XRD patterns of Cu<sub>3</sub>P/CdS samples containing different amount of Cu<sub>3</sub>P. (b) XPS spectrum, (c) SEM image, (d) TEM image of CP2 sample.

Figure 1c shows a low-magnification SEM image of Cu<sub>3</sub>P/CdS NRs (CP2 sample, Cu<sub>3</sub>P 0.44 wt %) with the diameter of ca.40-90 nm and length from 0.4  $\mu$ m to 2  $\mu$ m. The SEM image of pure CdS NRs is shown in Figure S1a for comparison, as well as the SEM image of Cu<sub>3</sub>P (Figure S1b). The slight increase of the diameter from CdS NRs to Cu<sub>3</sub>P/CdS NRs suggests that Cu<sub>3</sub>P has been successfully loaded on the CdS NRs. The TEM image shows an obvious core-shell structure of Cu<sub>3</sub>P/CdS NRs with a thin layer Cu<sub>3</sub>P (Figure 1d).

The HRTEM measurement can provide further insight into the crystalline morphologies and microstructural details of the core-shell nanostructure. Figure 2a shows the interfacial region of the Cu<sub>3</sub>P/CdS NRs, demonstrating that Cu<sub>3</sub>P nanoparticles were successfully loaded onto the surface of the single crystalline hexagonal CdS NRs. The crystalline planes for CdS can be recognized by the presence of the (002) lattice distance at 0.34 nm. Moreover, the lattice fringe distances of 0.22 nm and 0.25 nm can be assigned to the (211) and (112) planes of Cu<sub>3</sub>P, respectively. These results suggest that Cu<sub>3</sub>P nanoparticles are closely attached to the CdS nanorods, which could be highly beneficial to fast charge transfer between Cu<sub>3</sub>P and CdS materials. In addition, the selected-area electron diffraction (SAED) analysis showed ordered diffraction spots, revealing the single crystalline nature of the CdS NRs (Figure 2b).

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**Figure 2. (a)** HRTEM image, **(b)** Selected area electron diffraction (SAED) data of CP2 sample.

The EDX spectrum further confirmed the existence of Cd, S, Cu, and P, as shown in Figure 3a. Furthermore, the EDX elemental mapping images clearly show the distribution of Cd, S, Cu, and P elements (Figures 3b-3e). The observed Mo element is from the molybdenum grid substrate. The EDX mapping images of a single nanorod further reveal that both Cu and P elements are uniformly distributed in the whole nanorod. All these results clearly confirm the successful synthesis of core-shell Cu<sub>3</sub>P/CdS NRs by a facile solvothermal method.



Figure 3. (a) EDX spectrum of CP2. EDX-mapping of CP2: (b) bright field image, (c) Cu element mapping, (d) Cd element mapping, (e) S element mapping, and (f) P element mapping.

Photocatalytic hydrogen production experiments were carried out using the as-prepared Cu<sub>3</sub>P/CdS NRs photocatalysts in water under visible light irradiation ( $\lambda > 420$  nm). The experimental set up is shown in scheme S1. Figure 4a shows the rate of H<sub>2</sub> evolution using **Journal Name** 

Cu<sub>3</sub>P/CdS samples in the presence of Na<sub>2</sub>S and Na<sub>2</sub>SO<sub>3</sub> as the electron donors. It was found that an appropriate amount of Cu<sub>3</sub>P can significantly enhance the photocatalytic activity for H<sub>2</sub> production in water. The rate of H<sub>2</sub> evolution initially increased and then decreased with increasing amounts of Cu<sub>3</sub>P. With low content of Cu<sub>3</sub>P loading (0.03 wt %) on CdS (CP1), the H<sub>2</sub> evolution rate was ~96  $\mu$ mol h<sup>-1</sup> mg<sup>-1</sup>. The highest photocatalytic activity was achieved using CP2, presenting a H<sub>2</sub> evolution rate of ~184 µmol h<sup>-1</sup> mg<sup>-1</sup>. This phenomenon can be explained as follows: (1) excess Cu<sub>3</sub>P covered on the CdS surface probably inhibit the incident light absorption by CdS and then decrease the photogenerated electrons from the CdS NRs; (2) excess Cu<sub>3</sub>P could decrease the surface active sites of CdS.<sup>21, 40-41</sup> For comparison, in the absence of Cu<sub>3</sub>P, CdS showed very low activity, having a H<sub>2</sub> evolution rate of only  $\sim 28 \ \mu mol \ h^{-1} \ mg^{-1}$ . Although the mechanical mixture of Cu<sub>3</sub>P and CdS (0.44 wt % Cu<sub>3</sub>P) showed a slightly higher rate than CdS (~40 µmol h<sup>-1</sup> mg<sup>-1</sup>), it is still much less than the Cu<sub>3</sub>P/CdS samples prepared by the solvothermal method. In addition, the Cu<sub>3</sub>P alone showed no appreciable H<sub>2</sub> production with visible light irradiation, revealing that Cu<sub>3</sub>P is not an active photocatalyst.

The apparent quantum yield ( $\phi$ ) for photocatalytic H<sub>2</sub> production using CP2 photocatalyst was measured under 450 nm monochromatic light irradiation (Figure 4b). The result showed that the H<sub>2</sub> evolution rate reached ~88 µmol h<sup>-1</sup> mg<sup>-1</sup> with  $\phi$ ~25% after 8 hours of irradiation. The present efficient quantum yield from visible light energy into chemical energy indicated that Cu<sub>3</sub>P/CdS NRs is a highly efficient photocatalyst.



**Figure 4. (a)** Comparison of photocatalytic H<sub>2</sub> evolution rates of CdS, CP1, CP2, CP3, CP4, Cu<sub>3</sub>P, and mixed Cu<sub>3</sub>P-CdS samples at ambient temperature under visible light ( $\lambda > 420$  nm). The system contains 1.0 mg photocatalyst, 0.75 M Na<sub>2</sub>S, and 1.05 M Na<sub>2</sub>SO<sub>3</sub> in 20 mL deionized water. **(b)** Time courses of H<sub>2</sub> production and apparent quantum yield on CP2 under monochromatic 450 nm (±5 nm) light irradiation using 1.0 mg photocatalyst in a 20 mL aqueous solution containing 1.25 M Na<sub>2</sub>S and 1.75 M Na<sub>2</sub>SO<sub>3</sub>. **(c)** Cycling runs for photocatalytic hydrogen evolution in the presence of 1.0 mg CP2 photocatalyst in a 20 mL aqueous solution containing 1.25 M Na<sub>2</sub>S and 1.75 M Na<sub>2</sub>SO<sub>3</sub>. After every 3 hours, the evolved H<sub>2</sub> was evacuated. **(d)** Long-term evolution of H<sub>2</sub> under visible light irradiation ( $\lambda > 420$  nm) using CP2 photocatalyst. The system contains 1.0 mg photocatalyst, 1.25 M Na<sub>2</sub>SO<sub>3</sub> in 50 mL deionized water.

Besides its high activity, Cu<sub>3</sub>P/CdS NRs material exhibits good stability during photocatalysis. The reaction system was evacuated to

remove the evolved H<sub>2</sub> every 3 hours and the process was carried out repeatedly. As shown in Figure 4c, no significant decrease in the activity for photocatalytic H<sub>2</sub> production can be observed in 12 hours of cycling tests. Moreover, analysis of the photocatalytic efficiency after long-term operation was also performed, with the results shown in Figure 4d. After 50 hours of irradiation, a total of ~10 mmol H<sub>2</sub> was produced with a constant rate of ~200 µmol h<sup>-1</sup> mg<sup>-1</sup>. These results indicate the good stability of Cu<sub>3</sub>P/CdS NRs for photocatalytic H<sub>2</sub> production under visible light. In addition, the SEM image of Cu<sub>3</sub>P/CdS NRs had no significant change after photocatalysis. Figure S2 shows the SEM image of CP2 sample after 5 hours of irradiation. The image still has a rod-like structure, with nearly the same length and diameter as the sample before illumination, indicating that Cu<sub>3</sub>P/CdS NRs material is a robust photocatalyst in the present photocatalytic H<sub>2</sub> evolution system.



**Figure 5.** (a) The UV-vis diffuse reflectance spectra of and (b) the estimated band gap of the CdS, CP2, and Cu<sub>3</sub>P. (c) The steady-state photoluminescence (PL) spectra of the CdS and CP2. (d) The transient photocurrent responses of blank FTO, CdS, and CP2.

The optical properties of the pure CdS, Cu<sub>3</sub>P/CdS, and pure Cu<sub>3</sub>P were measured by using a UV-vis-NIR spectrophotometer. Figure 5a shows the UV-vis absorption spectra for all samples. The spectrum of pure CdS displayed a sharp edge at around 530 nm. The optical band gap energy (Eg) is estimated to be about 2.4 eV by the Tauc plot (Figure 5b),<sup>42</sup> which agrees well with the reported value.<sup>14, 43-44</sup> The absorption onset of Cu<sub>3</sub>P/CdS is also located at around 530 nm, probably originating from the absorption of CdS in Cu<sub>3</sub>P/CdS heterojunction photocatalysts. The same absorption edges of pure CdS and Cu<sub>3</sub>P/CdS. The increased absorption features of Cu<sub>3</sub>P/CdS in the region from 600 nm to 1200 nm should be ascribed to the contribution of Cu<sub>3</sub>P. The pure Cu<sub>3</sub>P shows the absorption edge at about 950 nm and the band gap energy (Eg) was estimated to be about 1.3 eV, which is consistent with the literature.<sup>34</sup>

In order to study the migration, transfer, and recombination processes of the photogenerated charge carriers over Cu<sub>3</sub>P/CdS, the photoluminescence (PL) spectra were examined. Figure 5c shows the PL spectra of CdS and Cu<sub>3</sub>P/CdS under excitation at 405 nm. It was observed that the as-prepared CdS displayed two distinct emission band at ~511 nm and ~713 nm, which is in agreement with previous studies.<sup>45-46</sup> The broad photoluminescence spectrum centered at ~713

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nm should be the trapped emission, which was mainly due to the surface defects.<sup>47-48</sup> Such surface defects would act as recombination centers for electron-hole pairs to lower the photocatalytic activity. The PL spectrum of Cu<sub>3</sub>P/CdS was similar to that of CdS. However, the emission intensity was much weaker than that of CdS, indicating the transfer of photogenerated charge carriers in Cu<sub>3</sub>P/CdS by forming p-n interfacial junctions.<sup>49</sup>

To further investigate the photocatalytic H<sub>2</sub> production mechanism over Cu<sub>3</sub>P/CdS, photocurrent tests were performed. As shown in Figure 5d, the Cu<sub>3</sub>P/CdS photocatalyst exhibits much higher photocurrent response than CdS NRs, revealing that the charge transfer process in Cu<sub>3</sub>P/CdS is more effective than pure CdS. The fast transfer of charge carriers can highly enhance photocatalytic activity for H<sub>2</sub> production according to the literature.<sup>19, 27, 50</sup>

Based on above results, a possible reaction mechanism for photocatalytic H<sub>2</sub> evolution using Cu<sub>3</sub>P/CdS heterojunction is proposed and the schematic diagram is illustrated in Scheme 1. Both Cu<sub>3</sub>P and CdS can absorb visible light to produce photoinduced electron-hole pairs. Since Cu<sub>3</sub>P is a p-type semiconductor<sup>34</sup> and CdS is an n-type material,<sup>12</sup> a large number of nanoscale p-n junctions could be generated when two such types of semiconductor materials are closely attached. It has been reported that the conduction band (CB) position of CdS (~-3.98 eV vs. AVS<sup>51</sup>) is lower than that of Cu<sub>3</sub>P (~- $3.8 \text{ eV vs. AVS}^{34}$ ). Therefore, the photoinduced electrons on the CB of Cu<sub>3</sub>P can directly transfer to the CB of CdS. The electrons, together with the electrons excited from the VB of CdS, would transfer to the surface of CdS to reduce  $H^+$  for  $H_2$  production. As for the valence band (VB), the corresponding VB position of CdS (~-6.38 eV vs. AVS) is also lower than that of Cu<sub>3</sub>P (~-5.1 eV vs. AVS). As a result, the photogenerated holes on the VB of CdS can migrate to the VB of Cu<sub>3</sub>P, which will greatly reduce the recombination process of electron-hole pairs and further lead to enhanced stability of the CdS and an improved H<sub>2</sub> evolution rate. Therefore, efficient charge separation was successfully achieved by the nanoscale p-n junctions in Cu<sub>3</sub>P/CdS. The close contact between CdS and Cu<sub>3</sub>P could facilitate the charge transfer, which would improve the photocatalytic H<sub>2</sub> evolution activity and stability.



Scheme 1. A possible reaction mechanism for photocatalytic H<sub>2</sub> evolution using the Cu<sub>3</sub>P/CdS heterojunction.

#### Conclusions

In conclusion,  $Cu_3P/CdS$  NRs core-shell heterostructures were successfully constructed by in situ deposition of  $Cu_3P$  onto CdS through a facile solvothermal method.  $Cu_3P$  nanomaterials were tightly attached to the surface of CdS, leading to the formation of p-n junctions between p-type  $Cu_3P$  and n-type CdS. The p-n junctions in Cu<sub>3</sub>P/CdS can promote fast charge transfer and reduce charge recombination. Under optimal conditions, the Cu<sub>3</sub>P/CdS exhibited a H<sub>2</sub> evolution rate of ~200  $\mu$ mol·h<sup>-1</sup>·mg<sup>-1</sup> with a maximum apparent quantum yield of ~25% when excited at 450 nm. The photocatalytic results indicate that Cu<sub>3</sub>P is a highly active promoter when loaded on CdS to form p-n junctions for H<sub>2</sub> production. This work not only supports the possibility of using cost-effective p-type Cu<sub>3</sub>P for photocatalytic H<sub>2</sub> production but also shows that a proper p-n junction structure is crucial for high photocatalytic activity in a hybrid photocatalyst.

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#### Notes and references

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<sup>†</sup> Electronic Supplementary Information (ESI) available: Experimental details, ICP-AES data of copper in Cu<sub>3</sub>P/CdS NRs, SEM images of CdS and Cu<sub>3</sub>P, and SEM image of Cu<sub>3</sub>P/CdS after 5 hours of visible light irradiation. See DOI: 10.1039/c000000x/

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#### TOC Figure

The present study shows that p-type copper phosphide ( $Cu_3P$ ) can be an efficient promoter to improve photocatalytic  $H_2$  production from water when loaded on n-type cadmium sulphide nanorods.

