Chemical Science

Accepted Manuscript



This is an *Accepted Manuscript*, which has been through the Royal Society of Chemistry peer review process and has been accepted for publication.

Accepted Manuscripts are published online shortly after acceptance, before technical editing, formatting and proof reading. Using this free service, authors can make their results available to the community, in citable form, before we publish the edited article. We will replace this Accepted Manuscript with the edited and formatted Advance Article as soon as it is available.

You can find more information about *Accepted Manuscripts* in the **Information for Authors**.

Please note that technical editing may introduce minor changes to the text and/or graphics, which may alter content. The journal's standard <u>Terms & Conditions</u> and the <u>Ethical guidelines</u> still apply. In no event shall the Royal Society of Chemistry be held responsible for any errors or omissions in this *Accepted Manuscript* or any consequences arising from the use of any information it contains.



www.rsc.org/chemicalscience

Chemical Science

ARTICLE

Received 00th January 20xx,

Accepted 00th January 20xx

DOI: 10.1039/x0xx00000x

www.rsc.org/



Phage Display Selected Magnetite Interacting Adhirons For Shape Controlled Nanoparticle Synthesis

Andrea E. Rawlings^a, Jonathan P. Bramble^a, Anna A.S. Tang^b, Lori A. Somner^a, Amy E. Monnington^c, David J. Cooke^c, Michael J. McPherson^b, Darren C. Tomlinson^b, Sarah S. Staniland^a

Adhirons are robust, well expressing, peptide display scaffold proteins, developed as an effective alternative to traditional antibody binding proteins for highly specific molecular recognition applications. This paper reports for the first time the use of these versatile proteins for material binding, and as tools for controlling material synthesis on the nanoscale. A phage library of Adhirons, each displaying two variable binding loops, was screened to identify specific proteins able to interact with [100] faces of cubic magnetite nanoparticles. The selected variable regions display a strong preference for basic residues such as lysine. Molecular dynamics simulations of amino acid adsorption onto a [100] magnetite surface provides a rationale for these interactions, with the lowest adsorption energy observed with lysine. These proteins direct the shape of the forming nanoparticles towards a cubic morphology in room temperature magnetite precipitation reactions, in stark contrast to the high temperature, harsh reaction conditions currently used to produce cubic nanoparticles. These effects demonstrate the utility of the selected Adhirons as novel magnetite mineralization control agents using ambient aqueous conditions. The approach we outline with artificial protein scaffolds has the potential to develop additives wider into toolkit of novel for nanomaterial fabrication. а

Introduction

Magnetic nanoparticles (MNP) have become an area of intense research interest thanks to their wide ranging applications from high density magnetic data storage, biomedical diagnostics, and therapies ^{1, 2}. Current manufacture of MNP with sufficiently high uniformity required for the demands of these applications typically requires high temperatures and expensive reagents². Therefore the development of a synthetic methodology is needed that offers mild reaction conditions and a high degree of molecular control over the type of mineral formed, the size, and the morphology. Such alternative approaches for MNP production have looked to nature where magnetotactic bacteria are able to synthesise highly uniform crystals of magnetite, often achieving nanoarchitectures that are unobtainable under any current synthetic method. The bacteria use proteins which are able to strictly regulate the nucleation of the magnetite nanoparticle and then direct its growth along certain crystal faces. Proteins from these bacteria, such as Mms6 and MmsF, can be isolated and shown to exert control over particle production during in vitro magnetite precipitation reactions ³⁻⁵. Being able to utilise proteins to regulate mineral formation in synthetic reactions offers a possible route to the precision manufacture of MNP with a narrow size distribution and uniform morphology. However, the use of these proteins limits the synthetic particle production to the characteristics of those found in nature, and the proteins found in the magnetosomes are often difficult and expensive to produce, purify, and therefore utilise in these reactions, making them ill-suited to industrial MNP manufacture.

Taking inspiration from these, and similar biomineralisation proteins, the process of biopanning has proven useful in uncovering novel peptides that can interact with specific mineral types ^{6, 7}. In this process, diverse libraries of random peptide sequences are expressed on the surface of bacterial cells (cell display) or bacteriophage (phage display) before application to a substrate.⁸ Through successive rounds o peptide binding, substrate washing, peptide elution and sequence amplification, peptides which show affinity to the substrate are selectively enriched. Binding peptides from these methods have now been identified for a range of materials and have been used for tethering of small molecules and enzymes, patterning surfaces, and also controlling the formation of materials^{9, 10}. This latter use has allowed the synthesis of nanoparticles with characteristics specified by the interacting peptides. Well studied systems include peptides which demonstrate affinity for metals such as gold¹¹⁻¹³, platinum¹⁴⁻¹⁶ palladium¹⁷ and titanium¹⁸ as well as oxides such as silica^{19, 20}, guartz²¹ and titania^{21, 22}.

^{a.} Department of Chemistry, The University of Sheffield, Sheffield, UK.

^b Faculty of Biological Sciences, The University of Leeds, Leeds, UK.
^c Chemical and Biological Sciences, University of Huddersfield, Huddersfield, UK.
Electronic Supplementary Information (ESI) available: [Detailed experimental methods, supporting experimental data, and details of the molecular dynamics study]. See DOI: 10.1039/x00x0000x



Page 2 of 10

ARTICLE

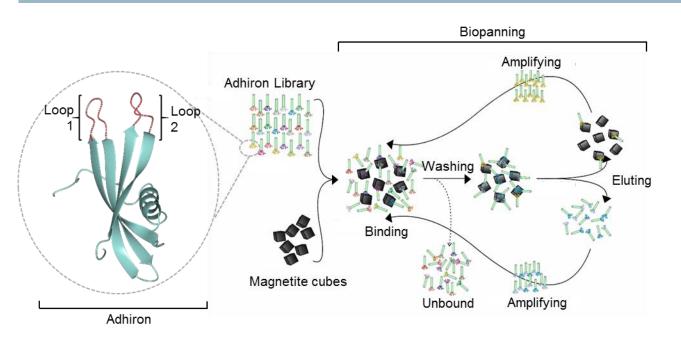


Figure 1. Overview of the magnetite interacting Adhiron selection process. A cartoon representation of the Adhiron scaffold is shown in green with the two variable loop regions depicted in red. Structure model of MIA-1 was generated using the Phyre2²³ server and the image produced in CCP4mg²⁴.

Recent advances have seen the development of material binding peptides which are constrained into cysteine bridged $\mathsf{loops}^{\mathsf{15, 25}}$ or by transposing the binding sequence into an alternative scaffold protein such as a protein $cage^{26}$. Constraining the binding peptides appears to alter the binding mode to the material it is selected for, offering different binding affinities and effects by limiting the conformational freedom of the peptide. By constraining peptide binders within a larger protein a number of benefits arise in that the system becomes suitable for large scale biological production, and is also easily amenable to genetic manipulation, thereby allowing the protein to be tailored for different applications. One scaffold protein capable of displaying variable binding loops is the recently developed Adhiron protein (commercially known as Affimers), Figure 1²⁷. Based upon a consensus sequence of the phytocystatin fold ²⁸, the Adhiron is engineered as a robust framework for the stable display of two variable nine amino acid long loop regions on the protein surface. Adhirons provide an ideal 'protein scaffold' as it is extremely robust and can be produced cost effectively in high quantities in bacterial expression systems. Here we report for the first time the selection by phage display of a suite of Magnetite Interacting Adhirons, MIA, which feature variable binding loops selected in situ for their affinity to cubic magnetite nanoparticles. These particles have specific planes, denoted [100], to which the MIA bind. These proteins possess the ability to alter the

characteristics of magnetite mineralisation products in favour of cubic nanoparticles

Results

Magnetite Interacting Adhirons selected via phage display

The Adhiron scaffold is a well expressing protein that has a melting temperature in excess of 100 °C 27 making it wel suited to exploitation in synthetic mineralisation reactions. The Adhiron library of over 1.3x10¹⁰ different sequences is fused to a truncated pIII minor coat protein of M13 bacteriophage²⁷ The phage library was exposed to clean cubic magnetite nanoparticles in the presence of a casein based blocking buffer. We anticipated that the highly charged magnetite surface would show a weak affinity to a wide range of proteins with compatible patches of surface charge, thereby diluting the eluted phage pool with large numbers of weak, nonspecific binding sequences which have potentially hindered previous studies²⁹. By including the blocking protein in excess we aimed to mitigate this by selecting binders which were able to successfully compete for binding to the magnetite rather than the casein.

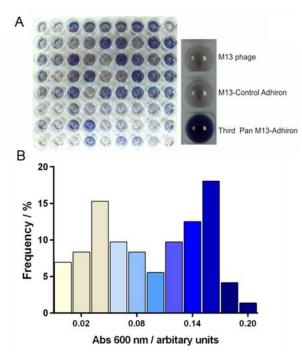


Figure 2. Phage ELISA of magnetite selected Adhirons. (A) Phage ELISA results where a blue colour-change indicates positive binding. The right panel shows ELISA controls of M13 phage, non-magnetite selected phage, and the selected phage pool after the final panning round. The left panel shows the range of binding of 72 clones from the final phage pool. (B) Frequency distribution showing the percentage of clones displaying a certain absorbance in the AP-Phage ELISA. Bar colouring corresponds to that generated in the assay.

Potentially the strength of association between some of the proteins and magnetite might mean that not all phage would be successfully eluted with the standard pH wash regime (acid then alkali), which has resulted in loss of the highest affinity library members in previous studies ³⁰. However, after each round of selection, in addition to the eluted phage, the remaining MNP with any residual phage attached were also used to directly infect *E. coli* cells. This approach was adopted to try and maximise the recovery of binding sequences carried through for

phage amplification and subsequent selection rounds despite the fact that the phage that remained associated with MNP may be sterically blocked from infecting. After three cycles of selection and amplification the phage pool from each panning round was assayed for magnetite binding using an alkaline phosphatase (AP) linked phage ELISA. The signal intensity increased with each panning round, showing that the phage pool was becoming increasingly enriched with magnetite binding proteins. As shown in Figure 2, M13 phage alone or with non-magnetite selected Adhirons showed no significant binding to the MNP in the same assay.

After the third and final panning round 72 individual library members were selected. These were amplified and assayed for binding to magnetite nanoparticles using an AP-phage ELISA. From the spectrum of intensities that we observed in the ELISA shown on Figure 2, we deduced an apparent range of different binding capabilities were present within this subset. Generation of a frequency distribution showed that there were two dominant groupings of lower and higher affinity binders.

DNA sequencing reveals a preference for basic residues

The phagemid vectors bearing the Adhiron coding sequences of the 48 clones ranked highest in terms of signal intensity inthe phage ELISA were extracted and sequenced. Alignment of the sequences revealed that approximately one third contained only the first of the variable binding loops, with the second loop completely absent. Previous detailed sequencing of the Adhiron library revealed that few library members were missing this binding loop²⁷, suggesting that the panning process has selectively enriched binding sequences which lack the second loop. We surmise that loop 1 may be the primary site of the magnetite interaction and the lack of a neighbouring second loop may promote improved binding The remaining two thirds of sequences showed the expected two binding loops. We analysed the binding loop sequences with a weighted Kullback Leibler logo plot, shown in Figure 3.

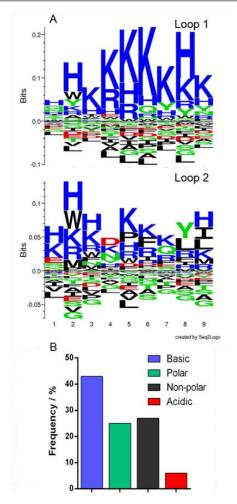


Figure 3. Sequence analysis of selected binding loops. (A) Seq2logo Kullback-Leibler plots³¹ of the amino acid sequences from the two binding loops. Residues in the positive area of the graph are enriched in the loops, and those in the negative area are depleted. (B) Frequency distribution of residue type in both binding loops. This revealed a strong preference for basic residues and in particular a predominance of lysine (26.4% of residues) and histidine (9.3%) in loop 1, and an almost total depletion of acidic amino acids. This is in contrast to the starting library which featured an approximately even distribution of amino acids²⁷. In loop 2 the sequences show a very similar but less stringent preference for basic residues with some hydrophobic and acidic residues reasonably well represented. This also suggests that loop 1 may be the principal magnetite interacting loop due to its more highly selective amino acid preference. The average isoelectric point, pl, of the loops also illustrates this with loop 1 being 10.12 and loop 2 being overall less basic, 8.74. The preference for basic residues (typically arginine) has been previously observed in cell display selected iron oxide binding peptides³².

Purification of Magnetite Interacting Adhirons

The coding regions of both the highest ranked (600 nm intensity) and median ranked MIA from the phage ELISA, were cloned into expression vectors encoding a C-terminal StrepII tag³³ with a final terminating cysteine residue for later gold surface attachment. This plasmid allows high level expression of the selected MIA proteins, referred to from here on in as MIA-1 and MIA-2. A control Adhiron was also introduced into the expression vector. This sequence was not selected during the MNP panning rounds and should not therefore show specific interaction with magnetite. The sequences are provided in Table1.

Table 1: Binding loops sequences of selected Adhirons					
Adhiron	Loop1	Loop2	Protein pl		
MIA-1	QKFVPKSTN	PKKSKIELK	9.6		
MIA-2	IKKKKKYKY	ETLTHKVIR	9.7		
Control	DWWEAGVFM	WNEINYMFD	5.5		

*Calculated using the ProtParam tool based on the complete Adhiron sequence.

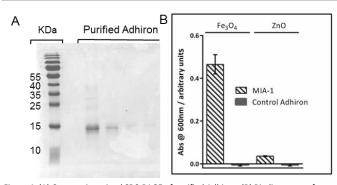


Figure 4. (A) Coomassie stained SDS-PAGE of purified Adhiron. (B) Binding assay of purified Adhirons with magnetite nanoparticle or zinc oxide nanoparticles.

Both MIA-1 and MIA-2, and the control Adhiron, were produced in E. coli and purified initially with Streptactin resin using the attached StrepII tag. SDS-PAGE analysis demonstrated high yield and good purity, figure 4A. However, the UV absorbance of MIA-1 and MIA-2 revealed a 260nm/280nm ratio of approximately 1.8. A ratio of greater than 1 is a common indicator that the protein sample also comprises DNA/RNA³⁴. This contamination appeared to be specific to MIA as the control-Adhiron was unaffected (ratio of 0.8) despite an identical purification procedure. We suspect that the high numbers of lysine residues in the variable loop regions were binding to endogenous cellular nucleic acids. Poly-lysine is a known binder of DNA³⁵ and is well characterised³⁶. Intriguingly this indicates that Adhirons could potentially be raised against specific nucleic acid sequences for molecular biology applications. To mitigate this we applied the proteins to a Heparin resin which mimics the sugar phosphate backbone of DNA and has been used previously for DNA. binding protein purification³⁷. After recovery from the Heparin resin via sodium chloride elution

the UV absorbance profile matched that of the control-Adhiron, confirming that nucleic acids had been successfully removed.

Assessing the specificity of interaction

To ascertain if the purified protein were capable of discriminating between different metal oxides we adapted the AP-Phage ELISA used earlier. In this modified assay the phage were substituted for the purified Adhiron and a primary antibody used which recognised the StrepII sequence of the purification tag. Equal amounts of magnetite and zinc oxide nanoparticles were screened for binding. The assay clearly showed that MIA-1 was able to bind to magnetite as demonstrated by the intense blue colour change, figure 4B. MIA-1 displayed a much weaker binding interaction with zinc oxide with a reduced absorbance at 600 nm. In both cases the control Adhiron recorded no colour change during the course of the assay indicating no binding had occurred to either material.

Probing the interaction of MIA with magnetite nanoparticles Quartz Crystal Microbalance

Quartz Crystal Microbalance with Dissipation (QCM-D) systems have been used extensively to study interactions of biomolecules with surfaces³⁸. Within a QCM-D system a thin piezoelectric crystal material, in this case, quartz coated in gold, is excited to oscillate in a shear mode. The surface of the crystal oscillates in a shear motion at its resonant frequency. If matter is deposited on the surface, the resonant frequency shifts. In simple cases the sensor response is described by the

Chemical Science



ARTICLE

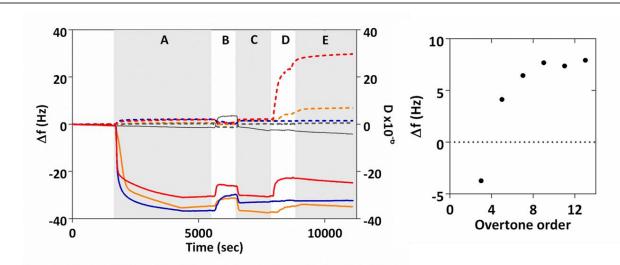


Figure 5. QCM analysis of MIA-nanoparticle interaction. Change in frequency, Δf, and Dissipation, D. Dotted lines represent dissipation and solid lines frequency. Red lines are MIA 1, gold lines Mia-2, blue lines control Adhiron, and grey lines blank experiments with no Adhiron. Phase A is injection of protein into the system, B and C are rinses with ultrapure water and PBS respectively. MNP are injected into the system during D, and Phase E is final rinsing with PBS. Time averaged frequency changes for MIA-1 at different overtone orders are also shown.

Sauerbrey equation which shows that the frequency shifts in a negative sense if the mass on the surface increases. More complex models use the oscillation decay time or dissipation which can be measured to quantify energy losses from visco-elastic layers.

The interaction of MIA-1 as well as the control-Adhiron with both gold surfaces and magnetite nanoparticles was investigated with QCM, figure 5. For comparison we have also included MNP interaction in the absence of any Adhiron. During phase A the proteins are injected into the system and the negative frequency shift (Δf) corresponds to the formation of a protein monolayer via cysteine attachment to the gold surface. At B, the samples are rinsed in ultra-pure (MilliQ) water causing a step change in the resonance position. After returning to PBS some of the adsorbed protein has been removed by the rinsing step, explained by the slight increase in the resonant frequency. Uncoated magnetite nanoparticles are injected into the system during D. A further decrease in Δf would indicate an increase in mass on the surface due to adsorption of nanoparticles to the MIA. If MIA do not interact with MNP we would expect to see no change in frequency with addition of particles.

However, in our experiments we observed a positive frequency shift for both MIA and no frequency change for the control Adhiron. The absence of a frequency change for the control Adhiron is suggestive of non-binding and is consistent with our ELISA results (figure 4B). The positive frequency shifts of the MIA-MNP experiment are clearly different to the lack of interaction observed with the control Adhiron which implies both MIA do indeed interact with the applied MNP. Interestingly, when MIA are applied to the QCM sensor surface with DNA occupying the binding site then we see no frequency shift with the addition of MNP (supplementary fig. 5). Taken together these results show that the positive Δf in figure 5 is due to an interaction between the binding loops and the nanoparticles.

Binding of colloidal particles and bacteria³⁹ in QCM have been shown to give rise to this effect which can be explained with a coupled-resonance model⁴⁰ whereby changes to the stiffness of the bond upon binding affect Δf , rather than simply the mass of the adsorbed material. In those cases the positive frequency shifts scale inversely with order of the overtone. We recorded the frequency shift caused by addition of MNP at different overtones and time averaged them after adsorption of MNP had ceased (phase E in figure 5). The average frequency shift was then plotted according to overtone order, n. ∆f clearly varies from negative at n=3 to positive for all higher overtone orders which is representative of a coupled resonance model³⁹ showing interaction between magnetite nanoparticles and surface bound MIA. Interestingly MIA-1 and MIA-2 appear to have different binding affinities, with MIA-1 displaying the larger resonance shift.

Modelling interactions between amino acids and the [100] magnetite surface

To complement our studies and to gain an insight into the interactions at the atomistic level we have performed a series of molecular dynamics (MD) simulations using the DL_POLY classic code⁴¹ to further understand the interaction of a magnetite surface with amino acids. In these simulations a selection of individual amino acids with capped termini were brought from an infinite distance away towards a magnetite [100] surface in order to calculate the adsorption energy in water. The [100] planes are found on cubic nanoparticles, which were the dominant nanoparticle form used in the Adhiron selection process. The MD simulations illustrate that the lysine sidechain is able to form strong hydrogen bonds with the surface as well as through the peptide carbonyl whereas glutamic acid is only able to interact through the side chain oxygen as shown in Figure 6.

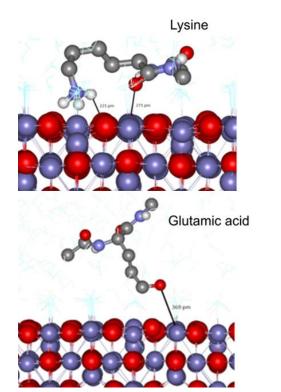


Figure 6. Simulated adsorption of amino acids on magnetite. A snap shot from the MD simulation of capped-lysine in the upper image, and capped-glutamic acid in the lower, on the [100] surface of magnetite. Hydrogen atoms bonded to carbon have been removed for clarity.

In both cases however, the zwitter-ion is considered, which indicates a likely pH dependence upon these adsorption interactions. The simulations reveal a clear trend with lysine showing the lowest adsorption energy (most negative), and acidic amino acids proving the least likely to interact with the surface, listed in Table 2. These data match extremely well with the amino acid preferences we observe in the variable loop regions of MIA.

Table 2: Comparison of simulated amino acid to magnetite adsorption energy with their occurrence in the binding loops

Amino acid	Adsorption energy (kJ/mol)	Frequency in binding loop (%)	
Lysine (K)	-52	26.4	
Arginine (R)	-45	7.0	
Leucine (L)	-31	3.9	
Glutamic acid (G)	-3	3.3	
Aspartic acid (D)	2	2.3	

MIA-1 can influence particle morphology in synthetic magnetite precipitation reactions

To ascertain the effect that MIA have upon magnetite nanoparticle synthesis we included MIA-1 and MIA-2 in nanoparticle formation reactions. In the room temperature coprecipitation reaction a mixture of ferrous and ferric ions are precipitated by the addition of hydroxide ions. MIA-1 was included at a ratio of 50 µg protein per 10 ml reaction volume. The particles were visualised by transmission electron microscopy (TEM) and compared to nanoparticles prepared in an identical process but without the addition of protein. Grain size analysis of over 500 nanoparticles from each batch revealed little difference in size, with modest improvements in homogeneity in the MIA-1-MNP compared to the control, shown in Figure 7D. However, we observed that the protein prepared particles had an overall more cubic and angular appearance when compared to the control particles, Figure 7A,B. This is significant as the MIA were enriched using affinity to magnetite cubes and previous studies of peptide selection to materials indicate that sequences which bind preferentially to a certain crystal face are able to stabilise the formation of that same face during material synthesis¹⁶. We therefore speculate that MIA-1 may show higher affinity for [100] faces and promote the formation of cubic particles. This effect was not observed for nanoparticles produced with either th€ addition of MIA-2 or the control Adhiron (supplementary figure 7). A partial oxidation of ferrous hydroxide reaction was also performed. This is a higher temperature reaction, which typically results in particles with an octahedral morphology. In these reactions we do not observe any effect from addition of MIA-1, with the control particles indistinguishable to those prepared with the protein (supplementary figure 6). It is likely that the extended high temperature incubation during these reactions causes denaturation of the Adhiron protein and ar accompanying loss of conformational integrity to the binding loops. This indicates that the conformation and structure of the binding loop could be necessary for functional activity of the protein. To test this hypothesis further we prepared MNP by the room temperature coprecipitation method with addition of free flexible peptides comprising the loop sequences of MIA-1. When supplied at an equivalent molar

Chemical Science



ARTICLE

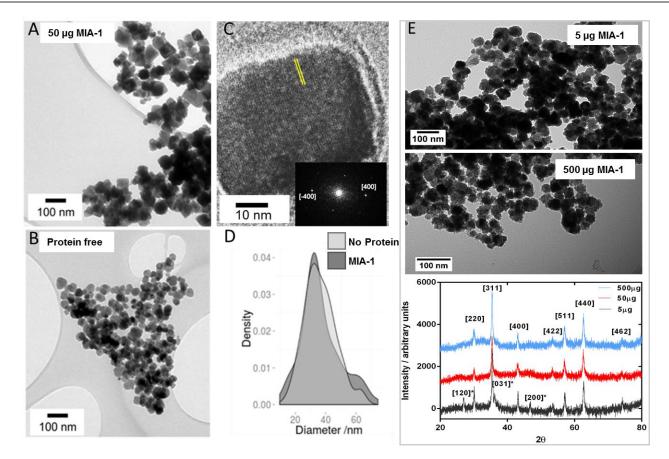


Figure 7. Transmission Electron Microscopy analysis of magnetite nanoparticles. (A) Particles prepared with MIA-1 at 50 µg per 10 ml reaction. (B) Particles prepared in the absence of MIA-1. (C) HRTEM analysis of a representative MIA-1 prepared MNP. A region of the lattice fringe is highlighted with yellow lines and the Fourier transform is shown inset with the [400] peaks identified. (D) Particle size analysis of MNP prepared with and without addition of MIA-1. (E)Transmission Electron Microscopy and X-ray diffraction analysis of magnetite nanoparticles at varying amounts of MIA-1. Powder XRD analysis of particles produced at 5 µg (grey), 50 µg (red) and 500 µg (blue) of MIA-1. Principal magnetite peaks have been assigned. Peaks present in the 5 µg sample assigned to lepidocrosite are denoted with an asterisk

ratio to the amount of MIA-1 we see no increased generation of cubic nanoparticles, and indeed the reaction appears to generate a more heterogeneous range of particles compared to even the Adhiron free particles (supplementary figure 7). This suggests that the constraint imposed by the Adhiron scaffold plays an important role in facilitating the activity of the binding loop sequences in cube production.

To confirm the crystal faces present on the nanoparticles synthesised with MIA-1 we performed high resolution TEM(HRTEM). Analysis of the lattice fringes, and the corresponding fourier transform, revealed a spacing of 0.20 nm which is consistent with the 0.20 nm lattice spacing of the [400] cubic plane of magnetite (Figure 7C).

We wanted to understand the effect that different concentrations of MIA-1 had upon nanoparticle formation. We

therefore used 5 μ g and 500 μ g of Adhiron per 10 mi magnetite coprecipitation (compared to the 50 μ g we had used previously) and analysed the products by both TEM and powder X-ray diffraction (XRD), figure 7E. Addition of 5 μ g produced particles, which appear similar to the control (protein free) nanoparticles, figure 7B. 500 μ g appeared to produce particles with a rough, uneven surface. This may be because the production of iron oxides is especially sensitive to changes in reaction conditions which may have been caused by the high amount of organic material present. We have estimated the amount of Adhiron required to produce an approximate monolayer coverage of the MNP produced in our experiments. Based on our calculations (supplementary data) we estimate that at 5 μ g the amount of Adhiron is insufficient to interact with the total surface area of particles during formation. 50 µg provides approximate monolayer coverage, and 500 µg provides a 10 fold excess of protein. This estimation matches well with our TEM data which indicates that 50 µg of Adhiron produces the largest effect. XRD analysis, Figure 7E, confirms that the samples are magnetite, however interestingly the MNP produced in the presence of 5 µg amount of MIA-1 reveals peaks assigned to the iron oxide hydroxide, lepidocrosite γ -FeO(OH). These peaks are not observed in samples produced with higher quantities of protein, suggesting that the addition of MIA-1 helps to prevent the formation of alternative iron oxide materials during the precipitation reaction.

Discussion

Using a phage display methodology we have selected a range of Adhiron proteins for their ability to bind to magnetite nanoparticles. By selecting binding sequences *in situ* within the confines of a robust scaffold protein we have established a novel approach to material binder development. Recent comparison between free peptide binders and antibody recognition of specific crystal planes shows that a proteinconstrained binding loop has greater potential for discriminating between rigid material surface than a flexible peptide due to its fixed spatial structure and molecular rigidity⁴². This is supported by other studies which indicate that a peptide will be interact with a material differently depending if it is free and flexible or constrained into a cyclic structure²⁵.

The magnetite selected Adhirons appear to have a clear interaction with magnetite nanoparticles in our QCM experiments, confirming that material binding is not dependent upon the presence of the pIII fusion protein, and is not an interaction with the scaffold itself, but a property derived from the variable loops. The positively charged surface of the MIA created by the lysine rich binding loops gives rise to a nucleic acid binding ability. This capability coupled with the proteins adsorption to magnetite nanoparticles gives these binding proteins a potential novel dual function which could have interesting applications.

The MIA we have analysed appear to share many of the properties of poly-lysine, which has been used previously as a magnetite coating material⁴³ and is also a known DNA binding molecule³⁶. However, the conformationally constrained binding loops and presence of other amino acids offers a route to produce changes in the formed nanoparticles when included in synthetic precipitation reactions. MIA-1 is able to exert shape control in room temperature reactions; tipping the balance of the MNP reaction in favour of the specific cubic crystal morphologies that these sequences were enriched against. In higher temperature reactions or when free peptides are used we do not observe this activity, indicating that the constraint imposed on the binding loops by the protein scaffold may be necessary for function. The presence of MIA-1 also appears to reduce the occurrence of alternative iron oxides in these reactions.

A key advantage of constraining the binding sequences is that with more defined conformation, differences in binding can be more easily interpreted in terms of sequence variability rather than alternative conformation. Our analysis of a large number of binding sequences reveals a high level of consensus with a strong preference for the positively charged residues of lysine histidine, and arginine. This matches particularly well with our molecular modelling of amino acid side chain interactions with a magnetite [100] surface which indicates that the most negative adsorption energy is observed with the amine group of lysine. The trend is in favour of interaction with positively charged amino acids at neutral pH with the negatively charged aspartate residue the least favoured. However, MIA-1 shows increased magnetite binding compared to MIA-2 in QCM-D and phage ELISA, and also displays morphology control during MNP synthesis. Yet MIA-1 contains fewer basic residues than MIA-2. Clearly the interplay between binding affinity and loop sequence is not a simple correlation to numbers of basic residues (although this does seem to be an important factor). but is most likely due to a subtle range of factors. A recent analysis of specific peptide binding to titania and silica surfaces revealed that positively charged residues are crucial for making the binding interactions, and that the balance of charged residues is important depending on the material used⁴⁴. In our case we hypothesise that due to the strong interaction between lysine and magnetite in our modelling studies, and the corresponding dominance of basic residues in the binder pool, lysine residues will be the main binding mediators. Their position and precise local topology within the binding loop are likely to govern how the residues interact with the specific arrangement of charges on the magnetite surface. The nuances of the MIA sequences and their implications for magnetite binding can only be unlocked through further in depth study of the entire binder pool.

Conclusions

Biological systems are able to synthesise materials with exquisite control down to the nanoscale into well-defined morphologies and nanoarchitectures using specialised biomineralisation proteins which have evolved over millennia. Mimicking this ambient, aqueous chemistry synthetically is a current challenge in nanomaterial development, and the use of screening strategies such as phage display and other combinatorial strategies offers a versatile fast-track route for achieving similar results in vitro. The manufacture of MNP with cubic faces currently requires high temperatures and a range of organic reagents. By using these cubic particles as a substrate for phage display we have selected a protein which demonstrates the ability to direct the production of particles with similar cubic morphology, thereby suggesting the potential to replicate an original extreme condition synthesis at room temperature and with mild reaction conditions. This highlights the power of combinatorial approaches in the development of novel nanomaterial green synthesis strategies By screening for binding sequences within the robust Adhiron scaffold we are able to limit the conformational freedom of

6, pp. 2681-2689.

Huang, Nat Chem, 2011, 3, 393-399.

Mater Med, 2010, **21**, 1103-1107.

Belcher, Nature, 2000, 405, 665-668.

Nanotechnol, 2002, 2, 95-100.

Knecht, ACS Nano, 2009, 3, 1288-1296.

15998-15999.

7900.

Acta Materialia Inc. Published by Elsevier Ltd, England, 2010, vol. 14. Y. Li, G. P. Whyburn and Y. Huang, J Am Chem Soc, 2009, 131 15. U. O. Seker, B. Wilson, S. Dincer, I. W. Kim, E. E. Oren, J. S. Evans, C. Tamerler and M. Sarikaya, Langmuir, 2007, 23, 7895 16. C. Y. Chiu, Y. J. Li, L. Y. Ruan, X. C. Ye, C. B. Murray and Y 17. D. B. Pacardo, M. Sethi, S. E. Jones, R. R. Naik and M. R 18. Y. Liu, J. Mao, B. Zhou, W. Wei and S. Gong, J Mater Sci 19. R. R. Naik, L. L. Brott, S. J. Clarson and M. O. Stone, J Nanosci 20. S. R. Whaley, D. S. English, E. L. Hu, P. F. Barbara and A. M

21. H. Chen, X. Su, K. G. Neoh and W. S. Choe, Anal Chem, 2006 **78**, 4872-4879. 22. Y. Fang, N. Poulsen, M. B. Dickerson, Y. Cai, S. E. Jones, R. R.

Naik, N. Kröger and K. H. Sandhage, 2008. 23. L. A. Kelley and M. J. Sternberg, Nature protocols, 2009, 4

363-371.

24. S. McNicholas, E. Potterton, K. S. Wilson and M. E. Noble, Acta crystallographica. Section D, Biological crystallography 2011, 67, 386-394.

25. M. Hnilova, E. E. Oren, U. O. S. Seker, B. R. Wilson, S. Collino, J. S. Evans, C. Tamerler and M. Sarikaya, 2008.

26. D. W. M. T. Klem, D. J. Solis, A. M. Belcher, M. Young, T. Douglas, Adv. Funct. Mater., 2005, 15.

27. C. Tiede, A. A. Tang, S. E. Deacon, U. Mandal, J. E. Nettleship, R. L. Owen, S. E. George, D. J. Harrison, R. J. Owens, D. C. Tomlinson and M. J. McPherson, Protein engineering, design &

selection : PEDS, 2014, 27, 145-155. 28. H. Kondo, K. Abe, Y. Emori and S. Arai, Febs Letters, 1991

278, 87-90.

29. J. Baumgartner, M. A. Carillo, K. M. Eckes, P. Werner and D. Faivre, Langmuir, 2014, **30**, 2129-2136.

30. R. R. Naik, S. E. Jones, C. J. Murray, J. C. McAuliffe, R. A. Vaia and M. O. Stone, Adv Funct Mater, 2004, 14, 25-30.

31. M. C. F. Thomsen and M. Nielsen, Nucleic Acids Res, 2012 40, W281-W287.

32. S. Brown, Proc Natl Acad Sci U S A, 1992, 89, 8651-8655.

33. T. G. Schmidt, J. Koepke, R. Frank and A. Skerra, J Mol Biol 1996, **255**, 753-766.

34. J. A. Glasel, Biotechniques, 1995, 18, 62-63.

35. D. E. Olins, A. L. Olins and Vonhippe.Ph, Journal of Molecular Biology, 1967, 24, 157-&.

36. G. Liu, M. Molas, G. A. Grossmann, M. Pasumarthy, J. C. Perales, M. J. Cooper and R. W. Hanson, Journal of Biological Chemistry, 2001, 276, 34379-34387.

37. T. Middleton and B. Sugden, J Virol, 1992, 66, 489-495.

38. K. A. Marx, Biomacromolecules, 2003, 4, 1099-1120. 39. A. Pomorska, D. Shchukin, R. Hammond, M. Cooper, G Grundmeier and D. Johannsmann, Analytical Chemistry, 2010, **82**, 2237-2242.

40. G. L. Dybwad, J Appl Phys, 1985, 58, 2789-2790.

the binding loops and focus instead on sequence preference which is dominated by lysine and other basic residues. This is in contrast to other iron oxide screening results which have shown the difficulty of generating significant sequence consensus from flexible peptide libraries ^{29, 32}. We believe the Adhiron scaffold represents a promising new approach for precision manufacture of magnetite nanoparticles and that the method we outline could potentially be expanded to other materials for future nanomaterial synthesis.

Acknowledgements

This work was supported by funding from the BBSRC (BB/H005412/2) and EPSRC (EP/L000202). Additional funding was from The University of Leeds via the Biomedical and Health Research Centre and The University of Huddersfield. The authors would like to thank J. Galloway for assistance with the QCM-D and figure drawing, C. Tiede for useful discussions during the phage display, and to A. Judge and J. Adams for help in preparing the cube MNP used in the Adhiron selection process. We also extend our thanks to S. Tsokov, C. Hill and the Sheffield Electron Miscroscopy unit for TEM and HRTEM, and to S. Spey for powder XRD data collection.

References

1. Q. A. Pankhurst, J. Connolly, S. K. Jones and J. Dobson, Journal of Physics D: Applied Physics, 2003, 36, R167-R181.

2. S. Laurent, D. Forge, M. Port, A. Roch, C. Robic, L. Vander Elst and R. N. Muller, Chem Rev, 2008, 108, 2064-2110.

3. Y. Amemiya, A. Arakaki, S. S. Staniland, T. Tanaka and T. Matsunaga, Biomaterials, 2007, 28, 5381-5389.

4. A. Arakaki, J. Webb and T. Matsunaga, J Biol Chem, 2003, **278**, 8745-8750.

5. A. E. Rawlings, J. P. Bramble, R. Walker, J. Bain, J. M. Galloway and S. S. Staniland, Proceedings of the National Academy of Sciences of the United States of America, 2014, 111, 16094-16099.

6. C. Tamerler, S. Dincer, D. Heidel, N. Karaguer and M. Sarikaya, Biomicroelectromechanical Systems (Biomems), 2003, 773, 101-110.

7. U. O. Seker and H. V. Demir, *Molecules*, 2011, 16, 1426-1451.

8. T. Bratkovic, Cell Mol Life Sci, 2010, 67, 749-767.

9. K. I. Sano and K. Shiba, Journal of the American Chemical Society, 2003, 125, 14234-14235.

10. C. R. So, J. L. Kulp, E. E. Oren, H. Zareie, C. Tamerler, J. S. Evans and M. Sarikaya, Acs Nano, 2009, 3, 1525-1531.

11. S. Brown, Nat Biotechnol, 1997, 15, 269-272.

12. J. M. Slocik, M. O. Stone and R. R. Naik, Small, 2005, 1, 1048-1052.

13. J. Kim, Y. Rheem, B. Yoo, Y. Chong, K. N. Bozhilov, D. Kim, M. J. Sadowsky, H. G. Hur and N. V. Myung, in Acta Biomater, 2010 41. W. Smith and T. R. Forester, *Journal of molecular graphics*, 1996, **14**, 136-141.

42. A. Artzy-Schnirman, E. Abu-Shah, M. Dishon, H. Soifer, Y. Sivan, Y. Reiter, I. Benhar and U. Sivan, *Journal of peptide science* : an official publication of the European Peptide Society, 2014, **20**, 446-450.

43. C. Riggio, M. P. Calatayud, C. Hoskins, J. Pinkernelle, B. Sanz, T. E. Torres, M. R. Ibarra, L. J. Wang, G. Keilhoff, G. F. Goya, V. Raffa and A. Cuschieri, *Int J Nanomed*, 2012, **7**, 3155-3166.

44. B. Z. Megumi Fukuta, Mutsunori Uenuma, Naofumi Okamoto, Yukiharu Uraoka, Ichiro Yamashita, Heiji Watanabe, *Colloids and surfaces. B, Biointerfaces*, 2014, **118C**.