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The stability of  $B_6$  octahedron in  $BaB_6$  under high pressure

 Xin Li, Xiaoli Huang, Defang Duan, Gang Wu, Mingkun Liu, Quan Zhuang, Shuli Wei, Yanping Huang, Fangfei Li, Qiang Zhou, Bingbing Liu and Tian Cui\*

 State key laboratory of Superhard Materials, College of Physics, Jilin University, Changchun 130012, People's Republic of China. E-mail: cuitian $\omega$ jlu.edu.cn; Fax: +86-431-85168825; Tel: +86-431-85168825

## **Abstract**

 We have performed in situ synchrotron X-ray diffraction and first-principles calculations to explore the compressive behaviors of Barium hexaboride  $(BaB<sub>6</sub>)$  under high pressure. No phase transitions in our experiment are observed up to 49.3 GPa at ambient temperature. It is found that the ambient cage structure (*Pm*-3*m*) is still stable with basic covalent - network during the experimental pressure run. Our theoretical calculations results show that the ambient structure might transform into three dynamically stable structures (*Cmmm*, *Cmcm*, *I*4/*mmm*) at 78 GPa, 97 GPa and 105 GPa respectively. The energy band calculations indicate that the sample is still semiconductor with narrow gap at 50 GPa.

## **Introduction**

The hexaborides  $MB_6$  (M = Y, La, Ca...) have a typical cage structure (*Pm*-3*m*). Owning to non-oxide structure, they have some excellent advantages such as high melting point, high chemical stability, high hardness, low density, and low coefficient of thermal expansion.<sup>1-4</sup> There are also some attractive properties in their various physical phenomena in recent decades. For example, rare-earth hexaborides exhibit Kondo behavior and valence-fluctuations in  $CeB_6^5$ ; La $B_6$  has low volatility and low work functions through enhancing thermionic emission at high temperature<sup>6</sup>; High superconducting critical temperature  $(T_c)$  is found to be 7 K in  $YB_6^7$  and narrow-gap semiconducting behavior exists in  $YbB<sub>6</sub>$ . Moreover, alkaline-earth metal hexaborides

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also have a fascinating feature in magnetism. There is coexistence phenomenon of weak ferromagnetism and antiferroelectricity in boron-deficient  $MgB<sub>6</sub>$ <sup>9</sup> Possibly because of a low-density free-electron gas,  $CaB<sub>6</sub>$  exhibits weak ferromagnetism in high temperature<sup>10</sup> and then subsequent experiments performed on undoped systems of  $CaB_6$ ,  $SrB_6$  and  $La-doped BaB_6$  indicating that magnetism might be an intrinsic property of alkaline-earth metal hexaborides, $10-12$  whose mechanism might be excitonic insulator by bound electron-hole pairs (excitons) condensing. Hexaborides of Alkaline-earth metal are simple polar semiconductors or semimetal with a fraction of an eV band gap because  $B_6$  molecule acquires two electrons from alkaline-earth metal to form a large divalent  $B_6$  anion and a stable metallic cation. A large number of theoretical calculations and experiments are extensively undertaken and an accredited conclusion is established that the electronic structure of  $CaB<sub>6</sub>$  is either semimetal or semiconductor.<sup>10, 11, 13-18</sup> At ambient conditions,  $MB_6$  (M = Ca, Sr, Ba) have a highly symmetrical CsCl-type structure with  $Pm-3m$  space group in which  $B_6$  molecule has the octahedral structure. Metal atoms and  $B_6$  molecules replace the position of Cs and Cl atoms respectively. Recently, Kolmogorov *et al* have found a new structure  $(I4/mmm)$  of CaB<sub>6</sub> under high pressure.<sup>19</sup>

 Nevertheless, Barium hexaboride, as an important Alkaline-earth metal hexaboride, is only researched on preparation and properties at ambient conditions.<sup>20</sup> Over the last several years, considerable experimental and theoretical investigations on the electronic band structures and ferromagnetism of alkaline-earth hexaborides have been performed.<sup>21-26</sup> From the results of D. C. magnetization measurements on single crystalline  $BaB<sub>6</sub>$ , the saturation magnetization at low temperatures is 8 times, in line with other weak ferromagnet of the hexaboride series. $27$ 

 Up to now, research on barium hexaboride under extreme conditions has not been reported. In this article, in order to explore properties of  $BaB<sub>6</sub>$  under high pressure, we have carried out in situ synchrotron angle dispersive X-ray diffraction (ADXRD) measurements and first-principles calculations.

## **Experimental and theoretical methods**

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 $BaB<sub>6</sub>$  powder with a purity of 99.5% was purchased from Alfa Aesar. The sample was ground finely before loaded into the sample chamber. A symmetric diamond anvil cell (DAC) with 300 *µ*m flat culets was used to generate pressure. A tungsten gasket was drilled a 100 um diameter hole in the center as a chamber, which had been precompressed to a thickness of 70  $\mu$ m. Pressure inside the chamber was determined by the standard ruby fluorescence method.<sup>28</sup> Furthermore, mixture of methanol-ethanol-water (16:3:1, MEW) was selected as the pressure transmitting medium (PTM).<sup>29</sup> In situ ADXRD measurements were carried out at the 4W2 High-Pressure Station of Beijing Synchrotron Radiation Facility (BSRF). The beam size was about 32(H)  $\times$  12(V)  $\mu$ m<sup>2</sup> and the incident wavelength was 0.6199 Å. A MAR3450 image plate detector was used to collect the diffraction patterns and the two-dimensional XRD images were radially integrated using the FIT2D software,  $30$ yielding intensity versus diffraction angle  $2\theta$  patterns. Prior to measurement, CeO<sub>2</sub> standard was used to calibrate the geometric parameters. The average acquisition time for each diffraction pattern to obtain sufficient intensity was 300 s. The fitted XRD patterns were completed by means of the Reflex module in the Material Studio Program.<sup>31</sup>

The first-principles calculations were performed with the pseudopotential plane-wave method based on density functional theory (DFT) implemented in the  $CASTEP<sup>32</sup>$  and VASP code.<sup>33</sup> The generalized gradient approximation (GGA) with the Perdew–Burke–Ernzerhof (PBE) exchange–correlation functional<sup>34</sup> was used in the calculation of optimizations. The hybrid functional of Heyd, Scuseria and Ernzerhof  $(HSE06)^{35, 36}$  was also used to verify the results of band structures. A plane-wave cutoff energy of 540 eV was employed for norm-conserving pseudopotentials and the Brillouin zone sampling grids of spacing  $2\pi \times 0.03$  Å<sup>-1</sup>. The project-augmented wave  $(PAW)^{37}$  method was adopted with valence electrons of  $2s^2 2p^1$  and  $5s^2 5p^6 6s^2$  for B and Ba atoms, respectively.

## **Results and discussion**

Like most of other hexaborides,  $BaB<sub>6</sub>$  has a simple cubic crystal structure with

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*Pm*-3*m* space group under ambient conditions. In this structure, six boron atoms form an octahedron connected by covalent bonds and each octahedron connects with six closest octahedrons by covalent bonds that forms a three-dimensional network. Due to this strong covalent network, the framework of *Pm*-3*m* structure is very stable. The schematic representation of  $BaB<sub>6</sub>$  crystal structure is shown in Fig. 1(a) and the electron localization function (ELF) of BaB<sub>6</sub> is plotted in Fig. 1 (b) and (c). B<sub>6</sub> octahedron pillages two electrons from Ba atom, and forms a stable divalent  $B_6$  anion. Ba cation inserts interstices of eight  $B_6$  octahedrons, which forms a typical cage structure. There are only two parameters to restrain  $BaB<sub>6</sub>$  crystal structure, one is lattice constant *a*, and another is fractional coordinates (Wyckoff site) of boron atom 6*f* (0.5, 0.5, *z*). The synchrotron XRD patterns have been obtained from 0.8 GPa to 49.3 GPa using MEW as the PTM. The refinement of  $BaB<sub>6</sub>$  has been carried out on an XRD pattern obtained at 0.8 GPa in Fig. 2(a). The refined lattice constant is  $a =$ 4.2614 Å and Wyckoff site of B atom is  $(0.5, 0.5, 0.2053)$  with unit cell volume  $V =$ 77.38  $\AA^3$ . Selected XRD diffraction patterns from this experimental run are illustrated in Fig. 3(a). It shows that there was no appearance of new peaks when the sample was compressed smoothly up to the highest pressure. In the process of compression, all Bragg diffraction peaks gradually broadened comparing with the initial sharp peaks and shifted toward higher  $2\theta$  angles, which indicates that the interplanar distance of crystal planes decreased. The original XRD peaks existed up to the highest pressure 49.3 GPa without appearance of new peaks indicating no occurrence of the phase transition. It confirms the stability of *Pm*-3*m* structure up to 49.3 GPa at least. Besides that, intensity of diffraction peaks were affected by preferred crystallite orientation. Compared with crystallite size, the X-ray beam size was relatively bigger, which could partly reduce mismatch in intensity caused by statistics of measurement.

In order to analyze the evolution of the lattice constants and unit cell volume with pressure, all XRD patterns were refined by Rietveld full profile structure. Fig. 3(b) shows the lattice constant  $a$ , B-B<sub>intra</sub> bond and B-B<sub>inter</sub> bond length as a function of pressure. The data were linearly fitted without any jumps:  $k_a = 0.0066 \text{ Å} \cdot \text{GPa}^{-1}$ ,  $k_{intra}$ 

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 $= 0.00274$  Å  $\cdot$  GPa<sup>-1</sup>,  $k_{inter} = 0.00270$  Å  $\cdot$  GPa<sup>-1</sup>. The lattice parameter shows the monotonically decrease with the increasing pressure indicating the reservation of the basic covalent network. We have presented the volume reduction as a function of pressure in Fig. 4. The experimental pressure-volume data were fitted by third-order Birch-Murnaghan (BM) equation of state  $(EOS)^{38}$ 

$$
P = \frac{3B_0}{2} \left[ \left( \frac{V}{V_0} \right)^{-\frac{7}{3}} - \left( \frac{V}{V_0} \right)^{-\frac{5}{3}} \right] \left\{ 1 + \frac{3}{4} (B_0' - 4) \left[ \left( \frac{V}{V_0} \right)^{-\frac{2}{3}} - 1 \right] \right\}
$$

where  $V_0$  is the volume per formula unit at ambient pressure,  $V$  is the volume per formula unit at pressure *P* given in GPa,  $B_0$  is the isothermal bulk modulus, and  $B_0'$  is the first pressure derivative of the bulk modulus. As a result of enough experimental data, we have set up  $B_0'$  of 4. Then, we obtained a bulk modulus of  $94 \pm 1$  GPa. The volume  $V_0$  is fitted to be 77.8  $\pm$  0.2 Å<sup>3</sup>. As is shown in Fig. 4(b), it is clearly seen that the volume of the sample at highest pressure is reduced by 20% compared with the original pressure point.

The present XRD results have confirmed that the structure *Pm*-3*m* is stable up to 49.3 GPa. Nevertheless, the recent theoretical and experimental results on another Alkaline-earth metal hexaboride  $CaB<sub>6</sub>$  have reported the phase transitions of  $CaB<sub>6</sub>$ under high pressure. Most strikingly, A. N. Kolmogorov *et al*. have proposed three dynamically stable structures by a systematic analysis of multiple imaginary-frequency phonon modes when ambient structure becomes dynamically unstable around 25 GPa. The novel structure of  $CaB<sub>6</sub>$  was successfully quenched down to ambient pressure, which revealed a transition from ambient structure *Pm*-3*m* to tetragonal *I*4*/mmm* structure after laser-heating in DAC at 31 GPa.<sup>19</sup> M. Li *et al.* also found a pressure-induced phase transition at ambient temperature by means of resistance measurement and synchrotron XRD.<sup>39</sup> Both Ca and Ba are alkaline earth metal elements and they have the similar valence electrons  $3s^2 3p^6 4s^2$  and  $5s^2 5p^6 6s^2$ respectively. So we suppose that  $CaB_6$  and  $BaB_6$  have the same or similar propertis under pressure. We have obtained three candidate crystal structures (*I*4*/mmm*, *Cmmm*, *Cmcm*) by replacing Ca atom with Ba atom and the cell parameters, which are given in reference 19. The crystal structure prediction were also performed by CALYPSO

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code<sup>40</sup> with two BaB<sub>6</sub> formula units at 100 GPa. The most stable structure is *Cmmm* which is the same structure with the replaced one. Then, we have calculated the enthalpy of the four structures mentioned above (*Pm*-3*m*, *I*4*/mmm*, *Cmmm*, *Cmcm*) by optimization method. Taking the enthalpy of *Pm*-3*m* structure as reference energy, we plotted relative enthalpy per formula unit of other three structures as a function of pressure in Fig. 5. The results illustrate that *Pm*-3*m* structure might not transform to *Cmmm* or *Cmcm* until pressure up to 78 GPa and 97 GPa respectively. The tetragonal structure ( $I4/mmm$ ) discovered in CaB<sub>6</sub> at 31 GPa would appear in BaB<sub>6</sub> when pressure reaches 105 GPa. The calculation results shows that *Cmmm* have a lower enthalpy value than others after 78 GPa, so *Cmmm* structure is the most possible candidate structure. But just only by compression, ambient structure might not transform into a new structure in the same situation with  $CaB<sub>6</sub>$ , it must be supplied some paths such as laser-heated in DAC to break the energy barrier.

It is confirmed that  $CaB<sub>6</sub>$  is a semiconductor with a narrow gap by means of transport measurements, $14$  angle-resolved photoemission spectroscopy (ARPES) and momentum-resolved inelastic X-ray scattering.<sup>16</sup> Considering that the calculations with GGA-PBE would underestimate band gap, so we calculated electronic band structures with HSE06. Fig. 6 plots the band structures of  $BaB<sub>6</sub>$  around the Fermi level at 0 GPa and 50 GPa, which illustrates that  $BaB<sub>6</sub>$  is semiconductor with so narrow band gap 0.22 eV at ambient conditions. The band gap is sensitive to lattice constant *a* and wyckoff position of boron atom  $(0, 0, z)$ ,  $(4, 41)$  and our experimental results shows that wyckoff position  $(z = 0.20534)$  would change scarcely. So it is obvious that lattice constant decreases by compressing in Fig 3 (b) and the band gap also changes, then  $BaB<sub>6</sub>$  is still semiconductor with 0.34 eV at 50 GPa.

# **Conclusion**

In this work, we have successfully investigated structural stability and compressive behaviors of  $BaB<sub>6</sub>$  by means of in situ high pressure synchrotron X-ray diffraction and first-principles calculations. Our experimental results reveal that it is very difficult to break the covalent network only by high pressure condition. During

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experimental pressure, no phase transitions occurred and lattice constant *a* decreases linearly with pressure. Our theoretical calculations propose that the BaB6 is still semiconductor at 50 GPa. Three candidate dynamic stable structures (*Cmmm*, *Cmcm*, *I*4/*mmm*) might replace the typical cage structure at 78 GPa, 97 GPa, and 105 GPa, respectively. Attempting to accomplish the phase transition, it must be supplied some paths to break energy barrier.

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Fig. 1 (a) The  $Pm-3m$  crystal structure of  $BaB<sub>6</sub>$  at ambient conditions. Cubic cage formed by eight  $B_6$  octahedrons and a central Ba cation. Octahedral surfaces shown in gray. (b) (c) Electron localization function (ELF) maps of different plan.



**Fig. 2** (a) The Rietveld full-profile refinement of synchrotron XRD pattern at 0.8 GPa. (b) Micrograph of  $BaB_6$  sample in DAC at 0.8 GPa just after loading pressure with PTM of methanol−ethanol−water.



Fig. 3 (a) Typical XRD patterns at selected pressures of BaB<sub>6</sub> as a function of pressure on compression up to 49.3 GPa at room temperature. The peaks with marked diamonds symbols (◆) represent the bragg peaks from tungsten gasket. The wavelength of the incident X-ray beam is 0.6199 Å. (b) Evolution of the lattice constant *a* and length of B-B<sub>intra</sub> bond and B-B<sub>inter</sub> bond with respect to pressure. The symbols represent experimental results, and lines represent fitted results respectively.



Fig. 4 (a) Pressure dependence of the unit cell volume of BaB<sub>6</sub>. The red line shows fitted results. Insert (b) represents the volume reduction upon compressions. The blue line only shows the trend.



**Fig. 5** The relative enthalpy per formula unit of four structures as a function of pressure for BaB6. The enthalpy of *Pm*-3*m* structure is taken as reference energy compared with others. The inset figure represents the most possible candidate *Cmmm* structure after 78 GPa.



**Fig. 6** Electronic band structures of *Pm*-3*m* structure (a) at 0 GPa and (b) at 50 GPa.