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COMMUNICATION

## Microfluidic electrochemical growth of vertically aligned TiO<sub>2</sub> nanotubes for SERS optofluidic devices.

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**The growth of TiO<sub>2</sub> nanotubes (NTs) array into a microfluidic electrochemical reactor is here demonstrated. The same microsystem can be also exploited to decorate the NTs with Ag nanoparticles by in-situ photoreduction thus fabricating an optofluidic sensor based on SERS effect, avoiding any ambient contamination of substrates.**

TiO<sub>2</sub> nanotubes (NTs) obtained by anodic oxidation have been largely employed in several fields thanks to their superior charge transport properties and high surface area. They have been successfully integrated as active material into dye sensitized solar cells,<sup>1,2</sup> Li-Ions batteries,<sup>3,4</sup> water splitting devices,<sup>5</sup> supercapacitors,<sup>6</sup> photodetector,<sup>7</sup> gas sensors,<sup>8</sup> cell culture substrates,<sup>9</sup> biosensing<sup>10</sup> and electrochromic devices.<sup>11</sup> The nanotubes growth by anodization of a Ti foil is usually performed in a standard electrochemical reactor using fluoride based electrolytes at a certain voltage for a given time. Typically the nanotube diameter can be tuned by modulating the applied anodization voltage,<sup>12</sup> while the length can be controlled imposing the suitable anodization time<sup>2</sup> and by using different electrolytes.<sup>13</sup> Recent reports in literature have shown effective TiO<sub>2</sub> NTs employment as Surface Enhanced Raman Scattering (SERS) active substrate after decoration with plasmonic nanoparticles.<sup>14-19</sup> Thanks to their 3D nanostructuring and the relatively high specific surface area, TiO<sub>2</sub> NTs allow an increased loading of metal particles yielding 3D plasmon "hot spots" on very large area which, synergically to charge transfer processes, can provide huge Raman enhancements.<sup>19,20</sup>

Actually, in Raman applications it can be crucial to avoid SERS substrate contamination in between its synthesis and the Raman analysis; this would prevent external molecule adsorption provoking vibrational backgrounds superimposed

to the effective analytes signal. In particular, the inclusion of organic impurities during SERS-active substrate synthesis and/or handling dramatically affect the SERS limit-of-detection.<sup>21</sup> Moreover, the advantages deriving from SERS substrate integration in microfluidics, which can benefit from a closed environment, have been only recently considered.<sup>19,22-24</sup> First of all, a microfluidic circuitry can solve problems dealing with a non-uniform molecular distribution on the surface of SERS substrates (i.e. "coffee ring" effect, where molecules likely accumulate on the edge of the droplet) since it can better confine the liquid analytes close to the SERS-active area. Secondly it can be really effective in reducing risks of sample contamination, which is particularly crucial for biological analysis. Finally, a microfluidic platform can be optimized in order to reduce the reagent volume, thereby decreasing the analysis cost.

A possible solution to overcome the discussed contamination problems of SERS-active substrates is to grow the TiO<sub>2</sub> NTs array few time before their application, avoiding intermediate steps in bare atmosphere. Recently Ladanov et al.<sup>25</sup> have reported the microfluidic growth of ZnO nanowires. In that case the microfluidic approach allows to overcome the problems related to the grown of nanostructures in hollow channels where the flowing of ever fresh precursor are needed. However there are only few reports in literature about the fabrication of electrochemical reactor exploiting microfluidics<sup>26,27</sup> and no one dedicated to the anodic oxidation process.

Herein, the anodization of Ti foil into a microfluidic chamber is reported for the first time. Vertically aligned NTs were grown into an microfluidic electrochemical cell. The same chamber is also used to decorate them with Ag NPs fabricating a SERS based optofluidic sensor without any exposure to environmental pollution.

The microfluidic architecture designed for this application is schematically depicted in Figure 1a and consists of a 200 μm thick PDMS membrane (5) sandwiched between a Ti foil (6) and a transparent conductive substrate (3) (F-doped Tin Oxide deposited on glass).

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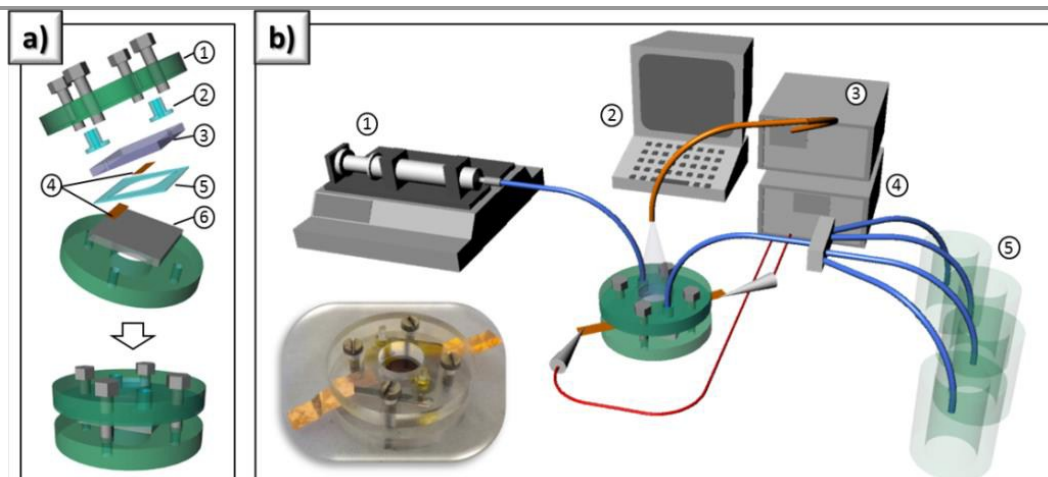


Figure 1. 3D schemes of microfluidic architecture (a) and of the microfluidic anodic oxidation setup (b). The inset in b) shows a digital photograph of the assembled microfluidic cell.

The PDMS membrane was fabricated by easy casting procedure<sup>28</sup> into micro-machined PMMA mould (by NC milling machine) with a double-drop shape in order to promote air bubble evacuation during the injection of the analytes/reagents.

The electrochemical cell was completed by a PMMA clamping system (1) which held PDMS press-fit interconnections (2) to avoid leakages under pressure. Two copper foils (4) were used as electric contacts by simply positioning them at the Ti foil-PDMS and FTO-PDMS interfaces, resulting insulated by the elastomeric membrane. The experimental setup (represented in Figure 1b) consisted on a motor-syringe (1) connected to the microfluidics via polymeric tubing, a power supply and connection cables (4), a manual microfluidic switch, four different reservoirs (5), a UV-lamp (3) connected with an optical fiber and a personal computer (2). A digital photograph of the assembled cell is reported as inset of Figure 1b.

The first step of the NTs growth (schematically represented in Figure 2a) involved the cleaning of the substrates by fluxing acetone and ethanol in the microfluidic chamber for 5 minutes for each. After that the chamber was filled with a fluorine based electrolytic solution (0.5wt.%  $\text{NH}_4\text{F}$  and 2.5wt.%  $\text{H}_2\text{O}$  into EG). The electrochemical process was conducted for 10 minutes applying an anodization potential of 60 V using a DC power supply (GW Instek SPD-3606) and a constant flow rate equal to 1 ml/min. This flow rate value was set in order to allow the evacuation of bubbles evolving from the electrodes. Indeed during anodic oxidation gas evolution ( $\text{H}_2$  and  $\text{O}_2$ ) occurred at the electrodes and, in a microfluidic chamber, these bubbles can create electrolytic-voids that could strongly affect the NTs grown. At the end of the process, the potential was switched down and the inlet of the chamber was connected to a water reservoir to eliminate contamination from the electrolyte for 5 minutes. The deposition of Ag nanoparticle was obtained by filling the microfluidic chamber with a silver precursor solution ( $\text{AgNO}_3$  in water and ethanol) under UV-light irradiation for 5 minutes using an optical fiber

illumination across the transparent ceiling of the cell. The photocatalytic properties of  $\text{TiO}_2$  nanotubes were exploited for the photo-reduction of silver ions to form Ag nanoparticles. After a second cleaning step in water, the in-situ fabricated SERS active substrate was ready to be employed as molecular sensor.

Thanks to the intrinsic reversible sealing of the microfluidic cell, Field Emission Scanning Electron Microscopy (FESEM) investigation is allowed in order to have information at the different process steps.

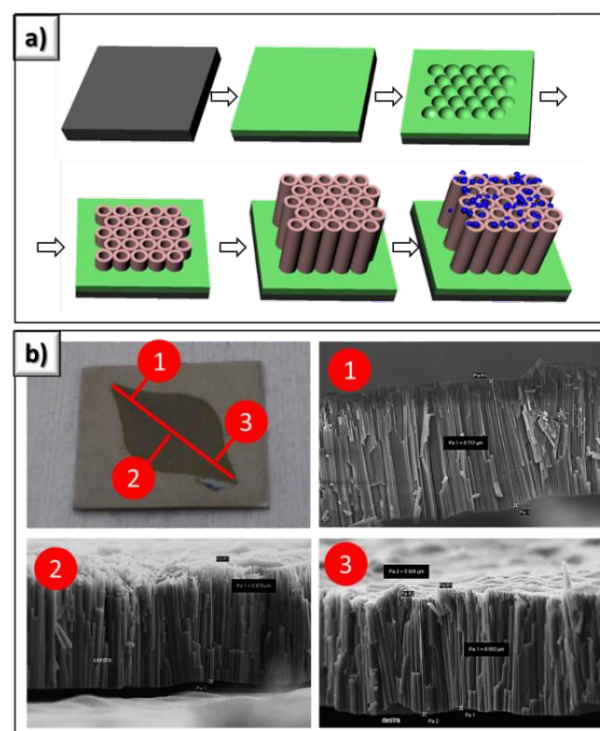


Figure 2. 3D scheme representing the  $\text{TiO}_2$  NTs growth by anodic oxidation (a) and FESEM micrographs (b) showing the cross-section of NT arrays in three different places along the diagonal of the sample (from inlet to outlet).

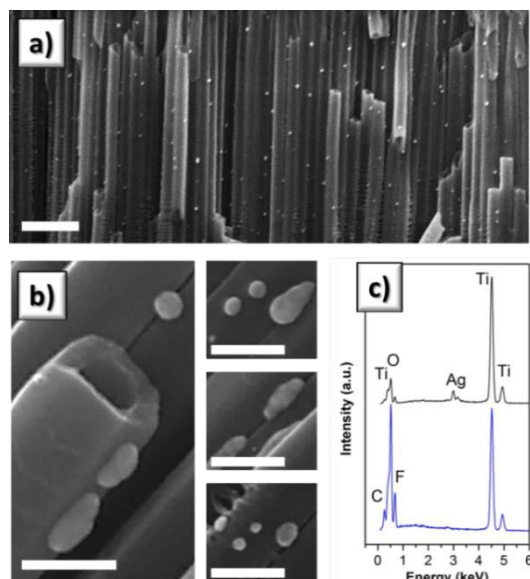


Figure 3. FESEM images (a-b) and EDX (c) spectra of  $\text{TiO}_2$  nanotubes before and after the  $\text{AgNO}_3$  photoreduction (scale bars in FESEM images are 500 nm in a) and 100 nm in b)).

The final appearance of the  $\text{TiO}_2$  NTs array after the microfluidic growth is shown in Figure 2b. FESEM cross section images collected in different places along the diagonal of the growth area confirm the uniformity of the microfluidic anodization: a constant thickness of about 6  $\mu\text{m}$  was obtained for 10 minutes treatment in the electrolytic solution. As comparison,  $\text{TiO}_2$  NTs were grown into a standard electrochemical cell and the obtained NTs lengths as a function of the synthesis time are reported in Figure S1 (see Supporting Information). Growth rate seems to be weekly dependent by the setup with NTs length slightly higher for the microfluidic configuration. The walls of the tubes are smooth (Figure S2a) thanks to the polar organic nature of the electrolytic solution.<sup>2</sup> The tubes have an average external diameter of around 110 nm with walls thickness of around 20 nm and exhibit a perfect vertical alignment and hexagonal assembly (see Figure S2b).

After the  $\text{AgNO}_3$  photoreduction the nanotubes surface was fully decorated by Ag nanoparticles. The FESEM images reported in Figure 3a confirm the uniform distribution of silver NPs along the whole thickness of the NTs array. Higher magnification FESEM images in Figure 3b allow to appreciate the quasi-hemispherical shape of the Ag NPs with dimension ranging from 10 up to 90 nm (size distribution evaluated from FESEM images is shown in Figure S3).

X-ray diffraction (XRD) patterns of as-grown and Ag-decorated  $\text{TiO}_2$  nanotubes are shown in Figure S4. The spectrum of the bare NTs present only peaks associated to titanium substrate since the as-anodized  $\text{TiO}_2$  nanotubes are fully amorphous. After the Ag synthesis, new peaks appear and can be assigned to the diffraction of (111), (200), and (220) planes of face-centered cubic (fcc) silver (JCPDS card n. 4-783,  $a = 4.08 \text{ \AA}$ ).

Chemical composition of the nanostructured thin film was investigated using EDX analysis. The EDX spectra (Figure 3c) exhibit C, F, O, Ag, and Ti peaks, showing that the prepared

materials are composed of Ag and  $\text{TiO}_2$  with the presence of fluorine and carbon atoms, as confirmed by the semi-quantitative standard-less analysis (Table S1). This contamination is consistent with the study of Albu et al.<sup>29</sup> that described the  $\text{TiO}_2$  NTs as composed by two layers: an outer part (shell) of pure and dense  $\text{TiO}_2$  and an inner porous part (core) containing electrolyte components.

The UV irradiation induces two secondary effects in addition to the Ag nanoparticles deposition: the first one is related to the heating of the sample which leads to a partial crystallization of amorphous titania.<sup>30</sup> Indeed it is possible to observe the appearance of two peaks in the XRD spectrum associated with the  $\text{TiO}_2$  in the anatase phase. The second and more important effect consists in the degradation of the contaminants (as reported in Table S1 – showing a reduction of C and F contents after the photoreduction process) arising from the wet electrochemical synthesis. This phenomenon can be attributed both to the above mentioned sample heating (previous studies demonstrate how heat treatments allow to eliminate the presence of contaminants in electrochemically grown  $\text{TiO}_2$  NTs)<sup>2</sup> and to the photocatalytic effect of  $\text{TiO}_2$ .

Optimized Ag decorated  $\text{TiO}_2$  NTs were employed into a SERS opto-fluidic sensor and they were able to detect cyanine (Cy5) molecules in ethanol solutions injected in the device microchamber at low concentrations (see the top spectrum in Figure 4 concerning with the Raman analysis of Cy5 diluted in ethanol solution at a concentration of  $10^{-6} \text{ M}$ ). The SERS measurements were performed by a micro-Raman spectrometer (Renishaw inVia, excitation at 514.5 nm) using a long working distance objective to focus/collect the exciting/scattered light (backscattering configuration) on the Ag- $\text{TiO}_2$  NT surface through the PDMS top-cover of the device (see Figure S5). Figure 4 shows the typical vibrational bands of the analytes in off-electronic resonance condition, since the excitation photon energy (2.54 eV) is far from the Cy5 absorption band maximum (1.9 eV).

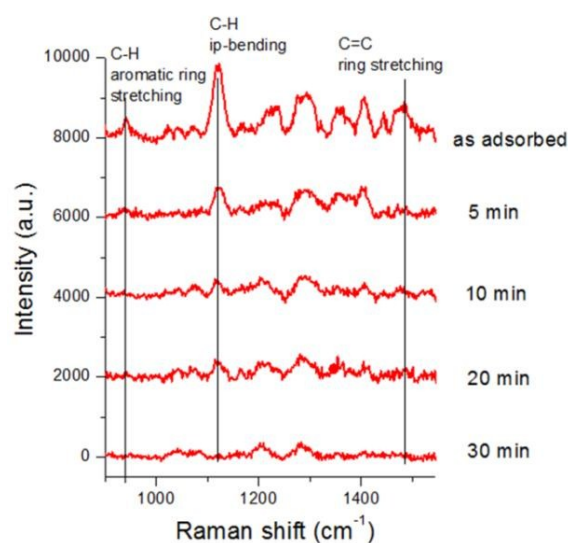


Figure 4. Raman spectra of Cy5 diluted in ethanol solution at a concentration of  $10^{-6} \text{ M}$  injected in the fluidic chamber (some main Raman bands are indicated) hosting the Ag-coated  $\text{TiO}_2$  NTs. The spectra are collected at different steps of UV-cleaning.

It is worth to underline that the most commonly used Raman substrates (i.e. Au or Ag nanoparticles on dielectric matrices) cannot be easily reused. This could be an important drawback from the economical point of view, but also from the substrates reproducibility standpoint. For these reasons a lot of efforts have been addressed in the last years in studying recyclable SERS substrates. The intrinsic photocatalytic activity of metal-oxide nanostructures could be exploited to degrade the analytes molecules after the SERS analysis allowing recyclability for a new detection run.<sup>31</sup>

The general mechanism of photocatalysis on metal-oxide nanostructures is based on the electron-hole pairs photogeneration under UV light, inducing the reaction with both oxygen and water molecules to promote the oxidation of organic substances. Nevertheless the fast recombination of electric charges in the semiconductor alone could reduce the resulting photocatalytic efficiency. To obtain effective molecules photodegradation the electrons and the holes in semiconductor must be separated to suppress their recombination. The noble metal-decorated metal-oxides are more efficient than their uncoated counterpart since for semiconductor/metal hybrid nanostructures with suitable energy level alignment, electrons can be quickly transferred to metal nanoparticles to achieve charges separation.<sup>32,33</sup>

Figure 4 shows the Raman spectra of Ag-coated TiO<sub>2</sub> NTs impregnated with Cy5 in ethanol solution after UV-exposure at increasing irradiation times. In particular, after an exposure for 30 min the vibrational spectrum of Cy5 is completely quenched. Actually, the photocatalytic process responsible of the shown spectra evolution can be explained by an electrochemical mechanism: the Ag decorated TiO<sub>2</sub> NTs array serves as an electron relay for an oxidant and reductant, and electron transfer occurs on the surface of the silver nanoparticles supported on the TiO<sub>2</sub> NTs. Cy5 molecules can be photodegraded thanks to electrons transfer from the Ag to the TiO<sub>2</sub> conduction band, and conversely, holes transfer from the TiO<sub>2</sub> valence band to the Ag under UV excitation. Exploiting the superior photocatalytic properties of combined Ag-TiO<sub>2</sub> nanostructures, the SERS substrate can be finally self-cleaned under UV irradiation in order to allow its recycled use. FESEM characterization on UV-cleaned samples have been performed in order to assess the structural stability of the Ag decorated TiO<sub>2</sub> NTs. As it is possible to see in Figure S6, UV-cleaning does not show any influence on the samples morphology. This observation is in line with previously reported analysis on UV-cleaned recyclable SERS substrates.<sup>18</sup>

## Conclusions

In summary, metal-semiconductor nanostructures consisting of Ag-coated TiO<sub>2</sub> NTs have been fully synthesized in a microfluidic electrochemical reactor exploiting in-situ anodic oxidation and in-situ silver nitrate photoreduction. Vibrational spectroscopy analysis confirms their promising application as recyclable SERS-active substrates for opto-fluidic molecular sensors. Moreover the suitable energy level alignment in the obtained semiconductor/metal hybrid nanostructures could be

exploited to degrade analyte molecules after the SERS analysis allowing substrate recyclability.

Additionally the here reported study can be exploited and extended to the microfluidic anodic oxidation of several other metals (valve metals such as Al, Ta, V, Zr, Nb, Hf, Mg, ...) leading to nanostructured functional oxides largely investigated for different applications from sensing to energy harvesting and storage.

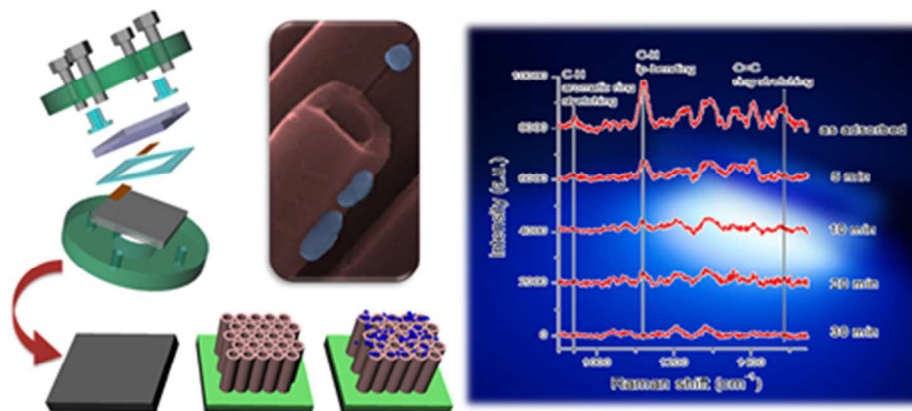
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## Notes and references

- H. P. Jen, M. H. Lin, L. L. Li, H. P. Wu, W. K. Huang, P. J. Cheng, E. W. G. Diau, *ACS Appl. Mater. Interfaces*, 2013, **5**, 10098.
- A. Lamberti, A. Sacco, S. Bianco, D. Manfredi, M. Armandi, M. Quaglio, E. Tresso, C. F. Pirri, *Sol. Energy*, 2013, **95**, 90.
- Z. Bi, M. P. Paranthaman, P. A. Menchhofer, R. R. Dehoff, C. A. Bridges, M. Chi, B. Guo, X.-G. Sun, S. Dai, *J. Power Sources*, 2013, **222**, 461.
- A. Lamberti, N. Garino, A. Sacco, S. Bianco, A. Chiodoni, C. Gerbaldi, *Electrochim. Acta*, 2015, **151**, 222.
- S. Hernández, D. Hidalgo, A. Sacco, A. Chiodoni, A. Lamberti, V. Cauda, E. Tresso, G. Saracco, *Phys. Chem. Chem. Phys.*, 2015, **17**, 7775.
- B. Chen, J. Hou, K. Lu, *Langmuir*, 2013, **29**, 5911.
- J. Zou, Q. Zhang, K. Huang, N. J. Marzari, *Phys. Chem.*, 2010, **114**, 10725.
- V. Galstyan, E. Comini, G. Faglia, G. Sberveglieri, *Sensors*, 2013, **13**, 14813.
- M. Y. Lan, C. P. Liu, H. H. Huang, J. K. Chang, S. W. Lee, *Nanoscale Res. Lett.*, 2013, **8**, 1.
- S. Liu, A. Chen, *Langmuir*, 2015, **21**, 8409.
- K. Lee, D. Kim, S. Berger, R. Kirchgeorg, P. Schmuki, *J. Mater. Chem.*, 2012, **22**, 9821.
- Y. Tang, J. Tao, Z. Dong, J. T. Oh, Z. Chen, *Adv. Nat. Sci.: Nanosci. Nanotechnol.*, 2011, **2**, 045002.
- S. So, K. Lee, P. Schmuki, *J. Am. Chem. Soc.*, 2012, **134**, 11316.
- R. Li, A. Zhou, Q. Lu, C. Yang, J. Zhang, *Colloids Surf. A*, 2013, **436**, 270.
- S. Nitta, A. Yamamoto, M. Kurita, R. Arakawa, H. Kawasaki, *ACS Appl. Mater. Interfaces*, 2014, **6**, 8387.
- Y. Huang, L. Sun, K. Xie, Y. Lai, B. Liu, B. Ren, C. J. Lin, *Raman Spectrosc.*, 2011, **42**, 986.
- Y. Chen, G. Tian, K. Pan, C. Tian, J. Zhou, W. Zhou, Z. Ren, H. Fu, *Dalton Trans.*, 2012, **41**, 1020.
- X. Li, G. Chen, L. Yang, Z. Jin, J. Liu, *Adv. Funct. Mater.*, 2010, **20**, 2815.
- A. Lamberti, A. Virga, A. Chiadò, A. Chiodoni, K. Bejtka, F. Rivolo, F. Giorgis, *J. Mat. Chem. C*, 2015, **3**, 6868.
- L. Yang, P. Li, H. Liu, X. Tang, J. Liu, *Chem. Soc. Rev.*, 2015, **44**, 2837.
- E. J. Blackie, E. C. L. Ru, P. G. Etchegoin, *J. Am. Chem. Soc.*, 2009, **131**, 14466.
- Q. Li, B. Li, Y. Wang, *RSC Adv.*, 2013, **3**, 13015.

- 23 J. Qi, J. Zeng, F. Zhao, S. H. Lin, B. Raja, U. Strych, R. C. Willson W. C. Shih, *Nanoscale*, 2014, **6**, 8521.
- 24 A. Lamberti, A. Virga, A. Angelini, A. Ricci, E. Descrovi, M. Cocuzza, F. Giorgis, *RSC Adv.*, 2015, **5**, 4404.
- 25 M. Ladanov, P. Algarin-Amaris, G. Matthews, M. Ram, S. Thomas, A. Kumar, J. Wang, *Nanotechnology*, 2013, **24**, 375301.
- 26 S. K. Yoon, G. W. Fichtl, P. J. Kenis, *Lab Chip*, 2006, **6**, 1516.
- 27 O. Scialdone, C. Guarisco, A. Galia, *Electrochim. Acta*, 2011, **58**, 463.
- 28 A. Lamberti, M. Quaglio, A. Sacco, M. Cocuzza, C. F. Pirri, *Appl. Surf. Sci.*, 2012, **258**, 9427.
- 29 S. P. Albu, A. Ghicov, S. Aldabergenova, P. Drechsel, D. LeClere, G. E. Thompson, J. M. Macak, P. Schmuki, *Adv. Mater.*, 2008, **20**, 4135.
- 30 O. Van Overschelde, R. Snyders, M. Wautelet, *Appl. Surf. Sci.*, 2007, **254**, 971.
- 31 X. Li, H. Hu, D. Li, Z. Shen, Q. Xiong, S. Li, H. J. Fan, *ACS Appl. Mater. Interfaces*, 2012, **4**, 2180.
- 32 A. Lamberti, Metal-Oxide Nanostructures for Surface Enhanced Raman Spectroscopy. In: Bhushan, B. (ed.) *Encyclopedia of Nanotechnology 2015*, Springer, Dordrecht (In press)
- 33 L. Yang, P. Li, J. Liu, *RSC Adv.*, 2014, **4**, 49635.



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