

RSC Advances



This is an *Accepted Manuscript*, which has been through the Royal Society of Chemistry peer review process and has been accepted for publication.

Accepted Manuscripts are published online shortly after acceptance, before technical editing, formatting and proof reading. Using this free service, authors can make their results available to the community, in citable form, before we publish the edited article. This *Accepted Manuscript* will be replaced by the edited, formatted and paginated article as soon as this is available.

You can find more information about *Accepted Manuscripts* in the [Information for Authors](#).

Please note that technical editing may introduce minor changes to the text and/or graphics, which may alter content. The journal's standard [Terms & Conditions](#) and the [Ethical guidelines](#) still apply. In no event shall the Royal Society of Chemistry be held responsible for any errors or omissions in this *Accepted Manuscript* or any consequences arising from the use of any information it contains.



Journal Name

COMMUNICATION

Facile synthesis of viologen and its reversible lithium storage property in organic lithium-ion batteries

Received 00th January 20xx,
Accepted 00th January 20xx

Arnab Ghosh and Sagar Mitra*

DOI: 10.1039/x0xx00000x

www.rsc.org/

The reversible lithium-ion storage property of a viologen has been explored for the first time. The viologen exhibits a reversible capacity of 175 mAh g⁻¹ against lithium at 50 mA g⁻¹ discharge rate, long term cycling performance and good rate capability. Reversible lithiation-delithiation property of the viologen has been confirmed by *ex-situ* NMR studies.

Due to high specific capacity and high energy density, rechargeable lithium ion battery technology¹ is becoming the most promising alternative to lead-acid, Ni-Cd, and Ni-MH batteries. The conventional lithium ion batteries are based on inorganic insertion compounds (e.g., LiCoO₂, LiMn₂O₄, LiFePO₄, LiV₃O₈ etc.) as cathode materials²⁻⁵ and Si, Sn etc.^{6,7} as anode materials. However, large scale production of these materials using ceramic processes causes a serious concern about environmental issues. Additionally, extracting of these transition metals from their ores and transforming them into electro-active materials is more expensive and produce toxic by-products. Therefore, there is a surging demand for cost-effective, sustainable electrode materials.^{8,9} As an alternative to conventional inorganic materials, the redox active organic compounds offer the opportunity to develop low cost and sustainable "green batteries" thanks to their sustainability, structural variety, flexibility¹⁰ and no ecological footprint. Not only that, the charge carrying function of elemental ions can be substituted by other molecular ions in redox reactions of organic compounds as reported by M. Yao *et al.*¹¹ very recently.

The organic electrode materials are not a new concept, since, the first organic cathode based on dichloroisocyanuric acid was proposed in 1969.¹² After that, for a long time, organic electrode materials have received much less attention because of their less capacity, poor rate performance due to their solubility in

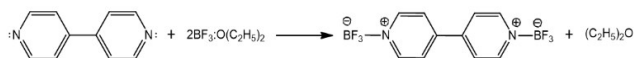
electrolytes and high electronic resistivity. Recently, Tarascon and co-workers provided a solution to overcome those problems by proposing conjugated carboxylic acid salts.¹³ Thus after, several organic electrode materials¹⁴⁻¹⁷ using conjugated carbonyl compounds, organosulfur compounds, electrochemically active radical polymers¹⁸⁻²⁰ etc. have been reported till the date.

In this paper, for the first time, a small organic compound of N, N-disubstituted- 4, 4'-bipyridine (commonly known as 'viologen') family, is described as an electrode material for lithium-ion batteries. Viologens are most often used as redox mediators,^{21,22} molecular electronics,²³ and potential electrochromic materials²⁴⁻²⁷ in commercially available smart windows and rear view mirrors. The electrochromic behavior of the viologen monomers is based on reversible interconversion between two redox states via 2e⁻ redox reaction, when externally negative electric potential is applied. Due to the excellent redox properties, the use of viologens are being shifted towards energy storage devices. Viologens has been used as charge carriers in a polymer based battery by Palmore *et al.*²⁸ Recently, poly(vinyl-benzyl ethyl viologen), a polymeric viologen has been reported as redox active polymer for non-aqueous redox flow batteries (NRFBS) by Nagarjuna *et al.*²⁹ Inspiring by these works, we synthesized a viologen monomer N, N-di(boron trifluoride)-4, 4'-bipyridine [or boron trifluoride viologen] (BTFV), comprising of a 4,4'-bipyridine core, sandwiched between two BF₃ molecules and used it as an electrode material for lithium ion battery. BTFV has a stable aromatic core, which offers a theoretical capacity of 183 mAh g⁻¹ (calculation is shown in ESI[†]) with respect to lithium, by following the same electrochemistry as in electrochromic devices and redox flow batteries. Hence, it could be an alternative electrode material for sustainable and cost-effective lithium-ion batteries.

Department of Energy Science and Engineering, IIT Bombay, Powai, Mumbai 400076, INDIA. Email: sagar.mitra@iitb.ac.in Tel.: + 91 22 2576 7849.

† Footnotes relating to the title and/or authors should appear here.

Electronic Supplementary Information (ESI) available: [details of any supplementary information available should be included here]. See DOI: 10.1039/x0xx00000x

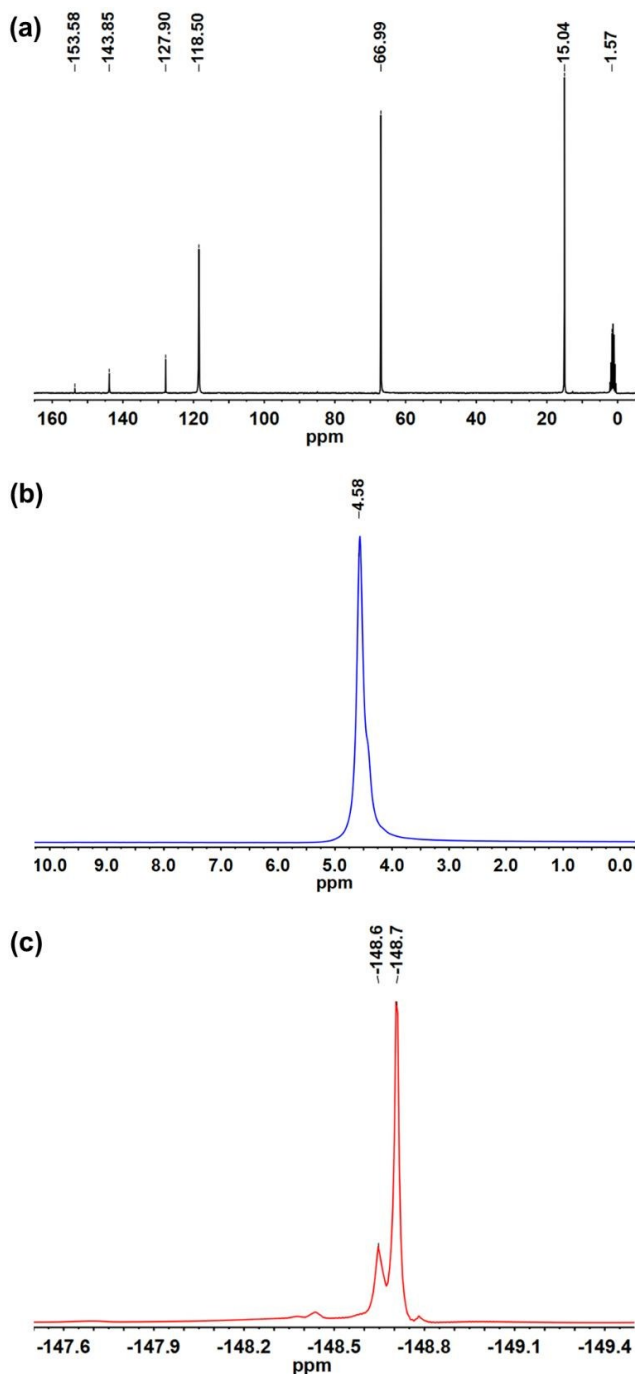


Scheme 1. Schematic representation of N, N-di-(boron trifluoride)-4, 4'-bipyridilium synthesis.

N, N-di-(boron trifluoride)-4, 4'-bipyridine (BTFV) was prepared via following steps (Scheme 1): firstly, a suspension of 4, 4'-bipyridine (1 mmol) was prepared in 10 ml diethyl ether. To this suspension, 2.2 mmol of boron trifluoride diethyl etherate $\text{BF}_3 \cdot \text{O}(\text{C}_2\text{H}_5)_2$ was added (excess etherated boron trifluoride was added to avoid any mono-quaternization of 4, 4'-bipyridine). Due to the lower electronegativity of nitrogen in comparison to oxygen, the ether molecules coordinately attached to BF_3 substituted by 4, 4'-bipyridine molecules and immediately a white colored precipitated was obtained due to instant Lewis acid-base adduct formation between BF_3 and 4, 4'-bipyridine. The colorless property of BTFV is due to absence of any charge transfer transition and could be beneficial for electrochromic applications. The precipitate was then filtrated and air-dried. The formation of product was confirmed by FTIR, ^{13}C -NMR, ^{11}B -NMR and ^{19}F -NMR. X-ray diffraction studies were performed for both 4, 4'-bipyridine and BTFV to understand the change of crystallinity of 4, 4'-bipyridine upon adduct formation. The morphology and microstructure of the product was understood by scanning electron microscopy (SEM). Energy dispersive X-ray (EDX) analysis was carried out for elemental analysis of the product.

The reversible redox property of BTFV was confirmed by cyclic voltammetry, carried out on an Autolab PGSTAT 302N electrochemical workstation, using three-electrode configuration with fluorine doped tin oxide (FTO) coated glass as working electrode, Pt-rod as counter electrode and $\text{Ag}/\text{AgCl}/\text{KCl}$ half-cell as reference electrode. The cyclic voltammetry experiment was performed with 20 mM solution of BTFV prepared in a 15 ml liquid electrolyte comprise of N, N-dimethylformamide (DMF) and 1-Ethyl-3-methylimidazolium bis(trifluoromethylsulfonyl) imide ionic liquid (2:1 v/v) at a scan rate of 5 mV s^{-1} . Furthermore, to investigate the preliminary battery performance of the viologen, another cyclic voltammetry was carried out against lithium metal in presence of lithium electrolyte. Working electrode was prepared using BTFV as active material, super P as conductive additive and polyethylene oxide (PEO) as binder with an overall ratio of 60:30:10 (by wt.). The slurry was prepared in acetonitrile and then blade cast onto a carbon coated copper foil. The active material loading was about 0.45 mg cm^{-2} . Swagelok-type cells are prepared using this organic electrode as working electrode with lithium foil as reference/counter electrode and borosilicate glass fiber sheet as a separator soaked with $50 \mu\text{L}$ of electrolyte comprising of 1 M LiPF_6 in 1:1 (v/v) ethylene carbonate (EC) and dimethyl carbonate (DMC). Cyclic voltammetry was carried out at a scan rate of $50 \mu\text{V s}^{-1}$ within the potential window of 0.01-3.0 V (vs. Li/Li^+) using Biologic VMP-3. The Galvanostatic charge-discharge experiments were performed using an Arbin BT-2000 within two different potential ranges of 0.01-3.0 V and 0.8-2.8 V vs. Li/Li^+ . All capacities were calculated based on the mass of BTFV in electrode.

FTIR spectra of N, N-di-(boron trifluoride)-4, 4'-bipyridine (BTFV) is shown in ESI[†] (Fig. S1). A characteristic peak is observed at 1050 cm^{-1} which is related to formation of nitrogen-boron co-ordination



bond ($\text{N} \rightarrow \text{B} \equiv \text{N}^+ - \text{B}^-$), while the peak intensity at around 1300 cm^{-1} (due to B-O bond) is found to decreased, indicating the replacement of coordinated diethyl ether molecules by 4, 4'-bipyridine. A medium and broad peak is observed around 1476 cm^{-1} corresponding to the aromatic C=C bond stretching. The band at 3199 cm^{-1} is due to aromatic hydrogens (Ar-H) attached to the bipyridine ring. The band at 781 cm^{-1} is assigned to aromatic C-H bending vibration.

^{13}C -NMR was taken (Fig. 1a) by dissolving the product (BTFV) in CD_3CN . The peaks at 1.57 ppm and 118.5 ppm are corresponding to the carbons present in $-\text{CD}_3$ and $-\text{CN}$ groups. The peaks at 15.04

Figure 1. (a) ^{13}C -NMR spectra, (b) ^{11}B -NMR spectra and (c) ^{19}F -NMR spectra of BTFV.

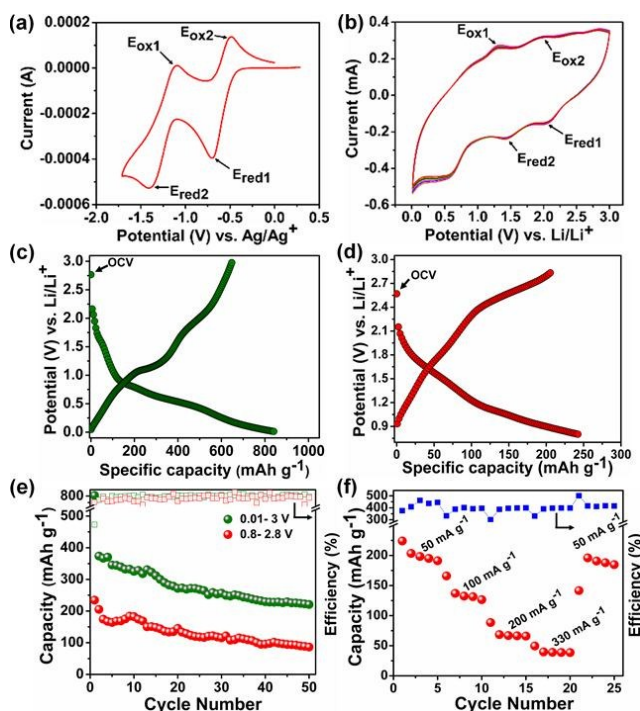


Figure 2. Cyclic voltammograms of BTFV (a) with respect to Ag/Ag⁺ at a scan rate of 5 mV s⁻¹ and (b) with respect to Li/Li⁺ at a scan rate of 0.05 mV s⁻¹. (c) First cycle charge-discharge profile of BTFV electrode within the potential range of 0.01-3.0 V. (d) First cycle charge-discharge profile of BTFV electrode within the potential range of 0.8-2.8 V. (e) Cycling performance of same at 50 mA g⁻¹ within different potential range. (f) Rate performance of BTFV at different current rates within the potential range of 0.8-2.8 V.

ppm (-OCH₂CH₃) and 66.99 ppm (-OCH₂CH₃) are because of residual amount of ether present in the product. The characteristic peaks of the product are assigned as follows: 127.9 ppm (the aromatic carbons next to the carbon directly attached with nitrogen), 143.85 ppm (aromatic carbons directly attached with nitrogen) and 153.58 ppm (the aromatic carbons opposite to nitrogen).

Boron NMR was taken to further confirm the formation of viologen. Fig. 1b shows the ¹¹B-NMR spectra of BTFV. It can be observed from the spectra that there is a significant peak shift from 0 ppm (for pure diethyl etherated BF₃³⁰) to 4.58 ppm, confirming the replacement of diethyl ether molecules and adduct formation between BF₃ and 4, 4'-bipyridine.

Fig. 1c shows the fluorine NMR spectrum of BTFV. Boron has two naturally occurring NMR active nuclei ¹⁰B and ¹¹B. Hence, two peaks were observed at -148.7 and -148.6 ppm, respectively. The strong peak at -148.7 ppm is due to the fluorine attached to ¹¹B (more sensitive) atom and the weak one at -148.6 ppm representing the fluorine attached to ¹⁰B. The significant peak shift from -153.2 ppm (pure diethyl etherated BF₃³⁰) to -148.7 indicates the replacement of coordinately bonded diethyl ether molecules by 4, 4'-bipyridine molecules. The XRD patterns of 4, 4'-bipyridine and BTFV showed a remarkable difference in crystallinity of 4, 4'-bipyridine upon adduct formation with boron trifluoride (Fig. S2)

The BTFV showed a very good redox property as it is clear from cyclic voltammogram (Fig. 2a) with respect to Ag/AgCl/KCl reference electrode ($E_{Ag^+/Ag} = 0.20$ V vs. SHE). The first reduction peak in the cathodic sweep is obtained at -0.71 V which is attributed to the reduction of the neutral N, N-di-(boron trifluoride)-4, 4'-bipyridine (BTFV⁰) to its corresponding radical-anion (BTFV^{-•}) form. Another peak is seen at relatively higher potential of -1.40 V, which corresponds to the further reduction leading to formation of di-anionic species (BTFV²⁻). On changing the sweep direction, the first and second anodic peaks were obtained at around -1.11 V and -0.47 V, respectively. The first anodic peak at -1.11 V indicates the oxidation of di-anionic viologen (BTFV²⁻) to radical-anion (BTFV^{-•}) and the second one implies the further oxidation to the neutral form (BTFV⁰). Furthermore, two pairs of approximately symmetrical sized cathodic and anodic peaks have almost same areas, suggesting good lithiation/de-lithiation reversibility for this organic electrode material. Following the same redox chemistry, a reversible two steps lithiation and delithiation process in BTFV is also observed in CV experiment against Li/Li⁺ (Fig. 2b). During reduction process, three prominent peaks are obtained. The peaks at around 2.15 V is related to the conversion of neutral BTFV to its radical-anion form and the following peak at 1.5 V is due to further reduction of radical-cation to di-anionic viologen as shown in ESI[†](Fig.S4). The peak at around 0.7-0.5 V is because of reversible insertion of Li⁺ ions into the carbon black.

The first cycle charge-discharge profile of BTFV electrode within the potential range of 0.01-3.0 V vs. Li/Li⁺ at the scan rate of 50 mA g⁻¹ is shown in Fig. 2c. The discharge profile exhibits three plateaus at around 2.15 V, 1.55 V and 0.7-0.5 V. This is in good agreement with the three reduction peaks in the CV operated between 0.01-3.0 V. The first cycle charge-discharge profile of BTFV electrode within the potential range of 0.8-2.8 V (vs. Li/Li⁺) at 50 mA g⁻¹ current rate exhibited only two discharge plateaus at around 2.15 V and 1.55 V (Fig. 2d). From Fig. 2e it can be observed that the specific capacity obtained from cycling performance operated between 0.01-3.0 V is much higher than that between 0.8-2.8 V. Since, the theoretical capacity of BTFV is 183 mAh g⁻¹, at least 160 mAh g⁻¹ comes from the reversible intercalation of Li⁺ ions into carbon at lower potential (0.5-0.7 V) and it is possible that at most 300 mAh g⁻¹ (= 600 mAh g⁻¹ × 30 wt% of super P/60 wt% BTFV) contribution from super P carbon as reported by C. W. Lee *et al.*³¹ Furthermore, during cycling between 0.01-3.0 V some irreversible capacity in the first discharge may be attributed to the rapid decomposition of electrolyte and formation of solid electrolyte interphase (SEI) at lower potential.

To further interrogate the electrochemical properties of the viologen, additional experiments to examine rate capability (Fig. 2f) and extended cycling performance (Fig. S7) were carried out. The rate performance has been carried out at 50, 100, 200 and 330 mAh g⁻¹ discharge rates and results are shown in Fig. 2f. The capacities at different rates found to be stable.

To interpret the reversible lithium storage property of this viologen, ex-situ NMR characterizations have been carried out. Two separate Swagelok cells are prepared with BTFV as the working electrode. First cell has been discharged to 0.8 V (vs. Li/Li⁺) and for the second cell first discharge to 0.8 V and then recharged to 2.8 V. Both the electrodes are washed properly in order to remove the electrolyte salt, if any. Then electrode materials are taken out from the current

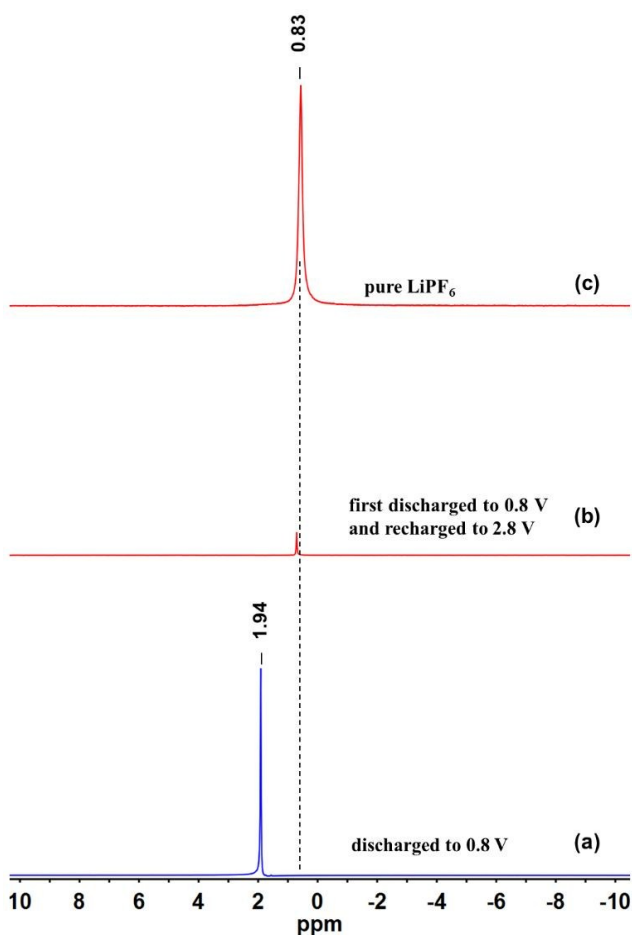


Figure 3. ^7Li -NMR spectrum collected in CD_3CN . (a) Lithiated BTFV where the cell was discharged from open circuit voltage to 0.8 V. (b) De-lithiated BTFV where the cell was first discharged from OCV to 0.8 V and then recharged to 2.8 V. (c) Pure LiPF_6 salt.

collector and dried at room temperature. All of these steps are carried out inside the argon-filled glove box. The electrode materials are then separately centrifuged in acetonitrile- d_3 (CD_3CN). A little volume of supernatant liquid was taken for NMR characterizations. Fig. 3 shows ^7Li -NMR spectrum for both the electrode materials. The electrode material discharged to 0.8 V exhibits a strong peak at 1.94 ppm (Fig. 3a) corresponds to lithiated BTFV. While for the electrode material which is first discharged to 0.8 V and then recharged to 2.8 V, the peak at 1.94 ppm get disappeared and a very weak peak is observed at 0.83 ppm (Fig. 3b) due to the minute amount of unwashed LiPF_6 salt. The disappearance of lithium peak at 1.94 ppm indicates the delithiation of BTFV.

In summary, an organic material of N, N-disubstituted-4, 4'-bipyridine (viologen) category synthesized in a very simple way. The viologen shows a good capacity retention and good rate capability at different current rates with respect to lithium and hence it can be an alternative electrode material for sustainable organic lithium ion batteries.

This work is financially supported by National Centre for Photovoltaic Research and Education (NCPRE), IIT Bombay. The authors are thankful to SAIF (IIT Bombay) for instrument facility. Author A. Ghosh would like to thank Sudeep Sarkar and Ananta Sarkar for their help to NMR characterization.

Notes and References

- 1 J. M. Tarascon and M. Armand, *Nature*, 2001, **414**, 359.
- 2 B. Dunn, H. Kamath and J. M. Tarascon, *Science*, 2011, **334**, 928.
- 3 C. M. Julien, A. Mauger, K. Zaghib and H. Groult, *Inorganics*, 2014, **2**, 132.
- 4 B. Scrosati and J. Garche, *J. Power Sources*, 2010, **195**, 2419.
- 5 S. Sarkar, H. Banda and S. Mitra, *Electrochim. Acta*, 2013, **99**, 242.
- 6 V. Etacheri, R. Marom, R. Elazari, G. Salitra and D. Aurbach, *Energy Environ. Sci.*, 2011, **4**, 3243.
- 7 A. M. Tripathi and S. Mitra, *ChemElectroChem*, 2014, **1**, 1327.
- 8 D. Larcher and J. M. Tarascon, *Nat. Chem.*, 2014, **7**, 19.
- 9 Z. Song and H. Zhou, *Energy Environ. Sci.*, 2013, **6**, 2280.
- 10 A. L. Goodwin, *Nat. Mater.*, 2010, **9**, 7.
- 11 M. Yao, H. Sano, H. Ando and T. Kiyobayashi, *Sci. Rep.*, 2015, **5**, 10962.
- 12 D. L. Williams, J. J. Byrne and J. S. Driscoll, *J. Electrochem. Soc.*, 1969, **116**, 2.
- 13 M. Armand, S. Grugeon, H. Vezin, S. Laruelle, P. Ribière, P. Poizot and J.-M. Tarascon, *Nat. Mater.*, 2009, **8**, 120.
- 14 Y. Liang, Z. Tao and J. Chen, *Adv. Energy Mater.*, 2012, **2**, 742.
- 15 L. M. Zhu, A. W. Lei, Y. L. Cao, X. P. Ai and H. X. Yang, *Chem. Commun.*, 2013, **49**, 567.
- 16 H. Chen, M. Armand, G. Demailly, F. Dolhem, P. Poizot and J.-M. Tarascon, *ChemSusChem*, 2008, **1**, 348.
- 17 S. Goriparti, M. N. K. Harish and S. Sampath, *Chem. Commun.*, 2013, **49**, 7234.
- 18 P. Novák, K. Müller, K. S. V. Santhanam and O. Haas, *Chem. Rev.*, 1997, **97**, 207.
- 19 X. Han, C. Chang, L. Yuan, T. Sun and J. Sun, *Adv. Mater.*, 2007, **19**, 1616.
- 20 T. Janoschka, M. D. Hager and U. S. Schubert, *Adv. Mater.*, 2012, **24**, 6397.
- 21 D. M. Scott, T. H. Tsang, L. Chetty, S. Aloji and B. Y. Liaw, *J. Power Sources*, 2011, **196**, 10556.
- 22 M. J. Lacey, J. T. Frith and J. R. Owen, *Electrochem. Commun.*, 2013, **26**, 74.
- 23 P. M. S. Monk, *J. Wiley & Sons*, 1998.
- 24 R. J. Mortimer, *Electrochim. Acta*, 1999, **44**, 2971.
- 25 R. J. Mortimer, A. L. Dyer and J. R. Reynolds, *Displays*, 2006, **27**, 2.
- 26 R. J. Mortimer and T. S. Varley, *Sol. Energy Mater. Sol. Cells*, 2012, **99**, 213.
- 27 R. Sydam, A. Ghosh and M. Deepa, *Org. Electron.*, 2015, **17**, 33.
- 28 S. Sen, J. Saraidaridis, S. Y. Kim and G. T. R. Palmore, *ACS Appl. Mater. Interfaces*, 2013, **5**, 7825.
- 29 G. Nagarjuna, J. Hui, K. J. Cheng, T. Lichtenstein, M. Shen, J. S. Moore and J. R. López, *J. Am. Chem. Soc.*, 2014, **136**, 16309.
- 30 E. L. Myers, C. P. Butts and V. K. Aggarwal, *Chem. Commun.*, 2006, 4434.

Journal Name

COMMUNICATION

31 R. M. Gnanamuthu and C. W. Lee, *Mater. Chem. Phys.*, 2011, **130**, 831.

RSC Advances Accepted Manuscript