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# Synthesis of Ag<sub>2</sub>CO<sub>3</sub>/Bi<sub>2</sub>WO<sub>6</sub> heterojunctions with enhanced photocatalytic activity and cycling stability

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## Abstract

Hierarchical Bi<sub>2</sub>WO<sub>6</sub> nanoarchitectures with scale of 2-3 μm were prepared *via* a facile microwave-assisted solution-phase reaction process. Monodispersed spherical Ag<sub>2</sub>CO<sub>3</sub> nanoparticles with an average size of about 10 nm were deposited onto the surface of Bi<sub>2</sub>WO<sub>6</sub> nanoarchitectures to form a novel Ag<sub>2</sub>CO<sub>3</sub>/Bi<sub>2</sub>WO<sub>6</sub> heterojunction structure through a facile *in situ* precipitation-deposition method. The obtained samples were characterized by XRD, XPS, SEM, TEM (HRTEM), UV-vis DRS and nitrogen adsorption-desorption techniques. The photocatalytic evaluation displays the decoration of Ag<sub>2</sub>CO<sub>3</sub> nanoparticles significantly enhanced the photocatalytic activity of Bi<sub>2</sub>WO<sub>6</sub> and the photocatalytic performance is greatly influenced by the content of deposited Ag<sub>2</sub>CO<sub>3</sub>. The 30 wt% Ag<sub>2</sub>CO<sub>3</sub>-loaded Bi<sub>2</sub>WO<sub>6</sub> sample exhibited the highest photocatalytic activity for degradation of Rhodamine B (RhB) under visible light irradiation. Meanwhile, it possess also excellent cycling stability and superior photocatalytic performance toward other pollutants. The dramatic enhanced photocatalytic activity and stability can be mainly ascribed to well-matched energy band and heterojunction relative between Ag<sub>2</sub>CO<sub>3</sub> and Bi<sub>2</sub>WO<sub>6</sub>, which can effective improve the separation of photo-induced electron-hole at the heterojunctional interfaces.

**Keywords:** Ag<sub>2</sub>CO<sub>3</sub>/Bi<sub>2</sub>WO<sub>6</sub>; Heterojunction structure; Visible light irradiation; Photocatalysis

## 1. Introduction

Heterogeneous photocatalysts have attracted tremendous attention due to their great potential for decomposing organic pollutants and for converting photon energy into chemical energy.<sup>1-4</sup> Among the novel photocatalysts, Bi<sub>2</sub>WO<sub>6</sub> have been proven to be promising catalysts for photo-chemical reactions owing to their suitable bandgap (~2.8 eV) and strong oxidizing power.<sup>5-7</sup> It is widely accepted that the photocatalytic performance of catalytic is very closely related to their surface structure and crystallinity including morphology, size, specific surface, crystal planes and defect.<sup>8-10</sup> Bi<sub>2</sub>WO<sub>6</sub> as a layer-structured semiconductor, constructed by cornersharing WO<sub>6</sub> octahedra (WO<sub>4</sub>)<sup>2-</sup> sheets and bismuth oxide (Bi<sub>2</sub>O<sub>2</sub>)<sup>2+</sup> sheets,<sup>11</sup> could be more easily obtained three-dimensional (3D) hierarchical architectures assembling by one-dimensional (1D) nanosheets. Many studies found that 3D hierarchical architectures could effectively increase the specific surface to volume ratio, organic pollutant adsorption and surface active sites.<sup>12,13</sup>

As is well known, excepting for adsorption behavior and photoresponse range of photocatalysts, as an ideal photocatalysts the separation efficiency and increase the oxidation power of photogenerated carriers is another vital consideration. Nonetheless, the bare Bi<sub>2</sub>WO<sub>6</sub> photocatalysts has high recombination of photoinduced electron-hole (e<sup>-</sup>-h<sup>+</sup>) pairs which leads to the low quantum efficiency.<sup>14,15</sup> Recent, numerous attempts have been made to prolong the lifetime of photogenerated e<sup>-</sup>-h<sup>+</sup> and promote carriers transport processes of Bi<sub>2</sub>WO<sub>6</sub>, including impurity doping,<sup>16,17</sup> noble metals sensitization,<sup>18-20</sup> and heterojunction structure formation.<sup>21-28</sup>

Coupling of two semiconducting structures with matching energy levels to form a heterojunction can effectively improve absorb and utilizes to visible light. More importantly, the nanojunction system can remarkably change the electrical properties of composite semiconductors, resulting in the enhanced separation, transport and oxidation (or reduction) power of photogenerated carriers.<sup>29,30</sup> Thus, coupling of

$\text{Bi}_2\text{WO}_6$  with other semiconductors have been widely employed to improve the photocatalytic performance of  $\text{Bi}_2\text{WO}_6$ . For *e.g.*  $\text{AgBr}/\text{Bi}_2\text{WO}_6$ ,  $\text{Ag}_2\text{S}/\text{Bi}_2\text{WO}_6$ ,  $\text{BiOBr}/\text{Bi}_2\text{WO}_6$ ,  $\text{TiO}_2/\text{Bi}_2\text{WO}_6$ ,  $\text{Bi}_2\text{S}_3/\text{Bi}_2\text{WO}_6$  and  $\text{g-C}_3\text{N}_4/\text{Bi}_2\text{WO}_6$  systems have been indicated as effectual in suppress the recombination of photogenerated  $\text{e}^-$ - $\text{h}^+$  pairs.<sup>21-28</sup>

Recently,  $\text{Ag}_2\text{CO}_3$  are regarded as a new high efficiency visible light photocatalysis material. Compared with the well-known N-doped  $\text{TiO}_2$ , the  $\text{Ag}_2\text{CO}_3$  usually shows obviously excellent utilization rate of visible light and photocatalysis activity.<sup>31,32</sup> Although,  $\text{Ag}_2\text{CO}_3$  is unstable in pure crystal form owing to its self-photocorrosion, previous studies suggested that  $\text{Ag}_2\text{CO}_3$  could maintain its stability through growth heterostructured composites. Yu et al.<sup>33</sup> reported that the  $\text{Ag}_2\text{O}/\text{Ag}_2\text{CO}_3$  heterostructured photocatalysts *via* phase transformation routes. This result indicates that the interface heterostructure effectively facilitates charge transfer and suppresses the recombination of photogenerated  $\text{e}^-$ - $\text{h}^+$ , resulting in exhibits extremely high activity and stability. Similarly, the synthesis of  $\text{Ag}_2\text{CO}_3$  composite photocatalysts have been reported, such as  $\text{Ag}_2\text{CO}_3/\text{g-C}_3\text{N}_4$ ,  $\text{Ag}_2\text{CO}_3/\text{AgX}$ ,  $\text{Ag}_2\text{CO}_3/\text{Ag}/\text{AgBr}$ , and their visible-light-driven photocatalytic activity and stability have also been explored.<sup>34-36</sup> Herein, we intend to design a novel  $\text{Ag}_2\text{CO}_3/\text{Bi}_2\text{WO}_6$  heterojunction photocatalyst with matching energy levels. This aim is to achieve improvement the visible-light photocatalytic activity of  $\text{Bi}_2\text{WO}_6$  and overcome  $\text{Ag}_2\text{CO}_3$  self-photocorrosion disadvantage.

Moreover, the microwave-assisted approach has been accepted as a promising method for the preparation of  $\text{Bi}_2\text{WO}_6$ .<sup>37</sup> Compared with traditional hydrothermal synthetic methods, microwave-assisted synthesis is simple, rapid, uniform, efficient, economical, and environmentally friendly.<sup>38</sup> Herein, in this work, we firstly prepared the  $\text{Bi}_2\text{WO}_6$  microspheres using a facile microwave-assisted solution-phase reaction process. Then, monodispersed  $\text{Ag}_2\text{CO}_3$  nanoparticles were deposited onto the surface of  $\text{Bi}_2\text{WO}_6$  nanoarchitectures to form a novel  $\text{Ag}_2\text{CO}_3/\text{Bi}_2\text{WO}_6$  heterojunction. As expected, as prepared the photocatalysts exhibited excellent photocatalytic activity and outstanding cycling performance for degradation the rhodamine B (RhB) dye

under visible light irradiation. Meanwhile, they also possess extremely high photocatalytic activities for other pollutants such as methyl orange (MO), methyl blue (MB), congo red (CR), and hexavalent chromium (Cr (VI) ). Furthermore, the possible mechanism of enhanced photocatalytic activity and stability for  $\text{Ag}_2\text{CO}_3/\text{Bi}_2\text{WO}_6$  heterojunctions was also discussed in detail.

## 2. Experiment section

### 2.1 Catalysts Synthesis

All the chemicals were analytic grade purity and used without further treatment. The hierarchical  $\text{Bi}_2\text{WO}_6$  microspheres were prepared using a MARS 5 Microwave System (Matthews, NC, USA). In the system, the parameters of time and temperature of reaction are regulated by a user-defined computer program. The automatic temperature-control system allowed continuous monitoring and control of the internal temperature of the reaction system by continuously adjusting the applied microwave power. In a typical preparation procedure,  $\text{Bi}(\text{NO}_3)_3 \cdot 5\text{H}_2\text{O}$  (3.5 mmol) were dissolved in 5 mL acetic acid (36wt%) to form a clear solution by magnetic stirring, then 30 mL aqueous solution of  $\text{Na}_2\text{WO}_4 \cdot 2\text{H}_2\text{O}$  (2 mmol) was dropwise added to the obtained solution. The mixture was allowed to stir for another 1 h, then transferred to a 50 mL microwave (MW) teflon container and kept at 200 °C for 6 h.

For the synthesis of  $\text{Ag}_2\text{CO}_3/\text{Bi}_2\text{WO}_6$  composites, the as-synthesized  $\text{Bi}_2\text{WO}_6$  (0.5 mmol) powders were ultrasonic dispersed into 20 mL of deionized water. Subsequently,  $\text{AgNO}_3$  solution (0.05 M) and  $\text{NH}_3 \cdot \text{H}_2\text{O}$  solution (0.1 M) were added dropwise to the above suspension in sequence under magnetic stirring. At last,  $\text{NaHCO}_3$  solution (0.05M) was added into the the mixture with a rate of less than  $0.3 \text{ mL min}^{-1}$ . The precipitate was collected by centrifugation, washed with deionized water and ethanol, and dried at 60 °C for 12 h. The  $\text{Ag}_2\text{CO}_3/\text{Bi}_2\text{WO}_6$  composites with different mass ratios were fabricated by changing the added amount of  $\text{AgNO}_3$ ,  $\text{NH}_3 \cdot \text{H}_2\text{O}$  and  $\text{NaHCO}_3$  solutions under the  $\text{AgNO}_3/\text{NH}_3 \cdot \text{H}_2\text{O}/\text{NaHCO}_3$  volume ratios of 1:1:2. The samples are denoted as A-BWO-1, A-BWO-2, A-BWO-3 and A-BWO-4 when the weight percentages of  $\text{Ag}_2\text{CO}_3$  in  $\text{Ag}_2\text{CO}_3/\text{Bi}_2\text{WO}_6$  composites are 10%, 20%, 30%, 40%,

respectively. The pure  $\text{Ag}_2\text{CO}_3$  were prepared by the same method without the addition of  $\text{Bi}_2\text{WO}_6$ . Moreover, the N-doped  $\text{TiO}_2$  was obtained according to the literature reported.<sup>39</sup>

## 2.2 Characterization

The X-ray diffraction (XRD) datas were recorded on a Bruker D8 X-ray diffractometer with  $\text{Cu-K}\alpha$  radiation source ( $\lambda=1.5406 \text{ \AA}$ ). Scanning electron microscopy (SEM) images were taken using a JSM-6701F scanning electron microscope with an accelerating voltage of 15 kV. Energy Dispersive X-ray (EDX) analysis was also performed on the JSM-6701F. Transmission electron microscopy (TEM) images were obtained on a JEM-2100 electron microscope at an accelerating voltage of 200 kV. The X-ray photoelectron spectroscopy (XPS) analysis was recorded though an AXIS-ULTRA DLD-600W photoelectron spectrometer with Al K1 radiation. The UV-vis diffuse reflectance spectroscopy (DRS) obtained using Hitachi U-4100 spectrophotometer. Nitrogen adsorption-desorption isotherms were collected on an Autosorb-iQ sorption analyzer and analyzed followed by the Brunauer-Emmett-Teller (BET) equation and Barret-Joyner-Halenda (BJH) model.

## 2.3 Photocatalytic activity test

The photocatalytic performance of the as-prepared samples was evaluated by measuring the degradation of RhB. In a typical photocatalytic measurement, the samples (0.1 g) were put into a solution of RhB dyes (100 mL, 10 mg/L), which was then irradiated with a 300W Xe arc lamp to provide visible light with  $\lambda \geq 420 \text{ nm}$  by an cutoff filter. Before the suspensions were irradiated they were magnetically stirred for 60 min in the dark to complete the adsorption-desorption equilibrium between dyes and photocatalysts. The degradation results were analyzed by using the UV-vis spectrophotometer (UV-2550, Shimadzu, Japan). Before the spectroscopy measurement, these photocatalysts were removed from the photocatalytic reaction systems by centrifugation. The same test for the RhB was also carried out on MB, MO and CR. The Cr (VI) was tested using the diphenylcarbazide (DPC) method introduced by Clesceri et al.<sup>40</sup>

### 3. Results and discussion

#### 3.1 Structure and property analysis

The structural characteristics and phase composition of samples were investigated by powder XRD. As shown in Fig. 1, the diffraction peaks of pure  $\text{Bi}_2\text{WO}_6$  and  $\text{Ag}_2\text{CO}_3$  samples can be match well with the standard crystalline structure of the orthorhombic phase of  $\text{Bi}_2\text{WO}_6$  (JCPDS no. 73-1126) and the monoclinic phase of  $\text{Ag}_2\text{CO}_3$  (JCPDS no. 70-2184), respectively. For  $\text{Ag}_2\text{CO}_3/\text{Bi}_2\text{WO}_6$  heterojunctions samples, excepting for all the diffraction peaks of  $\text{Bi}_2\text{WO}_6$ , the peaks at  $2\theta$  of  $18.6^\circ$ ,  $20.5^\circ$ ,  $32.6^\circ$ ,  $33.6^\circ$ ,  $37.1^\circ$ ,  $39.6^\circ$ ,  $44.3^\circ$ , and  $51.3^\circ$  are attributed to the pure monoclinic phase of  $\text{Ag}_2\text{CO}_3$ . Meanwhile, the intensities of corresponding diffraction peaks of  $\text{Ag}_2\text{CO}_3$  strengthened gradually along with the increase of the  $\text{Ag}_2\text{CO}_3$  content in  $\text{Ag}_2\text{CO}_3/\text{Bi}_2\text{WO}_6$  composites. Metallic Ag or impurity peaks was not detected from any of the samples, indicating that  $\text{Ag}_2\text{CO}_3/\text{Bi}_2\text{WO}_6$  composites are only composed of  $\text{Ag}_2\text{CO}_3$  and  $\text{Bi}_2\text{WO}_6$  phases.

The elemental composition and oxidation states of the as-prepared samples were further analyzed by XPS. Fig. 2a shows XPS survey spectra of pure  $\text{Bi}_2\text{WO}_6$ ,  $\text{Ag}_2\text{CO}_3$  and A-BWO-3 composite. The overview spectrum of composite demonstrates that Bi, W, O, C and Ag exist, further confirming that the sample was composed of  $\text{Bi}_2\text{WO}_6$  and  $\text{Ag}_2\text{CO}_3$ . Fig. 2b-f show the high-resolution spectra of Bi 4f, W 4f, O 1s, C 1s and Ag 3d regions for the A-BWO-3 sample, respectively. The peaks at binding energies ( $E_b$ ) of 163.3 and 158.0 eV (Fig. 2b) belong to Bi 4f<sub>5/2</sub> and Bi 4f<sub>7/2</sub> of  $\text{Bi}^{3+}$  ions,<sup>41</sup> which shift toward low binding energy compared with that of pure  $\text{Bi}_2\text{WO}_6$  (164.7 and 159.4 eV, Fig. S2 ). The appearance was also found in the XPS spectra of W 4f and O 1s. Such results could be ascribed to the interaction between  $\text{Bi}_2\text{WO}_6$  and  $\text{Ag}_2\text{CO}_3$  resulting in an inner shift of Bi 4f, W 4f and O 1s orbits.<sup>42</sup> The W 4f<sub>5/2</sub> and W 4f<sub>7/2</sub> peaks (Fig. 2c) located at 36.3 and 34.2 eV with a spin-orbit separation of 2.1 eV, which corresponds to  $\text{W}^{6+}$  according to previous reports.<sup>43</sup> The wide and asymmetric XPS of O 1s peaks (Fig. 2d) can be deconvoluted into four peaks, the peaks at  $E_b$  of 528.9, 529.7 and 530.5 eV correspond to Bi-O, W-O and crystal lattice oxygen of

Ag<sub>2</sub>CO<sub>3</sub>, respectively.<sup>44,45</sup> The weak peak at 531.7 eV correspond to surface absorbed oxygen species.<sup>46</sup> In Fig. 2e, the C 1s spectrum were deconvoluted into three peaks: the peaks at  $E_b$  of 283.6 eV could be attributed to carbon element in Ag<sub>2</sub>CO<sub>3</sub>, which is similar to what was reported by Dong et al,<sup>47</sup> and the rest of two peaks were mainly ascribed to the adventitious hydrocarbon from XPS itself. Fig. 2f gives the high-resolution XPS spectrum of Ag 3d. The peaks at  $E_b$  of 366.9 and 372.9 eV correspond to the Ag 3d<sub>5/2</sub> and Ag 3d<sub>3/2</sub> of Ag<sup>+</sup>, respectively.<sup>22</sup> It is clear suggesting that no metal Ag is formed, which is in accordance with the result of XRD analysis. Therefore, by combining XPS and XRD investigation, the results confirmed that there were both Bi<sub>2</sub>WO<sub>6</sub> and Ag<sub>2</sub>CO<sub>3</sub> species in the Ag<sub>2</sub>CO<sub>3</sub>/Bi<sub>2</sub>WO<sub>6</sub> composite samples.

The morphology of the as-synthesized samples was examined by SEM, as shown in Fig. 3. It can be clearly seen that the bare Bi<sub>2</sub>WO<sub>6</sub> is uniformly dispersed and presents 3D hierarchical architectures with a diameter of 2-3 μm (Fig. 3a). When Ag<sub>2</sub>CO<sub>3</sub> was deposited onto 3D hierarchical Bi<sub>2</sub>WO<sub>6</sub> microspheres via a facile precipitation-deposition process (Fig. 3b-e), the resulting Ag<sub>2</sub>CO<sub>3</sub>/Bi<sub>2</sub>WO<sub>6</sub> composite exhibits a similar morphology and size as compared to that of bare Bi<sub>2</sub>WO<sub>6</sub>. With increasing Ag<sub>2</sub>CO<sub>3</sub> content, it is easy to observe that the large amount Ag<sub>2</sub>CO<sub>3</sub> nanoparticles were dispersed uniformly onto the surface of hierarchical Bi<sub>2</sub>WO<sub>6</sub> microspheres. Furthermore, the EDX spectrum of A-BWO-3 composite is shown in Fig. 3f. The composite is only contain Bi, W, O, C and Ag elements except for the elements of Au from the supports, which is well consistent with the XRD and XPS results.

Further information regarding the microstructure of prepared samples was obtained from TEM and HRTEM images. Fig. 4a and b show TEM images of the bare Bi<sub>2</sub>WO<sub>6</sub> with different magnifications. It is easy to observe that the hierarchical Bi<sub>2</sub>WO<sub>6</sub> microspheres with a zigzag brim, which is in fact assembled by several laminar Bi<sub>2</sub>WO<sub>6</sub> nanoplates with quadrate shape. Specifically, each quadrate shape nanoplate has a relative smooth surface with an average length of about 60 nm and a width of



about 40 nm. Fig. 4c and d show TEM images of the A-BWO-3 composite. The locations of  $\text{Ag}_2\text{CO}_3$  nanoparticles on the surface of  $\text{Bi}_2\text{WO}_6$  nanoplates are indicated by arrows in TEM (Fig. 4d). It displays that some monodispersed spherical nanoparticles with size of about 10 nm covered the surface of  $\text{Bi}_2\text{WO}_6$  nanoplates. Fig. 4e is the corresponding selected area electron diffraction (SAED) pattern of composite, which shows several diffraction rings, indicating that the nanoplates are polycrystalline structure. The diffraction fringes can be indexed to (113), (006), (206) and (313) planes for the orthorhombic crystal structure of  $\text{Bi}_2\text{WO}_6$ , which is in accordance with XRD results. HRTEM (Fig. 4f) shows that  $\text{Ag}_2\text{CO}_3$  nanoparticles tightly adhere to the surfaces of  $\text{Bi}_2\text{WO}_6$  nanoplates. The lattice fringe spacings of 0.315 and 0.266 nm correspond to the (113) plane of orthorhombic  $\text{Bi}_2\text{WO}_6$  and the (130) plane of monoclinic  $\text{Ag}_2\text{CO}_3$ , respectively. Thus, the above results indicate the formation of heterojunctions between  $\text{Ag}_2\text{CO}_3$  and  $\text{Bi}_2\text{WO}_6$ .

Fig. 5 shows the fabrication method of the  $\text{Ag}_2\text{CO}_3/\text{Bi}_2\text{WO}_6$  heterojunction photocatalysts via a three-step process. Firstly,  $\text{Bi}_2\text{WO}_6$  hierarchical microspheres obtained by a facile microwave-assisted solution-phase reaction process. Afterwards, the  $\text{AgNO}_3$  solution and  $\text{NH}_3\cdot\text{H}_2\text{O}$  were added to the  $\text{Bi}_2\text{WO}_6$  suspension. The formation of  $\text{Ag}(\text{NH}_2)^{2+}$  can be bound tightly to the surface of hierarchical  $\text{Bi}_2\text{WO}_6$  microspheres due to chemical adsorption. After that, with the addition  $\text{NaHCO}_3$ ,  $\text{Ag}(\text{NH}_2)^{2+}$  ions attached to the surface of  $\text{Bi}_2\text{WO}_6$  react with  $\text{HCO}_3^-$  to generate  $\text{Ag}_2\text{CO}_3$  nanoparticles. Furthermore, added  $\text{NH}_3\cdot\text{H}_2\text{O}$  is an important influencing factor for the regulation of  $\text{Ag}_2\text{CO}_3$  grain size and formation of nano-heterojunctions. In the absence of  $\text{NH}_3\cdot\text{H}_2\text{O}$ , a large number of short nanorods of  $\text{Ag}_2\text{CO}_3$  are existing on the brim of  $\text{Bi}_2\text{WO}_6$  nanoplates (Fig. S3).

Fig. 6 displays the nitrogen adsorption-desorption isotherms and the corresponding pore size distribution curves of the as-prepared  $\text{Bi}_2\text{WO}_6$  and A-BWO-3 heterojunction photocatalysts. The isotherm of samples can be categorized as a type IV isotherm and an H3-type hysteresis loop ( $0.75 < P/P_0 < 0.99$ ), suggesting the presence of mesoporous structures with silt-like pores in these two samples.<sup>48</sup>

According to the fitting analysis with the BET equation, the surface area of hierarchical  $\text{Bi}_2\text{WO}_6$  microspheres is about  $19.2 \text{ m}^2\text{g}^{-1}$ . After decoration with  $\text{Ag}_2\text{CO}_3$  nanoparticles, the heterojunction photocatalysts still possesses a relatively large surface area of  $16.8 \text{ m}^2\text{g}^{-1}$ . Furthermore, the main pore size distribution (inset of Fig. 6) in these two samples is similarly in the range of 10-15 nm, thus implying the  $\text{Ag}_2\text{CO}_3$  nanoparticles only covered the surface of  $\text{Bi}_2\text{WO}_6$  and the pore size was retained well.

The light absorption properties of the as-prepared samples were investigated by UV-visible diffuse reflectance spectra. As shown in Fig. 7, the absorption spectrum of pure  $\text{Bi}_2\text{WO}_6$  extends from UV region to visible light at about 450 nm. Compared with pure  $\text{Bi}_2\text{WO}_6$ , the  $\text{Ag}_2\text{CO}_3/\text{Bi}_2\text{WO}_6$  heterojunctions shows a significant redshift of absorption edge and displays surprisingly strong absorption around 450-800 nm. This observation clearly indicates that the  $\text{Ag}_2\text{CO}_3/\text{Bi}_2\text{WO}_6$  heterojunctions show more intensive absorption within the visible light range. This could be extremely advantageous for the degradation of pollutants owing to the light harvest ability great impact to the photocatalytic reaction process. Furthermore, it could be seen that the improvement of visible light absorption intensity is unobvious with increasing  $\text{Ag}_2\text{CO}_3$  content. It might be because the excess  $\text{Ag}_2\text{CO}_3$  will impede the light contacting with  $\text{Bi}_2\text{WO}_6$  crystal and inhibit interfacial transport of electron-hole between  $\text{Ag}_2\text{CO}_3$  and  $\text{Bi}_2\text{WO}_6$ .<sup>49</sup> In addition, the optical band gaps was calculated following the equation:<sup>50</sup>

$$\alpha h\nu = A(h\nu - E_g)^{n/2}$$

where  $\alpha$ ,  $h$ ,  $\nu$ ,  $A$  and  $E_g$  are absorption coefficient, planks constant, frequency, proportionality constant and bandgap, respectively. And  $n$  value is determined by the type of transition ( $n$  value is 4 for  $\text{Ag}_2\text{CO}_3$  and  $\text{Bi}_2\text{WO}_6$  with indirect transition).<sup>19,33,51</sup> A plot of  $(\alpha h\nu)^{1/2}$  versus  $(h\nu)$  gives the bandgap as shown in the insets of Fig. 7. The bandgap of  $\text{Ag}_2\text{CO}_3$  and  $\text{Bi}_2\text{WO}_6$  estimated from the intercept of the tangent to the plot are 2.17 and 2.74 eV, respectively. The bandgap of as-prepared pure  $\text{Ag}_2\text{CO}_3$  is lower than that of  $\text{Ag}_2\text{CO}_3$  crystals ( $\sim 2.30$  eV) obtained by reaction

of  $\text{NaHCO}_3$  with  $\text{AgNO}_3$ .<sup>33,34</sup> It could be attributed to the different the preparation method and phase composition of  $\text{Ag}_2\text{CO}_3$ .

### 3.2 Photocatalytic activity

The photocatalytic performance of as-prepared samples was evaluated by the degradation of RhB and MO under visible light irradiation. For comparison, the respective performance of P25 and prepared N-doped  $\text{TiO}_2$  (N- $\text{TiO}_2$ ) photocatalysts was also presented and the results are given in Fig. 8. As shown in Fig. 8 a, the RhB can only be slightly degraded under visible light irradiation without catalysts, indicating that the self-degradation effect of RhB could be ignored. Compared to pure  $\text{Ag}_2\text{CO}_3$  and  $\text{Bi}_2\text{WO}_6$ , the  $\text{Ag}_2\text{CO}_3/\text{Bi}_2\text{WO}_6$  heterojunctions can effectively improve the photocatalytic activities. The highest activity is observed in A-BWO-3 sample, and the degradation rate to RhB can exceed 95% within 60 min. Additionally, it is noted that the A-BWO-3 heterojunction photocatalysts exhibit remarkably superior photocatalytic activity under the same conditions compared to P25 and N- $\text{TiO}_2$ . Fig. 8b shows the absorption spectral changes of RhB solution with the temporal evolution. The major absorption peaks of RhB around 553 nm diminished gradually with the increasing visible light irradiation time in the presence of the A-BWO-3. Meanwhile, the suspension losses of color gradually, which further indicates that the structure of RhB have been destroyed. The pseudo-first-order reaction kinetic of the photocatalytic for the degradation of RhB solution was investigated and the results are given in Fig. S4 and Table 1. The rate constant  $k$  is  $0.0489 \text{ min}^{-1}$  for A-BWO-3 heterojunction photocatalysts, which is much higher than those of P25 ( $0.0014 \text{ min}^{-1}$ ), N- $\text{TiO}_2$  ( $0.0124 \text{ min}^{-1}$ ), pure  $\text{Ag}_2\text{CO}_3$  ( $0.0195 \text{ min}^{-1}$ ) and pure  $\text{Bi}_2\text{WO}_6$  ( $0.0272 \text{ min}^{-1}$ ). This results indicated RhB could be degraded more efficiently by A-BWO-3 heterojunction photocatalysts.

Furthermore, MO was also selected as the other typical target compound to further evaluate the photocatalytic activity of as-prepared samples. As shown in Figure 8c and d, the A-BWO-3 is decomposed about 95.2% of MO after 50 min under visible light irradiation. For comparison, only 4.7%, 13.2%, 43.1%, and 54.8% of MO is

removed under the same conditions for pure  $\text{Bi}_2\text{WO}_6$ , P25, N-TiO<sub>2</sub> and pure  $\text{Ag}_2\text{CO}_3$ , respectively. The photodegradation process of MO was found to follow pseudo-first-order kinetics and the rate constant over the A-BWO-3 heterojunction photocatalysts is higher than the sum of pure  $\text{Bi}_2\text{WO}_6$  and pure  $\text{Ag}_2\text{CO}_3$  (Fig. S4 and Table 1). The photocatalytic decomposition rate constant of A-BWO-3 over MO is 35 and 5 times over P25 and N-TiO<sub>2</sub>. Moreover, the data, showed in Table 1, revealed the heterojunction photocatalysts for decomposition of RhB is more related to  $\text{Ag}_2\text{CO}_3$  content. It is ascribed that the synergistic effect between  $\text{Bi}_2\text{WO}_6$  and  $\text{Ag}_2\text{CO}_3$  play a dominant role in reaction process. Whereas, for degradation of MO, the significantly enhanced the photocatalytic activity of  $\text{Bi}_2\text{WO}_6$  is mainly due to the modification effect of  $\text{Ag}_2\text{CO}_3$  to  $\text{Bi}_2\text{WO}_6$ . It can thus be seen that the actual active component of heterojunction photocatalysts for the degradation of RhB and MO is different. At the same time, the experimental data showed that the photocatalytic activity of  $\text{Ag}_2\text{CO}_3/\text{Bi}_2\text{WO}_6$  enhanced remarkably with increasing  $\text{Ag}_2\text{CO}_3$  content. Nevertheless,  $\text{Ag}_2\text{CO}_3$  content exceeds 30 wt%, the photocatalytic activity decreased, suggesting that the optimal  $\text{Ag}_2\text{CO}_3$  content in  $\text{Ag}_2\text{CO}_3/\text{Bi}_2\text{WO}_6$  is 30 wt%. The optimum content of  $\text{Ag}_2\text{CO}_3$  in heterojunction should be related to the recombination rate of photogenerated  $e^-h^+$  pairs and light absorption efficiency. The similar phenomenon was also observed other systems such as  $\text{Ag}_2\text{CO}_3/\text{g-C}_3\text{N}_4$ ,<sup>34</sup>  $\text{AgBr}/\text{Ag}_2\text{CO}_3$ .<sup>52</sup>

To further investigate the validity of photocatalytic activity of A-BWO-3 heterojunction photocatalysts, the degradation experiments of A-BWO-3 for other pollutants such as MB, CR, and Cr (VI) were performed. As shown in Fig. 9, the A-BWO-3 heterojunction photocatalysts could remove more than 95% of these pollutants after 60 min. All these results convincingly demonstrate that the formation of heterogenous interface in  $\text{Ag}_2\text{CO}_3/\text{Bi}_2\text{WO}_6$  can effectively promote the photoactivity of pure  $\text{Bi}_2\text{WO}_6$ .

### 3.3 Catalyst stability

The stability of catalysts is a vital consideration except for the photocatalytic activity in practical applications. Thus, the stability of the pure  $\text{Ag}_2\text{CO}_3$  and A-BWO-3 was investigated. As shown in Fig. 10, it was found that the  $\text{Ag}_2\text{CO}_3$  nanoparticles almost lose their activity in after two recycles. This results can due to the reduction of large amounts  $\text{Ag}^+$  ions in  $\text{Ag}_2\text{CO}_3$  lattice by the photogenerated electrons ( $\text{Ag}_2\text{CO}_3 + 2\text{e}^- \rightarrow 2\text{Ag}^0 + \text{CO}_3^{2-}$ ) during the recycles reaction.<sup>31,32</sup> The excessive Ag could act as the recombination center of electrons and prevent the visible light absorption of  $\text{Ag}_2\text{CO}_3$ , leading to the photocatalytic activity extremely lower. Conversely, the A-BWO-3 still maintains 85% degradation rate after five recycles for RhB. Fig. S5 and S6 show the XRD and XPS of the freshly obtained A-BWO-3 and its recovered samples after cycles of photocatalysis. It is found that there is a slight change in position and intensity of peaks after five cycles of photodegradation reaction, demonstrating that the  $\text{Ag}_2\text{CO}_3$  remain stable and only a relatively small number of  $\text{Ag}^0$  nanoparticles were formed on the surface of  $\text{Ag}_2\text{CO}_3$ .

### 3.4 Possible photocatalytic mechanism

To confirm the main active species in the photocatalytic process, we performed *in situ* trapping experiments of free radicals and holes for degradation of RhB by A-BWO-3 heterojunction system. As shown in Fig. 11, disodium ethylenediaminetetraacetate (EDTA, a quencher of  $\text{h}^+$ ), 1,4-benzoquinone (BQ, a quencher of  $\bullet\text{O}^{2-}$ ) and isopropanol (IPA, a  $\bullet\text{OH}$  scavenger) was added into the reaction liquid. We observed that the addition of 2 mM EDTA had a significant effect on the  $k_{\text{app}}$  compared with no scavenger under same conditions, suggesting that  $\text{h}^+$  may play a dominant role in the photocatalytic reaction process of RhB degradation. Whereas, after the addition of 2 mM IPA or BQ had a relatively lesser influence on the  $k_{\text{app}}$ , meaning that  $\bullet\text{OH}$  and  $\bullet\text{O}^{2-}$  played a secondary role and to a lesser extent participated in the photocatalytic reaction process.

For the hierarchical  $\text{Ag}_2\text{CO}_3/\text{Bi}_2\text{WO}_6$  heterojunctions, the enhancement mechanism of photocatalytic activity could be further revealed by the energy band matching principle for two semiconductors. The top of the valence band (VB) and the bottom

of the conduction band (CB) of  $\text{Ag}_2\text{CO}_3$  were calculated to be 2.60 and 0.43 eV (vs. NHE), respectively. And the VB and CB of  $\text{Bi}_2\text{WO}_6$  were estimated to be 3.23 and 0.49 eV (vs. NHE), (some relevant details of the energy band estimation were provided in the *Supporting Information*). It was widely accepted that the excellent photocatalytic activity results from an efficient separation and smooth transport of photogenerated electrons and holes, a schematic illustration of the possible heterojunction enhancement mechanism for photocatalytic activity in  $\text{Ag}_2\text{CO}_3/\text{Bi}_2\text{WO}_6$  was shown in Fig. 12. It can be seen the electron is excited from the valence band to the conduction band in both  $\text{Ag}_2\text{CO}_3$  and  $\text{Bi}_2\text{WO}_6$ , thereby generating an electron-hole pairs under visible light irradiation. Once the formation of heterogeneous interfaces between  $\text{Bi}_2\text{WO}_6$  and  $\text{Ag}_2\text{CO}_3$ , the photogenerated electrons on the  $\text{Ag}_2\text{CO}_3$  nanoparticles will easily transfer to the CB of  $\text{Bi}_2\text{WO}_6$  through the closely contacted interfaces in that  $\text{Ag}_2\text{CO}_3$  has a more negative potential of the conduction band and valence band than that of  $\text{Bi}_2\text{WO}_6$ . Meanwhile the photoinduced holes could migrate from the VB of  $\text{Bi}_2\text{WO}_6$  to that of  $\text{Ag}_2\text{CO}_3$ . Therefore, the photoexcited electrons and holes are separated effectively due to the ingenious heterojunction design under visible light irradiation. Meanwhile, the photocorrosion of  $\text{Ag}_2\text{CO}_3$  nanoparticles can be further restrained because photoinduced electrons will fast transfer to the CB of  $\text{Bi}_2\text{WO}_6$  rather than react with the  $\text{Ag}^+$  ions of  $\text{Ag}_2\text{CO}_3$  ( $\text{Ag}^+ + \text{e}^- \rightarrow \text{Ag}^0$ ). The captured electrons on the CB of  $\text{Bi}_2\text{WO}_6$  can be react with dissolved oxygen to give  $\cdot\text{O}^{2-}$ .<sup>53</sup> And the holes with strong oxidizing power in the VB of  $\text{Ag}_2\text{CO}_3$  can be react with  $\text{H}_2\text{O}$  to generate  $\cdot\text{OH}$  or directly oxidize pollutants adsorbed.<sup>36,54</sup> Based on the above results, we believe that the well-matched band of the  $\text{Ag}_2\text{CO}_3/\text{Bi}_2\text{WO}_6$  heterojunctions can promote interfacial carriers transfer processes and prolong the lifetime of photogenerated holes and electrons, resulting in high photocatalytic activity and stability.

In addition, after cycle of photodegradation reaction, a small number of Ag were formed on the surface of  $\text{Ag}_2\text{CO}_3$  (Fig. S5 and S6), which can act as an electron-rich collective.<sup>32</sup> These enriched electrons on  $\text{Ag}^0$  nanoparticles will participate in the

multi-electron transfer routes ( $O_2 + 2e^- + 2H^+ = H_2O_2$  (aq.);  $O_2 + 4e^- + 4H^+ = 2H_2O$  (aq.)).<sup>55</sup> While  $Ag^0$  may also become a recombination center of  $e^-$ - $h^+$  pairs,<sup>56</sup> but also overmuch  $Ag^0$  nanoparticles on the  $Ag_2CO_3$  surface could prevent the visible light absorption of  $Ag_2CO_3$ ,<sup>34</sup> which would lead to the decrease of activity in recycling reactions. With increasing cyclic tests, a relatively stable  $Ag$ - $Ag_2CO_3$  system can be formed. The self-stability mechanism has also been discussed in some other silver-contained compounds.<sup>57,58</sup> Therefore,  $Ag$ - $Ag_2CO_3/Bi_2WO_6$  system may be a vital factor for high stability in the photocatalytic reaction process.

#### 4. Conclusion

In summary, a novel hierarchical  $Ag_2CO_3/Bi_2WO_6$  heterojunctions catalyst has been successfully fabricated by *in situ* depositing monodispersed spherical  $Ag_2CO_3$  nanoparticles onto the surface of  $Bi_2WO_6$ . The decoration of  $Ag_2CO_3$  nanoparticles significantly enhanced the photocatalytic activity of  $Bi_2WO_6$  and the 30 wt%  $Ag_2CO_3$  content sample exhibited the highest photocatalytic activity for degradation of RhB and MO. Furthermore,  $Ag_2CO_3/Bi_2WO_6$  heterojunctions have excellent cycling stability. The enhanced photocatalytic activity and stability is not only attributes to the narrow bandgap of  $Ag_2CO_3$  and extended the optical response, but also benefits by the appropriate band structure matching to enhanced the photo-generated carriers separation in the heterogenous interface between  $Ag_2CO_3$  and  $Bi_2WO_6$ . Consequently, we believe that such heterojunction system is promising extended to other silver-containing heterostructured photocatalysts with highly efficient photocatalytic performances.

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## Table

**Table 1.** Pseudo-first-order rate constants in photocatalytic degradation RhB ( $k_1$ ) and MO ( $k_2$ ).

Sample	$k_1$ (min <sup>-1</sup> )	$k_2$ (min <sup>-1</sup> )
Pure-Ag <sub>2</sub> CO <sub>3</sub>	0.0195	0.0132
Pure-Bi <sub>2</sub> WO <sub>6</sub>	0.0272	0.0005
A-BWO-1	0.0297	0.0110
A-BWO-2	0.0343	0.0342
A-BWO-3	0.0489	0.0598
A-BWO-4	0.0423	0.0532
Degussa P25	0.0014	0.0017
prepared N-TiO <sub>2</sub>	0.0124	0.0108

## Figure

**Fig. 1** XRD patterns of the pure Bi<sub>2</sub>WO<sub>6</sub>, Ag<sub>2</sub>CO<sub>3</sub> and Ag<sub>2</sub>CO<sub>3</sub>/Bi<sub>2</sub>WO<sub>6</sub> composite photocatalyst with different Ag<sub>2</sub>CO<sub>3</sub> content.

**Fig. 2** XPS survey spectra (a) and high-resolution XPS spectra of Bi 4f (b), W 4f (c) O 1s (d), C 1s (e) and Ag 3d (f) regions for A-BWO-3 composite.

**Fig. 3** SEM images of bare Bi<sub>2</sub>WO<sub>6</sub> (a), A-BWO-1 (b), A-BWO-2 (c), A-BWO-3 (c) and A-BWO-4 (d) composite photocatalysts; EDX spectrum of A-BWO-3 composite photocatalysts (f).

**Fig. 4** TEM images of bare Bi<sub>2</sub>WO<sub>6</sub> (a, b) and A-BWO-3 heterojunction photocatalysts (c, d); SAED and HRTEM of A-BWO-3 heterojunction photocatalysts (e, f).

**Fig. 5** Schematic diagram of the fabrication of Ag<sub>2</sub>CO<sub>3</sub>/Bi<sub>2</sub>WO<sub>6</sub> heterojunctions.

**Fig. 6** Nitrogen adsorption-desorption isotherms and the corresponding pore size distribution curves (inset) of the pure Bi<sub>2</sub>WO<sub>6</sub> and the A-BWO-3 heterojunction photocatalysts.

**Fig. 7** UV-vis diffuse reflectance spectra of the as-prepared samples and the bandgap (inset) of the pure Bi<sub>2</sub>WO<sub>6</sub> and pure Ag<sub>2</sub>CO<sub>3</sub>.

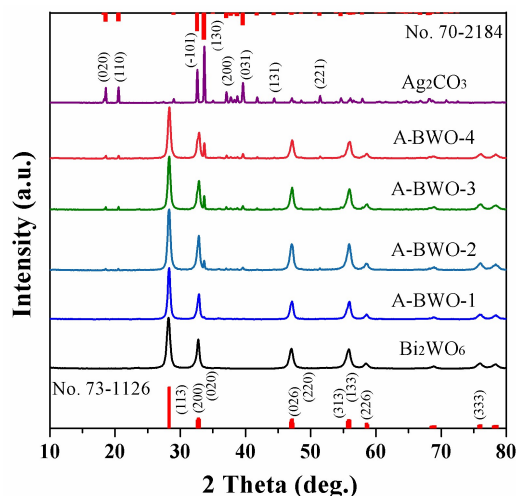
**Fig. 8** Photocatalytic activity of as-prepared samples and the adsorption spectra in the presence of A-BWO-3 for degradation of RhB (a, b) and MO (c, d) under visible irradiation.

**Fig. 9** Photocatalytic activity of A-BWO-3 for degradation of MB, CR, Cr (VI), MO and RhB after 60 min visible light irradiation.

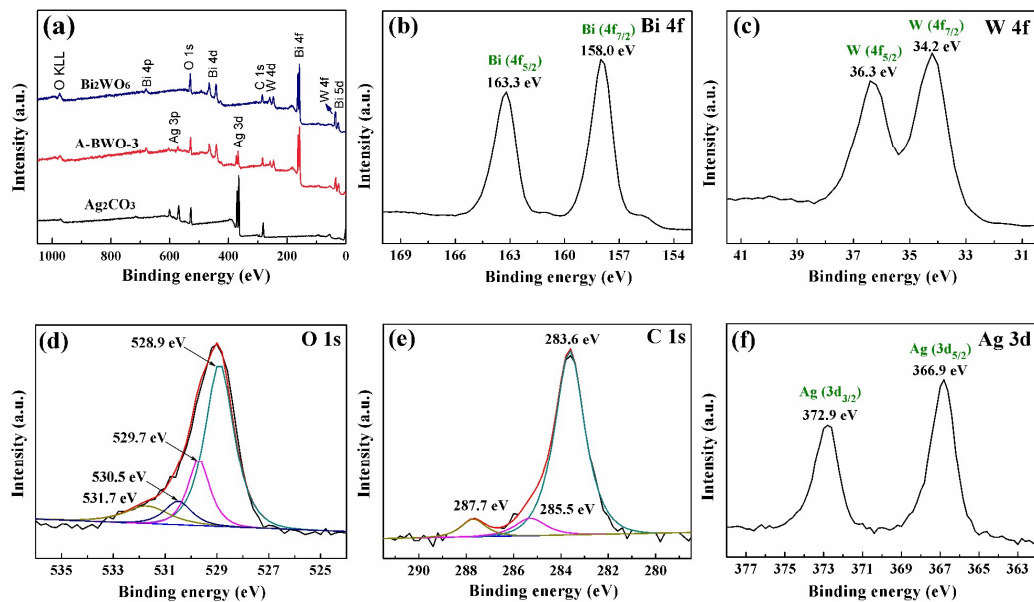
**Fig. 10** The recycling tests of the as-prepared  $\text{Ag}_2\text{CO}_3$  and A-BWO-3 heterojunction photocatalysts for photodegradation of RhB.

**Fig. 11** Comparison of the rate constant  $k_{\text{app}}$  of photocatalytic degradation RhB by A-BWO-3 in the presence of various scavengers, the concentration of all added scavengers is 2 mM.

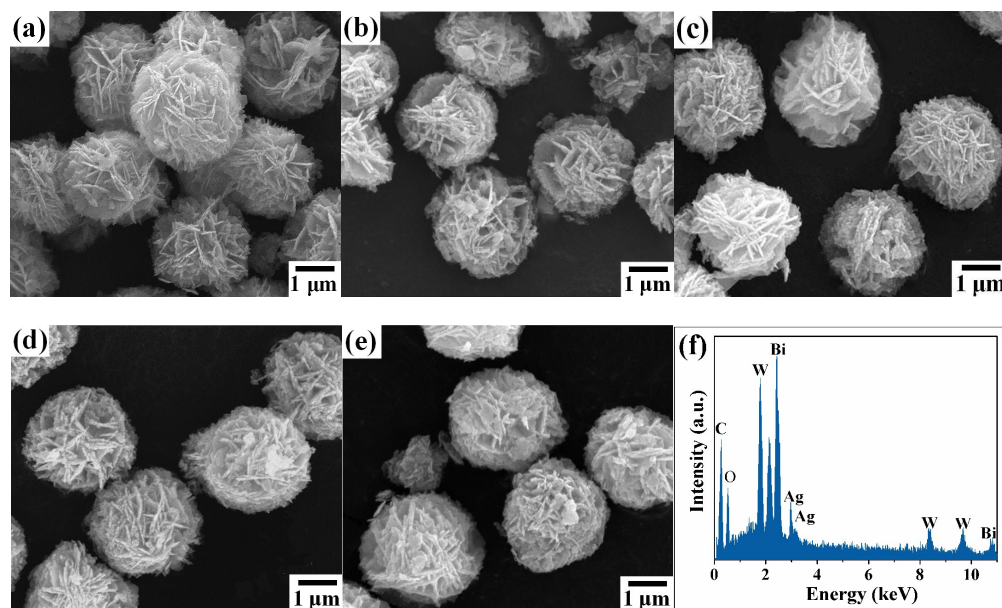
**Fig. 12** Schematic illustration photocatalytic reaction process and the possible charge carriers transfer of the  $\text{Ag}_2\text{CO}_3/\text{Bi}_2\text{WO}_6$  heterojunctions under visible light irradiation.



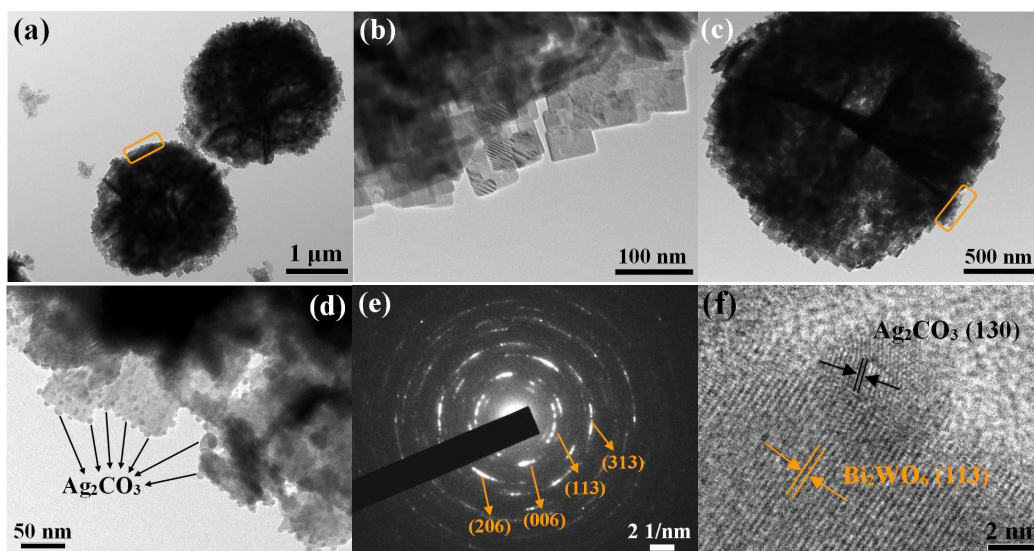
**Fig. 1** XRD patterns of the pure  $\text{Bi}_2\text{WO}_6$ ,  $\text{Ag}_2\text{CO}_3$  and  $\text{Ag}_2\text{CO}_3/\text{Bi}_2\text{WO}_6$  composite photocatalyst with different  $\text{Ag}_2\text{CO}_3$  content.



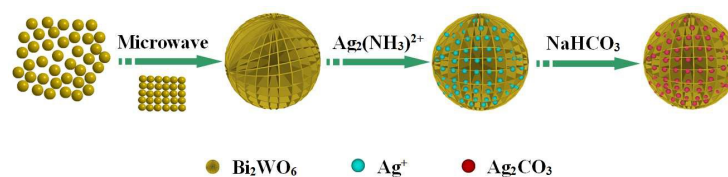
**Fig. 2** XPS survey spectra (a) and high-resolution XPS spectra of Bi 4f (b), W 4f (c) O 1s (d), C 1s (e) and Ag 3d (f) regions for A-BWO-3 composite.



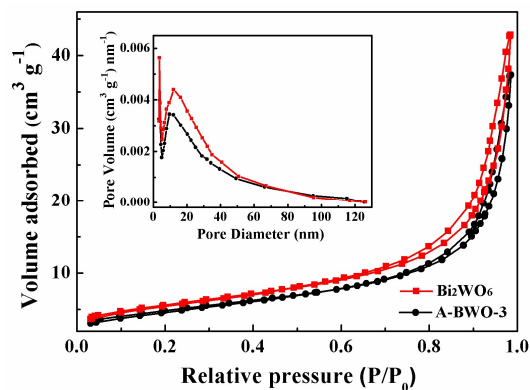
**Fig. 3** SEM images of bare  $\text{Bi}_2\text{WO}_6$  (a), A-BWO-1 (b), A-BWO-2 (c), A-BWO-3 (c) and A-BWO-4 (d) composite photocatalysts; EDX spectrum of A-BWO-3 composite photocatalysts (f).



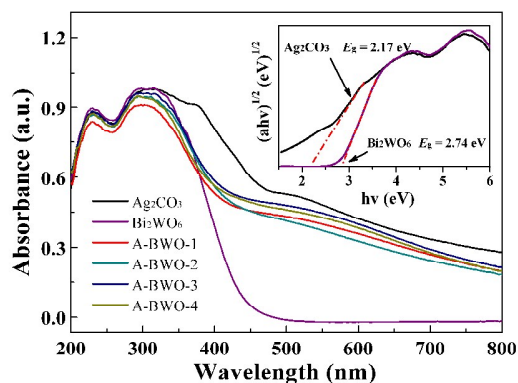
**Fig. 4** TEM images of bare  $\text{Bi}_2\text{WO}_6$  (a, b) and A-BWO-3 heterojunction photocatalysts (c, d); SAED and HRTEM of A-BWO-3 heterojunction photocatalysts (e, f).



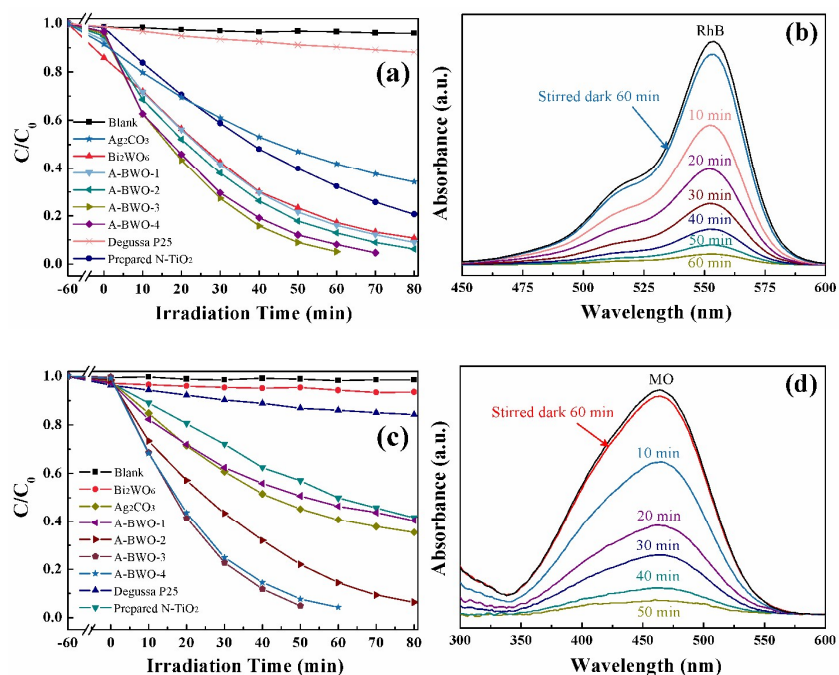
**Fig. 5** Schematic diagram of the fabrication of  $\text{Ag}_2\text{CO}_3/\text{Bi}_2\text{WO}_6$  heterojunctions.



**Fig. 6** Nitrogen adsorption-desorption isotherms and the corresponding pore size distribution curves (inset) of the pure  $\text{Bi}_2\text{WO}_6$  and the A-BWO-3 heterojunction photocatalysts.

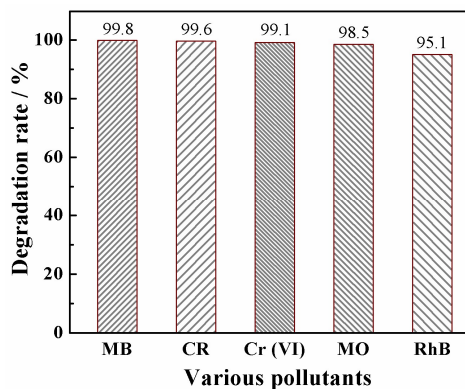


**Fig. 7** UV-vis diffuse reflectance spectra of the as-prepared samples and the bandgap (inset) of the pure Bi<sub>2</sub>WO<sub>6</sub> and pure Ag<sub>2</sub>CO<sub>3</sub>.

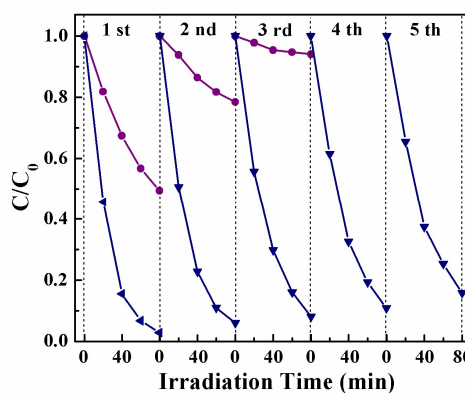


**Fig. 8** Photocatalytic activity of as-prepared samples and the adsorption spectra in the presence of A-BWO-3 for degradation of RhB (a, b) and MO (c, d) under visible irradiation.

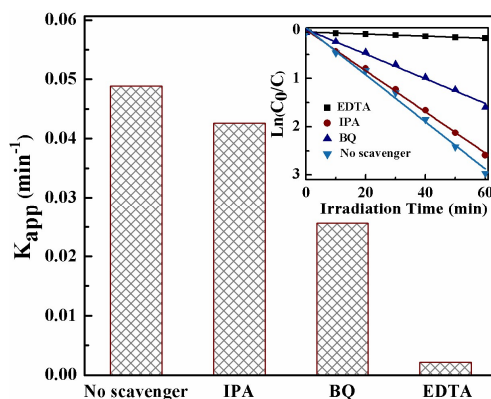




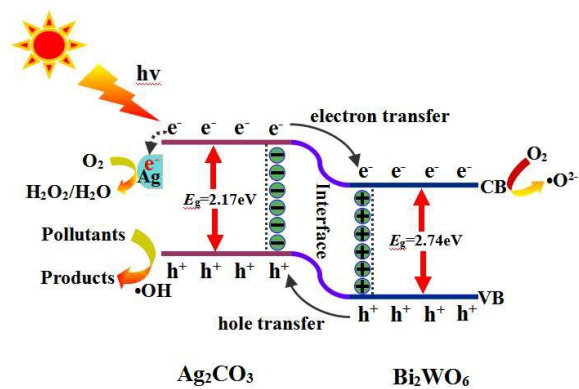
**Fig. 9** Photocatalytic activity of A-BWO-3 for degradation of MB, CR, Cr (VI), MO and RhB after 60 min visible light irradiation.



**Fig. 10** The recycling tests of the as-prepared  $\text{Ag}_2\text{CO}_3$  and A-BWO-3 heterojunction photocatalysts for photodegradation of RhB.



**Fig. 11** Comparison of the rate constant  $k_{app}$  of photocatalytic degradation RhB by A-BWO-3 in the presence of various scavengers, the concentration of all added scavengers is 2 mM.



**Fig. 12** Schematic illustration photocatalytic reaction process and the possible charge carriers transfer of the  $\text{Ag}_2\text{CO}_3/\text{Bi}_2\text{WO}_6$  heterojunctions under visible light irradiation.