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COMMUNICATION

7 **Na_{0.56}Ti_{1.72}Fe_{0.28}O₄: a novel anode material for Na-ion** 8 **batteries**

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9 Junke Hou^{a,b}, Yubin Niu^{a,b}, Wenjun Li^{a,b}, Fenglian Yi^{a,b}, Sangui Liu^{a,b}, Yutao Li^c and
10 Maowen Xu^{a,b*}

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14 **A novel Na_{0.56}Ti_{1.72}Fe_{0.28}O₄ material is explored as an anode in**
15 **Na-ion batteries for the first time. It delivers a reversible**
16 **discharge capacity of 210.3 mA h g⁻¹ at 20 mA g⁻¹ in Na-ion**
17 **batteries, exhibiting good capacity retention at a cut-off**
18 **voltage of 0.01~3 V.**

19 Na-ion batteries have been intensely concerned as a new
20 class of emerging energy storage devices in recent years.
21 Compared with Li-ion batteries, they possess wider resource
22 supplies, lower cost of production and higher system
23 security.¹⁻² Such advantages provide a broader development
24 space for Na-ion batteries and accelerate their practical
25 application in the future. In fact, the current research on
26 Na-ion batteries mainly focuses on the electrode materials
27 which can be divided into the cathode and anode. To date, a
28 lot of cathode materials, such as β-NaMnO₂,³ olivine
29 NaFePO₄,⁴ Na₃V₂(PO₄)₃⁵ and so on, have been investigated.
30 By contrast, the anode materials for NIBs are limited.
31 Although it is reported that many materials can be used as
32 anode materials for sodium-ion batteries, such as
33 (Cu₆Sn₅)_{1-x}C_x,⁶ Sb-C,⁷ hollow carbon,⁸ phosphorene,⁹
34 Si,¹⁰⁻¹¹ and so on. However, the rate performance and cycle
35 life of these materials still need to further improve in the
36 practical application. Thus, it is very necessary to find more
37 alternative anode materials for Na-ion batteries in the future.

38 In the past few decades, titanate materials have emerged
39 rapidly in the fields of biosensors,¹² adsorbents,¹³
40 photocatalysts,¹⁴ solar cells,¹⁵ supercapacitors,¹⁶ and Li-ion
41 batteries,¹⁷ due to their unique structural channel, one
42 dimensional (1D) nanostructure and large surface area.¹⁸ In

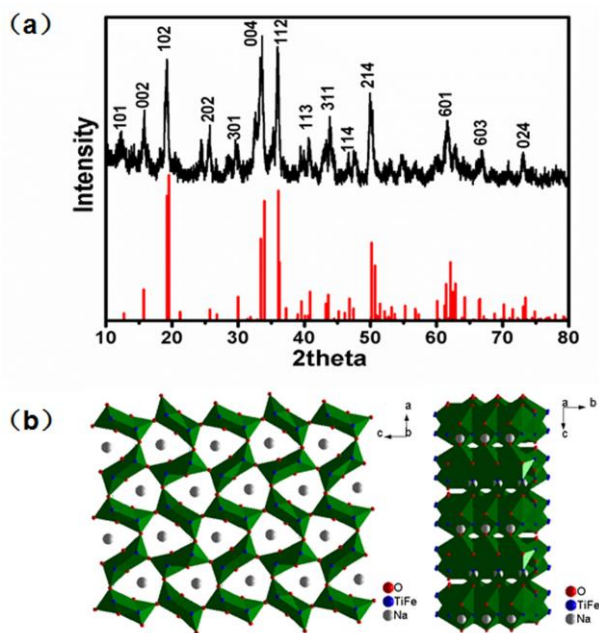
43 fact, the use of titanate as anode material for Na-ion batteries
44 has also been reported in recent years. Among them,
45 Na₂Ti₃O₇ shows a suitable capacity of more than 250 mA h
46 g⁻¹ at a current density of 354 mA g⁻¹ with long service life.¹⁹
47 Na_{2.65}Ti_{3.35}Fe_{0.65}O₉ delivers a capacity of 137.5 mA h g⁻¹ at a
48 current rate of 40 mA g⁻¹, which has been reported by our
49 group recently.²⁰ The monoclinic phase of Na₄Ti₅O₁₂ as an
50 anode for Na-ion batteries can obtain a reversible capacity of
51 64 mA h g⁻¹.²¹ Besides, Na₂Ti₆O₁₃ displays the ultra-fast
52 (30C or 2min) rate capacity and impressive long cycle life
53 (>5000cycles), but a first discharge capacity is only 65 mA h
54 g⁻¹.²² Therefore, it is very important to find a new kind of
55 titanate which could improve insertion of Na⁺ into per
56 formula units.

57 Here we used a simple solid-state reaction to synthesize
58 Na_{0.56}Ti_{1.72}Fe_{0.28}O₄, and tested its electrochemical
59 performance in Na-ion batteries for the first time. As an
60 anode material for Na-ion batteries, it can deliver a
61 reversible discharge capacity of 210.3 mA h g⁻¹ at 20mA g⁻¹,
62 and exhibit a low voltage plateau and long cycle life.

63 Na_{0.56}Ti_{1.72}Fe_{0.28}O₄ sample was synthesized by solid-state
64 reaction of stoichiometric amounts of Na₂CO₃, Fe₂O₃, Fe and
65 TiO₂. The powders were ball-milled for 10 h at 500 rpm, and
66 dried for 12 h at 100 °C. Finally, the obtained powder was
67 heated at 850 °C for 12 h in an Air atmosphere. The anode
68 electrode was prepared by mixing active material, super-P
69 carbon black, and polyvinylidene fluoride (PVdF) in a
70 weight ratio 80: 10: 10 with N-methyl pyrrolidone (NMP) as
71 solvent. The anode electrodes were pressed onto the
72 copper foil and dried at 120 °C under vacuum for 12 h. The
73 CR2032 coin-type cells consisting of a cathode and sodium
74 metal anode separated by a glass fiber were assembled in a
75 glove box filled with dry argon gas. The electrolyte was 1 M
76 NaPF₆ in ethylene carbonate–diethyl carbonate (EC/DEC,

77 1:1 (v/v) (Sigma). The galvanostatic charge–discharge tests
78 were performed using a Land battery tester at different rates
79 at 25 °C after 12 h rest.

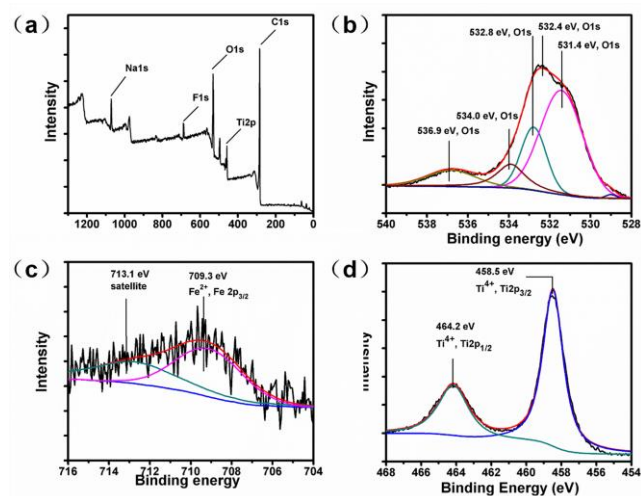
80 Powders X-ray diffraction were performed using Cu K α
81 radiation on a Bruker D8 Advance Diffractometer (XRD,
82 Maxima-X XRD-7000), X-ray photoelectron spectroscopy
83 (XPS) measurements were carried out on a spectrometer
84 (Escalab 250xi, Thermo Scientific) and the morphology of
85 as-prepared materials were also observed using
86 field-emission scanning electron microscopy (FESEM,
87 JSM-6700F) and transmission electron microscopy (TEM,
88 JEM-2100).



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90 **Figure 1** (a) XRD pattern of the as-prepared $\text{Na}_{0.56}\text{Ti}_{1.72}\text{Fe}_{0.28}\text{O}_4$ sample; (b)
91 Schematic illustration of the crystal structure of $\text{Na}_{0.56}\text{Ti}_{1.72}\text{Fe}_{0.28}\text{O}_4$.

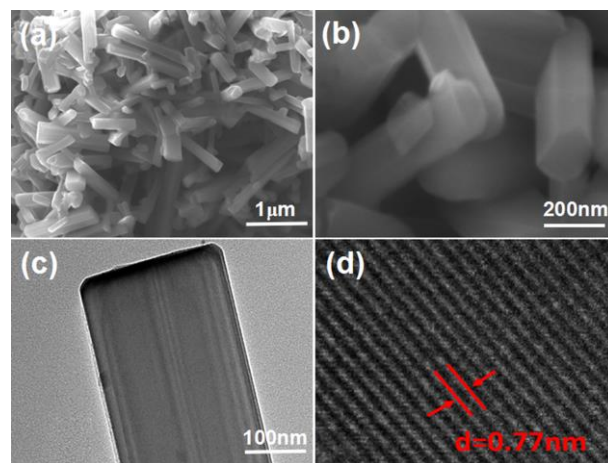
92 Fig. 1(a) displays the XRD patterns of $\text{Na}_{0.56}\text{Ti}_{1.72}\text{Fe}_{0.28}\text{O}_4$.
93 It is clear that the main diffraction peak positions of the
94 obtained materials agree well with $\text{Na}_{0.56}\text{Ti}_{1.72}\text{Fe}_{0.28}\text{O}_4$, and
95 the diffraction peaks were assigned to a crystallized
96 orthorhombic structure (space group: $Pnma(62)$). Only trace
97 amounts of impurities existed. Fig. 1(b) schematically shows
98 that the framework of $\text{Na}_{0.56}\text{Ti}_{1.72}\text{Fe}_{0.28}\text{O}_4$ is built of
99 $(\text{Fe}/\text{Ti})\text{O}_6$ octahedral with sharing edges. And we can see the
100 diffusion path of Na^+ is mainly along the b-axis direction for
101 free transport.



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103 **Figure 2** (a) XPS survey spectrum for the surface of $\text{Na}_{0.56}\text{Ti}_{1.72}\text{Fe}_{0.28}\text{O}_4$. XPS
104 spectra of (b) O 1s (c) Fe 2p and (d) Ti 2p.

105 The X-ray photoelectron spectroscopy (XPS) was used to
106 investigate the chemical states of Ti, Fe and O. The XPS
107 spectra of $\text{Na}_{0.56}\text{Ti}_{1.72}\text{Fe}_{0.28}\text{O}_4$ is displayed in Fig. 2. The
108 wide angle XPS (Fig. 2(a)) of the obtained sample shows the
109 predominant presence of O, Ti, Fe, and Na. From Fig. 2(b),
110 we can see the position of the peaks is 539.9 eV (O^{2-} , O 1s),
111 534.0 eV (O^{2-} , O 1s), 532.8 eV (O^{2-} , O 1s), 532.4 eV (O^{2-} , O
112 1s) and 531.4 eV (O^{2-} , O 1s). Fig. 2(c) shows the position of
113 the peak is 709.3 eV (Fe^{2+} , Fe 2p $_{3/2}$). And the position of the
114 peaks are 458.5 eV (Ti^{4+} , Ti 2p $_{3/2}$) and 464.2 eV (Ti^{4+} , Ti
115 2p $_{1/2}$) from Fig. 2(d).^{20, 23} The valence of the Fe, O, Ti is +2,
116 -2 and +4, respectively, which is consistent with the
117 molecular formula.



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119 **Figure 3** (a, b) SEM images of $\text{Na}_{0.56}\text{Ti}_{1.72}\text{Fe}_{0.28}\text{O}_4$ with different magnifications;
120 (c) TEM images of $\text{Na}_{0.56}\text{Ti}_{1.72}\text{Fe}_{0.28}\text{O}_4$; (d) High-resolution TEM image of
121 $\text{Na}_{0.56}\text{Ti}_{1.72}\text{Fe}_{0.28}\text{O}_4$.

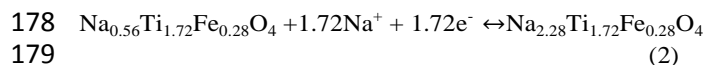
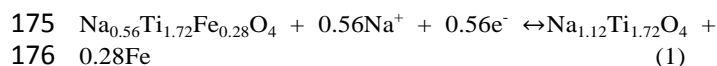
122 The FESEM and TEM images of the as-prepared
123 $\text{Na}_{0.56}\text{Ti}_{1.72}\text{Fe}_{0.28}\text{O}_4$ rods are shown in Fig. 3. Fig. 3(a) is
124 a typical SEM image of $\text{Na}_{0.56}\text{Ti}_{1.72}\text{Fe}_{0.28}\text{O}_4$ rods at relatively

low magnification, and the length of uniform rods is 1–2 μm . From the Fig. 3(b), it can be clearly observed that the rods with a diameter of 100–200 nm have been obtained. The width of rods is approximately 200 nm from the TEM image (Fig. 3(c)), while the HRTEM image (Fig. 3(d)) clearly shows lattice fringes indicating single crystallinity of the platelets. The width (0.69 nm) of neighbouring lattice fringes corresponds to the (101) plane of $\text{Na}_{0.56}\text{Ti}_{1.72}\text{Fe}_{0.28}\text{O}_4$.

Fig. S1 shows the FESEM images of $\text{Na}_{0.56}\text{Ti}_{1.72}\text{Fe}_{0.28}\text{O}_4$ after 20 cycles. From the figure we can see the structure of the material is substantially unchanged. As shown in Fig. S2, the energy-dispersive X-ray (EDX) mapping images of $\text{Na}_{0.56}\text{Ti}_{1.72}\text{Fe}_{0.28}\text{O}_4$ show no impurities, and all three elements – Na, Ti, Fe and O are present uniformly throughout the bulk of the sample.

The Na-ion insertion–extraction behavior of $\text{Na}_{0.56}\text{Ti}_{1.72}\text{Fe}_{0.28}\text{O}_4$ is studied by galvanostatic charge–discharge test as shown in Fig. 4. The differential specific capacity plots of the 1st, 2nd and 3rd cycle are shown in Fig. 4(a). The first charge–discharge capacity is 217.1/344.5 mA h g^{-1} at a current rate of 20 mA g^{-1} , respectively. The following charge–discharge capacities are 202.5/210.3 mA h g^{-1} and 195.5/202.3 mA h g^{-1} in the 2nd and 3rd cycle, respectively. To understand the electrochemical Na-ion insertion–extraction processes clearly in the first three cycles, cyclic voltammetry studies (CVs) were performed on $\text{Na}_{0.56}\text{Ti}_{1.72}\text{Fe}_{0.28}\text{O}_4$, the CVs were conducted in the voltage range 0.01 to 3 V at 0.1 mV s^{-1} and the results are shown in Fig. 4b. The first cycle reduction process consists of three peaks (at 0.766 V, 0.419 V and 0.144 V). On the subsequent cycling, the reduction peaks of 0.766 V and 0.419 V completely vanish. The first cycle oxidation process has four peaks (at 0.089 V, 0.322 V, 0.455 V and 0.768 V), the position of the oxidation peaks were basic unchanged after the subsequent cycling. The CV peaks overlap well for the 2nd and 3rd cycle, which indicates good reversibility of the charge–discharge reactions.

When the cell was charged to 3 V after 20 cycles, the position of peaks are 457.2 eV (Ti^{3+} , Ti 2 $p_{3/2}$), 458.5 eV (Ti^{4+} , Ti 2 $p_{3/2}$), 463.3 eV (Ti^{3+} , Ti 2 $p_{1/2}$), 464.5 eV (Ti^{4+} , Ti 2 $p_{1/2}$) and 709.3 eV (Fe^{2+} , Fe 2 $p_{3/2}$); and when the cell was discharged to 0.01 V after 20 cycles, the position of peaks are 456.8 eV (Ti^{3+} , Ti 2 $p_{3/2}$), 458.7 eV (Ti^{4+} , Ti 2 $p_{3/2}$), 464.3 eV (Ti^{4+} , Ti 2 $p_{1/2}$) and 706.7 eV (Fe^0 , Fe 2 $p_{3/2}$), 709.1 eV (Fe^{2+} , Fe 2 $p_{3/2}$) (Fig. S3). According to the reaction mechanism proposed for Fe and Ti in the earlier studies,^{19, 24} the capacity are provided by $\text{Ti}^{4+} \leftrightarrow \text{Ti}^{3+}$ and $\text{Fe}^{2+} \leftrightarrow \text{Fe}^0$ in this experiment. The electrochemical equation for sodium storage can hence be stated as follows:



The voltage plateau at approximately 0.766 V, representing the reduction of Fe ions into Fe metal (Fe^0). The other peak at 0.419 V is related to the process of Ti^{4+} transform into Ti^{3+} . Given by Equation 1 (2), respectively.

The cycling stability and the Coulombic efficiency are also displayed in Fig. 4 (c), the capacity is still maintained at 174.6 mA h g^{-1} at a current rate of 20 mA g^{-1} after 100 cycles, and the corresponding capacity retention is 83 %, and the Coulombic efficiency is close to 100% except the first cycle, showing good cycling stability. The rate performance of the $\text{Na}_{0.56}\text{Ti}_{1.72}\text{Fe}_{0.28}\text{O}_4$ composite electrode is summarized in Fig. 4 (d). The reversible capacities were 210.2, 171.8, 137.1, 111.2, 90.4, 69.1 and 49.4 mA h g^{-1} at the current rates of 20, 40, 80, 100, 200, 500 and 1000 mA g^{-1} , respectively. Even at the current density of 40 and 20 mA g^{-1} again, a discharge capacity of 133.3 and 166.5 mA h g^{-1} can still be achieved, respectively. The good electrochemical performance of $\text{Na}_{0.56}\text{Ti}_{1.72}\text{Fe}_{0.28}\text{O}_4$ cells are mostly attributed to the structure stability of the $\text{Na}_{0.56}\text{Ti}_{1.72}\text{Fe}_{0.28}\text{O}_4$ electrode material during Na-ion insertion–extraction processes.^{25–26} Therefore, the $\text{Na}_{0.56}\text{Ti}_{1.72}\text{Fe}_{0.28}\text{O}_4$ electrode material is promising in practical Na-ion batteries because of its good reversible structure change after Na-ion extraction–insertion processes.

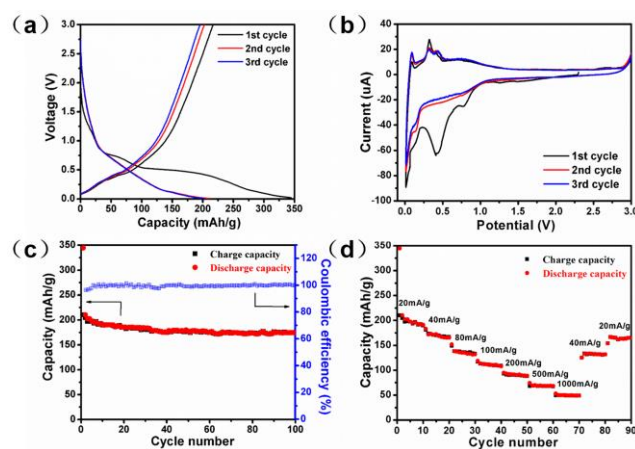


Figure 4 Performance of $\text{Na}_{0.56}\text{Ti}_{1.72}\text{Fe}_{0.28}\text{O}_4$. (a) The 1st, 2nd and 3rd charge/discharge curves of $\text{Na}_{0.56}\text{Ti}_{1.72}\text{Fe}_{0.28}\text{O}_4$ at a current rate of 20 mA g^{-1} in the voltage range of 0.01 and 3 V versus Na^+/Na ; (b) CV curves at a scan rate of 0.1 mV s^{-1} showing the first, the second and the third cycle of the $\text{Na}_{0.56}\text{Ti}_{1.72}\text{Fe}_{0.28}\text{O}_4$ composite electrode at a voltage window of 0.01 ~ 3 V; (c) Long-term cycling performance of the $\text{Na}_{0.56}\text{Ti}_{1.72}\text{Fe}_{0.28}\text{O}_4$ at a current rate of 20 mA g^{-1} in the voltage range of 0.01 and 3 V versus Na^+/Na ; (d) Rate capability of the $\text{Na}_{0.56}\text{Ti}_{1.72}\text{Fe}_{0.28}\text{O}_4$. The capacity versus cycle number at various current rates.

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In summary, we have synthesized $\text{Na}_{0.56}\text{Ti}_{1.72}\text{Fe}_{0.28}\text{O}_4$ by a simple solid-state route. The Na-ion extraction–insertion behavior of $\text{Na}_{0.56}\text{Ti}_{1.72}\text{Fe}_{0.28}\text{O}_4$ is investigated in detail for the first time. It delivered a discharge capacity of 210.3 mA h g^{-1} at a current density of 20 mA g^{-1} , and it presents a good capacity and cycle life compared with the aforementioned material. $\text{Na}_{0.56}\text{Ti}_{1.72}\text{Fe}_{0.28}\text{O}_4$ holds great promise as an anode material for Na-ion batteries applications.

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230 Notes and references

- 231 ^aInstitute for Clean Energy & Advanced Materials, Faculty of Materials
 232 and Energy, Southwest University, Chongqing 400715, P.R. China
 233 ^bChongqing Key Laboratory for Advanced Materials and Technologies of
 234 Clean Energies, Chongqing 400715, P.R. China
 235 ^cTexas Materials Institute, University of Texas at Austin, Texas 78712,
 236 USA
 237 Fax: +86-23-68254969; Tel: +86-23-68254969;
 238 E-mail: xumaowen@swu.edu.cn
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