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**Hierarchical Ni-Fe layered double hydroxide/MnO₂
sphere architecture as an efficient noble metal-free electrocatalyst for
ethanol electro-oxidation in alkaline solution**

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Abstract: Ni-Fe layered double hydroxide (LDH) nanosheets and hierarchical Ni-Fe LDH@MnO₂ spheres are synthesized by a facile and cost-effective approach for highly efficient ethanol electro-oxidation. The LDH@MnO₂ microsphere displays excellent catalytic activity and robust durability for ethanol electro-oxidation, compared with the Ni-Fe LDH nanosheets. According to the analyses for nitrogen adsorption isotherms and electrochemical impedance spectra (EIS), it is inferred that the performance enhancing could be attributed to that MnO₂ could increase the concentration of OH_{ads} species on Ni-Fe LDH surface and these OH_{ads} can react with C_{1ad} intermediate species to produce CO₂ or water soluble products, releasing the active sites on LDH for further electrochemical reaction. Therefore, it is expected that an effective noble-metal free catalysts for ethanol electro-oxidation could be obtained by tailoring structure and properties of LDHs and their composites.

Keywords: Ni-Fe LDH; MnO₂; Ethanol; Electro-oxidation; Electrocatalyst

Introduction

Considerable efforts have been fueled up to exploit green, sustainable and efficient power sources owing to the increasing global demand for energy, coupled with the depletion of fossil fuels and the associated detrimental environmental impact [1]. Ethanol is a promising fuel for low temperature direct fuel cell reactions for its high energy density, low toxicity, ease of storage and availability from different resources [2-6]. In direct ethanol fuel cells (DEFCs), noble metal based materials (Pt, Pd and their alloys) are the most effective catalysts due to their superior properties in the adsorption and dissociation of small organic molecules, but the high cost, limited availability as well as unsatisfactory cycle life restrict their practical applications [2-13]. In recent years, lots of explorations have been made to obtain advanced noble metal-based hetero-structures for improving the performance of DEFCs, such as graphene-supported Pt nanoparticles, free-standing Pd-Au bimetallic nanotube and noble metal membrane [14-22]. Despite all these progress, the catalysts are also noble metal-based. How to develop effective noble metal-free anode catalysts for ethanol electro-oxidation remains a challenging goal.

Layered double hydroxides (LDHs) are a family of layered materials consisting of positively charged brucite-type octahedral layers where the charge-balancing anions and water molecules occupy the interlayer space [14-15], which have been widely used in the fields of electrochemical sensors [16], super-capacitors [17] and alkaline secondary batteries [18], owing to their novel structure and desirable properties. Recently, LDHs have been reported as promising noble metal-free electrode materials

for the electrocatalytic oxidation of alcohol in alkaline medium [14-15, 19]. However, a key challenge for LDHs in the application of anode catalysts is to enhance the electrochemical activity by tailoring the structures and properties, such as a well-defined hierarchical architecture with high surface area and suitable pore-size distribution, in which all the electroactive species participates in faradaic redox reaction and a fast mass transport and electron transfer are guaranteed, and co-catalyst to influence intermediate adsorption-desorption.

Herein, we synthesized Ni-Fe LDH nanosheets and hierarchical Ni-Fe LDH@MnO₂ microspheres by a facile and cost-effective approach for highly efficient ethanol electro-oxidation. Furthermore, it is anticipated to have an essential insight into the synergistic mechanism for electro-catalytic activity enhancing.

Experimental

Ni-Fe LDH synthesis: In a typical synthesis, co-precipitated Ni-Fe LDH was synthesized by simultaneous dropwise addition of 30 mL of 1.0 M Ni(NO₃)₂ and 30 mL of 0.5 M Fe(NO₃)₃ solutions to 100 mL of 1.5 M Na₂CO₃ solution during constant stirring. The pH was adjusted to ~10 by addition of NaOH solution. Then the vessel was transferred to water bath at 65°C for 6 hours. Finally the precipitate was filtered, washed with deionized water, and dried in air at 80°C.

Ni-Fe LDH@MnO₂ sphere synthesis: To synthesize the composite electrocatalyst, 200 mg of MnO₂ sphere, prepared according to previous works [5], was dispersed into 100 mL deionized water and ultra-sonicated for 8 min. Then Ni-Fe LDH was synthesized and coated on the surface of MnO₂ sphere. The composite was

harvested and denoted as LDH@MnO₂

Material characterization: Powder X-ray diffraction (XRD) patterns of the as-prepared samples were collected on Smartlab XRD diffractometer using a Cu K α source, with a scan step of 0.02° and a scan range between 5 and 90°. The morphology of the samples was investigated using a transmission electron microscopy (TEM JEOL JEM-2010 HR-TEM). The accelerating voltage was 200 kV. The specific surface area was measured using the Brumauer-Emmett-Teller (BET) method based on the nitrogen adsorption-desorption isotherm at 77 K on a Micromeritics ASAP2020 sorption analyzer.

Electrochemical characterization: Electrochemical measurements were performed on an electrochemical workstation (CHI 660D, CH Instruments Inc., Shanghai) using a traditional three-electrode mode. Pt plate was used as a counter electrode, Hg/HgO as a reference electrode, and glassy carbon (3 mm in diameter) coated with the as-prepared samples, as a working electrode. The working electrode was fabricated as follows: 4 mg of catalysts was dispersed in 2 mL of ethanol solution and sonicated for 5 min; 10 μ L of the suspension was dripped onto the surface of a glassy carbon electrode and dried for 15 min; subsequently 2 μ L of 0.5% Nafion solution (Sigma-Aldrich) was coated on the electrode surface and dried for another 5 min. Electrocatalytic oxidation of ethanol on the working electrode was measured in 1.0 M KOH + 1.0 M ethanol solution by cyclic voltammetry in the potential range from 0.2 to 0.7 V. Electrochemical impedance spectra (EIS) were measured at 0.5 V from 100 KHz to 0.01 Hz and the perturbing AC amplitude was 5 mV. Electrical current density

was calculated by normalizing electrical current on the area of the 3 mm diameter glassy carbon electrode. The chronoamperometry (CA) was conducted at 0.5 V for 3600 s.

Results and discussion

The typical XRD pattern of the as-prepared Ni-Fe LDH is presented in Fig.1(a), showing the characteristic diffraction peaks at 2θ of 11.5, 23.3 and 34.6° corresponding to the (003), (006) and (012) plane reflections of 2D hydroxide-like materials, evidencing the formation of LDH particles (JCPDS card No.: 40-0215). As shown in Fig.1(b), all diffraction peaks can be unambiguously assigned to the single-phase γ -MnO₂ (JCPDS card No.: 14-0644). According to the XRD pattern shown in Fig. 1(c), it is found that the LDH@MnO₂ sphere is the phase mixture of γ -MnO₂ and Ni-Fe LDH. There is no new phase generated in the synthesis process.

The TEM image, as shown in Fig. 2 (a), displays MnO₂ particles that are mainly spherical and the diameter of these microspheres is about 1-2 μm . Fig. 2 (b) presents the morphology of Ni-FeLDH and it is apparent to see the nanosheet structure of this as-prepared sample. For LDH@MnO₂, as shown in Fig. 2(c), it holds the spherical morphology of the MnO₂ and is coated with Ni-FeLDH nanosheets on its surface. Due to the adsorption of Ni²⁺ and Fe³⁺ on MnO₂ spheres surface, Ni-Fe LDH nanosheets could cover the MnO₂ sphere uniformly when the precipitant was added. Furthermore, the heterogeneous interface between MnO₂ and solution could facilitate the nucleation of Fe-Ni LDH. It may be another reason for MnO₂ to be covered

uniformly with Fe-Ni LDH.

The N_2 adsorption and desorption isotherms shown in Fig. 3(a) and (c) clearly indicate the presence of mesopores in the prepared samples, classified as type IV as defined by the International Union of Pure and Applied Chemistry (IUPAC). A hysteric loop between the adsorption and desorption branches can be considered type H4, indicative of slit-like pores in the samples. The sample of Ni-Fe LDH shows a BET surface area of $208.33 \text{ m}^2 \text{ g}^{-1}$, which is much larger than that of LDH@MnO₂ composite ($103.47 \text{ m}^2 \text{ g}^{-1}$). The pore volumes of the as-prepared Ni-Fe LDH and LDH@MnO₂ microsphere are 0.883 and $0.208 \text{ cm}^3 \text{ g}^{-1}$, respectively. As shown in Fig. 3(b) and (d), the pore size distributions from the adsorption branch of the isotherms using the Barrett-Joyner-Halenda (BJH) method reveal that the pore size of Ni-Fe LDH focuses on 5~30 nm and for LDH@MnO₂, most of the pores have diameters under 20 nm.

The catalytic activities of the as-prepared catalysts towards ethanol electrooxidation were subsequently evaluated in an alkaline medium. The oxidation current was normalized to the electrode surface area; this allowed the current density to be directly used to compare the catalytic activity of different samples. The cyclic voltammograms of ethanol oxidation on the prepared catalysts are shown in Fig. 4. As shown in Fig. 4 (a), the cyclic voltammograms of MnO₂ in alkaline solutions indicate that MnO₂ does not have the catalytic activity to ethanol electro-oxidation. From Fig. 4 (b) and (c), it is seen that the CVs of Ni-Fe LDH and LDH@MnO₂ recorded in 1.0 M KOH solution without ethanol consist of a pair of redox peaks, corresponding to the reversible redox

of $\text{Fe}^{2+}/\text{Fe}^{3+}$ associated with OH^- . In the presence of 1.0 M ethanol, both Ni-Fe LDH and LDH@MnO₂ display electrocatalytic behavior to ethanol oxidation. The onset potentials of the forward anodic peak for both Ni-Fe LDH and LDH@MnO₂ were 0.557 V vs Hg/HgO. Furthermore, as shown in Fig. 4(d), the electrocatalytic behavior to ethanol oxidation is significantly enhanced for LDH@MnO₂ sphere in comparison with Ni-Fe LDH nanosheet. Generally speaking, a larger specific surface area and pore volume can provide more electroactive sites as well as effective diffusion channels for electrolyte ions. In particular, abundant mesopores benefit the mass diffusion and electron transfer, which guarantees highly efficient electrooxidation reaction. However, according to the analysis for nitrogen adsorption isotherms (Fig. 3), though the LDH@MnO₂ sphere possesses a smaller specific surface area and mesopore volume than Ni-Fe LDH nanosheet, but the electrocatalytic activity of the former is remarkably superior to that of the latter. Therefore, the performance improvement of LDH@MnO₂ cannot be attributed to the difference of surface and pore properties.

EIS was further applied to analyze the electrocatalytic performance of Ni-Fe LDH and LDH@MnO₂. Fig. 4(e) represents the EIS plots of Ni-Fe LDH and LDH@MnO₂ at 0.5 V in 1.0 M KOH+1.0 M ethanol solution and the data were analyzed by an equivalent circuit shown in Fig. 4(f). In the equivalent circuit, R_s is the sum of resistance of electrolyte, electrode material and the contact resistance at the interface of the active material/current collector; Q is a constant angle element, which represents the double layer capacitance; R_t is the charge transfer resistance; R_c is the resistance of

intermediate ad-layer and L is the inductance induced by the intermediate. The values of R_s , C , R_t , R_c and L were calculated from the CNLS fitting of the experimental impedances spectra and their resulting values are listed in Table 1.

According to the fitting results in Table 1, it is apparent to see the resistances for LDH@MnO₂, no matter R_t or R_c , are smaller than those for Ni-Fe LDH. In other words, the former sample has a smaller charge transfer resistance and intermediate ad-layer resistance. Moreover, it is found from the EIS results that mass diffusion is not a key factor for the electrooxidation process and thus both samples may provide effective diffusion channels despite the differences of surface area and pore volume. Additionally, it is well known that the adsorption of OH_{ads} species onto MnO₂ is more favorable [6]. Therefore, it is inferred that MnO₂ could increase the concentration of OH_{ads} species on Ni-Fe LDH surface, and these OH_{ads} can react with C_{1ad} intermediate species to produce CO₂ or water soluble products, releasing the active sites on LDH for further electrochemical reaction [5, 13-15]. This results in a better electrocatalytic performance in ethanol electro-oxidation process than Ni-Fe LDH. Further investigations of the mechanism are ongoing at our lab.

To evaluate the electrocatalytic activity and stability of Ni-Fe LDH and LDH@MnO₂ composite under continuous operating conditions, CV cycling tests were carried out in a 1.0 M KOH + 1.0 M ethanol solution, as shown in Fig. 4 (g). After 100 potential cycles, 90.8% of the initial catalytic activity was still maintained for LDH@MnO₂ sphere. This is highly superior to the Ni-Fe LDH nanosheet (60.2%), indicating the greatly improved stability of the LDH@MnO₂ sphere. In addition, the

cycling stability of these two samples was also studied by testing CA curves. Fig. 4 (h) shows the current density curves versus time recorded at 0.5 V for 3600s. It was found that the oxidation current density on the LDH@MnO₂ sphere is much higher than that on Ni-Fe LDH nanosheet over the whole time range, further demonstrating a significantly enhanced electrocatalytic activity. Moreover, this also indicates that the LDH@MnO₂ catalyst possesses good long-term durability for ethanol electro-oxidation in alkaline media.

Conclusions

In this work, Ni-Fe LDH nanosheet and hierarchical Ni-Fe LDH@MnO₂ sphere were synthesized by a facile and cost-effective approach for highly efficient ethanol electro-oxidation. According to the CV curves, it is manifested that MnO₂ has no catalytic activity for ethanol electro-oxidation, and the LDH@MnO₂ microsphere displays excellent catalytic activity and robust durability for ethanol electro-oxidation, compared with the Ni-Fe LDH nanosheet. This can be ascribed to that MnO₂ could increase the concentration of OH_{ads} species on Ni-Fe LDH surface, and these OH_{ads} can react with C_{1ad} intermediate species to produce CO₂ or water soluble products, releasing the active sites on LDH for further electrochemical reaction. It is expected that an effective noble-metal free catalyst for ethanol electro-oxidation could be obtained by tailoring structure and properties of LDHs and their composites.

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Figure caption

Fig.1 XRD patterns of Ni-Fe LDH (a), MnO₂ (b) and LDH@MnO₂ sphere (c)

Fig. 2 TEM images of the as-prepared samples, (a) MnO₂ sphere; (b) Ni-Fe LDH and (c) LDH@MnO₂

Fig. 3 N₂ adsorption-desorption isotherms of (a) Ni-Fe LDH; (c)LDH@MnO₂ and pore size distribution of (b) Ni-Fe LDH; (d)LDH@MnO₂

Fig. 4 (a-c) The cyclic voltammograms of MnO₂ (a), Ni-Fe LDH (b) and LDH@MnO₂ sphere (c); (d) cyclic voltammogramscomparison of three catalysts;(e) EIS plots of Ni-Fe LDH and LDH@MnO₂ sphere at 0.5 V; (f) the equivalent electrical circuit; (g) potential cycling stability of Ni-Fe LDH and LDH@MnO₂ sphere and (h) CA curves of Ni-Fe LDH and LDH@MnO₂ sphere in 1.0 M KOH+1.0 M ethanolsolution at a potential of 0.5 V at 25°C

Fig. 1 (a)

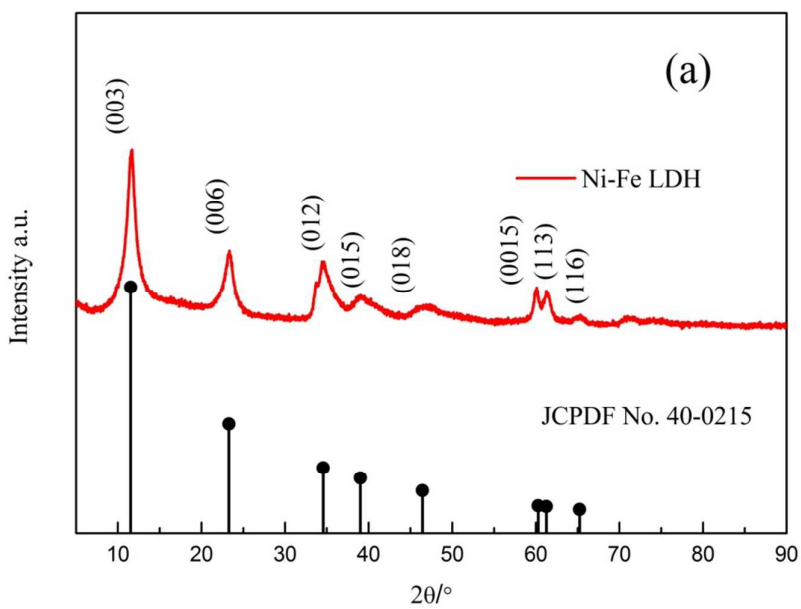


Fig. 1 (b)

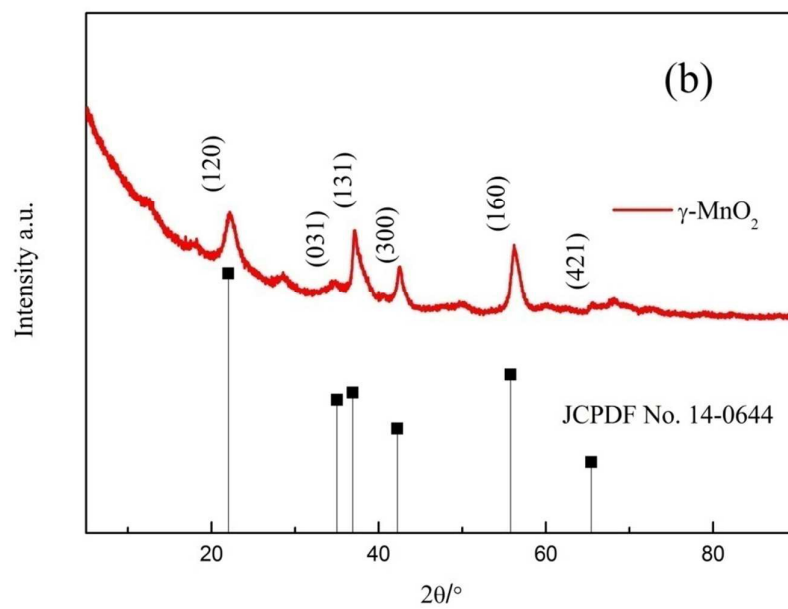


Fig. 1 (c)

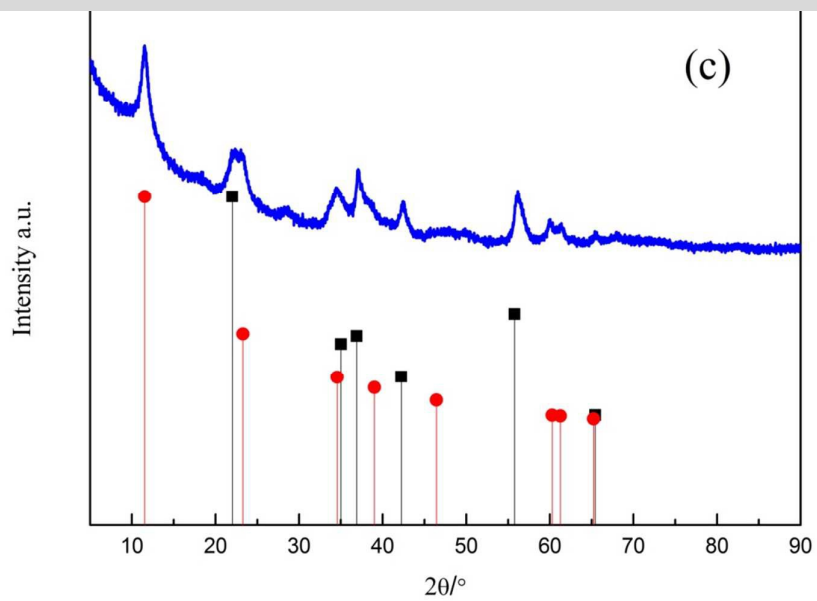


Fig. 2 (a)

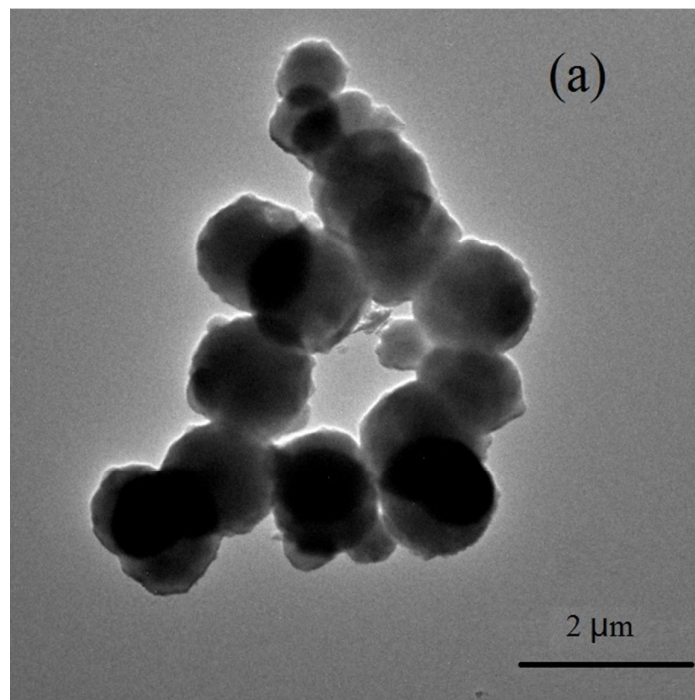


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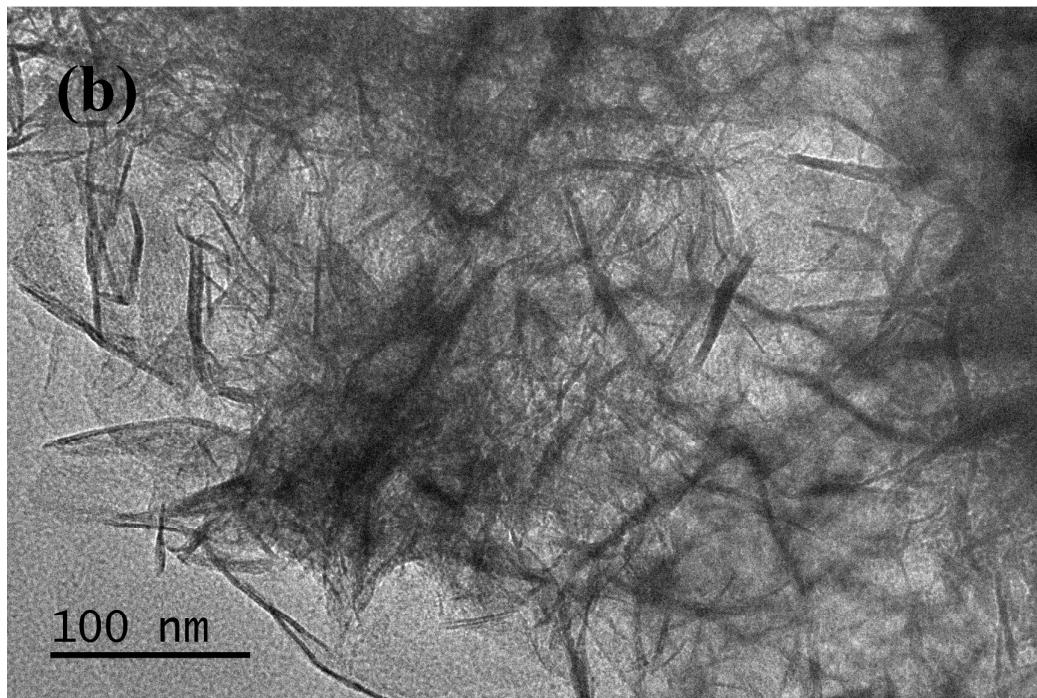


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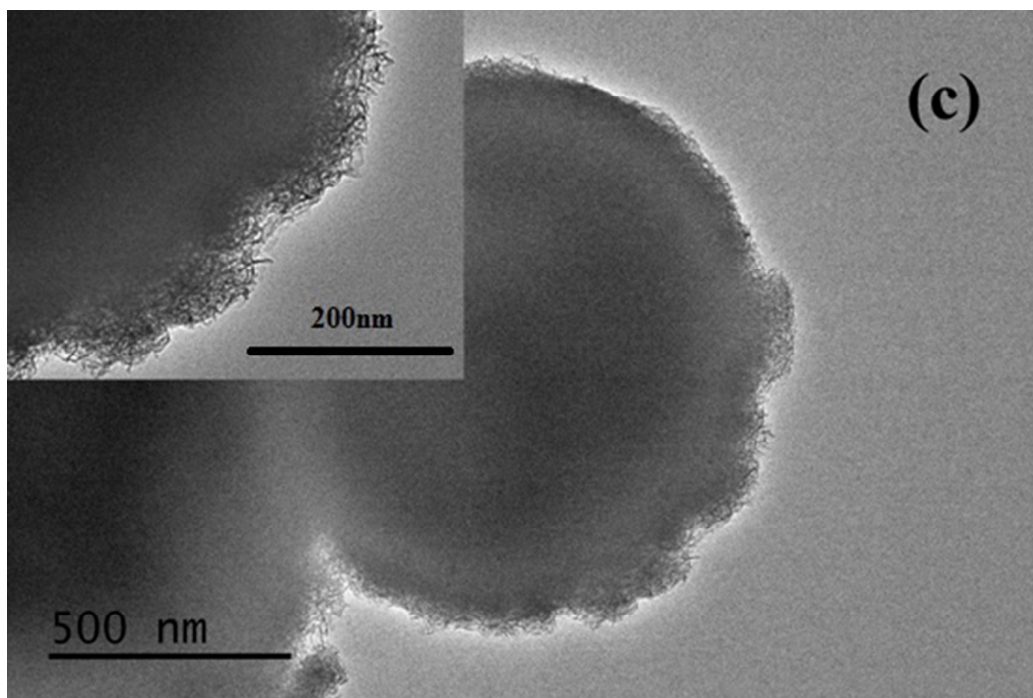


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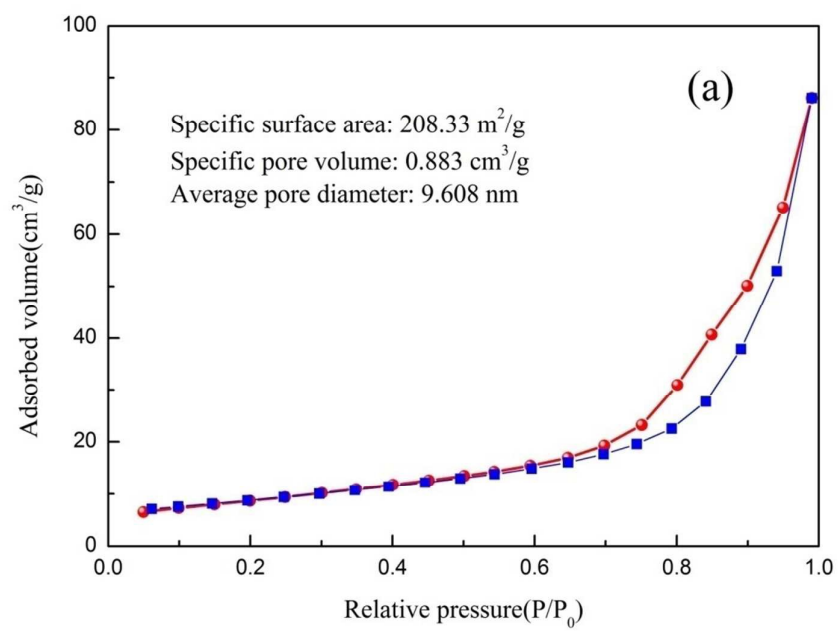


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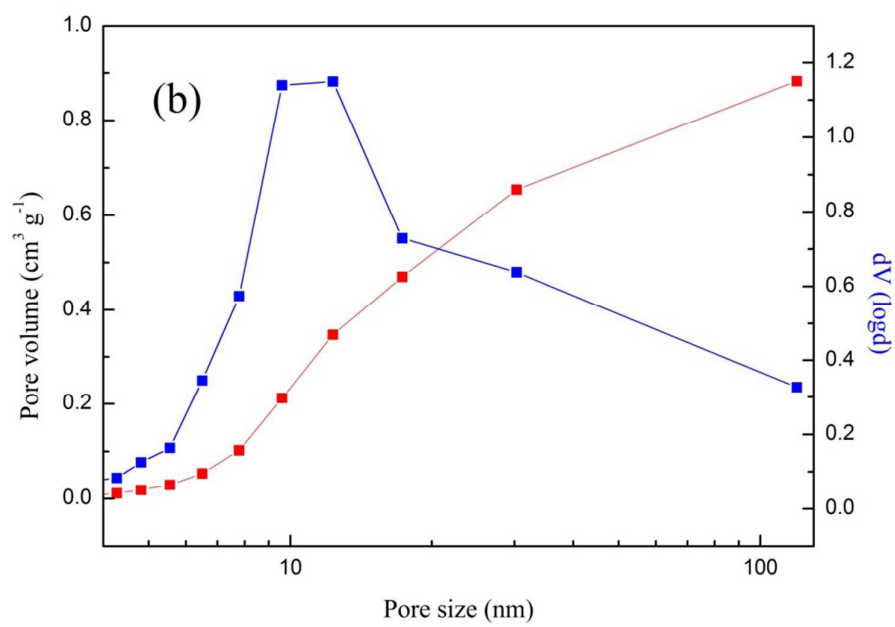


Fig. 3 (c)

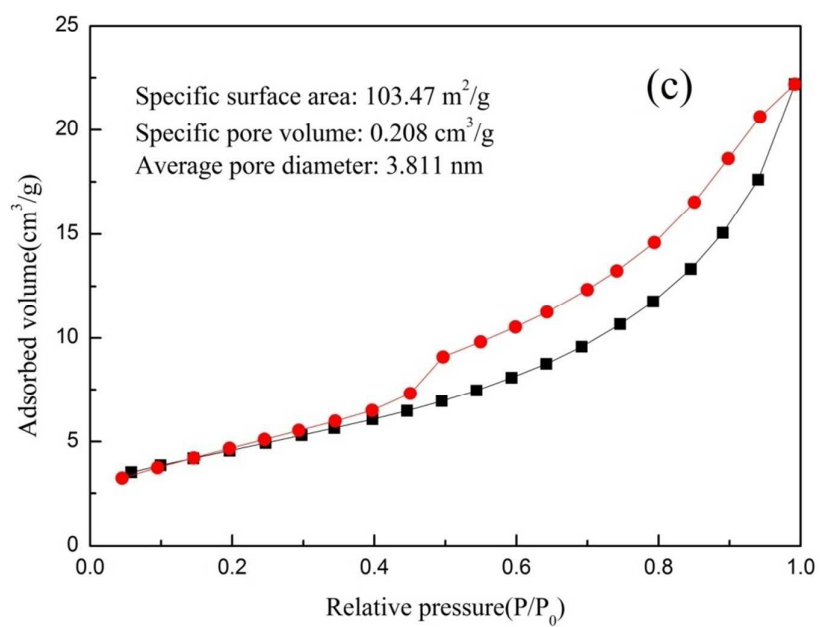


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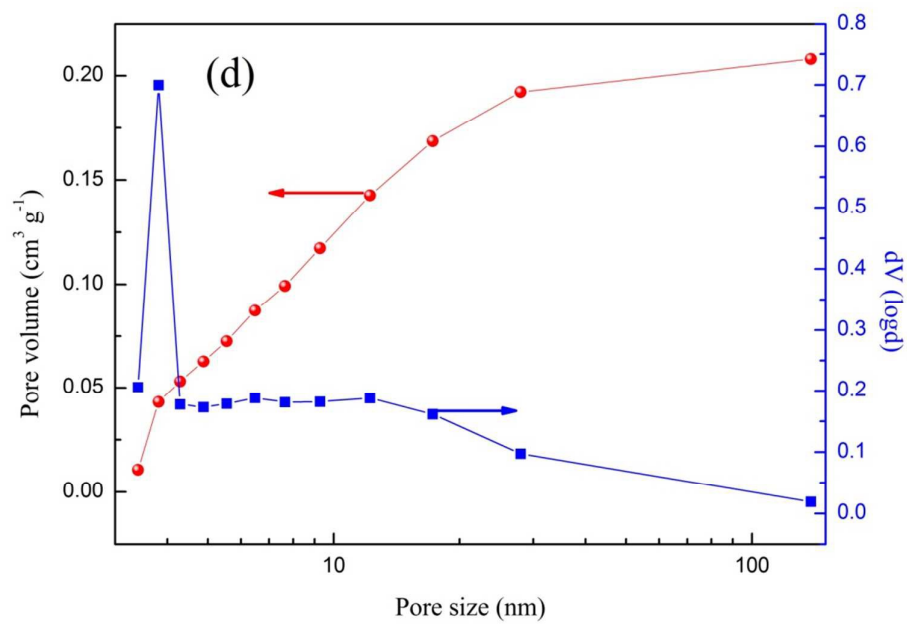


Fig. 4 (a)

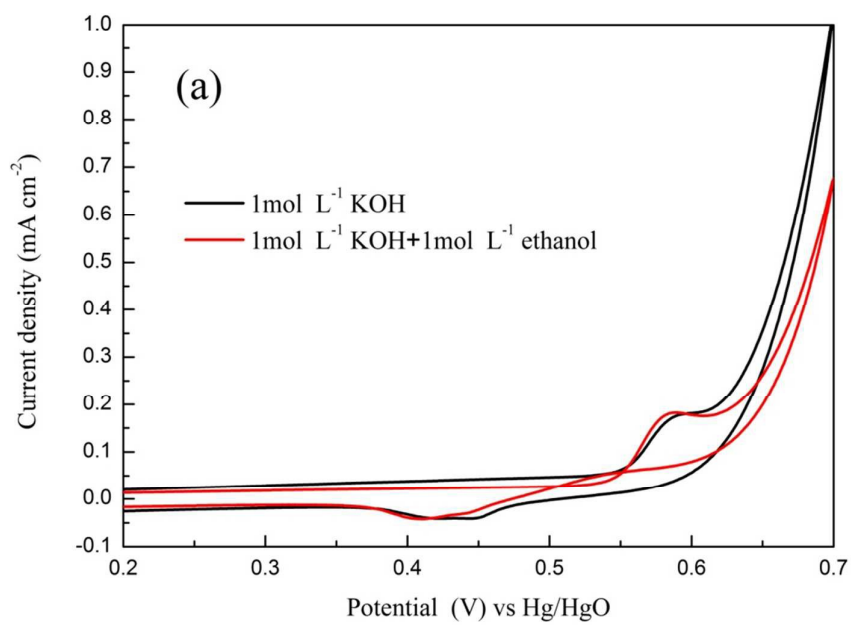


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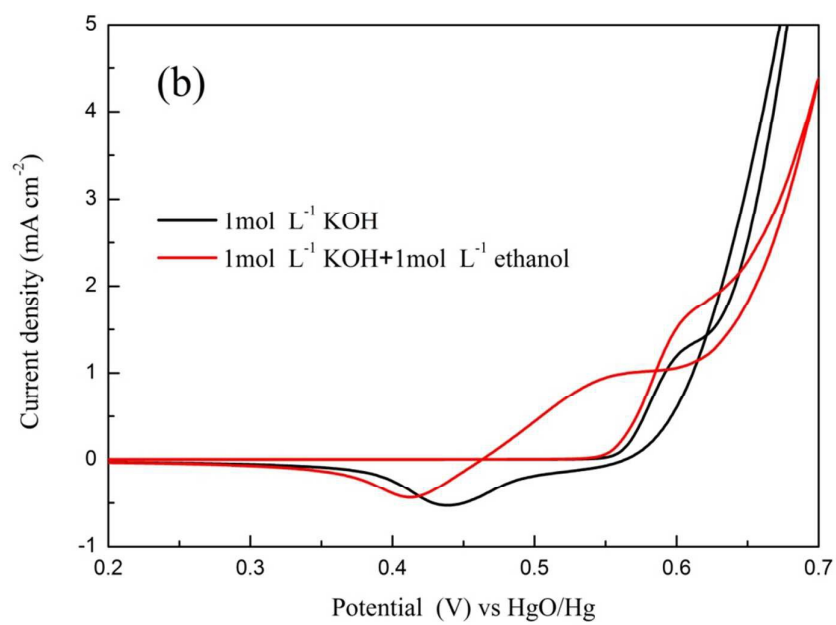


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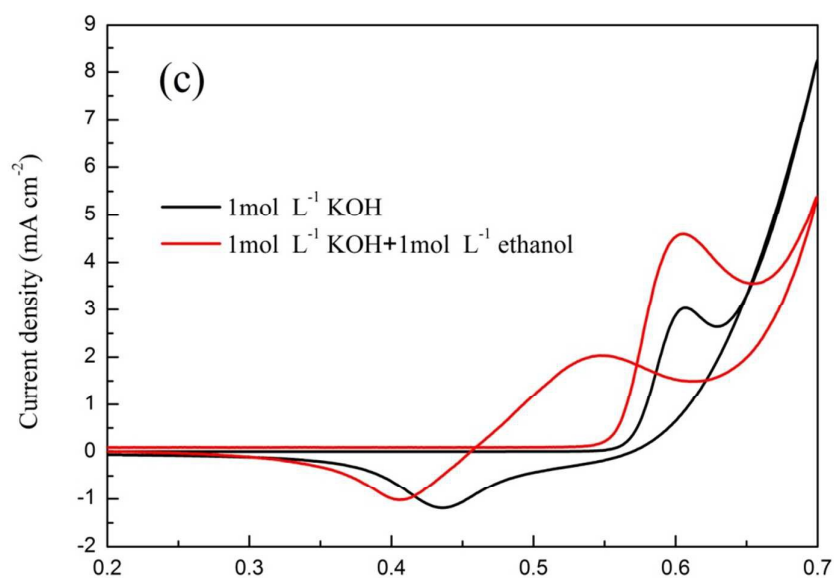


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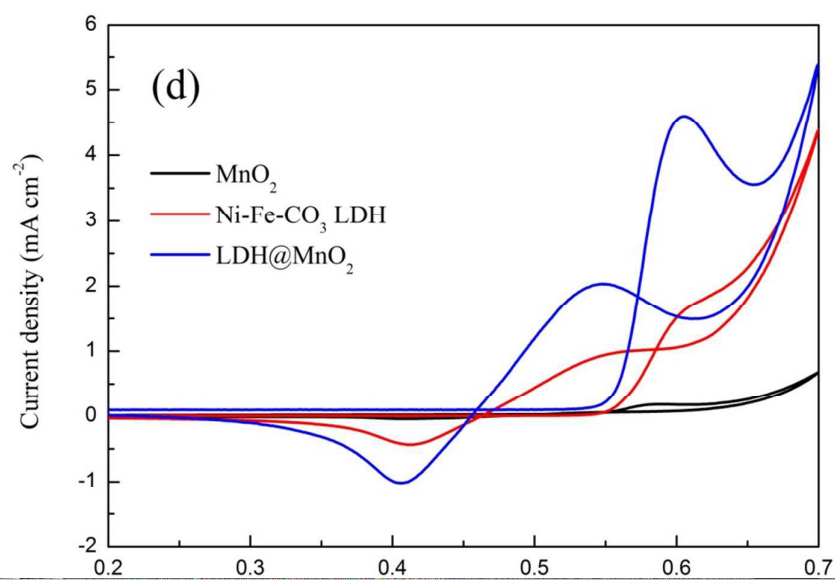


Fig. 4 (e)

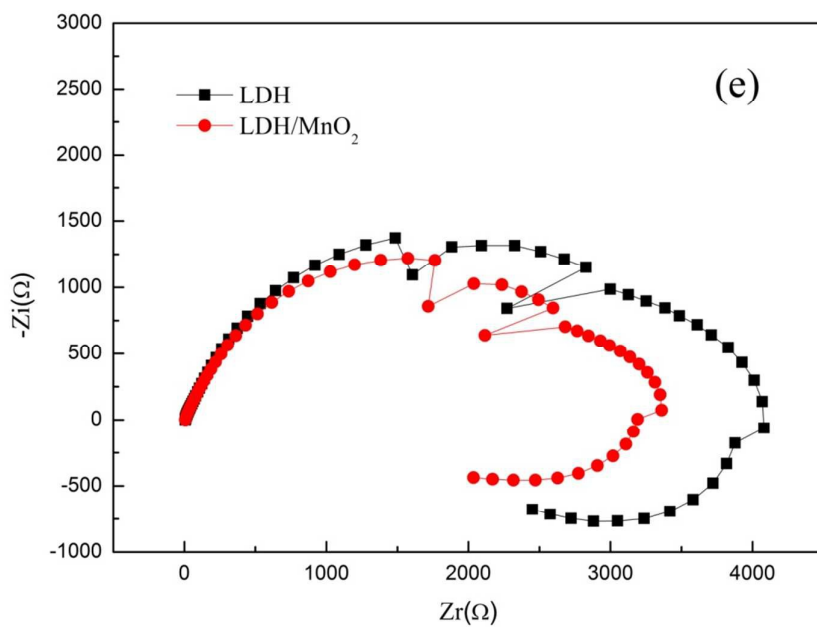


Fig. 4 (f)

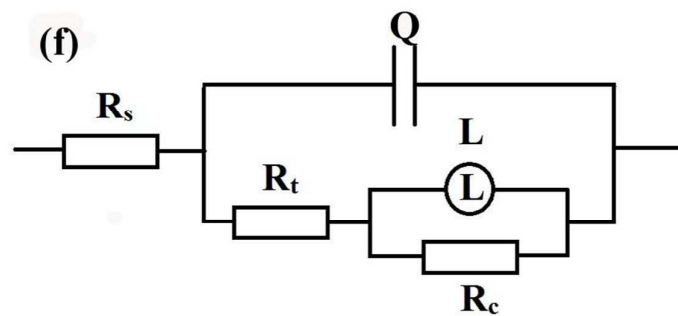


Fig. 4 (g)

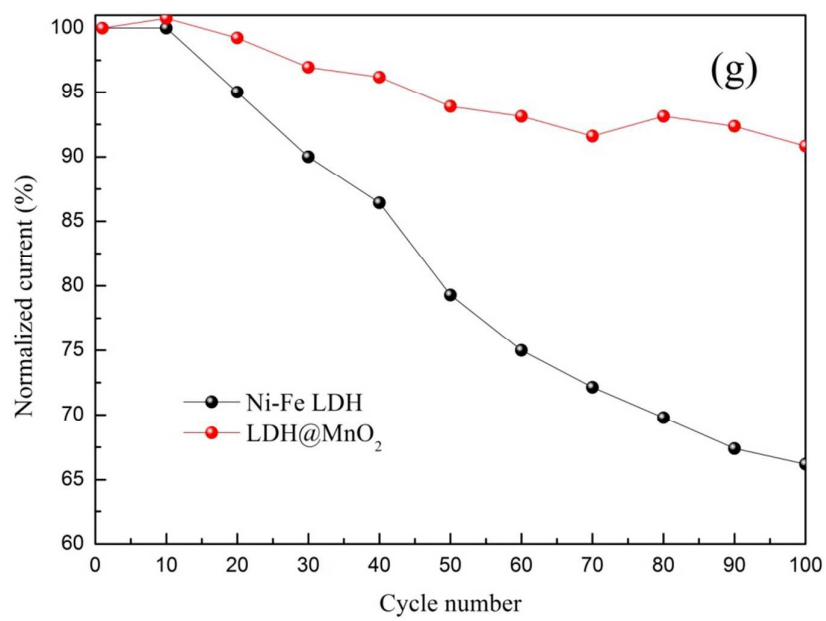


Fig. 4 (h)

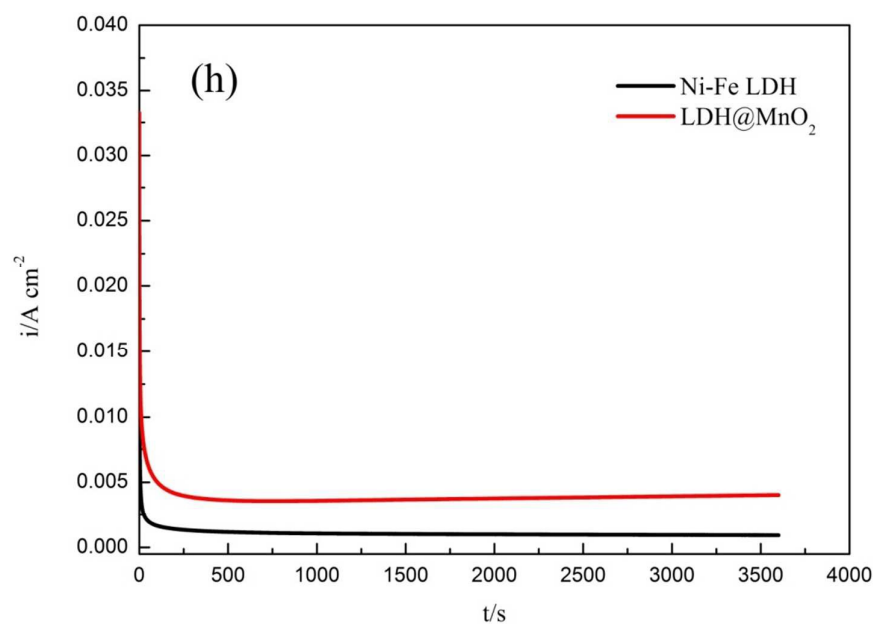


Figure caption

Table 1 Fitting results of EIS

Table 1

	$R_s(\Omega)$	Q (Y sec ⁻ⁿ)	n	$R_t(\Omega)$	$R_c(\Omega)$	L (H)
Ni-Fe LDH	6.149	2.88E-5	0.7833	2157	1747	1.463E4
LDH@MnO ₂	6.682	2.523E-5	0.7883	2053	1154	9610