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1 **Graphene oxide chemically decorated with Ag-Ru/chitosan nanoparticles:**
2 **Fabrication, Electrode Processing and Immunosensing Property**

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8
9 **Abstract**

10 Nanosheets of graphene oxide is chemically decorated with hybrid nanoparticles of
11 silver-ruthenium bipyridine complex (Ag@[Ru(bpy)₃]²⁺) core and chitosan shell. Oxygenated
12 groups of graphene oxide and abundant amine groups of chitosan layer on the surface of
13 hybrid nanoparticles allowed the functionalization reaction. Changes in optical, chemical and
14 structural properties of graphene oxide due to hybrid nanoparticles were studied from
15 spectroscopy and microscopic techniques. Electrodes modified with hybrid nanoparticles-
16 graphene oxide (HNPs-GO) displayed an amplified steady-state anodic (*I_pA*) and cathodic
17 (*I_pC*) peak currents, with a correlation coefficients of 0.9987 (*I_pA*₁), 0.9952 (*I_pA*₂) and 0.9964
18 (*I_pC*₁). Using monoclonal antibody *Listeria monocytogenes* the HNPs-GO immunosensor
19 could specifically detect *Listeria monocytogenes* contaminated in buffer and milk, with a
20 concentration range from 10² to 10⁷ cells/mL and detection limit of 2 cells/mL. Our results
21 suggest that selective optimization of bio-recognition elements on the HNPs-GO electrode
22 may find prospective use in food process industries.

23 *Keywords: Hybrid graphene oxide; Ag-Ru complex; Chitosan; electrochemical*
24 *immunosensor; Listeria monocytogenes*

25

26 1. Introduction

27 Development of hybrid nanomaterials with multiple structures and chemical
28 composition has attracted many researchers toward the advancement of functional
29 properties.¹ Recently, two-dimensional graphene oxide (GO) and reduced GO (rGO) have
30 been used in a variety of applications due to their cost-effective fabrication, ultra-thin layers,
31 large surface area and tunable oxygen functional groups.^{2,3} Surface treatment and
32 functionalization of active components on GO nanosheets influence the inherent sp^2/sp^3
33 carbon domains, resulting the change of crystallite size, lattice orientation and associated
34 properties.^{3,4} The different strategies employed to tune the physico-chemical and biomedical
35 functionality of GO are photoirradiation, elemental doping and chemical anchoring of
36 inorganic/organic materials.³⁻⁶

37 Significant effort has been devoted in the advancement of hierarchical hybrid GO
38 nanostructures. In particular, ternary/quaternary nanocomposite comprised of graphene-
39 derivatives, metal, metal-oxide and polymer has recently been shown to have improved
40 physico-chemical properties optimal for device construction (e.g., electrode materials for
41 biosensor platform and energy conversion). The typical enhanced electrochemical properties
42 of metalloid polymer hybrid (Ag@SiO₂-PEG)-GO,⁵ graphene/WO₃/Au⁷ and polyaniline-
43 Fe₂O₃-rGO⁸ are better suited to biosensor studies than the individual pristine derivatives.
44 Molecularly imprinted polymers based on CdTe/Cds and magnetic GO showed selective
45 recognition toward environmental pollutants.⁹ Pt-graphene-TiO₂¹⁰ and reduced GO-bismuth
46 ferrite (Bi₂Fe₄O₉)¹¹ have been reported to have better photocatalytic properties. Further,
47 studies show that the hierarchical structures of SnS₂-rGO-TiO₂/TiO₂ layered films¹² and
48 rGO/Fe₃O₄@SiO₂@polyaniline¹³ significantly improve photoelectric and electrochemical
49 properties, respectively. The accumulation of evidence indicates that the fabrication of hybrid
50 GO material has great potential for opto/electrochemical device development.

51 Nevertheless, achieving a durable structure of hybrid GO with inbuilt multi-
52 functionality is complicated. Physically-linked hybrid nanostructures are prone to leaching
53 and decreased synergistic functionality. Compared to physical adsorption of nanostructures
54 on GO surface, chemically-bonded active materials are expected to have durable
55 electrochemical properties and be more stable.^{5,6} However, only few studies have
56 demonstrated the chemical functionality between active materials and GO surface.
57 Incorporation of a durable single hybrid nanostructure on GO surface with optical,
58 electrochemical and biocompatible capabilities would be highly useful for various biosensing
59 applications. Previous work has shown that single hybrid core-shell nanoparticles made of a
60 metal-dye complex (AgNPs@[Ru(bpy)₃]²⁺) core and biopolymer (chitosan) shell can
61 influence optical, electrochemical and biocompatibility due to the electrical conductivity of
62 Ag, metal-to-ligand charge-transfer of [Ru(bpy)₃]²⁺ and abundant amino groups of chitosan.¹⁴
63 The three-in-one hybrid¹⁴ nanosystem (average particle size 54 nm) on the surface of GO
64 would be an ideal candidate for modification; it is multifunctional due to its opto-electronic
65 and biocompatible nature. GO chemically decorated with hybrid nanoparticles (HNPs) is
66 expected to have better optical, redox activity and biocompatible functional groups suitable
67 for various sensor studies. For example, introduction of Ag metal on GO improves the
68 electron transfer process and increases immunosensing ability.¹⁵ The presence of metallic and
69 hydrated Ru on the surface of GO electrodes enhanced electrochemical performance.^{16,17} As a
70 bio-derived linear polysaccharide with biocompatibility, biodegradability and film-forming
71 ability, chitosan is explored as an interface layer in the fabrication of chemically-modified
72 electrodes for biosensing.^{18,19}

73 Chemical functionalization of distinct materials on the surface of GO is highly
74 dependent on the reactivity of oxygen functional groups which exist on the edges and basal
75 planes of GO. Here, for the first time, the three-in-one HNPs of Ag@[Ru(bpy)₃]²⁺/chitosan

76 are used to chemically decorate the GO nanosheets. Abundant amino groups of chitosan-
77 coated on the surface of $\text{Ag}@\text{[Ru(bpy)}_3\text{]}^{2+}$ provided a significant modification on the
78 oxygenated edges/basal planes of GO. The influence of optical absorbance,
79 photoluminescence, zeta potential and structural integrities including morphology, chemical
80 structure and Raman shift of pristine GO and HNPs-GO materials were extensively studied to
81 understand the properties. Inherent electrochemical redox properties were studied on the
82 customized electrode modified with HNPs-GO.

83 As a proof-of-concept, electrochemical immunosensing property of HNPs-GO
84 electrodes has been demonstrated for the detection of *Listeria monocytogenes* (*Lm*). *Lm* is an
85 important Gram-positive rod-shaped food borne bacterium that causes an extremely life
86 threatening infection, listeriosis (high mortality rate ~30%).²⁰ Listeriosis is most prevalent in
87 pregnant women and their newborns, older adults >65 and people with weakened immune
88 systems. Recent report by CDC (Centers for Disease Control and Prevention), USA (dated
89 June 2015) declared a multistate outbreak of listeriosis linked to food poisoning.²¹
90 Government of Canada estimates that there are about 4 million cases of food borne illness
91 every year. Common symptoms associated with food poisoning include fever, vomiting,
92 diarrhoea and headache. In severe cases meningitis, septicaemia and abortion can occur.²⁰
93 Therefore a rapid and sensitive detection of *Lm* in food products is vital. Conventional cell
94 culture, microscopy, biochemical tests and luminescence methods are labor-intensive and
95 time-consuming. Although ELISA (enzyme-linked immunosorbent assay) and PCR
96 (polymerase chain reaction) may greatly reduce the assay time, they still consume hours to
97 identify target pathogens and lack the ability of real time detection.²² Electrochemical based
98 immunosensors are among the most promising because of their high specificity, rapid
99 analysis and field deployable possibility.^{20,22-24} Herein, electrochemical immunosensing of *Lm*
100 is proposed using an immobilized mouse monoclonal anti *Listeria monocytogenes* antibody

101 (anti-*Lm*) on the surface of HNPs-GO electrodes. Phosphate buffered saline and milk samples
102 were artificially contaminated with *Lm* cells and utilized for sensing studies. Cross-reactivity
103 of the proposed immunosensor to other bacterial strain (*Pseudomonas aeruginosa*, *Pa*) is also
104 measured and found to be insignificant. In addition to synergistic electrochemical property,
105 the biocompatible groups derived from HNPs-GO promote the affinity toward the
106 immobilized antibody, thereby enabling a sensitive detection of *Lm* contamination.

107 **2 Experimental Section**

108 **Chemicals**

109 Silver nitrate (AgNO_3), 3-mercaptopropionic acid (3-MPA), sodium borohydride
110 (NaBH_4), tris(2,2'-bipyridyl)dichloro ruthenium(II) hexahydrate, chitosan (low molecular
111 weight: 50 000-190 000 g mol^{-1} ; degree of deacetylation: 75-85%), graphite powder (<20 μm ,
112 synthetic), glutaraldehyde solution and phosphate buffered saline (PBS) were purchased from
113 Sigma-Aldrich. Mouse monoclonal IgG antibodies to *Listeria monocytogenes* was purchased
114 from AbD Serotec (Bio-Rad). Milk sample was obtained from local grocery store. Other
115 chemicals were of analytical grade and used as received without any further purification.
116 Milli-Q water (18.2 $\text{M}\Omega$) was used for all experiments.

117 **Synthesis of hybrid ($\text{Ag}@\text{[Ru(bpy)}_3\text{]}^{2+}/\text{chitosan}$) NPs**

118 Hybrid NPs of $\text{Ag}@\text{[Ru(bpy)}_3\text{]}^{2+}/\text{chitosan}$ were prepared according to the reported
119 procedure.¹⁵ At first, 5 mL of AgNO_3 (0.1 M), 25 mL of 14 N aq. NH_4OH and 5 mL of 3-
120 MPA (50 mM) were dissolved in 15 mL of deionized (DI) water (solution A). Separately, 5
121 mL of NaBH_4 (0.02 M) and 2 mL of 14 N aq. NH_4OH were dissolved in 15 mL of DI water
122 (solution B). At room temperature, solutions of vial A and B were slowly injected dropwise
123 into the 300 mL of DI water over 30 min with a magnetic stirring of 600 rpm. After 30 min of
124 reaction time, colloidal solution containing AgNPs-modified 3-MPA was separated by

125 centrifugation (13,000 rpm for 1 hr). The particles were then washed twice with DI water and
126 dispersed in DI water for further reaction.

127 $[\text{Ru}(\text{bpy})_3]^{2+}$ coating on AgNPs was achieved by mixing the above AgNPs-modified 3-
128 MPA (5 mL, 1 mg/mL) and ethanolic solution of $\text{Ru}(\text{bpy})_3\text{Cl}_2$ (5mL, 0.8 mg/mL). The
129 reaction mixture was left overnight under mild stirring, protected from light. Resulted
130 particles were centrifuged (13,000 rpm for 1 hr) and washed twice with ethanol and DI water
131 to remove unreacted $[\text{Ru}(\text{bpy})_3]^{2+}$. Prepared $\text{Ag}@[\text{Ru}(\text{bpy})_3]^{2+}$ were then surface modified
132 with chitosan by coordination chemical reaction using $\text{Ag}@[\text{Ru}(\text{bpy})_3]^{2+}$ (5 mL, 1 mg/mL)
133 and chitosan (5 mL, 0.01 wt%) under magnetic stirring of 600 rpm for 3 hrs at room
134 temperature. The final hybrid ($\text{Ag}@[\text{Ru}(\text{bpy})_3]^{2+}/\text{chitosan}$) NPs were isolated by
135 centrifugation (13,000 rpm for 1 hr), washed and re-dispersed in DI water for further
136 experimentation.

137 **Functionalization of HNPs on GO nanosheets**

138 Colloidal dispersions of GO nanosheets used in the current experiment were
139 synthesized according to the modified Hummers' method.²⁵ Functionalization of HNPs onto
140 the surface of GO sheets was achieved through a one-step process. An aqueous dispersion of
141 GO (25 mL, 0.5 mg/mL) and HNPs (25 mL, 2 mg/mL) was added to a reaction flask and kept
142 under magnetic stirring (600 rpm) at room temperature for 12 hrs. After the reaction time, the
143 HNPs-functionalized GO sheets were separated by centrifugation (13,000 rpm, 1 hr), washed
144 thrice with DI water and utilized for characterization.

145 **Construction of HNPs-GO sheets modified electrode**

146 An integrated gold printed circuit board (Au-PCB) chip served as the electrode system.
147 The central circle-shaped Au substrate with an area of 2 mm in diameter was used for the
148 modification of HNPs-GO sheets. The two crescent-shaped Au substrates with a length of 4.3
149 mm and a breadth of 0.8 mm were used as counter and reference electrodes, respectively.

150 After performing sequential washing with acetone, ethanol and DI water, the Au-PCB chip
151 was exposed to plasma treatment. Then, typically 4 μL of the aqueous dispersion of HNPs-
152 GO (1 mg/mL) was drop casted on the working substrate. To make uniform surface
153 modification of HNPs-GO sheets on the electrode surface, typically three layers of casting
154 were performed at regular intervals with an evaporation period of 1 hr at ambient
155 temperature.

156 **Fabrication of anti-*Lm* modified HNPs-GO electrodes**

157 HNPs-GO electrodes were first modified with 4 μL of 1.25% (vol/vol) PBS solution
158 of glutaraldehyde and kept in ambient temperature for 30 min, which allows the reaction
159 between amine groups of chitosan layer on the HNPs-GO, resulting in an aldehyde functional
160 group at the surface. The terminal aldehyde surface was then coupled to diluted anti-*Lm* (4
161 μL , 100 $\mu\text{g}/\text{mL}$ in PBS, pH 7.4) and kept at 4 $^{\circ}\text{C}$ for 1 h. After this, the modified electrode
162 surface was gently immersed into a PBS buffer (pH 7.4) for 30s, allowing the diffusion of
163 unbound antibodies away from the electrode surface.

164 **Preparation of bacterial cells and amperometric detection**

165 Bacterial cell suspensions (*Lm* and *Pa*) were prepared from overnight cultures, grown
166 in 4 mL of tryptone soy broth culture medium (TSB, Oxoid Canada, Nepean, Ontario). The
167 bacterial cultures were then isolated and washed twice in DI water by centrifugation at 4,500
168 rpm for 10 min at room temperature. Afterwards, the bacterial cells were resuspended in PBS
169 for further analysis. An optical density measurement at 600 nm was done for the stock
170 bacterial suspension and the desired test samples of various cell numbers were diluted
171 appropriately. Chronoamperometric detection of bacterial samples on the surface of HNPs-
172 GO/anti-*Lm* electrodes was measured at an applied potential of +0.55V. Typically 20 μL of
173 test samples (contaminated PBS and milk) were utilized onto the electrode surface.

174

175 Instrumentation

176 Cary 5000 UV-Vis-NIR spectrophotometer (Agilent Technologies) was used to analyze
177 the UV-vis absorbance spectra. Morphological characterizations were observed via a
178 transmission electron microscope (TEM) (Philips Tecnai 12) with an acceleration voltage of
179 120 kV. Samples used for imaging were prepared by casting 4 μL of (0.25 mg/mL) HNPs,
180 GO or HNPs-GO suspension onto a carbon-coated nickel grid. Zeta potential was studied
181 from Zetasizer Nano ZS (Malvern Instruments) equipped with a 4 mW, 633 nm He-Ne laser
182 using appropriate cells. Measurements were conducted in backscattering (173°) mode and
183 detected with an Avalanche photodiode. For accurate determination of zeta potential, thirteen
184 runs were averaged for each liquid sample. A Varian Cary Eclipse Fluorescence
185 spectrophotometer was used to examine the photo-luminescence properties of HNPs, GO and
186 HNPs-GO. The chemical structure and functional group modifications on pristine and hybrid
187 materials were identified by Fourier transform infrared (FTIR) spectra studied on a Nicolet
188 6700 FTIR spectrometer (in the ATR mode, diamond crystal). $^1\text{H-NMR}$ spectra in deuterated
189 dimethyl sulfoxide- d_6 (DMSO d_6) were measured on a Bruker AV 400 spectrometer
190 operating at 400 MHz (number of scan: 256). Raman spectral analysis was performed in
191 RENISHAW inVia Raman microscope equipped with CCD camera and a Leica microscope.
192 Aqueous dispersion of sample (~ 1 mg/mL) was drop casted on a cleaned silica wafer and
193 utilized for measurements. An excitation wavelength of 514 nm and laser power of 10% was
194 used. A short working distance 50 \times objective lens was used to focus the laser spot on the
195 sample surface. Measurements were taken in 30s of exposure time at varying numbers of
196 accumulations. Electrochemical properties of the pristine GO and HNPs-GO materials were
197 studied from the cyclic voltammetric technique using SP-150 potentiostat, Bio-Logic
198 instruments. All the cyclic voltammograms (CVs) were recorded in the 10 mM PBS solution
199 (pH 7.4) as supporting electrolyte, in the potential region between -0.25 to $+0.8$ V. A

200 reproducible voltammogram can be obtained under steady-state conditions after about five
201 cycles.

202 **3 Results and discussion**

203 **Synthesis of Ag@[Ru(bpy)₃]²⁺/chitosan NPs and functionalization on GO**

204 Fig. 1 illustrates the step-wise synthesis route for obtaining HNPs. HNPs made of a
205 metal-dye complex (Ag@[Ru(bpy)₃]²⁺) core and a chitosan shell are firmly bonded to each
206 other by electrostatic and coordination interaction, respectively.¹⁴ The thin layer of chitosan
207 on the surface of HNPs with abundant amine groups are reactive to the oxygenated functional
208 groups of GO. The presence of carboxyl and epoxy groups at the edges and basal planes of
209 GO provided multiple binding sites for chemical functionalization of HNPs. The two
210 important surface chemical reactions involved in this functionalization were formation of
211 amidation at the carboxyl groups and nucleophilic attack at the α -carbon by the HNPs. The
212 structural and chemical changes which resulted from the functionalization process were
213 characterized by FT-IR and ¹H-NMR spectroscopy as described later.

214 **UV-vis absorbance and photoluminescence**

215 In general, the optical absorption and emission band of metal or metal hybrid
216 nanostructures depend on the size, shape, nature of the surface functional layer and solvent
217 environment.²⁶ Here, an UV-vis absorbance spectroscopy was utilized to measure the optical
218 information of the prepared materials. Fig. 2A represents the spectra observed from aqueous
219 AgNPs and AgNPs-modified with 3-MPA. The peak at 402 nm denotes the existence of
220 characteristic surface plasmon resonance (SPR) of AgNPs. An SPR is the collective
221 oscillations of the conductive electrons that exist on the surface of metal NP. Depending on
222 the excitation of the localized surface plasmon, caused by strong light scattering at a specific
223 wavelength, strong SPR bands are produced.²⁷ At 423 nm, a significant red shift was
224 mediated by surface modification of AgNPs with 3-MPA. UV-vis absorption spectrum of

225 [Ru(bpy)₃]²⁺ shows the three specific peaks at 242, 290 and 450 nm (Fig. 2B) ascribed to
226 intra-ligand transition $\pi \rightarrow \pi^*$, bpy $\pi \rightarrow \pi_1^*$ transition and metal-to-ligand charge-transfer
227 (MLCT) band, respectively.²⁸ A shoulder peak at 420 nm is also attributed to MLCT (t_{2g}
228 (Ru) $\rightarrow \pi^*$ (bpy) transitions). Similarly, AgNPs modified with [Ru(bpy)₃]²⁺ also exhibit three
229 absorbance peaks with a moderate hump located at 423 nm due to the overlap of SPR from
230 AgNPs.²⁹

231 As shown in Fig. 2C, HNPs exhibited significant changes in the peak shape at 242, 290
232 and 450 nm, indicating that the chitosan modification altered the optical absorbance of
233 Ag@[Ru(bpy)₃]²⁺.¹⁴ Aqueous dispersion of GO nanosheets exhibited a wavelength of
234 maximum absorbance at 230 nm attributed to the $\pi \rightarrow \pi^*$ electron transition of the
235 polyaromatic C-C bonds of GO layers.⁵ The UV-vis absorbance spectrum of HNPs
236 functionalized GO exhibits peaks centered at 240, 284 and 450 nm. Compared to pristine
237 HNPs, the $\pi \rightarrow \pi^*$ electron transition signals of HNPs-GO are well resolved, probably due to
238 the associated signals of metal-dye complex and C-C bonds of GO. There is no significant
239 change observed from the bpy $\pi \rightarrow \pi_1^*$ transition peak position, however the peak centered at
240 450 nm is much more broad than that of pristine HNPs.

241 Excitation of specific wavelengths of light on the aqueous dispersion of optically active
242 nanomaterials could provide additional information such as photoluminescence (PL). It is
243 known that chemical oxidation of graphite results in the formation of mixed sp^2/sp^3 domains
244 in GO lattice, which creates a disruption of the π -network and generates an emission band.³⁰
245 The PL spectrum of GO and HNPs-GO was recorded using an excitation wavelength of 325
246 nm and is shown in Fig. 3. GO shows a sharp emission peak in the near UV region at around
247 365 nm due to the amorphous sp^3 matrix that surrounds the various graphitic sp^2 domains,
248 which act as a high tunnel barrier resulting in the generation of a band gap in GO. This is in
249 agreement with previous reports on PL of GO.^{30,31} Upon modification with HNPs, the near

250 band emission is quenched with slight broadening of the peak centered at 362 nm. This
251 observation is probably due to the formation of new metallic hybrid clusters on the GO
252 lattice. The observation of PL from HNPs-GO implies the existence of a band gap in the
253 electronic structure of the material. Recent studies identified that the surface modification of
254 GO could create a large band gap and decent carrier mobility suitable for advanced PL³² and
255 electrochemical biosensor.³³ As-prepared chemically decorated GO containing ternary
256 composite of metal-dye complex and biopolymer retained the inherent PL property and
257 electronic structure, and is expected to be a feasible option for dual (optical/electrochemical)
258 sensors.

259 **Morphology and surface zeta potential characterization**

260 Morphology of HNPs, GO and HNPs-GO nanostructures were visualized from TEM
261 and are shown in Fig. 4. HNPs with an overall spherical shape and coating of chitosan layer
262 are clearly visible in Fig. 4(A-B). Observed trace of particle's aggregation is possibly due to
263 the drying process done before imaging. Average particle size distribution of HNPs was
264 determined using the Malvern-dynamic light scattering-Zetasizer Nano ZS instrument and
265 found to be 54 nm (data not shown). Surface topography of GO (Fig. C-D) displays the
266 corrugated thin sheet-like membranous layer. The typical thin grooves or wrinkles on the
267 sheets are characteristic of GO nanostructures. Due to its two-dimensional thin layered
268 feature with reactive oxygenated groups, GO allows multiple chemical bonding with amine-
269 functionalized HNPs. HNPs were well decorated on the surface of GO nanosheets (Fig. 4 E-
270 F).

271 Zeta potential is a vital physical property used to study the stability of colloidal
272 dispersions and surface charge associated with the double layer around the colloidal particle.
273 The zeta potential varies depending on which chemical groups exist on the surface of
274 colloidal particles (supporting Fig. S1). Due to the ionization of the multiple surface

275 oxygenated functional groups, pristine GO showed the negative zeta potential of -39 mV.
276 HNPs containing chitosan shell with abundant amine groups displayed the positive zeta
277 potential of $+46.1$ mV. Upon surface functionalization, the zeta potential of HNPs-GO was
278 $+26.6$ mV, indicating that the chemically bonded HNPs modified the inherent surface zeta
279 potential of the GO nanosheets. These results supplement the morphological images. Such
280 modified hierarchical GO sheets with single hybrid of metal-dye complex and biopolymer
281 provide a unique set of physico-chemical properties that are promising for multi-functional
282 material.

283 FTIR and $^1\text{H-NMR}$ spectroscopy

284 In order to evaluate the chemical structure and functional group modifications on HNPs
285 or HNPs-GO, a comparative FTIR spectral analysis was performed on pristine chitosan and
286 GO samples. Fig. 5(A) shows the FTIR spectrum of chitosan: the C-H out of plane bend at
287 887 cm^{-1} and C-O stretch at 1026 and 1065 cm^{-1} . C-O-C stretch and C-H bend are located at
288 1154 and 1377 cm^{-1} . Vibrations at 1590 and 2870 cm^{-1} are attributed to N-H bend and C-H
289 stretch. The broad peak centered at 3317 cm^{-1} is associated with the N-H stretch and
290 hydrogen-bonded OH groups.^{34,35} HNPs containing chitosan modified $\text{Ag}@\text{[Ru(bpy)}_3\text{]}^{2+}$ (Fig.
291 5B) exhibit significant alterations in their group frequencies. For instance, the C-O stretch
292 shows distinct changes at 1040 cm^{-1} when compared with pristine chitosan. The primary N-H
293 bend at 1590 cm^{-1} is shifted to 1644 cm^{-1} , denoting the formation of a secondary amine. A
294 short but sharp peak at 2970 cm^{-1} is ascribed to the asymmetric stretching of C-H.³⁴ Further,
295 the N-H stretch and H-bonded OH stretch were much more intense than that obtained for
296 chitosan. Observed modifications in the group frequencies (C-O stretch and N-H bend) of
297 chitosan support their chemical bonding with $\text{Ag}@\text{[Ru(bpy)}_3\text{]}^{2+}$.¹⁴

298 The FTIR spectrum of GO samples (Fig. 5C) reveals the peaks relating to C-O
299 (carbonyl) at 1040 cm^{-1} , the C-O-C epoxy group frequencies at $1175\text{-}1250\text{ cm}^{-1}$ and

300 carboxyl-associated OH signal at 1405 cm^{-1} .^{13,36} Well resolved peaks at 1600 and 1725 cm^{-1}
301 are assigned to the C-C vibrations of un-oxidized graphitic domains and C=O stretching
302 vibrations, respectively.^{5,36} The relatively broad peak centered at 3240 cm^{-1} is associated with
303 the adsorbed water on the surface of the GO. As discussed previously, HNPs are expected to
304 form chemical bonds at the basal planes and edges of GO. After functionalization with HNPs,
305 the carbonyl peak of GO is broadened and shifted from 1040 cm^{-1} (Fig. 5C) to 1068 cm^{-1} (Fig.
306 5D). The epoxy group frequencies were almost dispersed and the peaks of carboxyl-
307 associated OH, C-C graphitic domains and C=O stretching vibrations were also changed.
308 Further, a new peak centered at 771 cm^{-1} , along with two shoulder peaks (735 and 830 cm^{-1})
309 attributed to C-H of chitosan,¹⁴ was observed. This provides the supporting information for
310 the functionalization of HNPs on the GO surface.

311 To gain further understanding of the chemical structure of pristine and hybrid
312 nanostructures, $^1\text{H-NMR}$ spectral analysis was utilized. Fig. 6A shows the $^1\text{H-NMR}$ spectrum
313 of HNPs, which exhibits the characteristic resonance peaks attributed to the functional groups
314 of chitosan such as $-\text{CH}-\text{CH}-$ (0.8 and 1.2 ppm), $-\text{NH}-$ (1.9 ppm) and OH (5.3 ppm).³⁷
315 Proton signals of GO nanostructures (Fig. 6B) are identified at 1.2 , 4.5 , 8.1 and 9.5 ppm and
316 attributed to the $-\text{CH}-\text{CH}-$, OH , $-\text{C}-\text{COOH}-$ and $=\text{C}-\text{COOH}$, respectively.^{37,38} The spectrum
317 of HNPs-GO (Fig. 6C) also shows the inherent $-\text{CH}-\text{CH}-$ protons. The peak shift located at
318 1.9 - 2.0 ppm , attributed to the amine protons of chitosan, is relatively weaker than that of
319 pristine HNPs (Fig. 6A), indicating the functionalization of HNPs on GO. Successful
320 chemical bonding of HNPs on the oxygenated functional groups of GO are validated by the
321 absence of free carboxyl proton signals (at 8.1 and 9.5 ppm) and appearance of multiple
322 amide proton signals (at 6.5 , 6.9 , 7.1 and 7.2 ppm). The presence of reactive epoxy and
323 carboxyl groups on the GO lattice structures offered the necessary binding sites for the
324 chemical decoration of HNPs.

325 Raman spectroscopy

326 Raman spectral analysis further revealed the structural integrity of GO after chemical
327 interaction with HNPs. The typical characteristics of Raman spectra of graphite materials are
328 a G-band at 1570 cm^{-1} attributed to the E_{2g} phonon of sp^2 C domains³⁹ and a D-band at 1345
329 cm^{-1} attributed to the vibrations of disordered C domains of graphite.^{36,39} The presence of D-
330 band at 1355 cm^{-1} and a G-band at 1596 cm^{-1} supports the oxygenation of graphite (Fig. 7).
331 Chemically decorated HNPs on the surface of the GO lattice displayed a broadened D-band at
332 1355 cm^{-1} and G-band at 1583 cm^{-1} (red shifted from inherent 1596 cm^{-1}), respectively. A
333 slight change in the intensity ratio of the D- and G-bands (I_D/I_G) of HNPs-GO (0.85)
334 compared to that in GO (0.81) indicated that functionalization of HNPs altered the in-plane
335 sp^2 graphitic domains of GO. According to an empirical formula known as the Tuinstra-
336 Koenig relation,⁴⁰ the average crystallite size of the ordered graphitic sp^2 C domains can be
337 calculated using the following equation,

$$L_a(\text{nm}) = [(2.4 \times 10^{-10}) (\lambda_1)^4] / [I_{(D)}/I_{(G)}]$$

338 where L_a is the average crystallite size of the sp^2 domains, λ_1 is the input laser energy, $I_{(D)}$ is
339 the intensity of the D band, and $I_{(G)}$ is the intensity of the G band. The calculated L_a values are
340 20.7 and 19.7 nm for GO and HNPs-GO, respectively. Observed changes in the size of sp^2
341 hybridized domains are ascribed to the chemical interaction with HNPs. These results are in
342 agreement with similar reports on GO hybridized with metal oxide nanoparticles⁴¹ and
343 biomaterials.⁴²

344 Fabrication and redox properties of HNPs-GO electrodes

345 Compared to conventional electrodes, carbon electrodes modified with conductive
346 hierarchical nanostructures exhibit an enhanced electron transfer rate and more durable
347 electrochemical properties.^{7,33} Immobilization of bio-friendly conductive nanostructures with
348 active chemical groups suitable for anchoring antibody or enzyme is certainly valuable for

349 fabrication of label-free biosensor platforms.⁴³ To understand its feasibility as transducer
350 material for an electrochemical biosensor platform, the primitive electrochemical response of
351 the HNPs-GO was evaluated in comparison with pristine GO. Fig. 8 shows the pictorial
352 representation of an integrated three-electrode system used for modification of pristine GO
353 and HNPs-GO. Unlike conventional electrochemical systems, there is no external counter or
354 reference electrodes utilized in the present study. In order to find an optimal potential region
355 suitable for the prepared materials, a pre-screening CV measurement was performed between
356 -1.0 to $+1.0$ V. From the analysis, it was found that -0.25 to $+0.8$ V is an optimal potential
357 region for studying the redox behavior of the HNPs-GO modified Au-PCB electrodes.

358 Fig. 9A represents the CV curves of bare Au-PCB, pristine GO and HNPs-GO modified
359 Au-PCB electrodes recorded at a constant scan rate of 50 mV/s. A 10 mM PBS solution
360 containing the final concentration of 0.0027 M potassium chloride and 0.137 M sodium
361 chloride, with a pH 7.4 was used as the supporting electrolyte. Under experimental
362 conditions, bare Au-PCB and pristine GO electrodes don't exhibit significant redox behavior.
363 Pristine HNPs without GO as an interface layer are poorly stable on the Au-PCB substrate,
364 which resulted in leaching and hindered the durable electrochemical response (data not
365 shown). On the other hand, HNPs-GO modified electrode showed well-defined and highly
366 amplified anodic peaks A_1 at $+0.38$ V and A_2 at $+0.52$ V; the former is related to the
367 oxidation of $Ag \rightarrow Ag_2O$ ⁴⁴ and the latter is derived from the oxidation reaction of
368 $[Ru(bpy)_3]^{2+} \rightarrow [Ru(bpy)_3]^{3+}$. Interestingly, the HNPs-GO electrode displayed a single
369 cathodic peak at -0.12 V, suggesting a coherent reduction reaction of $Ag_2O \rightarrow Ag$ and
370 $[Ru(bpy)_3]^{3+} \rightarrow [Ru(bpy)_3]^{2+}$. The anodic peak current (A_2 : $I_p = +43 \mu A$) generated from the
371 oxidation reaction of $[Ru(bpy)_3]^{2+}$ is higher than the peak current of A_1 ($I_p = +33 \mu A$). This is
372 probably attributed to the existence of $[Ru(bpy)_3]^{2+}$ on the surface of Ag core nanostructures
373 which supported the overall metal-to-ligand charge transfer process. Further, it is speculated

374 that the presence of chitosan layer on the core $\text{Ag}@\text{[Ru(bpy)}_3\text{]}^{2+}$ is expected to have
375 reasonable influence on the observed redox wave.

376 Earlier studies demonstrated that the insulating nature of GO nanosheets could be
377 transformed by functionalization of metallic composites, which not only provides better
378 electrical conductivity but also creates a 3D hierarchical environment with a large surface
379 area for rapid electron transfer. For instance, conductive polyaniline interconnected Fe_2O_3 -
380 rGO composites exhibited a surface-confined redox transition at the electrode interface.⁸
381 Enhanced redox waves generated from Ag-doped with organometallic or conductive polymer
382 composite electrodes have been reported previously in the literature.^{44,45} Likewise, herein the
383 hybrid combination of chemically interacted core $\text{Ag}@\text{[Ru(bpy)}_3\text{]}^{2+}$ and shell chitosan
384 enabled a significant redox reaction at the GO interface. Such hybrid nanoplatform is useful
385 for constructing advanced biosensor platforms.

386 In order to evaluate the constancy of the redox potentials and increasing peak currents
387 in respect to the scan rate, CVs for HNPs-GO electrodes were recorded at different scan rates
388 from 50 to 120 mV/s (Fig. 9B). The enhancement of the anodic (A_1 and A_2) and cathodic (C_1)
389 peak currents are in relation to the scan rate (Fig. 9C). The correlation coefficients for the
390 anodic peaks were 0.9987 (I_pA_1), 0.9952 (I_pA_2) and the cathodic peak was 0.9964 (I_pC_1),
391 indicating that it is a surface-confined process. This interesting redox behavior, which has
392 emerged from the current investigation on CVs of HNPs-GO is valuable and provides the
393 possibility of exploring their bio-affinity toward novel molecules, through a label-free, direct
394 electrochemical detection strategy.

395 **Immunosensing property of HNPs-GO/anti-*Lm* electrodes**

396 Using the chronoamperometric technique the electrochemical immunosensing
397 property of the HNPs-GO/anti-*Lm* electrodes has been carried out as a function of *Lm*
398 concentration. To show the detection ability, the desired concentrations of *Lm* cells were

399 diluted in PBS and in milk samples respectively for individual amperometric measurements.
400 In general, antibodies immobilized onto the electrode surface forms a layer that would
401 interfere the inherent electron transfer process at the electrode interface.⁴⁶ Studies have been
402 reported that the membranes of biological cells show a resistance of 10^2 – $10^5 \Omega \text{ cm}^2$.^{46,47}
403 Adhesion of bacterial cell membranes to the electrode interface, through a specific antibody-
404 antigen complex, could create a further barrier for the electrochemical process. The
405 magnitude of decrease in current is related to the number of bacterial cells captured by the
406 immobilized antibodies. Herein, the change of amperometric current is obtained by the
407 experimental group, with Lm and control group, without Lm ($\Delta I = I_{Lm} - I_{control}$). Based on the
408 broad and amplified anodic property of HNPs-GO electrodes in PBS buffer, an optimal peak
409 potential of +0.55 V, responsible for oxidation of $[\text{Ru}(\text{bpy})_3]^{2+} \rightarrow [\text{Ru}(\text{bpy})_3]^{3+}$, has been
410 applied to measure the change in amperometric response. As summarized in Fig.10A, the
411 amperometric current response from the electrodes shows a linear relationship for Lm
412 concentration in the range 10^2 to 10^7 cells/mL. The regression equation for the experimental
413 data is $y = A + B(X)$, where y is sensor response in change of current ($-\Delta I$), X is Lm
414 concentrations in cells/mL, A and B are sensor constants. Linear fitting of the PBS data gave
415 $y = -5.43 \times 10^{-6} + 1.82 \times 10^{-5}(X)$ with a correlation coefficient of 0.98. Likewise a good fit
416 ($R^2=0.98$) was obtained for the milk samples: $y = 2.55 \times 10^{-5} + 5.61 \times 10^{-6}(X)$. From these
417 correlations the sensitivity of the HNPs-GO based immunosensor in PBS is calculated to be
418 $1.82 \times 10^{-5} \text{ A}/10^1 \text{ Lm cells/mL}$ and in milk is $5.61 \times 10^{-6} \text{ A}/10^1 \text{ Lm cells/mL}$. Compared to
419 PBS the sensitivity of HNPs-GO electrodes in milk samples are decreased perhaps due to the
420 milk components such as fat, cholesterol, carbohydrates and vitamins. However, the change
421 of current values and concentration of Lm in milk samples have a linear relationship in
422 coherent with PBS. Observed sensitivity trend in PBS and milk samples are in agreement
423 with earlier immunodetection studies.^{24,48} The detection limit has been determined using $3\sigma/m$

424 criteria, where σ is the standard deviation of blank and m is the slope of the calibration plot,
425 and is found to be 2 cells/mL for both PBS and milk diluted with *Lm*. Observed lowest
426 detection limit is better than the recent reports.^{22-24,48} Immunosensing characteristics of the
427 proposed HNPs-GO/anti-*Lm* electrode have been summarized in Table S1. Fig.10B shows
428 the comparative absolute amperometric histogram with different species, *Pa*. The results
429 demonstrated that, due to the presence of monoclonal anti-*Lm*, the HNPs-GO based
430 immunosensor is specific for *Lm*, thereby interaction of other pathogen doesn't exhibit cross-
431 reaction on the sensor surface, resulting in uninterrupted electrochemical process. The
432 detection approach proposed herein has been shown to be fast in response time, <1 min. This
433 enhanced immunosensing property could be ascribed to the synergistic material's
434 composition exist on the electrode interface.

435 **4 Conclusion**

436 The fabrication of GO nanosheets chemically decorated with HNPs composed of
437 metal-dye complex ($\text{Ag}@\text{[Ru}(\text{bpy})_3\text{]}^{2+}$) core and biopolymer (chitosan) shell was
438 demonstrated through a scalable wet-chemical approach. Functionalized HNPs-GO facilitates
439 a profound improvement in the electron-transfer process at the electrode interface compared
440 to pristine Au, HNPs and GO. HNPs-GO based electrochemical immunosensor show a linear
441 correlation between change of current and various concentrations of *L. monocytogenes* in
442 PBS and milk samples. The monoclonal anti-*Lm* modified HNPs-GO electrode showed the
443 lowest detection limit of 2 cells/mL. Selectivity study with *P. aeruginosa* demonstrated that
444 the proposed immunosensor exhibit no cross-reactivity. The HNPs-GO immunosensing
445 system demonstrated advantages for its ease in fabrication and operation, sensitivity,
446 selectivity and rapid analysis toward the model food pathogen, *L. monocytogenes*. Additional
447 studies are warranted to understand the impedance change caused by interaction of antibody-

448 bacteria complex on HNPs-GO electrode and its related bacterial number that will further
449 optimize the sensitivity of the system.

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454 **Appendix A. Supplementary data**

455 Supplementary data associated with this article can be found, in the online version, at
456

457 **References**

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- 537

Figure Captions

538

539 **Fig. 1** Illustration of the stages in the fabrication of HNP (sequential wet-chemical synthesis
540 of AgNP stabilized with 3-MPA, $[\text{Ru}(\text{bpy})_3]^{2+}$ dye complex coating on AgNP and chitosan
541 modification on $\text{Ag}@\text{[Ru}(\text{bpy})_3]^{2+}$ *via in situ* reduction, electrostatic and coordination
542 reaction, respectively), and covalent functionalization on GO nanosheet.

543

544 **Fig. 2** UV-vis absorbance spectra of aqueous dispersion of (A) AgNPs and AgNPs stabilized
545 with 3-MPA, (B) $[\text{Ru}(\text{bpy})_3]^{2+}$ and $\text{Ag}@\text{[Ru}(\text{bpy})_3]^{2+}$, (C) HNPs, GO and HNPs-GO.

546

547 **Fig. 3** PL spectra of aqueous GO and HNPs-GO nanostructures ($\lambda_{\text{ex}} = 325$ nm) measured at
548 room temperature.

549

550 **Fig. 4** TEM images of (A and B) HNPs, (C and D) GO and (E-G) HNPs-GO nanostructures.

551

552 **Fig. 5** FTIR spectra of (A) pristine chitosan powder, (B) HNPs, (C) GO and (D) HNPs-GO.

553

554 **Fig. 6** $^1\text{H-NMR}$ spectra of (A) HNPs, (B) GO and (C) HNPs-GO samples in DMSO-d_6
555 solvent.

556

557 **Fig. 7** Raman spectra of GO and HNPs-GO.

558

559 **Fig. 8** Schematic illustration of an integrated three-electrode system and modification with
560 GO and HNPs-GO nanosheets.

561

562 **Fig. 9** (A) CVs of bare Au-PCB, pristine GO and HNPs-GO modified electrodes. (B) CVs of
563 HNPs-GO electrode at different scan rates (50-120 mV/s) in 10 mM PBS (pH 7.4) and (C)
564 the corresponding plots of anodic (A_1 and A_2) and cathodic (C_1) peak currents against the
565 square root of scan rates.

566

567 **Fig. 10** (A) Calibration curve fit illustrating the sensor response for various concentrations of
568 *Lm* cells in PBS and milk. Sensor response expressed in change of current obtained by the
569 experimental group, with *Lm* and control group, without *Lm* ($\Delta I = I_{Lm} - I_{control}$). (B) Absolute
570 amperometric current response of HNPs-GO/anti-*Lm* electrode, without and with bacterial
571 cells (*Lm* and *Pa*) in PBS. Measurements were performed at an applied potential of +0.55 V.
572 Each data point represents the average of three independent measurements at different
573 electrodes, and error bars denote the standard deviation of the mean.

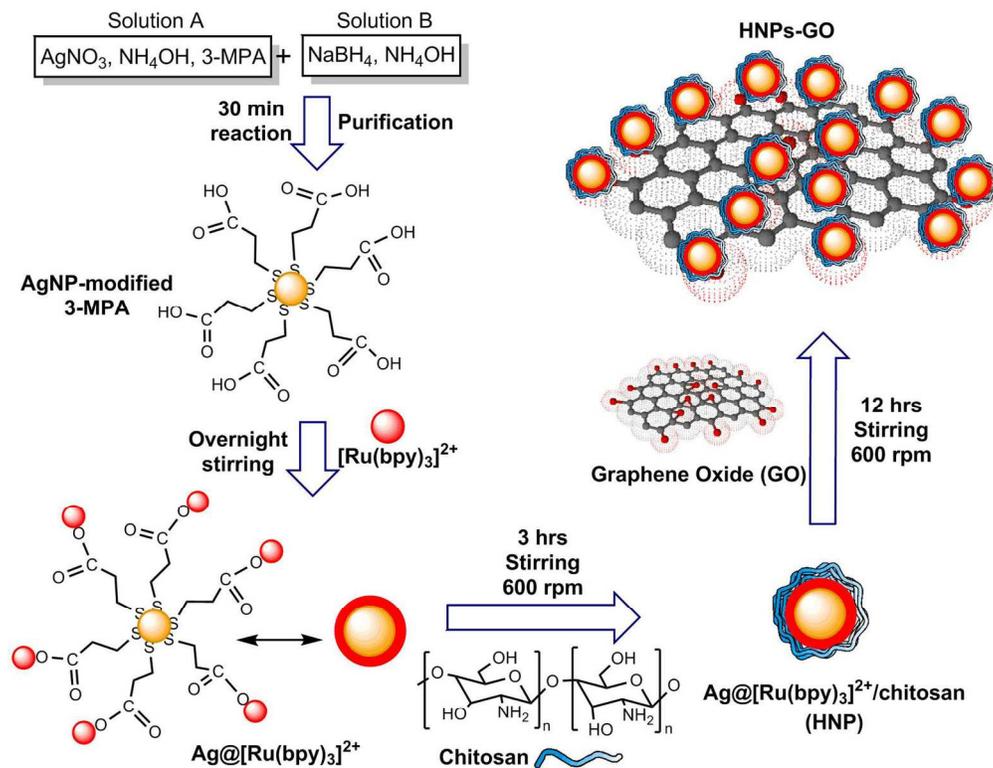
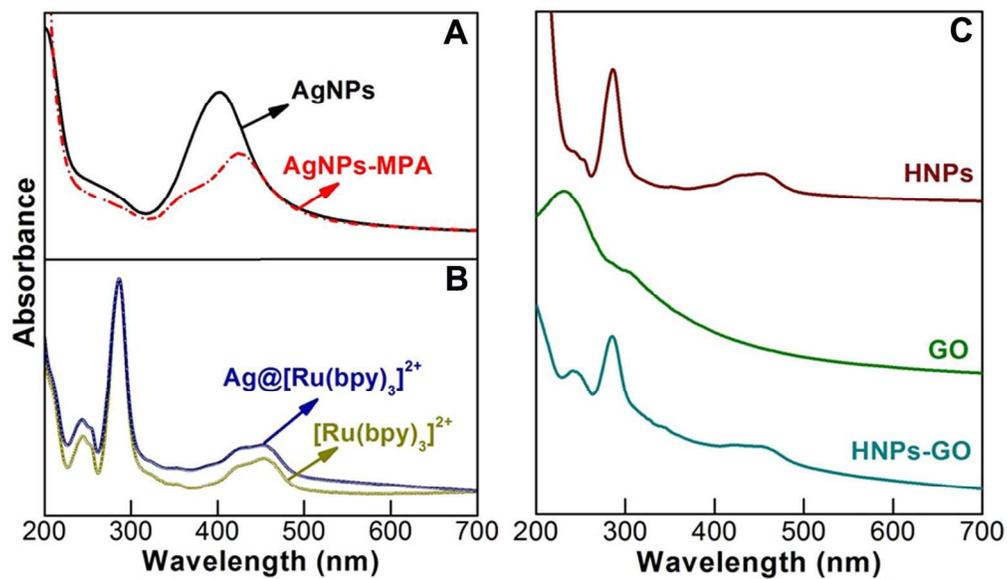
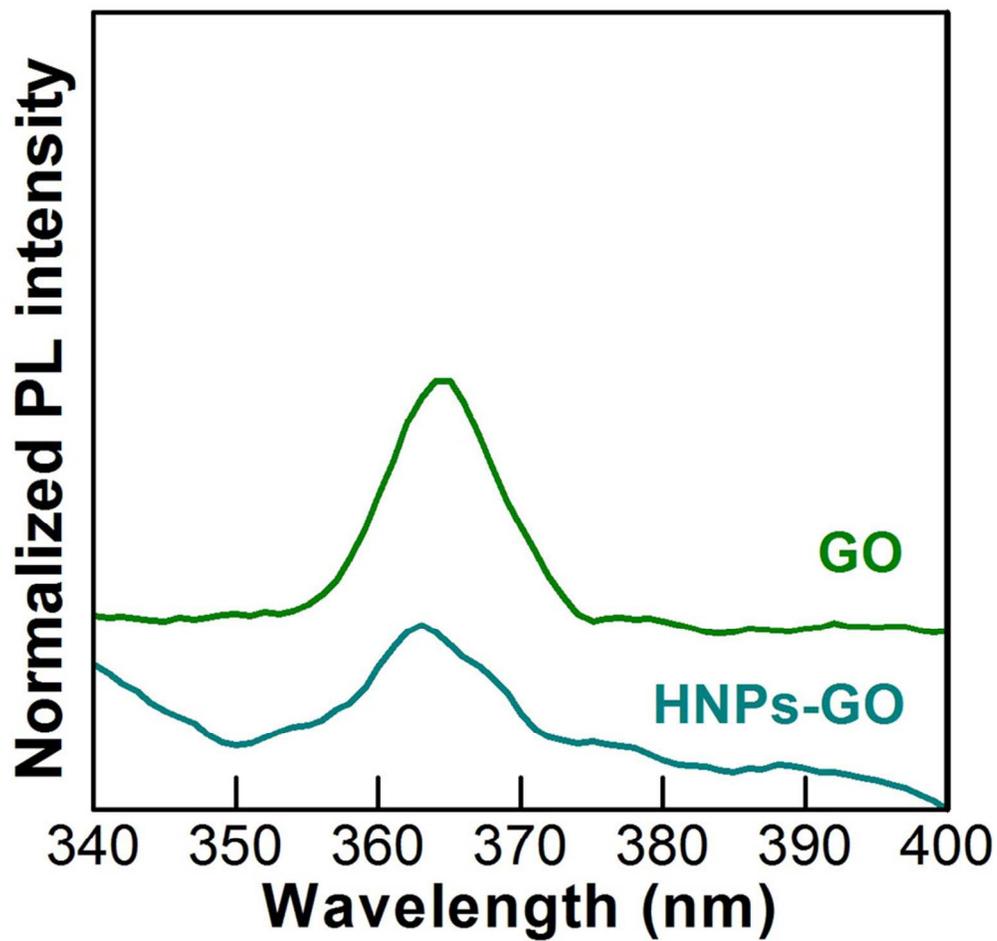


Illustration of the stages in the fabrication of HNP (sequential wet-chemical synthesis of AgNP stabilized with 3-MPA, $[\text{Ru}(\text{bpy})_3]^{2+}$ dye complex coating on AgNP and chitosan modification on $\text{Ag}@[\text{Ru}(\text{bpy})_3]^{2+}$ via in situ reduction, electrostatic and coordination reaction, respectively), and covalent functionalization on GO nanosheet.

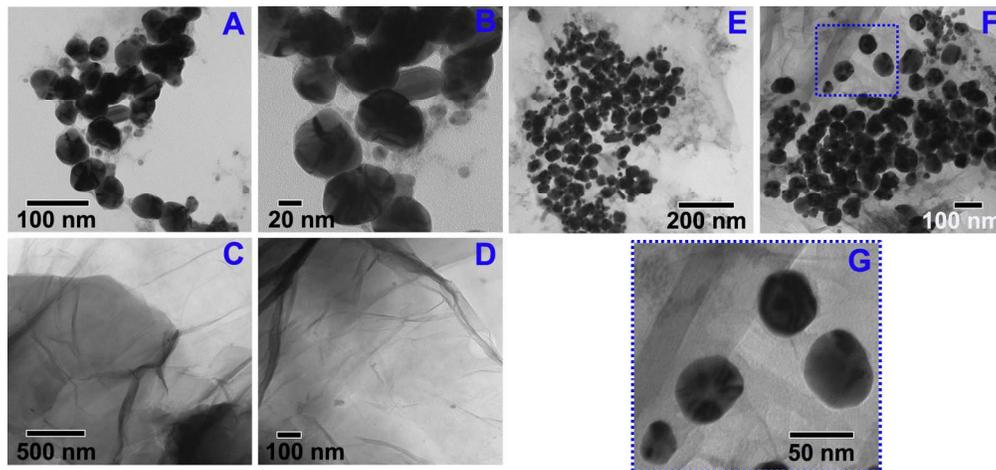
158x128mm (300 x 300 DPI)



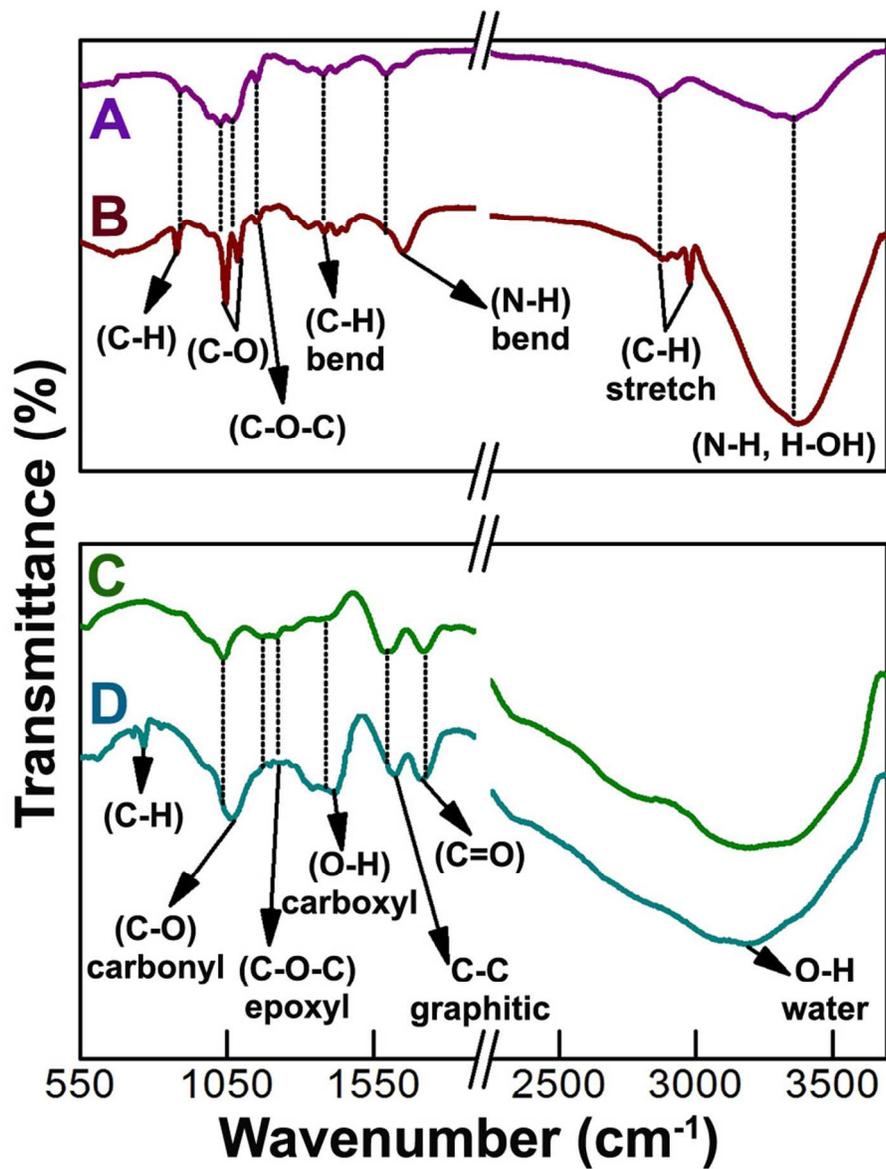
UV-vis absorbance spectra of aqueous dispersion of (A) AgNPs and AgNPs stabilized with 3-MPA, (B) $[\text{Ru}(\text{bpy})_3]^{2+}$ and $\text{Ag}@[\text{Ru}(\text{bpy})_3]^{2+}$, (C) HNPs, GO and HNPs-GO.
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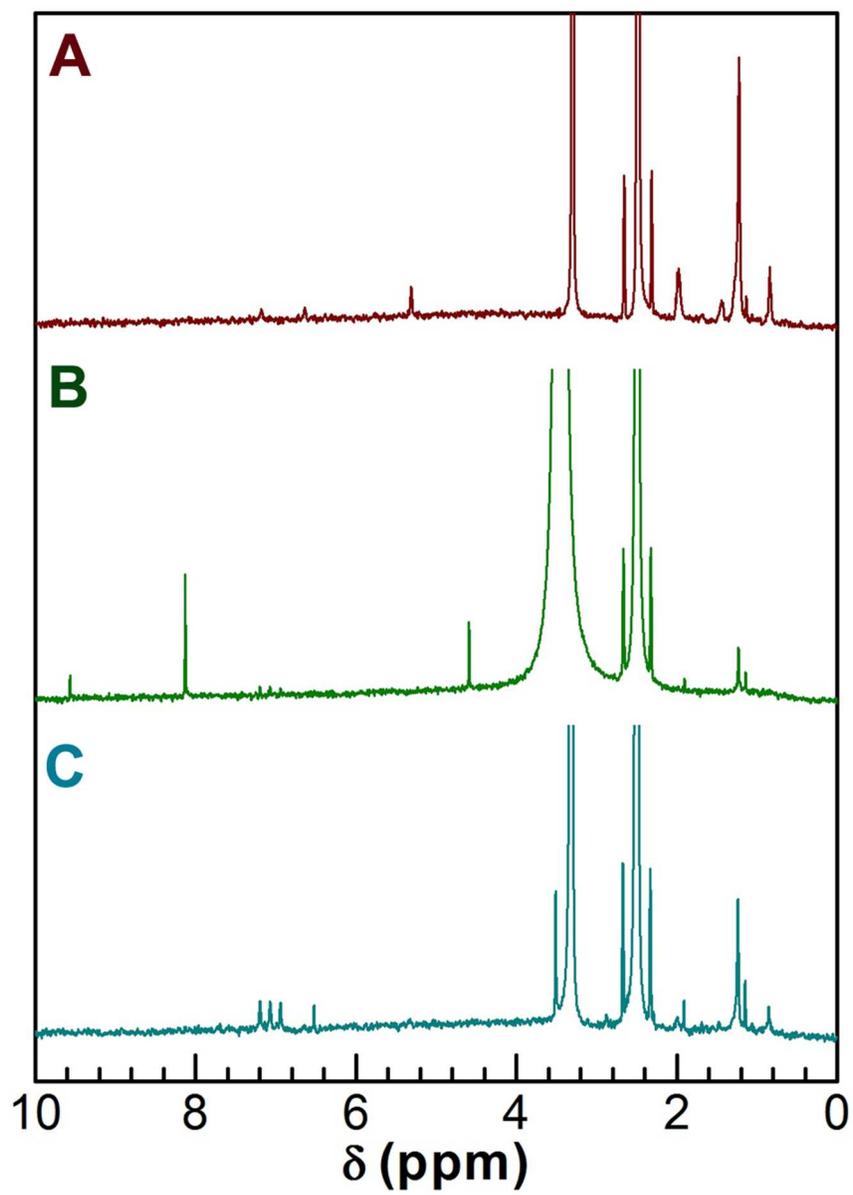
PL spectra of aqueous GO and HNPs-GO nanostructures ($\lambda_{\text{ex}} = 325$ nm) measured at room temperature.
82x78mm (300 x 300 DPI)



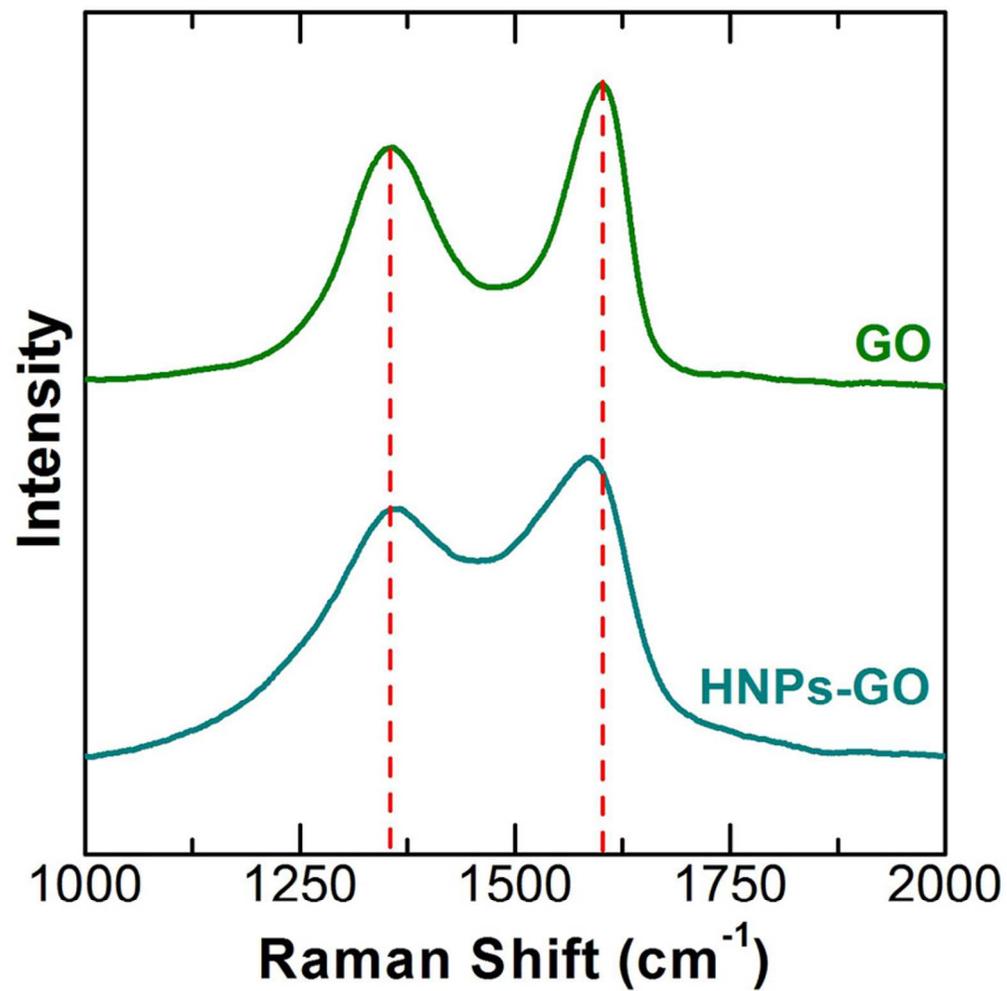
TEM images of (A and B) HNPs, (C and D) GO and (E-G) HNPs-GO nanostructures.
158x74mm (300 x 300 DPI)



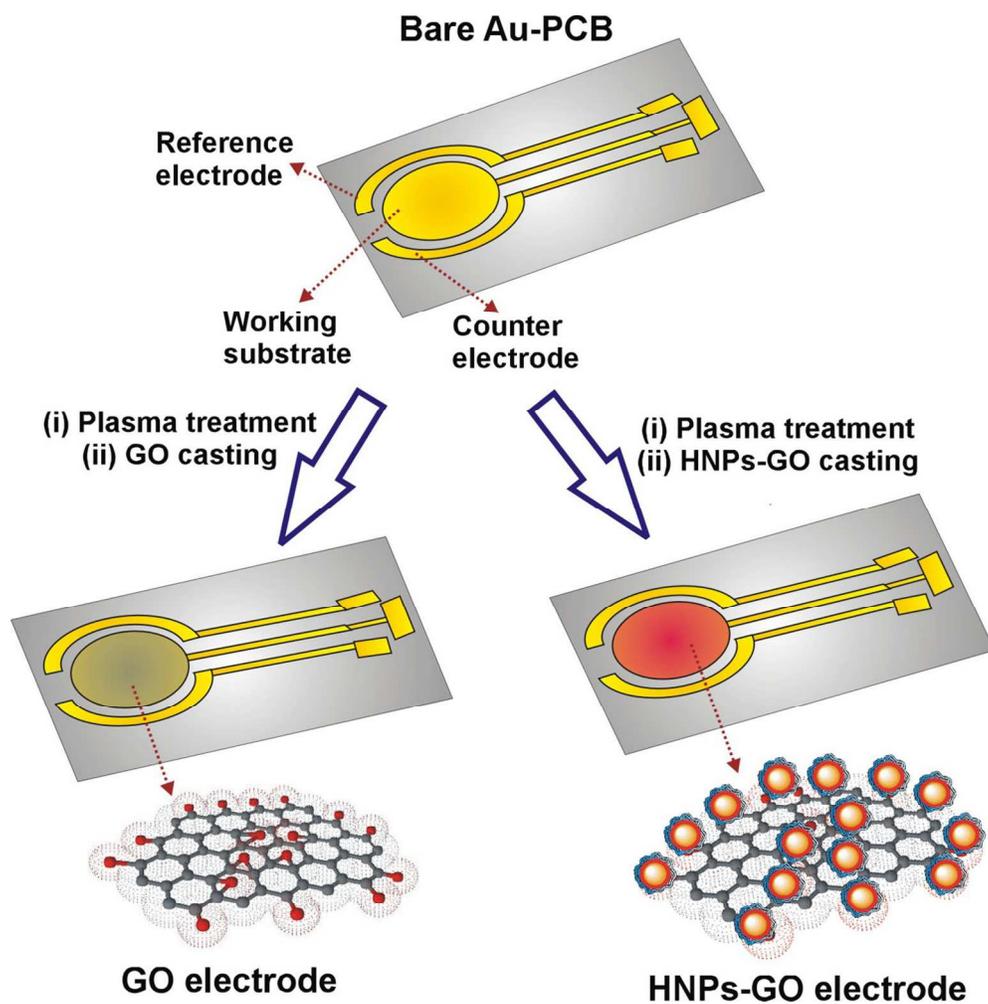
FTIR spectra of (A) pristine chitosan powder, (B) HNPs, (C) GO and (D) HNPs-GO.
82x107mm (300 x 300 DPI)



$^1\text{H-NMR}$ spectra of (A) HNPs, (B) GO and (C) HNPs-GO samples in DMSO- d_6 solvent.
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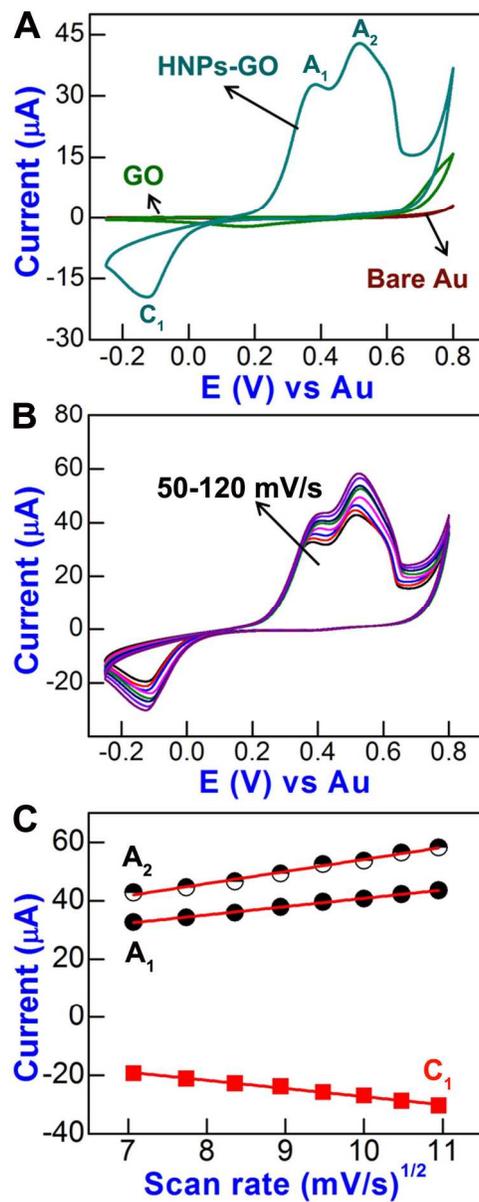


Raman spectra of GO and HNPs-GO.
76x76mm (300 x 300 DPI)

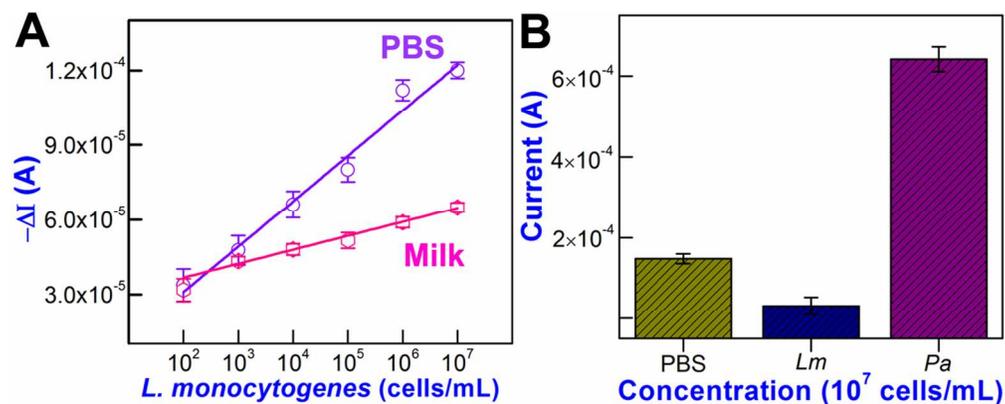


Schematic illustration of an integrated three-electrode system and modification with GO and HNPs-GO nanosheets.

123x121mm (300 x 300 DPI)

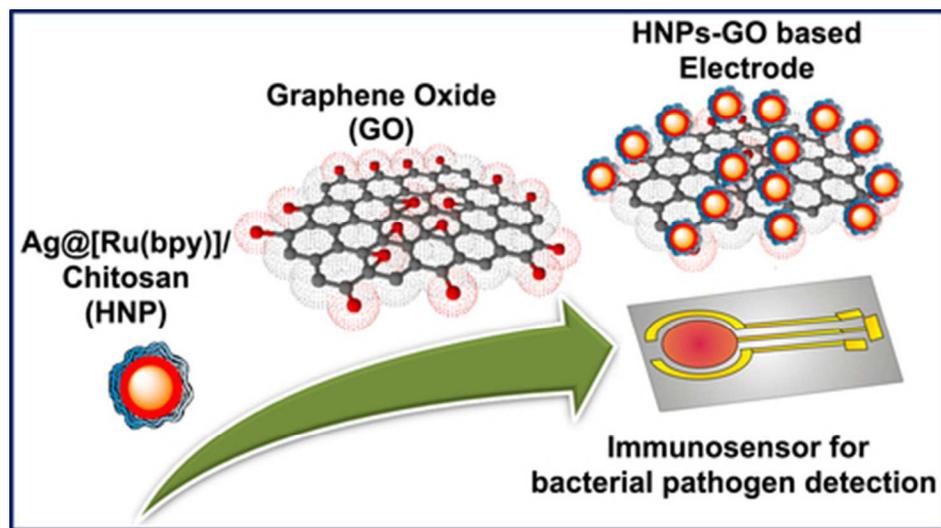


(A) CVs of bare Au-PCB, pristine GO and HNP-GO modified electrodes. (B) CVs of HNP-GO electrode at different scan rates (50-120 mV/s) in 10 mM PBS (pH 7.4) and (C) the corresponding plots of anodic (A_1 and A_2) and cathodic (C_1) peak currents against the square root of scan rates.
82x205mm (300 x 300 DPI)



(A) Calibration curve fit illustrating the sensor response for various concentrations of Lm cells in PBS and milk. Sensor response expressed in change of current obtained by the experimental group, with Lm and control group, without Lm ($\Delta I = I_{Lm} - I_{control}$). (B) Absolute amperometric current response of HNPs-GO/anti-Lm electrode, without and with bacterial cells (Lm and Pa) in PBS. Measurements were performed at an applied potential of +0.55 V. Each data point represents the average of three independent measurements at different electrodes, and error bars denote the standard deviation of the mean.

133x53mm (300 x 300 DPI)



39x22mm (300 x 300 DPI)