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### COMMUNICATION

### Solvothermal synthesis of GO/V<sub>2</sub>O<sub>5</sub> composites as cathode material for rechargeable magnesium batteries

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Herein  $GO/V_2O_5$  composites as cathode material for rechargeable magnesium batteries are presented. Synthesized by solvothermal reaction of vanadium oxytriisopropoxide (VOT) and graphene oxide (GO), the  $GO/V_2O_5$  composites exhibite greatly enhanced electrochemical performances, attained a high discharge capacity up to 178 mAh/g at rate of 0.2 C.

Rechargeable magnesium (Mg) batteries have been long considered as a future prospective alternative energy storage device <sup>1-3</sup> Mg possesses several characteristics that rank it as one of the most auspicious metal anode for high energy-density batteries. Due to its bivalency and density, its specific volumetric capacity reaches 3833 mAh /cm³, much higher than that for Li metal (2046 mAh/cm³). Meanwhile, Mg is a benign and abundant metal being in the earth's crust. Furthermore, Mg is not plagued by dendrite formation, which is a significant safety issue that has dissuaded the commercialization of rechargeable batteries utilizing a lithium metal anode.<sup>3-5</sup>

Although rechargeable Mg batteries possess attractive advantages in competition, researches of the batteries are still in infancy. The major obstacles that prevent Mg batteries from practices are (1) the kinetically sluggish diffusion of Mg in cathode materials; (2) the high polarizing ability of the divalent Mg²+ cation. Thus, the search for appropriate cathode material is intrinsically urging. Aurbach  $et\ al.$  combined Chevrel phase MgxMo3S4 cathode with 0.25M Mg(AlCl2EtBu)2/THF electrolyte in 2000, which achieved excellent performance. To date, various cathode materials such as Chevrel phase MxMo6T8 (M=metal, T=S, Se), MgxMySiO4 (M=Mn, Fe, Co, x+y=1), Hence Transition metal oxide (MnO2), MnO2 as suffur-containing material, have been successfully used as

cathode materials to improve the electrochemical properties of rechargeable Mg batteries.<sup>13</sup> Unfortunately, the actual performances of rechargeable Mg batteries achieved are not as desired due to the bad cycling.

In order to overcome the problems which are perplexed the development of rechargeable Mg batteries,  $\rm V_2O_5$  has long been recognized as a promising alternative.  $^{14\cdot17}$  Significantly results are presented for dry  $\rm V_2O_5$  gels,  $^{16\cdot17}$  but the measurement is under severe condition with aqueous electrolyte solution (e.g., 1M Mg(ClO\_4)\_2 + 1M H\_2O/acetonitrile) and lower current (C/40, i.e., duration of one cycle is about 40 h), as well as in rapid capacity fading. To improve the capacity and cycle stability for rechargeable Mg batteries,  $\rm GO/V_2O_5$  composites as an effective material were prepared by solvothermal reaction, which exhibited remarkably enhanced cycling performance as cathode material in LIBs.  $^{18\cdot21}$ 

In a typical procedure (Fig. 1), 10 mg of graphene oxides (GO) was dispersed in 35 mL of isopropanol (IPA) by ultrasonication for 2h,  $^{22}$  followed by addition of 200  $\mu L$  of vanadium

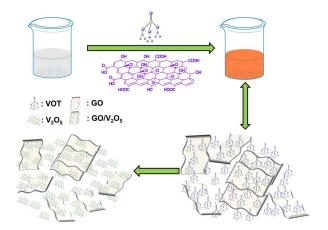


Fig. 1 Schematic illustration of the fabrication process of  $GO/V_2O_5$  composites for rechargeable Mg batteries.

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<sup>†.</sup>Electronic Supplementary Information (ESI) available: Experimental section, XRD pattern, SEM images and some more details of rechargeable Mg batteries. See DOI: 10.1039/x0xx00000x

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oxytriisopropoxide (VOT). Then, the mixture was sealed in a Teflon-lined autoclave and solvothermally treated at 200  $^{\circ}$ C for 12h. The precipitate was collected by centrifugation, washed thoroughly with isopropanol (IPA) and deionized water several times (see ESI). Ultimately,  $GO/V_2O_5$  composites were obtained from a calcinating process at 800  $^{\circ}$ C in Ar.

The electrochemical performances of the as-obtained composites are tested as cathode for rechargeable Mg battery. The electrode was prepared by pressing a powder mixture of  $GO/V_2O_5$  composites, acetylene black, and poly(vinylidene fluoride) (PVDF) in a weight ratio of 80:10:10 onto an Al foil. After drying, the foil was cut into pellets with 2.0 mg of electrode materials for every pellet. Then the pellets were dried at  $80\,^{\circ}\text{C}$  overnight under vacuum. Coin-type cells were assembled in a glove box, with magnesium foil as the counter and reference electrode, Celgard 2400 as the separator, and 0.25M Mg(AlCl<sub>2</sub>EtBu)<sub>2</sub>/THF as the electrolyte. The charge-discharge performance was tested between 1.0 and 2.8 V using the LAND CT2001A multichannel battery testing system at room temperature.

As-prepared  $GO/V_2O_5$  composites were characterized by X-ray diffraction (XRD) employing a Bruker D8 Focus X-ray Diffraction with Cu K $\alpha$  radiation. The morphology and structure were obtained on SEM (Hitachi S-4800) and TEM (Tecnai G2). X-ray photoelectron spectroscopy (XPS) was collected on Thermo ESCALAB 250 electron spectrometer.

Fig. 2 shows the powder X-ray diffraction (XRD) pattern of  $GO/V_2O_5$  composites. All the diffraction peaks can be indexed to orthorhombic  $V_2O_5$  (a=11.512 Å, b=3.564 Å, c=4.368 Å, JCPDS card no. 65-0131). To give insight into the crystalline structure evolution of  $GO/V_2O_5$  composites, XRD patterns of the precursors is exhibited in Fig. S2. Clearly, there is obvious change in the diffraction peak when  $GO/V_2O_5$  composites were heated at 800 °C.

The SEM and TEM micrographs of the as-prepared GO/V<sub>2</sub>O<sub>5</sub>

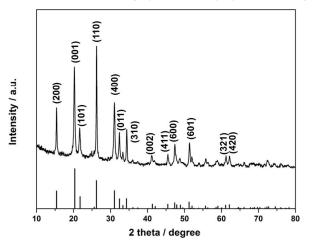
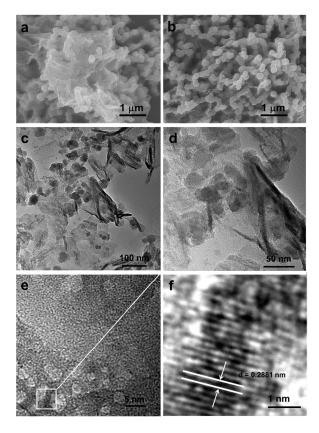


Fig. 2 The X-ray diffraction (XRD) pattern of as-synthesized  $GO/V_2O_5$  composites.



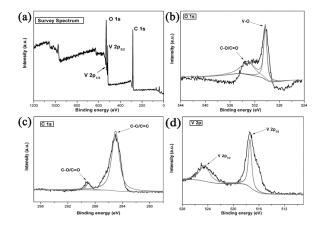
**Fig. 3** The SEM and TEM micrographs of as-synthesized  $GO/V_2O_5$  composites. (a)(b) SEM images. (c)(d) TEM images. (e)(f) HRTEM images.

composites are shown in Fig. 3. From SEM images in Fig. 3a and Fig. 3b, it can be seen that the GO and V<sub>2</sub>O<sub>5</sub> microparticles are well cohered. The V<sub>2</sub>O<sub>5</sub> microparticles are wrapped up around GO sheets, particularly exhibited in Fig. 3b. The TEM micrographs (Fig. 3c and Fig. 3d) clearly demonstrate the firm contact between GO and V2O5 microparticles. The SEM and TEM images disclose the *in-situ* growth relationship between GO and V<sub>2</sub>O<sub>5</sub> microparticles. In comparison with Fig. 3e, Fig. 3f and Fig. S7, the weakness of crystallinity degree of V<sub>2</sub>O<sub>5</sub> microparticles presented in Fig. 3f reveals that the existence of GO has an effect on structure of V<sub>2</sub>O<sub>5</sub>. In HRTEM image (Fig. 3f), the fringe spacing of 0.2881 nm is corresponding to the (400) plane of orthorhombic  $V_2O_5$ . The conclusion drawn from the discussion of SEM and TEM images is that the in-situ grown GO/V<sub>2</sub>O<sub>5</sub> composites could gain well-deserved electronic conductivity and obtain large reaction areas, where Mg ions would be absorbed, then exchanged and diffused.

XPS is recognized as a useful method to accurately determine the chemical compositions and electronic structures of nanomaterials. Fig. 4a shows the survey spectrum of as-obtained  $GO/V_2O_5$  composites, from which it

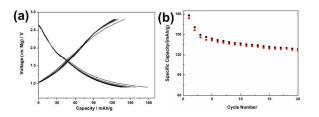
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could be obviously noted that the characteristic peaks of O, C and V exist in the  $GO/V_2O_5$  composites. The high-resolution



**Fig. 4** (a) Survey XPS spectrum of as-synthesized  $GO/V_2O_5$  composites. XPS spectrum for (b) O 1s, (C) C 1s, and (D) V 2p regions of as-synthesized  $GO/V_2O_5$  composites.

spectrum of as-obtained  $GO/V_2O_5$  composites, from which it could be obviously noted that the characteristic peaks of O, C and V exist in the GO/V<sub>2</sub>O<sub>5</sub> composites. The high-resolution XPS spectrum of the O 1s, C 1s and V 2p for the GO/V<sub>2</sub>O<sub>5</sub> composites are displayed in Fig. 4b-d. As shown in Fig. 4c, we can clearly find a peak at 284.6 eV, which is consistent with the binding energy of C 1s. One more broad peak at around 289 eV was observed, which is attributed to C-O/C=O bonding, confirming the existence of GO. The about 8 eV gap between V  $2p_{3/2}$  and V  $2p_{1/2}$  is a typical value for  $V^{5+}$  in the standard line of XPS spectrum. However, the high-resolution XPS spectra of V  $2p_{3/2}$  and V  $2p_{1/2}$  in Fig. 4d is obviously observed at 517.2 eV and 524.9 eV, respectively, which increases nearly 5 eV. According to rationale of XPS and the synthesis of GO/V2O5 composites, the increasing of binding energy would be caused by the multi-electron excitation and multiplet splitting. Valence shell electrons of vanadium element transit to the higher energy level. Relaxation effect is generated by the decline in kinetic energy of photoelectron. XPS spectrum



**Fig. 5** Electrochemical properties of as-synthesized  $GO/V_2O_5$  composites: (a)(b) Electrochemical cycling performance curves of  $GO/V_2O_5$  composites at a rate of 0.2 C.

reveals that the superior recombination of GO and  $V_2O_5$  microparticles, which is mentioned *in-situ* growth of  $V_2O_5$  microparticles on GO, transforms coordinating ability of vanadium element.

Intrigued by the structural features of GO/V<sub>2</sub>O<sub>5</sub> composites for rechargeable Mg batteries, we employed electrochemical measurements to test the behaviours of the as-obtained GO/V<sub>2</sub>O<sub>5</sub> composites. Fig. 5a and Fig. 5b show cycling performances and discharge-charge profiles of GO/V<sub>2</sub>O<sub>5</sub> composites, in which the current density was fixed at rate of 0.2 C, and the potential is in the range 1.0 to 2.8 V. The first discharge capacity of GO/V<sub>2</sub>O<sub>5</sub> composites is 178 mAh g<sup>-1</sup>. According to the equation mentioned by Gregory  $\it et~al.,~^{1,6,21}$ the GO/V<sub>2</sub>O<sub>5</sub> composites as cathode material for rechargeable Mg batteries could host about 0.6 Mg ions per formula unit (see ESI). Both discharge and recharge curves do not show clear plateaus. The reversible discharge capacity of the second and third cycles fades to 160 and 150 mAh g<sup>-1</sup> and diminishes tardily for the subsequent cycles, still remains 140 mAh g<sup>-1</sup> at  $20^{th}$  cycle, and the SEM micrographs of  $GO/V_2O_5$  composites after cycling can be seen in Fig. S11b. The significant capacity drop at the beginning cycles can be ascribed to the formation of MgO passivated film.

On the basis of the above results, the enhanced cycle life of the  $GO/V_2O_5$  composites cathode can be attributed to the architecture features of the *in-situ* grown composites structure as well as the synergistic effects of GO and  $V_2O_5$ : (1) The disordered structure and superior recombination can improve the electronic conductivity, shorten the  $Mg^{2+}$  and electrolyte molecules' transportation length and increase the electrolyte/active material contact area as well as facilitate the penetration ability of the electrolyte molecules. (2) The disordered structure and superior recombination can also retard the volume change during repeated charge–discharge cycles.

In summary, a new kind of  $GO/V_2O_5$  composites has been prepared as a cathode material for rechargeable magnesium batteries. The integration of GO and  $V_2O_5$  microparticles results in significantly enhanced performances. As-prepared  $GO/V_2O_5$  composites can deliver a high capacity of 178 mAh g<sup>-1</sup> and maintain 140 mAh g<sup>-1</sup> even after 20 cycles, which provide a new direction to explore cathode materials for rechargeable Mg batteries. This described synthetic strategy could motivate further exploration for next-generation, high-performance rechargeable Mg batteries or other fields of energy storage and conversion.

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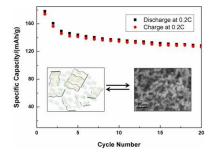
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### **Graphic abstract and table of contents**

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GO/V<sub>2</sub>O<sub>5</sub> composites prepared from solvothermal reaction exhibited greatly enhanced electrochemical performances as cathode material for rechargeable magnesium batteries.