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ARTICLE

Hierarchical LiMnPO₄ assembled from nanosheets via solvothermal method as high performance cathode material

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A series of LiMnPO₄ nanoparticles with different morphology have been successfully synthesized via solvothermal method. The samples have been characterized by X-ray diffraction (XRD), Scanning electron microscope (SEM), Transmission electron microscopy (TEM). The results show that morphology, particle size and crystal orientation are controllable synthesized by various precursor composite tailoring with various Li:Mn:P molar ratio. At 3:1:1, Li⁺-containing precursor Li₃PO₄ is obtained while at 2:1:1, only Mn²⁺-containing precursor involving Mn₃(PO₄)₂[(PO₃)OH]₂·4H₂O and MnHPO₄·2.25H₂O is detected. Specially, at 2.5:1:1, the precursor consists predominantly of Mn²⁺-containing precursor with minor amount of Li₃PO₄. From 2:1:1 to 3:1:1, the particle morphology evolves from sheet to spherical texture accompanied with the particle size reduced. In the presence of urea, highly uniform LiMnPO₄ with hierarchical micro-nanostructure is obtained, which is composed of nanosheets with the thickness of several tens of nanometers. Thus, those unique hierarchical nanoparticles with open porous structure play an important role in LiMnPO₄ cathode material. At the concentration of 0.16 mol L⁻¹ for urea, the hierarchical LiMnPO₄/C sample assembled from nanosheets with (010) facet exposed shows the best electrochemical performance, delivering higher reversible capacity of 150.4, 142.1, 138.5, 125.5, 118.6 mAh g⁻¹ at 0.1, 0.2, 0.5, 1.0, 2.0C, respectively. Moreover, the composites show long cycle stability at high rate, displaying a capacity retention up to 92.4% with no apparent voltage fading after 600 cycles at 2.0 C.

1. Introduction

LiMnPO₄ is one of the most promising cathode material for lithium ion-batteries, due to its high specific capacity, low toxicity, low cost, and excellent thermal stability, and has been expected to be applied in a wide-scale use in plug-in hybrid electric vehicles (PHEVs), electric vehicles (EVs) and smart grid storage that fast electronic transfer and superior ionic migration are demanded.^{1, 2} However, the bulk LiMnPO₄ suffers from poor electronic conductivity and slow Li⁺ diffusion, leading to low specific capacity, poor rate capacity, and inferior cycle stability.³⁻⁵ Up to now, the electrochemical performance of LiMnPO₄ cathode material has been improved by controlling the particle size and morphology. It has been demonstrated that the particles with dimensionally modulated nanostructure such as nanorod, nanosheets and nanosphere can enhance the electrochemical performance due to a shortening of both the electron and lithium ion diffusion path lengths within the particles.⁶⁻⁸ The cluster-like LiMnPO₄ composed of nanoplates with a thickness of ca. 35 nm and a width of ca. 400 nm via a solvothermal method exhibited a discharge capacity of 147 mAh g⁻¹ at 0.05 C.⁹ Xu et.al. reported the rod-like

LiMnPO₄ via a solvothermal method, showing a reversible capacity of 153.4 mAh g⁻¹ at 0.1 C, which is ascribed to the nanostructure with the average length and diameter of approximately 100 and 60 nm, respectively.¹⁰ In this sense, the nanostructure is key to synthesizing LiMnPO₄ cathode material with excellent electrochemical performance. Since the solvothermal process is a facile approach to prepare nanostructure and high crystalline materials with the notable virtues such as simple and straight ward operations, cost-effectiveness, and scalable production, it has been widely used to prepare the LiMnPO₄ cathode material. So far, nanostructured LiMnPO₄ cathode material has been successfully prepared via the solvothermal process, in which the expensive chemicals was used.^{11, 12} What's more, in a traditional hydrothermal/solvothermal method, to prepare the nanostructured LiMnPO₄, excess Li is used with the Li:Mn:P molar ratio of 3:1:1 or even more.^{12, 13} Up to now, EG based solvent has been successfully used to prepare micro- and nano-structure LiMPO₄ (M=Fe, Mn, Ni) for lithium ion-batteries, and the morphology could be easily tuned by altering the synthesis condition or selecting different raw materials, such as the concentration, solvent, temperature and time, pH value, feeding sequence and the surfactant.^{6, 8, 14-18} However, It is rarely reported the effect of the phase composition in the precursor solution or Li:Mn:P molar ratio on the morphology and structure of LiMnPO₄. Thus, it deserves further research to reveal the role that phase composition or Li:Mn:P molar ratio play in the preparation of LiMnPO₄. With this perspective, we

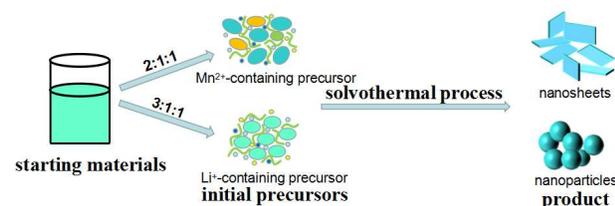
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report the morphology controllable synthesis of LiMnPO_4 tailoring with the molar ratio of Li:Mn:P by solvothermal method, using ethanol as green solvent, LiH_2PO_4 , $\text{LiOH} \cdot \text{H}_2\text{O}$ and $\text{MnSO}_4 \cdot \text{H}_2\text{O}$ as the starting material, as illustrated in Scheme 1.

On the other hand, it is hard to control the morphology, especially for highly uniform particles with nanostructure. Generally, the nanoparticles prefer to agglomerate together, which in turn constantly limits the demonstration of its intrinsic performance. In this regard, LiMnPO_4 nanoparticles with a novel hierarchical micro-nanostructure are expected as a new prototype to minimize the self-agglomeration and re-stacking and also facilitate physical properties at macroscale.

Herein, a urea assisted solvothermal method was proposed for the growth of hierarchical micro-nanostructured LiMnPO_4 by controlling the molar ratio of Li:Mn:P at 2:1:1. To fully demonstrate the effect of urea, the concentration of urea was optimized. With the morphology and composition advantages, the as-prepared LiMnPO_4/C is expected to manifest superior electrochemical performance, especially, the high rate capacity and long cycle stability.



Scheme. 1 the morphology controllable solvothermal process by tailoring with the Li:Mn:P molar ratio

2. Experimental

2.1 preparation

The LiMnPO_4 samples were prepared by solvothermal approach in ethylene glycol (EG) and de-ionized water mixing solvent (volume ratio of 9:1) as follows: various amount of LiOH aqueous solution was added drop-wise into the LiH_2PO_4 EG- H_2O solution under vigorous stirring, and then CTAB (hexadecyl trimethyl ammonium bromide, 0.015 mol L^{-1}) was dissolved in the above white suspension followed by MnSO_4 aqueous solution introduced into the mixture. The final mixture was transferred to a 300 mL stainless steel autoclave. The concentration of Mn^{2+} in the precursor solution was controlled to be 0.1 mol L^{-1} . The solvothermal reaction was carried out at $220 \text{ }^\circ\text{C}$ for 4 h to form a white precipitation. After that, white particles were collected via centrifugation, washed with distilled water three times, absolute ethanol twice. After dried in a vacuum oven at $60 \text{ }^\circ\text{C}$ for 12 h, the as-prepared LiMnPO_4 was mixed and ground with PVA (20 wt%), and then annealed at $670 \text{ }^\circ\text{C}$ for 4 h ($3 \text{ }^\circ\text{C}/\text{min}$) under an Ar atmosphere to carbonize glucose and increase the crystallinity of the LiMnPO_4 material. The mass percentage of carbon was determined by carbon sulfur analyzer to be about 3.4 wt%. The molar ratio of

Li:Mn:P was controlled at 2:1:1, 2.5:1:1 and 3:1:1 by adjusting the added amount of LiOH , and the product was named s-1, s-2 and s-3. During the series experiments in the presence of urea, the molar ratio of Li:Mn:P was 2:1:1, while the concentration of urea was controlled at 0.08, 0.12, 0.16, 0.2 mol L^{-1} , namely s-4, s-5, s-6, and s-7, respectively.

2.2 Characterization

The crystal structures of the synthesized powders were examined by X-ray diffraction (XRD) (Haoyuan DX-2500, Dandong Instrument Co., Ltd., China) analysis with nickel-filtered $\text{Cu K}\alpha$ radiation ($\lambda = 1.5418 \text{ \AA}$) over the 2θ range from 10° to 90° . The morphology was investigated with a field emission scan electron microscope (SEM, JEOL JSM-6360LV, and Japan) and a transmission electron microscope (TEM, Tecnai G2 F20).

2.3 Electrochemical measurement

The electrochemical performance of the samples as cathode material was evaluated using CR2025 coin cells assembled in an argon-filled glove box. For the cathode preparation, a mixture of LiMnPO_4/C , Super P carbon, and poly(vinyl difluoride) (PVDF) with a weight ratio of 75:15:10 was pasted on an Al foil, which works as a current collector. A Celgard 2340 microporous membrane was used as the separator and lithium foil was used as the counter electrode. The non-aqueous electrolyte was 1 M LiPF_6 solution in ethylene carbonate, dimethyl carbonate, and methyl-ethyl carbonate with a volume ratio of 1:1:1 (LIB 315, Guotai Huarong Chem. Co., Ltd., Zhangjiagang, China). The electrochemical tests were carried out at room temperature on a Land Test System (CT2001A, Wuhan Jinnuo Electronic Co., Ltd., Wuhan China) setting the cut off voltages of 2.50–4.50 V versus Li^+/Li ($1 \text{ C} = 160 \text{ mA g}^{-1}$).

3. Results and discussion

3.1 effect of Li:Mn:P molar ratio

The structure and morphology of the product are highly dependent on the Li:Mn:P molar ratio. The XRD pattern of the product obtained from various molar ratio are presented in Fig. 1. It can be clearly seen that all samples are single phase which can be indexed to the orthorhombic olivine structure with a $Pmnb$ space group (JCPDS card no.74-0375). Upon closer observation, the relative peak intensities change with the molar ratio variation of Li:Mn:P. The diffraction intensity from the (020) facet is strongest at 2:1:1, while the highest intensity at 3:1:1 comes from the (311) facet. This may indicate that the as-obtained sample at 2:1:1 has a preferential crystal orientation along the [010] direction, which is favorable to the Li^+ diffusion due to the fact that LiMnPO_4 has a one-dimension migration path along the b axis.^{9,19} Fig. 2 shows the SEM images of the samples tailoring with different Li:Mn:P molar ratio. At 2:1:1, a novel morphology was obtained. As shown in Fig. 2a, the sample consists of sheet-like crystals with the thickness of several tens of nanometers, a mean length of $\sim 450 \text{ nm}$, an average width of $\sim 300 \text{ nm}$, in which the small crystals attach together irregularly to assemble into hierarchical structure with open porousness. But

unfortunately, the hierarchical structure is not open enough, thus we carried out the urea assisted solvothermal method to effectively take advantage of those novel morphology which will be discussed later in detail. When the molar ratio is prolonged to 2.5:1:1, LiMnPO_4 composed of sheet-like crystals similar to 2:1:1 is obtained. It is clearly seen that the obtained sample was randomly dispersed with well-defined shape, in which the sheet-like crystals have a uniform particle size distribution. For instance, the length is varied from 250 nm to 500 nm, while the width from 150 nm to 450 nm. When the molar ratio increased to 3:1:1, the particles are near-spherical with an average diameter of several tens of nanometers. Obviously, the particles are seriously agglomerated, which will bring negative effects on the electrochemical performance. It is worth noting that, when increasing the molar ratio of Li:Mn:P, a smaller particle size is achieved, indicating that excess Li was favorable for attaining small particle size for LiMnPO_4 , in well agreement with the previous study.^{20,21}

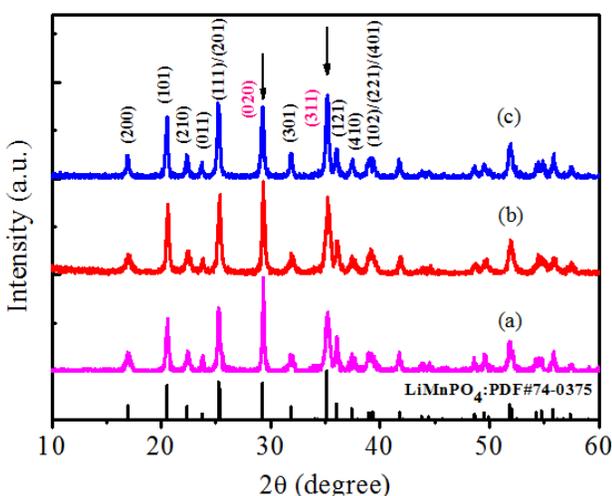


Fig. 1 the XRD pattern of the sample tailoring with different Li:Mn:P molar ratio: (a) 2:1:1; (b) 2.5:1:1; (c) 3:1:1.

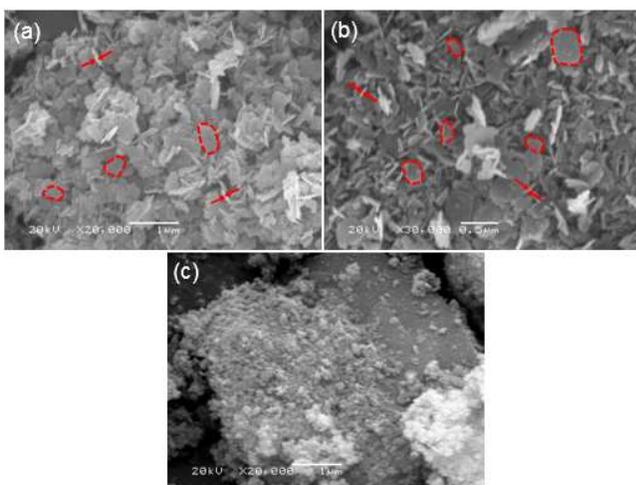


Fig. 2 the SEM images of samples tailoring with different Li:Mn:P molar ratio: (a) 2:1:1; (b) 2.5:1:1; (c) 3:1:1.

To further understand the evolution of morphology and the diversity of particle size. The structure and morphology of the precursors from the mother solution before solvothermal treatment were characterized. The XRD patterns of the dried precursors synthesized at room temperature are presented in Fig.3. The phase composition of the precursor is strongly affected by the molar ratio of Li:Mn:P. From the XRD patterns, obviously, Mn^{2+} -containing precursor involving $\text{Mn}_3(\text{PO}_4)_2[(\text{PO}_3)\text{OH}]_2 \cdot 4\text{H}_2\text{O}$ (JCPDS card no.34-0146) and $\text{MnHPO}_4 \cdot 2.25\text{H}_2\text{O}$ (JCPDS card no.47-0199) is stable at the off-stoichiometry, leaving Li^+ freely existing in the mother solution, as the molar ratio of Li:Mn:P increased, Li_3PO_4 phase (JCPDS card no.25-1030) is detected. When increased to 3:1:1, no other phase but Li^+ -containing precursor Li_3PO_4 is identified, while the manganese salt phases cannot be detected. Fig. S1 shows the SEM images of the dried precursors, clearly, at 3:1:1, the particles exhibit spherical texture similar to that of the resultant product LiMnPO_4 , indicating that LiMnPO_4 is most likely formed by an in situ evolution from the precursor Li_3PO_4 in the solvothermal method, in agreement with previous research.²² In the EG based solvothermal condition in which water is almost highly deficient is not likely able to dissolve the starting material Li_3PO_4 to form the intermediate phases such as $\text{Mn}_3(\text{PO}_4)_2$, MnHPO_4 , which are known as the dissolution-reprecipitation process in the hydrothermal method similar to LiFePO_4 .^{23,24} Thus, the Li_3PO_4 may play as a hard template to form LiMnPO_4 directly. At 2:1:1, the precursor also presents the spherical texture different with the subsequent product, which may be due to the multi-phases in the precursor solution, in which case the crystal units may contain more than one nucleus with different orientation and growth rates. In addition, such remarkable divergence in precursor solution results in different released ions from the precursor with various types or priority orders, leading to the different nucleation or growth rates. In this regard, the distinction in morphology is caused by the different solution composition generated by adjusting the molar ratio of Li:Mn:P.

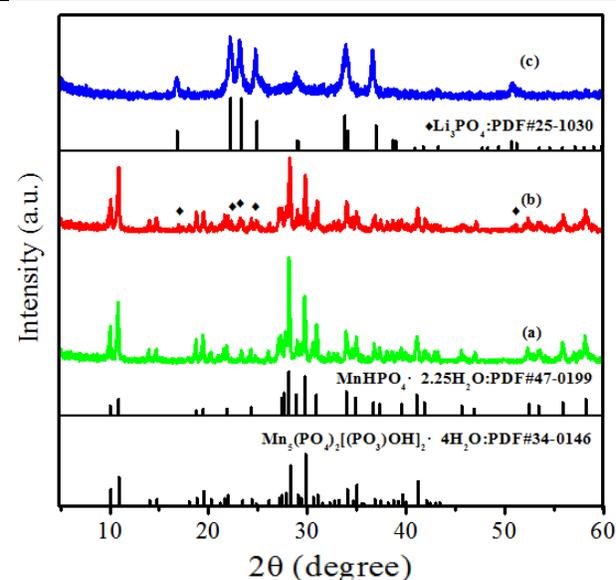


Fig. 3 the XRD pattern of the precursors tailoring with different Li:Mn:P molar ratio: (a) 2:1:1; (b) 2.5:1:1; (c) 3:1:1.

Based on the above results, it can be reasonably understood that the molar ratio of Li:Mn:P has a great effect on the phase composition of the precursor, which will control the morphology and particle size of the LiMnPO_4 crystals in the solvothermal process. To further confirm the roles of molar ratio of Li:Mn:P, also, taking advantage of the sheet-like crystals with preferred orientation of [010] direction, a set of independent experiments were carried out where urea was mainly used as the alkaline reagent.

3.2 effect of urea addition

Crystal structure is significantly important for cathode materials, including the phase composition, crystallinity, and crystal orientation. To characterize the crystal structure of LiMnPO_4 , the series samples were tested by XRD after thermal treatment. Fig. 4 shows the XRD pattern of the series samples. Among the samples after thermal treatment, the diffraction peaks can be well indexed to the orthorhombic olivine type LiMnPO_4 (JCPDS card no.74-0375), and no other impurity phase is found except that of the sample s-7, in which the impurity phase Mn_3O_4 (JCPDS card no.24-0734) is observed. In addition, the sample s-7 before thermal treatment, the impurity phase MnO_2 (JCPDS card no.12-0141) is detected. Thus, the formation of Mn(IV) impurity may be caused by the oxidation of Mn^{2+} with oxygen from the solution and air in the dissolving and feeding processes, because Mn^{2+} is strongly sensitive to the oxygen in a strong alkaline environment due to the increasing amount of urea. Also, the similar results are found in LiFePO_4 under the hydrothermal/solvothermal condition.²⁵⁻²⁷ Under the thermal treatment, the MnO_2 was further reduced by the decomposed carbon to form Mn_3O_4 . Therefore, excess urea would cause the side reaction, leading to impurity phase. It is worth noting that all the samples are well crystallized with the strongest peak (020), indicating an enlarged (010) facet exposed is expected.

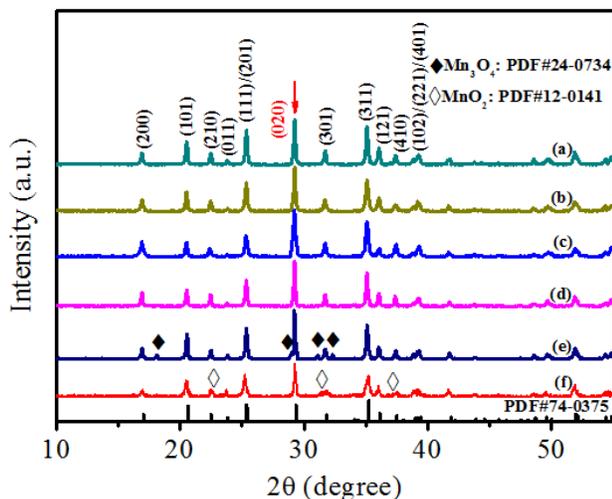


Fig. 4 the XRD pattern of the samples with various concentration of urea under thermal treatment: (a) 0 mol L^{-1} ; (b) 0.08 mol L^{-1} ; (c) 0.12 mol L^{-1} ; (d) 0.16 mol L^{-1} ; (e) 0.2 mol L^{-1} ; (f) 0.2 mol L^{-1} without thermal treatment

According to the SEM images in Fig. 5, the morphologies are not obviously changed by varying the concentration of urea, and all the samples are composed of sheet-like crystals, which indicates that urea can act as an alkaline reagent for the formation of LiMnPO_4 phase and has no obvious role in changing the shape of the LiMnPO_4 crystals. Therefore, it is believed that the phase composition of the precursor has a great effect on the morphology of LiMnPO_4 instead of the pH when tailoring with the molar ratio of Li:Mn:P. In addition, the LiMnPO_4 crystals assemble regularly to form the flower-like structure with more open porousness gradually, as increasing the concentration of urea. For the closed observation in Fig. 5, we can see that the particle size become smaller with the increasing amount of urea, as well as less agglomeration, indicating that urea is useful to reduce agglomeration and inhibit particles from growing. Importantly, when the concentration of urea is increased to 0.16 mol L^{-1} , the sample consists of nanosheets with flower-like texture. As shown in Fig. 6a-c, the flower-like sample is uniformly dispersed with well-defined micro-nanostructure. Obviously, the nanosheets crosslink together forming an open porous structure with an average thickness of several tens of nanometers, which further demonstrates their loose-packed flowery architecture. From the Fig. 6d, it is clear that the hierarchical structure can be maintained after the thermal treatment, which may be due to the fact that the relative low calcination temperature and short calcination time was employed during the thermal treatment, also the decomposed carbon can hinder agglomeration. Thus, those highly hierarchical structure constructed from the attached sheet-like crystals is beneficial to reduce the transport lengths for both electron and ion for the large surface areas readily accessible to electrolyte compared with the bulk material. Furthermore, the micro-nanostructure enables easy operation in terms of electrode fabrication and high tap density, compared with the nanoparticles. In addition, from the HRTEM image in Fig. 6e, the sample exhibits a set of lattice fringe with the d spacing value of 0.43 nm, which is indexed to the (101) facet of LiMnPO_4 . Importantly, the sample has a uniform and perfect carbon coating on the surface of LiMnPO_4 with a thickness of around 2 nm, which is beneficial to the electron transfer. It also reveals that the flower-like sample is a single crystal from the clear lattice fringe, which is further confirmed by the FFT pattern with regular spots. A more detailed analysis of FFT pattern reveals that the sample exhibits an exposed (010) as shown in Fig. 6f. Combination the XRD, TEM, HRTEM and FFT pattern, we could further confirm that the b axis of the sheet-like crystals are parallel to the electron beam direction [010], indicating that the product of the flower-like sample exposes a large (010) facet and the [010] direction is the thinnest direction of the particle, which is an optimal situation for LiMPO_4 (M=Mn, Fe) cathode material with favorable lithium ion migration owing to the known fact that lithium (de)insertion in LiMPO_4 (M=Mn, Fe) is along the b-axis.^{28, 29}

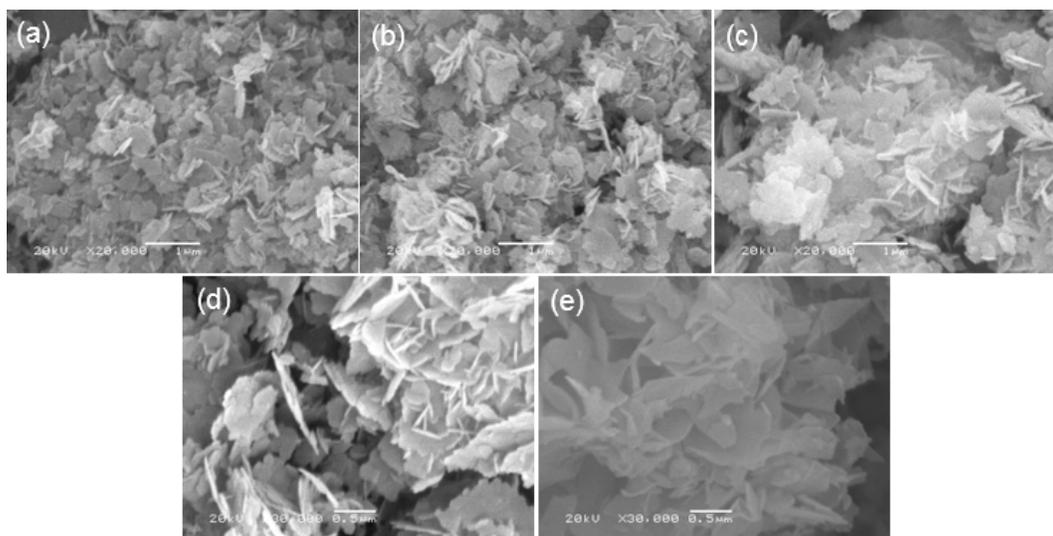


Fig. 5 the SEM images of the samples with various concentration of urea: (a) 0 mol L⁻¹; (b) 0.08 mol L⁻¹; (c) 0.12 mol L⁻¹; (d) 0.16 mol L⁻¹; (e) 0.2 mol L⁻¹

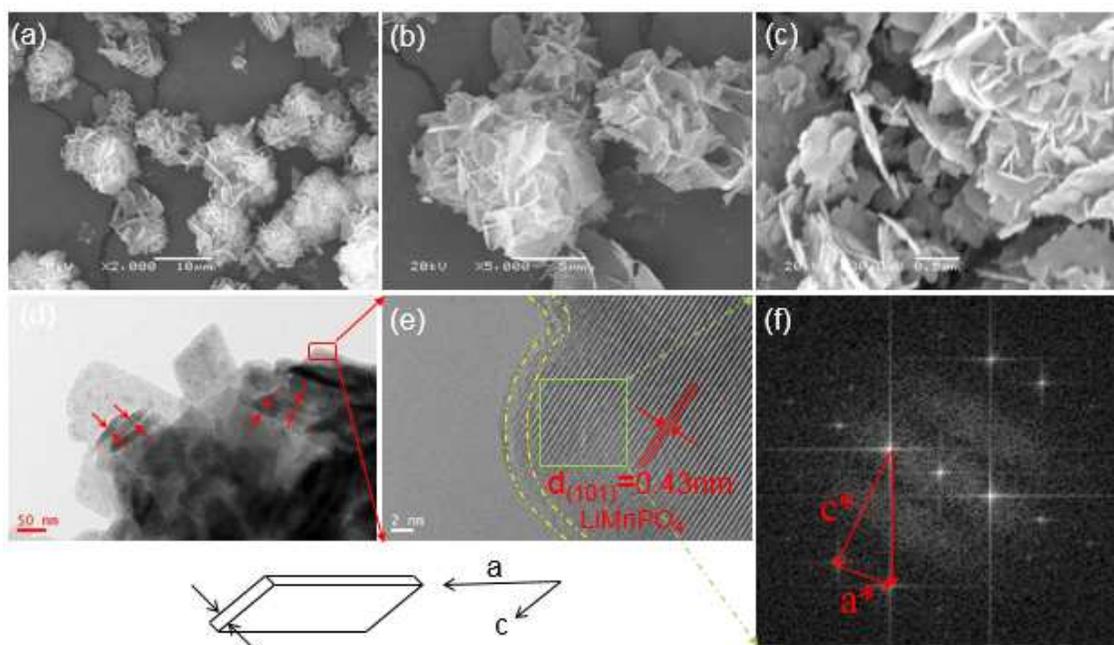


Fig. 6 the morphology of the flower-like LiMnPO₄: (a), (b), (c) the SEM images of LiMnPO₄; (d) the TEM images of LiMnPO₄/C; (e) the HRTEM images from the red region of (d); (f) the FFT of the off-green region of (e)

3.3 electrochemical performance

The electrochemical performance of the as-prepared LiMnPO₄ cathode materials with various morphology and particle size

was studied for its potential as a cathode material for lithium-ion batteries. Fig. 7a shows the typical discharge profiles of LiMnPO_4/C at 0.1 C, in the voltage range of 2.5–4.5 V. All the LiMnPO_4/C exhibits a single discharge plateau around 4.0–4.1 V, corresponding to the redox couple of $\text{Mn}^{3+}/\text{Mn}^{2+}$, in well agreement with the typical lithium extraction-insertion reaction in pure olivine LiMnPO_4 phase. During the initial charge-discharge process, the flower-like LiMnPO_4/C demonstrates a reversible capacity of 150.4 mAh g^{-1} , while 126.3, 128.0, 130.0, 110.2 mAh g^{-1} for s-1, s-2, s-3, s-7, respectively. Obviously, the flower-like LiMnPO_4/C exhibits the longest and the most stable plateau with a highest discharge potential of about 4.1 V vs Li^+/Li , which indicates that the flower-like LiMnPO_4/C has the best reversibility of electrode reaction by reducing electrode polarization owing to the novel morphology with (010) facet exposed. In particular, the sample s-7 shows the shortest plateau with a discharge potential of about 4.0 V, while no other plateau related to the impurity phase is observed. The rate performance of the typical samples are presented in Fig. 7b. The cells are tested at various rate from 0.1 C to 2.0 C. It is noted that the flower-like sample shows a superior rate capacity, in particular at high rate, for example, it exhibits $125.5, 118.6 \text{ mAh g}^{-1}$ at 1.0 C, 2.0 C, respectively. As for s-7, the sample shows the worst electrochemical performance, showing a capacity of $65.2, 50.9 \text{ mAh g}^{-1}$ at 1.0 C, 2.0 C, respectively, which is significantly influenced by the manganese oxide impurity phase. Thus, the amount of urea is a crucial parameter to control the electrochemical properties of LiMnPO_4 in terms of controllable impurity phase. As shown above, these samples excluding s-7 do not contain any impurity phases and differ only in their morphology, particle size and crystal orientation. Therefore, we can imply that the morphology, particle size and crystal orientation play an important role in the electrochemical performance. In terms of the flower-like sample, it has such advantages listed below: (1) the as-prepared porous nanoparticles with the 3D hierarchical structure are beneficial to reduce the Li^+ diffusion pathway, and favorable to the full permeability and infiltration between electrolyte and active material; also, it can decrease the agglomeration, which has been a rather severe challenge to maintain the excellent performance; (2) the flower-like sample is composed of sheet-like nanocrystals with mainly (010) facet exposed, which has large facet in the a-c plane and is thin in the b axis, resulting in an increase in the electrochemical reaction surface area, and enhancement in electrical conductivity and Li^+ diffusion. (3) The well decorated carbon coating provides easy electron transfer, which is beneficial to improve the kinetic process. Thus, it is self-evident that the flower-like LiMnPO_4/C behaves superior electrochemical performance. The samples including s-1, s-2 and s-3 exhibit almost the same electrochemical performance due to the various morphology and structure. For example, s-1 has a desired exposed facet (010) with a relative thick nanosheets, while s-2 shows a less uniform particle size distribution. As the Li:Mn:P molar ratio increased further, the sample s-3 exhibits the spherical nanoparticles with smaller particle size. However, it exhibits agglomeration to some degree, which is adverse to the effective

contact between electrolyte and active material, as well as additive Super P carbon between nanoparticles during the fabrication of the composite electrodes. In addition, the mainly exposed facet is different with the (010). In a word, the electrochemical performance of LiMnPO_4 is significantly affected by the morphology and structure.

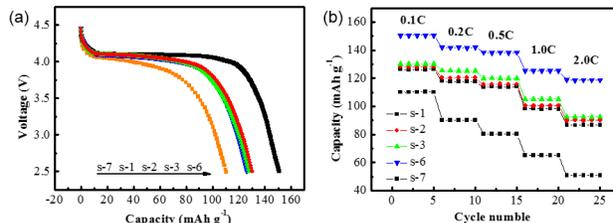


Fig. 7 the electrochemical performance of the typical LiMnPO_4/C samples: (a) the initial discharge curve at 0.1 C; (b) the rate capacity from 0.1 C to 2.0 C

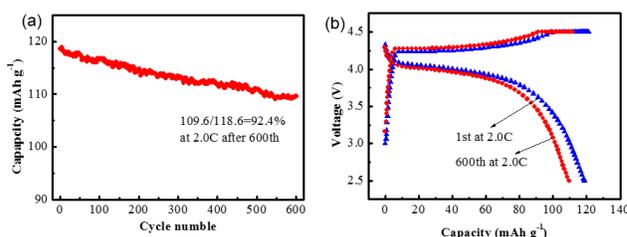


Fig. 8 the cycle performance of the flower-like LiMnPO_4/C : (a) the cycle curve at 2.0 C for 600th; (b) the charge-discharge profile after 1st and 600th

Long cycle performance is one of the significant electrochemical aspects of lithium-ion batteries for high power applications. The cycle performance of the flower-like LiMnPO_4/C is shown in Fig. 8a. The cells were cycled at 2.0 C in the voltage range of 2.5–4.5 V. As shown in Fig. 8b, the initial discharge capacities of LiMnPO_4 sample is 109.6 mAh g^{-1} with a potential plateau of about 4.0 V. As usual, upon increasing the cycle number, the samples exhibit slight polarization, which is reflected in the term of potential difference between the charge-discharge plateaus, thereby declining the storage capacity of the electrode. However, the flower-like sample displays excellent cycle stability with a negligible voltage fading after 600 cycles at 2.0 C, showing a capacity retention up to 92.4%, which indicates that those novel morphology can obviously improve the cycle behaviour, especially at high rate.

Conclusions

In summary, we successfully synthesized the LiMnPO_4 nanoparticle with different morphology and particle size via solvothermal method. The results show that morphology and particle size was controllable synthesized with various precursor composite tailoring with the molar ratio of Li:Mn:P.

From 2:1:1 to 3:1:1, the particle morphology evolves from sheet-like to spherical texture while the particle size reduced, accompanied with the phase composition from Mn²⁺-containing precursor to Li⁺-containing precursor. In addition, the urea is used to further confirm the conclusions, in which the urea has no obvious effect on the morphology, but prevent crystals from growing and reduce the agglomeration. At the concentration of 0.16 mol L⁻¹, the sample consists of small sheet-like crystals with flower-like texture, delivering excellent reversible capacity of 150.4, 125.5, 118.6 mAh g⁻¹ at 0.1, 1.0, 2.0 C, respectively. The composites show long cycle stability at high rate, displaying a capacity retention up to 92.4% with no apparent voltage fading after 600 cycles at 2.0 C. The attracting electrochemical performance of the flower-like LiMnPO₄/C composites suggests that the unique 3D hierarchical structure with (010) facet exposed and well carbon decorated is a promising way for high power Li-ion batteries electrode materials. Combination of the simple and facile synthesis method, it may also be applicable for the other olive type cathode materials such as LiFePO₄, LiCoPO₄ that require electrodes with high electronic and ionic conductivity.

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Notes and references

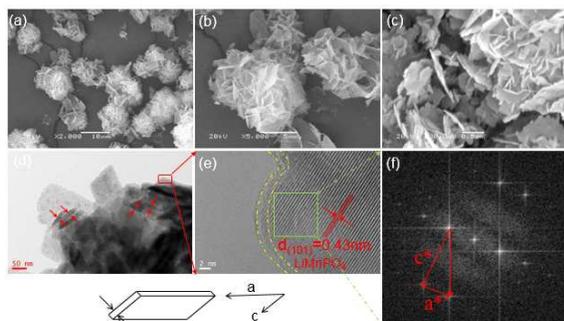
- A. K. Padhi, K. S. Nanjundaswamy and J. B. Goodenough, *J. Electrochem. Soc.*, 1997, **144**, 1188-1194.
- M. Yonemura, A. Yamada, Y. Takei, N. Sonoyama and R. Kanno, *J. Electrochem. Soc.*, 2004, **151**, A1352-A1356.
- C. Delacourt, L. Laffont, R. Bouchet, C. Wurm, J. B. Leriche, M. Morcrette, J. M. Tarascon and C. Masquelier, *J. Electrochem. Soc.*, 2005, **152**, A913-A921.
- D. Arčon, A. Zorko, P. Cevc, R. Dominko, M. Bele, J. Jamnik, Z. Jagličić and I. Golosovsky, *J. Phys. Chem. Solids*, 2004, **65**, 1773-1777.
- A. Yamada, M. Hosoya, S.-C. Chung, Y. Kudo, K. Hinokuma, K.-Y. Liu and Y. Nishi, *J. Power. Sources*, 2003, **119-121**, 232-238.
- N. H. Kwon and K. M. Fromm, *Electrochim. Acta*, 2012, **69**, 38-44.
- A. Vadivel Murugan, T. Muraliganth, P. J. Ferreira and A. Manthiram, *Inorg Chem*, 2009, **48**, 946-952.
- Y. Wang, Y. Yang, Y. Yang and H. Shao, *Solid State Commun.*, 2010, **150**, 81-85.
- F. Wang, J. Yang, P. Gao, Y. NuLi and J. Wang, *J. Power. Sources*, 2011, **196**, 10258-10262.
- L.-e. Li, J. Liu, L. Chen, H. Xu, J. Yang and Y. Qian, *RSC Adv.*, 2013, **3**, 6847-6852.
- J. Liu, X. Liu, T. Huang and A. Yu, *J. Power. Sources*, 2013, **229**, 203-209.
- J. Su, B.-Q. Wei, J.-P. Rong, W.-Y. Yin, Z.-X. Ye, X.-Q. Tian, L. Ren, M.-H. Cao and C.-W. Hu, *J. Solid State Chem.*, 2011, **184**, 2909-2919.
- X.-L. Pan, C.-Y. Xu, D. Hong, H.-T. Fang and L. Zhen, *Electrochim. Acta*, 2013, **87**, 303-308.
- Z. Dai, L. Wang, F. Ye, C. Huang, J. Wang, X. Huang, J. Wang, G. Tian, X. He and M. Ouyang, *Electrochim. Acta*, 2014, **134**, 13-17.
- K. Dokko, S. Koizumi, H. Nakano and K. Kanamura, *J. Mater. Chem.*, 2007, **17**, 4803-4810.
- L. Hu, B. Qiu, Y. Xia, Z. Qin, L. Qin, X. Zhou and Z. Liu, *J. Power. Sources*, 2014, **248**, 246-252.
- Z. Dai, L. Wang, X. He, F. Ye, C. Huang, J. Li, J. Gao, J. Wang, G. Tian and M. Ouyang, *Electrochim Acta*, 2013, **112**, 144-148.
- D. Fujimoto, Y. Lei, Z.-H. Huang, F. Kang and J. Kawamura, *Int. J. Electrochem.*, 2014, **2014**, 1-9.
- D. Wang, H. Buqa, M. Crouzet, G. Deghenghi, T. Drezen, I. Exnar, N.-H. Kwon, J. H. Miners, L. Poletto and M. Grätzel, *J. Power. Sources*, 2009, **189**, 624-628.
- K. Dokko, S. Koizumi, K. Shiraishi and K. Kanamura, *J. Power. Sources*, 2007, **165**, 656-659.
- K. Dokko, T. Hachida and M. Watanabe, *J. Electrochem. Soc.*, 2011, **158**, A1275-A1281.
- Z. Li, K. Zhu, J. Li and X. Wang, *CrystEngComm*, 2014, **16**, 10112-10122.
- B. Ellis, W. H. Kan, W. R. M. Makahnouk and L. F. Nazar, *J. Mater. Chem.*, 2007, **17**, 3248-3254.
- J. Chen, J. Bai, H. Chen and J. Graetz, *J. Phys. Chem. Lett.*, 2011, **2**, 1874-1878.
- K. Dokko, K. Shiraishi and K. Kanamura, *J. Electrochem. Soc.*, 2005, **152**, A2199-A2202.
- K. Shiraishi, K. Dokko and K. Kanamura, *J. Power. Sources*, 2005, **146**, 555-558.
- C. M. Burba and R. Frech, *J. Electrochem. Soc.*, 2004, **151**, A1032-A1038.
- M. S. Islam, D. J. Driscoll, C. A. J. Fisher and P. R. Slater, *Chem. Mat.*, 2005, **17**, 5085-5092.
- C. Ouyang, S. Shi, Z. Wang, X. Huang and L. Chen, *Phys. Rev. B*, 2004, **69**, 104303.



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The hierarchical flower-like LiMnPO_4/C micro-nanostructures with (010) exposed and uniform carbon coated were prepared to provide a superior Li^+ migration and electron transfer via the solvothermal method with the assisted of urea. The as-prepared material showed excellent high rate capacity, and behaved long cycle performance.

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