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6 Biotic oxidation of polyethylene by using bio-surfactant 7 produced by *B.licheniformis*: A novel technique.

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9 Abstract:

10 Polyethylene was incubated with bio-surfactant producing bacterium
11 *B.licheniformis* for 2 months in a suitable media. Lower concentrations of NaCl were added to study its effect
12 on bio-surfactants activity. Being amphiphilic, surfactant has the unique ability to decrease surface energy and
13 this decrease measured through surface tension of the medium was 50%. Surfactant was able to oxidize both
14 control (unoxidized) and pre-oxidized polyethylene during incubation. Oxidation level of control
15 polyethylene sample increased in the presence of NaCl and oxidation level was higher in the presence of 1%
16 of NaCl than that of 0.5% NaCl. Higher amount of surfactant was also produced, observed as comparatively
17 low surface tension in the presence of NaCl. During the bio-oxidation of polyethylene, higher amount of
18 unsaturated hydrocarbons were formed than carbonyl group. This oxidation was also observed through
19 reduced crystalline property and cracked polyethylene surface under SEM. It was also observed that the
20 oxidation product formed during oxidation by bio-surfactant was solubilising into the liquid media. For this
21 rapid loss of oxidation product, deterioration of mechanical property of all the treated polyethylene samples
22 was observed and this deterioration was highest in case of pre-oxidised polyethylene incubated with bio-
23 surfactant for 2 months. In this study, a novel unique method of bio-oxidation of polyethylene by bio-
24 surfactant was established.

25 A

26 1. Introduction:

27 The use of polyethylene films rise immensely in the last
28 decade for its low cost, strong mechanical property with
29 relatively lower thickness, multiple chemical resistances and

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30 longevity. Due to the absence of any polar groups in closely
31 packed carbon-hydrogen backbone and high molecular weight,
32 polyethylene is resistant towards abiotic oxidation and
33 microbial attack under natural condition. Therefore,
34 polyethylene once produced, does not degrade naturally,
35 resulting in high amount of waste polyethylene accumulation
36 (25 million tons per year) in the environment. This
37 environmental pollution becomes one of the most important
38 concerns of researchers and lots of studies have been done to
39 find a way for degradation of polyethylene waste^{1,2}. Decades
40 of researches have shown that biotic i.e. microbial degradation
41 of polyethylene is the only way for environmental friendly
42 degradation process. Microbial degradation is mainly
43 achieved by formation of bio-film on the polyethylene surface.
44 Lots of microbes i.e. *Lysinibacillus xylanilyticus*, *Aspergillus*
45 *niger*, *Penicillium pinophilum*, *Rhodococcus ruber* have been
46 used for microbial degradation of polyethylene^{3, 4, 1}. To
47 enhance microbial attachment to the polyethylene surface,
48 hydrophobic property of polyethylene should be changed into
49 hydrophilic one through abiotic oxidation¹. U.V has been
50 used in several studies for abiotic oxidation of polyethylene to
51 introduce polar groups in the polymer backbone and thereby to
52 reduce the hydrophobic property; for example, polyethylene
53 containing pro-oxidant has been oxidised by using U.V. for 60
54 hour with the increase in the carbonyl index^{1,3,5}. In another
55 study, polyethylene mixed with pro-oxidant additives has been
56 oxidised by U.V and highest oxidation level is achieved after
57 600 hour². Another way to increase degradation rate, other
58 than the pre-oxidation step is by increasing contact surface
59 between polyethylene and water by adding surfactant in the
60 biodegradation system.

61 Surfactants are wide varieties of surface active
62 amphiphilic (both hydrophilic and hydrophobic part is present

63 in the same molecule) molecules⁶. Surfactants have the
64 ability to reduce surface tension and interfacial tension of the
65 solution. Bioavailability of non-soluble/ hydrophobic material
66 is enhanced by the reduction of surface tension. Surfactants
67 also have the ability to enhance solubility of petroleum
68 hydrocarbons. Surface tension reduction ability of surfactants
69 can be used to increase biodegradation of polyethylene. In
70 some of the polyethylene biodegradation studies chemical
71 surfactant has been used into the system. Increased
72 biodegradation of pre-oxidised polyethylene has been reported
73 in soil after addition of surfactant into the system⁷. In another
74 study, enhanced bio-film formation has been observed by
75 adding tween 80 in the polyethylene biodegradation system
76 containing *Pseudomonas aeruginosa*⁸. Another types of
77 surfactant known as bio-surfactant, are produced by different
78 microbes⁹. Main advantage of bio-surfactants over chemical
79 surfactants is its environmental friendly nature and they are
80 also bio-degradable. One such bio-surfactant producing
81 bacterium is *Pseudomonas aeruginosa* and its bio-surfactant is
82 called rhamnolipid^{10, 11}. This specific bacterium has been
83 used to study biodegradation of polypropylene and bio-film
84 formation has been enhanced by rhamnolipid production¹².
85 Bio-surfactant producing bacteria *Bacillus pumilus*, *B.*
86 *halodenitrificans*, *B. cereus* have been used for biodegradation
87 of pre-oxidised polyethylene containing pro-oxidant and
88 enhanced biodegradation has been reported². There are
89 mainly 6 types of bio-surfactants produced by bacteria, among
90 which Lipopeptides are most important and effective bio-
91 surfactants for high surface activity, mainly produced by
92 different *Bacillus* strain^{13, 6}. Such bio-surfactant producing
93 bacterium is *Bacillus licheniformis*, isolated from oil reservoir
94¹⁴. This bacterium also has been used to enhance the oil
95 recovery through the production of biosurfactant^{15, 16, 5}. Bio-

96 surfactant produced by this bacterium is known as Lichenysin
97 and is a lipopeptide, which is reported for its ability to reduce
98 the surface tension¹⁶. *B.licheniformis* can be used for
99 changing hydrophobic property of polyethylene into
100 hydrophilic one by the action of bio-surfactant produced by
101 this bacterium in the presence of polyethylene. For this
102 treatment, a suitable growth medium should be used for
103 optimum production and optimum activity of the bio-
104 surfactant. Previously, it is reported that higher amount of
105 bio-surfactant is produced by *B.licheniformis* in a growth
106 medium without any trace of NaCl compared to the medium
107 containing 0.5% NaCl¹⁷. But, in another study, 5% NaCl
108 concentration in a growth medium is reported as optimum
109 condition for maximum production of bio-surfactant¹⁶.

110 Bio-surfactant producing *Bacillus licheniformis*
111 has not yet been used to study its effect on the oxidation of
112 polyethylene. Oxidation of unoxidized polyethylene and
113 chemical modifications of pre-oxidized polyethylene by the
114 bio-surfactant produced by *B.licheniformis* is reported in this
115 study in the presence or absence of NaCl during the incubation
116 of polyethylene for 2 months.

117

118 2. Materials and Methods:

119 2.1. Test materials:

120 Daily used 0.01 mm thick, transparent colourless polyethylene
121 bags were collected from the waste bins of Kolkata Municipal
122 Corporation. Bags were then cut into rectangular pieces (5mm
123 × 5mm) and washed vigorously with soap water and distilled
124 water, consecutively to remove any debris and bio-material
125 attached with the polyethylene surface. Rectangular pieces

126 were then dried at 60°C overnight in a hot-air oven. These
127 unoxidised polyethylene films were used as control
128 polyethylene films.

129 For heat-U.V. treatment, rectangular pieces of polyethylene
130 films were taken into a beaker and kept in custom made
131 chamber where the temperature was kept at 60°C under
132 continuous U.V. light for 1 month. Wavelength of the U.V.
133 light was within a range of 350 nm to 200 nm.

134 2.2. Microbial Culture:

135 *Bacillus licheniformis* JF2 (ATCC No. 39307, MTCC
136 No. 2454) was used for bio-treatment study. This microbial
137 culture was obtained from Institute of Microbial Technology,
138 Chandigarh, India. Microbial culture was maintained in
139 nutrient broth (Himedia). Bio-treatment was carried out in
140 YPD medium containing 10 g of yeast extract, 20 g of glucose
141 and 20 g of peptone in 1 litre of double distilled H₂O at 37°C
142 for different time period. 0.5% and 1% of Sodium Chloride
143 (NaCl) was added to the YPD medium incubated with
144 *B.licheniformis* containing control polyethylene to study its
145 effect on the stability of the bio-surfactant and its ability to
146 reduce the surface tension. Control polyethylene samples
147 incubated with *B.licheniformis* in the YPD growth medium
148 without NaCl and with NaCl of 1%, 0.5% and U.V. treated
149 polyethylene films incubated with *B.licheniformis* in YPD
150 growth medium were kept for 2 months. After 1 and 2
151 months, samples from each case were harvested, washed and
152 dried. For negative control, polyethylene samples were kept
153 in YPD growth medium without any bacterial sp. All the
154 samples were incubated at 37°C and in triplicate.

155 Surface tension (σ) of microbial culture medium was
156 measured by stalagmometer at 25°C at day zero and at

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157 different intervals of time¹⁸. Surface tension was calculated
 158 by following formula.

$$159 \quad \sigma_{water} \times \frac{\text{weight of } N \text{ drops of solvent}}{\text{weight of } N \text{ drops of water}} = \sigma_{solvent}$$

160
 161 σ_{water} is the surface tension of distilled water at 25°C and
 162 $\sigma_{solvent}$ is surface tension of solvent. N is the number of
 163 drops and this was same for both water and solvent. Every
 164 measurement was done in triplicate.

165 2.3. Characterization of Polyethylene:

166 FTIR analysis was carried out with ATR-FTIR
 167 (model alpha, Bruker, Germany) spectrometer, scanning from
 168 4000 cm⁻¹ to 500 cm⁻¹ at room temperature. The resolution
 169 was set at 4 cm⁻¹ with 42 scans per spectrum. Carbonyl index
 170 (C.I.) and double bond index (D.B.I.) were calculated using
 171 the ratio of absorbance frequency of the carbonyl peak (1740
 172 cm⁻¹) and double bond (1650 cm⁻¹) to that of the CH₂ group
 173 bending frequency (1465 cm⁻¹) respectively.

174 All polyethylene samples were sputter coated with gold layer
 175 by a Hitachi sputter coater (model-E1010 Ion Sputter), Japan.
 176 Photomicrographs were observed under scanning electron
 177 microscope (EVO 18, Carl Zeiss, Germany).

178 X-ray diffraction study of all types of polyethylene samples
 179 were recorded with an X-ray diffractometer (PANalytical,
 180 Netherlands) at an angle of 2θ from 3° to 50° and a fixed scan
 181 rate of 1° min⁻¹. Percentage (%) of crystallinity was calculated
 182 by using the following formula.

$$183 \quad \%Crystallinity = \frac{\text{Area under crystalline peaks}}{\text{Total Area under all peaks}} \times 100\%$$

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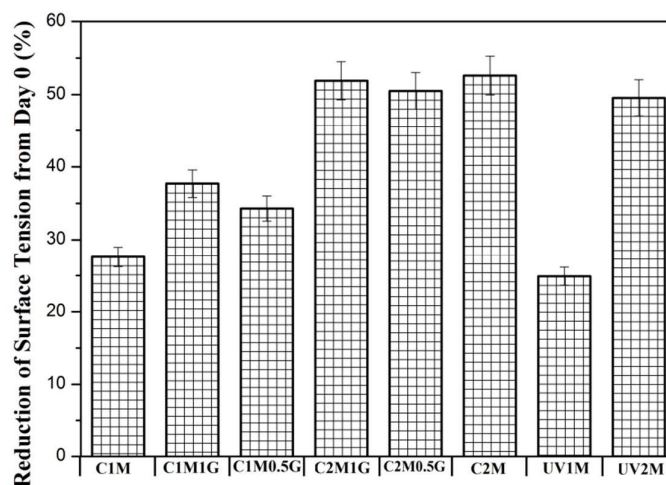


Figure 1: Surface reductions. **C1M**-control polyethylene incubated for 1 month with *B.licheniformis* without NaCl ; **C1M1G**-control polyethylene incubated for 1 month with *B.licheniformis* in the presence of 1% of NaCl; **C1M0.5G**- control polyethylene incubated for 1 month with *B.licheniformis* in the presence of 0.5% of NaCl; **C2M1G**- control polyethylene incubated for 2 months with *B.licheniformis* in the presence of 1% NaCl; **C2M0.5G**- control polyethylene incubated for 2 months with *B.licheniformis* in the presence of 0.5% of NaCl; **UV1M**-UV treated polyethylene incubated for 1 month with *B.licheniformis*; **UV2M**- UV treated polyethylene incubated for 2 months with *B.licheniformis*.

184 All samples were observed under AFM
 185 (Bruker- Germany, model: Innova) without doing any kind
 186 pre-treatment. Depth analysis of the surface of treated
 187 polyethylene was calculated by ‘Nanoscape analysis’
 188 software.

189 Tensile property of all polyethylene samples were tested by a
 190 tensile tester (Tinius Olsen H5KT, ASTM D638 standard) at
 191 23°C with a cross head speed of 20 mm/min. All eight
 192 samples were cut into rectangular shape of 100 mm in length
 193 and 10 mm in width and 0.01 mm thick.

194 3. Results and Discussion:

195 3.1. Surface Tension:

196 Reduction of surface tension from day 0 in
 197 percentage is presented in Figure 1. Minimum surface tension
 198 is achieved after 2 months of incubation with *B.licheniformis*
 199 with both control and pre-oxidized polyethylene with or
 200 without NaCl. The surface tension reduction from day 0, after
 201 2 months of incubation with *B.licheniformis*, are 51.9% in
 202 case of YPD medium with control polyethylene and 1% of
 203 NaCl, 50.5% in case of YPD medium with control
 204 polyethylene and 0.5% of NaCl, 52.6% in case of YPD
 205 medium with control polyethylene and without NaCl and
 206 49.6% in case of YPD medium with UVPE.. After 1 month,
 207 minimum surface tension is achieved in case of YPD medium
 208 incubated with *B.licheniformis* in the presence of 1gm of NaCl
 209 and control polyethylene. Presence of surface active
 210 molecules can be indirectly predicted from this reduction of
 211 surface tension. In a previous study, surface tension reduction
 212 from 70mN/m to 58.8 mN/m in the presence of pro-oxidant
 213 containing polyethylene by *B. pumilus*, *B. halodenitrificans*
 214 and *B. cereus* is reported ². But in the present study, the
 215 minimum surface tension achieved in the presence of control
 216 polyethylene (unoxidised) and UVPE by *B. licheniformis* is
 217 much lower than the previously reported one. NaCl was
 218 added in the growth medium to observe its effect on surface
 219 tension reduction ability of bio-surfactant. It is clear from the
 220 above mentioned result that after 1 month, lowest surface
 221 tension is observed in case of growth medium containing 1%
 222 of NaCl. As reported by previous studies, lower concentration
 223 of NaCl stabilizes the bio-surfactant and increases its activity;
 224 this may be the reason for observed comparatively lower
 225 surface tension by the bio-surfactant produced by

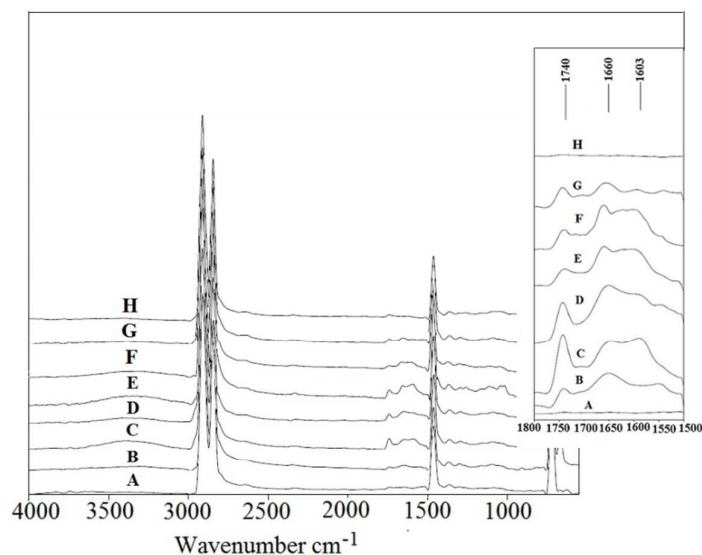


Figure 2: FTIR spectra of treated control (unoxidized) polyethylene. A-control PE, B-C1M; C-C1M1G; D-C1M0.5G; E-C2M1G; F-C2M0.5G; G-C2M; H-negative control.

226 *B.licheniformis* in the presence of 1% of NaCl¹⁶. After 2
 227 months, minimum surface tension achieved which is slightly
 228 higher than reported minimum surface tension by
 229 *B.licheniformis*, is almost same in the all the cases of YPD
 230 medium incubated with *B.licheniformis* with control and pre-
 231 oxidized polyethylene with or without any NaCl¹⁶. May be for
 232 this reason presence of NaCl do not affect surface tension
 233 reduction ability of bio-surfactant after 2 months of
 234 incubation, though presence of NaCl stabilizes bio-surfactant
 235 produced by *B.licheniformis* in the presence of polyethylene
 236 after 1 month of incubation. In case of negative control,
 237 polyethylene kept in YPD medium without any bacteria, no
 238 change in the surface tension is observed during 2 month of
 239 bio-treatment time (Figure 13).

240 **3.2.Characterization of Polyethylene Incubated with**
 241 ***B.licheniformis*:**

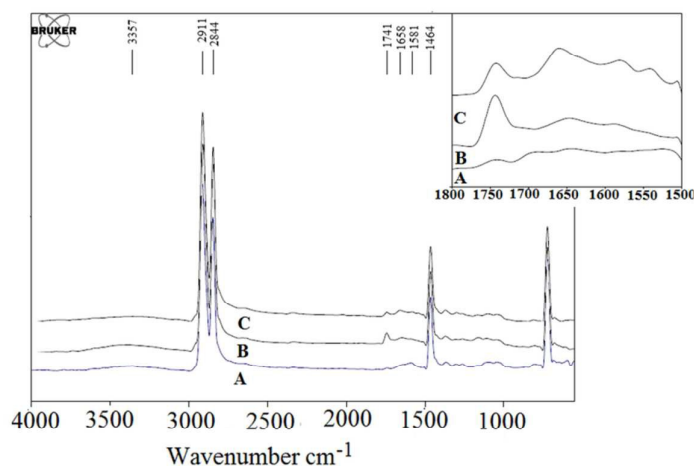


Figure 3: FTIR spectra of treated UVPE (Preoxidized) polyethylene. A-UVPE, B-UVIM, C-UV2M,

242 Control polyethylene films were incubated with *B.*
 243 *licheniformis* for 1 (C1M) and 2 (C2M) months in YPD
 244 medium without NaCl. Control polyethylene samples were

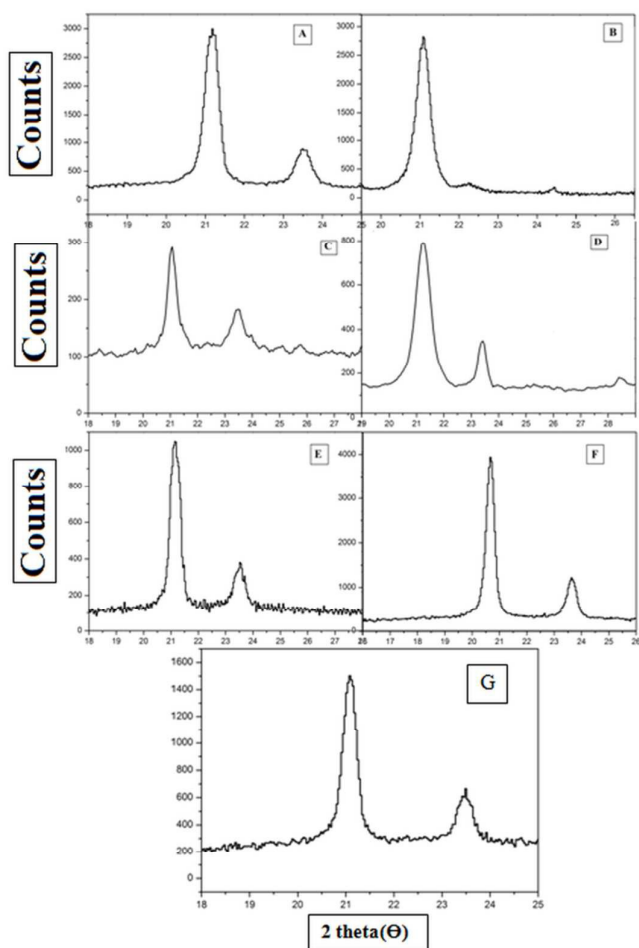


Figure 5: XRD spectra A-control PE, B-C1M, C-C1M1G, D-C1M0.5G, E-C2M, F-C2M1G, G-C2M0.5G

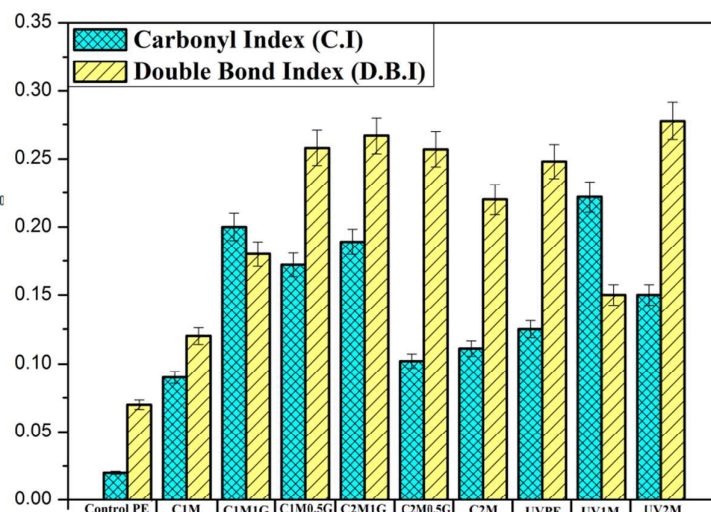


Figure 4: Carbonyl index and double bond index of treated and untreated polyethylene.

245 also incubated with *B.licheniformis* in YPD medium
 246 containing 1% (C1M1G) and 0.5% (C1M0.5G) of NaCl for 1
 247 month and 2 months (C2M1G for 1% NaCl and C2M0.5G for
 248 0.5% NaCl) separately. To study the effect of surfactant on
 249 pre-oxidized polyethylene sample, oxidized polyethylene
 250 samples by U.V. were also incubated with *B. licheniformis*
 251 for 1 (UV1M) and 2 (UV2M) months in YPD medium.
 252 Polyethylene incubated in YPD growth medium without any
 253 bacterial sp. was also characterized. No bio-film formation
 254 was observed on the treated polyethylene sample, incubated
 255 with *B.licheniformis*.

3.2.1. FTIR Analysis:

257 FTIR spectra of treated control (unoxidized)
 258 polyethylene samples are compared to the untreated and
 259 negative control one in Figure 2. Appearance of new peaks in
 260 the 1800-1500 cm^{-1} region can be observed in all types of
 261 treated control polyethylene samples. Another peak at 3400
 262 cm^{-1} appears in case of C1M1G, which is due to -OH group
 263 formation. Peaks at 1800-1500 cm^{-1} region are resulted by
 264 overlapping of 1740 cm^{-1} (for ketones), 1730 cm^{-1} (for
 265 aldehydes) and 1660 cm^{-1} for the formation of unsaturated

266 hydrocarbons (-C=C-). After 2 months of incubation of
 267 *B.licheniformis* in YPD medium, in case of C2M1G (1% of
 268 NaCl), C2M0.5G (0.5% of NaCl) and C2M (without NaCl)
 269 peak intensity at 1660 cm⁻¹ (for unsaturated hydrocarbons)
 270 increase than that of the C1M1G, C1M0.5G and C2M
 271 respectively. In case of C2M peak intensity of 1740 cm⁻¹ (for
 272 ketones) increases slightly from C1M and in case C2M1G and
 273 C2M0.5G peak intensity of 1740 cm⁻¹ (for ketones) decrease
 274 from that of the C1M1G and C1M0.5G respectively (Figure
 275 2). This indicates that more unsaturated hydrocarbons are
 276 formed compared to the ketone groups. In case of treated
 277 UVPE (Figure 3), after 1 and 2 months, peak intensity of
 278 1800-1500 cm⁻¹ region increase. In case of UV1M, an
 279 increase in the intensity of peak at 1740 cm⁻¹ is observed due
 280 to the formation of ketones. But after 2 months, intensity of
 281 peak at 1740 cm⁻¹ decreases and the intensity of peak at 1660
 282 cm⁻¹ increases. Carbonyl Index (C.I) and Double bond Index
 283 (D.B.I) of treated polyethylene films are compared to that of
 284 the untreated polyethylene i.e control PE and UVPE in Figure
 285 4. Both of C.I. and D.B.I. of treated polyethylene films
 286 increases from that of the untreated polyethylene (control and
 287 UVPE) films. Formation of unsaturated hydrocarbons is much

288 higher than that of the C=O bonds in case of all the treated
 289 control polyethylene except C1M1G as observed from FTIR
 290 spectra and C.I.-D.B.I graph (Figure 2, 4). But after 2 months,
 291 increase of D.B.I is observed in case of C2M1G, C2M0.5G
 292 and C2M. In case of UVPE, after 1 month, formation of C=O
 293 group is comparatively higher than formation of unsaturated
 294 hydrocarbons. But after 2 months, in case of UV2M, amount
 295 of C=O group is comparatively low than amount of
 296 unsaturated hydrocarbons. From this result, it is evident that
 297 the polar group are formed in the polymer backbone. After
 298 oxidation by means of U.V and heat, polyethylene shows
 299 similar changes observed through increased carbonyl index¹.
 300 Bio-surfactant being an amphiphilic molecule has the ability
 301 to increase the solubilisation of hydrocarbons. Hydrophobic
 302 part of the bio-surfactant remains attached with the
 303 polyethylene surface with hydrophilic part protruding towards
 304 the aqueous solution. This phenomenon enhances
 305 polyethylene's availability to the dissolved oxygen, which
 306 further results in the oxidation of polyethylene. In case of pre-
 307 oxidized polyethylene samples, oxidation rate is higher than
 308 control polyethylene. Previously formed oxidation product
 309 present in pre-oxidized polyethylene may be helping in the
 310 bio-surfactant initiated oxidation. But after 2 months, already

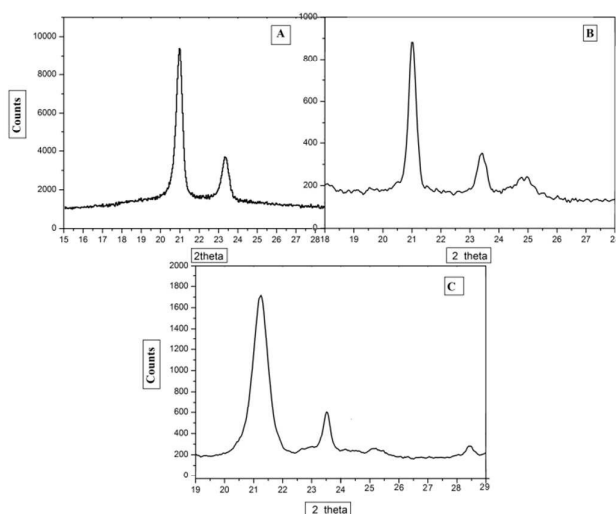


Figure 6: XRD spectra A-UVPE, B-UV1M, C-UV2M.

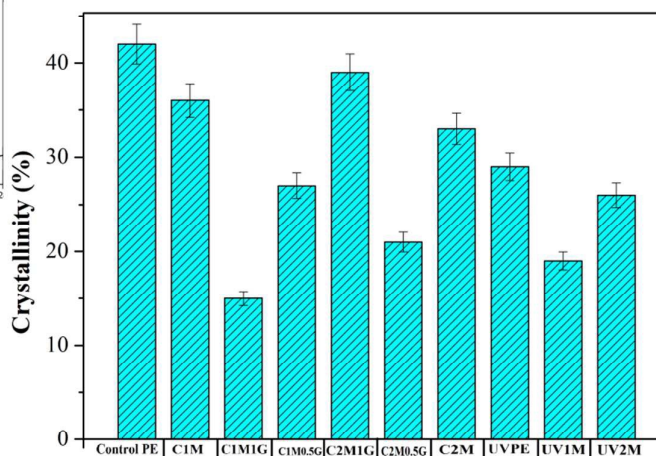


Figure 7: Crystallinity (%) of treated and untreated polyethylene

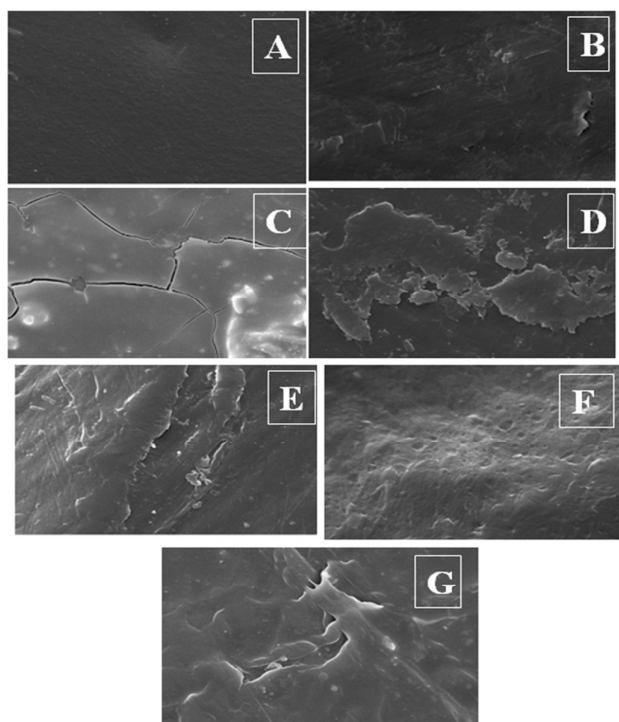


Figure 8: SEM Image A-Control PE, B-C1M, C-C1M1G, D-C1M0.5G, E-C2M, F-C2M1G, G-C2M0.5G

311 formed oxidation product may get solubilised into the aqueous
 312 medium, resulting in a decrease of the intensity of the peak at
 313 1740 cm^{-1} . Another reason for this phenomenon can be
 314 conversion of carbonyl groups into double bonds due to its
 315 further oxidation. Similar phenomenon is also reported in
 316 another study, where the conversion of carbonyl groups into
 317 double bonds has been observed during degradation process
 318 by *Lysinibacillus sp.* In this study, 42% reduction of carbonyl
 319 index and 200% increase of double bond index is reported
 320 after 18 weeks of incubation with *lysinibacillus sp* with U.V
 321 irradiated films³. But this reduction in carbonyl index is 32%
 322 and increase in double bond index is 85% in case of UVPE
 323 incubated in the presence of bio-surfactant for 2 months which
 324 is much higher than previously reported data. In another
 325 study, such 75% reduction of carbonyl index of pre-oxidized
 326 polyethylene containing pro-oxidant by bacteria strain in 21
 327 days is reported. Formation of unsaturated hydrocarbons after
 328 bacterial treatment is also reported in GC-MS studies².

329 No change in peak intensity is observed in case of negative
 330 control from the untreated control polyethylene.

331 3.2.2.XRD analysis:

332 XRD spectra of biologically treated samples are
 333 compared to the untreated control polyethylene sample in the
 334 Figure 5. In Figure 6, XRD spectra of biologically treated
 335 UVPE samples are compared to the untreated UVPE sample.
 336 Peaks at 21° and 23.5° are the characteristic peaks of semi-
 337 crystalline polyethylene molecule. Crystallinity in percentage
 338 (%), calculated from these XRD spectra of biologically treated
 339 polyethylene samples are represented in Figure 7. After
 340 biological treatment, crystallinity (%) of control polyethylene
 341 samples incubated with 1% and 0.5% of NaCl and UV treated
 342 polyethylene samples decrease from that of the untreated
 343 control polyethylene and UVPE samples after 1 month of
 344 bacterial incubation. Lowest crystallinity (%) is observed in
 345 case of C1M1G. After 1 month of bacterial treatment
 346 crystallinity decrease and 2 months crystallinity increase as
 347 observed in XRD analysis of C2M1G (Figure 7). In case of
 348 UVPE, similar trend is observed. But in case of control
 349 polyethylene incubated with *B.licheniformis* without NaCl and
 350 with 0.5% of NaCl, after 1 and 2 months crystallinity decrease
 351 from the untreated control polyethylene (figure 7). During the

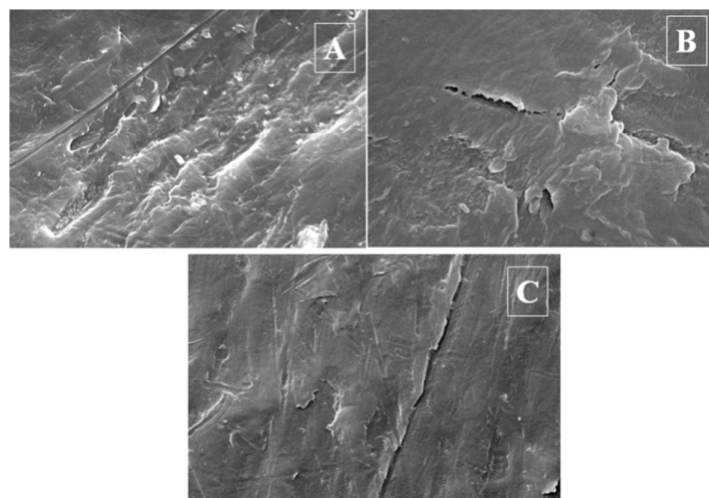


Figure 9: SEM Image A-UVPE, B-UV1M, C-UV1M.

352 process of oxidation by natural aging or by U.V., the
 353 crystallinity of polyethylene is found to increase during initial
 354 phase followed by decrease in the second stage. Increase in
 355 crystallinity during abiotic oxidation is observed with the
 356 increase in oxidation level^{19,20,21}. Mainly Short hydrocarbon
 357 chains produced during the process of oxidation, can initiate
 358 secondary crystallization, due to which crystallinity increases
 359 consequently. Such increase due to secondary crystallization
 360 is a common phenomenon in oxidation by natural aging. Such
 361 phenomenon of increase in crystallinity (55%) after oxidation
 362 of polyethylene by natural aging and accelerated aging is
 363 reported by Benitez et al²². Solubilisation of oxidised
 364 amorphous part into aqueous medium can also result in the
 365 increase in crystallinity as reported by Sepulveda et al^{2,4,19,20,}
 366²¹. And during second stage decrease in the crystallinity is
 367 observed to further oxidation of crystalline phase. But in this
 368 study, reduction of crystallinity is observed during initial stage
 369 followed by increase in crystallinity during second stage of

370 oxidation. Attached surfactant on the polyethylene samples in
 371 this bio-surfactant initiated oxidation can cause structural
 372 irregularities in crystalline structure, leading to the reduction
 373 in crystallinity during initial stage. But following increase in
 374 crystallinity can be due to formation of short hydrocarbon
 375 chain initiating secondary crystallization. Solubilisation of
 376 amorphous oxidation product formed during 1 month
 377 incubation of *B.licheniformis* can cause the increase in the
 378 crystalline region, therefore increasing crystallinity after 2
 379 months of incubation. Reduction of crystallinity even after 2
 380 months in case C2M, can be lower oxidation level even after 2
 381 months as observed in FTIR spectra. Another reason can be
 382 lower amount of solubilisation of amorphous region into
 383 aqueous media.

385 3.2.3.Morphological analysis:

386 3.2.3.1Scanning Electron Microscope:

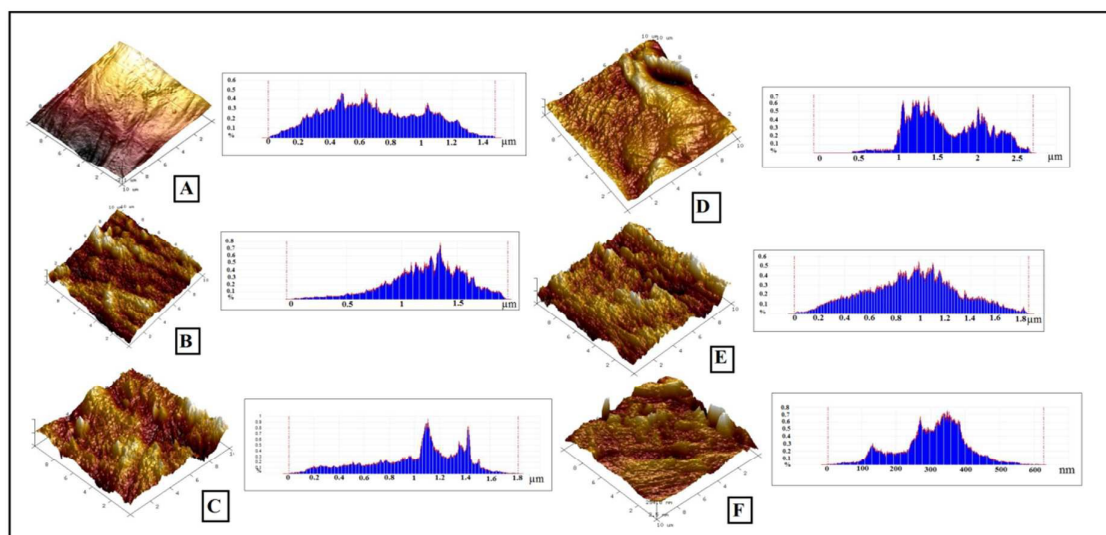


Figure 10: AFM image. A-Control PE, B-C1M, C-C1M1G, D-C1M0.5G, D-C2M,E-UVPE, F-UV2M

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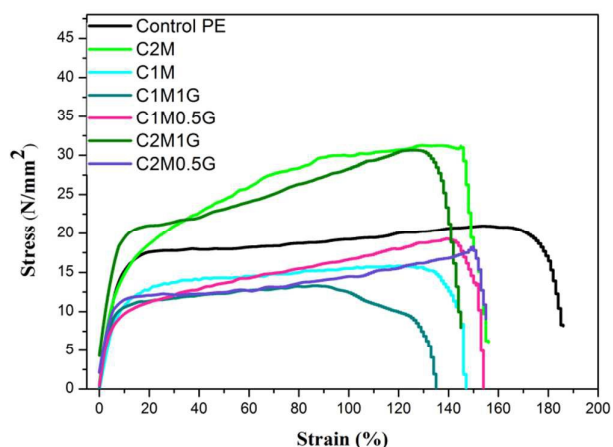


Figure 11: Stress Vs Strain graph of treated and untreated control polyethylene

387 Surface morphology of biologically treated
 388 and untreated control polyethylene is observed under SEM
 389 and images are represented in Figure 8. Images of surface
 390 morphology of biologically treated and untreated UVPE are
 391 represented in Figure 9. Surface morphology of all
 392 biologically treated polyethylene is rough and severely
 393 cracked. This kind of crack formation on the surface is
 394 usually observed in case of oxidized polyethylene samples
 395 resulted from natural contractions. The cracks are formed due
 396 to cross linking during oxidation or due to loss of oxidation
 397 product by solubilisation into bacterial medium from the
 398 surface of polyethylene^{2,3,24}.

399 3.2.3.2 Atomic Force Microscope:

400 Contact mode AFM height images of biologically
 401 treated and untreated polyethylene samples are represented in
 402 Figure 10. The AFM graph shows the amount and depth of
 403 cavity formed during treatment on the polyethylene surface.
 404 Depth of the cavity formed during biological treatment
 405 increases and more cavity forms on the surface of
 406 polyethylene in the case of C1M sample. This cavity
 407 formation process becomes slower after 1 month. After 2
 408 months of biological treatment, deeper cavity formed on the
 409 surface of C2M but the amount is much less than C1M. As it

410 is already reported that amorphous region is readily available
 411 for oxidation than crystalline region of semi-crystalline
 412 polyethylene molecule. These cavity forms due to formation
 413 of nodule made of little or unoxidized crystalline part of
 414 polyethylene molecule and due to solubilisation of oxidation
 415 product into liquid media, resulted during oxidation of
 416 amorphous region by bio-surfactant^{18,24}. During 1 month of
 417 biological treatment, in case of C1M, amorphous region is
 418 oxidised. After 2 months, oxidation process proceeds further
 419 oxidizing amorphous region. But, in case of C2M, depth of
 420 the cavity increase rather than amount of cavity formed on the
 421 polyethylene surface. Also, C2M oxidize more than
 422 solubilised less in amount which can be reason for slight
 423 increase in depth of the cavity than C1M. Depth of the cavity
 424 is higher in case of the C1M0.5G than that of the C1M1G. In
 425 case of the C1M1G, both amorphous region and crystalline
 426 region is oxidised, from where oxidation product solubilises
 427 into liquid medium forming lesser deep cavity. This can also
 428 be observed through reduce crystallinity through XRD data
 429 [Figure 7]. For these reason, depth of the cavity is less in case
 430 of C1M1G. In case of C1M0.5G, only amorphous region is
 431 oxidised leaving crystalline region very little or unoxidized

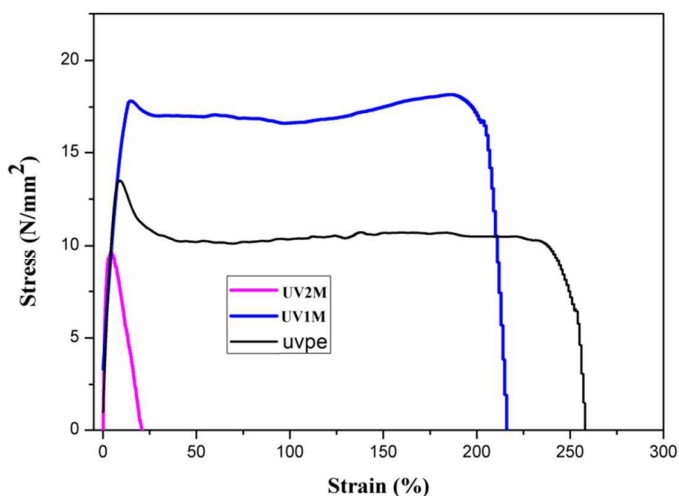


Figure 12: Stress Vs Strain graph of treated and untreated U.V. treated polyethylene

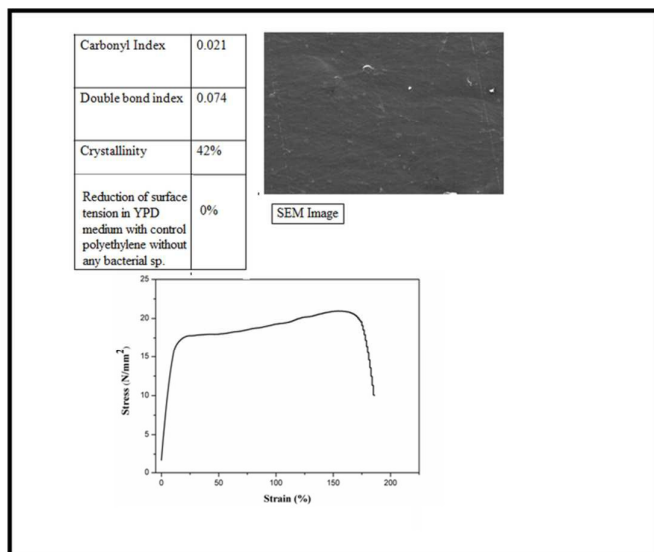


Figure 13: Analysis of Negative control

432 which creates nodules on the surface with deeper cavity. This
 433 result is also in accordance with the observed crystallinity
 434 level in Figure 7. This little or unoxidized crystalline part of
 435 C1M0.5G is further oxidized during 2 months of bacterial
 436 incubation, in case of C2M0.5G. Reduction of crystallinity of
 437 C2M0.5G may be resulted due to this phenomenon.
 438 Crystallinity (%) of C1M0.5G is slightly lower from untreated
 439 control polyethylene. Although in both C2M and C1M0.5G
 440 cases, only amorphous region is oxidised, but deeper cavity is
 441 formed in case of C1M0.5G as oxidation level was higher in
 442 case of C1M0.5G than that of the C2M. In case of UV2M,
 443 after 2 months, due to loss of oxidation product from
 444 amorphous and crystalline region, cavity depth is very less
 445 than other treated polyethylene samples. After 1 month, both
 446 crystalline and amorphous region is oxidised as observed
 447 through reduced crystallinity from Figure 7. After 2 months,
 448 in case of UV2M, oxidation product formed during U.V
 449 treatment and during 1 month of incubation of UVPE with
 450 surfactant, solubilised in to the liquid medium resulting lesser

451 deep cavity. This phenomenon can be co-related with the
 452 slightly increased crystallinity level.

453 3.2.4. Mechanical analysis:

454 In Figure 11, stress versus strain curve of treated
 455 and untreated control polyethylene is plotted. Although,
 456 mechanical property of polyethylene samples do not depend
 457 on the oxidation level, but changes in tensile property is
 458 observed in case of treated control polyethylene samples. It is
 459 previously reported that elongation at break directly depends
 460 on carbonyl index of the polyethylene sample²⁵. This
 461 phenomenon is observed in case of CIM1G which shows
 462 highest reduction in elongation at break. This reduction in
 463 elongation at break of CIM1G can be correlated with its
 464 highest C.I. among all treated control polyethylene samples
 465 after 1 month of bacterial incubation [Figure 4]. Higher C.I.
 466 value is related to higher oxidation level and chain breakage
 467 during oxidation is another possible reason for loss of
 468 mechanical property as observed in these cases through the
 469 loss of elongation at break. In case of mechanical property,
 470 control polyethylene incubated with *B.licheniformis* in YPD
 471 growth medium with 1% of NaCl for 2 months (C2M1G), with
 472 0.5% of NaCl (C2M0.5G) and without NaCl for 2 months
 473 (C2M) shows similar effects (Figure 11). Similar trend of
 474 increasing elongation at break during bacterial treatment is
 475 also reported by Lee et al⁵.

476 Stress versus Strain graph of treated and
 477 untreated UVPE is plotted in Figure 12. After 1 month of
 478 biological treatment, elongation at break of UVIM decreases.
 479 But this decrease in elongation at break is major in case of
 480 UV2M. This polyethylene sample shows total loss of tensile
 481 property. Higher oxidation rate and structural modification
 482 can be a reason for this loss of tensile property. Oxidation
 483 products formed during U.V treatment and during two months

484 of bacterial incubation dissolve into the liquid media, as
 485 observed through SEM and depth analysis by AFM; this is
 486 another reason for the loss of mechanical property.
 487 Negative control polyethylene, kept in YPD medium without
 488 any added bacterial sp., does not show any change of
 489 chemical, physical or mechanical property during 2months of
 490 bio-treatment (Figure 13).

491
 492 Polyethylene is oxidized in the presence of the bio-
 493 surfactant produced by *B.licheniformis*. In the presence of
 494 NaCl, oxidation level is higher. During 2 months incubation,
 495 already formed oxidation products get solubilised into the
 496 aqueous medium which is a characterized property of
 497 surfactant. This property is also enhanced in the presence of
 498 NaCl. Therefore, NaCl present in the medium can stabilise
 499 and enhance activity of bio-surfactant as reported earlier.
 500 Similar level of oxidation of polyethylene films in high
 501 density form is reported by Ojeda et al by natural weathering
 502 in 161 days; this can be achieved in 60 days by bio-surfactant
 503 initiated oxidation process²⁶. Mostly natural weathering or
 504 accelerated weathering is used for oxidation of polyethylene.
 505 If polyethylene is mixed with pro-oxidant additives, then the
 506 oxidation process by this method is very effective and fast.
 507 But, in case of polyethylene films without pro-oxidant or
 508 commercial polyethylene with added antioxidant, oxidation by
 509 this process can take 9 months to 1 year. Although pro-
 510 oxidant initiates the process of oxidation very fast, but this
 511 process is not that much economical. On the other hand, bio-
 512 surfactants are environmental friendly and more effective than
 513 above mentioned processes. In addition, it can easily be
 514 isolated after polyethylene treatment and can be used for any
 515 other application. Also solubilisation of oxidation products

516 into aqueous medium can further be resulted into degradation
 517 process with weight loss if observed for long duration.

518

519 4. Conclusion:

520 Continuous production of bio-surfactant by
 521 *B.licheniformis* in the presence of polyethylene proved to be
 522 an effective process of oxidation of polyethylene. The
 523 presence of lower concentration of NaCl in ypd growth
 524 medium not only stabilized the bio-surfactant produced but
 525 also enhanced its activity as observed through higher level of
 526 oxidation of polyethylene in the presence of lower
 527 concentration of NaCl. In case of pre-oxidized polyethylene,
 528 oxidization and solubilisation of oxidation products into
 529 aqueous medium was also observed.

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