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1 2	Aerosol Assisted Chemical Vapour Deposition of a ZrO ₂ -TiO ₂ Composite Thin Film with Enhanced Photocatalytic Activity
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12	Abstract
13	ZrO ₂ -TiO ₂ composite thin films were deposited by aerosol assisted chemical vapour
14	deposition onto a glass substrate at 450 °C and then annealed at 600 °C. For comparison ZrO ₂
15	thin films and TiO_2 thin films were deposited under the same conditions. X-ray diffraction
16	(XRD), scanning electron microscopy (SEM), energy dispersive X-ray spectroscopy (EDX),
17	X-ray photoelectron spectroscopy (XPS), UV-Vis and Raman spectroscopy were used to
18	characterize all films. Photocatayltic activities were tested by degradation of intelligent ink
19	containing resazurin redox dye under UVA irradiation. The formal quantum efficiency (FQE)
20	and yield (FQY) for the ZrO_2 .Ti O_2 composite thin films were determined as 2.03 x 10^{-3} dye
21	molecules per incident photon and 4.91×10^{-3} dye molecules per absorbed photon
22	respectively. Surprisingly the $ZrO_2 - TiO_2$ composite was a more efficient photocatalyst than

the comparable TiO₂ coating.

26 **1. Introduction**

Titanium dioxide (TiO₂) is a well-known and highly versatile semiconducting material with applications in gas-sensing, photovoltaics and anti-microbial devices.¹⁻⁵ It is also the best established and most widely used photocatalyst due to its relatively high photo-activity compared other similar semiconductors (ZnO) and its chemical stability.^{6, 7} It has been commercialised for use in self-cleaning windows and the current market worth over £ 1.5 billion at the distributor level.

33 TiO_2 exists in three common phases - anatase, rutile and brookite. The anatase phase is considered the most photocatalytically active due to its more efficient transport of bulk 34 excitons to the surface and the indirect bandgap that enables longer exciton lifetimes.^{4, 8} 35 However, the wide bandgap of anatase TiO_2 means that it is transparent to visible light and 36 37 hence much of the solar spectrum is not absorbed, resulting in sub-optimum photocatalytic activity.^{3, 9, 10} Efforts to enhance the photoactivity of TiO₂ involve doping with anionic 38 species such as fluorine, nitrogen and sulphur to decrease the bandgap to the visible region.³, 39 ^{10, 11} Furthermore, photocatalytic activity in anatase is also hindered by the recombination of 40 excitons before they migrate to the surface where they can carry out useful work.¹ 41 Recombination can be reduced by composite formation with other metal oxides that act as 42 electron or hole sinks due to favourable band positions.¹²⁻¹⁴ Furthermore, composite 43 formation results in surface hydroxyl groups that accept holes after irradiation and enable 44 spatial separation of charger carriers and hence reduce recombination.^{12, 15} Composite 45 formation also enhances crystallinity and increases surface area, two properties that are 46 important for high photocatalytic activity.¹⁶ 47

Zirconium oxide, ZrO₂, is an important material with applications in fuel cells¹⁷⁻¹⁹, gas 48 sensors²⁰, optical-dielectrics²¹ and antimicrobial coatings²². It is generally synthesised *via* sol-49 gel^{23, 24}, hydrothermal²⁵ and solid-state reactions. Thin films have also been produced *via* 50 metal-organic CVD^{26, 27} and pulsed laser deposition^{28, 29}. ZrO₂ crystallises in the monoclinic, 51 tetragonal or cubic structure depending on the temperature. Monoclinic ZrO₂ that is observed 52 under ambient conditions converts to the tetragonal form above 1150 °C and above 2300 °C 53 54 to the cubic phase. However there are literature reports of lower temperature cubic ZrO_2 synthesis that involve the use dopant amounts of yttrium or calcium as stabilisers. 55 Furthermore, Prakashbabu et at. have shown the formation of cubic ZrO₂ nanopowders at 400 56 ^oC without the need for a stabilising agent.³⁰ 57

Composite ZrO₂-TiO₂ material has applications as catalysts/catalyst supports in many 58 59 reactions, gas sensors and photoconductive thin films. Synergistic effects arising coupling of ZrO_2 with TiO_2 is known to enhance the latters photocatalytic properties. Furthermore, it has 60 been previously noted that ZrO₂-TiO₂ composites (as well as other binary oxide systems) 61 have an increased surface acidity and hence reactivity compared to pure TiO2.³¹ Increased 62 surface acidity in such systems is thought to derive from greater amounts of surface hydroxyl 63 64 groups. These groups trap photo-induced holes and enhance photocatalytic activity by 65 reducing electron-hole pair recombination. Furthermore, as the hydroxyl groups are 66 concentrated on the surface, the holes are trapped near the surface hence it allows efficient oxidation of pollutant molecules directly or *via* the formation of hydroxyl radicals.^{31, 32} 67

In this paper we report the first preparation of ZrO₂-TiO₂ composite thin films via aerosol 68 69 assisted chemical vapour deposition (AACVD). AACVD is a simple and easily tuneable 70 CVD technique that has been employed to fabricate photocatalytic, optoelectronic and photocoltaic films.³³⁻³⁸ In AACVD the precursors are dissolved in a suitable solvent, then the 71 solution is atomised and transported into the deposition chamber using a carrier gas. 72 73 Compared to other solution based methods like sol-gel techniques (which is often a multistep 74 process) where coating large areas can be an issue, AACVD is a single step technique that is easily scalable.^{39, 40} 75

The composite film was made along with pure ZrO_2 and TiO_2 under the same conditions for comparison. The photocatalytic properties of the films under UVA irradiation showed the composite to have superior activity.

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2. Experimental

81 **2.1. Materials**

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Zirconium acetylacetonate [Zr(acac)₄] and titanium isopropoxide [Ti(OCH(CH₃)₂)₄] were
purchased from Sigma-Aldrich Chemical Co; absolute ethanol from Merck Chemicals. All
chemicals were used as received.

85 2.2. Precursors solution for AACVD

Three types of films were prepared, ZrO_2 -TiO₂ composite, ZrO_2 and TiO₂. For the composite, (1 mmol = 0.48 g) of [$Zr(acac)_4$] was dissolved into 40 ml of absolute ethanol and (1 mmol \approx 0.30 ml) of [Ti(OCH(CH_3)_2)_4] was added to [$Zr(acac)_4$] and left to stir 30 minutes before use by AACVD. For the ZrO_2 film, (1 mmol = 0.48 g) of [$Zr(acac)_4$] was dissolved into 40 ml of absolute ethanol and added to a humidifier flask (100 mL). For TiO₂, \approx 0.30 ml of [Ti(OCH(CH_3)_2)_4] was dissolved at 40 ml of absolute ethanol then added to the humidifier flask (100 mL).

93 2.3. Aerosol assisted chemical vapour deposition (AACVD)

94 Aerosol assisted chemical vapour deposition (AACVD) was used to deposition the films. The reactor contained a carbon block, containing a Whatmann cartridge heater. A Pt-Rh 95 thermocouple was used to control the temperature on the substrate. The reactor has top and 96 bottom plates for deposition and the top plate was placed 8 mm above the substrate. The 97 aerosol mist was generated by a PicoHealthTM ultrasonic humidifier at room temperature. The 98 deposition was carried out on SiO₂ coated float-glass, this prevented diffusion of ions from 99 100 the glass into the film as it acted as a blocking layer. Prior to use the glass was washed with water, acetone and isopropanol and allowed to dry in the oven at 100 °C. Depositions were 101 102 carried out in a cold-walled horizontal-bed CVD reactor at 450 °C. The deposition was 103 conducted by generation of an aerosol and using nitrogen (flow rate 1.4 L min⁻) to drive the 104 aerosol into reactor, this took about 40-45 minutes for all the precursors to transfer. When the 105 deposition was completed, the heat was turned off and the glass was left to cool under 106 nitrogen to room temperature. The glass substrates were handled in air and annealed at 600 107 °C for one hour.

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110 **2.4. Films characterization**

111 XRD data were collected using a microfocus Bruker GAADS powder X-ray diffractometer 112 using a monochromated Cu Ka radiation. (XPS) X-ray photoelectron spectroscopy was 113 carried out using a Thermo Scientific K-Alpha instrument with monochromatic Al-K α 114 source. Raman spectroscopy was obtained from a Renishaw 1000 spectrometer equipped with a 514.5 nm laser. SEM images were carried out on a JEOL 6301F instrument with 115 116 acceleration voltage of 5 kV. Samples were prepared by cutting to 10 mm \times 10 mm and then coated with gold in order to avoid charging. UV-Vis spectroscopy was carried out using both 117 118 a Lamda 25 and 950 instruments. Water droplet contact angles were carried out using an 119 FTA-1000 drop shape instrument. A Fujifilm Finepix HS25 EXR camera captured image at 120 1000 frames per second.

121 **2.5.** Photocatalytic test

122 Prior to photocatalytic testing the samples were cleaned with distilled water, rinsed in 123 isopropanol and acetone and placed under UVC light to irradiate for 30 minutes to activate 124 the surface. The photocatalyic activity was measuring using a resazurin based "intelligent ink" devised by Mills and *et al.*⁴¹ Surfaces of the films were covered evenly with the dye. 125 Photocatalytic reduction of the dye by UVA light was measured by UV-Vis spectroscopy. 126 The formal quantum efficiency (FQE) was calculated by dividing the rate of dye molecules 127 128 destroyed by the photon flux. The formal quantum yield (FQY) was calculated as a UVX 129 radiometer with a detector for 365 nm radiation was used to measure the photon flux and 130 photon absorption for each film.

131 **2.6.** Synthesis of 'intelligent ink'

Resazurin (92%), hydroxy ethyl cellulose and Glycerol (99.6%) were all purchased fromSigma-Aldrich Chemical Co.

40 mg of resazurin redox dye added to 40 ml of aqueous solution and added 3 g of glycerol,
0.45 g of d hydroxylethyl cellulose. That was placed one day in the fridge between 3-5 °C for
one day.⁴¹

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139 **3. Result and discussion**

Aerosol assisted chemical vapour deposition (AACVD) was used to deposit TiO₂, ZrO₂ and 140 ZrO₂-TiO₂ composite films on glass substrates at 450 °C. The samples were then annealed at 141 600 °C for one hour. The pure TiO₂ and ZrO₂ films were deposited from 142 [Ti(OCH(CH₃)₂]₄/ethanol and [Zr(acac)4]/ethanol solution respectively. The composite ZrO₂-143 144 TiO_2 films were synthesised from the one-pot molar equivalent ethanol solution of $[Zr(acac)_4]$ and [Ti(OCH(CH₃)₂]₄. All films were translucent with interference fringes indicating a 145 variation in thickness along the substrate. The films also passed the ScotchTM tape test and 146 147 were indefinitely stable under visible light and in air.

148 **3.1 Material characterisation**

A range of techniques were used to characterize and understand the material and functional
 properties of the ZrO₂-TiO₂ thin films.

151 **3.1.1 X-ray diffraction**

152 Figure 1 shows the XRD patterns of the ZrO₂, TiO₂ and ZrO₂-TiO₂ composite films are deposited by AACVD along with the standard patterns of ZrO₂ and TiO₂. The ZrO₂ films 153 surprisingly match the high temperature cubic phase with reflections for the (111), (200), 154 155 (220) and (311) planes at 2θ values of 30.3, 35.3, 50.6 and 60.3° respectively. Although the formation of the cubic phase below 2300 °C is not common without the use of stabilisers 156 there are sporadic literature examples of cubic ZrO₂ formation at temperatures as low as 400 157 °C.³⁰ The TiO₂ film was pure anatase with reflection at (101), (112), (200) and (204) at 2θ 158 values of 25.3, 38.6, 48.0, 55.1 and 62.8°. XRD pattern for ZrO2-TiO2 composite films 159 160 showed a dual phase system corresponding to cubic ZrO₂ with the (111) at 30.9 2θ and TiO₂ 161 anatase with reflections corresponding to the expected (101), (200) and (211) planes.

162 Applying the Debye - Scherrer formula to the diffraction data showed that the TiO_2 film was 163 made up of crystallites almost twice and four times the size as those found in the ZrO_2 and 164 $ZrO_2 - TiO_2$ composite films respectively (see supporting information).⁴²

Furthermore, the peaks from the composite film show shifts towards lower 2θ values for the anatase phase and to higher 2θ for the cubic ZrO₂ phase indicating some formation of a solid solution. The shift to lower 2θ values of the anatase phase indicates an expansion of the TiO₂ unit cell due to the substitutional replacement of Ti⁴⁺ (ionic radii 0.60 Å) with the larger Zr⁴⁺

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(ionic radii 0.72 Å) ions. This has previously been observed (via both XRD and EXAFS 169 experiments) for Zr doped anatase TiO2.^{43, 44} Conversely the peaks corresponding to the ZrO2

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cubic phase are shifted to higher 2θ values, which is evident of a contraction in the ZrO₂ unit 171

cell. There is further evidence for solid solution formation in the composite ZrO₂-TiO₂ film 172

173 from the Raman data shown below.



Figure 1: Shows XRD for a) ZrO_2 and TiO_2 and b) ZrO_2 - TiO_2 composite thin films grown by AACVD from

177 $[Ti(OPr^{i})_{4}]$ and $[Zr(acac)_{4}]$ in ethanol carrier solvent at 450 °C followed by annealed at 600 °C for 1 hour.

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179 **3.1.2 Raman spectroscopy**

180 Raman spectra for TiO₂, and ZrO₂-TiO₂ composite thin films are shown in Figure 2. The spectrum for the anatase TiO₂ (figure 3) film showed peaks at 143 cm⁻¹ for (E_g, strong peak), 181 198 cm⁻¹ for (E_g , weak peak), 396 cm⁻¹ for (B_{1g}), 517 cm⁻¹ for (A_{1g} , B_{1g}) and 638 cm⁻¹ for (E_g) 182 matching well with literature values.⁴⁵ The ZrO₂-TiO₂ composite thin film (Figure 2) gave a 183 spectrum corresponding to the anatase phase of TiO₂ again with peaks at 138, 389, 514 and 184 632 cm⁻¹ Interestingly, the principal Eg peak in the ZrO₂-TiO₂ composite film was blue 185 shifted to 138 cm⁻¹ corresponding to an expansion in the TiO₂ unit cell due to some 186 substitutional doping of Ti⁴⁺ with Zr⁴⁺ in the TiO₂ matrix as also seen from the XRD analysis 187 (Figure 1).46 188









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3.1.3 X-ray photoelectron spectroscopy (XPS)

194 XPS was preformed on the ZrO₂-TiO₂ composite film to determine the oxidation states of the Zr and Ti (Figure 3). The Zr $3d_{5/2}$ peak appeared at 182.5 eV with Ti $2p_{3/2}$ peak at 459.2 eV 195 corresponding to Zr and Ti in the 4+ oxidation state as expected.⁴⁴ There were two oxygen 196 197 environments detected that matched chemisorbed oxygen (at 531.4 eV) and oxygen bound to Zr and Ti (529.5 eV) (See supporting information).⁴⁷ XPS also showed the ratio of Zr : Ti on 198 the surface of the composite film was 1 : 4 and bulk analysis using energy dispersive X-ray 199 200 spectroscopy (EDX) found the overall ratio to be 1 : 2.7 therefore showing that the composite 201 film was rich in Ti with surface segregation of Ti also occurring. This was in spite of 202 equimolar quantities of both precursors being used in the AACVD system hence it suggest 203 that the $[Ti(OCH(CH_3)_2)_4]$ is a more efficient CVD precursor at the deposition temperature 204 $(500 \ ^{\circ}C)$ compared to $[Zr(acac)_4]$. This has been previously observed for other related Ti containing composite systems grown via AACVD.¹³ 205



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Figure 3: XPS spectra of ZrO₂-TiO₂ composite thin film showing the Ti 2p_{3/2} and 2p_{1/2} transitions (above) and Zr 3d_{5/2} and 3d_{3/2} transitions (below).

211 **3.1.4 Scanning electron microscopy (SEM)**

212 The mophology of the AACVD deposited thin films (Figure 4) was analysed by scanning 213 electron microspy (SEM). The morphology of the ZrO₂ film consists of domes ranging from 214 100 to 500 nm in diameter on an otherwise flat surface (Figure 4a). The pure TiO_2 film 215 however is made up of a dense array of facets that are on average 400 nm long (Figure 4b). 216 Composite formation of these two system results in a surface morphology made up of densly 217 packed partilees (ca. 50 nm wide) and like in the pure ZrO₂ film, the composite film also 218 contains domes that are almost a micron wide that appear cracked (Figure 4c). Side on 219 images (Figure 4d-f) indicate the thickness of the ZrO₂ and TiO₂ films to be ca. 250 nm 220 whereas the composite film was almost 500 nm as expected.

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Figure 4: SEM images of the a) ZrO₂, b) TiO₂ and c) ZrO₂ – TiO₂ composite films with the high magnification
 images inset. The side on images - d) ZrO₂, e) TiO₂ and f) ZrO₂ - TiO₂ composite – show the film thickness.

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227 3.1.5 UV-visible measurements (UV-Vis)

228 Figure 4 illustrates UV-Vis transmission spectrum for TiO₂, ZrO₂ and ZrO₂-TiO₂ composite 229 thin films. All films show interference fringes that are expected for ZrO₂ and TiO₂ based films. The transparency of the ZrO₂-TiO₂ composite thin film is between 60 and 70% in the 230 231 visible region (380–760 nm) and near infrared region (760–2500 nm). This is slightly lower 232 than the transparency values obtained for pure ZrO_2 and TiO_2 films, which were between 80 233 and 90% again in both the visible and infrared regions. Reflectance measurements shown in 234 Figure 5 (b) show that the composite film has higher reflectance (between 35 and 45%) than 235 the pure films, which is at 10% and 25% for ZrO₂ and TiO₂ respectively.

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Figure 5: Illustrates a) transmittance and b) reflectance spectra of ZrO₂, TiO₂ and ZrO₂-TiO₂ composite thin
 films

The indirect optical bandgaps of the TiO₂ and ZrO₂-TiO₂ composite thin films were calculated using the Tauc plot (see supporting information). The anatase TiO₂ film gave a bandgap of 3.2 eV matching literature values whereas the determination of the ZrO₂ film's bandgap was not possible due to absorption from the glass substrate in the UV-Vis spectrum. For the composite film a bang gap of 3.3 eV was calculated owing to the anatase component of the film.

247 **3.2 Functional testing**

248 3.2.1 Photocatalysis testing

249 The photocatalytic study of the TiO₂ and ZrO₂-TiO₂ thin films were determined under the 250 same condition by the resazurin dye (intelligent ink) under 365 nm UV light irradiation. The 251 photoreduction reaction was induced by UVA radiation (flux = 3.67×10^{14} photons per cm² per s) and observed by UV-Vis spectroscopy. The photocatalytic activity of the pure ZrO₂ 252 was not tested due to the inability of UVA radiation to activate the ZrO₂ due to its wide 253 bandgap. The rate of dye degradation on the TiO₂ film was 3.3 x 10^{-4} dye molecules cm⁻² s ⁻¹ 254 taking for the resazurin that is a royal blue colour to be photoreduced to the pink resorufin. 255 The composite film showed a degradation rate an order of magnitude better $(2.0 \times 10^{-3} \text{ dye})$ 256 molecules cm^{-2} s⁻¹). 257

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Fig

Figure 8: Shows the degradation of resazurin redox dye on the ZrO₂-TiO₂composite thin films by UVA irradiation.

261 The formal quantum efficiencies (FQE) and formal quantum yields (FQY) for the TiO_2 and 262 ZrO₂-TiO₂ composite films were calculated by taking into account the UVA photon flux on 263 the films and the number of photons absorbed by the films. Figure 9 shows FQE and FQY for the pure TiO₂ and composite films. The FQE for the TiO₂ was 3.3×10^{-4} dye molecules 264 destroyed / incident photon and the ZrO2-TiO2 composite film was an order of magnitude 265 better at 2 x 10⁻³ dye molecules destroyed / incident photon. The FQY calculated for both 266 films also showed that the composite film $(4.9 \times 10^{-3} \text{ dye molecules destroyed / absorbed})$ 267 photon) was again an order of magnitude better than the pure TiO₂ film (5.1 x 10^{-4} dye 268 269 molecules destroyed / absorbed photon). The enhanced photocatalytic activity of the composite film compared to the pure TiO₂ film is most likely due to the greater concentration 270 of hydroxyl groups found on the composite surface that amplify surface acidity and reactivity 271 towards pollutants.³¹ Furthermore, the superior performance of the ZrO₂-TiO₂ composite in 272 the degradation of resasurin is also helped by the supressed charge carrier recombinations as 273 the surface hydroxyl groups act as hole traps.^{31, 32} 274



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Figure 9: Illustrates the formal quantum efficiency (FQE) and the formal quantum yield (FQY) of the UVA
 induced photocatalytic reduction of resazurin redox dye for ZrO₂-TiO₂ composite and TiO₂ thin films

279 3.2.2 Photoinduced wettability

The water contact angle for ZrO_2 , TiO_2 and ZrO_2 - TiO_2 composite thin films was assessed before and after UVA radiation (flux= 3.67×10^{14} photons per cm² per s) for 16 hours by 5 µL droplet of water on the films (see supporting information). There was no significant change in the water contact angle from pre and post irradiation. This is expected for pure ZrO_2 as UVA irradiation is too weak to induce wettability, however literature examples show TiO₂ and ZrO_2 -TiO₂ films to be readily activated by UVA resulting in a significant decrease in the post irradiation water contact angle.⁴⁸

287 4 Conclusions

In this paper we have shown the novel use of AACVD to synthesis a $ZrO_2 -TiO_2$ composite thin film. The composite film that shows enhanced photocatalytic activity under UVA activation compared to a TiO₂ film (deposited under the same conditions). The enhancement is believed to result from synergistic effects such as improved surface acidity. Further work on the effects of temperature and ratios of TiO₂ and ZrO₂ on the photocatalytic properties are of interest.

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