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1 **Thermal decomposition and desorption of PFPE Zdol on a DLC**
2 **substrate using quartic bond interaction potential**

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8

9 **Abstract**

10 At heat assisted magnetic recording (HAMR) system heating of the hard disk magnetic
11 layer is carried out by applying laser rays during the movement of the head on the
12 carbon overcoat for the purpose of reading and writing on its magnetic layer. Depletion
13 of PFPE Zdol occurs for the cause of its thermal decomposition and desorption on a
14 DLC substrate due to laser heating and this model is developed using the coarse-grained
15 bead spring based on quartic and Van der Waals interaction potential. Effect of
16 temperature on the bond breaking phenomenon of PFPE Zdol due to thermal
17 decomposition and thermal desorption are studied. To support the reliability of the
18 present simulation results by quartic potential, the end bead density and total bead
19 density on a DLC substrate obtained by the finitely extensible non-linear elastic (FENE)
20 and quartic potential are shown as a comparative manner.

21

22 **Keyword:** Thermal decomposition, thermal desorption, PFPE Zdol, quartic potential

23 **1. Introduction**

24 In MEMS/NEMS, PFPE films are used as permanent lubricant to protect these
25 structures from erosion [1-2]. Hard disk industry uses thin PFPEs (with a thickness of
26 1-2nm) deposited onto amorphous carbon overcoat to protect underlying hard disks
27 from mechanical damage during intermittent contacts with the read-write heads as the
28 disk rotates. Conventional reading-writing technique to increase the memory of a hard
29 disk has a limitation to reach above 1TB/inch². But the only way is heat-assisted
30 magnetic recording technology is the alternative way to reach the memory up to 10
31 TB/inch². But the main difficulties are the stability of lubricants on the hard disk is a
32 great concern to the scientific community. Ganesh et al. [3] showed the successful
33 fabrication of a thin, transparent, and homogenous coating of perfluoropolyether (PFPE)
34 on a smooth glass surface by electro spraying technique. The wetting behavior of three

1 nanometer thick perfluoropolyether (PFPE) polymers with the same backbone but
2 different end-groups were studied by contact angle tests and simultaneous governing
3 mechanisms like oleophobicity/hydrophilicity were investigated [4]. Li and Wong [5]
4 carried out molecular dynamics simulations to study the thermal stability of thin
5 lubricant films. Lubricant film thickness, bonding ratio, molecular weight and laser
6 heating rate play important roles to deplete lubricant films [6-9]. Wu [8] described a
7 model for lubricant flow dynamics under a scanning laser beam and studied the
8 mechanisms of lubricant on both glass and aluminum disks. The main causes of the
9 depletion of the lubricant film on a hard disk surface are thermal decomposition and
10 desorption but the mechanisms of thermal decomposition and desorption are not
11 understood either by theoretical or experimental way till now [9]. Magnetic layer of a
12 hard disk is heated above its Curie temperature in heat-assisted magnetic recording
13 systems [10]. Pacansky and Waltman [11] reported that PFPEs degrade due to main
14 chain scission when subjected to electron irradiation. Heller et al.[12] investigated the
15 effect of localized laser heating on molecularly thin PFPE lubricants for short times.
16 Zhu et al.[13] used mass spectroscopy to study the thermal chemistry of laser-irradiated
17 PFPEs on disk media. It was found that there is a stronger likelihood of laser-induced
18 scission and rearrangement of the PFPE backbone chain than that of its end group with
19 the laser power density being in a range 60-140 mJ/cm². Li et al.[14] employed
20 molecular dynamics (MD) simulation to investigate the evolution and degradation
21 characteristics of ultrathin PFPEs under laser irradiation. They observed that the
22 lubricant diffused radially away from the heating spot and raised ridge was formed
23 around the depleted zone due to non-uniform distribution of surface tension and thermo
24 capillary stress. Waltman and Tyndall [15-16] measured the evaporation rates of both
25 polydisperse and monodisperse PFPEs from a CH_x surface at various temperature. Lim
26 and Gellman [9] investigated the kinetics of laser induced desorption and decomposition
27 of Fomblin Zdol on carbon overcoats. They observed that under HAMR conditions
28 lubricant films with molecular weight of less than 3000 g/mole desorb more than that of
29 decomposition on the disk surface during rapid heating, while decomposition is
30 performed at temperatures (<400⁰C) expected in HAMR for higher molecular weighs
31 lubricants. The decomposition behavior of Fomblin Zdol and Fomblin Z on amorphous
32 carbon overcoat using thermal desorption spectroscopy [17], the decomposition
33 mechanisms of PFPE Zdol at the head disk interface under sliding conditions [18],
34 desorption mechanisms of Fomblin Zdol on graphite surface [19], thermal stability of
35 Fomblin Z on Al₂O₃ surface [20], thermal decomposition of Fomblin Zdol on graphite
36 and amorphous carbon surfaces [21], thermal and electron-induced decomposition of

1 Fomblin Zdol on a carbon overcoat [22], the desorption and decomposition of
2 polyperfluorinated ether using transmission infrared spectroscopy [23] were studied
3 experimentally and reported. Lei et al. [17] observed that the decomposition of Fomblin
4 Zdol on amorphous carbon surface occurs in the range of temperature 600-700K and
5 they suggest that desorption is the results of the decomposition of Fomblin Zdol. Kasai
6 et al.[20] observed during the experiment that Fomblin Z on Al₂O₃ surface degrades at
7 473K at two stages and the vigorous second stage leads to complete loss of fluid.
8 Vurens and Mate [21] concluded from their study on Fomblin Zol that the two
9 desorption features of Fomblin Zdol on graphite surface happens at temperatures 620
10 and 770K; on the other hand the main desorption features of Fomblin Zdol on
11 amorphous carbon surface are at the temperature 510 and 650K.

12 Tagewa et al. [24-25] carried out a series of experiments to investigate the lubricant
13 depletion characteristics of HAMR systems. They found that the lubricant depletion due
14 to laser heating is largely affected by film thickness, bonding ratio, molecular weight
15 and end groups of lubricant. Ma et al.[26] studied the lubricant depletion on a
16 self-developed HAMR tester and found that the lubricant depletion depth is proportional
17 to the logarithm of laser heating duration. Moreover, they suggested that almost all the
18 lubricant on the disk would be depleted over the lifetime of the hard disk drive. Bei et
19 al.[39] employed molecular dynamics simulation coupled with a course-grained
20 bead-spring polymer model to study the depletion of PFPE lubricants. Their study
21 shows that desorption of PFPE lubricants is favored over decomposition at high
22 temperature. The photo-thermal stability and tribological properties of Zdol lubricant,
23 modified by substituting the OH groups with benzophenone, were investigated using
24 laser beam exposure by Gauvin et al.[27] and they concluded that their modified Zdol
25 lubricant is suitable for heat assistant magnetic recording systems.

26

27

28 From the above study, it is observed that desorption and decomposition rate of
29 functional lubricant film on a thin DLC film using a bond-breaking potential are not
30 available. In the present study we study desorption and decomposition of PFPE Zdol on
31 a thin DLC film considering interaction of all carbon atoms with the lubricant film. New
32 potential quartic is used instead of FENE potential which allows bond-breaking. To
33 satisfy the current results obtained by quartic potential, at low temperature, end bead
34 density and total bead density with the film thickness are computed by using quartic and
35 FENE potential and are compared to each other. To study the validity of the present
36 simulation results, the number density as a function of the film thickness, the

1 decomposition and desorption rate with the film thickness are compared with the
2 published results. Effects of temperature on the desorption and decomposition rate of
3 PFPE Zdol are investigated in details. With the graphical results, some snapshots during
4 the depletion of lubricants are described to understand the physics of lubricant on a
5 DLC substrate during heating.

7 **2. Simulation details**

8 Figure 1 shows the interaction between the beads of lubricant molecules and DLC
9 substrate. DLC substrate consists of carbon atoms. DLC substrate is prepared
10 subsequent heating and quenching of BCC or FCC diamond crystals by molecular
11 dynamics simulations following the procedures [28-31] using Tersoff potential [32-33].
12 DLC substrate obtained from atomistic simulation is compressed half of its original
13 dimension so that no lubricant molecules can diffuse inside the DLC substrate during
14 simulation [34-35]. Each lubricant molecule of PFPE Zdol is constructed considering
15 bead spring model. Each beads consists of several PFPE Zdol atoms. Each lubricant
16 molecules consists of 10 beads and among them end beads are considered as polar or
17 functional beads and the rest of the beads are considered as backbone beads. The
18 neighboring beads in each chain are connected by bonds which is modeled using quartic
19 potential and this bond can break anytime if the bond exceeds its allowable extension
20 limit. Practically DLC substrate consists of many dangling bonds and for this reason the
21 functional beads of lubricant molecules are attracted more than the rest of the lubricant
22 molecules [36-37]. Besides, the interaction between the functional beads is more than
23 that of non-functional beads or that of functional and non-functional beads. So for this
24 reason, a special treatment is done between the interaction of functional beads to DLC
25 substrate and among functional beads. The detailed description of the potentials and
26 their interaction are described in the theoretical section using Fig.1.

27
28 900 lubricant molecules are considered in the present study. 900 lubricant molecules are
29 kept on a DLC substrate and simulated for long time by the subsequent isothermal
30 heating to obtain the initial configuration of PFPE Zdol film which is shown in Fig.2
31 [34-35]. After getting an initial configuration of PFPE Zdol film on a DLC substrate,
32 three dimensional simulation is considered considering periodic boundary conditions in
33 x and y directions and in the z direction shrink-wrapped boundary conditions [38] are
34 used because during the simulation the simulation box dimension alters in z direction.
35 Used time step during the simulation 0.005τ . During simulation the positions of carbon
36 atoms in a DLC substrate is kept fixed. Isothermal heating is carried out on the

1 molecules of PFPE Zdol lubricant molecules on a DLC substrate at different
 2 temperature separately. During simulations at high temperature bond breaking happens
 3 and new short chains are generated which enhance the thermal decomposition and
 4 desorption. The simulations are carried out separately at different isothermal heating for
 5 2000,000 steps for example $T^*=1$ to 6 using NVT (N=same number of atoms,
 6 V=volume, T=temperature) canonical ensemble, and we calculate how many bonds
 7 break and how many lubricant molecules washed away by desorption phenomenon
 8 during the simulation at specific temperature at last time step. Besides, gradual heating
 9 effects are also considered heating the lubricant molecules from $T^*=1$ to 6 at slow rate
 10 at time step 2000,000 using the NVT canonical ensemble at the same simulation. The
 11 simulation is rescaled at every 20 steps. Thermal decomposition and thermal desorption
 12 of PFPE Zdol on a DLC substrate is calculated for each case at a time step 2000,000. To
 13 verify the soundness of the present simulation results, PFPE Zdol lubricant molecules
 14 on a DLC substrate is simulated using FENE and Quartic potentials for bonding
 15 between beads of PFPE Zdol molecules, separately keeping the other potential
 16 parameter same at a very low temperature $T^*=1.3$.

17

18 **3. Theories of functional lubricant on a thin DLC film**

19 To describe the bond interaction between the neighboring beads in each lubricant
 20 molecule, FENE potential [39-40] is used in the coarse-grained bead spring model. As
 21 the FENE potential is unable to bond breakage during the simulation, quartic potential
 22 [41-42] is used instead of FENE potential [39-40].

$$23 \quad U_4 = \begin{cases} k_4(y - b_1)(y - b_2)y^2 + U_0 & r < r_b \\ U_0 & r > r_b \end{cases} \quad (1)$$

24 Where r is the distance between two neighboring beads, $y = r - r_b$ shifts the quartic center
 25 from the origin and r_b is the cutoff length where the potential goes smoothly to a local
 26 maximum. When the bond breaks and the quartic term is deactivated to impede the
 27 bond from reforming. This potential mimics the FENE potential by setting the following
 28 parameters as

$$29 \quad k_4 = 1434.3\epsilon/\sigma^4, b_1 = -0.7589\sigma, b_2 = 0.0, r_b = 1.5\sigma, \text{ and } U_0 = 67.223\epsilon [43-44]$$

30 The expression of non-bonded potential LJ 12/6 is

$$U_{LJ12/6} = 4\epsilon \left[\left(\frac{\sigma}{r} \right)^{12} - \left(\frac{\sigma}{r} \right)^6 \right] \quad (2)$$

Here σ is the L-J diameter of the beads and ϵ is the potential depth.

The equation (2) is used for the interaction between non-functional beads; nonfunctional bead to functional bead of PFPE Zdol. As dangling bonds in the DLC film are available, polar beads attract toward the DLC surface to form bond with hydrogen from their OH group and form an ionic bond with the DLC surface. Therefore, an extra attractive potential is added with the van der Waals interactions to satisfy the conditions [44]. For the extra interaction due to the functional effects of the end beads of functional lubricant PFPE Zdol [34-35, 39-40, 44] used expression of the extra interaction between them by an exponential potential is

$$U_{\text{EXP1}} = -\epsilon_p \left(-\frac{r-r_c}{d} \right) \quad (3)$$

For the extra interaction due to the functional effects of the end beads of functional lubricant PFPE Zdol and DLC C atom [34-35,39-40,44] used expression of the extra interaction between them by an exponential potential is

$$U_{\text{EXP2}} = -\epsilon_p^{\text{dlc}} \left(-\frac{r-r_c}{d} \right) \quad (4)$$

Combining non-bonded potential LJ 12/6 and exponential potential for the interaction between the end beads of functional lubricant PFPE Zdol, the expression of the resultant energy between polar end bead to polar end bead [34-35, 39-40,44] is

$$U_{\text{cb-cb}} = U_{LJ12/6} + U_{\text{EXP1}} = 4\epsilon \left[\left(\frac{\sigma}{r} \right)^{12} - \left(\frac{\sigma}{r} \right)^6 \right] - \epsilon_p \left(-\frac{r-r_c}{d} \right) \quad (5)$$

Combining the non-bonded potential LJ 12/6 and exponential potential for the interaction between the end beads to DLC C atoms, the expression of the resultant energy between polar end bead and DLC C atom [34-35, 39-40,44] is

$$U_{\text{eb-dlc}} = U_{\text{LJ12/6}} + U_{\text{EXP2}} = 4\varepsilon \left[\left(\frac{\sigma}{r} \right)^{12} - \left(\frac{\sigma}{r} \right)^6 \right] - \varepsilon_p^{\text{dlc}} \left(-\frac{r-r_c}{d} \right) \quad (6)$$

2

3 The detailed descriptions of the parameters in the above equations are given in the
4 reference [34-35, 39-40, and 44-45].

5

6 4. Results and Discussion

7 From the experimental studies it is observed that depletion of PFPE Zdol lubricant on a
8 DLC substrate occurs at high temperature. The reason of the depletion of PFPE Zdol on
9 a DLC substrate is not clear till now. For this reason, theoretical analysis of the
10 depletion of PFPE Zdol on a DLC substrate is necessary to understand the actual
11 physics and chemistry of PFPE Zdol on a DLC substrate at low to high temperature.
12 Two main reasons of the depletion of PFPE Zdol on a DLC substrate are assumed which
13 are thermal decomposition and thermal desorption. Generally static, kinetic and
14 rheological properties of PFPE Zdol on a DLC substrate can be investigated using
15 coarse-grained bead spring model based on FENE potential but it can not be applied at
16 higher temperature because at higher temperature lubricant bond breaks due to
17 increased its bond length. So for this reason, coarse-grained bead spring model based on
18 quartic potential which allows bond breaking during the simulation at higher
19 temperature. As at a low temperature, $T^* = 1.3$, PFPE Zdol on a DLC substrate is studied
20 using two different potentials such as quartic and FENE keeping the same Van der
21 Waals interaction and extra attractive potential to verify the reliability of quartic
22 potential on the coarse-grained bead spring model. Figure 3(a) illustrates the end bead
23 density of PFPE Zdol on a DLC substrate as a function of film thickness using two
24 different potentials such as quartic and FENE based on coarse-grained bead spring
25 models. Two layers of PFPE Zdol on a DLC films are formed in which the first peak of
26 the end bead density of PFPE Zdol is largely higher than that of second peak for the
27 case of both potentials. In the 1st layer, the end bead density of PFPE Zdol using quartic
28 potential is slightly smaller than that of FENE potential but except the first layer, the
29 end bead density of PFPE Zdol for the cases of both potentials coincide each other.
30 Figure 3(b) illustrates the total bead density of PFPE Zdol on a DLC substrate using two
31 different potentials such as quartic and FENE as a comparative manner. The first peak
32 of the total bead density of PFPE Zdol using quartic potential is higher than that of the

1 first peak of FENE potential but except the first peak, the total bead density of PFPE
2 Zdol on a DLC substrate for both of the potentials are almost same with film thickness.
3 From Figs.3 (a) and 3(b), it is clearly observed that the phase transition of PFPE Zdol
4 originates between the first and second layers. Figure 4 shows the comparative study of
5 number density of PFPE Zdol on a DLC substrate as a function of the film thickness
6 using quartic potential based coarse-grained bead spring model with that of PFPE on a
7 single wall obtained by Bei et al., 2013. The obtained number density of PFPE Zdol on
8 a DLC substrate by us qualitatively agree well with that of Bei et al., 2013 and some
9 discrepancies of our result with that of Bei et al., 2013 occurs because Bei et al., 2013
10 used non-functional PFPE lubricant instead of functional PFPE Zdol and single wall
11 instead of DLC substrate in their simulations. Figure 5 illustrates the total bead density
12 of PFPE Zdol on a DLC substrate at higher temperature such as $T^*=3.1, 3.4, 3.5$ and 3.9 .
13 From the this figure, a clear phase transition zone of PFPE Zdol is observed where the
14 liquid phase of PFPE Zdol transforms into a gaseous phase. Besides, from this figure, it
15 is clear that with the increase of temperature, the total bead density not only decreases
16 with the film thickness but also the film thickness of PFPE Zdol on a DLC substrate
17 decreases with the increase of temperature. From figure 5, it is evident that heating of
18 PFPE Zdol on a DLC substrate at a higher temperature is responsible for higher lub
19 depletion than the heating of PFPE Zdol at lower temperature. Figure 6(a) shows the
20 initial configuration of PFPE Zdol on a DLC film at a temperature $T^*=3$ at time step=0.
21 Figure 6(b) shows the final configuration of PFPE Zdol on a DLC substrate during
22 isothermal heating at time =10000 τ . After time 10000 τ due to isothermal heating, the
23 film of PFPE Zdol on a DLC substrate deplete which are clearly observed if is
24 compared the snapshots of PFPE Zdol on a DLC substrate as observed in Fig.6(a) and
25 6(b). At time 10000 τ , it is observed that in few spots on the DLC substrate, PFPE Zdol
26 lubricant is removed due to isothermal heating at temperature $T^*=4.0$.

27

28 Figure 7 illustrates the comparative study of the bond percentage of PFPE Zdol on a
29 DLC substrate with that of non-functional PFPE on a single wall obtained by Bei et al.,
30 2013. Up to temperature $T^*=4$, the bond percentage of PFPE Zdol on a DLC substrate
31 by us quantitatively agree well with that of non-functional PFPE on a single wall
32 obtained by Bei et al., 2013 but the deviation between the bond percentage of PFPE
33 Zdol on a DLC substrate and non-functional lubricant PFPE on a single wall obtained
34 by Bei et al., 2013 increases after temperature $T^*=4$ with the increase of temperature
35 because functionality of PFPE Zdol lubricants and DLC substrate play an important role
36 such types of deviation from that of Bei et al., 2013. For the case of PFPE Zdol

1 lubricants, functional beads form strong bond with the DLC substrate. As the
2 non-functional PFPE has no functional beads, they cannot form strong bond with the
3 carbon overcoat. For this reason, at the high temperature the decomposition rate of
4 PFPE Zdol should be lower than that of non-functional PFPE lubricants and this effect
5 is strongly observed in the comparative study as shown in Fig.7. Figure 8(a) illustrates
6 the bond percentage of PFPE Zdol on a DLC substrate for the temperature range T^*
7 $=1.5$ to 4 . Figure 8(b) illustrates the bond percentage of PFPE Zdol on a DLC substrate
8 as a function of elapsed time during the simulation at the temperature range $T^*=4$ to 6 .
9 From Fig.8(a), at the temperature $T^* =1.5$, the bond percentage of PFPE Zdol is nearly
10 100%, which means that bond of PFPE Zdol on a DLC substrate does not break which
11 ensures that at $T^*=1.5$, no thermal decomposition of PFPE Zdol on a DLC substrate
12 happens at this temperature during the simulation. From Fig.8(a), it is observed that at
13 temperature $T^* =2$, the total bond percentage of PFPE Zdol on a DLC substrate is very
14 high as compared to temperature $T^* =3$, i.e. at temperature $T^* =2$, the rate of bond
15 breaking phenomenon is extremely lower as compared to higher temperature $T^* =3$ or
16 other temperatures. From Figure 8(a) and 8(b), it is evident that with the increase of
17 temperature, bond percentage of PFPE Zdol on a DLC substrate decrease which means
18 the increased temperature enhances the bond breaking phenomenon of PFPE Zdol on a
19 DLC substrate which also increases the thermal decomposition of PFPE Zdol on a DLC
20 substrate. After elapsed time 1000τ , the rate of bond percentage with time during the
21 simulation gradually decreases as before because after elapsed time 1000τ , many beads
22 of PFPE Zdol lubricant exceeds their force cut off distance and many short chains form
23 after 1000τ , the percentage of long chain decreases and for which the increasing rate of
24 bond breaking decreases in the range of elapsed time 1000τ to 10000τ .

25

26 To validate the result of thermal desorption of PFPE Zdol on a DLC substrate obtained
27 by us is compared with that of non-functional PFPE lubricant on a single wall obtained
28 by Bei et al., 2013 at two different temperatures such as $T^*=3$ and 3.3 as shown Fig.9.
29 For the case of temperature $T^*=3$, it is observed that weight percentage of PFPE Zdol
30 on a DLC substrate by us agree well qualitatively with that of non-functional PFPE on a
31 single wall obtained by Bei et al., 2013 but at $T^*=3.3$, the deviation of the weight
32 percentage of PFPE Zdol on a DLC substrate obtained by us and the weight percentage
33 of non-functional PFPE on a single wall obtained by Bei et al., 2013 is quietly larger as
34 compared to temperature $T^*=3$ due to the functionality of PFPE Zdol and DLC
35 substrate. Figure 10(a) illustrates the weight percentage of PFPE Zdol on a DLC
36 substrate as a function of elapsed time for the temperature range $T^* =1.5$ to 4 . Figure

1 10(b) illustrates the weight percentage of PFPE Zdol on a DLC substrate as a function
2 of elapsed time for the temperature range $T^* = 4$ to 6. At temperature $T^* = 1.5$ and 2, the
3 bond percentage of PFPE Zdol on a DLC substrate as a function of elapsed time
4 decreases at constant rate. At temperature $T^* = 1.5$, the thermal desorption of PFPE Zdol
5 is very low which can be negligible because at elapsed time, the weight percentage of
6 PFPE Zdol on a DLC substrate is close to 100%. With the increase of temperature, the
7 weight percentage of PFPE Zdol on a DLC substrate decreases. The desorption rate of
8 PFPE Zdol is several times of thermal decomposition during heating PFPE Zdol on a
9 DLC substrate. Up to elapsed time 1000τ , the desorption rate of PFPE Zdol on a DLC
10 substrate is constant and higher than that of rest of the elapsed time. Thermal
11 decomposition enhances the thermal desorption of PFPE Zdol because during heating
12 many small lubricant chain forms due to bond breaking which increases the thermal
13 desorption. Figure 11(a) illustrates the initial configuration of PFPE Zdol on a DLC
14 substrate which is heated at varying temperature $T^* = 1-6$ during elapsed time 10000τ as
15 observed in Fig. 11. At the final stage, most of the PFPE Zdol molecules are washed
16 away from the DLC substrate due to having higher temperature $T^* = 6$ when it reaches
17 at final stage of heating time 10000τ as shown in Fig.11(b). Some lubricant molecules
18 are attached on the DLC substrate which form strong bond on the DLC substrate. PFPE
19 Zdol lubricant molecules on the DLC substrate form strong bond and are strongly
20 bonded with the DLC substrate and the lubricant molecules on the top surface of
21 lubricant film are non-bonded. During heating, the kinetic energy of each bead increases
22 and for this reason, the bond length increases with the rise of temperature during heating.
23 When the bond length exceeds its allowable limit, bond breaks and new short chains
24 appear on the film. During heating at high temperature, non-bonded lubricants are
25 removed rapidly by desorption and the PFPE Zdol thickness decreases. As the lubricant
26 on the DLC substrate form strong bond with the DLC substrate, all lubricant molecules
27 are not washed away from the DLC substrate as shown in Fig.11 (b). Bond percentage
28 of PFPE Zdol on a DLC substrate is also studied in two different cases as shown in
29 Figs.12(a) and 12(b) such as in the first case, bond percentage of PFPE Zdol is studied
30 as a function of elapsed time during heating temperature $T^* = 1-6$ as shown in Fig.12(a).
31 In the second case, bond percentage of PFPE Zdol is studied as a function of
32 temperature T^* during heating up to elapsed time 10000τ as shown in Fig.12 (b).
33 During heating of PFPE Zdol on a DLC substrate, bond percentage of PFPE Zdol
34 decrease at a very low constant rate up to temperature $T^* = 1$ to 3 or up to elapsed time
35 upto 4000τ during the simulation. After reaching temperature $T^* = 3$ or elapsed time
36 4000τ , the decreasing rate of bond percentage of PFPE Zdol on a DLC substrate

1 increases upto $T^* = 5$ or elapsed time 7000τ and after reaching temperature $T^* = 5$ or
2 elapsed time 7000τ , the bond percentage of PFPE Zdol on a DLC substrate increases at
3 a constant rate to reach the final stage at time 10000τ or temperature $T^* = 6$. Thermal
4 decomposition of PFPE Zdol is dominating in the temperature range $T^* = 4$ to 6 which
5 agree well qualitatively with that of the experimental study of the thermal
6 decomposition of Fomblin Zdol on a carbon surface by Lei et al.[17]. In the theoretical
7 study of non-functional PFPE lubricants [39] and the experimental study of Fomblin
8 Zdol on a carbon surface favors desorption over decomposition. And the similar effects
9 are also observed in our study and from present study of thermal decomposition and
10 desorption of PFPE Zdol on a DLC substrate, desorption of PFPE Zdol favors over the
11 decomposition.
12
13

14 **5. Conclusion**

15 From the comparative study of the end bead density and total bead density of PFPE
16 Zdol on a DLC substrate at a temperature $T^* = 1.3$ obtained by quartic potential and
17 FENE potential agree well each other. But at higher temperature bond breaking
18 phenomenon happens which cannot be solved using FENE potential but quartic
19 potential allows bond breaking phenomenon during heating at high temperature. From
20 the total bead density of PFPE Zdol using quartic potential, clear phase transition zone
21 is observed in where liquid PFPE Zdol turns into gaseous phase. With the increase of
22 temperature, thermal decomposition and thermal desorption of PFPE Zdol on a DLC
23 substrate increases. Thermal desorption is the principle cause of lub desorption. But due
24 to thermal decomposition, many long lubricant chain turn into short chains which
25 enhance thermal desorption of PFPE Zdol lubricant molecules on a DLC substrate.
26

27 **References**

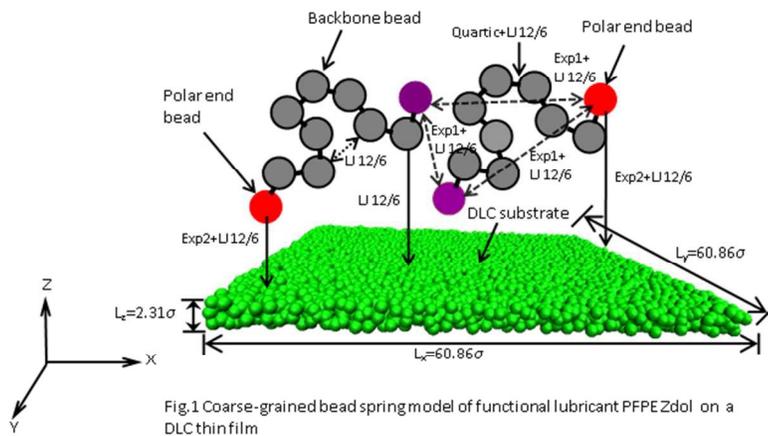
- 28 1. B.Marchon, *IEEE Trans. Magn.*, 2009, **45**,872-876.
- 29 2. Z. Tao and Bhusan, *Wear*, 2015, **259**,1352-1361.
- 30 3.V.A.Ganesh, S.S.Dinachali, S.Jayaraman, R.Sridhar, H.K.Raut, A.Góra, A.Baji,
31 A.S.Nair and S.Ramakrishna, *RSC Advances*, 2014, **4**,55263-55270.
- 32 4.Y.Wang,J.Knapp,A. Legere,J.Raney and L.Li, *RSC Advances*,2015,5,30570-30576
- 33 5. B. Li and C.H.Wong, *Computational Materials Science*, 2014, **93**, 11-14.
- 34 6. N. Tagawa, H. Andoh and H. Tani, *Tribology Letters*, 2010, **37**,411-418.

- 1 7. N.Tagawa, T.Miki and H.Tani, *Tribology Letters*, 2012, **47**,123-129.
- 2 8. L.Wu, *Nanotechnology*, (2007), 18, 215702.
- 3 9. M.S.Lim and A.J.Gellman, *Tribology International*, 2005, **38**,554-561.
- 4 10.W.A.Challener, C.Peng, A.V.Itagi, D.Karns, W.Peng,
- 5 Y.Peng,X.Yang,X.Zhu,N.J.Gokemeijer,Y.-T.Hsia,G.Ju,R.E.Rottmayer,M.A.Seigler and
- 6 E.C.Gage, *Nature Photonics*, 2009, **3** ,220-224.
- 7 11. J.Pacansky and R.J.Waltman, *Chem. Mater.* 1993, **5**,486-494.
- 8 12.J.Hellor, C.M.Mate, C.C.Poon and A.C.Tan, *Langmuir*, 1991, **15**,8282-8287.
- 9 13. L.Zha,T.Liew and T.C.Chang, *Applied Physics A*, 2002,**75**,633-636.
- 10 14. Y.Li, C.H.Wong,B.Li, S.K.Yu, W.Hua, and W. Zhou, *Soft Matter*, 2012,
- 11 8,5649-5657.
- 12 15. R.J.Waltman, G.W.Tyndall, J.Pacansky and R.J.Berry, *Tribology Letters*, 1999,
- 13 **7**,91-102.
- 14 16. G.W.Tyndall, and R.J.Waltman, *J.Phys.Chem. B.*, 2000, **104**,7085-7095.
- 15 17.R.Z.Lei, A.J.Gellman and P. Jones, *Tribology Letters*, 2001, **11**,1-5.
- 16 18. J.We, W.Fong, D.B.Bogy and C.S.Bhatia, *Tribology Letters*, 1998, **5**,203-209.
- 17 19.K.R.Paserba and A.J.Gellman, *J.Phys.Chem B*, 2001,**105**,12105-12110.
- 18 20.P.H.Kasai, W.T.Tang and P.Wheeler, *Applied Surface Science*,1991,**51**,201-211.
- 19 21.G.H.Vurens and C.M.Mate, *Applied Surface Science*,1992,**59**,281-287.
- 20 22.J-L. Lin, C.S. Bhatia and J.T.Yates, Jr, *Journal of Vacuum Science & Technology A*,
- 21 1995,**13**,163-168.
- 22 23.L.M.Ng, E.Lyth, M.V.Zeller and D.L.Boyd, *Langmuir*,1995,**11**,127-135.
- 23 24.N. Tagawa, R.Kakitani, H. Tani, N.Iketani and I.Nakano, *IEEE Trans.*
- 24 *Magn.*,2009,**45**,877-882.
- 25 25.N.Tagawa and H. Tani, *IEEE Trans. Magn.*, 2011,**47**,105-110.
- 26 26.Y.S.Ma, X.Y.Chen and B.Liu, *Tribology Letters*, 2012,**47**,175-182.
- 27 27.M.Gauvin, H.Zhang, B.Suen, J.Lee, H.J.Kang and F.E.Talke, *IEEE Transactions on*
- 28 *Magnetics* ,2011,**47**,1849-1854.
- 29 28.H.Lan, T.Kato and C.Liu, *Tribology International*, 2011,**44**,1329-1332.
- 30 29.Z.D.Sha, P.S.Branicio, V.Sorkin, Q. X. Pei and Y.W.Zhang, *Diamond & Related*
- 31 *Materials*, 2011,**20**,1303-1309.
- 32 30.J.F.Ziegler, M.D.Ziegler and J.P.Biersack, *Nuclear Instruments and Methods in*
- 33 *Physics Research B*,2010,**268**,1818-1823.
- 34 31.I.N. Remediakis, G.Kopidakis and P.C.Kelires, *Acta Materialia*, 2008,**56**,5340-5344.
- 35 32.J. Tersoff, *Physical Review B*,1988,**39**,5566-5568.
- 36 33.J.Tersoff, *Physical Review Letters*,1988,**61**,2879-2882.

- 1 34.S.K. Deb Nath., C.H. Wong, V. Sorokin, Z.D. Sha, Y.W. Zhang and S.-G. Kim,
2 *Tribology Transactions*, 2013,**56**,255-267.
- 3 35.S.K. Deb Nath, *Applied Physics A*, 2014,**117**,857-870.
- 4 36.P.H.Kasai, A.Wass and B.K.Yen, *Journal of Information Storage and Processing*
5 *Systems*,1999,**1**,245-258.
- 6 37.P.H.Kansai and A.M.Spool, *IEEE Transactions on Magnetics*, 2001,**37**,929-933.
- 7 38.S.J. Plimpton, *Journal of Computational Physics*, 1995,**117**,1-19.
- 8 39.L.Bei, C.H.Wong and Q.Chen, *Soft Matter*, 2013, **9**,700-708.
- 9 40.Q. GuO, S. Izumisawa, D.M. Phillips and M.S. Jhon, *Journal of Applied Physics*,
10 2003,**93**,8707-8709.
- 11 41.M.Tsige and M.J.Stevens, *Macromolecules*,2004,**37**,630-637.
- 12 42.M.J.Stevens, *Macromolecules*, 2001,**34**,1411-1415.
- 13 43.S. Izumisawa and M.S. Jhon, *Journal of Applied Physics*, 2002,**91**,7583-7585.
- 14 44.P.S. Chung, H. Park and M.S. Jhon, *IEEE Transactions on*
15 *Magnetics* ,2009,**45**,3644-3647.
- 16 45.X. Li, Y. Hu, L. Jiang and J. Zhang, *The Journal of Chemical Physics*,
17 2008,**128**,194904.

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216x121mm (96 x 96 DPI)

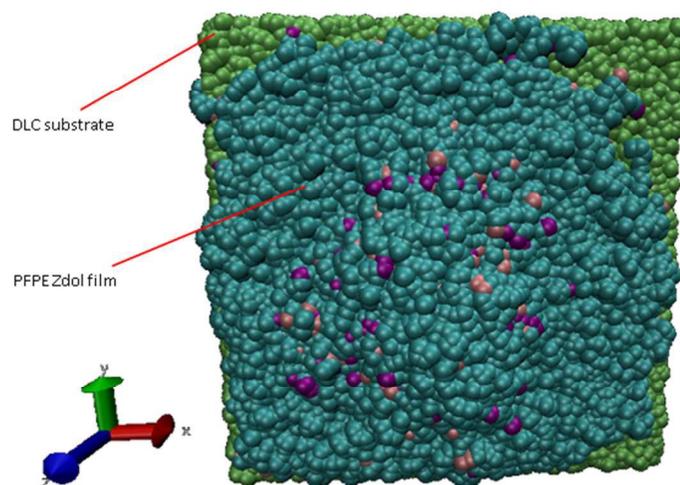
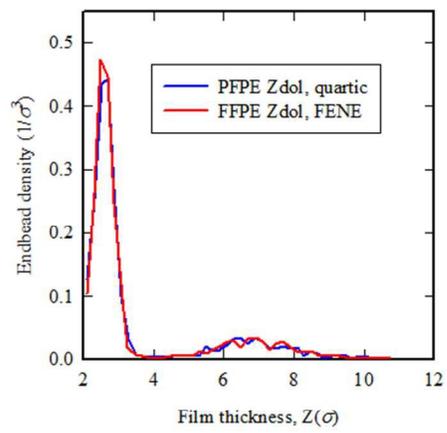


Fig.2 Front view of initial configuration of PFPEZdol film on a DLC substrate

216x129mm (96 x 96 DPI)



3(a)

216x121mm (96 x 96 DPI)

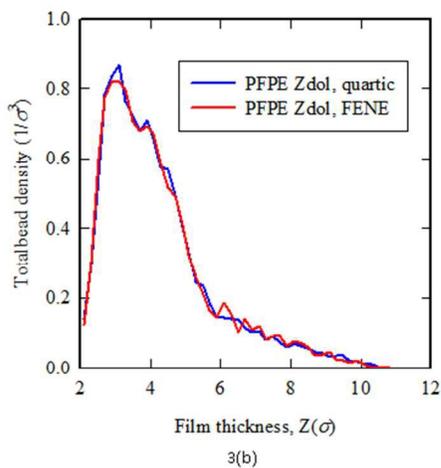


Fig.3 Effects of two different potentials such as quartic and FENE :(a) on the film thickness as a function of end bead density; (b) film thickness as a function of total bead density of PFPE Zdol lubricant molecules on a DLC film at a temperature, $T^* = 1.3$

216x134mm (96 x 96 DPI)

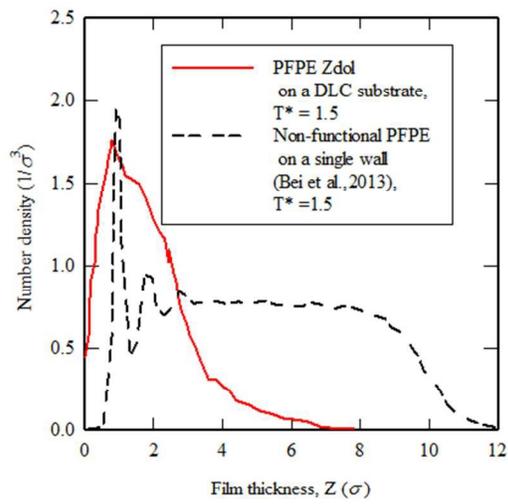


Fig.4 Comparative study of the number density of beads of PFPE Zdol on a DLC substrate with that of non-functional PFPE on a single wall(Bei et al., 2013) at a reduced temperature $T^*=1.3$

216x129mm (96 x 96 DPI)

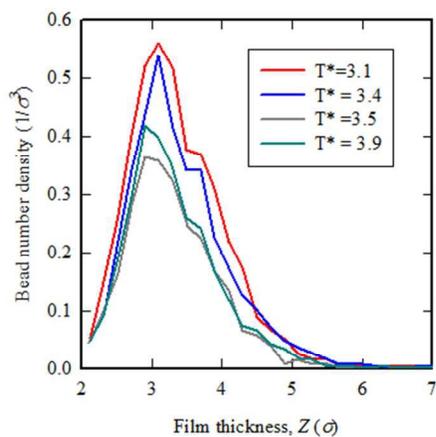
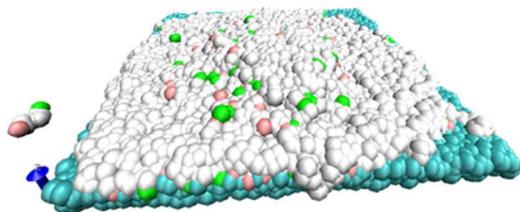


Fig.5 Effect of temperature on the total bead density of PFPE Zdol lubricant as a function of film thickness at temperature $T^* = 3.1, 3.4, 3.5, 3.9$

216x129mm (96 x 96 DPI)



6(a)

216x129mm (96 x 96 DPI)

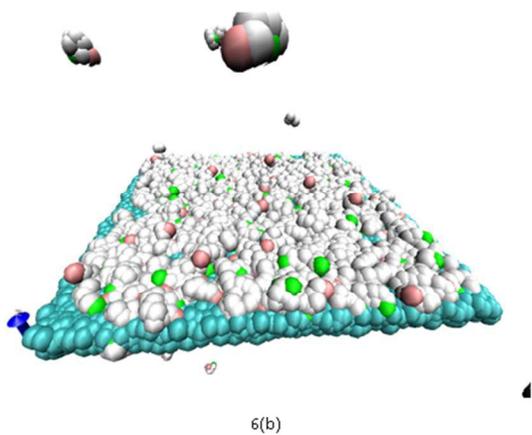


Fig.6 Snapshots of PFPE Zdol lubricant molecules on a DLC substrate at constant temperature, $T^*=3$: (a) at time, $\tau=0$; (b) at $\tau=10000$ during isothermal heating

216x129mm (96 x 96 DPI)

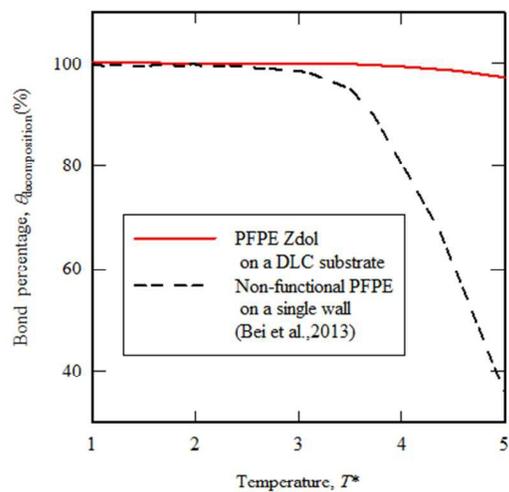
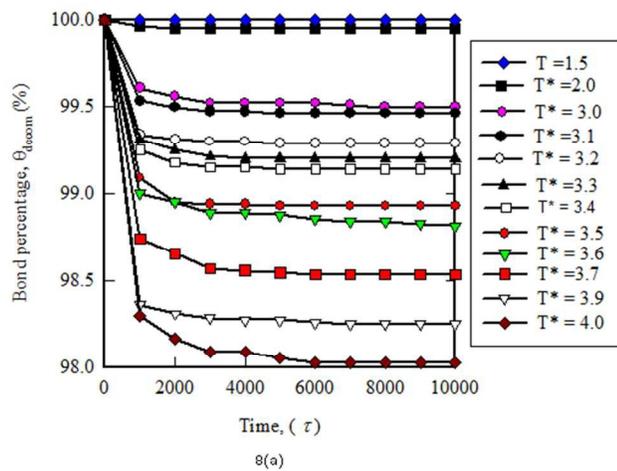


Fig. 7 Comparative study of bond change of PFPE Zdol on a DLC substrate with that of non-functional PFPE on a single wall (Bei et al., 2013) with reduced temperature

216x138mm (96 x 96 DPI)



216x138mm (96 x 96 DPI)

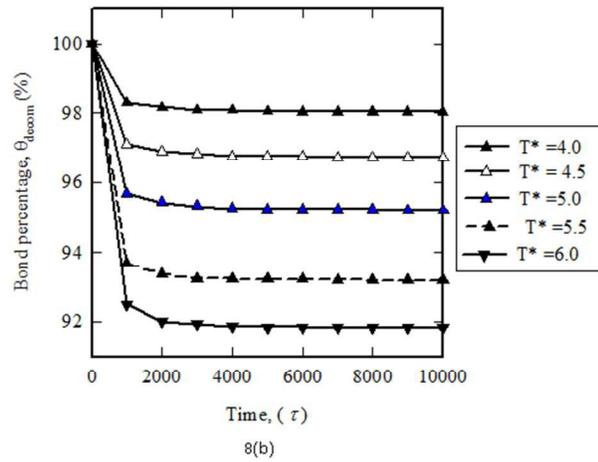


Fig.8 Effect of temperature on the bond percentage of PFPE Zdol lubricant molecules on a DLC substrate: (a) at temperature range, $T^*=1.5-4$; (b) at temperature range, $T^*=4-6$

216x138mm (96 x 96 DPI)

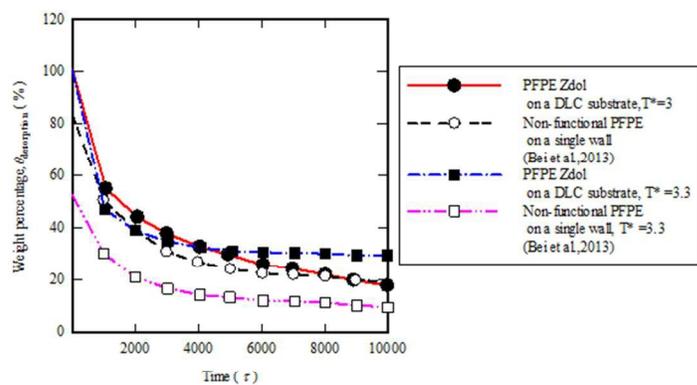
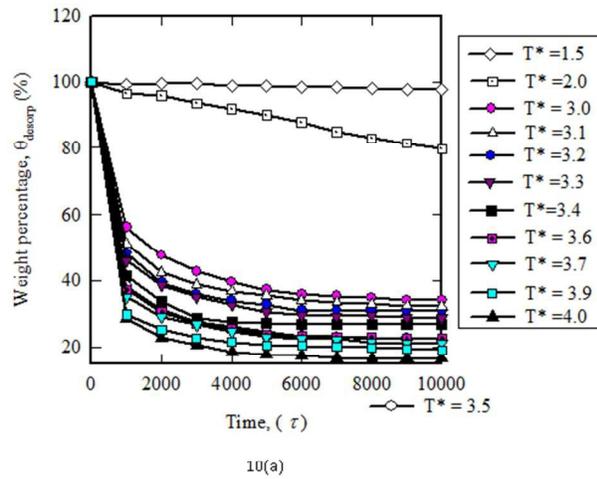


Fig.9 Comparative study of mass change of PFPE Zdol on a DLC substrate with that of non-functional PFPE on a single wall (Bei et al., 2013) with time t

216x138mm (96 x 96 DPI)



216x138mm (96 x 96 DPI)

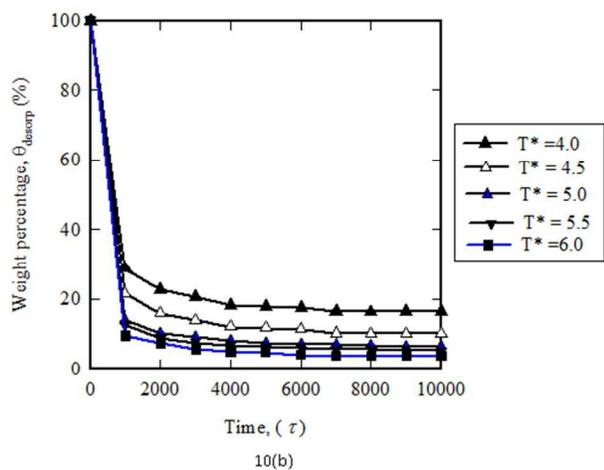
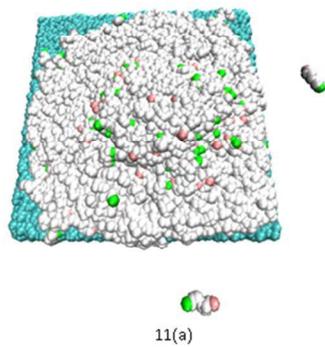
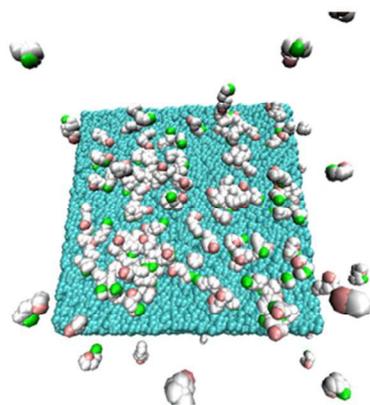


Fig.10 Effect of temperature on the weight percentage of PFPE Zdol lubricant molecules on a DLC substrate: (a) at temperature range, $T^* = 1.5 - 4$; (b) at temperature range, $T^* = 4 - 6$

216x133mm (96 x 96 DPI)



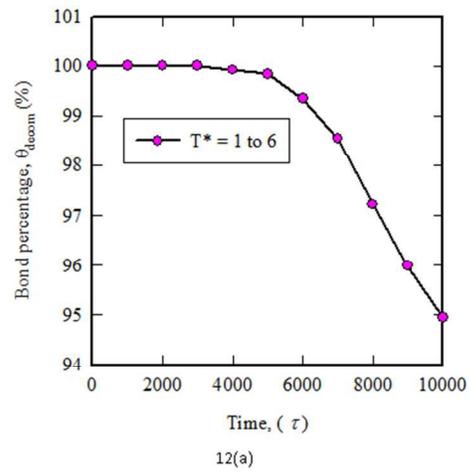
216x138mm (96 x 96 DPI)



11(b)

Fig.11 Snapshots of PFPEZdol lubricant molecules on a DLC substrate at varying temperature, $T^*=1-6$: (a) at time, $\tau=0$; (b) at $\tau=10000$

216x138mm (96 x 96 DPI)



216x138mm (96 x 96 DPI)

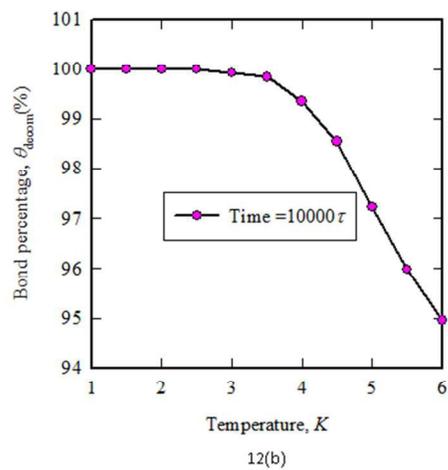


Fig.12 Bond percentage of PFPE Zdol lubricant molecules on a DLC substrate: (a) as a function of elapsed time; (b) as a function of temperature during heating at increased temperature $T^*=1$ to 6

216x138mm (96 x 96 DPI)