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# Ba<sub>7</sub>(BO<sub>3</sub>)<sub>3</sub>GeO<sub>4</sub>X (X=Cl, Br): the borogermanate halides with rigid GeO<sub>4</sub> tetrahedra and flexible XBa<sub>6</sub> octahedra

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The first borogermanate halides,  $Ba_7(BO_3)_3GeO_4X$  (X=Cl, Br) have been synthesized by high temperature solution method.  $Ba_7(BO_3)_3GeO_4X$  (X=Cl, Br) are isostructural and crystallize the orthorhombic space group *Pbam* (no. 55). They feature a Ba-based 3D framework, while isolated GeO\_4 tetrahedra and BO\_3 triangles are filled in the space created by the framework. Significantly,  $Ba_7(BO_3)_3GeO_4X$  (X=Cl, Br) possess similar formula with the borosilicate halides,  $Ba_7(BO_3)_3GeO_4X$  (X=Cl, Br), while they are not isostructural. A interpretation for the structural transformation between them is demonstrated here considering rigid GeO\_4 tetrahedra and flexible XBa<sub>6</sub> (X=Cl, Br) octahedra. IR spectrum, UV-Vis-NIR diffuse reflectance spectrum and first-principles calculations have been carried out on title compounds.

### Introduction

Borates have been widely investigated owing to their rich chemistry structures.<sup>1-9</sup> Especially the borates with planar BO<sub>3</sub> units have attracted considerable attention, because the  $\pi$ -delocalization of BO<sub>3</sub> group is beneficial for strong second-harmonic generation (SHG) response and large birefringence based on anionic group theory. A statistical analysis of borate fundamental building blocks, carried out by P. Becker,<sup>10</sup> indicates that the borates with isolated BO<sub>3</sub> groups will be found in the system with appropriate cations/boron ratios (>1). Thus, in the alkaline-earth borates the formula  $M_7(BO_3)_3F_5$  (M=Ba or Sr) with a large cations/boron ratio have attracted scientist's interests.

In the  $M_7(BO_3)_3F_5$  (M=Ba or Sr) system, two compounds  $Ba_7(BO_3)_3F_5^{11}$  and  $Ba_3Sr_4(BO_3)_3F_5^{11}$  were first identified by Keszler *et al* in 1994. In 2008, Zhang's group has reinvestigated the crystal structure of  $Ba_3Sr_4(BO_3)_3F_5$  and grown large crystal by the top-seeded solution growth method, which shows an SHG intensity about 0.5 times as large as that of KH<sub>2</sub>PO<sub>4</sub> (KDP).<sup>12</sup> Then in 2012, Bekker *et al* has systematically studied the  $M_7(BO_3)_3F_5$  (M=Ba or Sr) system. They found that the system exists disorders between  $(4F)^{4-}$  and  $(BO_3F)^{4-}$  and a series compounds of  $Ba_7(BO_3)_{4-x}F_{2+3x}$  and  $Ba_{4-x}Sr_{3+x}(BO_3)_{4-y}F_{2+3y}^{13-14}$  have been identified. However, in the crystal structure, the existence of disorder in the crystal is not favorable for the excellent physical or chemistry properties.

The basic structural units  $SiO_4$  have been introduced into borate to replace the  $(4F)^{4-}$  or  $(BO_3F)^{4-}$  owing to  $SiO_4$  tetrahedra possessing the same charge with  $(4F)^{4-}$  or  $(BO_3F)^{4-}$ , which may solve the disorders in the crystal structure and improve the properties. Inspired by this, the compounds Ba7(BO3)3SiO4X (X=Cl, Br)<sup>15</sup> have been synthesized by our group, in which the structural disorder disappears. Ba7(BO3)3SiO4X (X=Cl, Br) have SHG response as strong as that of KDP. However, as is known to all, high reaction temperature and high viscosity of the silicate are not conducive for the growth of crystals. Comparing to silicates, germanates have lower reaction temperature. And in previous research, many alkaline and alkaline-earth metal borogermanates have been reported.<sup>16-20</sup> In addition, the Ge and Si elements are in the same main group. and possess similar coordination environment. Guided by these ideas, the first borogermanate halides, Ba7(BO3)3GeO4X (X=Cl Br) have been successfully synthesized. Owing to the large cations/boron ratio (7:3), the BO<sub>3</sub> triangles and GeO<sub>4</sub> tetrahedra are all isolated, which is quite rare in inorganic borogermanates.<sup>21</sup> It is worth noted that although the Ge and Si atoms possess similarity, Ba7(BO3)3GeO4X (X=Cl, Br) and Ba<sub>7</sub>(BO<sub>3</sub>)<sub>3</sub>SiO<sub>4</sub>X (X=Cl, Br) crystallize in different space groups, centrosymmetric Pbam (no. 55) and polar P63mc (r 186), respectively.

Thus, we have investigated the substitution of Ge for Si in other reported borates determined by single crystal X-r<sub>n</sub>, diffraction structure.<sup>17-18, 22-28</sup> We found that most of the compounds are isostructural (Table S1 in the Supporting Information). In these borates,<sup>17-18, 22-28</sup> we define a parameter of R (O/M, M=Si or Ge), which represents the mole ratio of the

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J. Name., 2013, **00**, 1-3 | 1

Page 2 of 7

O element and the substitution element. In general, a higher value of R means a lower mole ratio of the substituted element in the compounds, which causes smaller change from the substitution and more likely results in the isostructural substitution. Therefore when R is higher than 3, all the compounds are isostructural.<sup>17,23-28</sup> It is unusual that  $Ba_7(BO_3)_3GeO_4X$  (X=Cl, Br) and  $Ba_7(BO_3)_3GeO_4X$  (X=Cl, Br) and  $Ba_7(BO_3)_3GeO_4X$  (X=Cl, Br) are not isostructural with a quite high R value of 13. The structural transformation between  $Ba_7(BO_3)_3GeO_4X$  (X=Cl, Br) and  $Ba_7(BO_3)_3GeO_4X$  (X=Cl, Br) have been discussed in detail in the paper. We have also reported the spectrum and the first-principles calculations of the title compounds.

### **Experimental**

### Solid-State synthesis

Polycrystalline samples of Ba<sub>7</sub>(BO<sub>3</sub>)<sub>3</sub>GeO<sub>4</sub>X (X=Cl, Br) were synthesized via conventional solid-state reactions. Stoichiometric mixtures of BaCO<sub>3</sub>, BaCl<sub>2</sub>, GeO<sub>2</sub>, and B<sub>2</sub>O<sub>3</sub> at a molar ratio 6.5:0.5:1:1.5 for Ba<sub>7</sub>(BO<sub>3</sub>)<sub>3</sub>GeO<sub>4</sub>Cl and BaCO<sub>3</sub>, BaBr<sub>2</sub>, GeO<sub>2</sub>, and B<sub>2</sub>O<sub>3</sub> at a molar ratio 6.5:0.5:1:1.5 for Ba<sub>7</sub>(BO<sub>3</sub>)<sub>3</sub>GeO<sub>4</sub>Br have been ground and packed into alumina crucibles. Raw materials were preheated at 600 °C for 14 h. The mixtures were heated to 900°C for Ba<sub>7</sub>(BO<sub>3</sub>)<sub>3</sub>GeO<sub>4</sub>Cl and 950 °C for Ba<sub>7</sub>(BO<sub>3</sub>)<sub>3</sub>GeO<sub>4</sub>Br, then held for 4 days, and finally cooled to room temperature. During the sintering processes the samples have been ground thoroughly. The purity of the samples was confirmed by a powder X-ray diffraction (XRD), which agree well with the theoretical patterns of the compounds (Figure S1 in the Supporting Information).

### Crystal growth

The single crystals of Ba<sub>7</sub>(BO<sub>3</sub>)<sub>3</sub>GeO<sub>4</sub>X (X=Cl, Br) were grown from the high temperature solutions using NaCl–LiCl, NaBr–LiBr and B<sub>2</sub>O<sub>3</sub> as the flux systems. The solutions were obtained in platinum crucibles by melting mixtures of BaCO<sub>3</sub>, GeO<sub>2</sub>, B<sub>2</sub>O<sub>3</sub>, NaCl and LiCl at a molar ratio of 7:1:2.5:2:5 for Ba<sub>7</sub>GeB<sub>3</sub>O<sub>13</sub>Cl and BaCO<sub>3</sub>, GeO<sub>2</sub>, B<sub>2</sub>O<sub>3</sub>, NaBr and LiBr at a molar ratio of 7:1:2.5:1:4 for Ba<sub>7</sub>GeB<sub>3</sub>O<sub>13</sub>Br. The solutions were heated to 900 °C, dwell at this temperature for 10 h, then slowly cooled to 700 °C at a rate of 3 °C·h<sup>-1</sup> and finally cooled down to room temperature at a rate of 10 °C·h<sup>-1</sup>. The crystals of Ba<sub>7</sub>(BO<sub>3</sub>)<sub>3</sub>GeO<sub>4</sub>X (X=Cl, Br) have been obtained, which were used for the single-crystal X-ray diffraction analysis.

### Structure determination

The crystal structures of Ba<sub>7</sub>(BO<sub>3</sub>)<sub>3</sub>GeO<sub>4</sub>X (X=Cl, Br) were collected by a Bruker SMART APEX II CCD diffractometer with monochromatic Mo K $\alpha$  radiation at 293(2) K, which were futher integrated with the SAINT program.<sup>29</sup> The calculations were completed by programs from the SHELXTL crystallographic software package, while their structures were determined by direct methods with SHELXS-97. Final least-squares refinements are on F<sub>o</sub><sup>2</sup> with data having F<sub>o</sub><sup>2</sup>  $\ge 2\sigma(F_o^2)$ .

PLATON has been used to checke for missing symmetry.<sup>30</sup> Crystal data and structure refinement information are shown in Table 1. And final refined atomic positions and isotropic thermal parameters are displayed in Table S2 in the Supporting Information.

Table 1Crystaldataandstructurerefinementsfor $Ba_7(BO_3)_3GeO_4Cl$  and  $Ba_7(BO_3)_3GeO_4Br$ .

			_
Empirical formula	Ba7(BO3)3GeO4Cl	Ba7(BO3)3GeO4B	r
Temperature	296(2) K		
Wavelength	0.71073 Å		
Crystal system	Orthorhombic		
Space group	Pbam		
Formula weight	1309.85	1354.31	
a (Å)	20.353(18)	20.381(8)	
b (Å)	7.416(6)	7.477(3)	YU
<i>c</i> (Å)	11.123(10)	11.175(4)	
Z, Volume (Å <sup>3</sup> )	4, 1679(3)	4, 1702.9(11)	
$\rho_{\text{Calcd}} (\text{Mg/m}^3)$	5.183	5.283	
μ(/mm)	18.121	20.054	
R(int)	0.0776	0.0651	$\bigcirc$
Goodness-of-fit on F <sup>2</sup>	1.037	1.043	
Final R indices	R1 = 0.0351,	R1 = 0.0315,	U
$[F_o^2 > 2\sigma (F_o^2)]^a$	wR2 = 0.0612	wR2 = 0.0585	
R indices (all data) <sup>a</sup>	R1 = 0.0545,	R1 = 0.0426,	
	wR2 = 0.0675	wR2 = 0.0630	
Extinction coefficient	0.00061(4)	0.00207(7)	1
Largest diff. peak and hole $(e; Å^{-3})$	2.403 and -2.209	2.158 and -2.070	

<sup>a</sup>  $R_1 = \Sigma ||F_o| - |F_c|| / \Sigma |F_o|$  and  $wR_2 = [\Sigma w (F_o^2 - F_c^2)^2 / \Sigma w F_o^4]^{1/2}$ for  $F_o^2 > 2\sigma (F_o^2)$ 

### Infrared and UV-vis-NIR diffuse reflectance spectroscopy

The infrared spectra with the range of 400–4000 cm<sup>-1</sup> have been performed on Shimadzu IR Affinity-1 Fourier transform infrared spectrometer. UV–vis–NIR diffuse reflectance data have been performed on the Solid Spec-3700DUV spectrophotometer with a range from 190 to 2600 nm. Then Polytetrafluoroethylene is used as a standard. Finally we converted the reflectance spectra to absorbance by the function,  $F(R) = (1 - R)^2/2R$ . R represents the reflectance while F(R)represents the Kubelka–Munk remission function.<sup>31</sup>

### Numerical calculation details

The electronic structure calculations were performed using a plane-wave basis set and pseudopotentials within density functional theory (DFT) implemented in the total-energy module CASTEP.<sup>32</sup> The exchange and correlation effects wer treated by Perdew–Burke–Ernzerhof (PBE) in the generalized gradient approximation (GGA).<sup>33-35</sup> The interactions between the ionic cores and the electrons were described by Ultrasc a pseudopotentials. The following orbital electrons were treated as valence electrons: Ba  $5s^25p^66s^25d^0$ , Ge  $4s^24p^2$ , B  $2s^22p^1$ , O

Journal Name

The linear optical response properties of the Ba<sub>7</sub>(BO<sub>3</sub>)<sub>3</sub>GeO<sub>4</sub>X (X=Cl, Br) were examined through calculating the complex dielectric function  $\varepsilon(\omega) = \varepsilon_I(\omega) + i\varepsilon_2(\omega)$ . The imaginary part of the dielectric function  $\varepsilon_2$  is given in the following equation:<sup>37</sup>

$$\varepsilon_{2}\left(\mathbf{q}\rightarrow O_{\hat{u}},h\omega\right)=\frac{2e^{2}\pi}{\Omega\varepsilon_{o}}\sum_{kcv}\left|\left\langle \varphi_{k}^{c}\left|\boldsymbol{u}\cdot\boldsymbol{r}\right|\varphi_{k}^{v}\right\rangle \right|^{2}\delta\left[E_{k}^{c}-E_{k}^{v}-E\right]$$

The real part  $\varepsilon_1(\omega)$  can be obtained from the imaginary part  $\varepsilon_2(\omega)$  by the Kramers–Kronig transformation. All the other optical constants, such as the absorption spectrum, refractive index, and reflectivity are derived from  $\varepsilon_1(\omega)$  and  $\varepsilon_2(\omega)$ .

### **Results and discussion**

### Structure

Both of the title compounds crystallize in an orthorhombic space group, *Pbam* (no. 55). As  $Ba_7(BO_3)_3GeO_4X$  (X=Cl, Br) are isostructural, only the structure of  $Ba_7(BO_3)_3GeO_4Br$  is described in detail here. There are six unique Ba sites, two unique B sites, one unique Ge site, one unique Br site and eight unique O sites in the asymmetric unit (Table S2 in the Supporting Information). In the structure of  $Ba_7(BO_3)_3GeO_4Br$ ,



**Figure 1** The structure of  $Ba_7(BO_3)_3GeO_4Br$ . (a) Ba-based 3D framework. (b) 3D network of  $Ba_7(BO_3)_3GeO_4Br$ . All the Ba-O and Ba-Br bonds have been omitted for clarity.

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Selected bond distances and angles for  $Ba_7(BO_3)_3GeO_4X$  (X=Cl, Br) are presented in Figure S3 in the Supporting Information. The Ba–O, B–O, Ge–O and Ba-Br distances varying over 2.567(7)–3.217(5) Å, 1.362(12)–1.391(9) Å, 1.737(7)–1.766(5) Å and 3.1424(15)–3.7463(13) Å, respectively. These values are observed in other reported compounds.<sup>16a, 38</sup>

# Comparing the structure of Ba<sub>7</sub>(BO<sub>3</sub>)<sub>3</sub>GeO<sub>4</sub>X (X=Cl, Br) with Ba<sub>7</sub>(BO<sub>3</sub>)<sub>3</sub>SiO<sub>4</sub>X (X=Cl, Br).

Ba<sub>7</sub>(BO<sub>3</sub>)<sub>3</sub>SiO<sub>4</sub>X (X=Cl, Br) are isostructural and crystallize in the hexagonal  $P6_{3}mc$  space group, while Ba<sub>7</sub>(BO<sub>3</sub>)<sub>3</sub>GeO<sub>4</sub>X (X=Cl, Br) crystallize in orthorhombic *Pbam*. Three unique Ba atoms, one unique B atom, one unique Si atom, four unique atoms and one unique Br atom are in the asymmetric unit  $Pa_7(BO_3)_3SiO_4Br$ . The structure of Ba<sub>7</sub>(BO<sub>3</sub>)<sub>3</sub>SiO<sub>4</sub>Br are composed of a 3D Ba(3)-based framework with tunne<sup>1</sup>s viewing along *c* axis, in which the BrBa(2)<sub>3</sub> chains are resided (Figure 2a). In addition,  ${}_{\infty}{}^{1}$ [Ba(1)O<sub>6</sub>(SiO<sub>4</sub>)] single chains and isolate BO<sub>3</sub> triangles are filled in the framework to form its 3D network structure (Figures 2b and 2c). In the structure, the  ${}_{\infty}{}^{1}$ [Ba(1)O<sub>6</sub>(SiO<sub>4</sub>)] single chain consists of isolated SiO<sub>4</sub> tetrahedra and Ba(1)O<sub>10</sub> polyhedra.



**Figure 2** The structure of  $Ba_7(BO_3)_3SiO_4Br.$  (a)  $_{\infty}^{-1}[BrBa_3]$ . (b) Ba(3)-based 3D framework. (c) 3D network of  $Ba_7(BO_3)_3SiO_4Br$ . All the Ba–O bonds have been omitted for clarity.

Similarly, the structure of  $Ba_7(BO_3)_3GeO_4Br$  composes of a 3D Ba(1, 4, 5)-based framework with tunnels viewing along *b* axis, in which the BrBa(2, 6)<sub>3</sub> chains are located (Figure 2a). In

J. Name., 2012, 00, 1-3 | 3

 $Ba(3)O_{10}$  polyhedra.

addition,  ${}_{\infty}{}^{1}\{[Ba(3)O_{4}(GeO_{4})]_{2}\}$  double chains and isolate BO<sub>3</sub> the striangles are filled in the framework to form its 3D network (Fig structure (Figures 2b and 2c). The  ${}_{\infty}{}^{1}\{[Ba(3)O_{4}(GeO_{4})]_{2}\}$  double chain are built of isolated GeO<sub>4</sub> tetrahedra and

Accordingly, the structural transformation between  $Ba_7(BO_3)_3SiO_4Br$  and  $Ba_7(BO_3)_3GeO_4Br$  will be discussed from the following two parts: (**A**) the arrangements of chains and (**B**) the Ba-based 3D frameworks.



(c) 3D network of Ba7(BO3)3GeO4Br

**Figure 3** The structure of  $Ba_7(BO_3)_3GeO_4Br.$  (a)  ${}_{\infty}^{-1}$  [BrBa<sub>3</sub>]. (b) Ba-based 3D framework. (c) 3D network of  $Ba_7(BO_3)_3GeO_4Br.$  All the Ba-O bonds have been omitted for clarity.

(A) The transformation from  ${}_{\infty}^{-1}[Ba(1)O_6(SiO_4)]$  single chains to  $\infty^{1}{[Ba(3)O_{4}(GeO_{4})]_{2}}$  double chains. In the structure of Ba7(BO3)3SiO4Br, the parallel single chains of  ${}_{\infty}^{1}$ [Ba(1)O<sub>6</sub>(SiO<sub>4</sub>)] are linked by Ba(3)O<sub>9</sub> polyhedra (Figure 4a). When the Si atoms are replaced by the Ge atoms, it causes a weak squeeze between GeO<sub>4</sub> tetrahedra and Ba(3)O<sub>9</sub> polyhedra owing to the dimensional effect (the average bond lengths of Si–O is 0.155 Å smaller than that of Ge–O). In addition, as we know, the rigid units are hard to be compressed and squeezed while the flexible ones are easy. In the structure, the SiO<sub>4</sub> and GeO<sub>4</sub> tetrahedra can be regarded as rigid due to their small bond distance variations comparing to that of Ba-O polyhedra. However, comparing with Ba-O polyhedra, the BrBa<sub>6</sub> octahedra can be regarded as flexible units for their largest bond distance variations in the structures. Thus, the  $Ba(3)O_9$ polyhedra are squeezed by the rigid GeO<sub>4</sub> tetrahedra to the flexible BrBa<sub>6</sub> octahedra (Figure 4b). As a result, the Ba(3)O<sub>9</sub> polyhedra have been squeezed out of the chains, which leads to the single chains connect each other to form the double chains (Figures 4c and 4d).



**Figure 4** The transformation from  ${}_{\infty}{}^{1}$ [Ba(1)O<sub>6</sub>(SiO<sub>4</sub>)] single chains to  ${}_{\infty}{}^{1}$ {[Ba(3)O<sub>4</sub>(GeO<sub>4</sub>)]<sub>2</sub>} double chains. (a)  ${}_{\infty}{}^{1}$ [Ba(1)O<sub>6</sub>(SiO<sub>4</sub>)] single chains. (b) the arrangement of sing chains and BrBa<sub>6</sub> octahedra in Ba<sub>7</sub>(BO<sub>3</sub>)<sub>3</sub>SiO<sub>4</sub>Br. (c)  ${}_{\infty}{}^{1}$ {[Ba(3)O<sub>4</sub>(GeO<sub>4</sub>)]<sub>2</sub>} double chains. (d) the arrangement of single chains and BrBa<sub>6</sub> octahedra in Ba<sub>7</sub>(BO<sub>3</sub>)<sub>3</sub>GeO<sub>4</sub>Br.

(B)The transformation of the Ba-based 3D frameworks Both Ba<sub>7</sub>(BO<sub>3</sub>)<sub>3</sub>SiO<sub>4</sub>Br and Ba<sub>7</sub>(BO<sub>3</sub>)<sub>3</sub>GeO<sub>4</sub>Br have a Ba:Si or Ba:Ge molar ratio of 7:1, but the Ba atoms are arranged in different configurations. In the structure of Ba7(BO3)3SiO4Br, the BrBa<sub>6</sub> octahedra are in the center of neighboring six Si tetrahedra and the SiO<sub>4</sub> tetrahedra connect with the Ba(3) atoms to form a single chain viewing along the c axis (Figure 5a). From Figure 5b, we can also observe that every six Ba atoms are lied around one SiO<sub>4</sub> tetrahedron. The rigid SiO<sub>4</sub> tetrahedra have pushed the Ba(2) atoms to the Br atoms, which makes the six Ba atoms like a pattern of equilateral triangle (Figure 5b). Six equilateral triangles constitute a hexagon configuration template. These suggest that Ba7(BO3)3SiO4Br crystallizes in the hexagonal crystal system. While in the structure of Ba<sub>7</sub>(BO<sub>3</sub>)<sub>3</sub>GeO<sub>4</sub>Br, similarly, the BrBa<sub>6</sub> octahedra are located at the center of the SiO<sub>4</sub> tetrahedra and the Ba atoms lie around the SiO<sub>4</sub> tetrahedra (Figures 5c and 5d). Owing to the squeeze from the  $SiO_4$  tetrahedra to the Ba(2) and Ba(6) atoms, the Ba atoms form an analogous pattern of rectangle. And four rectangles compose a bigger one, which can become a rectangle template (Figure 5d). These imply that Ba<sub>7</sub>(BO<sub>3</sub>)<sub>3</sub>GeO<sub>4</sub>P crystallizes in the orthogonal crystal system.

Based on above analyses, rigid  $SiO_4$  tetrahedra and flexible BrBa<sub>6</sub> octahedra play key roles in the structural transformatic a from hexagonal Ba<sub>7</sub>(BO<sub>3</sub>)<sub>3</sub>SiO<sub>4</sub>Br to orthogon.' Ba<sub>7</sub>(BO<sub>3</sub>)<sub>3</sub>GeO<sub>4</sub>Br.

### **RSC Advances**



**Figure 5** The comparison of different Ba-based frameworks in  $Ba_7(BO_3)_3SiO_4Br$  and  $Ba_7(BO_3)_3GeO_4Br$ . (a) The arrangements of Si and Br atoms in  $Ba_7(BO_3)_3SiO_4Br$ . (b) The arrangements of Si, Ba and Br atoms in  $Ba_7(BO_3)_3SiO_4Br$ . (c) The arrangement of Ge and Br atoms in  $Ba_7(BO_3)_3GeO_4Br$ . (d) The arrangement of Ge, Ba and Br atoms in  $Ba_7(BO_3)_3GeO_4Br$ .

### Infrared and UV-vis-NIR diffuse reflectance spectroscopy

The infrared spectroscopy of Ba<sub>7</sub>(BO<sub>3</sub>)<sub>3</sub>GeO<sub>4</sub>X (X=Cl, Br) are quite same. For both of them, the bands appeared in the range of 1350–1450 cm<sup>-1</sup> and 1150–1250 cm<sup>-1</sup> can be assigned to the asymmetric stretching vibration of BO<sub>3</sub> groups (Figure S2 in the Supporting Information). The peaks observed in the region of 733–641 cm<sup>-1</sup> are attributed to the out-of-bending of BO<sub>3</sub>. The peaks in the region of 860-900 cm<sup>-1</sup> can be assigned to an asymmetrical stretch of Ge–O of the tetrahedral germanium. The absorption bands in the range of 450–600 cm<sup>-1</sup> correspond to the bending vibrations of the Ge–O bonds. Infrared spectra further confirm the existence of BO<sub>3</sub> triangles and GeO<sub>4</sub> tetrahedra, which is consistent with the results obtained from single crystal X-ray structural analyses.

The UV-vis-NIR diffuse reflectance spectra of  $Ba_7(BO_3)_3GeO_4X$  (X=Cl, Br) are deposited in Figure S3 in the Supporting Information. It indicates that the the experimental optical band gap of  $Ba_7(BO_3)_3GeO_4Cl$  and  $Ba_7(BO_3)_3GeO_4Br$  are 5.28 and 5.31 eV, respectively. The large band gaps of the title compounds suggest that they may be used as the optical window materials.

### Band structures, density of states and optical properties

The band structures of  $Ba_7(BO_3)_3GeO_4X$  (X=Cl, Br) are presented in Figure S4 in the Supporting Information. The valence band maximum (VBM) and the conduction band minimum (CBM) are located at  $\Gamma$  point of the Brillouin zone indicating both of the two compounds belong to direct bandgap semiconductors. The extrapolated experimental optical gaps of 5.28 eV for  $Ba_7(BO_3)_3GeO_4Cl$  and 5.31 eV for  $Ba_7(BO_3)_3GeO_4Br$  are slightly larger than the calculated values 4.93 eV and 4.91 eV respectively, the underestimation of the

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band gap is generally due to insufficiently accuracy of exchange correlation energy under DFT methods.<sup>39</sup> We can see that the results of our calculation are good and give reasonable explanations for the optical absorption spectra. The highest occupied and lowest unoccupied orbitals determine the way of electronic transitions and the absorption edge. It is shown in Figure S5 in the Supporting Information that the absorption edges of  $Ba_7(BO_3)_3GeO_4Cl$  and  $Ba_7(BO_3)_3GeO_4Br$  are mainly decided by O 2p states in BO<sub>3</sub> units.

The total and partial densities of states (DOS and PDOS) are shown in Figure 6. The top region of VBs extends wide range from -6.7 eV to VBM. This bands mostly originate from Ge 4s4p, B 2s2p and O 2s2p states with the mixings of Ba 5s5p and Cl 3p/ Br 4p states, especially near the Fermi level. It is worthy to note that strong hybridizations occur among the Ge 4s4p, B 2s2p and O 2s2p states in the range of -6.7 eV to -0.7 eV. For  $Ba_7(BO_3)_3GeO_4Cl$  and  $Ba_7(BO_3)_3GeO_4Br$ , the conduction bands from CBM to 8.5 eV are derived from Ba 5 Ge 4s4p and B 2p states with mixings of Ba 5s5p, B 2s and O 2s2p states, implying strong interactions in Ge–O and B–O bonds for the compounds. The valence bands and conduction bands near the gap are dominated by O 2p and Ba 5drespectively. Accordingly, the absorption spectrum near the UV-Visible cut-off wavelength can be assigned as the charge transfers from O 2p states to the Ba 5d states.



Figure 6 TDOS and PDOS plots of  $Ba_7(BO_3)_3GeO_4Cl$  (a) and  $Ba_7(BO_3)_3GeO_4Br$  (b).

On the basis of the electronic structures, the linear optical is properties can be estimated. The imaginary part of dielectric function  $\varepsilon_2$  is calculated, and its real part  $\varepsilon_1$  is determined by the function  $\varepsilon_2$  is calculated, and its real part  $\varepsilon_1$  is determined by the function  $\varepsilon_2$  is calculated, and its real part  $\varepsilon_1$  is determined by the function  $\varepsilon_2$  is calculated, and its real part  $\varepsilon_1$  is determined by the function  $\varepsilon_2$  is calculated, and its real part  $\varepsilon_1$  is determined by the function  $\varepsilon_2$  is calculated, and its real part  $\varepsilon_1$  is determined by the function  $\varepsilon_2$  is calculated, and its real part  $\varepsilon_1$  is determined by the function  $\varepsilon_2$  is calculated, and its real part  $\varepsilon_1$  is determined by the function  $\varepsilon_2$  is calculated by the birefringence  $\Delta n$  were obtained.<sup>40</sup> The dispersion  $\varepsilon_1$  and  $\varepsilon_2$  is calculated by the formula  $n^2(\omega) = \varepsilon(\omega)$  displaying a similar optical anisotropy behavior:  $n^x \approx n^y > 1$  $n^z$  for Ba<sub>7</sub>(BO<sub>3</sub>)<sub>3</sub>GeO<sub>4</sub>Cl and Ba<sub>7</sub>(BO<sub>3</sub>)<sub>3</sub>GeO<sub>4</sub>Br. And the birefringences of Ba<sub>7</sub>(BO<sub>3</sub>)<sub>3</sub>GeO<sub>4</sub>Cl and Ba<sub>7</sub>(BO<sub>3</sub>)<sub>3</sub>GeO<sub>4</sub>Br are about 0.03 with the wavelength at 1064 nm.

### Conclusion

The first borogermanate halides, Ba<sub>7</sub>(BO<sub>3</sub>)<sub>3</sub>GeO<sub>4</sub>X (X=Cl, Br) have been obtained by high temperature solution method. We have discussed the structural transformation from Ba<sub>7</sub>(BO<sub>3</sub>)<sub>3</sub>SiO<sub>4</sub>X (X=Cl, Br) to Ba<sub>7</sub>(BO<sub>3</sub>)<sub>3</sub>GeO<sub>4</sub>X (X=Cl, Br). The unusual structural transformation mainly originates from rigid GeO<sub>4</sub> tetrahedra and flexible XBa<sub>6</sub> (X=Cl, Br) octahedra. We believe that this interpretation can give new insights into structural transformation. The first-principles calculations show that the calculated band gaps (4.93 eV for Ba<sub>7</sub>(BO<sub>3</sub>)<sub>3</sub>GeO<sub>4</sub>Cl and 4.91 eV for Ba7(BO3)3GeO4Br) agree well with the experiment ones (5.28 and 5.31 eV). The top of the valence band is dominated by the mixture of B 2p and O 2p states, while the Ba 5d, Ge 4s4p, B 2s and O 2s2p states dominate the bottom of the conduction band.

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### Notes and references

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- (a) P. Becker, Adv. Mater., 1998, 10, 979. (b) C. T. Chen, B. C. Wu,
  A. Jiang and G. M. You, Sci. Sin. B, 1985, 28, 235. (c) C. T. Chen,
  Y. C. Wu, A. Jiang, B. C. Wu, G. M. You, R. K. Li and S. J. Lin, J.
  Opt. Soc. Am. B, 1989, 6, 616. (d) Y. C. Wu, T. Sasaki, S. Nakai, A
  Yokotani, H. Tang and C. T. Chen, Appl. Phys. Lett., 1993, 62, 2614
- 2 (a) Y. Mori, I. Kuroda, S. Nakajima, T. Sasaki and S. Nakai, *Appl. Phys. Lett.*, 1995, **67**, 1818. (b) C. T. Chen, Y. B. Wang, B. C. Wu, K. C. Wu, W. L. Zeng and L. H. Yu, *Nature*, 1995, **373**, 322. (c) N. Ye, W. R. Zeng, J. Jiang, B. C. Wu, C. T. Chen, B. H. Feng and L. Zhang, *J. Opt. Soc. Am.*, 2000, **17**, 764. (d) L. Y. Li, G. B. Li, Y. X. Wang, F. H. Liao and J. H. Lin, *Chem. Mater.*, 2005, **17**, 4174.
- (a) H. W. Huang, L. J. Liu, S. F Jin, W. J. Yao, Y. H. Zhang and C. T. Chen, *J. Am. Chem. Soc.*, 2013, **135**, 18319. (b) H. W. Huang, Y He, Z. S. Lin, L. Kang and Y. H. Zhang, *J. Phys. Chem. C*, 2013, **117**, 22986. (c) R. H. Cong, J. L. Zhu, Y. X. Wang, T. Yang, F. H Liao, C. Q. Jin and J. H. Lin, *CrystEngComm*, 2009, **11**, 1971.
- 4 (a) S. G. Zhao, P. F. Gong, S. Y. Luo, S. J. Liu, L. N. Li, M. A. Asghar, T. Khan, M. C. Hong, Z. S. Lin and J. H. Luo, *J. Am. Chem. Soc.*, 2015, **137**, 2207. (b) S. L. Pan, J. P. Smit, M. R. Marvel, E. S. Stampler, J. M. Haag, J. Baek, P. S. Halasyamanin and K. R. Poeppelmeier, *J. Solid State Chem.*, 2008, **181**, 2087. (c) T. Pilz and M. Jansen, *Z. Anorg. Allg. Chem.*, 2011, **637**, 1.
- 5 (a) H. P. Wu, H. W. Yu, Z. H. Yang, X. L. Hou, X. Su, S. L. Pan, K. R. Poeppelmeier and J. M. Rondinelli, *J. Am. Chem. Soc.*, 2013, 135, 4215. (b) X. W. Zhang, H. W. Yu, H. P. Wu, S. L. Pan, A. Q. Jiao, B. B. Zhang and Z. H. Yang, *RSC Adv.*, 2014, 4, 13195. (c) H. Y. Li, H. P. Wu, X. Su, H. W. Yu, S. L. Pan, Z. H. Yang, Y. Lu, J. Han and K. R. Poeppelmeier, *J. Mater. Chem. C*, 2014, 2, 21704. (d) K. Wu, S. L. Pan and Z. H. Yang, *RSC Adv.*, 2015, 5, 33646.
- (a) C. Y. Bai, S. J. Han, S. L. Pan, B. B. Zhang, Y. Yang, L. Li, Z. H Yang, *RSC Adv.*, 2015, 5, 12416. (b) Y. Yang, S. L. Pan, X. Su, Y. Wang, Z. H. Yang, J. Han, M. Zhang and Z. H. Chen *CrystEngComm*, 2014, 16, 1978. (c) Z. Wang, Q. Jing, M. Zhang, X. Y. Dong, S. L. Pan and Z. H. Yang, *RSC Adv.*, 2014, 4, 54194. (d) H P. Wu, H. W. Yu, S. L. Pan, A. Q. Jiao, H. Y. Li, J. Han, K. Wu and S. J. Han, *Dalton Trans.*, 2014, 43, 4886.
- (a) A. H. Reshak, X. A. Chen, S. Auluck, H. Kamarudin, J. Chyský, A. Wojciechowski and I. V. Kityk, *J. Phys. Chem. B*, 2013, 117, 14141. (b) X. A. Chen, H. Yin, X. N. Chang, H. G. Zang and W. Xiao, *J. Solid State Chem.*, 2010, 183, 2910.
- 8 (a) C. D. McMillen, J. T. Stritzinger and J. W. Kolis, *Inorg. Chem.*, 2012, **51**, 3953. (b) C. Heyward, C. McMillen and J. W. Kol., *Inorg. Chem.*, 2012, **51**, 3956.
- 9 (a) H. Emme, M. Valldor, R. Pöttgen and H. Huppertz, *Chem Mater.*, 2005, 17, 2707. (b) J. S. Knyrim, H. Emme, M. Döbling r,
  O. Oeckler, M. Weil, and H. Huppertz, *Chem. Eur. J.*, 2008, 14,

### Page 7 of 7

6149. (c) M. J. Xia, B. Xu and R. K. Li, J. Cryst. Growth, 2014, 404, 35 A. M. Rappe, K. M. Rabe, E. Kaxiras and J. D. Joannopoulos, Phys. 65.

- 10 P. Becker, Z. Kristallogr., 2001, 216, 523.
- 11 D. A. Keszler, A. Akella, K. I. Schaffers and T. Alekel, Mater. Res. Soc. Symp. Proc., 1994, 329, 15.
- G. C. Zhang, Z. L. Liu, J. X. Zhang, F. D. Fan, Y. C. Liu and P. Z. 12 Fu, Cryst. Growth & Des., 2009, 9, 3139.
- 13 T. B. Bekker, S. V. Rashchenko, V. V. Bakakin, Y. V. Seryotkin, P. P. Fedorov, A. E. Kokha and S. Y. Stonogaa, CrystEngComm, 2012, 14. 6910.
- S. V. Rashchenko, T. B. Bekker, V. V. Bakakin, Y. V. Seryotkin, V. 14 S. Shevchenko, A. E. Kokh and S. Y. Stonoga, Cryst. Growth & Des., 2012, 12, 2955.
- 15 X. X. Lin, F. F. Zhang, S. L. Pan, H. W. Yu, F. Y. Zhang, X. Y. Dong, S. J. Han, L. Y. Dong, C. Y. Bai and Z. Wang, J. Mater. Chem. C, 2014, 2, 4257.
- (a) X. Xu, C. L. Hu, F. Kong, J. H. Zhang and J. G. Mao, Inorg. 16 Chem., 2011, 50, 8861. (b) D. B. Xiong, J. T. Zhao, H. H. Chen and X. X. Yang, Chem. Eur. J., 2007, 13, 9862.
- 17 (a) J. H. Zhang, F. Kong and J. G. Mao, Inorg. Chem., 2011, 50, 3037. (b) Y. C. Hao, C. L. Hu, X. Xu, F. Kong and J. G. Mao, Inorg. Chem., 2013, 52, 13644. (c) J. B. Parise and T. E. Gier, Chem. Mater., 1992, 4, 1065.
- Z. E. Lin, J. Zhang and G. Y. Yang, Inorg. Chem., 2003, 42, 1797. 18
- 19 (a) H. X. Zhang, J. Zhang, S. T. Zheng, G. M. Wang and G. Y. Yang, Inorg. Chem., 2004, 43, 6148. (b) D. B. Xiong, H. H. Chen, M. R. Li, X. X. Yang and J. T. Zhao, Inorg. Chem., 2006, 45, 9301.
- 20 (a) P. Becker, Cryst. Res. Technol., 2001, 2, 119. (b) B. Petermüller, L. L. Petschnig, K. Wurst, G. Heymann and H. Huppertz, Inorg. Chem., 2014, 53, 9722.
- 21 B. F. Dzhurinskii, A. B. Pobedina, K. K. Palkina and M. G. Komova, Russ. J. Inorg. Chem., 1998, 43, 1488.
- 22 M. Ihara and F. Kamei, J. Ceram. Soc. Jpn., 1980, 88, 32.
- H. P. Wu, H. W. Yu, S. L. Pan, Z. J. Huang, Z. H. Yang, X. Su and 23 K. R. Poeppelmeier, Angew. Chem. Int. Ed., 2013, 52, 3406.
- X. Xu, C. L. Hu, F. Kong, J. H. Zhang, J. G. Mao and J. L. Sun, 24 Inorg. Chem., 2013, 52, 5831.
- 25 T. Berger and K. J. Range, Z. Naturforsch. B, 1996, 51, 172.
- 26 C. Heyward, C. D. McMillen and J. J. Kolis, Solid State Chem., 2013, 203.166
- 27 V. R. Samygina, E. A. Genkina, B. A. Maksimov and N. I. Leonyuk, Kristallografiya, 1993, 38, 61.
- 28 A. A. Kaminskii, B. V. Mill, E. L. Belokoneva and A. V. Butashin, Izv. Akad. Nauk SSSR, 1990, 26, 1105.
- 29 SAINT, Version 7.60A; Bruker Analytical X-ray Instruments, Inc. Madison, WI, 2008.
- 30 A. L. Spek, J. Appl. Cryst., 2003, 36, 7,
- (a) P. Kubelka and F. Z. Munk, Tech. Phys., 1931, 12, 593. (b) J. 31 Tauc, Mater. Res. Bull., 1970, 5, 721.
- 32 S. J. Clark, M. D. Segall, C. J. Pickard, P. J. Hasnip, M. J. Probert, K. Rrfson and M. C. Payne, Z. Kristallogr., 2005, 220, 567.
- 33 J. P. Perdew, K. Burke and M. Ernzerhof, Phys. Rev. Lett., 1996, 77, 3865.
- 34 J. S. Lin, A. Qteish, M. C. Payne and V. Heine, Phys. Rev. B., 1993, 47, 4174.

- Rev. B, 1990, 41, 1227.
- 36 H. J. Monkhorst and J. D. Pack, Phys. Rev. B, 1976, 13, 5188.
- 37 E. D. Palik, Handbook of Optical Constants of Solids. Orlando. Academic Press: New York, 1985.
- 38 X. Y. Dong, H. P. Wu, Y. J. Shi, H. W. Yu, Z. H. Yang, B. B. Zhang Z. H. Chen, Y. Yang, Z. J. Huang, S. L. Pan and Z. X. Zhou, Chem Eur. J., 2013, 19, 7338.
- 39 R. W. Godby, M. Schluter and L. J. Sham, Phys. Rev. B, 1987, 36 6497.
- 40 F. Wooten, Optical Properties of Solid, Academic, New York, 1972.

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