

RSC Advances



This is an *Accepted Manuscript*, which has been through the Royal Society of Chemistry peer review process and has been accepted for publication.

Accepted Manuscripts are published online shortly after acceptance, before technical editing, formatting and proof reading. Using this free service, authors can make their results available to the community, in citable form, before we publish the edited article. This *Accepted Manuscript* will be replaced by the edited, formatted and paginated article as soon as this is available.

You can find more information about *Accepted Manuscripts* in the [Information for Authors](#).

Please note that technical editing may introduce minor changes to the text and/or graphics, which may alter content. The journal's standard [Terms & Conditions](#) and the [Ethical guidelines](#) still apply. In no event shall the Royal Society of Chemistry be held responsible for any errors or omissions in this *Accepted Manuscript* or any consequences arising from the use of any information it contains.

1 **Effect of A/B-site Substitution on Oxygen Production Performance of**
2 **Strontium Cobalt Based Perovskites for CO₂ Capture Application**

3 Qiuwan Shen ^{a,b}, Ying Zheng ^{a,*}, Cong Luo ^a, Ning Ding ^c, Chu-Guang Zheng ^a, Marcus Thern ^{b,*}

4 ^a State Key Laboratory of Coal combustion, Huazhong University of Science & Technology, Wuhan 430074, China

5 ^b Department of Energy Sciences, Lund University, P.O. Box 118, SE-221 00 Lund, Sweden

6 ^c Energy Research Institute of Shandong Academy of Sciences, Jinan 250014, China

7 **Abstract:** Oxy-fuel combustion is one of the proposed technologies which have the
8 potential to achieve a zero CO₂ emission. Strontium cobalt based perovskite oxygen
9 carriers are promising materials for air separation with a high selectivity for oxygen.
10 And these perovskites can produce an oxygen enriched carbon dioxide stream for
11 oxyfuel combustion process. The relative low oxygen production yield may be a
12 drawback of this type of materials for this technology. This paper presents an effective
13 approach by A/B-site substitution to improve the oxygen production performance of
14 the perovskites. In this study, a series of different A/B-site substituted SrCo_{0.8}Fe_{0.2}O_{3-δ}
15 were prepared by an EDTA-citrate sol-gel combustion synthesis method. Fixed-bed
16 experiments and TGA measurements were performed to study the effects of A/B-site
17 substitution on cyclic oxygen adsorption/desorption performance of the synthesized
18 samples. The experimental results indicate that the oxygen desorption amount of
19 different A-site substituted perovskites decrease in the order of BaCo_{0.8}Fe_{0.2}O_{3-δ} >
20 Ba_{0.5}Sr_{0.5}Co_{0.8}Fe_{0.2}O_{3-δ} > SrCo_{0.8}Fe_{0.2}O_{3-δ} > Sr_{0.5}Ca_{0.5}Co_{0.8}Fe_{0.2}O_{3-δ} > MgCo_{0.8}Fe_{0.2}O_{3-δ}.
21 Moreover, B-site substitution by different transition metal ions can significantly
22 modified oxygen adsorption capacity and oxygen desorption performance of
23 SrCo_{0.8}Fe_{0.2}O_{3-δ}. Furthermore, oxygen desorption performance can be improved when

Corresponding author. Tel.: +86 27-63120550. +46 462224112

E-mail address: Y.Zheng1967@gmail.com (Y. Zheng). Marcus.Thern@energy.lth.se (M.Thern)

24 Fe ions of the perovskite $\text{SrCo}_{0.8}\text{Fe}_{0.2}\text{O}_{3-\delta}$ were substituted by Zr, Cr, Zn, Ni ions.

25 **Key words:** Perovskite-type; $\text{SrCo}_{0.8}\text{Fe}_{0.2}\text{O}_{3-\delta}$; A/B-site substitution; Oxygen
26 adsorption/desorption

27

28

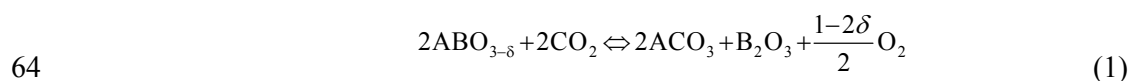
29 1. Introduction

30 It is acknowledged that the combustion of fossil fuels contributes to the emission
31 of CO_2 into the atmosphere, which in turn causes global warming [1-3]. Oxy-fuel
32 combustion is one of the proposed technologies which have the potential to achieve a
33 zero CO_2 emission. Oxy-fuel combustion is well known as the O_2/CO_2 recycle
34 combustion process, in which oxygen is fed to the combustion chamber, and a major
35 part of the CO_2 rich exhaust gas is recycled back to maintain the combustion
36 temperature. Although the oxy-fuel combustion process offers obvious advantages
37 over regular combustion processes, one of the key barriers to implementation of
38 oxy-combustion, however, is the cost of producing the oxygen. Therefore, significant
39 reduction in the cost of oxygen production is a key requirement in making the
40 oxy-fuel combustion power plant a viable future option when carbon dioxide capture
41 becomes a necessity [4-5].

42 Perovskite-type metal oxides have been receiving increasing attention for a wide
43 variety of applications, such as a component of capacitors, microwave technology,
44 varistors, electrodes, and immobilization of nuclear wastes, as well as being catalysts
45 for oxidation and hydrogenation [6-9]. Perovskite-type metal oxides have the general
46 formula ABO_3 , where A-site can be an alkali, alkaline earth, rare earth or other large ion,

47 and B-site can be transitional metal ion. If A- and/or B-sites are substituted with lower
48 valent cations, in order to maintain charge neutrality, some oxygen ions are removed
49 from the lattice thereby creating oxygen vacancies [10]. Perovskite metal oxides have
50 excellent mixed ionic-electronic conductivity and exhibit extremely high selectivity
51 for oxygen due to the existence of oxygen vacancies. Perovskites are highly selective
52 for oxygen over nitrogen or other non-oxygen containing gases which makes them
53 ideal candidates as oxygen permeable membrane. However, this technology faces
54 major challenges in regard to the scale-up manufacturing and stability of the
55 membranes [11].

56 Lin et al. proposed a new process of producing pure O₂ or O₂/CO₂ gas streams by
57 using a perovskite-type oxygen carrier in oxy-fuel combustion [11]. A general scheme
58 of such a process is shown in Figure 1: (1) oxygen adsorption in air reactor; (2)
59 oxygen desorption in CO₂ reactor. For oxygen adsorption step, air is used as feed gas
60 to saturate the perovskite oxygen carrier with O₂; while in the oxygen desorption step,
61 using CO₂ as a sweep gas to desorb O₂ from the perovskite to produce an O₂-enriched
62 CO₂ flue gas stream. The reversible adsorption/desorption processes based on the
63 perovskite-type oxygen carrier is described as below [12]:



65 Relative low oxygen desorption amount may be a major drawback of this type of
66 materials for this technology. This issue may cause challenges to achieving a high
67 efficiency of O₂ product amount in practical application. Therefore, improving the

68 oxygen desorption performance is of great importance to develop the promising
69 perovskite for oxygen production.

70 A common approach to improve the properties of perovskite-type oxygen carrier
71 is through A/B site substitution in the metal oxides. $\text{SrCo}_{1-x}\text{Fe}_x\text{O}_{3-\delta}$ is a promising
72 perovskite that has drawn considerable attention because of its high oxygen
73 permeability [13-14].

74 This paper is an extension of our previous study [14]. We found that the oxygen
75 production performance of $\text{SrCo}_{1-x}\text{Fe}_x\text{O}_{3-\delta}$ is improved by Co doping. $\text{SrCo}_{0.8}\text{Fe}_{0.2}\text{O}_{3-\delta}$
76 has the best oxygen production property among the $\text{SrCo}_{1-x}\text{Fe}_x\text{O}_{3-\delta}$ ($x= 0.2, 0.4, 0.6,$
77 0.8) and multiple cycles demonstrated that $\text{SrCo}_{0.8}\text{Fe}_{0.2}\text{O}_{3-\delta}$ also displays high stability
78 and regeneration capacity. The present work was focused on understanding the effects
79 of A/B site substitution in $\text{SrCo}_{0.8}\text{Fe}_{0.2}\text{O}_{3-\delta}$ perovskite on its oxygen production
80 performance.

81 2. Experimental

82 2.1. Materials and Preparation Method

83 Perovskite samples used in this study were synthesized by an EDTA-citrate
84 sol-gel combustion synthesis [15-20]. The starting materials were $\text{Sr}(\text{NO}_3)_2 \cdot 4\text{H}_2\text{O}$,
85 $\text{Co}(\text{NO}_3)_2 \cdot 6\text{H}_2\text{O}$, $\text{Fe}(\text{NO}_3)_3 \cdot 9\text{H}_2\text{O}$, citric acid and ethylenediamine-tetraacetic acid
86 (EDTA), all of which were of analytical purity. As for synthesizing 0.1 mol of
87 $\text{SrCo}_{0.8}\text{Fe}_{0.2}\text{O}_{3-\delta}$ powders, the detail procedures are described as follows: First, 0.1 mol
88 of EDTA was mixed with 125 ml of 13 N NH_4OH solutions to make a NH_3 -EDTA
89 solution. Then, 0.05 mol of $\text{Sr}(\text{NO}_3)_2 \cdot 4\text{H}_2\text{O}$, 0.04 mol of $\text{Co}(\text{NO}_3)_2 \cdot 6\text{H}_2\text{O}$ and 0.01

90 mol of $\text{Fe}(\text{NO}_3)_3 \cdot 9\text{H}_2\text{O}$ were dissolved to the NH_3 -EDTA solution. The solution was
91 mixed and stirred, then 0.15 mol of citric acid was added to the mixed solution. The
92 mole ratios of EDTA: citric acid: total metal ions were 1:1.5:1. Then the precursor
93 solution was heated and stirred at 70 °C until it gelled. The resulting viscous gel was
94 dried at 105°C for 24 h and self-ignited at 400 °C or 4 h to burn out the organic
95 compounds. Finally, the black ash was sintered at 850 °C for 20 h and ground into
96 fine powders for characterization.

97 *2.2. Fixed-bed experiments and TGA measurements*

98 Oxygen adsorption/desorption experiments were performed in a fixed-bed
99 reactor system as shown in Figure 2. It consisted of a gas feeding system, a tube
100 furnace with a quartz reactor, a gas analyzer (Gasboard 3100) and a computerized
101 data-acquisition system. Oxygen concentration during desorption process were
102 recorded to investigate the oxygen production performance of perovskite powders.
103 About 1.0 g of powders was packed in the middle of the quartz reactor. Air and CO_2
104 were respectively used as the feed gas for adsorption step and sweep gas for
105 desorption step.

106 In the adsorption step, the powders were heated to a desired adsorption
107 temperature in a flow of air in 1 atm at flow rate of 200 ml/min for 10 min. The
108 adsorption step was followed by the desorption step with a switch of the sweep gas
109 from Air to CO_2 stream at a flow rate of 50 ml/min, and setting the temperature to the
110 predetermined desorption temperature. The desorption step was stopped when the O_2
111 concentration nearly dropped to zero. Then the CO_2 stream was switched to air to start

112 the next cycle of oxygen adsorption and desorption processes.

113 The detailed oxygen adsorption properties of perovskite were also studied by
114 thermogravimetric analysis (DSC/TGA, STA449C) in a temperature range
115 25-850 °C at a constant gas flow rate of 80 ml/min and the heating rate was
116 15 °C/min. For a typical TGA measurement, approximately 10 mg of dry powder was
117 placed in an alumina crucible. The sample was heated to the designed temperature in
118 1 atm atmosphere (air or N₂) and until a stable weight was observed from the TGA
119 curve. The feed gas was switched quickly between the air and N₂. Weight change was
120 recorded continuously by TGA analysis.

121 3. Results and discussions

122 3.1 Effects of A-site substitution on oxygen adsorption/desorption performance

123 Figure 3 compares the oxygen adsorption breakthrough curves of different A-site
124 total/partial substituted SrCo_{0.8}Fe_{0.2}O_{3-δ} perovskite samples denoted as SrCo_{0.8}Fe_{0.2}O_{3-δ}
125 (SCF), BaCo_{0.8}Fe_{0.2}O_{3-δ}(BCF), Ba_{0.5}Sr_{0.5}Co_{0.8}Fe_{0.2}O_{3-δ} (BSCF), Sr_{0.5}Ca_{0.5}Co_{0.8}Fe_{0.2}O_{3-δ}
126 (SCCF) and MgCo_{0.8}Fe_{0.2}O_{3-δ} (MCF) at the air flow rate of 200 ml/min at 850 °C. A
127 standard oxygen adsorption breakthrough curve was introduced for comparison. The
128 results show that the breakthrough curves are similar in shape with a sharp increase
129 after a breakthrough time (50-100 s), and soon afterward all the curves are
130 characterized by a long tail. For example, for BSCF, no outgoing oxygen was detected
131 by the oxygen sensor before 100 s, and after 150 s, the outlet oxygen concentration
132 was equal to the feed gas concentration (20.9 %). Figure 3 indicates that MCF sample
133 have the shortest breakthrough time (oxygen adsorption capacity), on the contrary,

134 BSCF have the longest breakthrough time.

135 Figure 4 depicts the comparison of oxygen desorption curves of SCF with different
 136 A-site total/partial substitution. It is clear that A-site total/partial substitution has
 137 significant influence on oxygen desorption properties of SCF. Oxygen desorption
 138 amount of each desorption process shown in Figure 4 for different samples are given
 139 in Table 1. The oxygen desorption amount was calculated by the integral scheme
 140 based on the obtained oxygen concentration distribution. The following equation can
 141 be used:

$$142 \quad m_{O_2} = \frac{\Sigma C_{O_2} \times F_{out} \times M_{O_2}}{V_m \times m} \quad (2)$$

143 where ΣC_{O_2} is the integration of the entire oxygen concentration during desorption
 144 and F_{out} (ml/s) is the flow rate of desorption effluent. We suppose that $F_{out} \approx F_{CO_2}$,
 145 M_{O_2} (g/mol) is the molecular weight of O_2 , m (g) is the mass of perovskite sample,
 146 and m_{O_2} (g/g-sample) is the oxygen desorption amount for 1 g of perovskite sample.

147 The results show that oxygen desorption amount is in the order
 148 BCF>BSCF>SCF>SCCF>MCF. It is also can be seen from Figure 4 that the slope of
 149 the curves of BCF and BSCF are bigger than others which demonstrate the oxygen
 150 desorption rates of BCF and BSCF are faster than the rest. The outlet oxygen
 151 concentration of BCF reached the maximum value of 25 % after 150 s. On the other
 152 hand, substitution of Sr^{2+} with Ca^{2+}/Mg^{2+} reduces the oxygen desorption amount for
 153 $SrCo_{0.8}Fe_{0.2}O_{3-\delta}$. The lower desorption amount may be related to the smaller ionic
 154 radius, as ionic radius is in the order $Ba(1.75\text{\AA}) > Sr(1.58\text{\AA}) > Ca(1.48\text{\AA}) > Mg(1.03\text{\AA})$,

155 the bigger ionic radius of Ba resulting in the increase of lattice volume and leading
 156 contribution to oxygen ions transition in the crystal. Therefore, though high
 157 temperature oxygen (β -oxygen) desorption is usually related to the B-site substitution,
 158 it is also affected by A-site substitution [21]. Moreover, for a fixed B-site
 159 composition, A-site ionic with the same valency but different ionic radius affects the
 160 oxygen desorption property.

161 **Table 1 oxygen desorption amount of a unit mass of A(A=Sr, Ba, Mg,**
 162 **Ca) $\text{Co}_{0.8}\text{Fe}_{0.2}\text{O}_{3-\delta}$**

Samples	BCF	BSCF	SCF	SCCF	MCF
Time range to collect the product in desorption ($t_1 \sim t_2$)	128s-400s	126s-400s	143s-400s	143s-400s	135s-400s
Oxygen desorption amount (mg O_2 /g·sample)	45.5	35.8	33.0	31.5	10.2

163

164 3.2 Effects of fixed-bed operation condition on oxygen production performance

165 3.2.1 Desorption temperature

166 Figure 5 presents the desorption curves of $\text{BaCo}_{0.8}\text{Fe}_{0.2}\text{O}_{3-\delta}$ at different
 167 desorption temperatures from 750 °C to 950 °C (with the same adsorption temperature
 168 at 850 °C). As shown in Figure 5, it is clear that 850 °C is the ideal desorption
 169 temperature for $\text{BaCo}_{0.8}\text{Fe}_{0.2}\text{O}_{3-\delta}$. It is known that the carbonation reaction between
 170 CO_2 and $\text{BaCo}_{0.8}\text{Fe}_{0.2}\text{O}_{3-\delta}$ is a typical gas-solid reaction. The higher desorption
 171 temperature contributes to the higher carbonation rates which can produce more
 172 oxygen in the desorption process. However, when the temperature is higher than
 173 900 °C, the oxygen desorption performance of $\text{BaCo}_{0.8}\text{Fe}_{0.2}\text{O}_{3-\delta}$ begins to fall. It

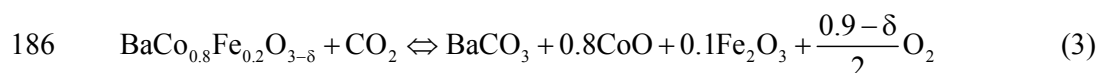
174 might be caused by the decrease of the sample surface area during the carbonation
175 reaction. Therefore an optimum oxygen desorption temperature exist in the oxygen
176 desorption process.

177

178 3.2.2 CO₂ partial pressure

179 In the carbonation reaction of perovskite with CO₂, except for desorption
180 temperature, partial pressure in gas phase also plays a key role in the desorption
181 process. Figure 6 shows desorption curves at different CO₂ partial pressure during the
182 desorption process of BaCo_{0.8}Fe_{0.2}O_{3-δ} at 850 °C. As shown in Figure 6, a higher CO₂
183 partial pressure leads to a larger amount of oxygen produces in the desorption process.

184 The carbonation reaction of BaCo_{0.8}Fe_{0.2}O_{3-δ} perovskite-type oxygen carriers with
185 CO₂ is described as follows:



187 It is clear that the pressure dependence of carbonation reaction for BaCo_{0.8}Fe_{0.2}O_{3-δ} is
188 consistent with the known properties of gas-solid reaction [13]. The mass-action law
189 demonstrates that the carbonation reaction rate (oxygen desorption rate) increase with
190 CO₂ partial pressure.

191 3.3 Effects of B-site substitution on oxygen adsorption/desorption performance

192 Effects of B-site substitution by different transition metal ions on oxygen
193 adsorption/desorption performance of SrCo_{0.8}B_{0.2}O₃ (B=Ni,Zn,Zr,Cr,Cu,Fe) were
194 studied. Oxygen adsorption capacities of all the samples were investigated by TGA

195 analysis. Based on the weight change measurement by TGA, the oxygen adsorption
 196 capacities under a given temperature can be calculate by:

$$197 \quad q = \frac{1}{2M_o} \frac{W_{\text{air}} - W_{N_2}}{W_0} \quad (4)$$

198 where q is the oxygen adsorption capacity (mol/g); M_o is the molecular weight of O;
 199 $W_{\text{air}}(\text{g})$ and $W_{N_2}(\text{g})$ is the equilibrium weight in air and N_2 at particular temperature,
 200 respectively; W_0 is the weight of the sample at initial condition (110 °C in 1 atm air).

201 Equilibrium weight (mg) of $\text{SrCo}_{0.8}\text{B}_{0.2}\text{O}_{3-\delta}$ (B=Fe, Ni, Cu, Zn, Cr, Zr) at
 202 different conditions were listed in Table 2. Figure 7 compares the oxygen adsorption
 203 capacity which was calculated from data in Table 2. The results show that the
 204 substitution of Fe by Zr, Cr and Cu in $\text{SrCo}_{0.8}\text{B}_{0.2}\text{O}_{3-\delta}$ increases the oxygen adsorption
 205 capacity, the adsorption capacity decreases in the order $\text{SrCo}_{0.8}\text{Zr}_{0.2}\text{O}_{3-\delta}$ (Sr-Co-Zr)>
 206 $\text{SrCo}_{0.8}\text{Cr}_{0.2}\text{O}_{3-\delta}$ (Sr-Co-Cr)> $\text{SrCo}_{0.8}\text{Cu}_{0.2}\text{O}_{3-\delta}$ (Sr-Co-Cu)> $\text{SrCo}_{0.8}\text{Fe}_{0.2}\text{O}_{3-\delta}$ (Sr-Co-Fe)>
 207 $\text{SrCo}_{0.8}\text{Ni}_{0.2}\text{O}_{3-\delta}$ (Sr-Co-Ni)> $\text{SrCo}_{0.8}\text{Zn}_{0.2}\text{O}_{3-\delta}$ (Sr-Co-Zn).

208 Generally speaking, substitution of B-site by higher valent cations, such as Zr,Cr
 209 results in less adsorption capacity because the charge neutrality rule restrains the
 210 formation of oxygen vacancy. Since the results contradicts the results of this study, it
 211 may appear that except for oxygen vacancy of perovskite, the oxygen adsorption
 212 capacity also relates to the surface adsorption activity.

213 **Table 2** Equilibrium weight (mg) of $\text{SrCo}_{0.8}\text{B}_{0.2}\text{O}_{3-\delta}$ (B=Fe,Ni,Cu,Zn,Cr,Zr) at initial
 214 conditions, and 850 °C in air and N_2 , respectively.

Samples	Sr-Co-Ni	Sr-Co-Zn	Sr-Co-Zr	Sr-Co-Cr	Sr-Co-Cu	Sr-Co-Fe
---------	----------	----------	----------	----------	----------	----------

$W_{\text{air(mg)}}$	10.500	10.400	8.969	9.319	9.372	9.655
$W_{\text{N2(mg)}}$	10.465	10.370	8.907	9.259	9.314	9.615
$W_{\text{O(mg)}}$	10.67	10.64	9.789	9.646	9.563	9.867

215

216 Figure 8 shows a comparison of cyclic oxygen desorption performance for
 217 $\text{SrCo}_{0.8}\text{B}_{0.2}\text{O}_{3-\delta}$ (B=Fe, Ni, Cu, Zn, Cr, Zr). It indicates that substitution of Fe in B-site
 218 with different transition metal ions has significantly effects on the oxygen desorption
 219 performance. The substitution of Fe by by Zr, Ni, Zn and Cr in $\text{SrCo}_{0.8}\text{Fe}_{0.2}\text{O}_{3-\delta}$
 220 enhanced the oxygen desorption amount. On the other hand, substitution of Fe by Cu
 221 reduced the oxygen desorption amount which results in the lowest cyclic ability for
 222 $\text{SrCo}_{0.8}\text{Cu}_{0.2}\text{O}_{3-\delta}$ among the samples as showed in Figure 8. These results indicate that
 223 substitution of Fe with Zr and Cr in $\text{SrCo}_{0.8}\text{Fe}_{0.2}\text{O}_{3-\delta}$ enhanced both oxygen adsorption
 224 capacity and desorption amount; substitution with Zn and Ni increased the oxygen
 225 desorption amount but adversely reduced the oxygen adsorption capacity. Considering
 226 both the adsorption/desorption, the performance of $\text{SrCo}_{0.8}\text{Zr}_{0.2}\text{O}_{3-\delta}$ was the best
 227 followed by $\text{SrCo}_{0.8}\text{Cr}_{0.2}\text{O}_{3-\delta}$ among the different composition of $\text{SrCo}_{0.8}\text{B}_{0.2}\text{O}_{3-\delta}$ with
 228 different B-site transition metal iron substitution.

229

230 4. Conclusions

231 In this study, oxygen production performance of a new series strontium cobalt based
 232 perovskites with A/B-site substitution was investigated. The following conclusions
 233 can be drawn from this study:

234 1. Oxygen desorption amount of different A-site substituted perovskites decrease in

235 the order of $\text{BaCo}_{0.8}\text{Fe}_{0.2}\text{O}_{3-\delta} > \text{Ba}_{0.5}\text{Sr}_{0.5}\text{Co}_{0.8}\text{Fe}_{0.2}\text{O}_{3-\delta} > \text{SrCo}_{0.8}\text{Fe}_{0.2}\text{O}_{3-\delta} >$
236 $\text{Sr}_{0.5}\text{Ca}_{0.5}\text{Co}_{0.8}\text{Fe}_{0.2}\text{O}_{3-\delta} > \text{MgCo}_{0.8}\text{Fe}_{0.2}\text{O}_{3-\delta}$. The lower desorption amount is related
237 to the smaller ionic radius, the bigger ionic radius of Ba resulting in the increase
238 of lattice volume and leading contribution to oxygen ions transition in the crystal.

239 2. B-site substitution by different transition metal ions can significantly modified
240 oxygen adsorption capacity and oxygen desorption performance of
241 $\text{SrCo}_{0.8}\text{Fe}_{0.2}\text{O}_{3-\delta}$. Fe substituted by Zr, Cr and Cu in $\text{SrCo}_{0.8}\text{B}_{0.2}\text{O}_{3-\delta}$ increasing the
242 oxygen adsorption capacity, it may appear that except for oxygen vacancy of
243 perovskite, the oxygen adsorption capacity is also related to the surface adsorption
244 activity. Moreover, substitution of Fe ion in $\text{SrCo}_{0.8}\text{Fe}_{0.2}\text{O}_{3-\delta}$ with Zr, Ni, Zn and Cr
245 ions all enhanced the oxygen desorption amount. On the contrary, substitution a
246 Fe ion with Cu reduced the oxygen desorption amount. Considering both the
247 adsorption/desorption, the performance of $\text{SrCo}_{0.8}\text{Zr}_{0.2}\text{O}_{3-\delta}$ was the best followed
248 by $\text{SrCo}_{0.8}\text{Cr}_{0.2}\text{O}_{3-\delta}$ among the different composition of $\text{SrCo}_{0.8}\text{B}_{0.2}\text{O}_{3-\delta}$ with
249 different B-site transition metal iron substitution.

250

251

252 **Acknowledgment**

253 The authors acknowledge the financial supports of National Natural Science
254 Foundation of China (No.51276078), and the National Basic Research Program of
255 China (No.2011CB707301). The authors also gratefully acknowledge the
256 financial supports from China Scholarship Council

257

258

259

260 **References**

- 261 1. Q.W. Shen, Y. Zheng, C. Luo, and C.G. Zheng. Development and
262 characterization of $\text{Ba}_{1-x}\text{Sr}_x\text{Co}_{0.8}\text{Fe}_{0.2}\text{O}_{3-\delta}$ perovskite for oxygen production in
263 oxyfuel combustion system, *Chem. Eng. J.* 2014, **225**, 462-470.
- 264 2. B. W. Wang, G. Xiao, X. Y. Song, H.B. Zhao, C.G.Zheng, Chemical looping
265 combustion of high-sulfur coal with NiFe_2O_4 -combined oxygen carrier, *J Therm.*
266 *Anal. Calorim.* 2014,**118**, 1593-1602.
- 267 3. C. Luo, Y. Zheng, J. J. Yin, C. L. Qin, N. Ding, C. G. Zheng, B. Feng, Effect of
268 sulfation during oxy-fuel calcination stage in calcium looping on CO_2 capture
269 performance of CaO-based sorbent, *Energy Fuels* 2013,**27**,1008-1014.
- 270 4. E. Croiset, K. V. Thambimuthu, NO_x and SO_2 emissions from O_2/CO_2 recycle
271 coal combustion, *Fuel* 2001, **80**, 2117-2121.
- 272 5. B.J.P. Buhre, L. K. Elliott, C. D. Sheng, R. P. Gupta, T. F.Wall, Oxy-fuel
273 combustion technology for coal-fired power generation, *Prog. Energy Combust.*
274 *Sci.* 2005, **31**, 283-307.
- 275 6. M. Sun, X. W. Chen and L. Hong. Leveraging the A-site Ba^{2+} - Sr^{2+} ratio in the
276 designated perovskite to enhance oxygen transport and structural/interfacial
277 stability. *RSC Adv.*, 2014, **4**, 5618-5625.

- 278 7. H.J. Zhan, F. Li, P. Gao, N. Zhao, F.K. Xiao, W. Wei and Y.H. Sun, Influence of
279 element doping on La–Mn–Cu–O based perovskite precursors for methanol
280 synthesis from CO₂/H₂. *RSC Adv.*, 2014, **4**, 48888-48896.
- 281 8. K. Zhang, J.K. Sunarso, Z.P. Shao, W. Zhou, C.H. Sun, S.B. Wang and S.M. Liu.
282 Research progress and materials selection guidelines on mixed conducting
283 perovskite-type ceramic membranes for oxygen production. *RSC Adv*, 2011, **1**,
284 1661–1676.
- 285 9. C. Zhang, R. Ran, G.H. Pham, K. Zhang, J. Liu and S. M. Liu. Ce_{0.9}Gd_{0.1}O_{2-δ}
286 membranes coated with porous Ba_{0.5}Sr_{0.5}Co_{0.8}Fe_{0.2}O_{3-δ} for oxygen separation.
287 *RSC Adv*, 2015, **5**, 5379-5386.
- 288 10. N. O. Vitoriano , C. B. López , I. R. Larramendi , R. Knibbe , K. Thydén , A.
289 Hauch , P. Holtappels , T. Rojo , Optimizing solid oxide fuel cell cathode
290 processing route for intermediate temperature operation, *Applied Energy*
291 2014,**104**, 984-991.
- 292 11. S. Guntuka, S. Banerjee, S. Farooq, M. P. Srinivasan, A- and B-site substituted
293 lanthanum cobaltite perovskite as high temperature oxygen sorbent, *Ind. Eng.*
294 *Chem. Res.* 2008, **47**, 154-162.
- 295 12. Z. Yang, Y. S. Lin, High-temperature sorption process for air separation and
296 oxygen removal, *Ind. Eng. Chem. Res.* 2002, **41**, 2775-2784.

- 297 13. Q. Yang, Y. S. Lin, Improved sorbent for high-temperature production of
298 oxygen-enriched carbon dioxide stream. *Ind. Eng. Chem. Res.* 2007, **46**,
299 6025-6031.
- 300 14. Z. B. Rui, J. J. Ding, Y. D. Li, $\text{SrCo}_{0.8}\text{Fe}_{0.2}\text{O}_{3-\delta}$ sorbent for high-temperature
301 production of oxygen-enriched carbon dioxide stream, *Fuel* 2010, **89**, 1429-1434.
- 302 15. Q. W. Shen, Y. Zheng, C. Luo, and C. G. Zheng, Characteristics of $\text{SrCo}_{1-x}\text{Fe}_x\text{O}_{3-\delta}$
303 perovskite powders with improved O_2/CO_2 production performance for oxyfuel
304 combustion, *Bull. Korean. Chem. Soc.* 2014, **35**, 1613-1616.
- 305 16. A. A. Yaremchenko, E. V. , A. V. Kovalevsky et al., Stability, oxygen permeability
306 and chemical expansion of $\text{Sr}(\text{Fe},\text{Al})\text{O}_{3-\delta}$ - and studies of $\text{Sr}(\text{Co},\text{Fe})\text{O}_{3-\delta}$ - based
307 membranes, *Solid State Ionics* 2011, **192**, 259-268.
- 308 17. H. Patra, S. K. Rout, S. K. Pratihara, S. Bhattacharya, Effect of process parameters
309 on combined EDTA–citrate synthesis of $\text{Ba}_{0.5}\text{Sr}_{0.5}\text{Co}_{0.8}\text{Fe}_{0.2}\text{O}_{3-\delta}$ perovskite,
310 *Powder Technol.* 2011, **209**, 98-104.
- 311 18. Z. Shao, S. M. Haile, A high-performance cathode for the next generation of
312 solid-oxide fuel cells, *Nature* 2004, **431**, 170-173.
- 313 19. H. Wang, C. Tablet, A. Feldhoff, J. Caro, Investigation of phase structure,
314 sintering, and permeability of perovskite-type $\text{Ba}_{0.5}\text{Sr}_{0.5}\text{Co}_{0.8}\text{Fe}_{0.2}\text{O}_{3-\delta}$ membranes,
315 *J. Membrane Sci.* 2005, **262**, 20-26.

- 316 | 20. S. Lee, Y. Lima, E. A. Lee, $\text{Ba}_{0.5}\text{Sr}_{0.5}\text{Co}_{0.8}\text{Fe}_{0.2}\text{O}_{3-\delta}$ (BSCF) and
317 | $\text{La}_{0.6}\text{Ba}_{0.4}\text{Co}_{0.2}\text{Fe}_{0.8}\text{O}_{3-\delta}$ (LBCF) cathodes prepared by combined citrate-EDTA
318 | method for IT-SOFCs, *J. Power Sources* 2006,**157**, 848-854.
- 319 | 21. W. Zhou, Z. Shao, R. Ran, Z. Chen, High performance electrode for
320 | electrochemical oxygen generator cell based on solid electrolyte ion transport
321 | membrane, *Electrochim. Acta.* 2007, **52**, 6297-6303.
- 322 | 22. M.A.Pena, J.L. Fierro, Chemical structures and performance of perovskite oxides.
323 | *Chem. Rev.* 2001, **101**, 1981-2017.
- 324 |
325 |
326 |
327 |
328 |
329 |
330 |
331 |
332 |
333 |
334 |
335 |
336 |

337 **Figure Captions:**

338

339 **Figure 1.** Scheme for O₂/CO₂ production integrated oxy-fuel combustion process.

340

341 **Figure 2.** Schematic diagram of fixed-bed reaction system. (1) Gas cylinder; (2) valve;
342 (3) flow controller; (4) thermocouple; (5) temperature controller; (6) quartz reactor; (7)
343 horizontal tube furnace; (8) gas analyzer; (9) data acquisition system.

344

345 **Figure 3.** Oxygen adsorption curves for A (A=Sr, Ba, Mg, Ca) Co_{0.8}Fe_{0.2}O_{3-δ}

346

347 **Figure 4.** Oxygen desorption curves for A (A=Sr, Ba, Mg, Ca) Co_{0.8}Fe_{0.2}O_{3-δ}

348

349 **Figure 5.** Oxygen desorption curves of BaCo_{0.8}Fe_{0.2}O_{3-δ} at different desorption
350 temperatures

351

352 **Figure 6.** Oxygen desorption curves of BaCo_{0.8}Fe_{0.2}O_{3-δ} at different CO₂ partial
353 pressure at 850 °C

354

355 **Figure 7.** Comparison of oxygen adsorption capacity for SrCo_{0.8}B (B=Fe, Ni, Cu, Zn,
356 Cr, Zr) _{0.2}O_{3-δ}

357

358 **Figure 8.** Comparison of Oxygen desorption amount for SrCo_{0.8}B (B=Fe, Ni, Cu, Zn,
359 Cr, Zr) _{0.2}O_{3-δ} with the number of cycles.

360

361

362

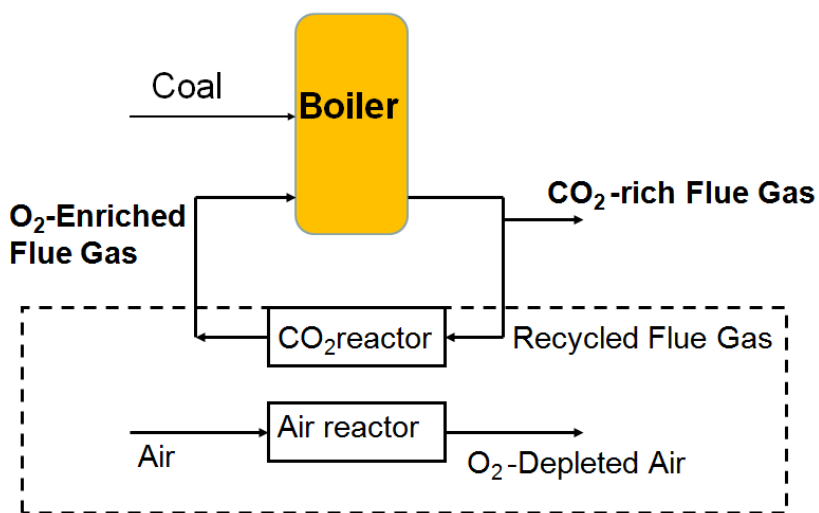
363

364

365

366 | **Figure 1.**

367



368

369

370

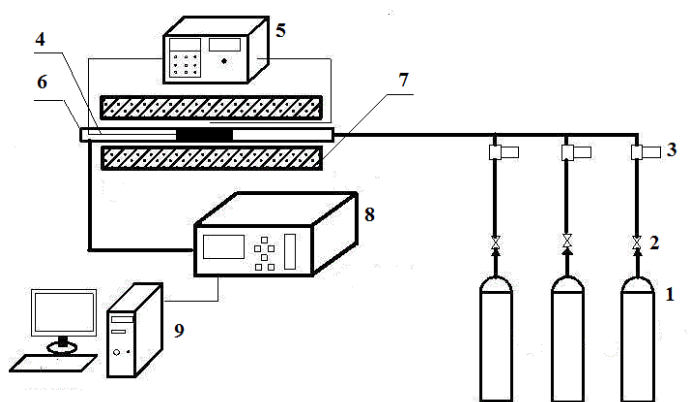
371

372

373

374 | **Figure 2.**

375



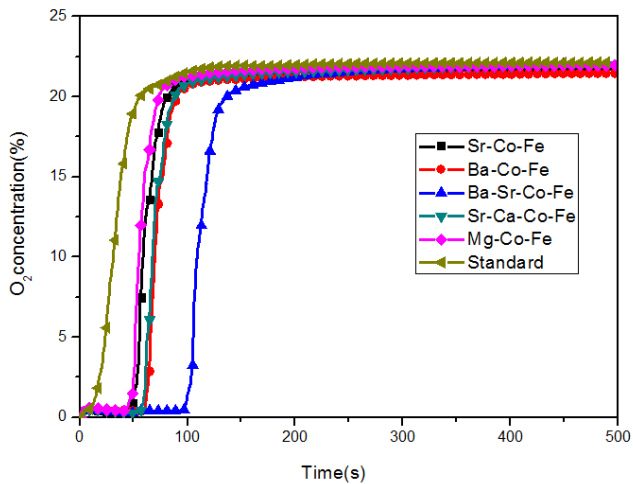
376

377

378

379 **Figure 3.**

380



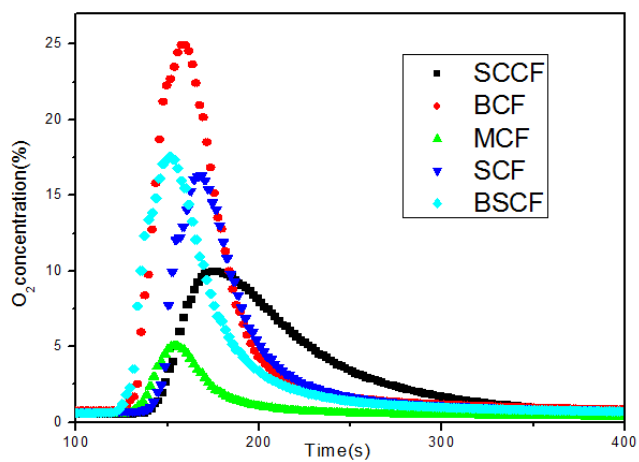
381

382

383

384 **Figure 4.**

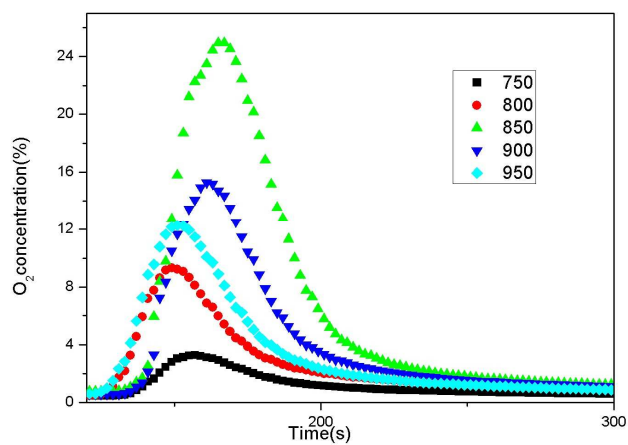
385



386

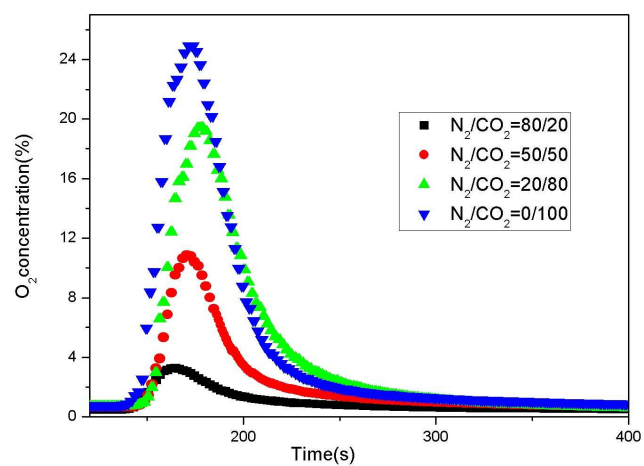
387

388

389 **Figure 5.**

390

391

392 **Figure 6.**

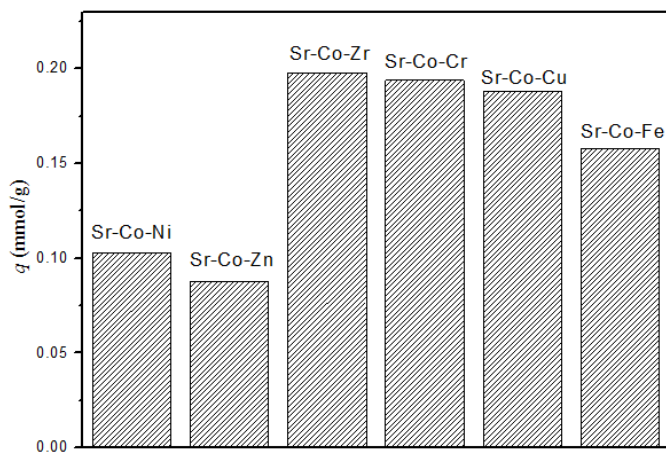
393

394

395

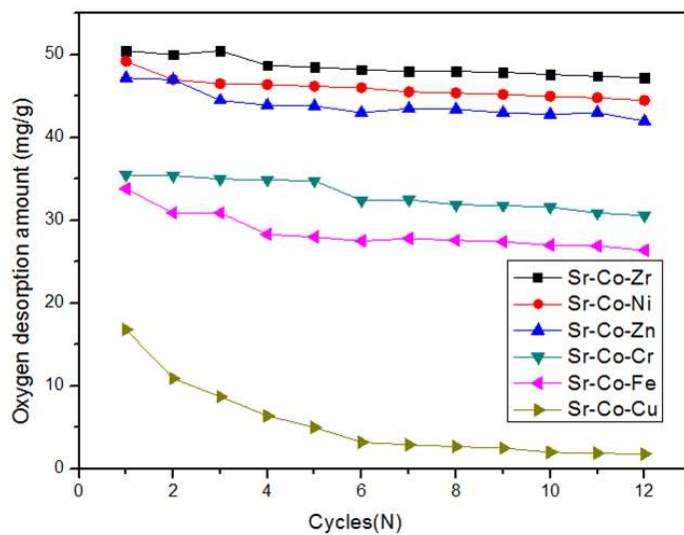
396

397

398 **Figure 7.**

399

400

401 **Figure 8.**

402