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Nanomaterials with few nanometers (1-100 nm) in dimension possess exceptional properties making them suitable for a wide range of applications in various fields such as chemistry, physics, electronics, optics and biomedical sciences. Modern approaches in nanotechnology allow us to design and control the size, shape, structure and assembly of the nanomaterials to suite the required applications. Nanomaterials with controlled and tunable properties are commonly synthesized using techniques such as: chemical vapor deposition method (CVDM), co-precipitation, solid-state thermal decomposition, colloidal synthesis etc., however almost all of these methods involve synthesis in controlled environments and further post-synthesis treatments (e.g. separation, cleaning, size classification of nanoparticles etc.) resulting in cumbersome and time consuming process. Another synthesis method, known as combustion synthesis (CS) has gained considerable research attention in a short period of time due to fast, economic and simple synthesis procedure. Traditionally CS was applied in solid-solid reactants systems by mixing the reactant powders and pressing them to form a pellet, which is ignited to start the exothermic combustion reaction. After the reactive mixture is ignited, the combustion reaction is reported to proceed in two modes; self-propagating high temperature synthesis (SHS) and volume combustion synthesis (VCS). In SHS, the reactive medium is ignited locally to initiate the combustion reaction which propagates through the unreacted sample in a controlled manner. In case of VCS, the entire medium is heated uniformly and the reaction occurs simultaneously throughout the volume. Another variation to CS was introduced later, known as Solution Combustion Synthesis (SCS) where the synthesis precursors are dissolved in water to get a homogeneous solution.

The solution is heated uniformly over a hot plate heater to start combustion reaction leading to nanomaterials synthesis with uniform properties. Single step synthesis of pure metals and their oxides can be achieved using SCS technique which is simple and versatile to synthesize nano-compounds with high surface area and porosity in a cost effective manner for catalytic applications. Typically, it involves a redox reaction in a homogeneous mixture of metal precursor (oxidizing agent e.g. metal nitrate) and fuel (reducing agent e.g. glycine) resulting in reduced reaction between metal nitrates as oxidizer and glycine as a fuel can be described by the following equation:

$$M^n(NO_3)_\nu + \left(\frac{5}{9}v\varphi\right)NH_2CH_2COOH + v\frac{5}{4}(\varphi - 1)O_2$$

$$\downarrow$$

$$M^nO_{\nu}s + \left(\frac{10}{9}v\varphi\right)CO_2(g) + \frac{25}{18}vH_2O(g) + v\left(\frac{5\varphi + 9}{18}\right)N_2(g)$$

(1)

Where $M^n$ is a $n$-valent metal. The parameter $\varphi$, known as fuel to oxidizer ratio, is defined such that $\varphi = 1$ corresponds to a
stoichiometric oxygen concentration, meaning that the initial mixture does not require atmospheric oxygen for complete oxidation of the fuel, while \( \phi > 1 \) \((-1\) implies fuel-rich (or lean) conditions.

Previous works \(^{14,15}\) on SCS indicate the energy required for sustaining the combustion synthesis comes from the reaction between \( \text{NH}_3 \) obtained during the decomposition of glycine and \( H\text{NO}_3 \) released from the decomposition of metal nitrate. As indicated by scheme (1) above and also from the literature survey, SCS is commonly used for synthesizing metal oxides, however recent publication \(^{12,21}\) have shown a higher value of \( \phi \) (large amount of fuel) could be used synthesize metallic phases. Thermodynamic studies indicate that a higher value of glycine helps in creating a reducing environment by releasing hydrogen as one of the byproducts \(^{14,15}\). Overall, it is expected that a suitable condition for the synthesis of pure metal nanoparticles would involve a high value of \( \phi \), so that the synthesized metal oxide could be reduced in presence of excess hydrogen. The combustion temperature plays an important role in determining the synthesized nanoparticles properties such as crystallite size, surface area etc., and a lower combustion temperature could help in synthesizing smaller nanoparticles with high surface area suitable for catalytic applications. There are various ways in which combustion temperature can be decreased; such as by decreasing the energy density of the system by changing the type of fuel and amount of fuel, or by mixing an inert material. Recently a novel CS method was introduced by restricting the active solution in a thin layer, known as Impregnated Layer Combustion Synthesis (ILCS). \(^{12,21}\) The thin layer could be active (e.g. cellulose paper) or inert (e.g. high surface area metal oxides) depending on the requirements. The combustion temperature in ILCS method is relatively lower due reduced energy density of the mixture, and also a thin media helps in cooling the synthesized product faster by quenching any crystal growth after synthesis. \(^{12,21,22}\) Also, the ILCS method provides a suitable platform for large scale synthesis of nanomaterials as the entire process can be automatized for synthesis in continuous mode rather than batch-wise synthesis. \(^{23}\) Recent theoretical studies on ILCS method indicate a better control on combustion and tunable product properties can be obtained by choosing the impregnation media with suitable thermo-physical properties. \(^{22}\)

Although the ILCS method has numerous advantages, however, so far a detailed study relating the interaction of reactive solution with the impregnation media is lacking. The previous publications \(^{14,15}\) describe the effect of amount of fuel on SCS, but no such study is available to indicate the effect of fuel content on product properties when the reaction media is impregnated in a cellulose paper, which is the main focus of this paper. A mixture containing Cu and Ni is used for this study to synthesize nanoparticles using SCS and ILCS method with various fuel (glycine) to oxidizer (metal nitrate) ratio. The choice of Cu and Ni is based on their importance as catalysts for hydrocarbon reforming reactions. \(^{24}\) At later stage, selected nanoparticles will be used as catalysts for ethanol dehydrogenation reaction. The synthesized products are characterized and compared using BET, XRD SEM and TEM. A mechanistic study is also conducted using TGA/DTA to understand the development of phases in ILCS process.

**Experimental**

The nanopowders of Cu-Ni were synthesized by SCS and ILCS methods. Firstly, a homogeneous aqueous solution of Nickel nitrate hexahydrate, \( \text{Ni(NO}_3\text{)}_2 \cdot 6\text{H}_2\text{O} \), Copper nitrate trihydrate, \( \text{Cu(NO}_3\text{)}_2 \cdot 3\text{H}_2\text{O} \) and glycine is prepared in desired molar ratio. The amount of precursors was calculated based on the production of 3 g of pure metal/metal-oxide in the product using stoichiometric equation (1). All the reagents were dissolved in 75 ml deionized water and kept at room temperature for 1 hr while continuously stirred to get a homogenous mixture with uniform clarity. In case of SCS method, the prepared solutions were placed on a hot plate heater (Barnstead Thermolyne, model no: sp 46925), to heat the entire solution, leading to evaporation of water after sometime. Thereafter the temperature starts to increase and once it reaches a critical temperature, known as ignition temperature, the combustion reaction is initiated inside the beaker. This reaction can proceed in two modes VCS or SHS similar to solid-solid combustion synthesis, depending on the fuel to oxidizer ration. The chart in supplementary Fig. S1 describes the synthesis of Cu-Ni alloy using SCS method. In case of ILCS method, the homogeneous solution is impregnated in a cellulose paper and dried for 24 hr at room temperature in open air to remove excess solution and moisture. A fixed amount of cellulose paper is completely dipped inside the solution to absorb the reaction mixture until it gets saturated, which is removed thereafter and hanged in air for drying. The steps involved in ILCS process are presented in supplementary Fig. S1b. A detailed description can be found in earlier publications. \(^{12,22}\) The dried cellulose is locally ignited by a hot wire to initiate the combustion in smoldering mode without any flame, which is controllable and gives low temperature. The combustion can also take place in an uncontrolled manner leading to flames and high temperature, which is not desirable. The transition from smoldering to flame mode depends on exothermicity of the reaction mixture, thickness of the active solution loaded and availability of oxygen for combustion. \(^{25,26}\) The temperature-time profile during the synthesis process has been recorded using a high speed data logger (Daq 3005 personal board, Omega); to get the ignition temperature along with the combustion temperature. The study of combustion synthesis mechanism was conducted using differential thermal and thermo-gravimetric analysis (DTA-TGA, Pyris 6, Perkin Elmer), of the sample under continuous flow of nitrogen at 1 atm pressure and 20 ccm at different heating rates of 5, 10, 15 and 20 K/min within the temperature range of 300 K – 900 K. The BET surface area measurements of the synthesized nanoparticles were carried out on Micromertics ASAP 2420 using nitrogen as an absorbent gas. XRD characterizations were done using Rigaku MiniFlexII Desktop X-ray powder diffractometer with a wavelength of 0.154056 nm by Cu-Ku radiation and a scan range of 10 – 90 degrees. Crystallite sizes were determined by using the Scherrer’s equation. \(^{27}\)

The surface morphology were analyzed with Scanning Electron Microscope (Nova Nano 450, FEI), with magnification up to 200 kx. The TEM analysis of nanopowders were performed using a high resolution transmission electron microscopy (HRTEM, TECNAI G2 F20, FEI). The TEM samples were prepared by dispersing the nanoparticles in a water/ethanol solution and sonicating them for a
fixed time period. A drop from this solution containing dispersed nanoparticles was loaded on TEM grid and dried before placing them on the TEM holder for analysis.

Results and discussions

A typical temperature-time profile of the SCS in copper nitrate-nickel nitrate-glycine system is shown in Fig. 1. After evaporation of water from the reactive solution, it is heated to a temperature, $T_i$, known as ignition temperature, where the combustion reaction initiates. Thereafter the temperature rises quickly to maximum temperature (known as combustion temperature) before cooling down. In some cases, depending on the value of the parameter $\phi$, the combustion reaction may start at one point inside the beaker then propagate to cover the entire reaction media as in SHS mode. After combustion the resultant metal oxide may be converted into metal, depending on the fuel content in the solution. In Fig. 1, the temperature profile of Cu-Ni synthesis with fuel to metal nitrate ratio of approximately 1.75 (fuel rich) is presented. In this case more than one peak is observed, correlating with the combustion reaction and subsequent reduction of metal oxide to metallic phase. Depending on the oxidation state of metals and the fuel amount more than two peaks are also possible.

Figure 1: Temperature-time profile in Copper nitrate hexahydrate-Nickel nitrate hexahydrate-Glycine using SCS mode

Our current work mainly emphasizes the decomposition mechanism of the precursors in both methods of synthesis, SCS and ILCS, leading to final solid products. The decomposition of individual precursors; cellulose, glycine, copper nitrate and nickel nitrate are summarized as follows:

Decomposition of Reactants

Fig. 2 shows the TGA analysis of Cellulose and glycine under nitrogen atmosphere and temperature range of 300 K – 1000 K. In case of cellulose, decomposition proceeds in three stages as shown in Fig 2(a):

(I) A small weight loss of (10%) in the initial stage (310 K - 325 K), possibly due to the removal of absorbed water from the cellulose.

(II) The next degradation happens at 565 K attributed to the formation of alkenes and other hydrocarbons by the glycosidic cleavage in cellulose with a weight loss from 85% to 24% of the initial value.

(III) The residual decomposition at higher temperature, commonly at 673 K could be due to the release of CO$_2$, H$_2$O, CO and char as mentioned in previous work.

Similarly in Fig. 2(b) the decomposition of glycine can be explained in different steps. The first weight loss is seen in the temperature range 510 K - 585 K possibly releasing the products NH$_3$, H$_2$O, CO$_2$ and reaching a weight loss of 40% of initial weight. Besides that, at higher temperature of 583 K - 730 K, the decrease in weight could be associated with the formation of HNCO and CO$_3$.

Furthermore, the degradation of each metal precursors were identified by using TGA analysis of copper and nickel nitrates as in Fig. 3. The disintegration of the copper nitrate Cu(NO$_3$)$_2$·3H$_2$O,
shown in details in Fig. 3(a), can be explained in three exothermic stages as below.

(I) The crystalline water molecules are released in the initial stage of degradation in the temperature below 425 K making a weight loss of 18%.

(II) The second stage of decomposition start at 425 K till 488 K showing a major weight loss of 55%, as a consequence of the formation of HNO₃ and Cu₂(OH)₃·NO₃ in this temperature range.¹⁴

(III) Finally, the weight loss in the temperature of 488 K – 700 K initiates the production of CuO by further decomposition of Cu₂(OH)₃·NO₃ and giving a weight loss down to ~18%. After that residual mass doesn’t follow any further decomposition between 700 K – 1000 K.

Figure 3: TGA/DTA analysis of (a) Copper nitrate (b) Nickel nitrate in temperature range 300-1000K at a heating rate of 5 K/min.

TGA curve of the nickel nitrate, Ni(NO₃)₂·6H₂O in the Fig. 3(b) displays the presence of various weight losses starting in the temperature range 310 K – 350 K which could be by virtue of the water loss in the initial stage. Later, it follows the weight loss of 55% of its initial value in the temperature 525 K resulted in the release of HNO₃.¹² The final stage is the decomposition of the remaining residuals into NiO occurring in 523 K – 590 K. After this stage, the residual mass of 25% remains constant from 590 K to 1000 K without any further decomposition.

On account of above two sections and available literatures¹⁵, it is anticipated that the HNO₃ decomposed from the metal nitrate and NH₃ from glycine in the same temperature range would react exothermically and provide the energy released during the combustion synthesis.¹⁵

TGA/DTA study of ILCS method

Copper nitrate + Cellulose + Glycine system (φ=1)

The TGA/DTA profile for (Cu(NO₃)₂·3H₂O)-cellulose-glycine system obtained at 5 K/min heating rate in continuous flow of N₂ (20 cc/min) is shown in Fig. 4. The thermogram can be divided into three regions. In region I, the weight loss between temperature 300 K – 400 K could be related to water removal from the sample (from copper nitrate hydrate and extra moisture due to cellulose impregnation) as indicated in the 1st peak of DTA curve. The second region between 400 K – 550 K comprises of two weight loss (peak 2nd and 3rd) which could be associated with the combustion reaction (at 425 K) to form CuO and subsequent partial reduction (495 K) of CuO to Cu₂O or Cu metal. As reported elsewhere¹⁴ there is a possibility of multiple peaks during combustion reaction depending of the fuel to oxidizer ratio (φ). In case of copper system, (at φ = 1) a partial reduction of CuO to Cu₂O or Cu is possible as indicated by the peak 3 in Fig. 4.

Figure 4: TGA/DTA result of Copper nitrate-Cellulose-Glycine system in temperature range 300-850 K at a heating rate of 5 K/min.

The study of ILCS method in previous works¹² show that ILCS combustion front is comprised of two fronts; (1) low temperature front due to ignition in the impregnated reactive solution, and (2) high temperature front due to combustion of cellulosic material. The solid products obtained after first front are amorphous and do not show any crystallinity, while the products after second front were found to be crystalline showing well defined XRD peaks¹², indicating the combustion of cellulose to be the source of energy for calcination. Following a similar analysis, the DTA peak in the higher
temperature zone (region III) results from the combustion of cellulose, playing an important role of conversion of the amorphous products to crystalline form without requiring any external calcination process. Hence, it is clear that cellulose paper has two important functions; firstly being a thin medium enhances the cooling of synthesized products, and secondly it also acts as a source of energy for synthesizing crystalline nanomaterials. The role of cellulose in ILCS and sequence of events taking place for the formation of nanoparticles can be summarized as follows:

(I) The homogeneous mixture of precursors (metal nitrates and glycine) in the initial stage form a good binding with the cellulose paper. This bonding could be associated with glycosidic linkages of cellulose as reported elsewhere by Wei Zhou et al.33, to form a thin layer of reactive media.

(II) Combustion in the reactive thin layer results in a layer of the amorphous metal/oxide over the cellulose fibres

(III) The combustion of cellulose converts the amorphous metals/oxides into crystalline form without requiring external calcination.

Nickel nitrate + Glycine system + Cellulose (φ=1)

The TGA/DTA analysis of the dried nickel nitrate-glycine impregnated in cellulose is shown in the Fig. 5 indicating three regions with sharp weight losses which can be explained as following:

Region I: weight loss reaches to 20% of its initial value, with the temperature of 320 K – 385 K as cause of the dehydration of water from the dried sample.

Region II: In case of nickel on the contrary to copper, only single weight loss at 450 K is observed. The φ value used in this case is unity, which is used to synthesize only NiO resulting in only one peak due to combustion reaction. A higher value of φ could reduce NiO to Ni showing multiple peaks. In this stage the residue mass came down to 55% of its initial value due to degradation of the nickel nitrate to NiO.

Region III: Appearance of weight loss in this region at 550 K results in 95% loss of its initial mass. This could be associated with burning of cellulose similar to the case of copper nitrate system.

Copper nitrate + Nickel nitrate + Glycine + Cellulose System (φ=1)

The TGA/DTA of the mixture of copper nitrate-nickel nitrate-glycine sloution impregnated in cellulose displays three regions with four DTA peaks as shown in Fig. 6.

Region I: A weight loss of 20% at 325 K is due to the devolatilization of water molecules linked with the cellulose.

Region 2: The sample is a mixture of the nitrates of copper and nickel, and the combustion peak in region II (425 K – 550 K, peak 2 and peak 3) could be due to combustion reaction between metal nitrate and glycine and subsequent partial reduction of metal oxides to metal with possibilities of giving Cu$_2$O, Cu, Ni or Cu-Ni alloy along with their oxides (CuO, NiO). The weight loss in this region reaches to 50 % of initial value.

Region III: The smooth weight loss of 25% from 50% in range of 560 K – 634 K ensures combustion of cellulose and formation of crystalline nanoparticles.
where $x$ is the fraction of precursors degraded, $\frac{dx}{dt}$ is the heating rate, $R$ is the gas constant, $T$ is the Kelvin temperature, $A$ and $E_a$ are material dependent properties in which $A$ is the probability of the presence of molecule with energy $E_a$ in the reaction, known as frequency factor, while $E_a$ is the activation energy. Following the analysis shown by Kissinger$^{34}$, a logarithmic equation can be obtained to calculate activation energy from experimental data using TGA/DTA system

$$-\ln \left( \frac{\beta}{T_m^2} \right) = -\ln \left( \frac{R}{A} \right) + \frac{E_a}{R} \left( \frac{1}{T_m} \right)$$  \hspace{1cm} (3)

The activation energy was calculated by plotting $-\ln \left( \frac{\beta}{T_m^2} \right)$ versus $1/T_m$ and calculating the slope of the straight line obtained.

The reactive medium (Cu-ILCS, Ni-ILCS, and Cu-Ni-ILCS) with stoichiometric fuel ratio were analyzed at different heating rates of 5, 8 and 10 K/min. From the above equations, the activation energy of Cu-ILCS was found to be 100.93 KJ/mol. Subsequently the Ni-ILCS and Cu-Ni-ILCS showed activation energies of 132.38 KJ/mol and 116.125 KJ/mol. The activation energies thus calculated shows a good agreement with a chemical reaction in a kinetic controlled stage as it is found to be in the range of 98 KJ/mol - 133 KJ/mol$^{35}$. The surface area of combustion synthesized (both SCS and ILCS) nanoparticles of copper-nickel has been examined using BET for different values of $\phi$. In SCS mode, the surface area shows a decreasing trend with increasing value of $\phi$. The surface area for $\phi=0.5$ was found to be 7.43 m$^2$/g, and decreases to 4.39 m$^2$/g and 4.08 m$^2$/g for $\phi=1$ and 1.75 respectively. Likewise, in ILCS the surface area decreases with $\phi$, giving a value of 18.78 m$^2$/g, 7.12 m$^2$/g and 3.88 m$^2$/g for $\phi$ values of 0.5, 1, 1.75 respectively. At lower value of $\phi$, the ILCS method gives better surface area (more than twice as compared to SCS) which gradually decreases and reaches approximately the same value of SCS eventually at $\phi=1.75$. Considering both cases of combustion, there is a considerable decrease in the Surface area with increasing fuel ratio. As discussed earlier, the combustion temperature increases with increase in $\phi$, reaches to the maximum point at stoichiometric value of $\phi$ and then decreases. The temperature is known to affect the crystallite size of nanomaterials synthesized (due to sintering) which could be the main cause for the decrease in surface area for lower $\phi$ values (e.g. $\phi=0.5$ and 1.0), however the decrease in surface area at higher $\phi$ value could not be associated with temperature as the expected combustion temperature would be lower as compared to $\phi=1$ value.

The release of gaseous products during combustion process, as presented in equation (1), would have an effect on total surface area and porosity which seems to play an important role here. BJH desorption analysis was used to get information regarding the pore volume and pore width to investigate further the surface area trend observed with $\phi$.

Fig. 7 shows the total pore volume and average pore size observed wrt $\phi$ during combustion synthesis. Both SCS and ILCS methods exhibit a similar trend in the variation of average pore volume and diameter towards the fuel amount. As the fuel amount increases, the average pore volume decreases possibly due to its surface transport mechanism. A similar effect in pore volume has been reported by E.J. Bosze$^{36}$ showing that the combustion temperature rises with the fuel amount and at higher fuel value the synthesized particles hold a very little amount of gases to create the porosity; in turns it accelerate the crystallite growth by escaping the inert gasses fastly. Conversely, with lower combustion temperature moderate growth kinetic allow the inert gas to be trapped and set in the final product of larger porosity. Considering the study of pore diameter with fuel ratio, it follows a volcano shaped curve indicating the development of wide pores in the stoichiometric ratio. When compared with the SCS, ILCS method exhibit large pore volume and diameter, could be commonly due to the evolution of more amount of gases in latter approach of synthesis as it is accompanied by burning of cellulose along with the combustion in the reaction mixture.

![Figure 7: Evaluating the relation of average pore volume and average pore diameter w.r.t. variation of fuel ratio. (— Average pore volume in SCS, ----- Average pore volume in ILCS, .......... Average pore diameter in SCS and ~~~~~~ Average pore diameter in ILCS)](image)
observed in XRD pattern. While increasing the value of fuel, a hydrogen rich environment can be generated to reduce the oxide products into their metallic form.\(^\text{12,14}\) At \(\varphi\) value of 1.75 the complete oxides were converted to pure copper nickel alloy. The XRD patterns of the nanoparticles in ILCS are shown in the Fig. 9b. In spite of changing the value of \(\varphi (0.5 \leq \varphi \leq 1.75)\) in the ILCS mode of synthesis, only metal oxides (CuO, NiO) were observed and no metallic form was found. This could be due to the presence of cellulose which may consume the extra fuel added at higher \(\varphi\) values preventing the creation of a reducing environment as observed in case of SCS mode. Perhaps, a higher value of \(\varphi\) is required in this case for the reduction of the metal oxides during combustion.

![Figure 8: BJH desorption pore volume distribution against pore width for different value of \(\varphi\) in both synthesis technique a) SCS b) ILCS.](image)

The Crystalline sizes of the particles were calculated using Scherrer’s equation\(^\text{27}\) with increasing fuel ratio for the two mode of synthesis. During the increment of \(\varphi\) in SCS, the crystalline size (associated with peak at 20 =~44°) increases from 8.55 nm for fuel lean (\(\varphi=0.5\)) to 10.05 nm for stoichiometric value (\(\varphi=1\)) and 13.1 nm for fuel rich (\(\varphi=1.75\)) condition. This result is consistent with the BET surface area observed and pore volume pattern in Fig. 8 with \(\varphi\). The larger crystallites at higher value of \(\varphi\) would result in lower surface area. However the effect of \(\varphi\) on crystallite size in ILCS was not so pronounced and a constant crystallite size of approximately 13.55 nm was found for all value of \(\varphi\). Perhaps this behavior could be due to dominant cellulose material which reduces the total energy density of the reaction system (consisting of metal nitrate and glycine). So, it is clear that high surface area in ILCS products is due to large porosity rather than the crystallite size. A large crystallite size in case of ILCS is possibly due high temperature of combustion resulting from cellulose burning. As the major energy for calcination comes from cellulose combustion, the \(\varphi\) value will not affect the crystallite size much, which is consistent with the XRD patterns observed (only oxides). It is anticipated that a considerably large amount of fuel (\(\varphi\geq 1\)) would be required to have any effect on the crystallite size and the nature of products observed in case of ILCS method. The nanoparticles microstructure was studied using high magnification SEM analysis. Typical SEM microstructures of Cu-Ni nanoparticles synthesized in SCS under different values of \(\varphi\) are shown as in Fig. 10a-f.

![Figure 9: XRD pattern of Ni(NO\(_3\))\(_2\)·6H\(_2\)O/Cu(NO\(_3\))\(_2\)·3H\(_2\)O and glycine system for different fuel ratios in a) SCS b) ILCS; ● CuNi, ■ CuO, ♦ NiO](image)

As it can be seen, for all the values of \(\varphi\) the products are highly porous in nature consisting of agglomerated nanoparticles. The agglomeration of nanoparticles is very common in combustion synthesized nanopowders. In most cases these particles appear like thin flakes or sheets (Fig. 10a-c, low magnification). As the \(\varphi\) value is increased, some changes in the microstructure is observed.
showing a gradual increase in the size of agglomerated particles (Fig. 10d-f, high magnification). The SEM images for ILCS synthesized nanoparticles are shown in Fig. 11a-f. In this case, the nanoparticles appear to be agglomerated generating a shape similar to the cellulosic microfibers (Fig. 11a-c, low magnification). There is no significant change in the fibrous structure of synthesized nanoparticles observed by changing the φ value. High magnification SEM images (Fig. 11d-f) show that these fibers are composed of highly agglomerated nanoparticles. No major change in nanoparticle size is observed by changing the φ and adding more fuel in ILCS process. These results are in agreement with the crystallite size result observed using XRD analysis. Fig. 12 shows TEM images of the synthesized CuENi nanoparticles by SCS (a) and ILCS (b) method using φ=1.75. In both, SCS and ILCS, TEM images show the presence of smaller nanoparticles agglomerated together to form bigger particles. The porous structure in these nanopowders is also apparent through the contrast images displaying slightly more porosity in case of ILCS method, consistent with the BET results discussed earlier.

Figure 10: Typical SEM image of CuNi alloy synthesized in SCS method under (a-c) lower magnification, scale bar = 10µm (d-f) higher magnification with φ=0.5, 1, 1.75 respectively, scale bar = 1 µm

Figure 11: Typical SEM image of CuNi alloy synthesized in ILCS method under (a-c) lower magnification, scale bar = 10 µm (d-f) higher magnification with φ=0.5, 1, 1.75 respectively, scale bar = 4 µm

The particle size distribution is difficult to count due to agglomeration, however, just from visual inspection, particles do not appear to be of uniform size and particles from 10 nm – 100 nm size are clearly visible, with slightly better uniformity in case of ILCS method. The SCS samples appear to have smaller particle as expected from the XRD results. Agglomeration of nanoparticles in combustion synthesis is very common and reported in many systems studied23,37. Few research groups have been working modifying the synthesis technique by adding leachable soluble salts during synthesis38,39 where the synthesized nanoparticles will be encapsulated with deposited salt, which can be leached out after synthesis, therefore reducing the agglomeration to a considerable extent. Future study in our group will explore the catalytic properties of the synthesized materials using a probe reaction to correlate the effect of synthesis parameters with observed catalytic behavior.

Figure 12: TEM image of as-synthesized nanoparticles of CuNi at φ=1.75 using (a) SCS and (a) ILCS techniques.

Conclusions
Nanopowders of Cu-Ni were synthesized using two modes of combustion synthesis (SCS and ILCS or cellulose assisted combustion synthesis) and characterized by various techniques
products. The higher surface area was found to be due to large crystallite size. The ILCS process was found to be very sensitive to the amount of fuel used. Within the investigated range of \( \phi \) values, only metal oxides were observed in ILCS method. The crystallite size and microstructure were also remained unaffected although a pattern was visible in case of total surface area and porosity. The ILCS products displayed higher surface area and larger pore volume as compared to SCS products. The higher surface area was found to be due to large porosity rather than crystallite size of the nanoparticles as ILCS products were found to have larger crystallite size. The TGA/DTA analyses of the precursors impregnated in cellulose were used to identify the sequence of events taking place during ILCS process. The synthesis of nanoparticles (metals or oxides) in ILCS method is initiated by an exothermic reaction between metal nitrate and fuel, and followed by combustion of cellulose at higher temperature (approximately 600 K), which is anticipated to be the main energy source for generating crystalline products. Both steps, the exothermic synthesis reaction and the cellulose combustion, release considerable amount of gases which form channels while escaping and help in synthesizing porous products.

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