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# Sheet-like structure FeF<sub>3</sub>/graphene composite as novel cathode material for Na ion batteries

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**Abstract:** A sheet-like structure FeF<sub>3</sub>/graphene composite is successfully synthesized by a novel and facile sol-gel method. The structure and electrochemical performance of the as-synthesized FeF<sub>3</sub>/graphene composite are investigated by X-ray diffraction (XRD), scanning electron microscope (SEM), transmission electron microscopy (TEM), high-resolution TEM (HRTEM) and electrochemical measurement. The results indicate that the FeF<sub>3</sub> nanosheets are loaded on the surface of the graphene sheets to form the sheet-like structure hybrid. Fourier transform infrared (FTIR) spectrum confirms that C-F bonds exist in FeF<sub>3</sub>/graphene composite, and it further indicates that a chemical bond between FeF<sub>3</sub> and graphene has been formed and FeF<sub>3</sub> can preferably stick to the surface of the graphene. The FeF<sub>3</sub>/graphene composite as cathode material of rechargeable Na ion batteries (NIB) exhibits a fairly high initial discharge capacity of 550 mAh g<sup>-1</sup> at 0.1 C, and it still keeps a capacity of 115 mAh g<sup>-1</sup> after 50 cycles at 0.3 C at a range of 1.0-4.0 V for NIB.

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22 **Keywords:** Na ion batteries; Iron fluorides; Graphene; Sheet-like structure; Sol-gel  
23 method

24

## 25 **1. Introduction**

26 For the past decade, Li ion batteries (LIB) have been extensively used in the  
27 consumer electronics, military, electric vehicle and aerospace fields due to their  
28 excellent energy density, good power capability, and long cycle life. However, along  
29 with the development of low cost and high efficient recycling technology, the low  
30 abundance of lithium in the earth's crust and large-scale applications of LIB become  
31 controversial. Alternatively to the widely studied LIB, the NIB can be a suitable  
32 choice for smart grid, energy conversion and storage applications in terms of the low  
33 cost, safety and natural abundance of sodium resources [1]. For the past decade, many  
34 efforts have been devoted to the development of novel cathode active materials.  
35 Transition metal oxide compounds and polyanion compounds with layered structure  
36 have been investigated as the cathode materials for NIB, such as  $\text{Na}_{0.44}\text{MnO}_2$  [2-3],  
37  $\text{Na}_4\text{Mn}_9\text{O}_{18}$  [4],  $\text{Na}_{0.71}\text{CoO}_2$  [5],  $\text{NaNi}_{1/3}\text{Mn}_{1/3}\text{Co}_{1/3}\text{O}_2$  [6],  $\text{Na}[\text{Ni}_{1/3}\text{Fe}_{1/3}\text{Mn}_{1/3}]\text{O}_2$  [7],  
38  $\text{Na}_{2/3}\text{Ni}_{1/3}\text{Mn}_{2/3-x}\text{Ti}_x\text{O}_2$  [8],  $\text{Na}_2\text{FeP}_2\text{O}_7$  [9],  $\text{Na}_2\text{CoP}_2\text{O}_7$  [10],  $\text{Na}[\text{Fe},\text{Mn}]\text{PO}_4$  [11],  
39  $\text{Na}_3\text{V}_2(\text{PO}_4)_3$  [12],  $\text{Na}[\text{Mn}_{1-x}\text{M}_x]\text{PO}_4$  (M = Fe, Ca, Mg) [13]. Recently, transition metal  
40 fluorides have been considered as a promising new class of cathode materials for LIB,  
41 which exhibit large theoretical capacities and high discharge voltages due to their  
42 highly ionic metal-ligand bonds and small atomic weight. Unlike conventional  
43 intercalation reaction, transition metal fluorides, based on reversible conversion

44 reaction, enable the full utilization of redox during the charge/discharge process and  
45 thus possess high theoretical specific capacity. Especially, due to high theoretical  
46 capacity and theoretical energy density,  $\text{FeF}_3$  have been researched as new and  
47 promising cathode materials for LIB [14-17, 18-20]. Besides, iron based fluorides  
48 have attracted interest as a promising positive electrode for rechargeable Na batteries.  
49 The important fluoride materials are currently reported as Na-ion batteries cathodes,  
50 such as  $\text{FeF}_3$  [14-17] and  $\text{NaFeF}_3$  [21, 22]. Li and his coworkers reported  
51  $\text{FeF}_3 \cdot 0.33\text{H}_2\text{O}$  and  $\text{FeF}_3 \cdot 0.5\text{H}_2\text{O}$  wired by carbon nanotubes through  
52 ionic-liquid-based synthesis method for Na-storage, and which exhibited a  
53 considerable capacity and rate performances as cathode materials for NIB [15, 16].  
54 Because of the dense structure and high insulating character of pure  $\text{FeF}_3$ , the  
55 electrochemical performance deteriorates promptly as the cathode material of LIB, let  
56 alone NIB [15].

57 Usually,  $\text{FeF}_3$  used as cathode materials of LIB or NIB is its hydrated compounds.  
58 Besides, it has also been reported that lithium aluminate nanosheet and  $\alpha\text{-Fe}_2\text{O}_3$   
59 nanoplates well-dispersed on the graphene can enhance their ionic conductivity [23,  
60 24]. In order to overcome the intrinsic drawback of the  $\text{FeF}_3$ , a sheet-like structure  
61  $\text{FeF}_3$  loaded by graphene is first designed and synthesized via a novel and facile  
62 sol-gel route in this work. The  $\text{FeF}_3$ /graphene nanosheets are obtained by controlling  
63 the dry temperature and the amount of graphene in a bottom-up synthesis. Herein,  
64 graphene is both used as the conductive agent to further improve the electrical  
65 conductivity of  $\text{FeF}_3$  and a supporter for the stabilization of iron fluoride nanosheets.

66 The FeF<sub>3</sub> nanosheets and graphene stack each other to form a hierarchical electron/ion  
67 conducting network, arising from the bilateral interaction of iron fluoride nanosheets  
68 and graphene sheet. The morphology and electrochemical performances of the  
69 sheet-like FeF<sub>3</sub>/graphene composites are subsequently investigated as cathode  
70 material for NIB.

71

## 72 **2. Experimental**

73 Fe(NO<sub>3</sub>)<sub>3</sub>·9H<sub>2</sub>O (10 mmol, 4.04 g) and conduction type graphene 0.1 g (8% of  
74 total weight) (The Sixth Element Inc) were dissolved or suspended in 50 mL methanol,  
75 then 1.5 ml (30 mmol) of 40% HF acid was added with stirring to gain a solution. The  
76 obtained FeF<sub>3</sub>·3H<sub>2</sub>O sol was aged for 24 h before dried at 60 °C. The product xerogel  
77 was further dried at 180 °C for 8h in a vacuum drying oven and ground into fine  
78 powders.

79 X-ray powder diffraction was performed using Rigaku D/MAX-2500/PC  
80 equipped with Cu-Ka source (40 kV, 250 mA) to get the crystal structure. The sizes  
81 and morphologies of compound particles were characterized by a field emission  
82 scanning electron microscope using JEOL JSM-3500N. Transmission electron  
83 microscopy (TEM), selected area electron diffraction (SAED) and high-resolution  
84 TEM (HRTEM) images were taken on a JEOL-2100 microscope instrument at an  
85 acceleration voltage of 200 kV.

86 The electrochemical performance of the as-synthesized material was  
87 characterized on 2025 type coin cells as a cathode and a sodium disk as anode for

88 Na-ion batteries. The cathodes for Na cells were fabricated by mixing the cathode  
89 material, Super carbon (SP), and polyvinylidene fluoride (PVDF) binder with a  
90 weight ratio of 80:15:5 in N-methyl pyrrolidinone, which were then pasted on  
91 aluminum foil followed by drying under vacuum at 110 °C for 24 h. The average mass  
92 loading of materials on aluminum foil is about 1.83 mg/cm<sup>2</sup>. The test Na cells were  
93 assembled with the cathodes thus fabricated, metallic sodium anode, Glass fiber  
94 (GF/D) from Whatman was employed as the separator, and the electrolyte was 1 mol  
95 L<sup>-1</sup> NaClO<sub>4</sub> in a solvent of propylene carbonate (PC). The assembly of the testing  
96 cells was carried out in an argon-filled glove box, where water and oxygen  
97 concentration were kept less than 5 ppm. The charge-discharge experiments were  
98 conducted on a battery cycler (Newell Battery Test) at 25 °C. Charge-discharge  
99 measurements of fluoride-based cathodes versus Na/Na<sup>+</sup> were performed at room  
100 temperature under various rates of 0.1–5 C in a voltage range of 1.2–4.0 V and  
101 1.0–4.0 V for Na-storage. All of the specific capacities were calculated based on the  
102 mass of as-synthesized material (including graphene). Specifically, 0.1 C represents  
103 23.7 mA g<sup>-1</sup>, 0.3 C represents 71.1 mA g<sup>-1</sup>, 1 C represents 237 mA g<sup>-1</sup> and so on.

104 The galvanostatic intermittent titration technique (GITT) was performed at room  
105 temperature at a voltage range of 1.0–4.0 V. The GITT measurements were performed  
106 on the second cycle to determine the diffusion coefficient of Na ions ( $D_{\text{Na}}$ ) in  
107 electrode active materials. To achieve nearly equilibrium state (Es), relaxing 60 min  
108 after an interval of 10 min at a current density of 0.1 C have been combined with the  
109 GITT test. The procedure continued until the cutoff voltage was reached. As shown in

110 Fig. S4, the Na ion diffusion coefficient can also be determined by GITT by Fick's  
111 second laws of diffusion and calculated according to the following equation:

$$112 \quad D_{Na} = \frac{4}{\pi} \left( I_0 \frac{Vm}{FS} \right)^2 \left( \frac{dE/dx}{dE/dt^{1/2}} \right)^2$$

113 Where  $I_0$  (A) is the applied current in the charge/discharge process,  $V_m$  ( $\text{cm}^3$   
114  $\text{mol}^{-1}$ ) is the molar volume of active materials,  $F$  (96485 C/mol) is the Faraday  
115 constant,  $S$  ( $\text{cm}^2$ ) is the total contact area between the electrolyte and electrodes and  $L$   
116 (cm) is the thickness of the electrode.

117

### 118 3. Results and discussion

119 Fig. 1a illustrates  $\text{FeF}_3$  nanosheets anchored on the surface of graphene sheet to  
120 form the  $\text{FeF}_3/\text{graphene}$  composite with sheet-like structure. As schematically shown  
121 in Fig. 1a, firstly,  $\text{FeF}_3 \cdot 3\text{H}_2\text{O}$ , which is quickly formed after  $\text{Fe}(\text{NO}_3)_3 \cdot 9\text{H}_2\text{O}$  and HF  
122 added in methanol solution, generates C-F bond with graphene for nucleation sites to  
123 induce surface nucleation of iron fluoride during the aged process. The new produced  
124 C-F bonds are expected to act as nucleation sites for  $\text{FeF}_3 \cdot 3\text{H}_2\text{O}$  due to intermolecular  
125 interactions. Besides, controlling the graphene amount, the graphene sheets, which  
126 can anchor  $\text{FeF}_3 \cdot 3\text{H}_2\text{O}$  crystal, can provide more growing spots for  $\text{FeF}_3 \cdot 3\text{H}_2\text{O}$   
127 crystals and can also prevent the growth of big fluoride nanocrystals, thus  $\text{FeF}_3 \cdot 3\text{H}_2\text{O}$   
128 nanocrystals distribute homogeneously on the surface of the graphene sheet, a  
129  $\text{FeF}_3 \cdot 3\text{H}_2\text{O}/\text{graphene}$  composite precursor is obtained. Finally, the precursor was  
130 heated in a vacuum drying oven at the temperature of 180 °C to remove  $\text{H}_2\text{O}$   
131 molecules and the sheet-like  $\text{FeF}_3/\text{graphene}$  composite was obtained.

132 From the SEM images in Fig. 1b, the FeF<sub>3</sub>/graphene composite is actually the  
133 hierarchical sheet-like structure. A rough wavy structure could originate from the  
134 intrinsic wrinkles and ripples of graphene. The TEM images (Fig. 1(c, d)) confirm  
135 further that the FeF<sub>3</sub> nanosheets (red arrows) anchor on the surface of the graphene  
136 sheet (blue arrows) and form the hybrid with both sheet-like structure. The sizes of the  
137 nanosheets FeF<sub>3</sub>/graphene range from several hundred nanometers to a few  
138 micrometers. As being seen from the inset of Fig. 1c, the SAED pattern shows a set  
139 of broad diffuse rings instead of spots due to the random orientation of the crystallites,  
140 corresponding to diffraction from different planes of the nanocrystallites, which  
141 indicate the FeF<sub>3</sub> nanosheets are formed by tiny the FeF<sub>3</sub> nanocrystallites instead of  
142 growing along a certain direction. The SAED patterns are consistent with a hexagonal  
143 phase structure of FeF<sub>3</sub> with strong ring patterns due to (012) and (024) planes.  
144 Moreover, it can also be found from Fig. 1d that the FeF<sub>3</sub> nanosheets are formed by  
145 self-assembly of numerous nanoparticles with various sizes from 10 nm to 100 nm,  
146 which stretch outwards from the aggregate core, thus presenting the nanosheets  
147 morphology. HRTEM image (Fig. 1e) also reveals that FeF<sub>3</sub> tiny nanocrystals were  
148 formed. Lattice fringes can be discerned from the HRTEM image, suggesting that the  
149 FeF<sub>3</sub> nanoparticles are well-crystallized. Fig. 1(e) show that the interplanar spacing is  
150 about 0.381 nm and 0.265 nm, which also corresponds to the distance of (012) and  
151 (104) planes of FeF<sub>3</sub>, respectively, indicating that the nanoparticles are the iron based  
152 fluoride.



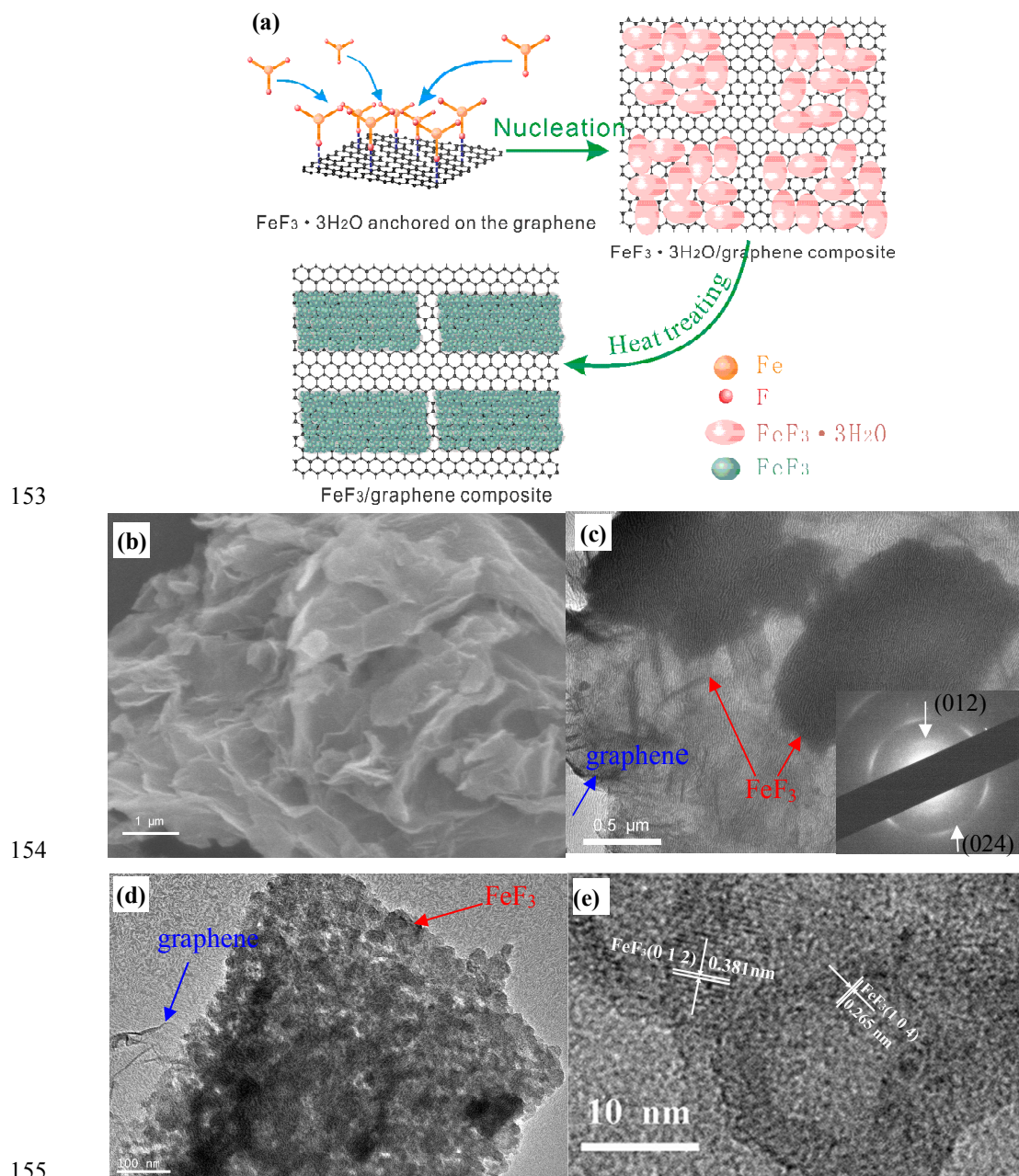
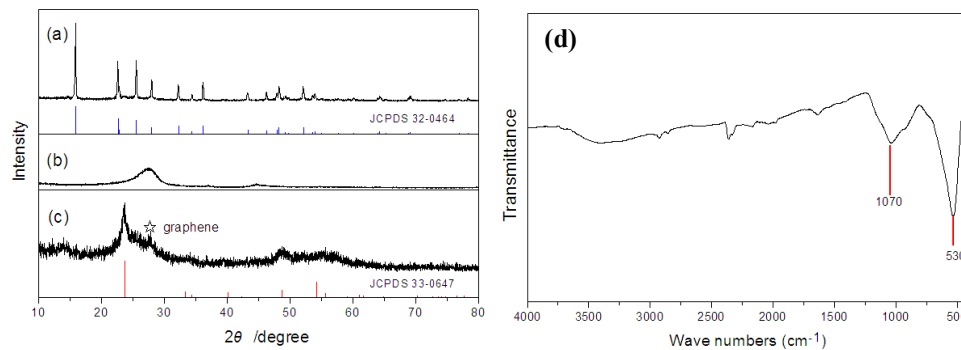


Fig. 1 (a) schematic illustration of fluoride nanosheet with graphene to form  $\text{FeF}_3/\text{graphene}$ , (b) SEM image, (c, d) TEM images, SAED image (the inset of (c)), and (e) HRTEM images of nanosheets  $\text{FeF}_3/\text{graphene}$ .

Fig. 2a, 2b and 2c represent the typical XRD patterns of the  $\text{FeF}_3 \cdot 3\text{H}_2\text{O}$  precursor, graphene, and the synthesized  $\text{FeF}_3/\text{graphene}$  by sol-gel method, respectively. According to the Scherer formula:  $D = R\lambda / (\beta \cos \theta)$ , when  $2\theta$  is  $15.940^\circ$ ,  $22.641^\circ$  and

162 25.580° in XRD pattern of FeF<sub>3</sub>·3H<sub>2</sub>O precursor, it can be calculated that the average  
163 particle sizes are 53.8 nm, 46 nm and 44.8 nm, respectively. The results indicate that  
164 the precursor consists of nanosized particles. The peak locations in Fig. 2c indicate  
165 the formation of FeF<sub>3</sub>/graphene composite. Moreover, a very small intensity reflection  
166 of graphene in Fig. 2c due to the presence of graphene can also be observed for the  
167 FeF<sub>3</sub>/graphene composite. The broad reflections indicate its low crystallisation degree,  
168 which is one common feature shared by many nanoscopic metal fluorides [25]. In  
169 order to analyze the interaction of FeF<sub>3</sub> nanosheets with graphene sheets, Fourier  
170 transform infrared (FTIR) spectrum was used. The samples prepared as KBr pellets  
171 were recorded on a Perkin-Elmer Spectrum One FTIR spectrophotometer in a scan  
172 range of 400-4000 cm<sup>-1</sup>, and the results are shown in Fig. 2d. The characteristic peak  
173 at approximately 530 cm<sup>-1</sup> is associated with the typical vibration band of Fe-F  
174 bond of FeF<sub>3</sub>. Meanwhile, the C-F stretching vibrations are generally found at around  
175 1070 cm<sup>-1</sup>[17], which confirms that FeF<sub>3</sub> nanosheets tightly anchor on the surface of  
176 graphene sheets (as Fig. 1a) due to interaction of F atom with C atom each other, then  
177 the FeF<sub>3</sub>/graphene composite with hybrid sheet-structure is formed.

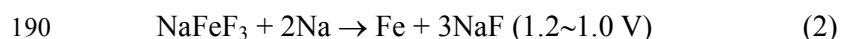
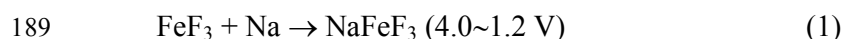


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179 Fig. 2 XRD patterns of (a) the FeF<sub>3</sub>·3H<sub>2</sub>O precursor, (b) graphene and (c) FeF<sub>3</sub>/graphene, (d) IR

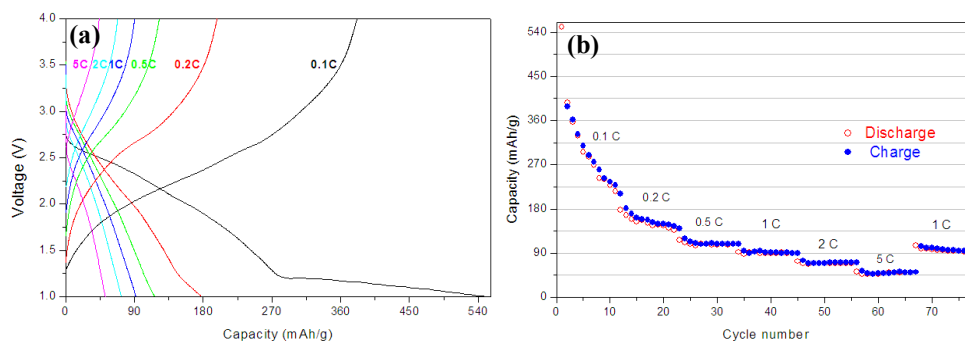
180 spectra of nanosheets FeF<sub>3</sub>/graphene.

181 Fig. 3a indicates the comparison of the rate capability for the FeF<sub>3</sub>/graphene  
182 composite at a range of 1.0-4.0 V. The FeF<sub>3</sub>/graphene cathode presents apparently a  
183 high irreversible discharge capacity of 550 mAh g<sup>-1</sup> at the first cycle at 0.1 C. The  
184 discharge capacity is close to the theoretical capacity of FeF<sub>3</sub>. It can also find from  
185 Fig. 3a that there is two reduction peaks at about 2.6 V and 1.2 V, which can  
186 correspond to the insertion/deinsertion reaction and the reversible conversion reaction,  
187 respectively. Usually, the electrode reaction mechanism of FeF<sub>3</sub> as the cathode  
188 material of Na ion battery can be expressed as following [26]:

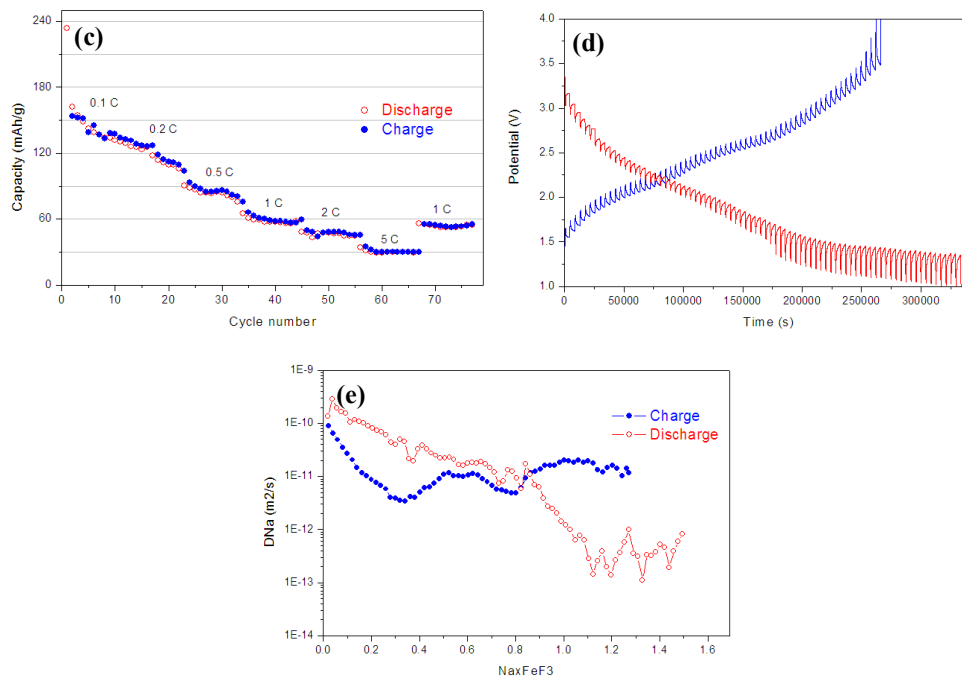


191 Equ.1 corresponds to the Na ion intercalation/unintercalation reactions between  
192 phases containing Fe<sup>3+</sup> and Fe<sup>2+</sup>. Equ.2 should be attributed to the redox reactions  
193 between phases containing Fe<sup>2+</sup> and metallic Fe<sup>0</sup> based on chemical conversion  
194 mechanism. It can be found from Fig. 3b that although the first discharge capacity is  
195 high, the second discharge capacity drops down to 396 mAh g<sup>-1</sup> and charge capacity  
196 dives to 388 mAh g<sup>-1</sup>. The reasons of the sudden descent of capacity are likely to  
197 originate from not only the formation of insulating phases during conversion reaction  
198 but also the formation of solid electrolyte interphase (SEI) layers. The presence of a  
199 relatively stable conversion reaction for the FeF<sub>3</sub>/graphene composite provides extra  
200 capacity for the first few cycles. However, the Fe<sup>0</sup> originated from Equ.2 will  
201 probably aggregate to form big particles during the charge/discharge cycle process,

202 thus it will result in the slow kinetics of the conversion reaction and the electrode  
203 process exhibits mainly the  $\text{Na}^+$  insertion reaction after the 50th cycle. Besides, GITT  
204 test results in Fig. 3d and 3e show that the Na ion ( $D_{\text{Na}}$ ) diffusion coefficient during  
205 insertion reaction is higher than during conversion reaction. Fig. 3c indicates the rate  
206 performance at different rates at the range of 1.5-4.0 V, in this potential window the  
207 electrode reaction only behaves the insertion/deinsertion reaction. To our surprise, the  
208 nanosheet  $\text{FeF}_3/\text{graphene}$  composite delivers  $234 \text{ mAh g}^{-1}$  in the insertion process at  
209 the first cycle at 0.1 C, which is nearly accordance with the theoretical capacity of 1  
210 Na insertion. This high capacity is maybe ascribed to the special sheet-like structure  
211 of the  $\text{FeF}_3/\text{graphene}$  composite. Especially, the  $\text{FeF}_3/\text{graphene}$  electrode can deliver  
212  $90 \text{ mAh g}^{-1}$  at 1 C rate at the range of 1.0-4.0 V (Fig. 3b), while it can only deliver  
213 about  $60 \text{ mAh g}^{-1}$  at the same rate at the range of 1.5-4.0 V (Fig. 3c). Compared Fig.  
214 3b with Fig.3c, it can be found that the capacity provide by the insertion reaction is  
215 about two-thirds of the total capacity. Therefore, the sheet-like structure of  
216  $\text{FeF}_3/\text{graphene}$  composite plays likely an important role for improving its capacity.



217



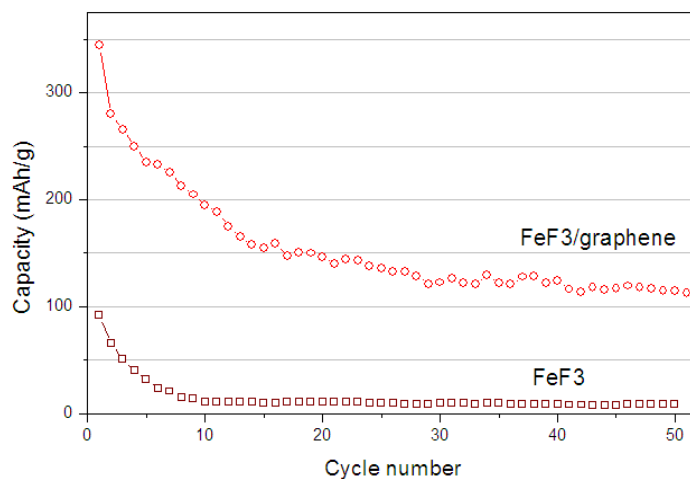
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219

220 Fig. 3 electrochemical behavior for  $FeF_3/graphene$ : (a) Prolonged cycling behavior at different  
 221 C rates, respectively at the range of 1.0-4.0 V, (b) rate performance at different rates at the range of  
 222 1.0-4.0 V, (c) rate performance at different rates at the range of 1.5-4.0 V, (d) the discharge/charge  
 223 GITT curves of  $FeF_3/graphene$  electrode as a function of time in the potential range of 1.0-4.0 V,  
 224 and (e) calculated  $D_{Na}$  for  $FeF_3/graphene$  as a function of  $x$  (the total Na insertion and conversion  
 225 number) during charge/discharge process, respectively.

226 Fig. 4 shows the cycling performances of  $FeF_3/graphene$  and  $FeF_3$  as the cathode  
 227 material of NIB at 0.3 C at a range of 1.0-4.0 V. The capacities of  $FeF_3$  and  
 228  $FeF_3/graphene$  composite during the first discharge are 91 and 344 mAh  $g^{-1}$ ,  
 229 respectively. The capacities of discharge decay faster in the first few cycles, but  
 230 capacities during the later cycles decline slowly, and the discharge capacity of the  
 231  $FeF_3/graphene$  composite can still keep 115.8 mAh  $g^{-1}$  after 50 cycles. However, for  
 232 the  $FeF_3$  electrode the discharge capacity is only 10 mAh  $g^{-1}$  after the 50th cycle.

233 Apparently, the addition of graphene and formation of sheet-like structure between  
234  $\text{FeF}_3$  and graphene can well improve the electrochemical performance.



235  
236 Fig. 4 the discharge curves of  $\text{FeF}_3$  cathode and  $\text{FeF}_3/\text{graphene}$  cathode at 0.3 C at the range of  
237 1.0-4.0 V for NIB.

238

#### 239 4. Conclusions

240 In conclusion, the sheet-like structure  $\text{FeF}_3/\text{graphene}$  composite for the  
241 application of NIB was successfully synthesized by a facile sol-gel route. The  
242 sheet-structural formation between  $\text{FeF}_3$  and graphene for  $\text{FeF}_3/\text{graphene}$  composite  
243 can contribute positively to the large reversible Na-storage capacity compared with  
244  $\text{FeF}_3$  during insertion/deinsertion process as well as conversion reaction. In the aspect  
245 of electrochemical behavior,  $\text{FeF}_3/\text{graphene}$  exhibits a fairly high initial discharge  
246 capacity of  $550 \text{ mAh g}^{-1}$  at 0.1 C, and it can still retain  $115.8 \text{ mAh g}^{-1}$  after 50 cycles  
247 at 0.3 C at a range of 1.0-4.0 V. Although the cycling performance and specific  
248 capacity of  $\text{FeF}_3/\text{graphene}$  need to be improved further, the design and preparation for  
249 the hybrid sheet-like structure  $\text{FeF}_3/\text{graphene}$  composite as the cathode material of

250 NIB is still a valuable exploration.

251

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